

Macromolecular Chemistry

MC 2.1.3

Makro – „Crashkurs“

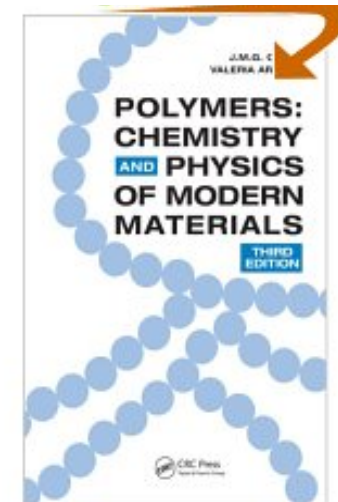
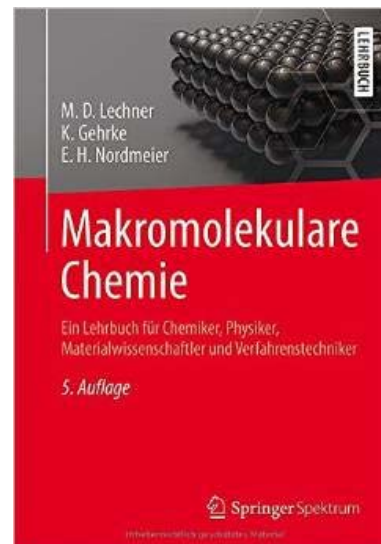
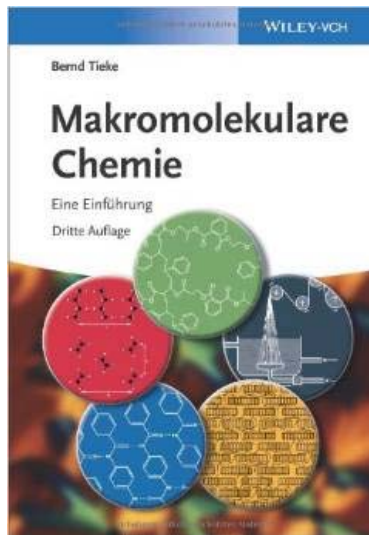
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Textbooks

- Tieke, Bernd: *Makromolekulare Chemie*, Wiley-VCH (2014, 49.90 €)
- Lechner, Gehrke, Nordmeier: *Makromolekulare Chemie*, Birkhaeuser (2014, 69,99 €)
- Elias: *Makromoleküle 1-5*, Wiley-VCH (2003, > 800 €)
- Cowie: *Polymers: Chemistry and Physics of Modern Materials* CRC-Press Inc. (2011, 65,95 €)



Why use synthetic polymers?

Advantages:

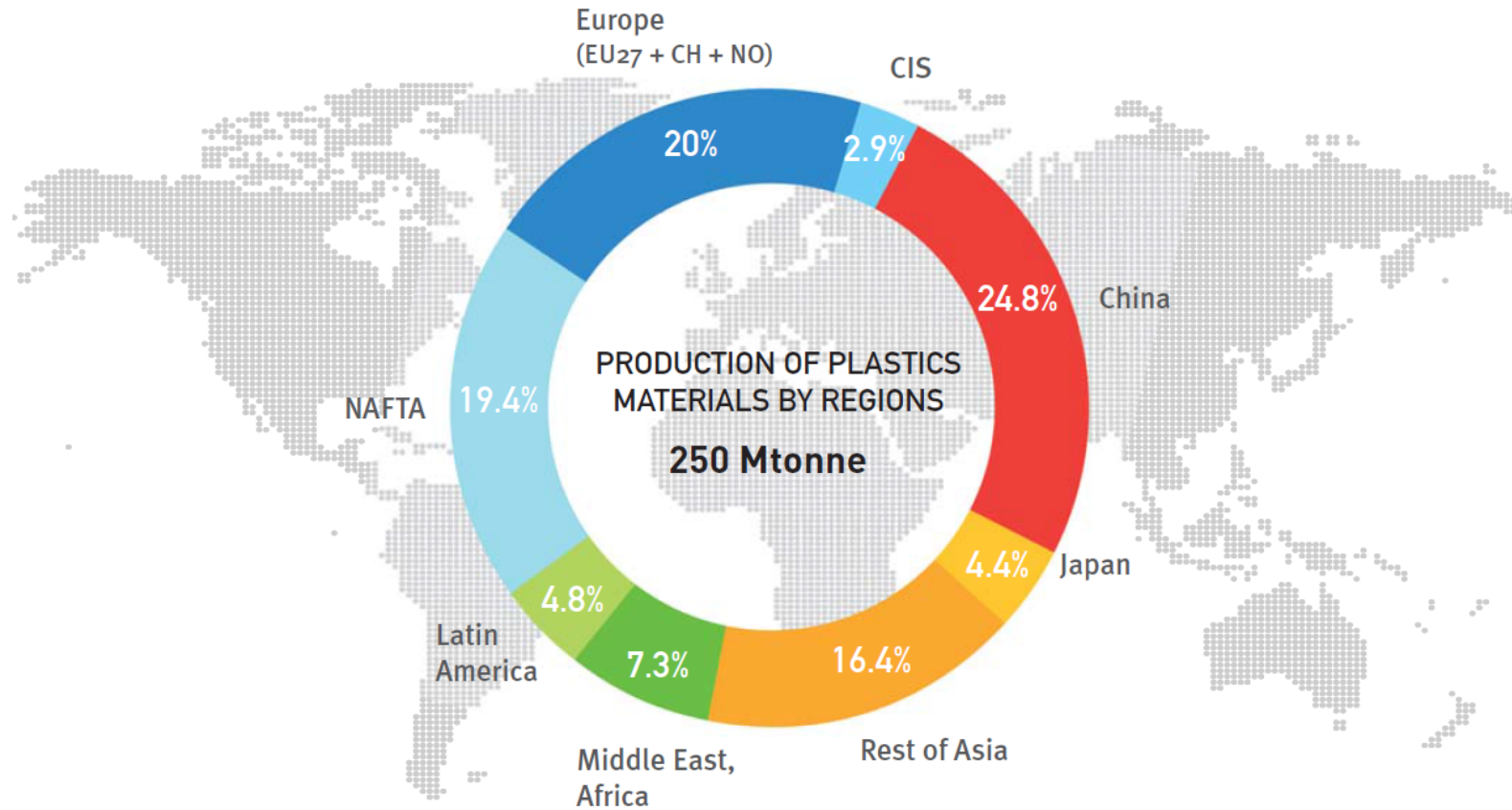
- Cheap
- Good quality
- Easy to process, fast to process, *i.e.* to convert into certain forms, such as bottles or fibers (lower temperatures than needed for metals)
- Light-weight (energy-saving in cars or planes)
- Sufficient petrochemical resources
(at present, more than 90% are used for energy production)
- Possibility to switch to natural resources (poly(lactic acid), soy beans, cellulose)
- Easy formulation compounding → a broad spectrum of different properties is available
- Recycling possibilities (thermal recycling, reuse)
- Tunable properties (choice of monomer, constitution, molar mass, ...)

Applications

- Plastics (*e.g.*, packaging materials)
- Coatings & adhesives (*e.g.* paints)
- Fibers (*e.g.*, clothes)
- Rubbers & elastomers (*e.g.* tires)
- Agriculture (*e.g.*, promoting plant growth)
- Medicine (*e.g.*, replacement of heart valves)
- Specialty polymers (high thermal stability)
- Plastic electronics (*e.g.*, PLED, solar cells)

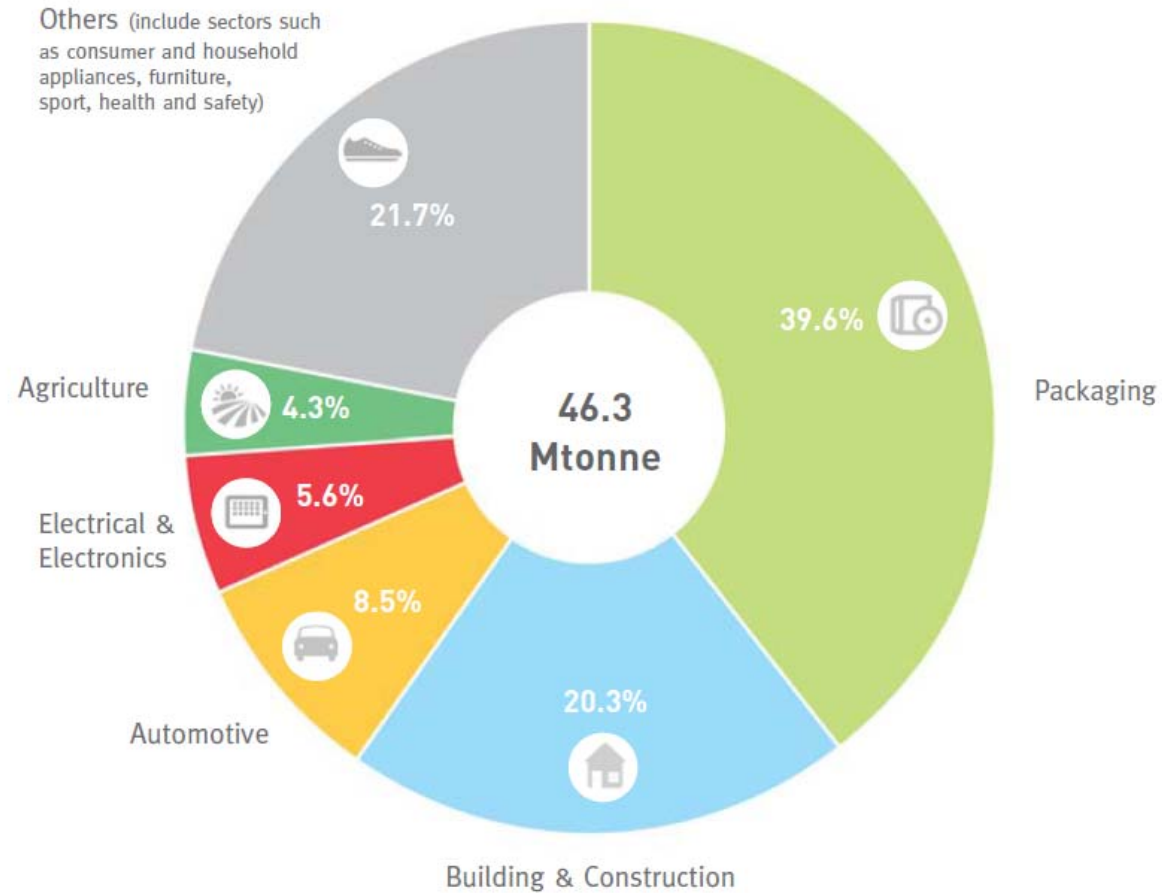
Source: British Plastics and rubber, www.mpes.co.uk

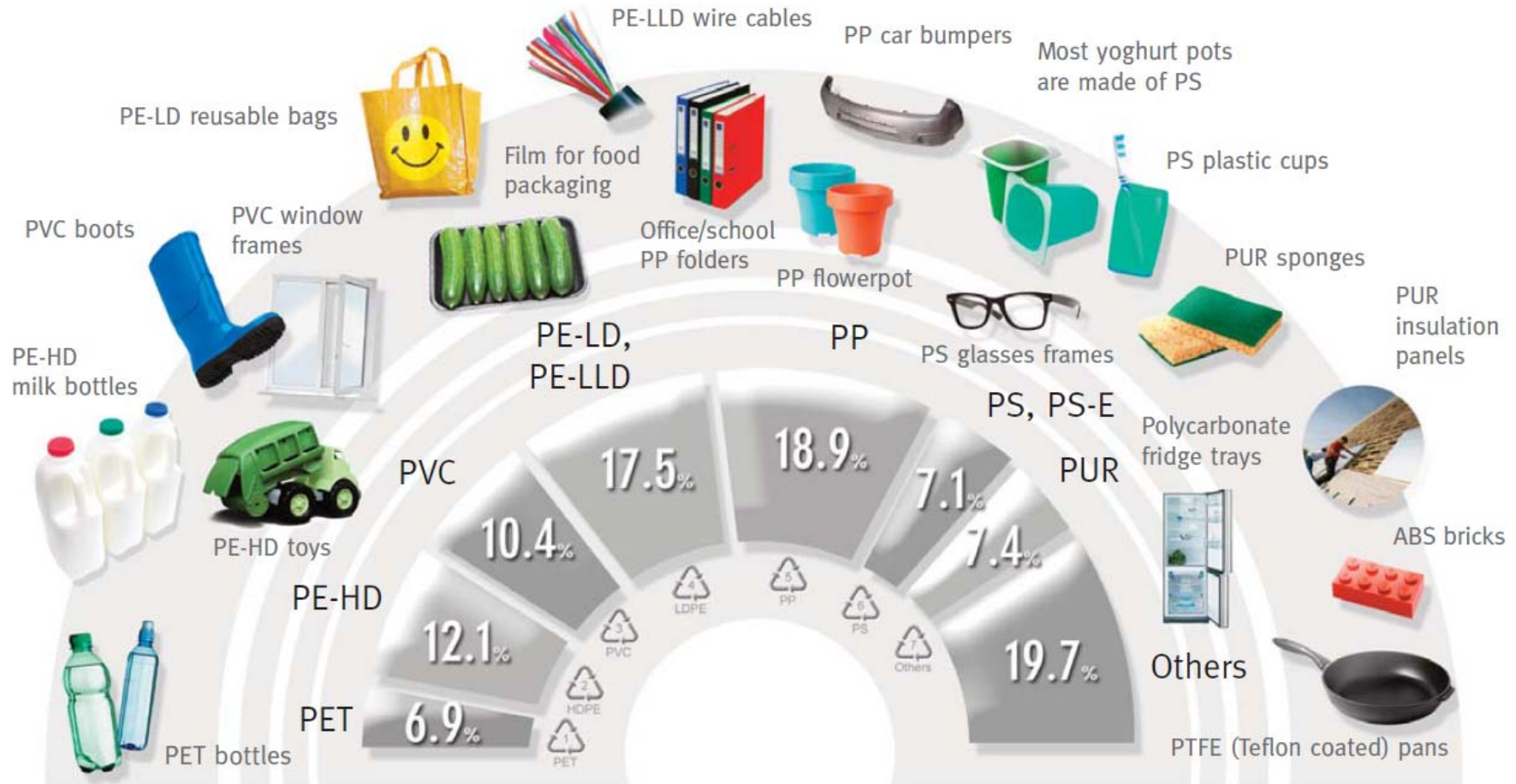
World plastics production



2013 World production of plastics materials (thermoplastics and polyurethanes)
Does not include other plastics (thermosets, adhesives, coatings and sealants) nor PP-fibers.
Source: PlasticsEurope (PEMRG) / Consultic

European plastic demand





European plastics demand* by polymer type 2013

Plastics – the Facts 2014/2015 - PlasticsEurope

European plastics industry

Jobs

The plastics industry gives direct employment to over

1.45 million



people in Europe

Companies

An industry in which more than

60.000

companies operate, most

of them SME's



Turnover

The European plastics industry had a turnover of

320 billion euro in 2013



Multiplier effect

The European plastics industry has a multiplier effect of **2.4** in GDP and almost **3** in jobs*



Innovation

The plastics sector is part of the

top 5 most innovative sector

in the EU representing 1 in 25 patents submitted by the industry between 2003 and 2012



Recycling

More than **6.6** million tonnes of plastics waste recycled in 2012



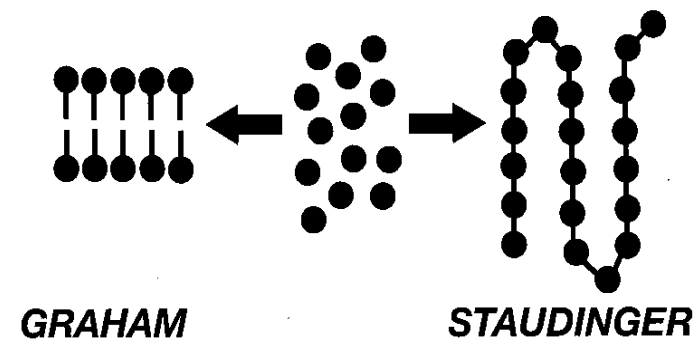
On the other hand....rubber was once thought (~1910) to be two isoprene units (-CH₂-C(CH₃)=CH-CH₂-) joined in a circle. The high viscosity was attributed to secondary forces grouping the circles into great aggregates. If so, then brominating the isoprene should dramatically alter the viscosity. It did not.

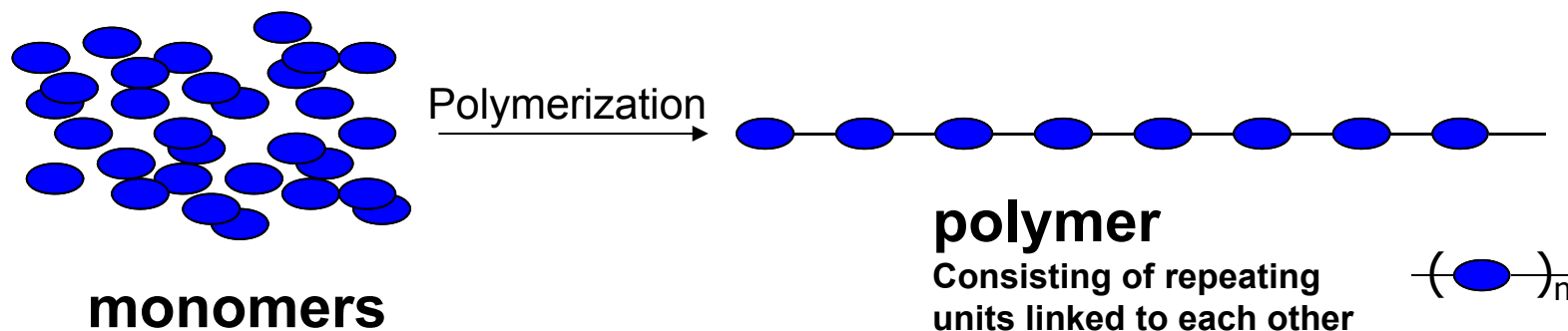
Still people were not convinced...because where were the end groups if the chains were really linear?

No one could find them, and no one could admit such a failure of chemical analysis...so linear polymerization was a mystery. Maybe polymers were large rings???

(Leaping ahead for awhile: In fact, it was a failure of chemical analysis that the end groups were not found. On a polystyrene of M=104,000 there are only 2 ends: Only 1 in 500 styrene units is different.)

→ **Polymers were assigned as organic colloids (loose physical aggregates) or as impure stuff.**





Polymer: a substance composed of macromolecules

Oligomer: A molecule of **intermediate relative molar mass**, the structure of which essentially comprises a **small plurality of units** derived, actually or conceptually, from molecules of lower relative molecular mass. A molecule is regarded as having an intermediate relative molecular mass if it has **properties which do vary significantly with the removal of one or a few of the units**. If a part or the whole of the molecule has an intermediate relative molar mass and essentially comprises a small plurality of units derived, actually or conceptually, from molecules of lower relative molecular mass, it may be described as **oligomeric**, or by **oligomer** used adjectivally.

Polymer (IUPAC-definition):

A substance composed of molecules characterized by the multiple repetition of one or more species of atoms or groups of atoms (constitutional units) linked to each other in amounts sufficient to provide a set of properties that does not vary markedly with the addition or removal of one or a few of the constitutional units.

Timeline

- Use of natural polymers: cellulose, skins, wools, resin, natural rubber since prehistoric times
- 1839 first trials of vulcanization of natural rubber by Charles Goodyear
- 1865 Parker: 1. Patent for celluloid
- 1901 Röhm: Polyacrylicacidesters
- 1906 Hofmann: Polyisoprene
- 1907 Bakeland: Bakelite (phenol-formaldehyd-resin); typical black
- 1915 Bayer: polydimethylbutadiene (synthetic rubber)
- Not until 1920 development of macromolecular chemistry by Hermann Staudinger (nobelprize 1953)
- 1926 PVC: Waldon Semeon of B.F. Goodrich (method to plastize)
- 1930 BASF: Polystyrene, Polyacrylonitrile
- 1933 Röhm: Plexiglass (PMMA)
- 1937 IG Farben: BUNA (first synthetic rubber)
 - DuPont: Nylon-6,6 (Carothers)
 - Bayer: Polyurethanes by using isocyanates (Otto Bayer)

- 1938/39 BASF: Nylon-6 = Perlon (ϵ -Caprolactam) (Dr. P. Schlack)
- 1939 ICI: HD-PE
- 1945 ICI: Terylen (Poly(ethylene terephthalate))
Hoechst: Trevira (Poly(ethylene terephthalate))
- 1946 DuPont: Teflon (Polytetrafluoroethylene)
- 1952/53 Ziegler: ZIEGLER-catalysts
-> Polymerization of ethylene at normal pressure
Natta: ZIEGLER-catalysts for the synthesis of substituted polyethylenes -> regular (= isotactic) polypropylene
result: Ziegler-Natta-Polymerization for the synthesis of tactic polymers
- 1956 DuPont: Polyoxymethylene $-(\text{-CH}_2\text{-O-})-$ (POM)
- 1957 General Electric and Bayer: Polycarbonates
- 1962 DuPont: Polyimides and Polyphenylenes
- 1964/65 Polyphenylenoxides, -sulfides, - sulfones
- 1970 PET/PBT (Polybutyleneterephthalate),
SBS (Stryrene-Butadiene-Styrene)
- 1975 Kevlar (aromatic polyamide, high performance polymer)
- 1970's Intensive investigations of block-, grafted- and star polymers
- 1977 Electrically conducting polymers (doped polyacetylene)

- Since 80s: Dendritic polymers, highly branched polymers
- Since 90s: Controlled radical polymerization
- 1986: First solar cell consisting of conducting polymer and C₆₀, Sariciftci
- 1989: Patent (Bayer) PEDOT for antistatic applications
→ later OLEDs, solar cells
- 90s: Polylactide: Discovery 1990, biodegradable polymers

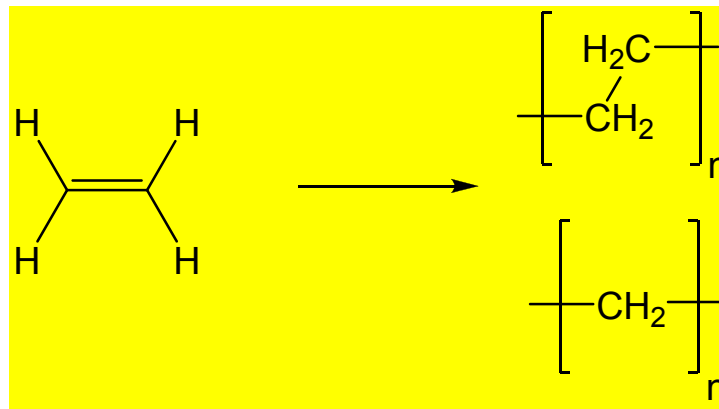
Four generations of polymeric materials

1. Polymers introduced to the market before 1950, simple plastics (PS, PVC, PMMA, LD-PE, Nylon)
2. Polymers with improved properties as construction materials, 1950-65 (HD-PE, PC, isotact. PP, PU)
3. Speciality polymers, since 1965, more complex chemical structure, high mechanical strength, high softening temperature (Poly(phenylenesulfide), Polyimide, Kevlar, Teflon)
4. Electrically conducting polymers, since 1977

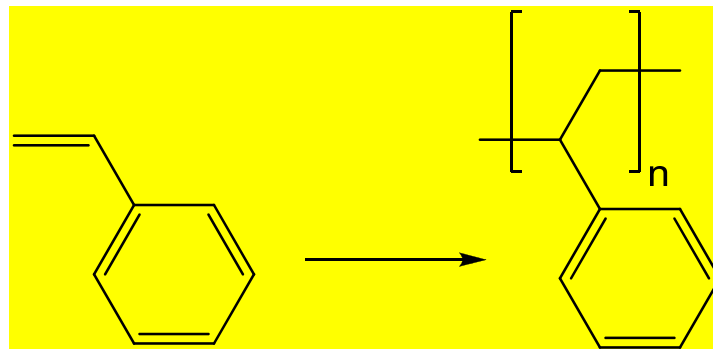
B. Rånby: Background – polymer science before 1977

Normally the monomer determines the name of the polymer:

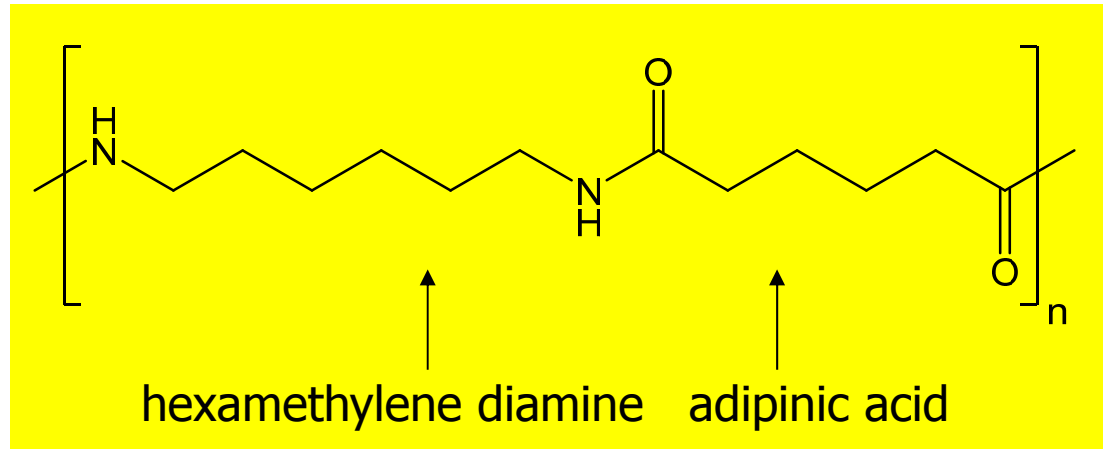
Poly(ethylene)/Polymethylene



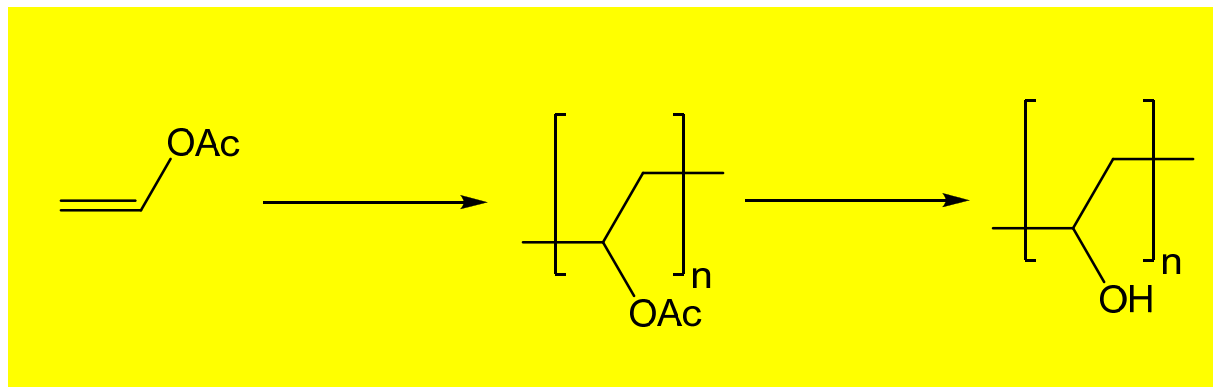
Polystyrene

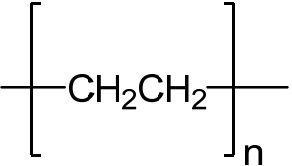
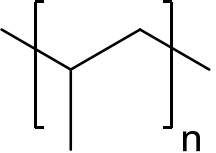
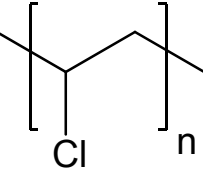
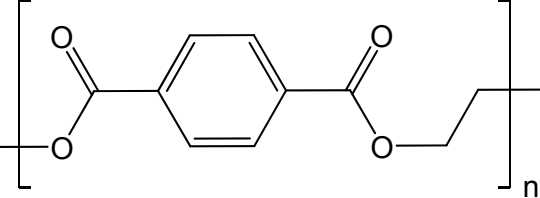
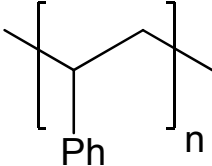


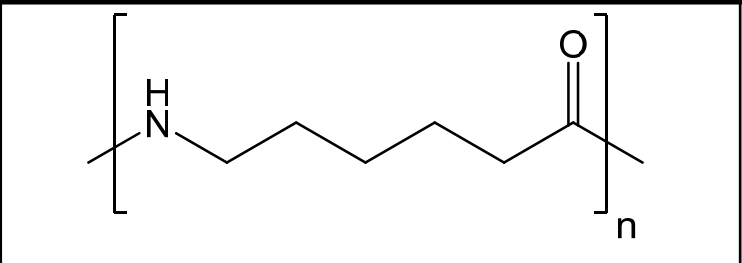
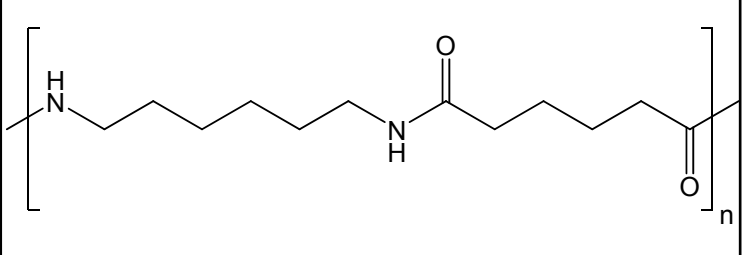
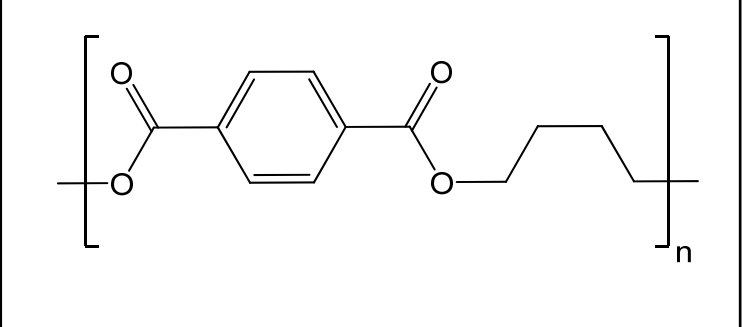
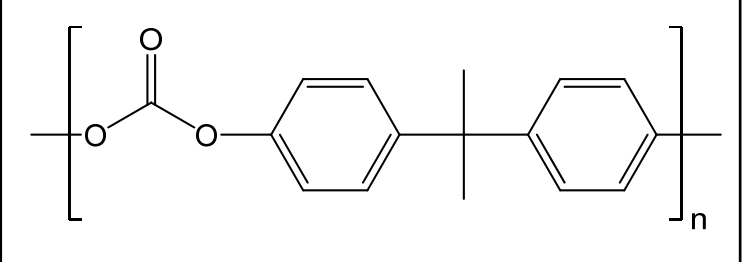
Poly(hexamethylene-adipinic acidamide) Nylon 66



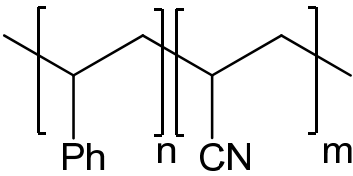
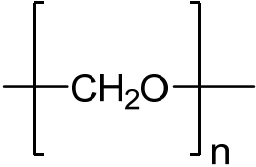
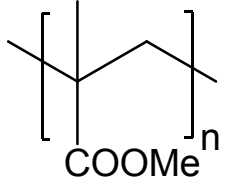
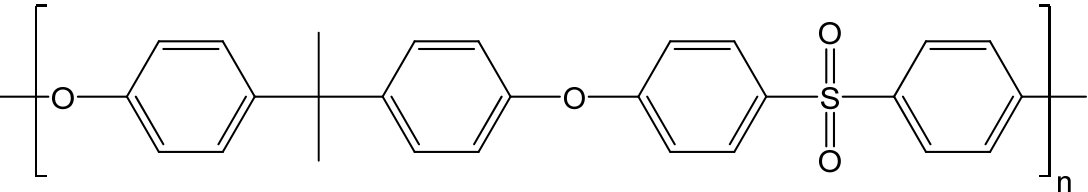
But: Polyvinylalcohol

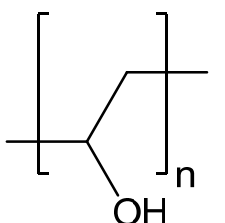
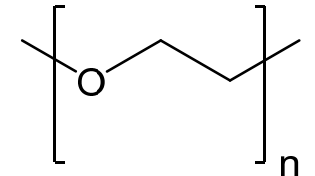
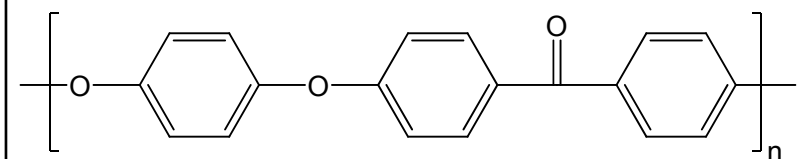
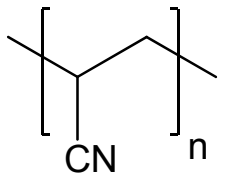


PE	Polyethylene		packaging gas pipes
PP	Polypropylene		packaging plastic mouldings
PVC	Polyvinylchloride		pipelines building material
PET	Poly(ethylene terephthalate)		food and liquid containers
PS	Polystyrene		insulation packaging

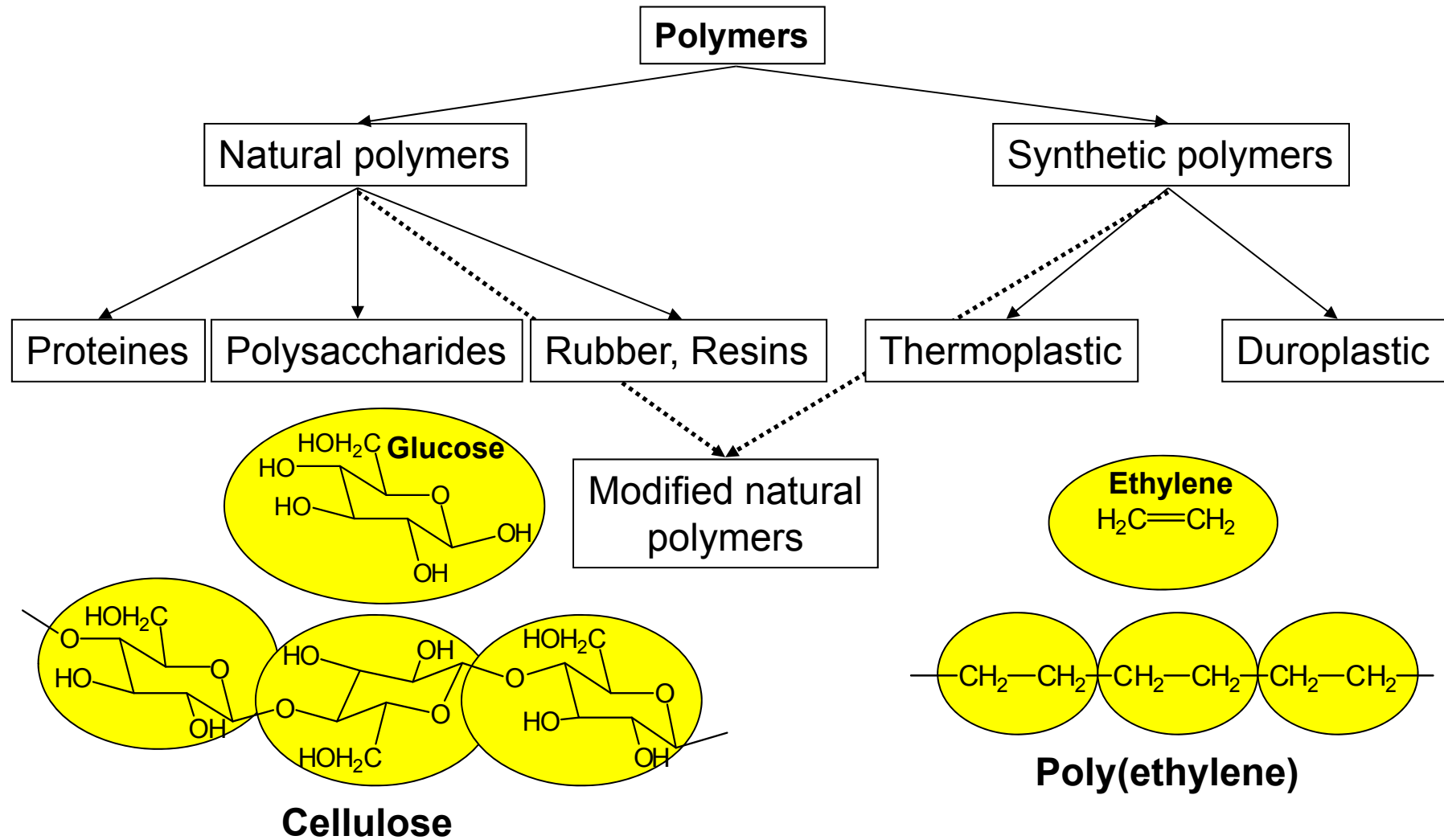
PA6	Polyamide 6		synthetic fibers
PA6	Polyamide 66		synthetic fibers
PBT	Poly(butylene-terephthalate)		enclosure of electric devices
PC	Polycarbonate		greenhouses CD, DVD

	Polyimide		electronic industry
PEI	Poly(ethyleneimine)		cell extraction
PI	Poly(isoprene)		rubber
PTFE	Poly(tetrafluoroethylene)		Teflon

SAN	Poly(styrene-acrylonitrile)		optics reflectors
POM	Poly(oxymethylene)		automobile parts fixative
PMMA	Poly(methylmethacrylate)		optics glass-like appl.
PSU	Poly(sulfone)	<p>Membranes, composite material</p> 	

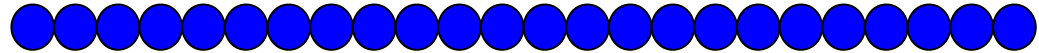
PVA	Polyvinyl alcohol		carbon dioxide barrier
PEG	Polyethylene glycol		clinical use cosmetics
PEEK	Poly(etheretherketone)		medical implants parts of pumps
PAN	Poly(acrylnitrile)		fibers

Distinction of origin and manufacturing



Homopolymer

only one monomer species



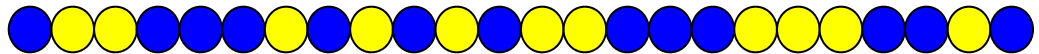
poly(A)

Copolymers

more than one monomer species

Bipolymers/Copolymers

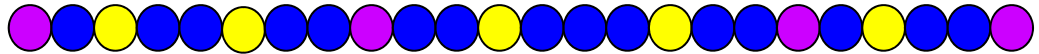
two different monomers



poly(A-co-B)

Terpolymers

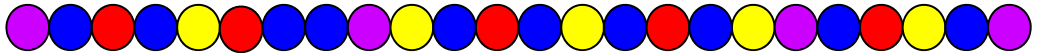
three monomer species



poly(A-co-B-co-C)

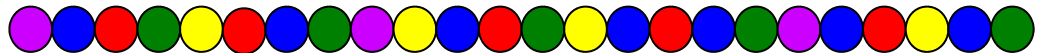
Quarterpolymers

four different monomers



poly(A-co-B-co-C-co-D)

Pentapolymer,

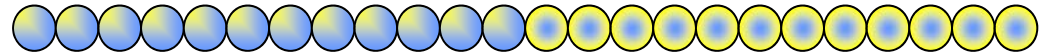


poly(A-co-B-co-C-co-D-co-E)

No constitution, only composition

Block polymers

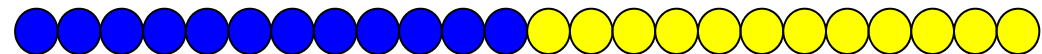
composed of blocks in sequence



Block copolymer

more than one monomer, sequence of blocks, each block consists only out of one monomer

Diblock copolymer



poly(A)-block-poly(B)

Triblock terpolymer



poly(A)-block-poly(B)-block-poly(C)

Tetrablock quaterpolymer



poly(A)-block-poly(B)-block-poly(C)-block-poly(D)

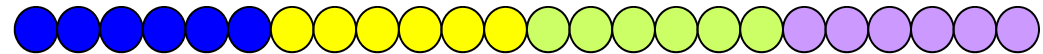
Multiblock copolymer



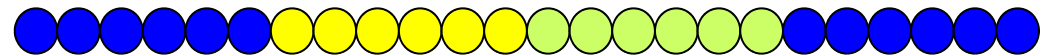
poly(A-B)_n

Examples

Tetrablock quarterpolymer



Tetrablock terpolymer



Triblock terpolymer



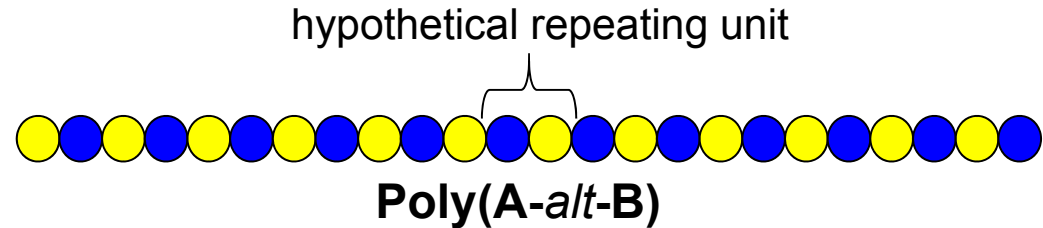
Triblock bipolymer



number of blocks, number of different monomers
(latin) (greek)

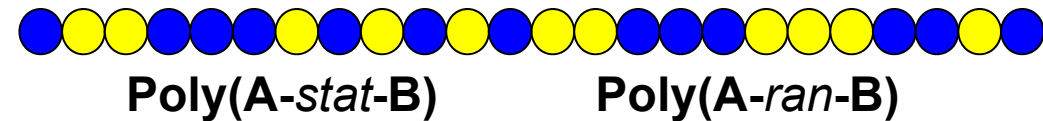
Alternating copolymer

may be considered as homopolymer



Statistical copolymer

distribution obeys statistical laws,
otherwise random

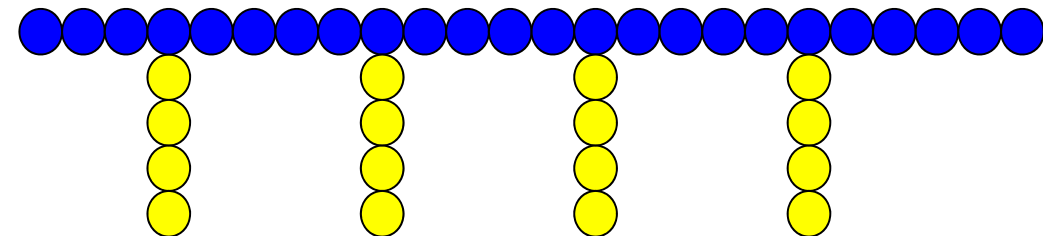


Gradient copolymer

two monomers (one end many of
monomer A, monomer B *vice versa*)



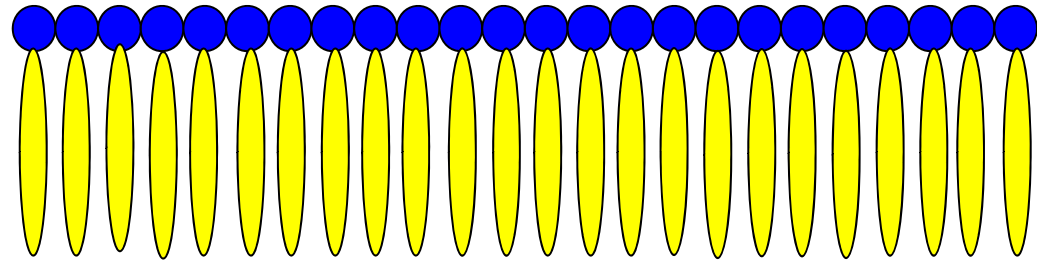
Graft copolymer



Poly(A)-*graft*-poly(B)

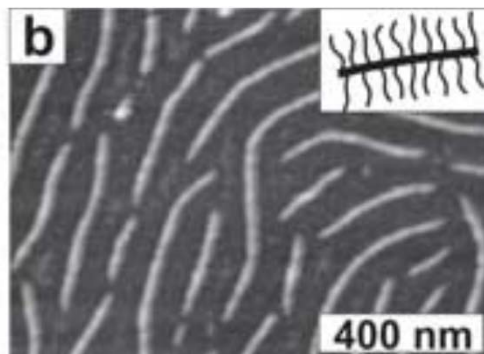
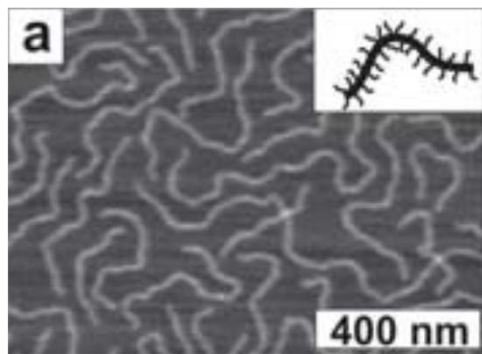
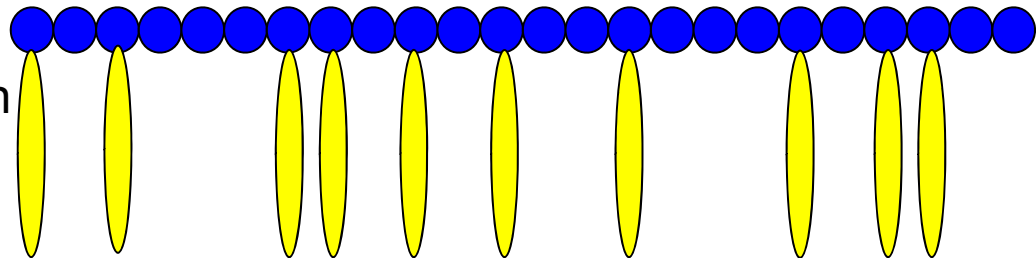
Comb copolymers

each monomer has a side chain



Graft copolymers

some monomers have a side chain

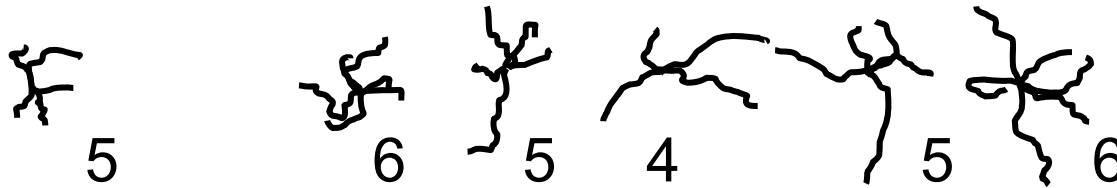
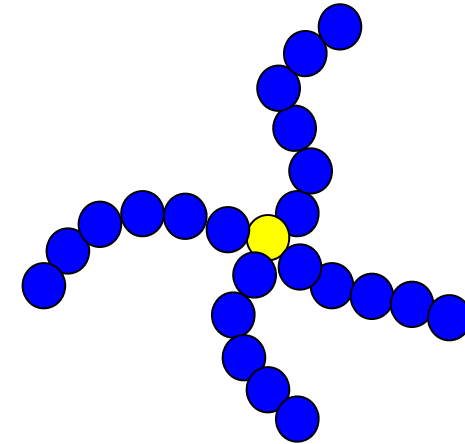
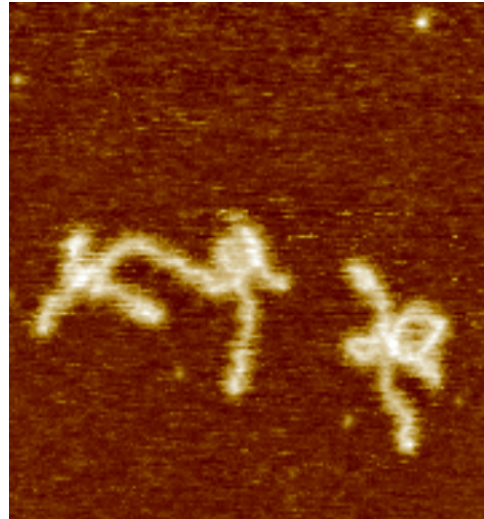
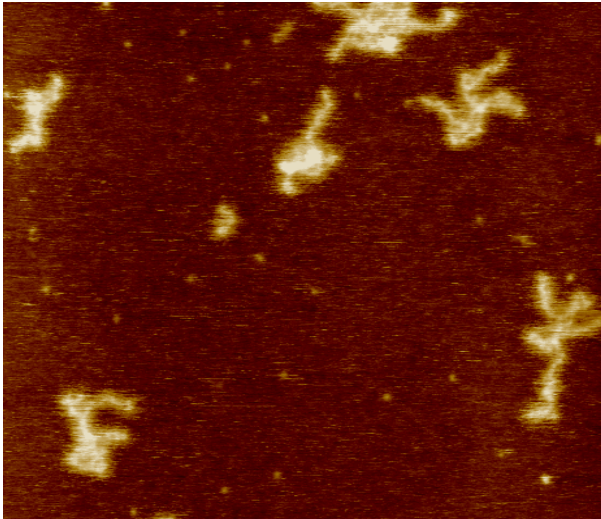


- a) Short side chains
- b) Long side chains

M. Matyjaszewski, *et al.*, *Nature* **2006**, 440, 191

Star-shaped polymers

several side chains (arms) are connected with one core



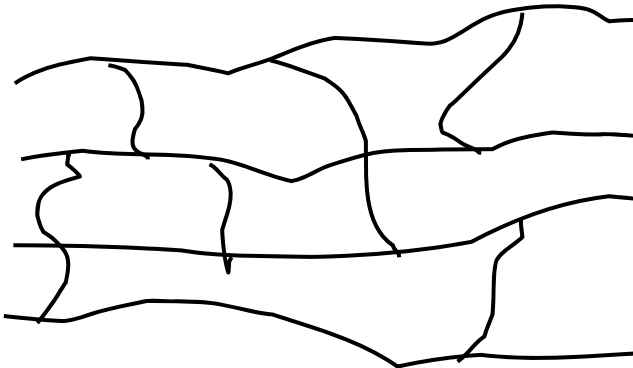
linear polymers



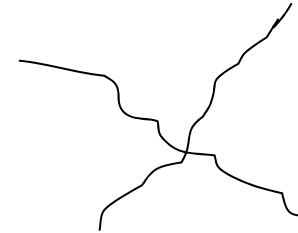
branched polymers



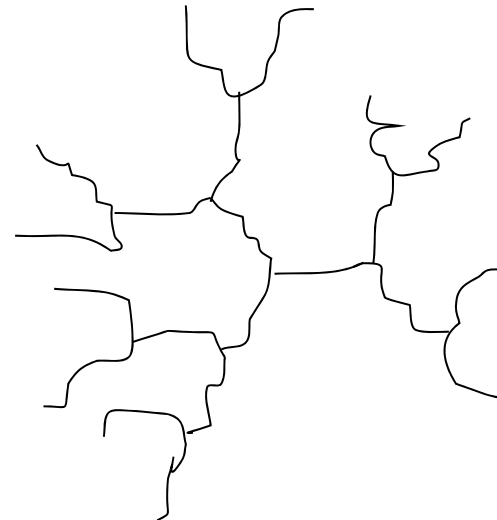
crosslinked polymers



star-polymers



dendrimers (3rd generation)



branched polymers: short-chain branches = oligomeric subunits
long-chain branches = polymeric subunits

Polymerization

Step-growth reactions

Polycondenzation

Polyaddition

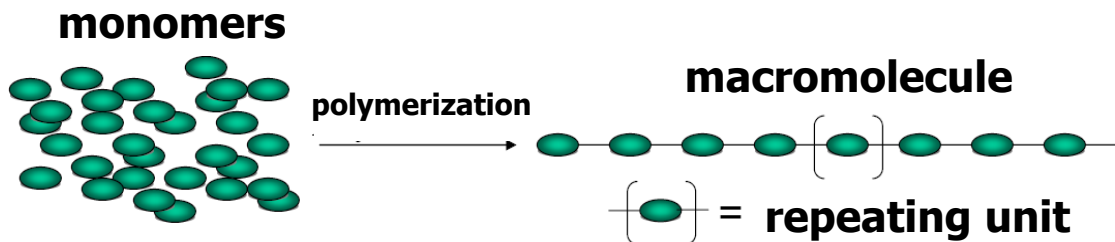
Chain-growth reactions

radical polymerization

cationic polymerization

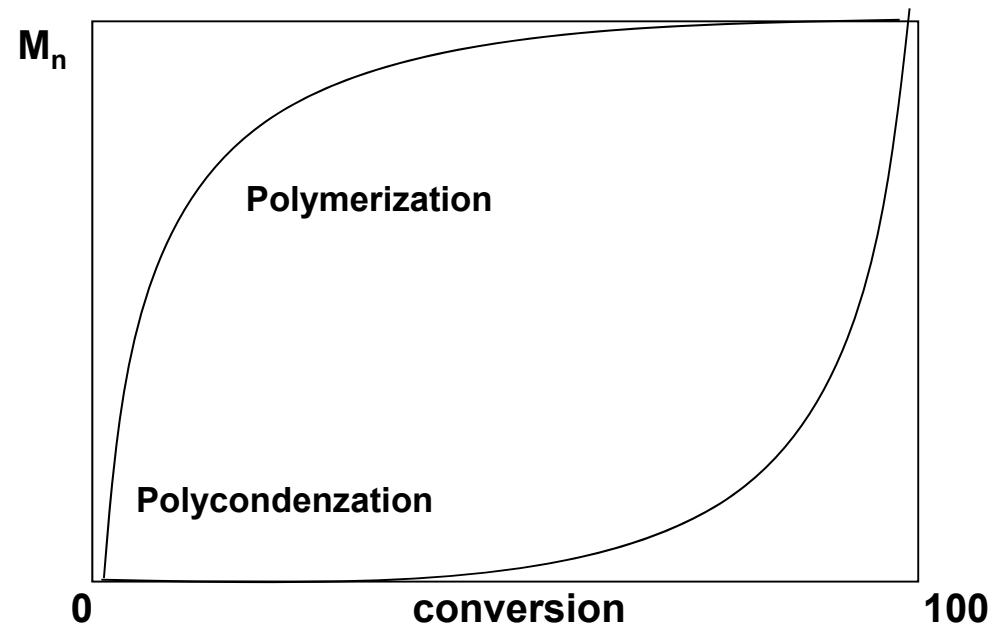
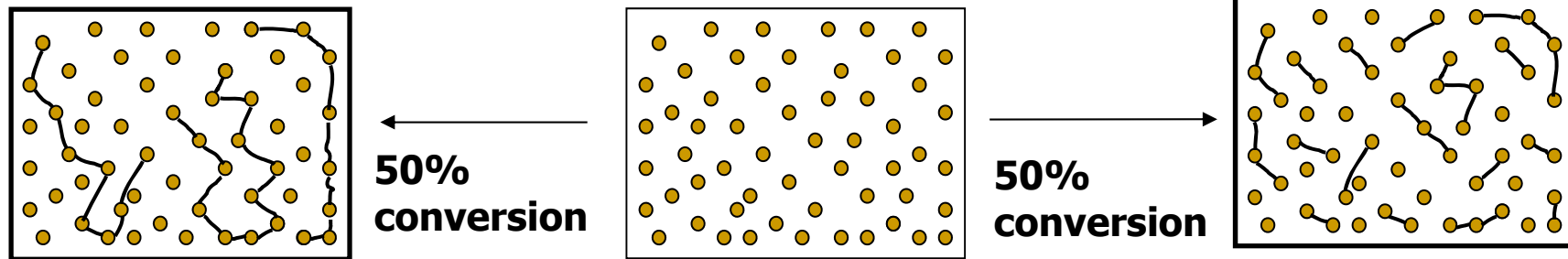
anionic polymerization

coordinative polymerization



Chain-growth reaction

Step-growth reaction

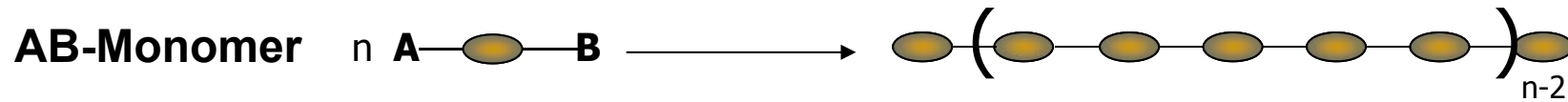


Chain growth:

- Chain reaction
- Intermediate products and stages not always isolatable
- Highly reactive intermediates (radicals, ions)
- High energy of activation
- High molar masses are achievable very fast

Step growth:

- Stepwise build-up of the polymer
- Same energy of activation for each reaction step
- Usually: intermediate products and stages isolatable
- High molar masses only at high conversions

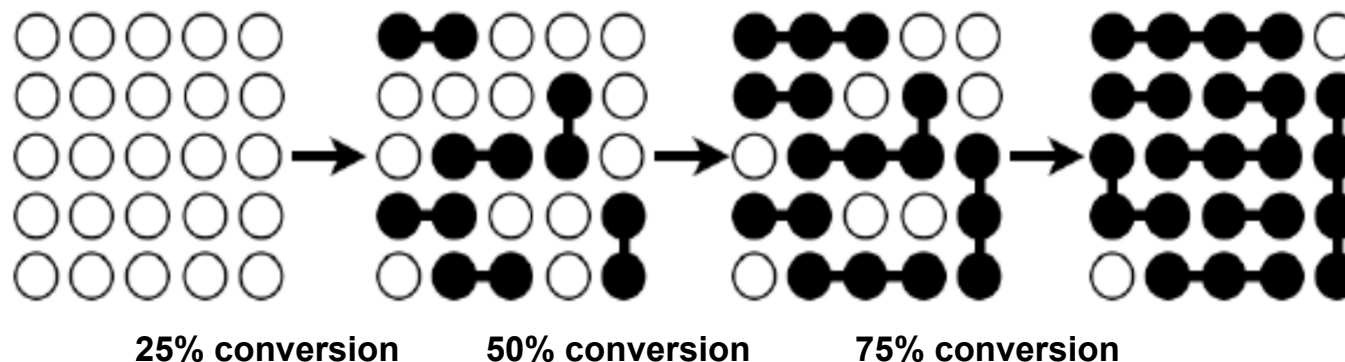


Polycondenzation: Coupling of two M/O/P involving the elimination of small molecule (e.g., water)

Polyaddition: Coupling of two M/O/P without loss of molecules

- No initiator necessary
- Use of catalysts is possible
- Theoretically: All molecules (M/O/P) have the same reactivity
- All molecules (M/O/P) react

A high conversion is necessary for high molar masses



AB-system:



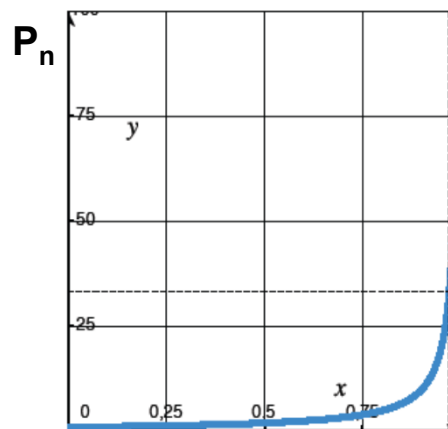
$$p = \frac{N_0 - N_t}{N_0}$$

and
$$P_n = \frac{N_0}{N_t} = \frac{1}{1-p}$$

Carothers equation

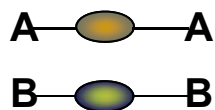
p	P _n
0.5	2
0.95	20
0.990	100
0.9990	1000
0.9999	10000

Tieke.



p – conversion
 P_n – degree of polymerisation
 N₀ – number of functional units at t = 0
 N_t – number of Functional units at t

Carothers equation for AA/BB-systems:



$$P_n = \frac{q + 1}{q + 1 - 2pq}$$

$q = n_A / n_B$ $n_A \leq 1$; $p = \text{conversion}$,
 n_A, n_B – mols of monomer A and B resp.

$$q = 1 \Rightarrow P_n = 1/(1-p); \quad p = 1 \Rightarrow P_n = (1 + q)/(1 - q)$$

aberrations: Formation of cycles, aberrations from equilibrium, reactivity of the endgroups is not independent of the molecule length, side reactions

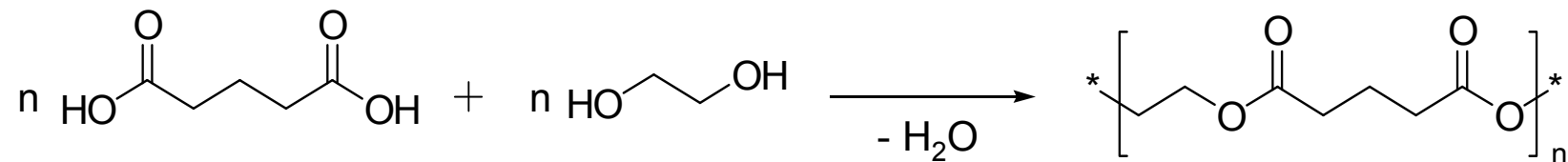
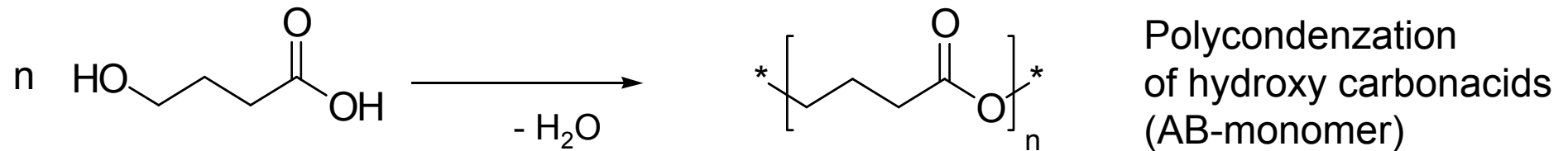
r	p	P_n
0.9524	0.999	39.4
1	0.999	1000
0.9524	1	41.0

$$r = 1/1.05 \\ = A/B$$

Important:

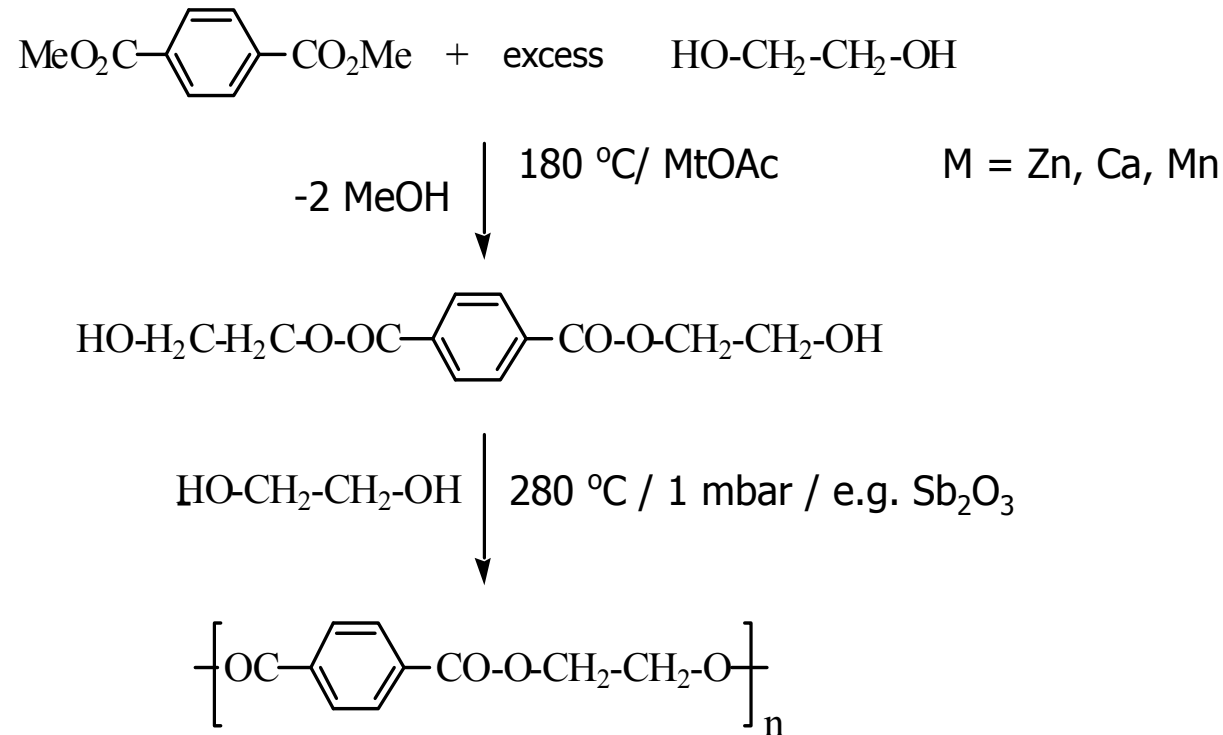
**High molar masses
are only achievable when
the exact same monomer
amounts are used !**

Synthesis of polyesters



Polycondensation of dicarboxylic acids and dialcohols (AA- und BB-monomer)

Polyethyleneterephthalate (PET)

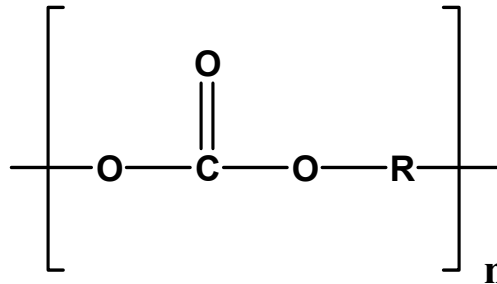


Also: other diols (1,4-butandiol \rightarrow polybutyleneterephthalate)

Semi-crystalline ($T_m = 265 \text{ }^\circ\text{C}$, $T_g = 80 \text{ }^\circ\text{C}$)

Melt-spinning of fibers (textile industry)

Bottles



Thermoplasts

Semi-crystalline with low crystallinity → transparent

Hard, impact-proof

Heat resistant (glass transition ~150 °C, processing 300 °C)

Solubility: Insoluble in water and alcohols

Soluble in many organic solvents (e.g. CH₂Cl₂, CHCl₃, pyridine)

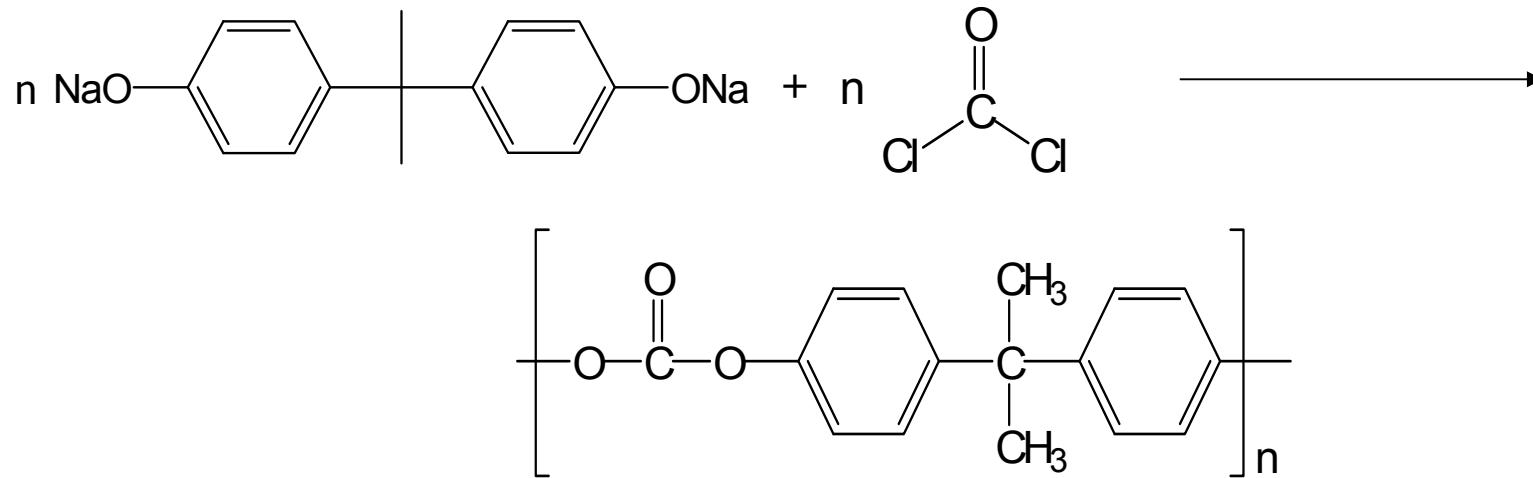
Application: Injection molding, isolation foils, Makrolon (Bayer), CD and DVD



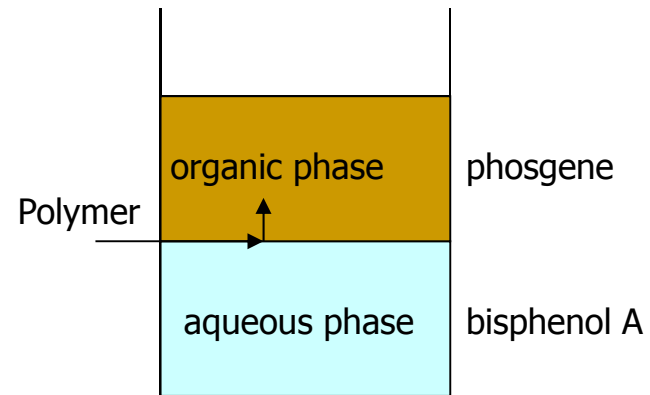
Bayer - Makrolon

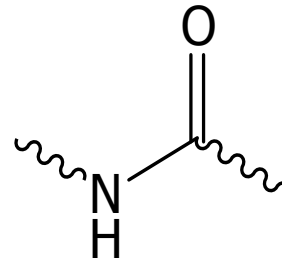
Synthesis I

Interface polycondensation (Schotten-Baumann)

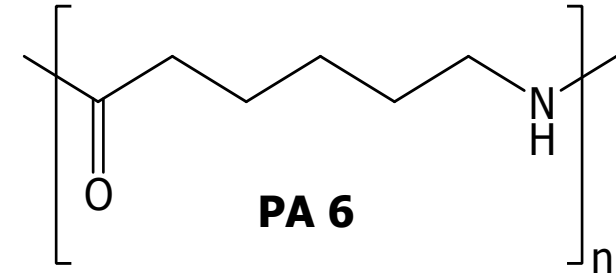


high molar masses are achievable
 but
 removing of NaCl is difficult



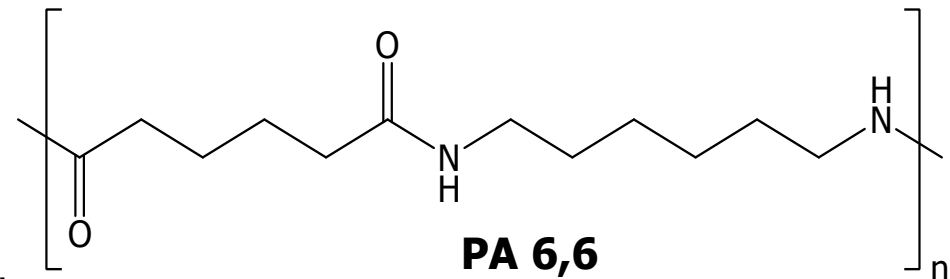


Perlons: constitutional unit and monomer identical
 ω -amino carboxylic acids or lactams



Nylons: constitutional unit consists of two different monomers
diamino and dicarboxylic acid monomer

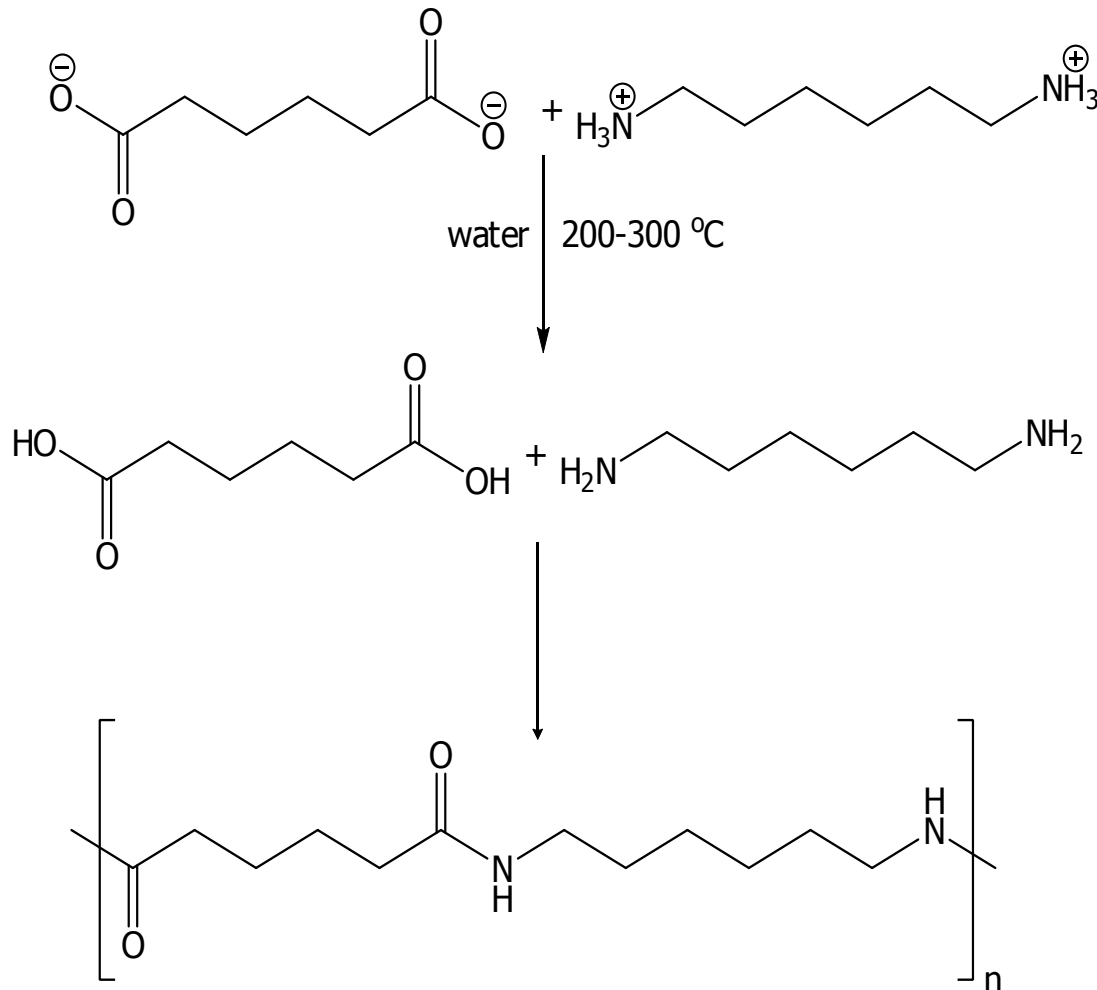
- Polyamides contain –NHOC–group
- Proteines, polyaramides (aromatic amines and carboxylic acids)
- Nylon xy
x = number of the C-atoms of the amine
y = number of the C-atoms of the carboxylic acids
- Hydrogen bonding → high mechanical strength



Applications: Fibers, engineering plastics
Melting point 260 °C, 30-50% crystallinity,
 $M_n = 15000-50000$ g/mol, PDI ~ 2

Synthesis

Nylons:

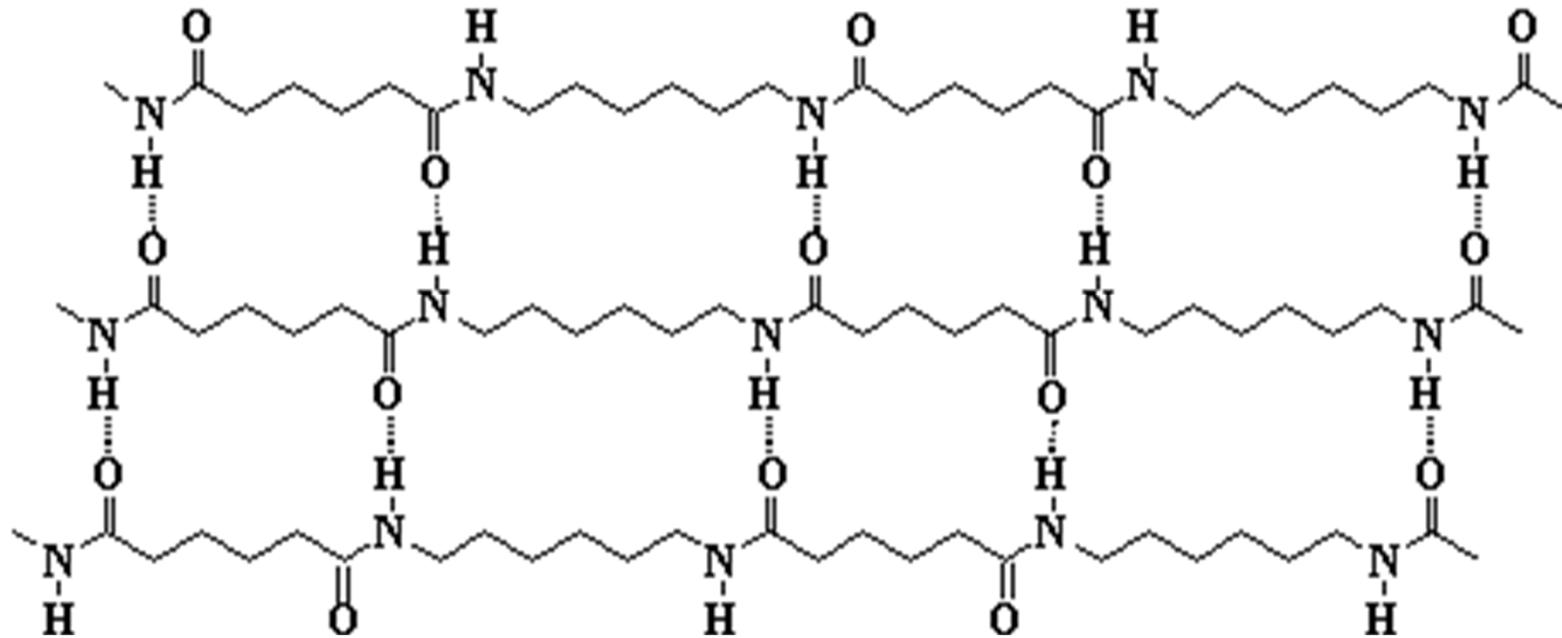


AH-salt (made of dicarboxylic acid [adipinic acid] and diamine [hexamethylene diamine])
Purification by recrystallization
→ exact stoichiometry
Molar mass is adjusted by addition of mono carboxylic acid

But also interface polycondensation is possible:



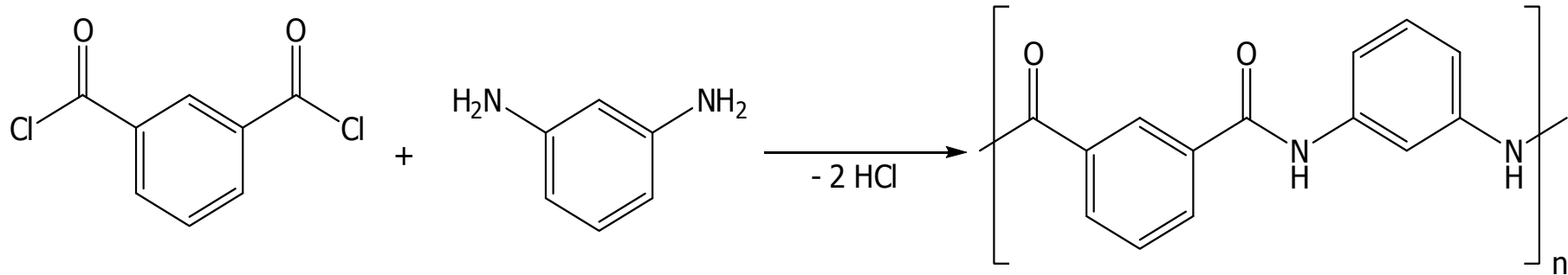
Hydrogen bonding in Nylon 6,6



high mechanical strength



lower mechanical strength

poly(*m*-phenylene-isophthalamide) (DuPont: Nomex)

High melting point (380-390 °C), soluble in DMF/LiCl

High permanent heat resistance

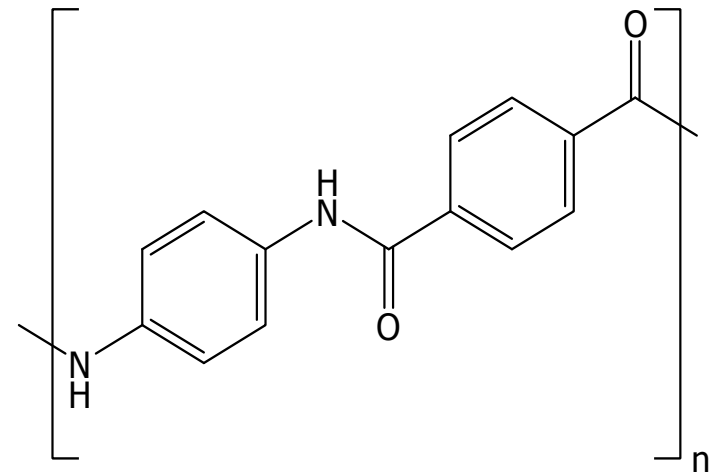
poly(*p*-phenylene-terephthalamide) (DuPont: Kevlar)

Unmeltable (decomposition 425 °C)

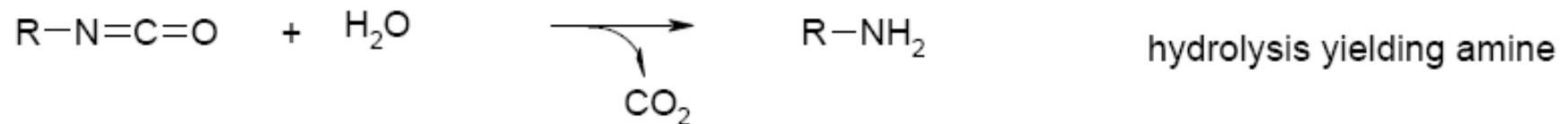
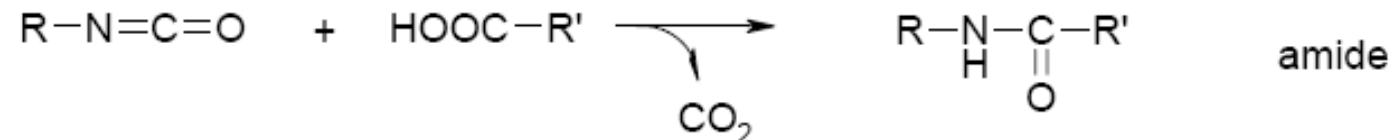
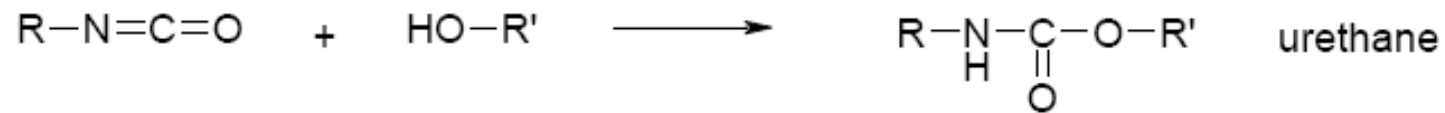
Soluble in DMAc/LiCl, sulfuric acid

High permanent heat resistance, mechanical strength

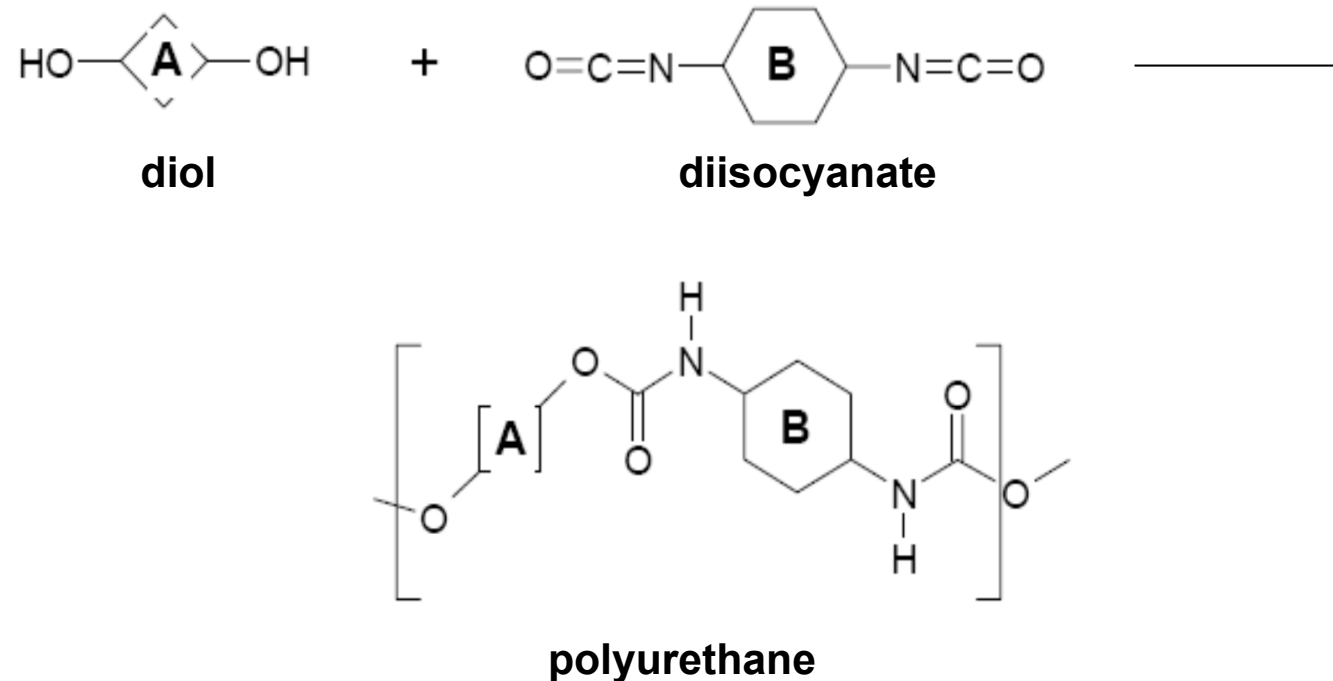
Polycondensation has to be carried out in solution



Isocyanate chemistry – basic reactions



Isocyanate chemistry – polyurethanes



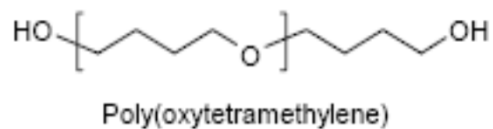
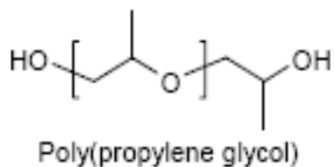
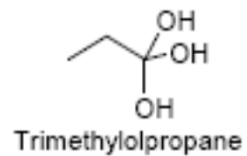
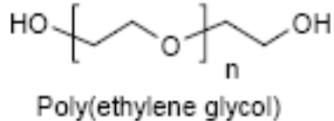
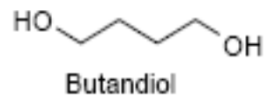
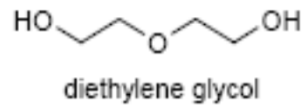
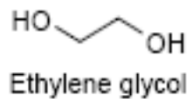
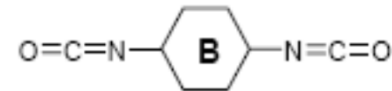
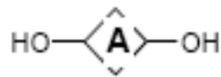
Variation of diol and diisocyanate → adjustment of chain flexibility
→ adjustment of physical properties

Alkyl polyurethanes → reaction in melt; aromatic polyurethanes →
reaction in solution

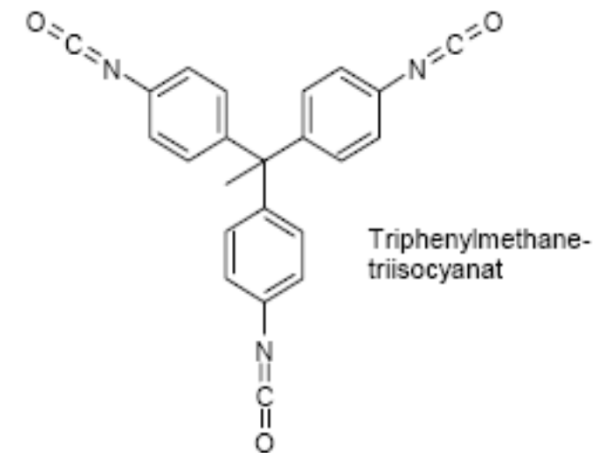
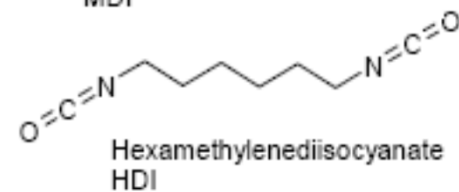
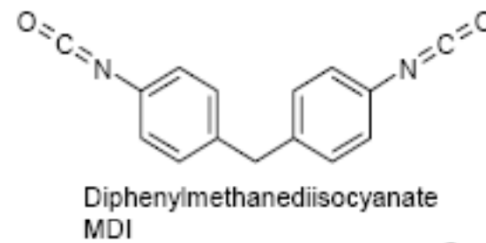
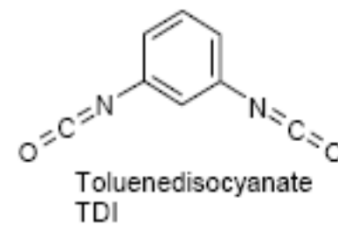
Catalysts: Tertiary amines (DABCO), tin compounds [$n\text{Bu}_2\text{Sn}(\text{OAc})_2$]

Lecture Prof. Menzel, TU Braunschweig.

Isocyanate chemistry – different building blocks



and many more



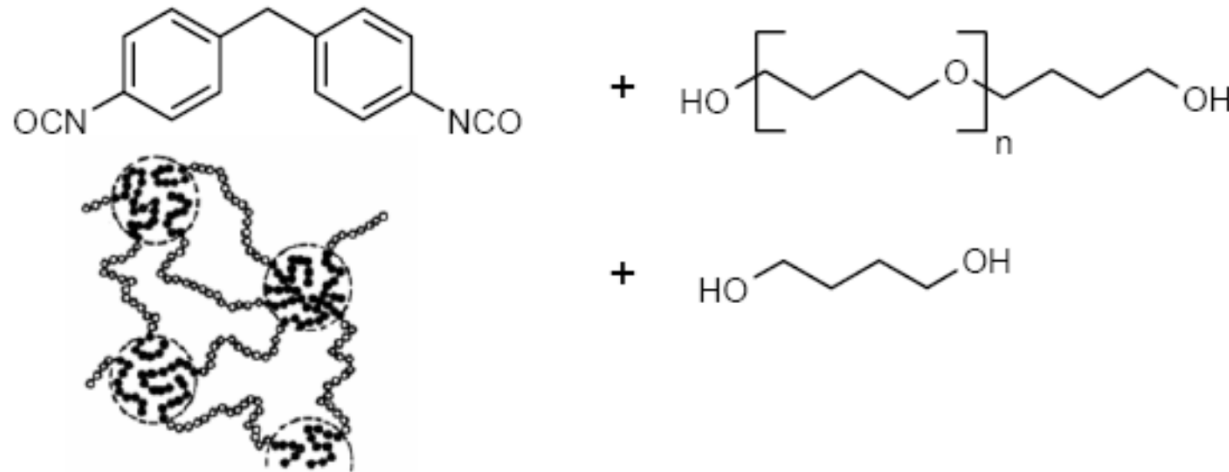
and many more

Thermoplastic elastomers (TPU):

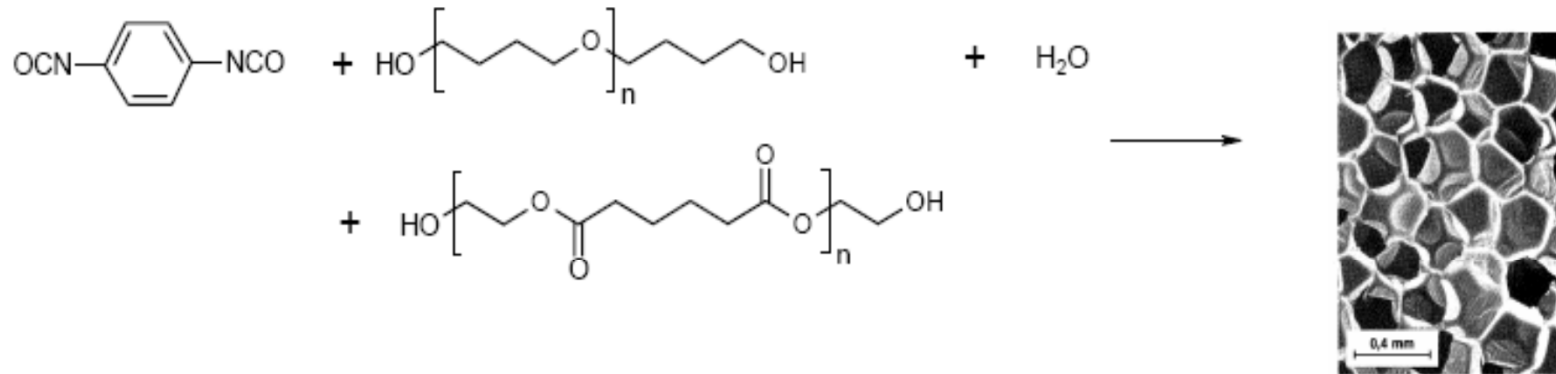
Diisocyanates and oligomers (diols) → prepolymer (isocyanate endgroups)
soft segment

Diisocyanate and butanediol → prepolymer (OH-endgroups)
hard segment

Combination of the two prepolymers



Polyurethane foams



two phase system:
gas and polymer

└─ gas for foaming



Hardness and pore structure depends on ratios of monomers, water content, ...

Lecture Prof. Menzel, TU Braunschweig.

Applications:



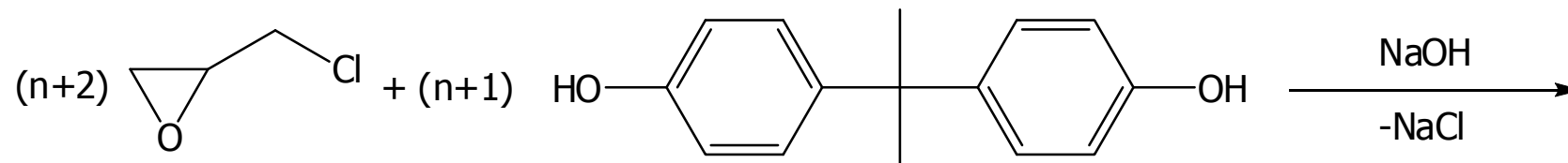
insulation



www.basf.pfe.com, www.technicalfoams.nl.

History

1936 Pierre Castan (Switzerland) produced an amber colored, low melting solid from the reaction of bisphenol A with epichlorhydrin, at the same time S. O. Greenlee (USA) discovered this reaction



Ciba (licensed Castan's work) became one of the major epoxy resin manufacturers

1946 introduction of Araldite adhesive resin by Ciba

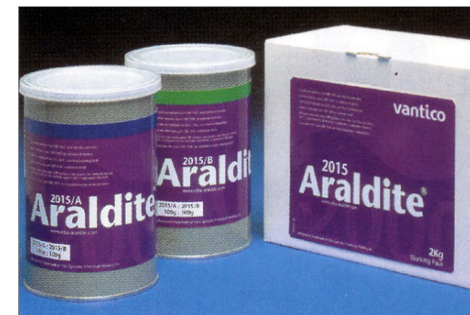
Still present today → Huntsman

1952 start of the industrial production of epoxyresins



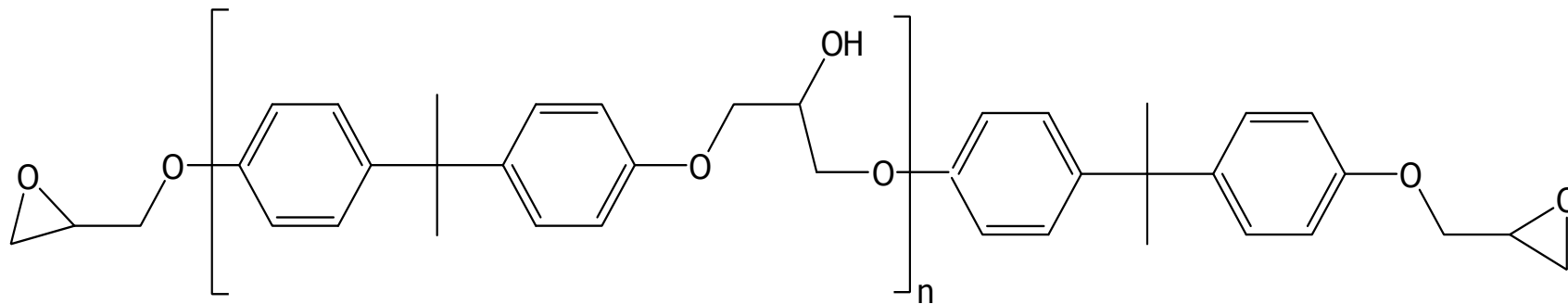
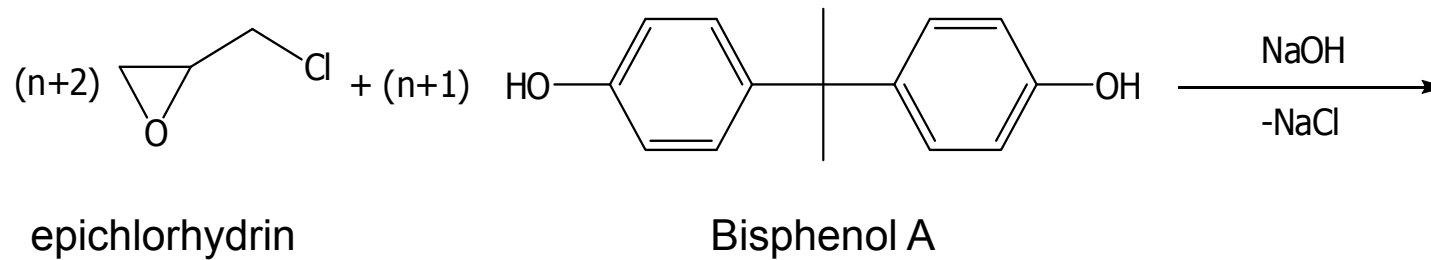
two component
adhesive

www.glue-shop.com, www.it-logistics.de.

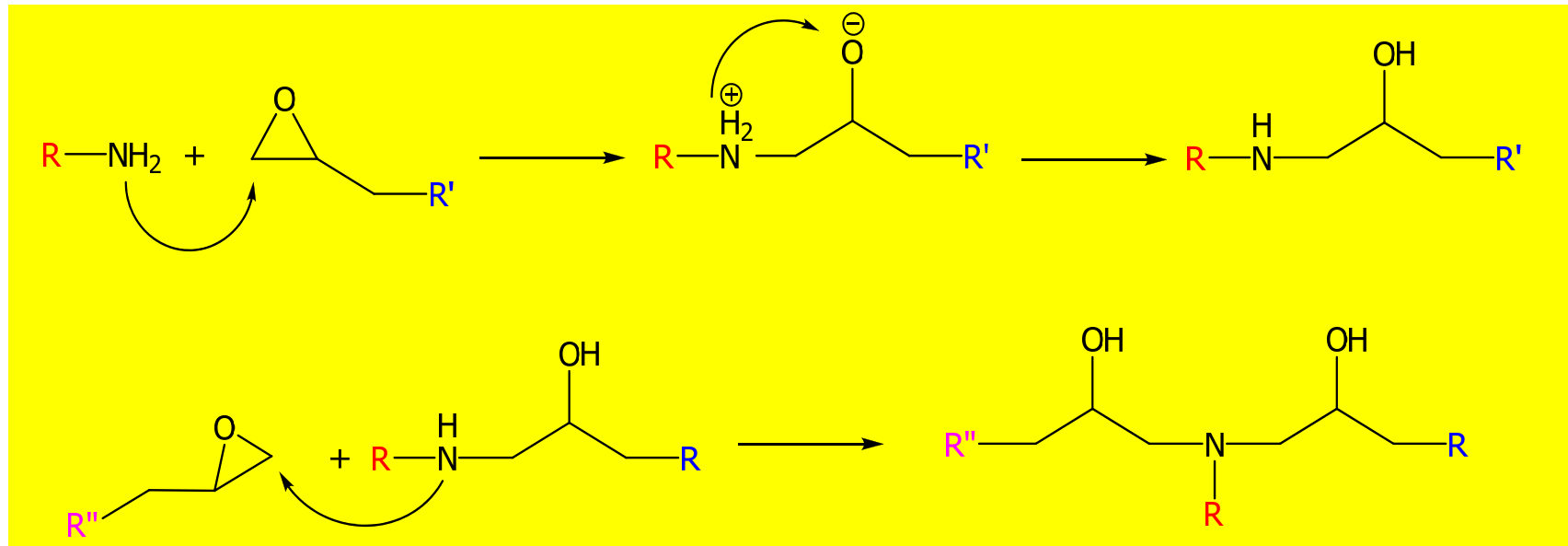
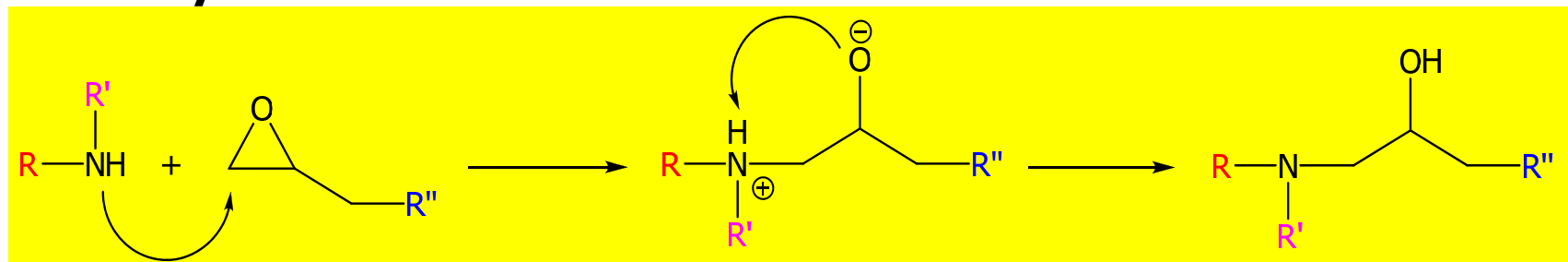


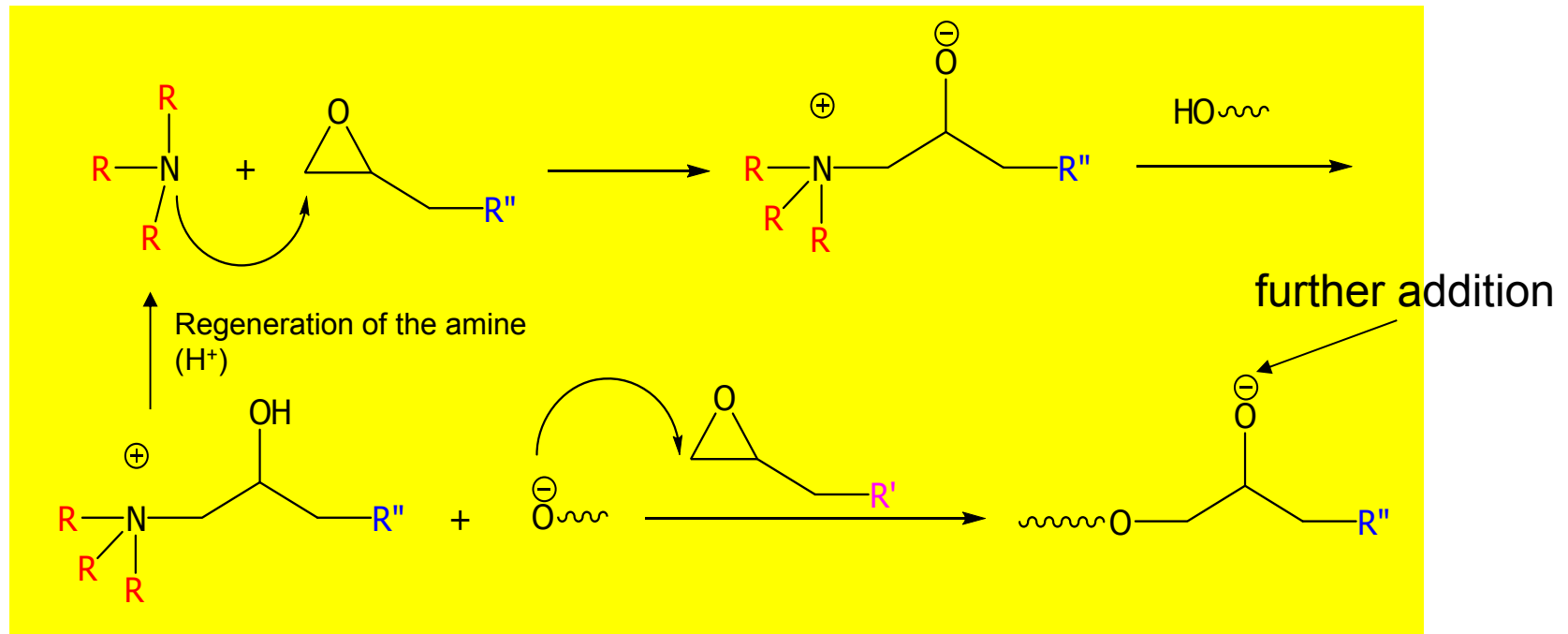
Epoxy resins

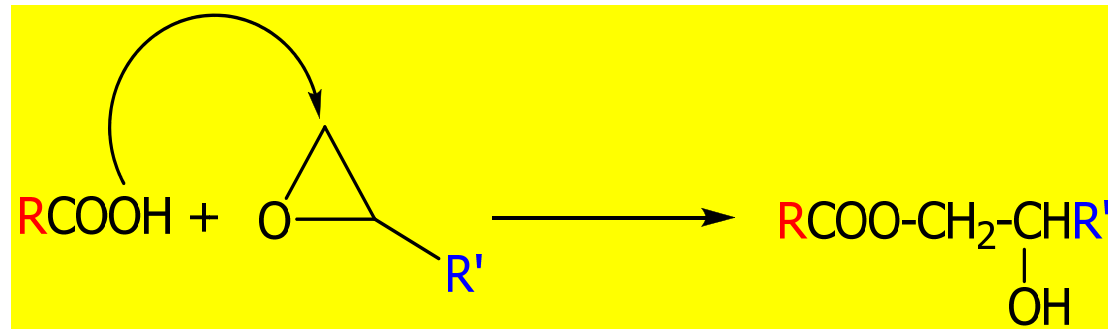
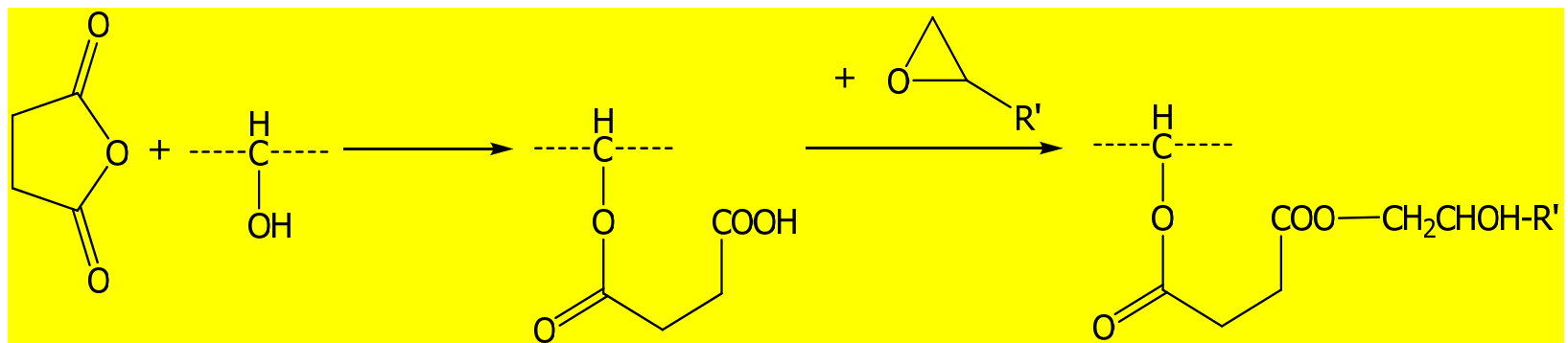
No pure polyaddition, also polycondenzation steps are involved.



Bisphenol-A-diglycidyl ether ($n = 0-30$)

Crosslinking:**Primary amines****Secondary amines**

Crosslinking:**Tertiary amines**

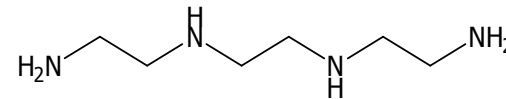
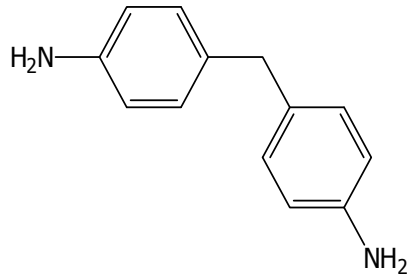
Crosslinking:**Carboxylic acids****Anhydrides**

Epoxy resins

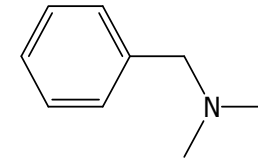
Typical curing agent:

Amines

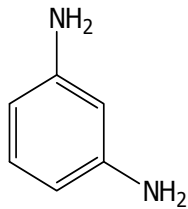
4,4'-diamino-diphenylmethane



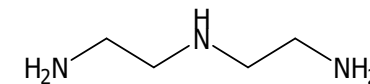
triethylene diamine



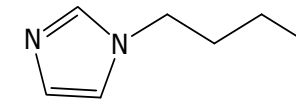
benzyldimethylamine



m-phenylene diamine

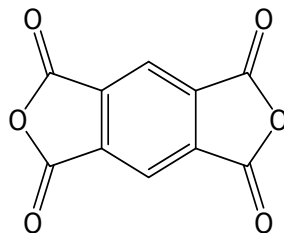


diethylene diamine

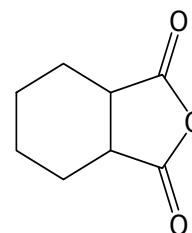


N-n-butylimidazole

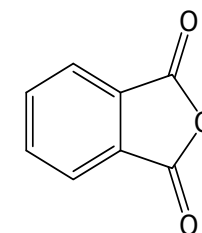
Anhydrides



pyromellithic acid anhydride



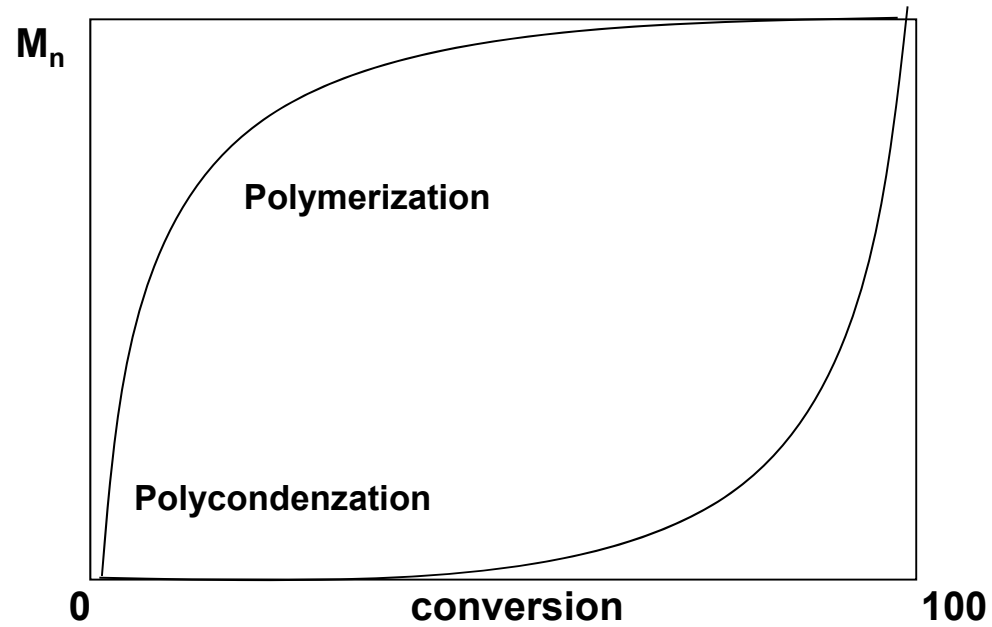
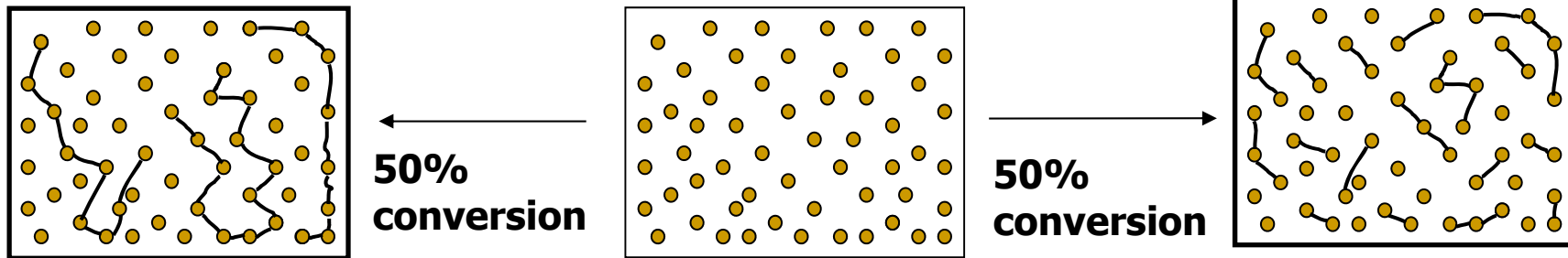
hexahydrophthalic anhydride



phthalic anhydride

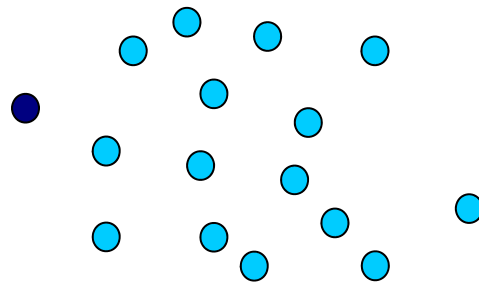
Chain-growth reaction

Step-growth reaction



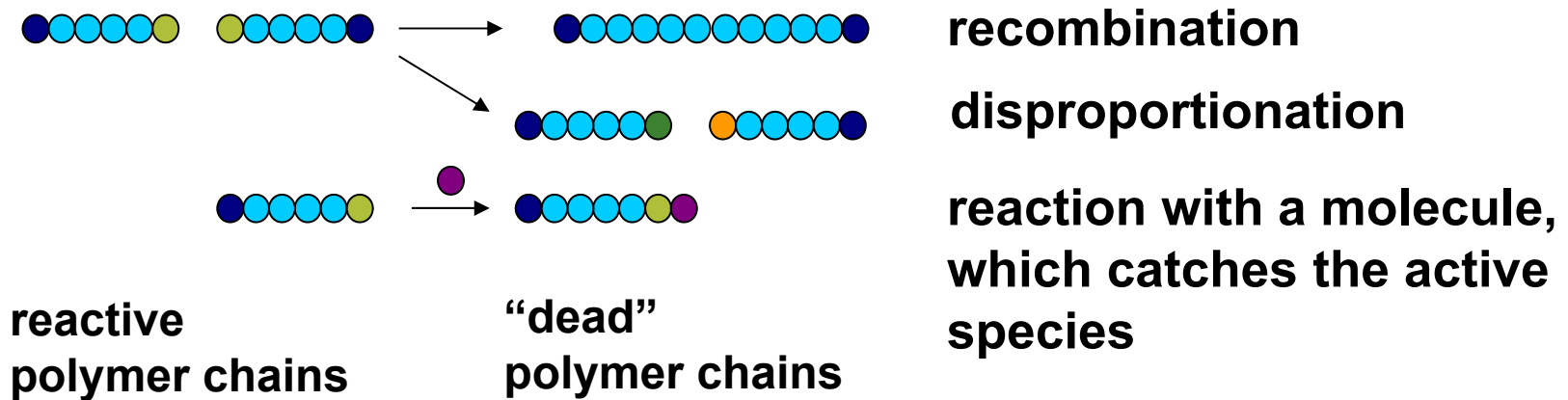


radical polymerization $R\cdot$ radical former (e.g., peroxy-compounds)	anionic polymerization R^{\ominus} anions (e.g., <u>BuLi</u>)	cationic polymerization R^{\oplus} cations (e.g., <u>TosOMe</u>)	coordinative polymerization $\square ML_n$ metall complexes (e.g., Ti-complex)
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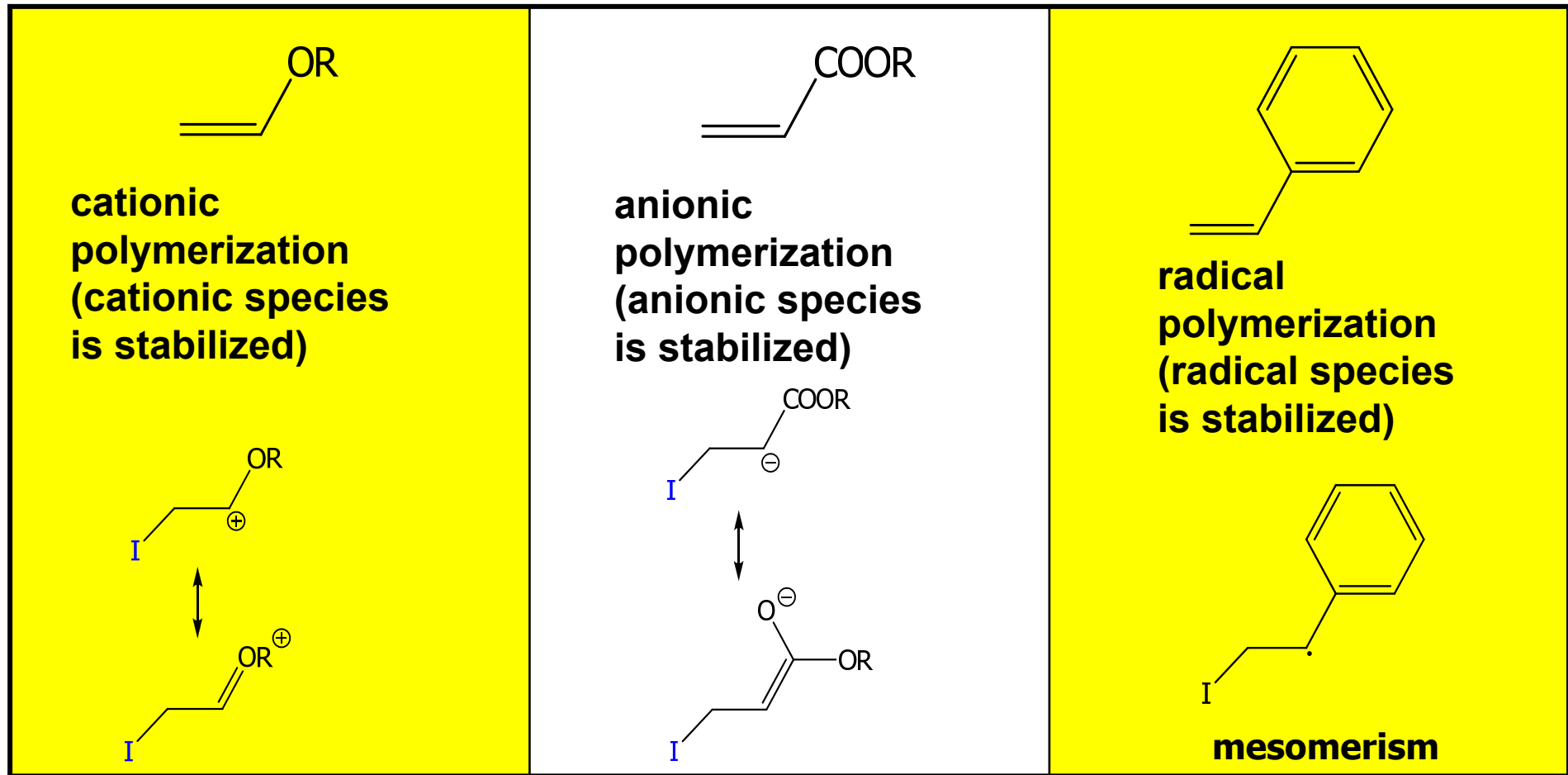


monomers

0. addition of initiator ●
 initiating of a monomer → “reactive monomer”
 growing chain
 termination



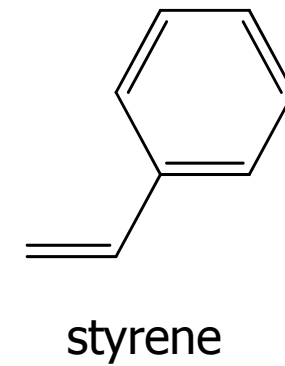
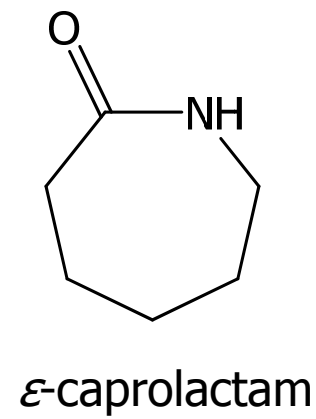
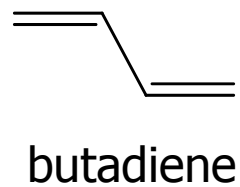
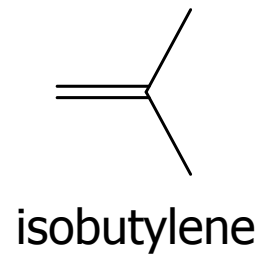
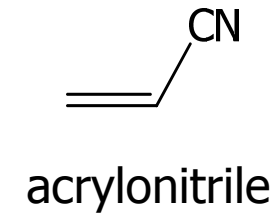
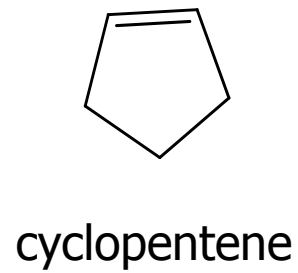
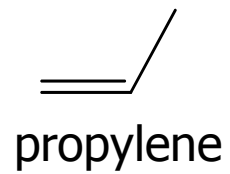
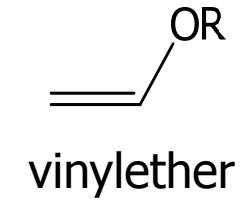
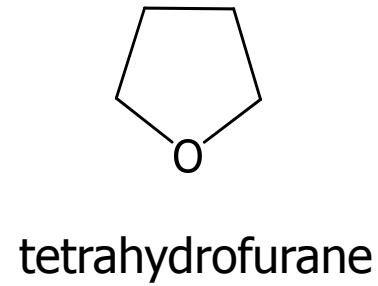
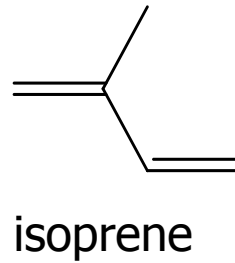
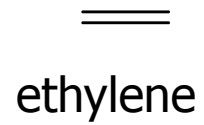
Not all monomers can be polymerized by all four methods
 Kind of the monomer and substitution of the double bond
 determines the reactivity of the monomer
 → possibilities to polymerize it

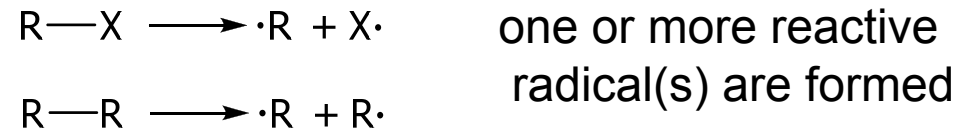
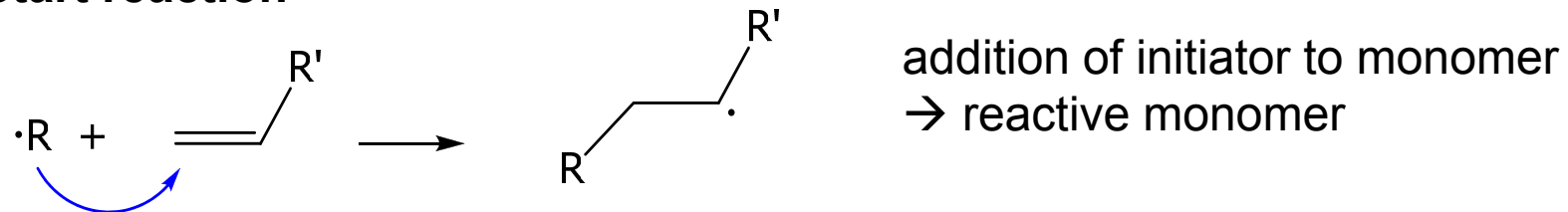
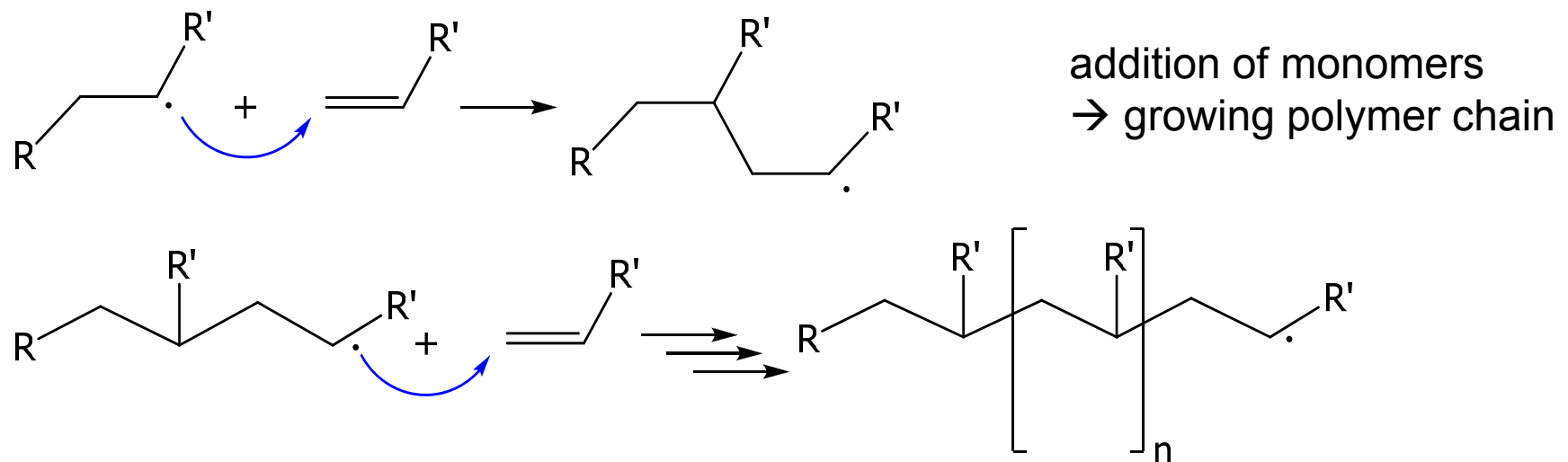


Polymerizations of selected monomers:

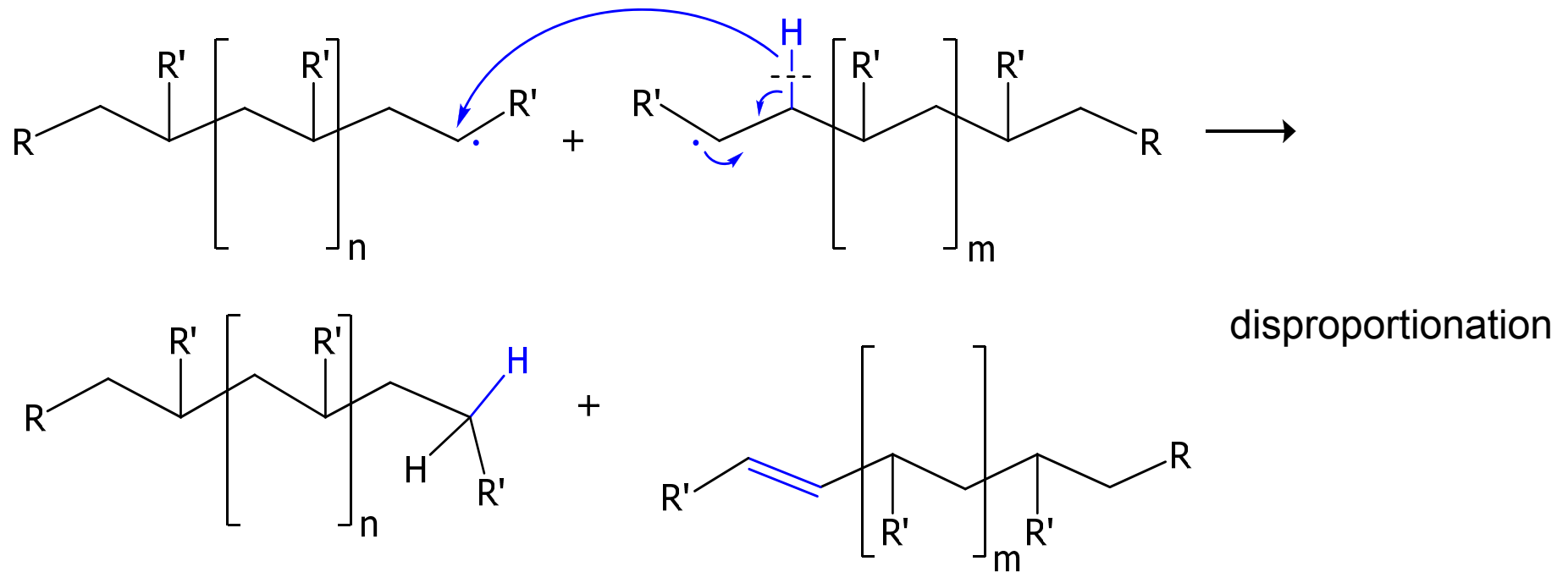
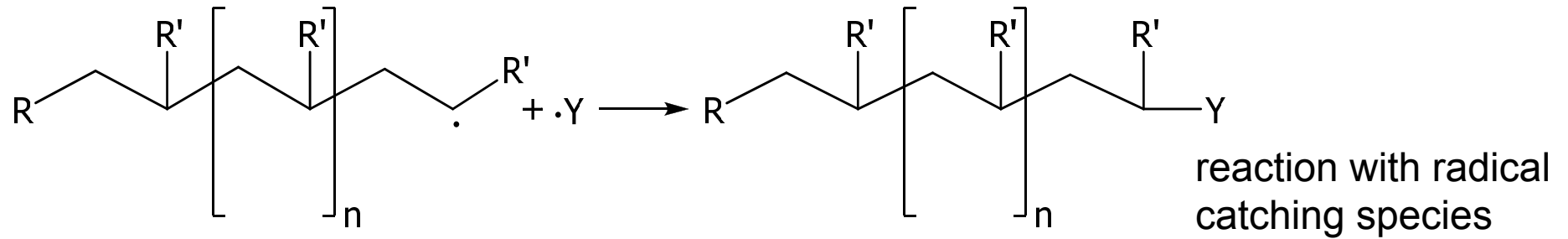
☑ industrial scale ✓ possible

Monomer	radical	anionic	cationic	polyinsertion
ethylene	☑			☑
propylene			✓	☑
isobutylene			☑	✓
isoprene	✓	✓	✓	☑
cyclopentene				☑
butadiene	☑	☑	✓	☑
tetrahydro- furane			☑	
ethylene oxide		☑	✓	
caprolactam		☑	✓	
vinylether			☑	✓
acrylonitrile	☑			
styrene	☑	✓	✓	✓

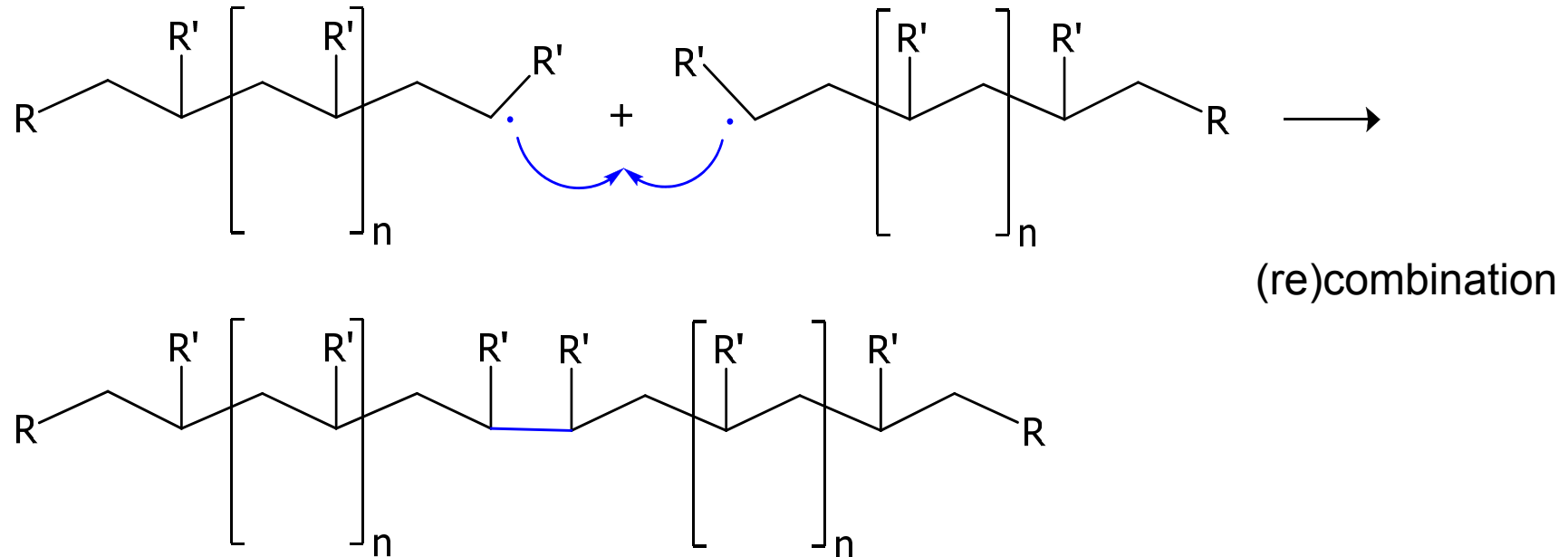


1) Initiation**2) Start reaction****3) Chain reaction (propagation)**

4) Termination I



4) Termination II



termination depends on monomer and temperature:

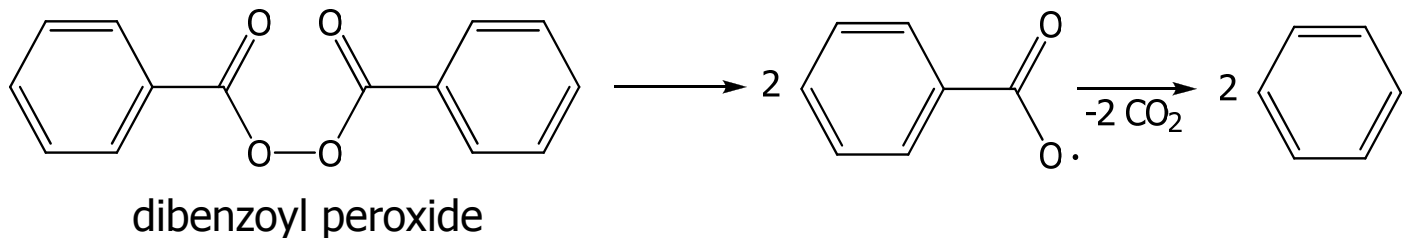
styrene (up to 160 °C)	mainly recombination
MMA (> 60 °C)	mainly disproportionation
acrylonitrile	recombination
vinylacetate	disproportionation

Radicals are produced by homolytic thermal decomposition of:
carbon-carbon-bonds (no great importance)
organic and inorganic peroxides
azoinitiators

By high energetic radiation

By redox reactions

Peroxides



Organic peroxides are potentially explosive!
Organic peroxides are used in technical scale
Mass polymerization of styrene, ethylene, vinylchloride
Both radicals can initiate the polymerization
(influenced by environment, monomer)

Inorganic peroxides and redoxreactions

Predominantly for polymerizations in aqueous phase

H_2O_2 : unsafe (formation of oxygen), used in redoxsystems



also others:



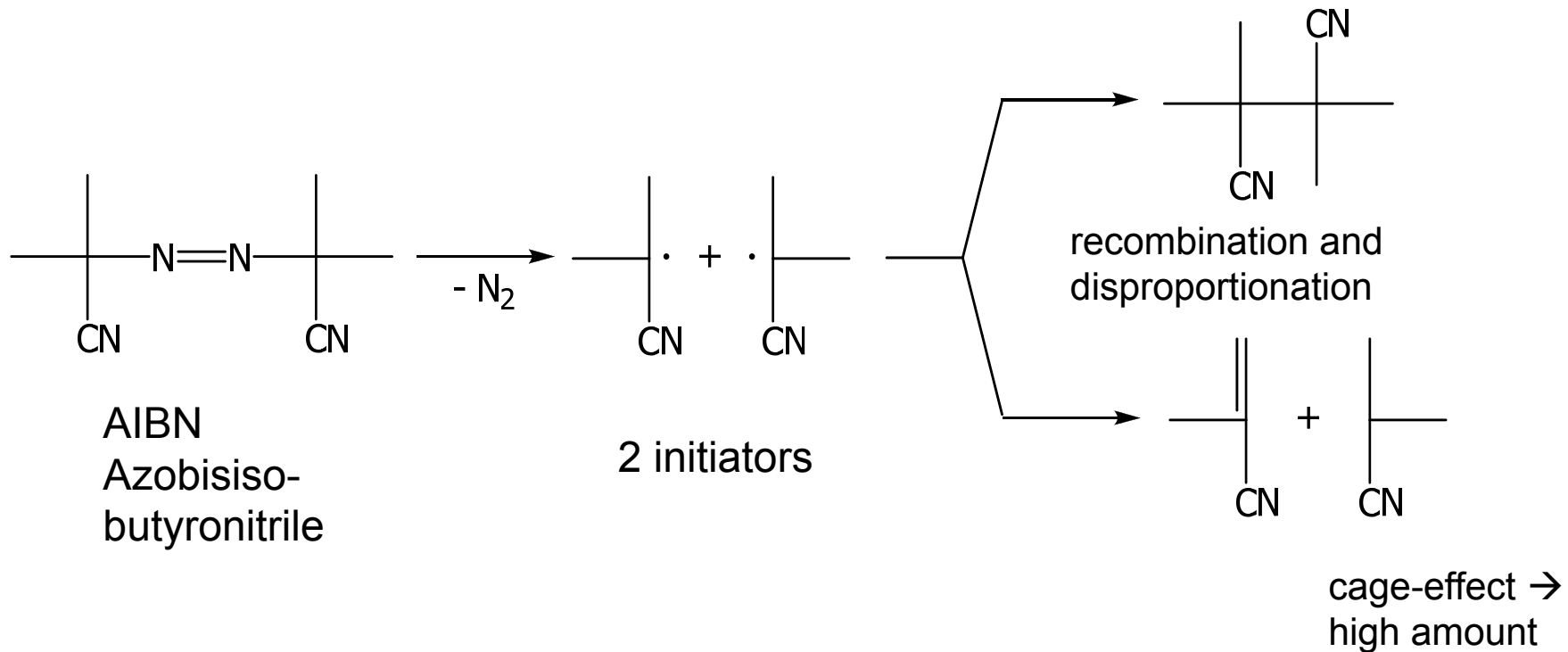
Peroxodisulfate $\text{S}_2\text{O}_8^{2-}$ (potassium, ammonium salt),
reaction mechanism still unclear:

Homolytic cleavage: $\text{S}_2\text{O}_8^{2-} \rightarrow 2 \text{SO}_4^{\cdot-}$ or

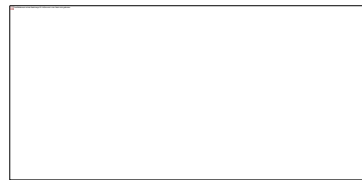
radical formation: $\text{S}_2\text{O}_8^{2-} + \text{H}_2\text{O} \rightarrow \text{SO}_4^{\cdot-} + \text{HSO}_4^- + \text{OH}\cdot$

Both reactions have first order kinetics

Azoinitiators



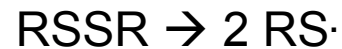
Thermal decomposition or photolysis (360 nm)
 Two radicals are formed
 Also unsymmetrical initiators



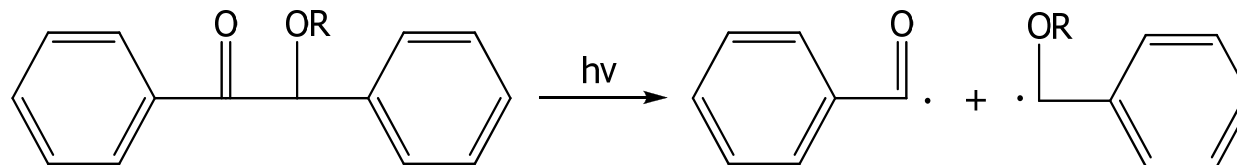
Photoinitiators

Two different types: type I \rightarrow bond cleavage; type II \rightarrow H-abstraction

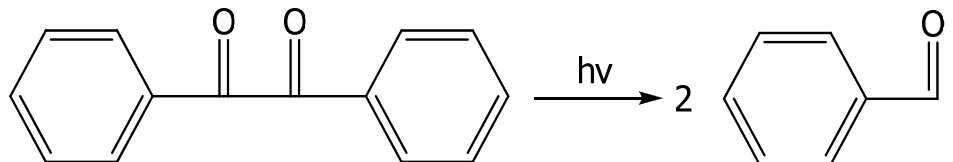
Disulfides (I)



Benzoin and α -ethers (I) (approx. 250 nm)

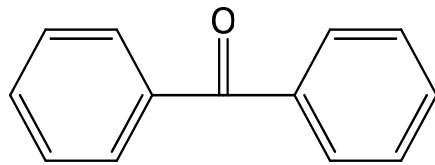


Benzil (approx. 260 nm) (I)

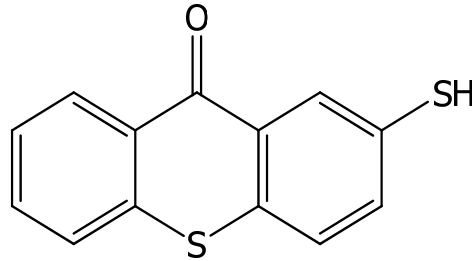


Photoinitiators

Ketones (II)

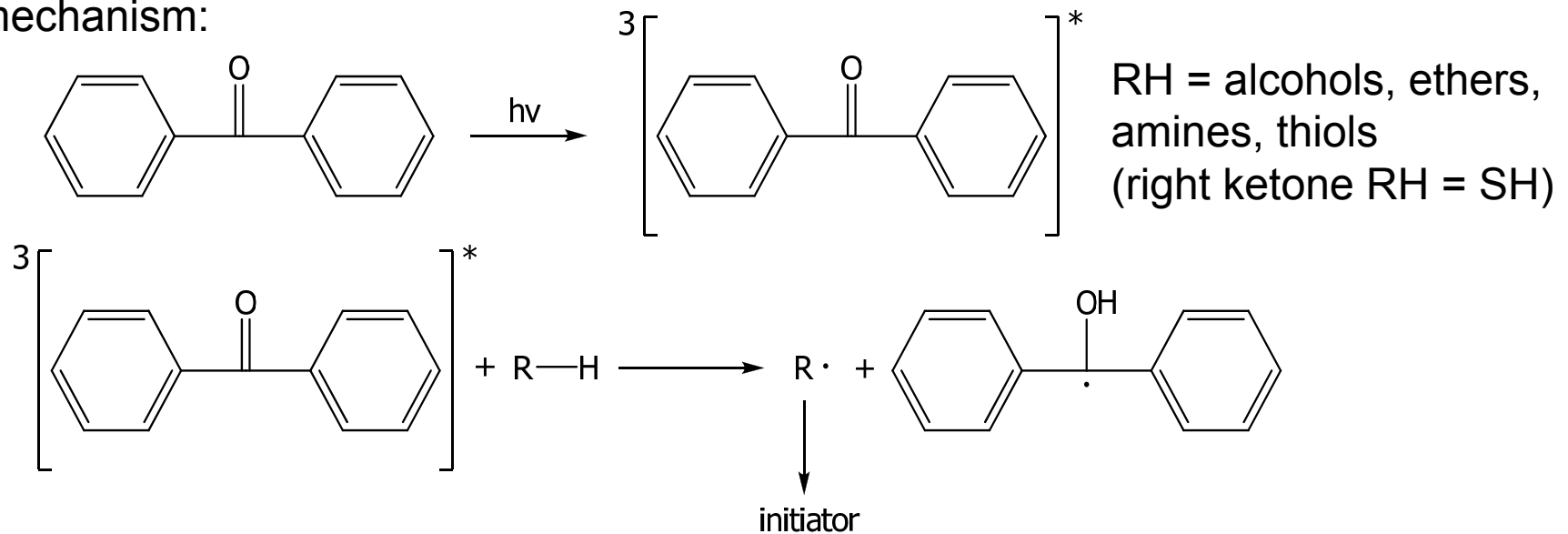


benzophenone
(250 nm)



thioxanthone-thiol
(260 nm)

mechanism:



Self-initiation

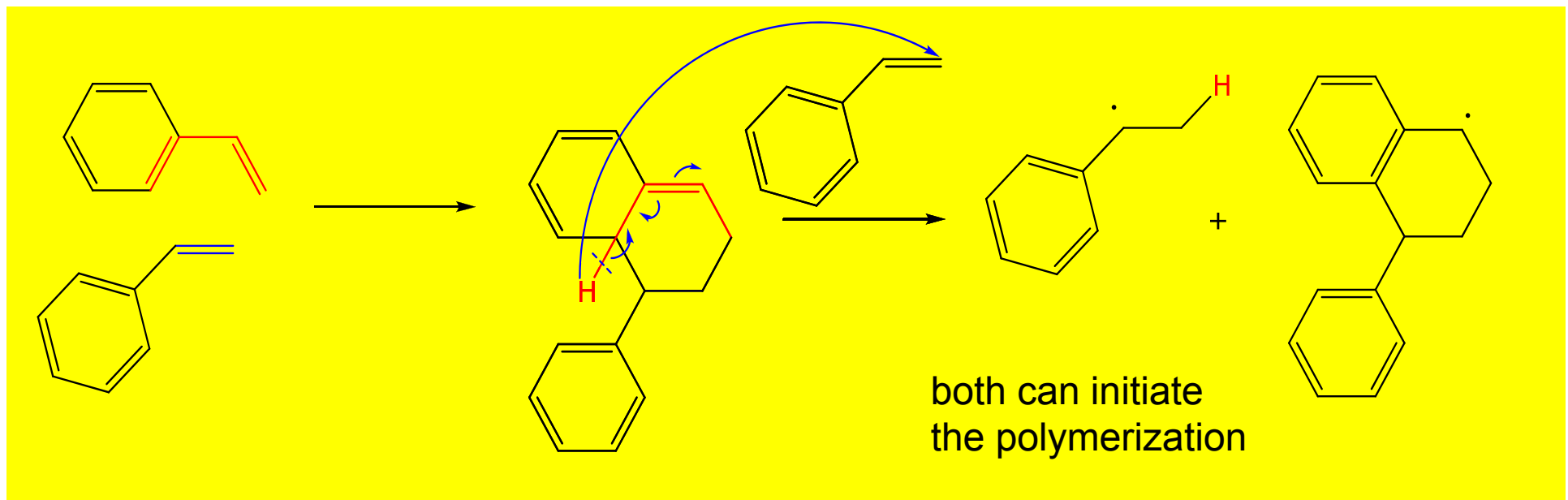
Rare cases, monomers themselves can start reaction

No “spontaneous” polymerization (light, impurities, ...)

e. g. styrene and derivatives, MMA, 2-vinyl-pyridine

Not all self-initiated reactions are radical reactions, but difficult to investigate: MMA conversion of 50% after 5.3 years

Styrene (self-initiated in the dark): 50% conversion of styrene
400 days at 29 °C, 4 hours at 127 °C

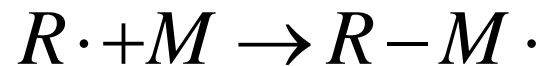


Initiation



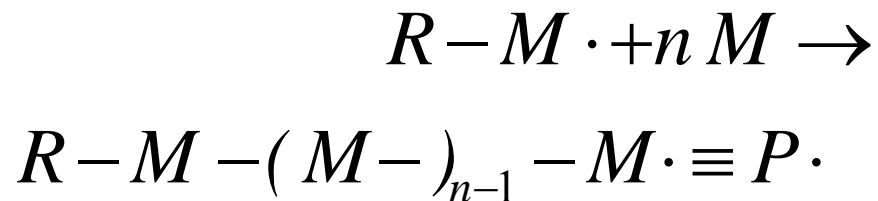
$$v_{init} = k_i [I]$$

Startreaction



$$v_{start} = k_{st} \cdot [R\cdot] = k_{st} \cdot k_i \cdot 2 \cdot f \cdot [I]$$

Growth

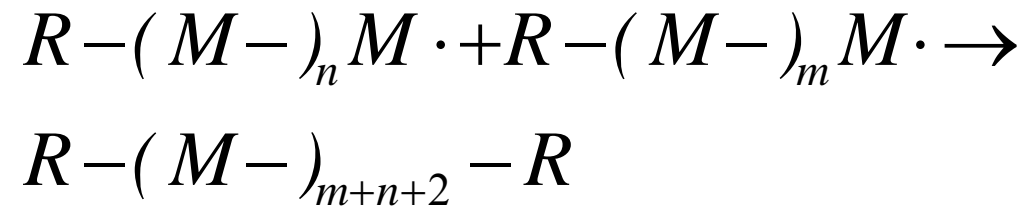


$$v_{growth} = k_{gr} \cdot [P\cdot][M]$$

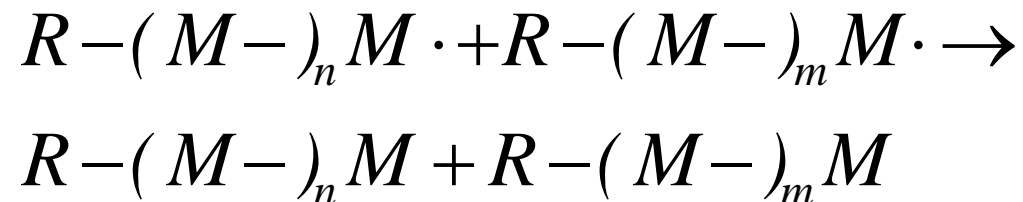
4. Termination

$$v_{term} = k_t \cdot [P\cdot]^2$$

4.1 Coupling



4.2 Disproportionation



**Steady state: formation of new radicals = consumption of radicals
otherwise reaction will stop or explode**

$$\frac{d[R\cdot]}{dt} = 0 = \underbrace{2 \cdot f \cdot k_i \cdot [I]}_{\text{formation}} - \underbrace{k_{st} \cdot [R\cdot] \cdot [M]}_{\text{consumption}}$$

rearrange:

$$[R\cdot] = \frac{2 \cdot f \cdot k_i \cdot [I]}{k_{st} \cdot [M]}$$

growing chains also constant

$$\frac{d[P_1\cdot]}{dt} = 0 = k_{st} \cdot [R\cdot] \cdot [M] - k_{gr,1} \cdot [P_1\cdot] \cdot [M]$$

$$\frac{d[P_2\cdot]}{dt} = 0 = k_{gr,1} \cdot [P_1\cdot] \cdot [M] - k_{gr,2} \cdot [P_2\cdot] \cdot [M]$$

$$\frac{d[P_1\cdot]}{dt} = 0 = k_{st} \cdot [R\cdot] \cdot [M] - k_{gr,1} \cdot [P_1\cdot] \cdot [M]$$

$$\frac{d[P_2\cdot]}{dt} = 0 = k_{gr,1} \cdot [P_1\cdot] \cdot [M] - k_{gr,2} \cdot [P_2\cdot] \cdot [M]$$

all growing chains possess same reactivity \rightarrow all $[P_n\cdot]$ are equal \rightarrow
replacement by $[P\cdot]$

$$\frac{d[P\cdot]}{dt} = 0 = \underbrace{k_{st}}_{\text{startreaction}} \cdot [R\cdot] \cdot [M] - \underbrace{k_t}_{\text{termination}} \cdot [P\cdot]^2$$

rearrange:

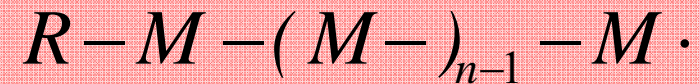
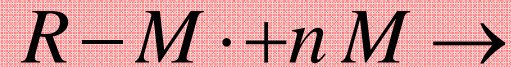
$$[P\cdot] = \sqrt{\frac{k_{st} \cdot [R\cdot] \cdot [M]}{k_t}}$$

monomer concentration decreases with time: $v_{overall} = -\frac{d[M]}{dt}$

monomer is used in the **start reaction** and in the **growth reaction**



$$v_{start} = k_{st} \cdot k_i \cdot 2 \cdot f \cdot [I]$$



$$v_{growth} = k_{gr} \cdot [P\cdot][M]$$

much more monomers react in the growth reaction
 → start reaction can be neglected

$$v_{overall} = -\frac{d[M]}{dt} = k_{gr} \cdot [P\cdot] \cdot [M]$$

$$v_{overall} = -\frac{d[M]}{dt} = k_{gr} \cdot [P\cdot] \cdot [M]$$

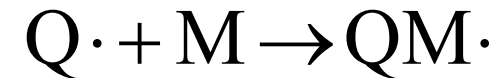
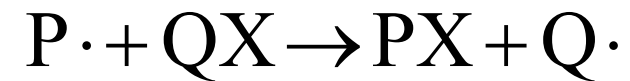
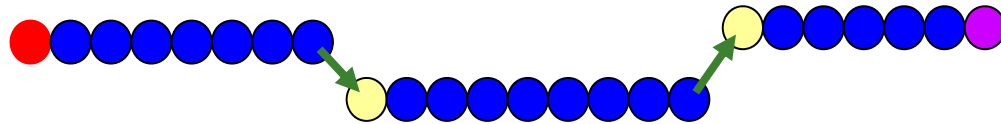
$$[R\cdot] = \frac{2 \cdot f \cdot k_i \cdot [I]}{k_{st} \cdot [M]}$$

$$[P\cdot] = \sqrt{\frac{k_{st} \cdot [R\cdot] \cdot [M]}{k_{ab}}}$$

$$v_{overall} = k_{gr} \cdot \left[\frac{k_{st} \cdot [M]}{k_{ab}} \cdot \frac{2 \cdot f \cdot k_i \cdot [I]}{k_{st} \cdot [M]} \right]^{0.5} \cdot [M]$$

$$v_{overall} = K \cdot \sqrt{[I]} \cdot [M]$$

Chain transfer: Termination of the molecule growth without termination of the kinetic chain



QX can be any molecule:

Monomer

Initiator

Polymer (\rightarrow branching)

Solvent

Chain transfer agents

thiol:



CBr_4 :



$$v_{transfer} = -\frac{d[QX]}{dt} = k_{transfer} \cdot [P\cdot] \cdot [QX]$$

Kinetic chain length:

$$v' = \frac{v_{growth}}{v_{termination} + \sum v_{transfer}}$$

Mayo-equation

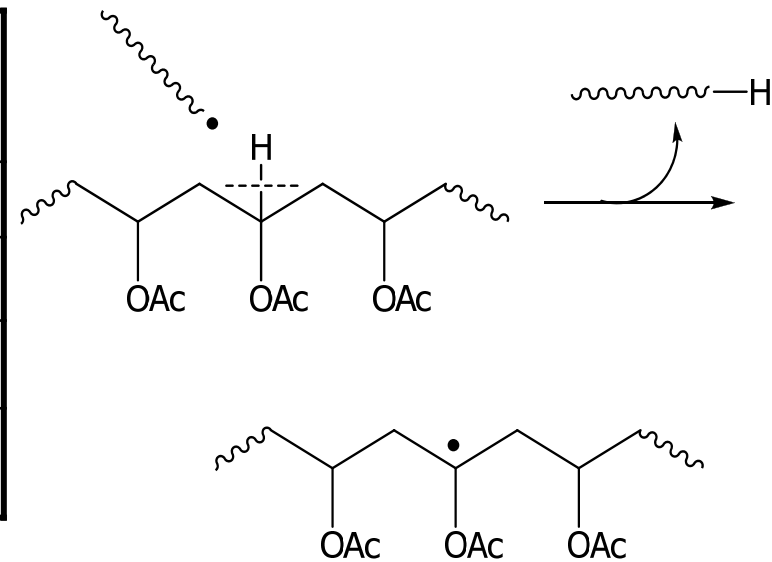
$$\frac{1}{\overline{P}_n} = \frac{1}{\overline{P}_{n,0}} + C_{transfer} \frac{[QX]}{[M]}$$

$$C_{transfer} = \frac{k_{transfer}}{k_{growth}}$$

↑
degree of polymerization
without chain transfer

↑
amount and efficiency of
chain transfer agent

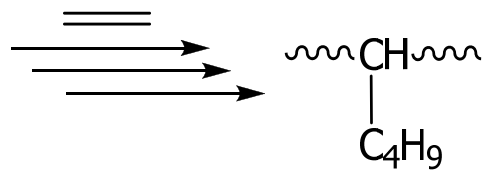
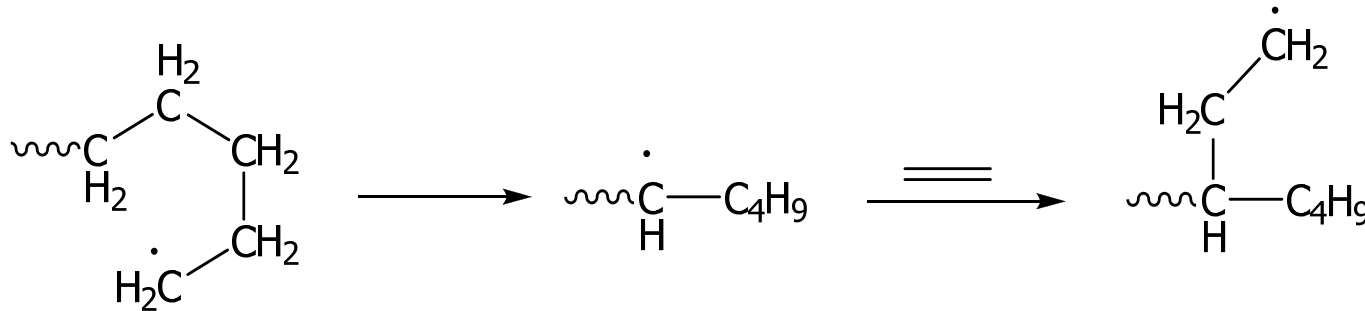
	to monomer $C_{transfer,M} \times 10^5$	to polymer $C_{transfer,P} \times 10^5$
acrylonitrile	2.6	35
MMA	1.0	21
styrene	6.0	19
vinylacetate	20.0	30



intermolecular transfer

Elias.

intramolecular transfer - backbiting



Anzahl / 1000 C-Atome

Methyl - 0

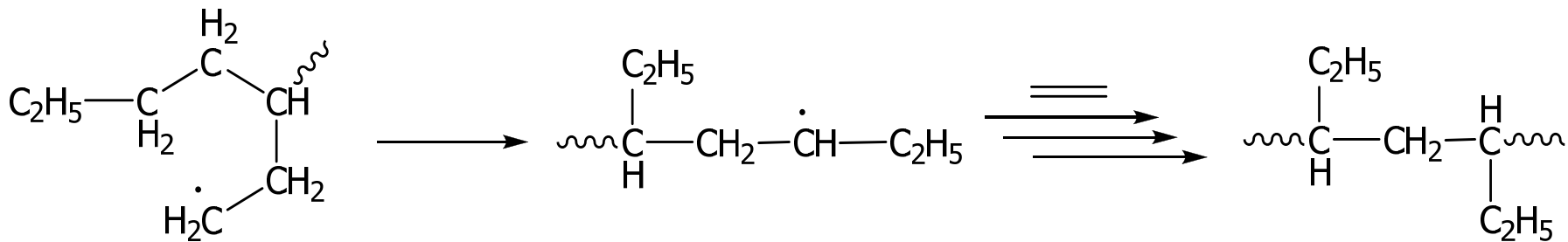
Ethyl - 1

Propyl - 0

Butyl - 9.6

Pentyl - 3.6

Hexyl - 5.6



→ short chain branching

transfer to solvent

	$C_{\text{transfer},M} \times 10^5$
benzene	0.23
toluene	1.6
ethylbenzene	8.3
isopropylbenzene	10.4
<i>tert</i> -butylbenzene	0.6
triphenylmethane	65
butylalcohol	0.6
propargylalcohol	70
allylalcohol	15.0

transfer to modifier

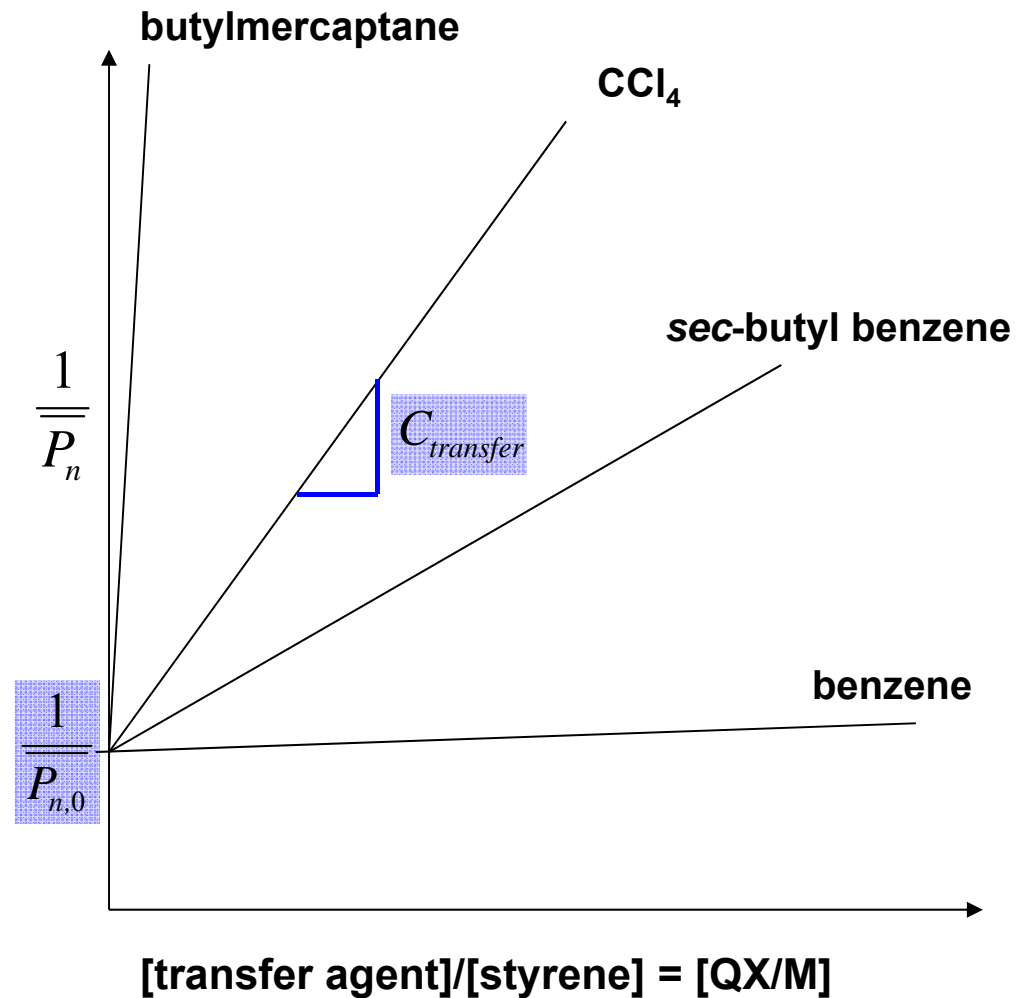
	$C_{\text{transfer},M} \times 10^5$
CCl_4	900
CBr_4	220,000
dodecylmercaptane	1,480,000
butylmercaptane	2,100,000

Chain transfer leads to a decrease of the polymer chain length.
 Large $C_{transfer}$ \rightarrow telomerization
 (only oligomers are formed)

$$\frac{1}{\overline{P}_n} = \frac{1}{\overline{P}_{n,0}} + C_{transfer} \frac{[QX]}{[M]}$$

↑
intercept

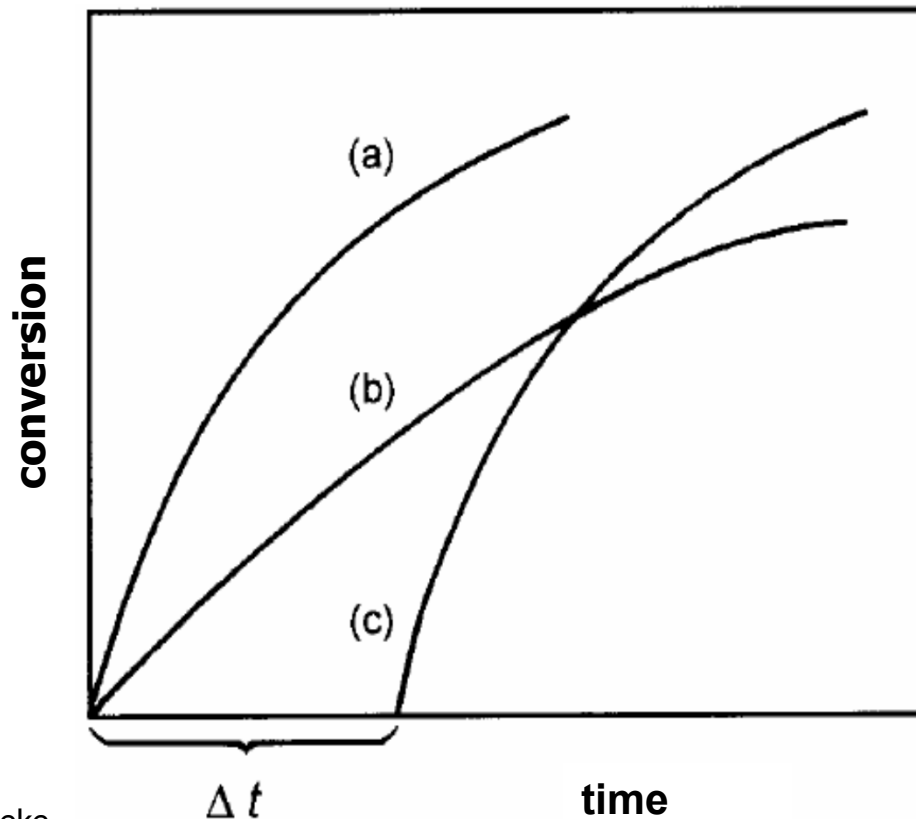
↑
slope



Inhibition = intended chain termination (e. g. storage of monomers)

Inhibitor = substance that suppresses a radical polymerization completely
(formation of stable inactive radicals)

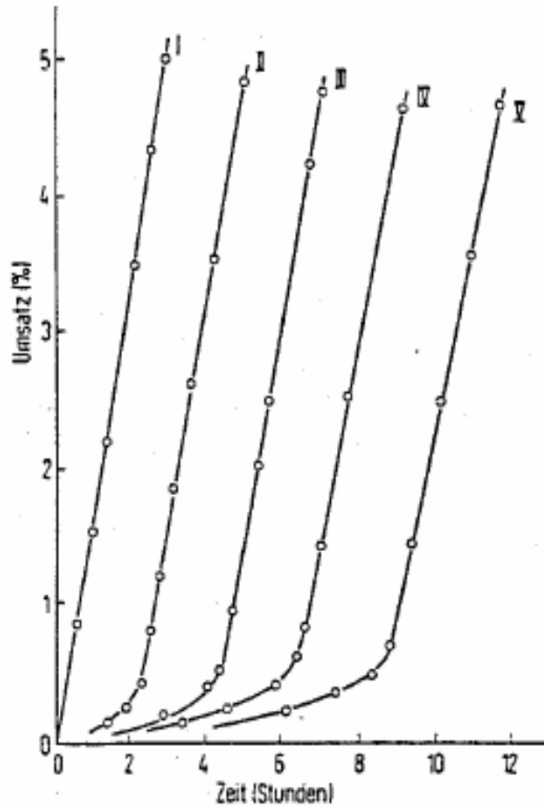
Retarder = substance that decelerates a radical polymerization
(chain transfer, only few reactive radicals, slow transfer)



Radical polymerization
Radical polymerization
in the presence of a **retarder**
→ rate of polymerization is
lowered, lower degree of poly-
merization

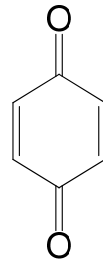
Radical polymerization in the
presence of an **inhibitor**,
polymerization is stopped
until all inhibitor is consumed

Single substance can be inhibitor or retarder depending on the reaction conditions and the used monomer.

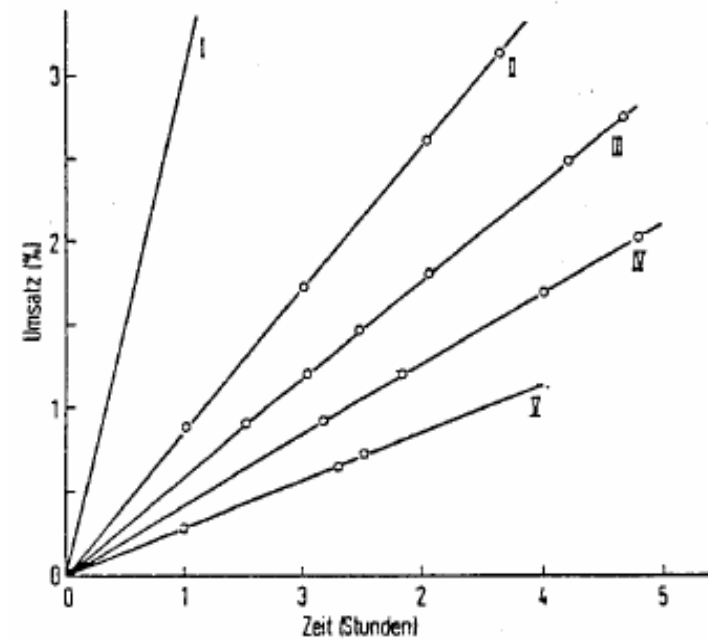


styrene

→ inhibitor



I-V
increasing
concentration

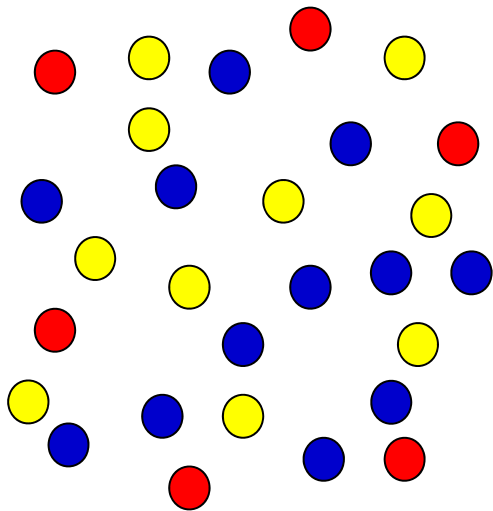


MMA

→ retarder

Copolymerization:

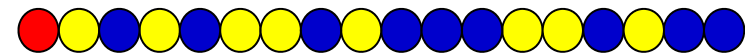
Polymerization of at least two different monomers → copolymers



 **Initiator**

 **Monomer A**

 **Monomer B**

Statistical copolymer

approx. same reactivities of A and B

Alternating copolymer

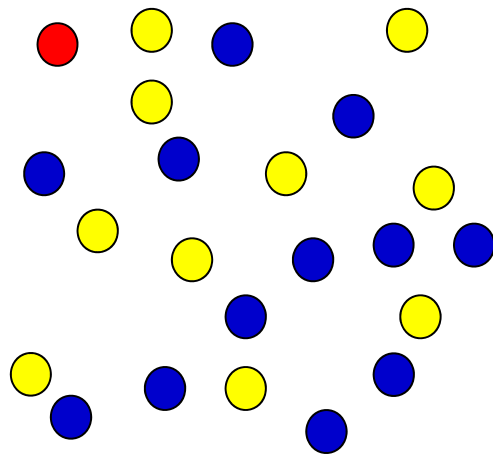
A and B do not homopolymerize or A or B does not homopolymerize, but one monomer is very reactive

Gradient copolymer

B is more reactive than A

Homopolymer(s)

B is much more reactive than A
(unreactive)



● Initiator

● Monomer A

● Monomer B

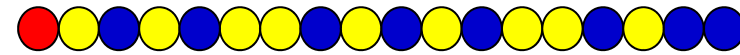
Molar fractions in
monomer feed

$$F_A = \left(\frac{[A]}{[A] + [B]} \right)_{mon}$$

$$F_B = 1 - F_A$$

$$F_A = \frac{12}{10 + 12} = 0.54$$

$$F_B = 0.46$$



Molar fractions in
copolymer

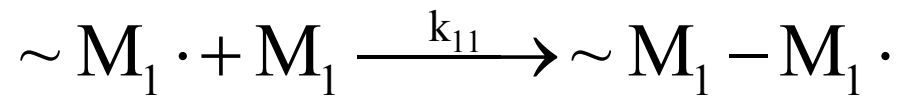
$$f_A = \left(\frac{[A]}{[A] + [B]} \right)_{poly}$$

$$f_B = 1 - f_A$$

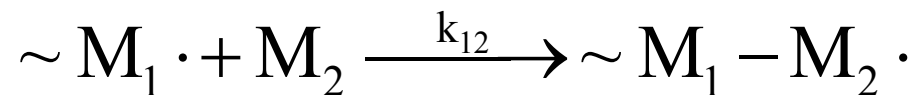
$$f_A = \frac{8}{8 + 9} = 0.47$$

$$f_B = 0.53$$

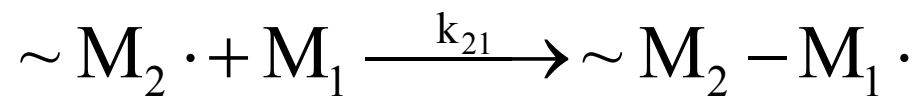
$$f_A = \frac{d[A]_{pol}}{d[A]_{pol} + d[B]_{pol}}$$

Assumptions:**Reactivity independent from chain length****Growth via an active chain end****Start reaction, termination and chain transfer can be neglected****4 different propagation steps****(terminal model – only last monomer unit of the chain influences reactivity)****reaction rates of all steps**

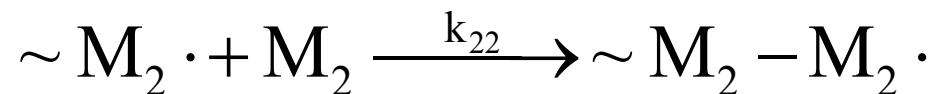
$$v_{11} = k_{11} \cdot [\sim M_1 \cdot] \cdot [M_1]$$



$$v_{12} = k_{12} \cdot [\sim M_1 \cdot] \cdot [M_2]$$



$$v_{21} = k_{21} \cdot [\sim M_2 \cdot] \cdot [M_1]$$



$$v_{22} = k_{22} \cdot [\sim M_2 \cdot] \cdot [M_2]$$

Addition of monomer M₁

$$v_{11} = k_{11} \cdot [\sim M_1 \cdot] \cdot [M_1]$$

$$v_{21} = k_{21} \cdot [\sim M_2 \cdot] \cdot [M_1]$$

$$-\frac{d[M_1]}{dt} = k_{11} \cdot [\sim M_1 \cdot] \cdot [M_1] + k_{21} \cdot [\sim M_2 \cdot] \cdot [M_1]$$

Addition of monomer M₂

$$v_{12} = k_{12} \cdot [\sim M_1 \cdot] \cdot [M_2]$$

$$v_{22} = k_{22} \cdot [\sim M_2 \cdot] \cdot [M_2]$$

$$-\frac{d[M_2]}{dt} = k_{22} \cdot [\sim M_2 \cdot] \cdot [M_2] + k_{12} \cdot [\sim M_1 \cdot] \cdot [M_2]$$

Reactivity ratio of monomers M₁ and M₂

$$\frac{d[M_1]}{d[M_2]} = \frac{k_{11} \cdot [\sim M_1 \cdot] \cdot [M_1] + k_{21} \cdot [\sim M_2 \cdot] \cdot [M_1]}{k_{22} \cdot [\sim M_2 \cdot] \cdot [M_2] + k_{12} \cdot [\sim M_1 \cdot] \cdot [M_2]}$$

$$\frac{d[M_1]}{d[M_2]} = \frac{k_{11} \cdot [\sim M_1 \cdot] \cdot [M_1] + k_{21} \cdot [\sim M_2 \cdot] \cdot [M_1]}{k_{22} \cdot [\sim M_2 \cdot] \cdot [M_2] + k_{12} \cdot [\sim M_1 \cdot] \cdot [M_2]}$$

Assumptions:
 $[\sim M \cdot]$ is constant
 equilibrium: $v_{21} = v_{12}$

Steady state conditions



**Monomer ratio
 in polymer**

$$\frac{m_1}{m_2} = \frac{[M_1]}{[M_2]} \cdot \frac{r_1 \cdot [M_1] + [M_2]}{[M_1] + r_2 \cdot [M_2]}$$

$$r_i = \frac{\text{rate of add. of same M}}{\text{rate of add. of other M}} \Rightarrow r_1 = \frac{k_{11}}{k_{12}} ; r_2 = \frac{k_{22}}{k_{21}}$$

$$r_i = \frac{\text{add. of same M}}{\text{add. of other M}} \Rightarrow r_1 = \frac{k_{11}}{k_{12}} ; r_2 = \frac{k_{22}}{k_{21}}$$

Five different cases for r_i :

- $r_i = 0$** Rate constant for homopolymerizations are zero
→ active chain end adds only the other type of monomer
- $r_i < 1$** The other type of monomer is added preferentially
- $r_i = 1$** Both monomers are added with equal probability
if $[M_1] = [M_2]$
- $r_i > 1$** The same monomer is added preferentially
- $r_i = \infty$** Only homopolymerization, no copolymerization

M_1	r_1	M_2	r_2
tetrafluoroethylene	1	chlortrifluoroethylene	1
ethylene	0.88	vinylacetate	1.03
styrene	56	vinylacetate	0.01
styrene	17.24	vinylchloride	0.058
styrene	0.84	butadiene	1.44
vinylchloride	0.04	butadiene	8.8
acrylonitrile	0.04	butadiene	0.36
methylmethacrylate	0.027	butadiene	0.5
methylmethacrylate	0.41	styrene	0.48
acrylonitrile	1.68	butylacrylate	1.06
stilbene	0.03	maleic acid anhydride	0.03

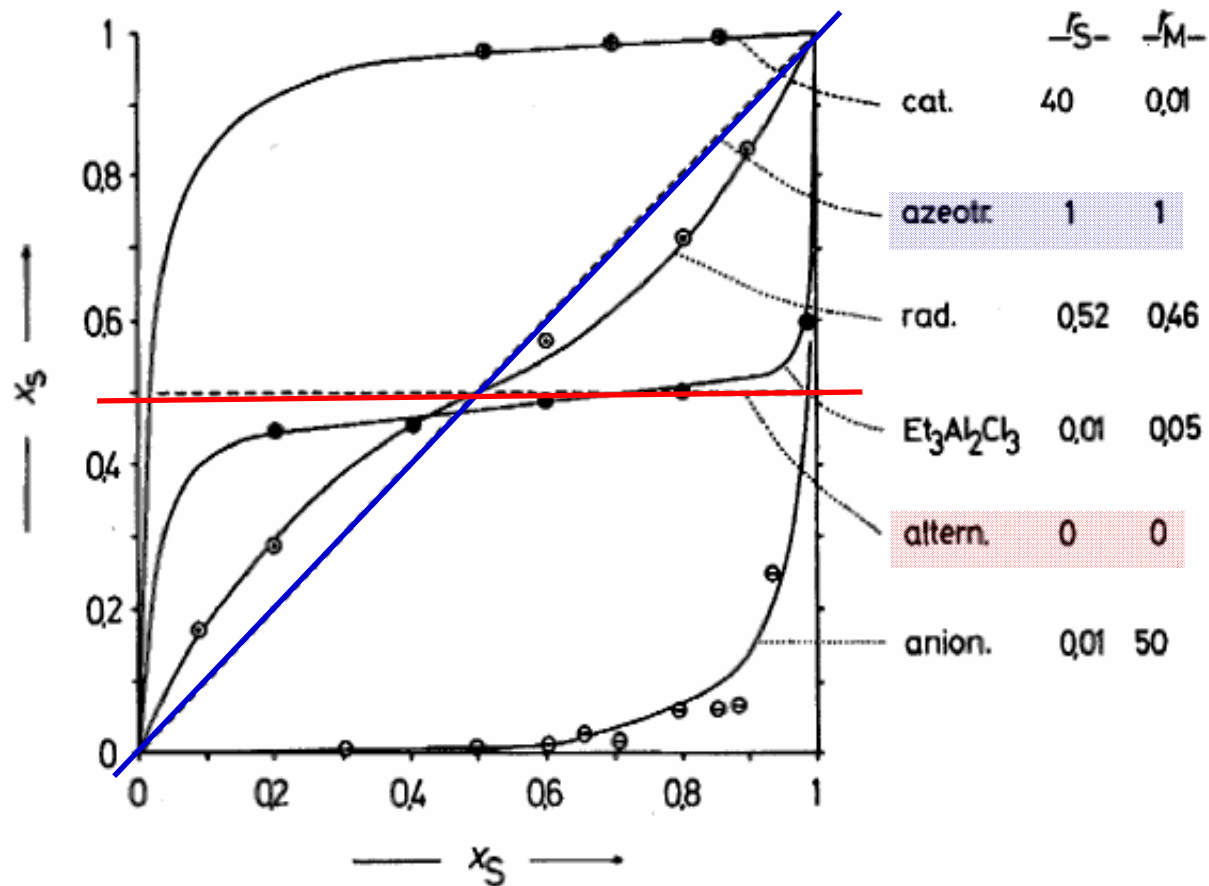
$r_1 \approx r_2; r_1 \cdot r_2 = 1 \Rightarrow$ random copolymerization, ideal copolymerization

$r_1 > 1, r_2 < 1; r_1 \cdot r_2 = 1 \Rightarrow$ composition drift, ideal copolymerization

$r_1 > 1, r_2 < 1; r_1 \cdot r_2 > 1 \Rightarrow$ composition drift, deviation from ideal copolymerization

$r_1 = r_2 = 0; r_1 \cdot r_2 = 0 \Rightarrow$ strictly alternating copolymer

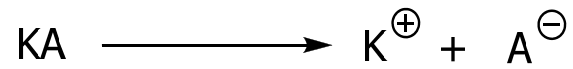
$r_1 < 1, r_2 < 1; 0 < r_1 \cdot r_2 < 1 \Rightarrow$ more or less alternating copolymer



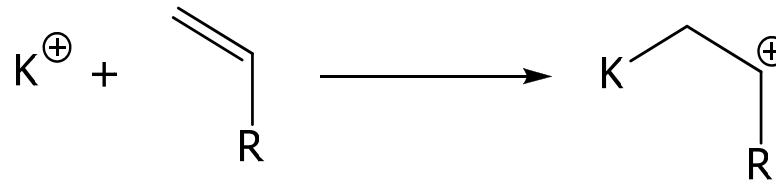
Copolymerization of styrene (S) and methylmethacrylat (M) with different initiators (cationic, radical, anionic, coordinative polymerization) (x_s = ratio of styrene in polymer vs. x_s = ratio of styrene in monomer mixture)

Cowie.

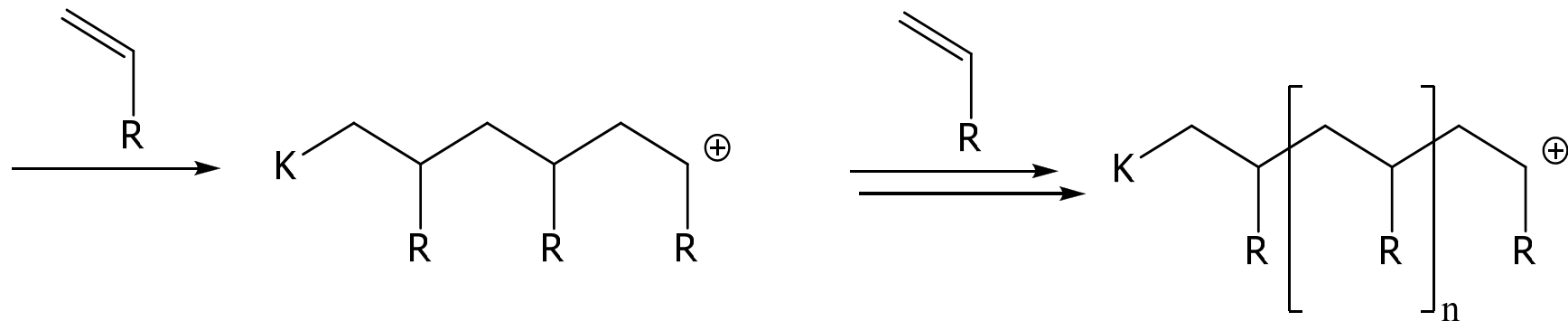
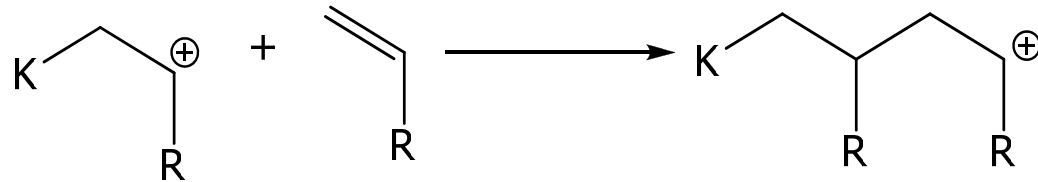
1) Initiation



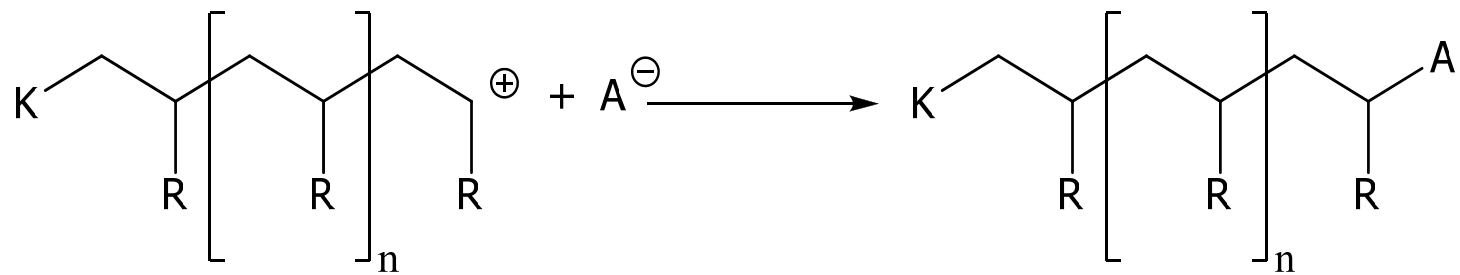
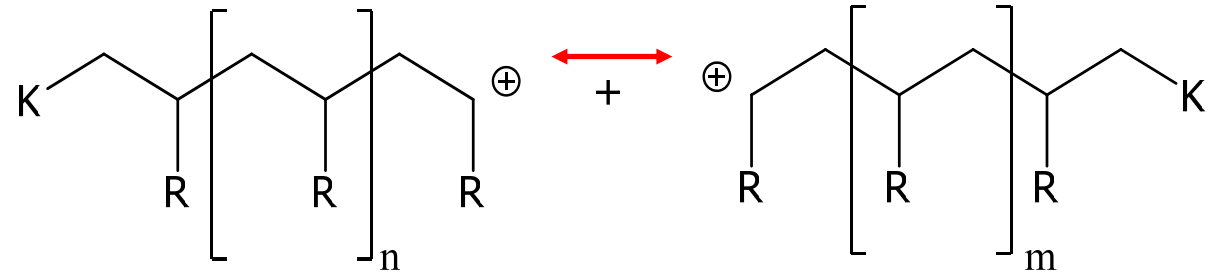
2) Start reaction



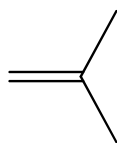
3) Growth reaction



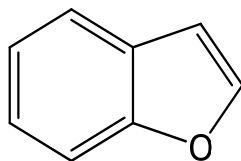
4) Termination



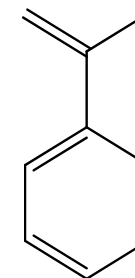
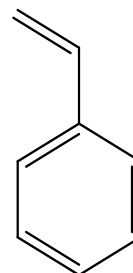
+ many other reactions



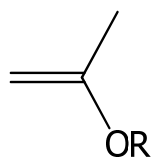
isobutene



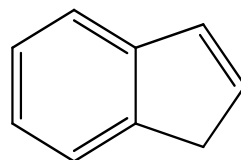
benzofurane



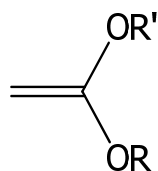
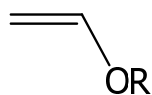
styrene derivatives and
 α -methylstyrene derivatives



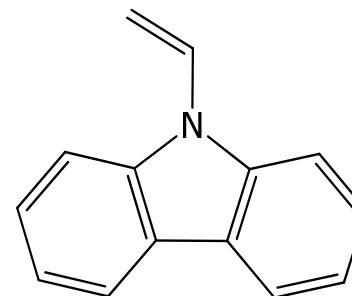
vinylethers



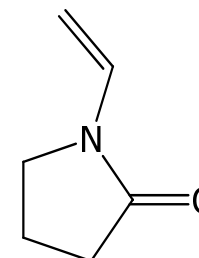
indene



vinylacetals



vinylcarbazole



N-vinylpyrrolidone

Monomers are electronrich and stable against deprotonation.

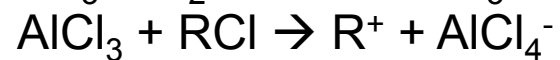
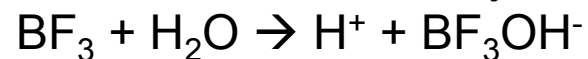
➤ **Protic acids**

H_2SO_4 , HClO_4 , FSO_3H , $\text{F}_3\text{CSO}_3\text{H}$, F_3CCOOH → sterically demanding anion

➤ **Lewis acids**

BF_3 , AlCl_3 , AlBr_3 , SnCl_4 , SbF_5 , SbCl_5 , TiCl_4

Coinitiator is necessary:

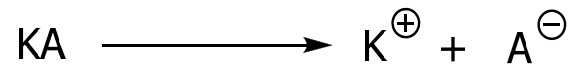


higher molar masses

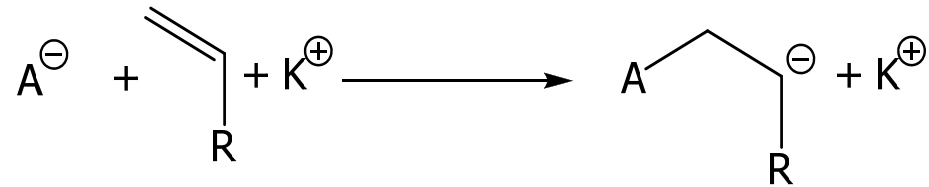
acidity of cocatalyst determines reaction rate:

acetic acid > nitroethane > phenol > water

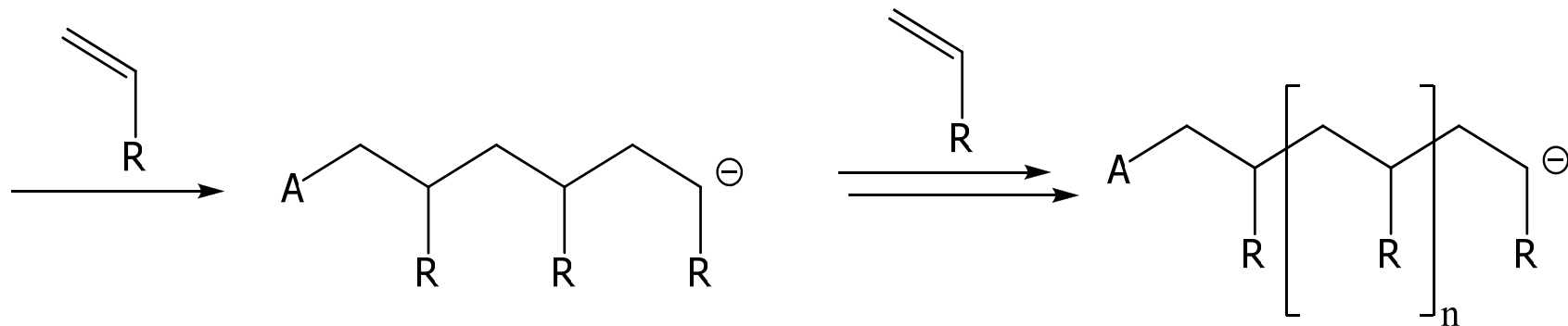
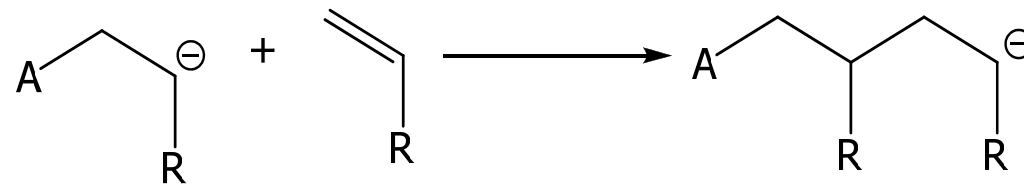
1) Initiation



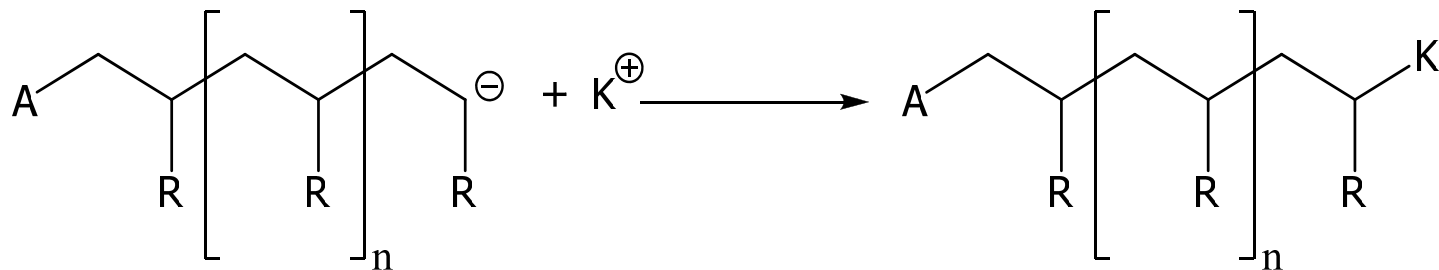
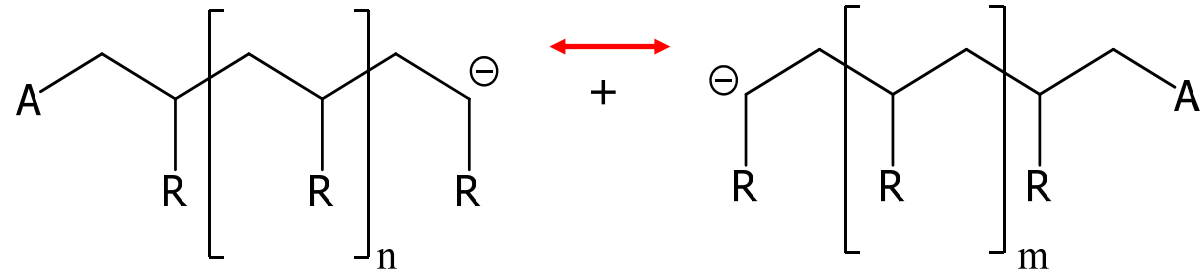
2) Start reaction



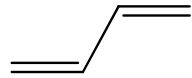
3) Growth reaction



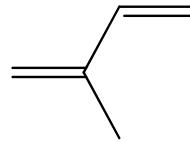
4) Termination



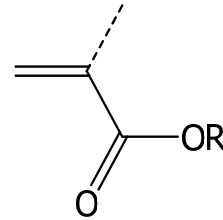
+ many other reactions



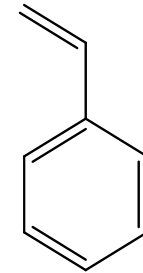
butadiene



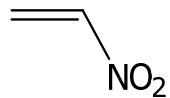
isoprene



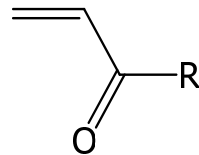
acrylates and
methacrylates



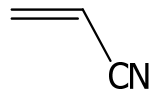
styrene



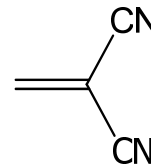
nitroethene



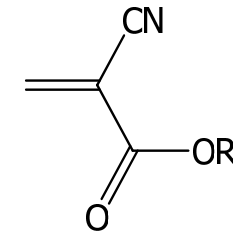
vinylketons



acrylonitrile

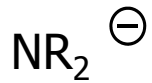
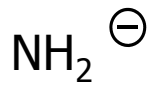


vinylidene cyanide

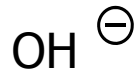


cyanoacrylate

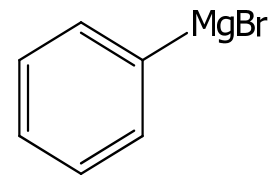
Monomers are electronpoor and have electronwithdrawing substituents.



metal amides



“normal”
anions



Grignard
compounds

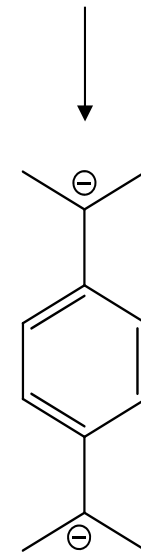
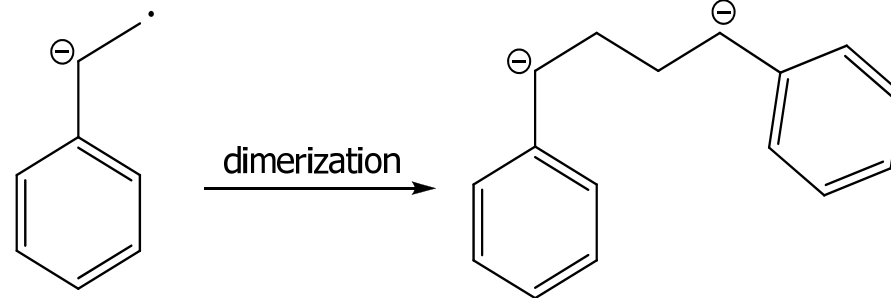
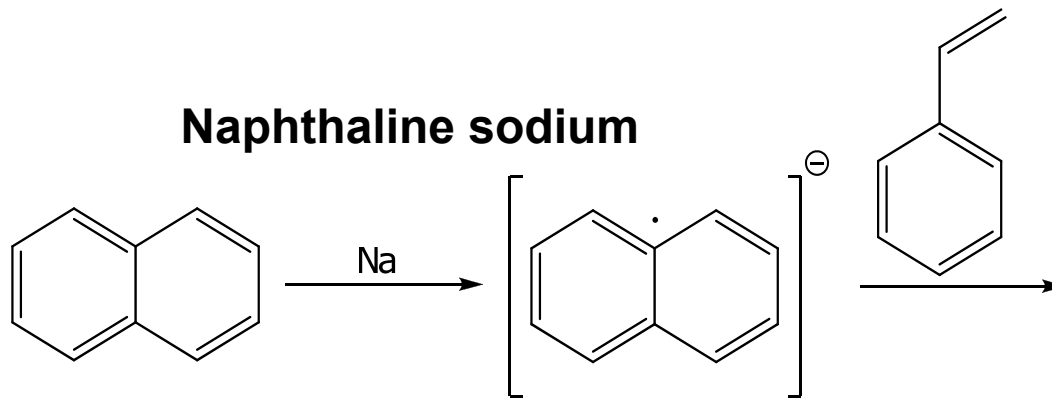
n-BuLi
sec-BuLi

Lithium
compounds



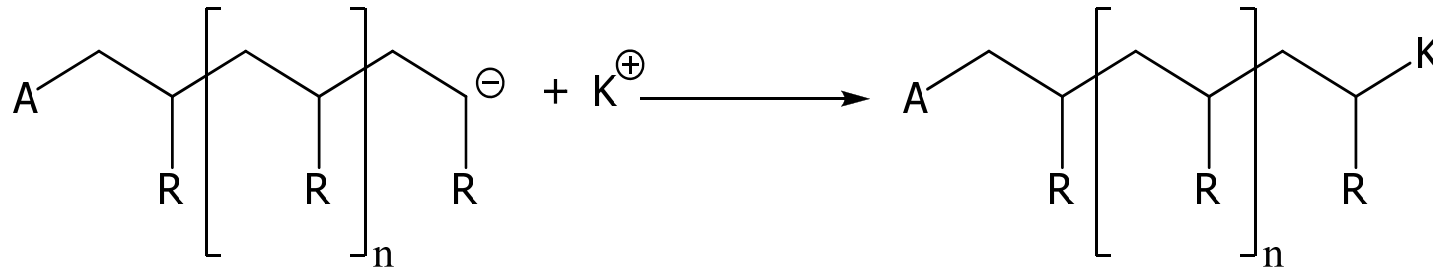
alkoxides

Naphthalene sodium

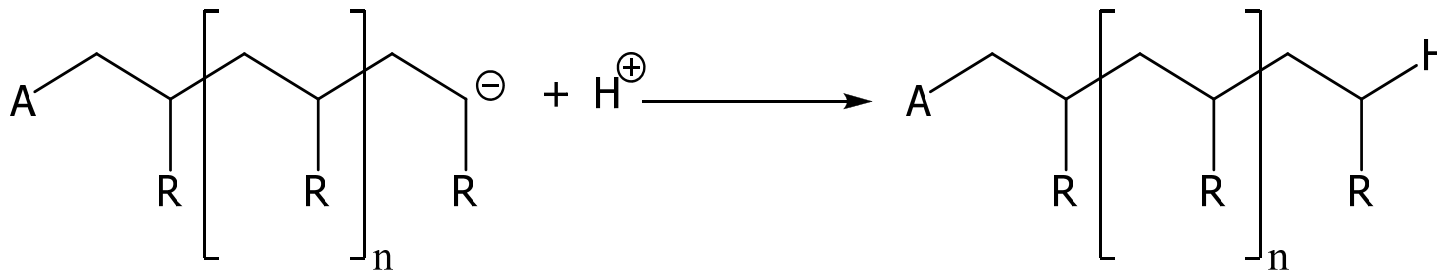


Ideal anionic polymerization (“living polymerization”) → no termination

Reaction of anion with counter ion



Anion (base) reacts with proton



Special considerations for experimental work

high nucleophilicity of the initiators (and propagating chain ends)
absolutely necessary to **avoid oxygen, water and protic impurities**

aprotic solvents (dry)

polar: THF, diethyl ether, ...

non-polar: toluene, cyclohexane, hexane, ...

purification of all reagents

handling of reagents in vacuum or under inert gas

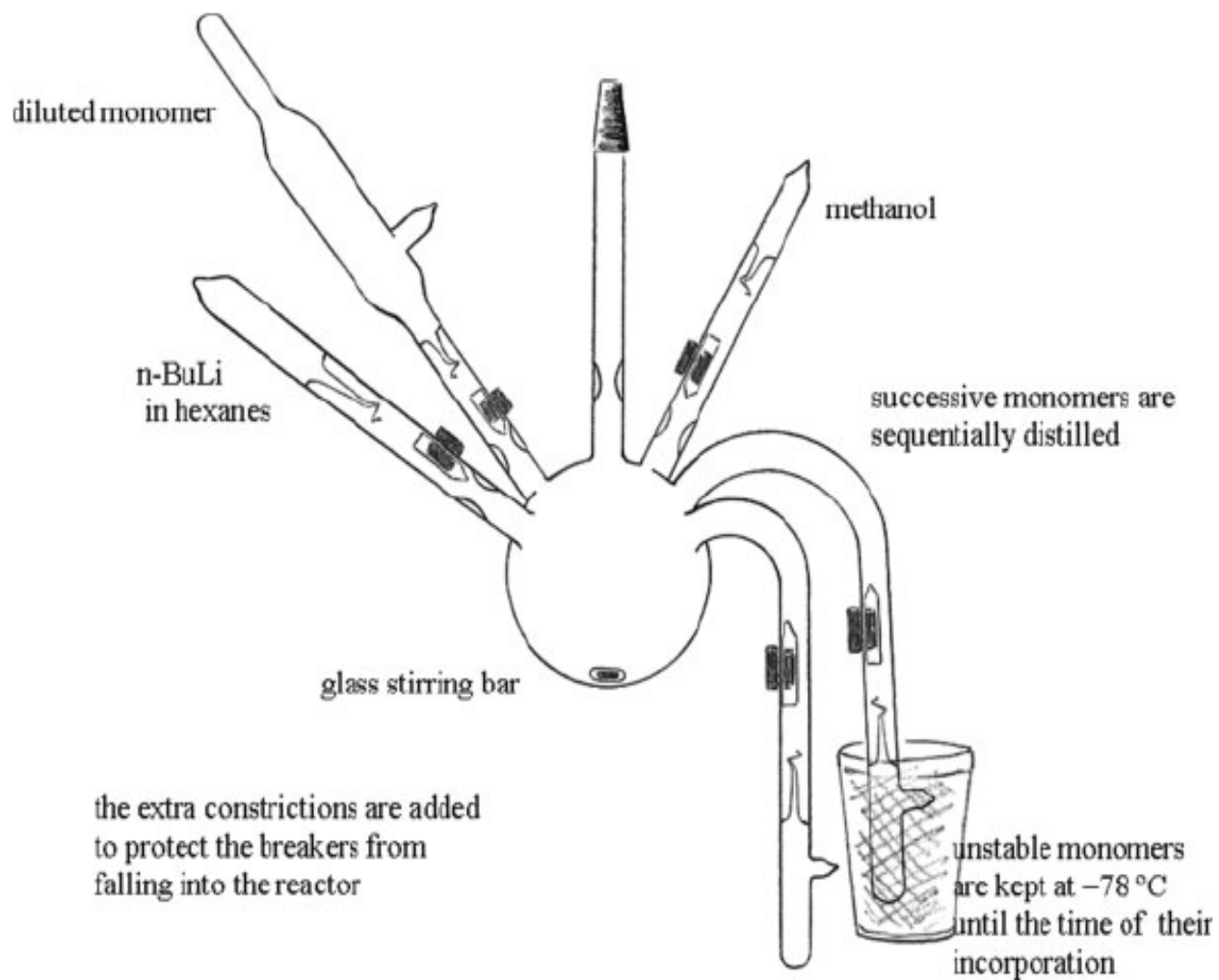
no termination → high concentration of active species → very high reaction rate

(e.g. the half of the amount of MMA reacts in less than 1 s in THF at room temperature)

→ polymerization at very **low temperatures**

„The high-vacuum technique employed in classical anionic polymerization methodology requires highly developed and specialized skills. The polymer scientist must possess both an adept glassblowing ability and a thorough understanding of the behavior of compounds, volatile and nonvolatile, in a vacuum environment.“

J. Polym. Sci., Part A., Polym. Chem. **2005**, *43*, 6179–6222.



N. Hadjichristidis, University of Athens

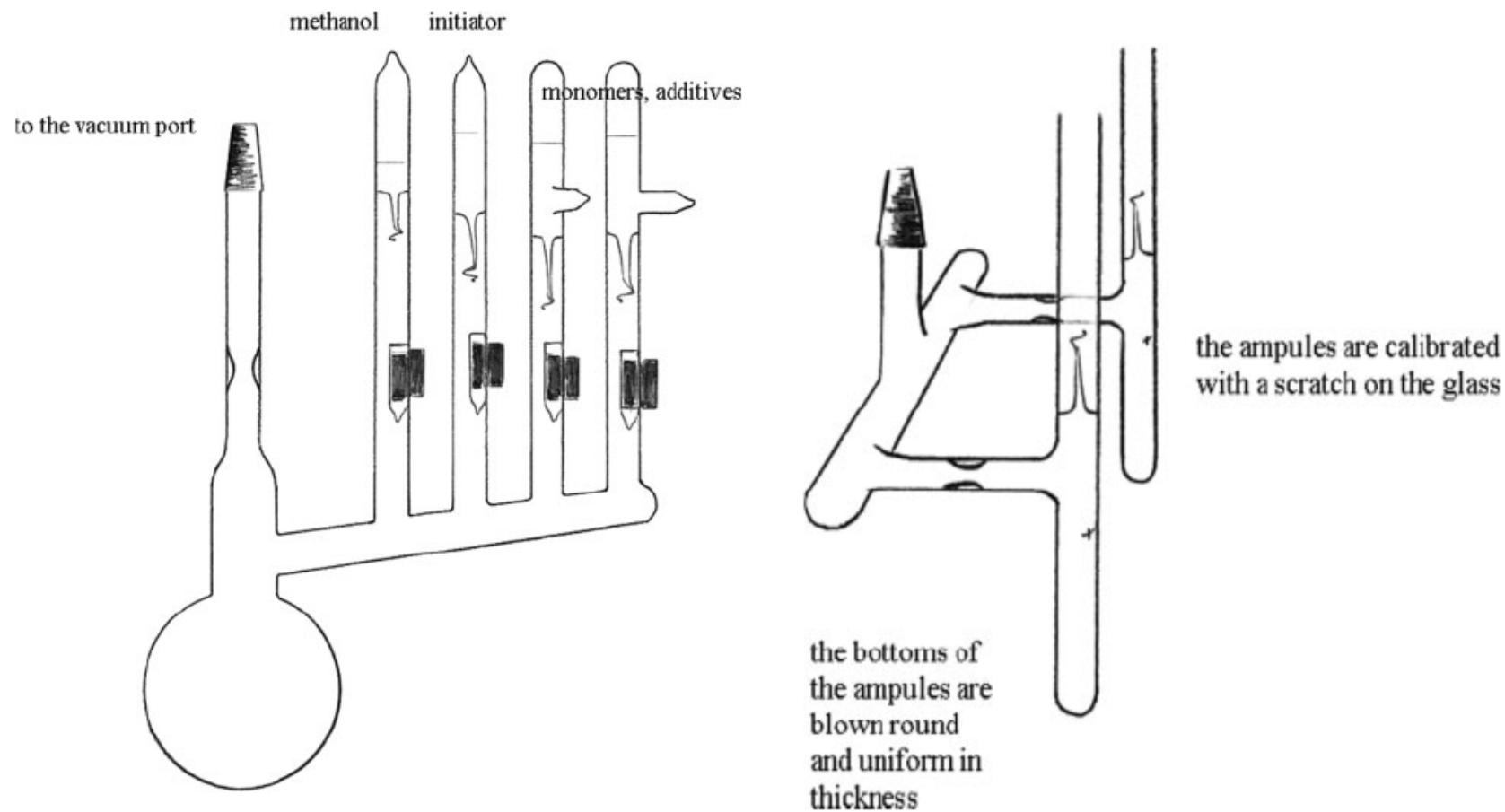
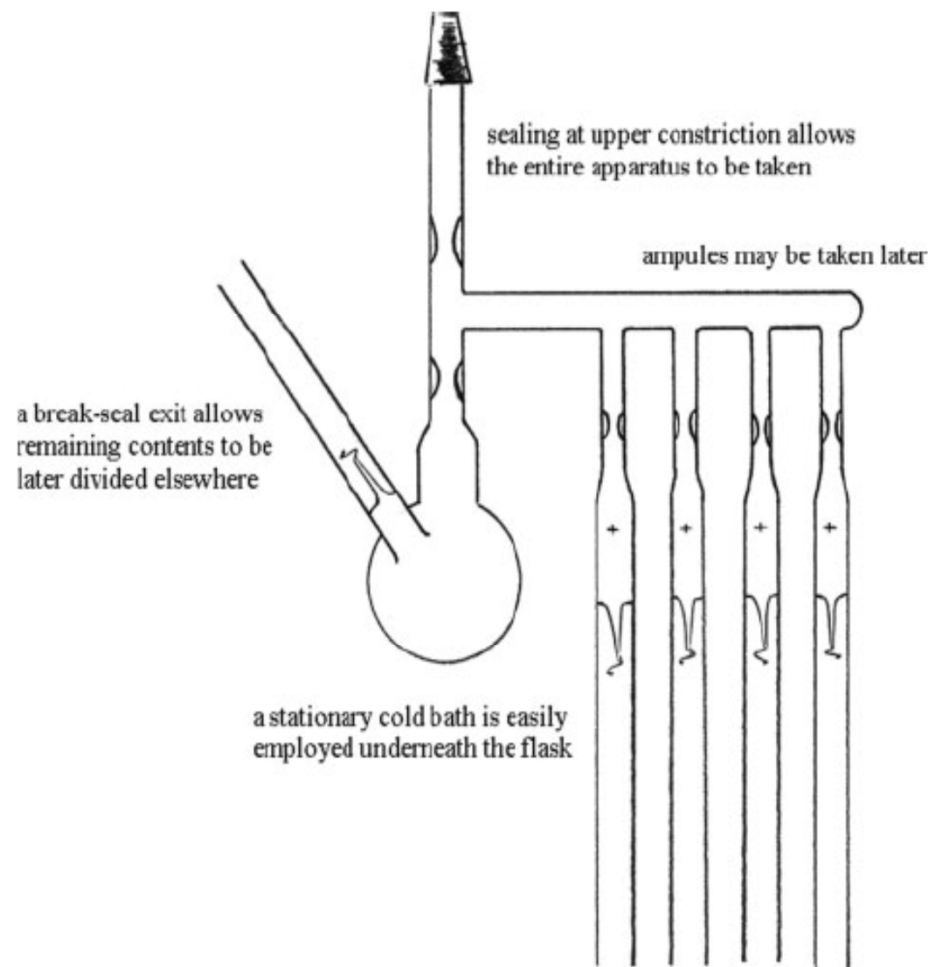
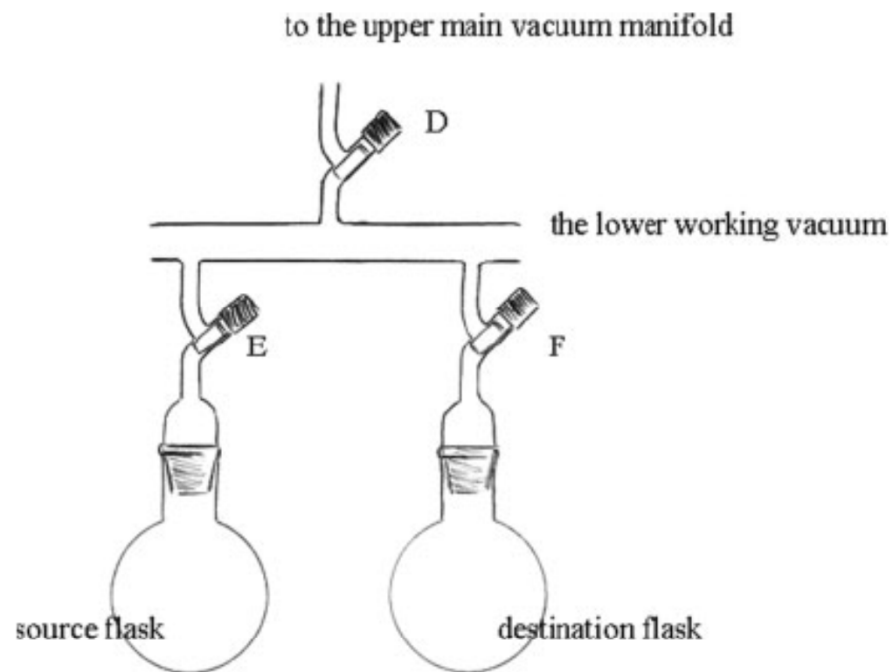
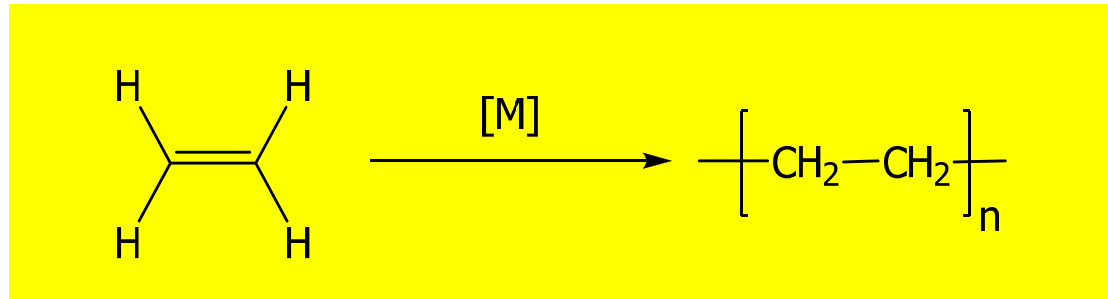


Figure 46. Simple polymerization apparatus.

N. Hadjichristidis, university of Athens





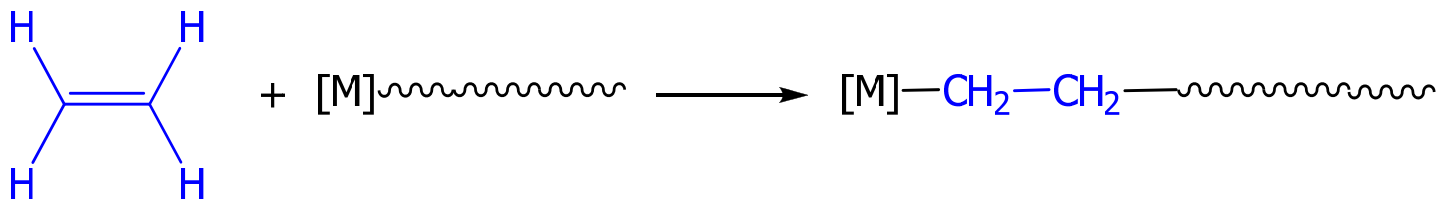
History:

1953 Karl Ziegler and his coworkers discovered the polymerization of ethene with triethylaluminium and nickel complexes

1954 Giulio Natta discovered the stereoregular polymerization of propylene

1963: Nobel prize for K. Ziegler and G. Natta

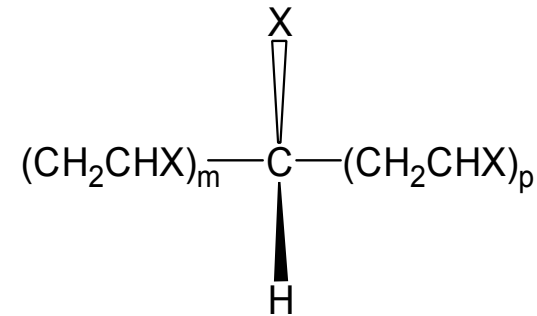
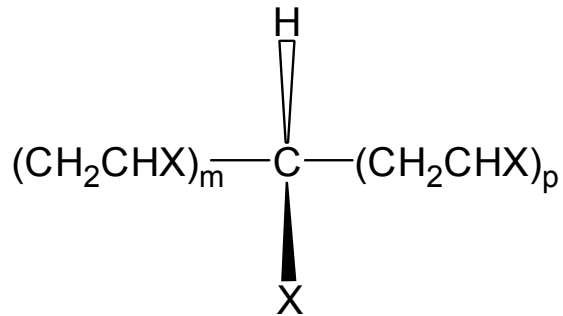
also: polyinsertion



Stereoisomerism

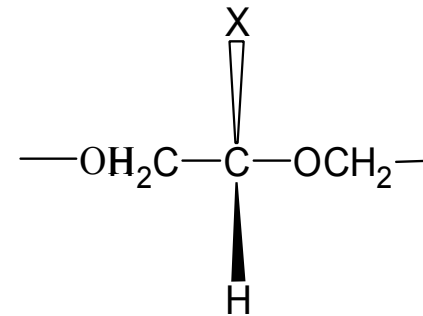
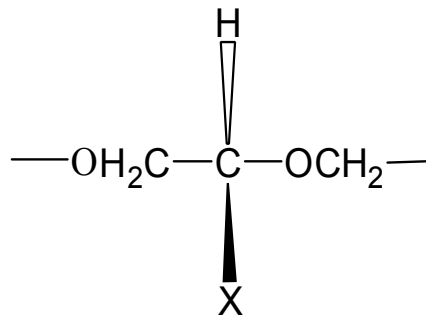
- configuration of monomers (stereoisomers)
- configurational isomers → high energy barrier
- conformational isomers → low energy barrier

two different configurations



asymmetric, but usually no optical activity

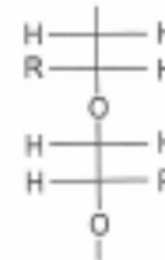
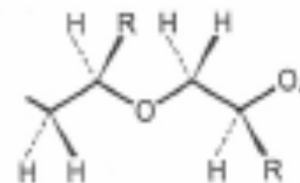
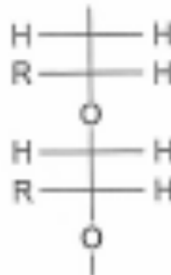
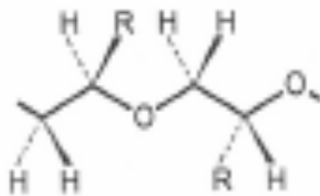
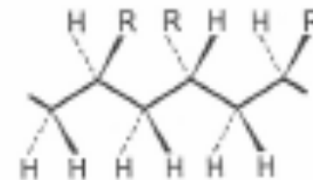
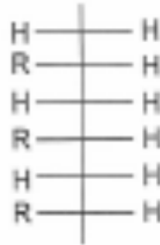
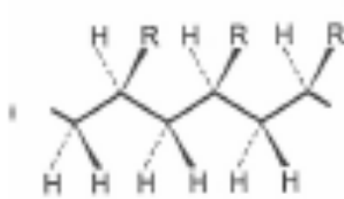
heteroatoms (main chain) → absolute configuration → optical activity



Tacticity: steric order of R-substitution along the main chain
 → Fischer projection

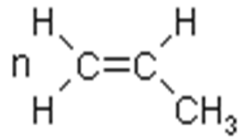
When all substituents are
 on one side → isotactic

alternating → syndiotactic

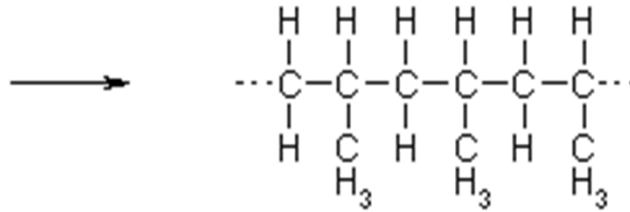


irregular polymers → atactic

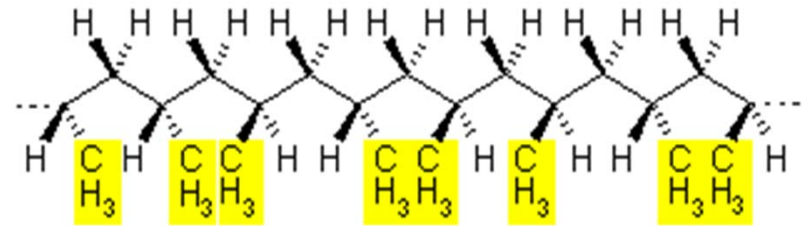
Example for tacticity: poly(propylene)



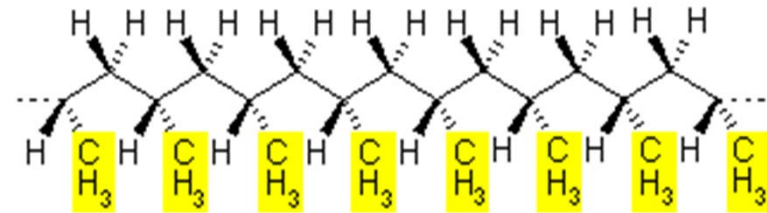
propene



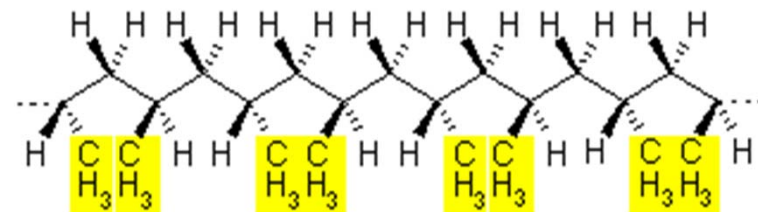
poly(propylene)



atactic poly(propylene)



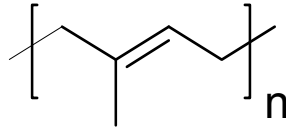
isotactic poly(propylene)



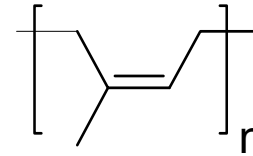
syndiotactic poly(propylene)

- atactic: no long-range order, amorphous
- isotactic: helical, crystalline; best mechanical properties
- syndiotactic: crystalline, zig-zag conformation
- melting point: at < st < it
120 < 130 < 165 °C
- crystallinity: st < it
- industry: mainly isotactic PP

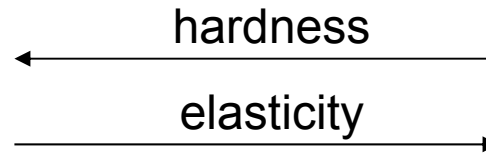
cis – trans isomers:
poly(isoprene)



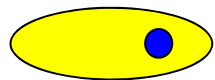
trans: guttapercha



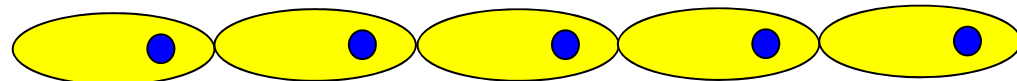
cis: natural rubber



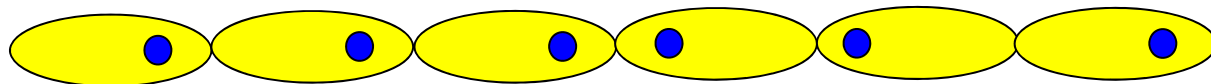
head-tail/head-head and tail-tail-polymers:
Vinyl polymers



tail head



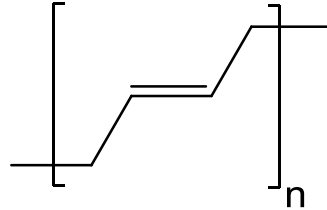
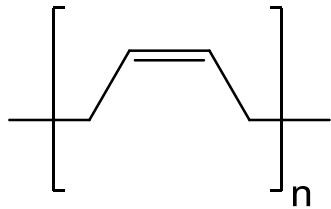
head-tail



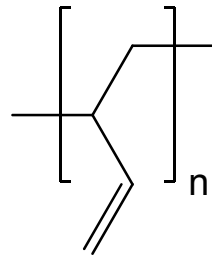
head-head

tail-tail

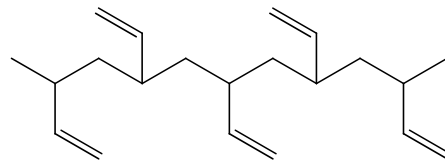
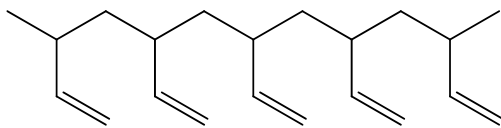
Geometric isomerism - poly(butadiene)



cis-1,4-poly(butadiene) (a) *trans*-1,4-poly(butadiene) (b)

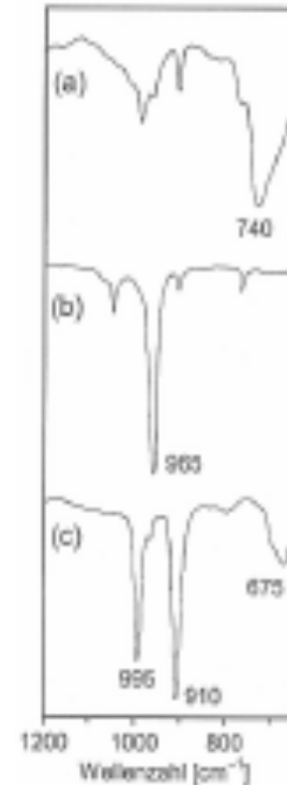


1,2-poly(butadiene) (c)



isotactic

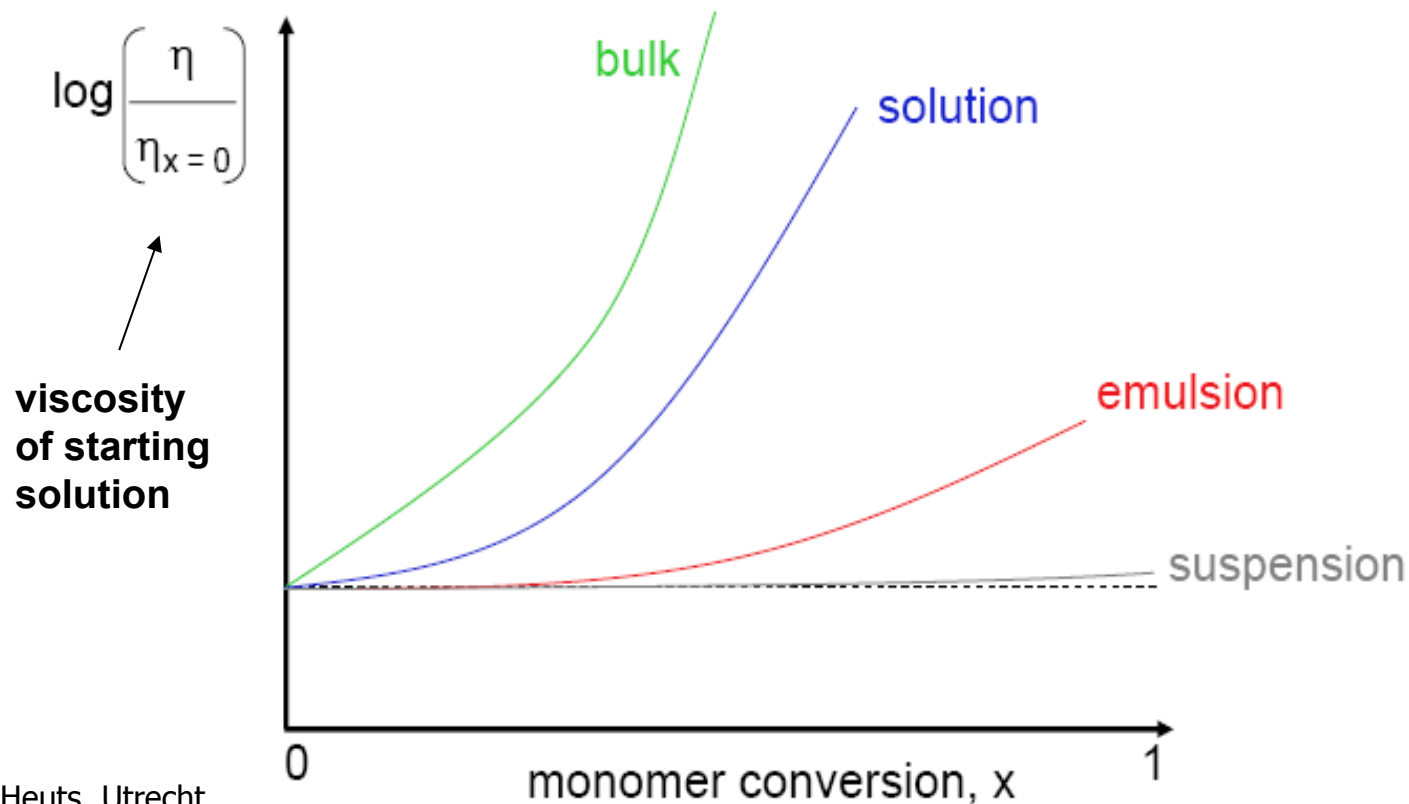
syndiotactic 1,2-poly(butadiene)



out-of-plane CH vibrations

Homogeneous: **Bulk:** With monomer only
Solution: In a solvent

Heterogeneous: **Emulsion:** With monomer dispersed in an aqueous phase
Suspension: In a solvent
[Gas phase: In gaseous monomer, solid catalyst]



Polymerization of the monomer without any solvent
Polymer is dissolved in the monomer
Low conversions → residual monomer is stripped off

Advantages:	Disadvantages:
<ul style="list-style-type: none">➤ Pure products, freedom from contaminations → high optical clarity➤ No solvent➤ Polymerization → endform (e.g. sheets)	<ul style="list-style-type: none">➤ Highly increasing viscosity → difficult to stir → difficult heat removal➤ Very broad molar mass distribution➤ Removal of residual monomer is necessary➤ Viscous polymer/monomer solution is difficult to handle

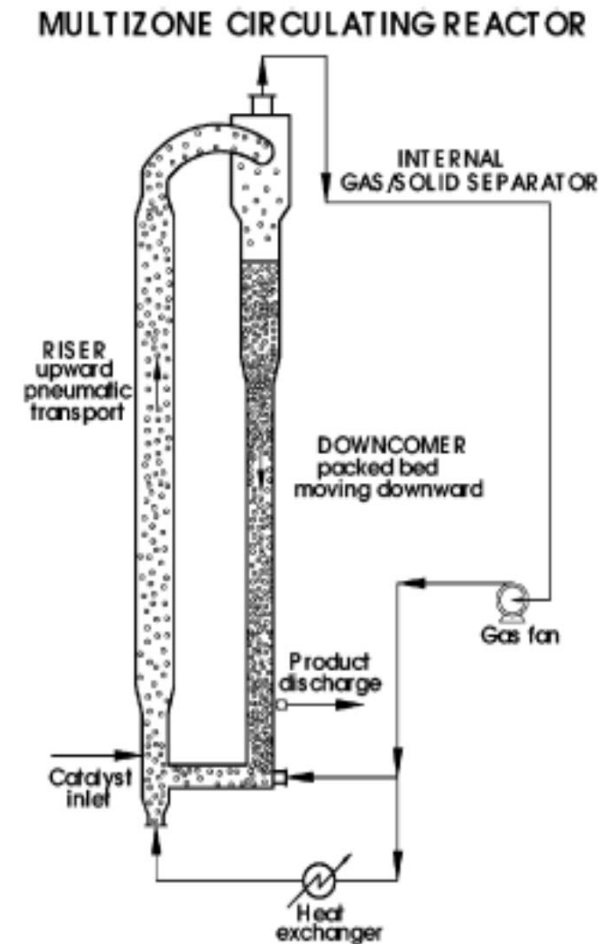
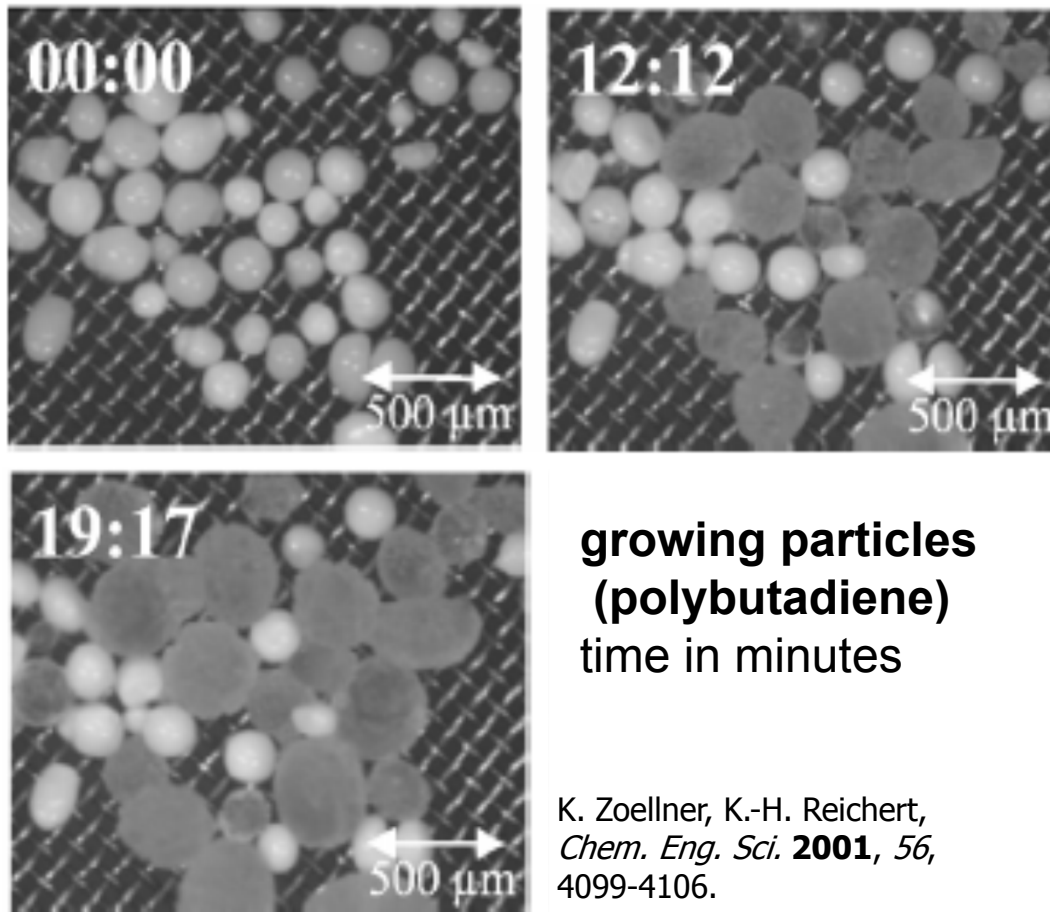
Used for polystyrene, poly(vinyl chloride), PMMA
PMMA sheets: Unstirred reaction: Prepolymer (lower molar mass)
main polymerization to higher molar mass
→ better heat control

Polymerization of the monomer in the presence of a solvent
Polymer and monomer are dissolved in the solvent

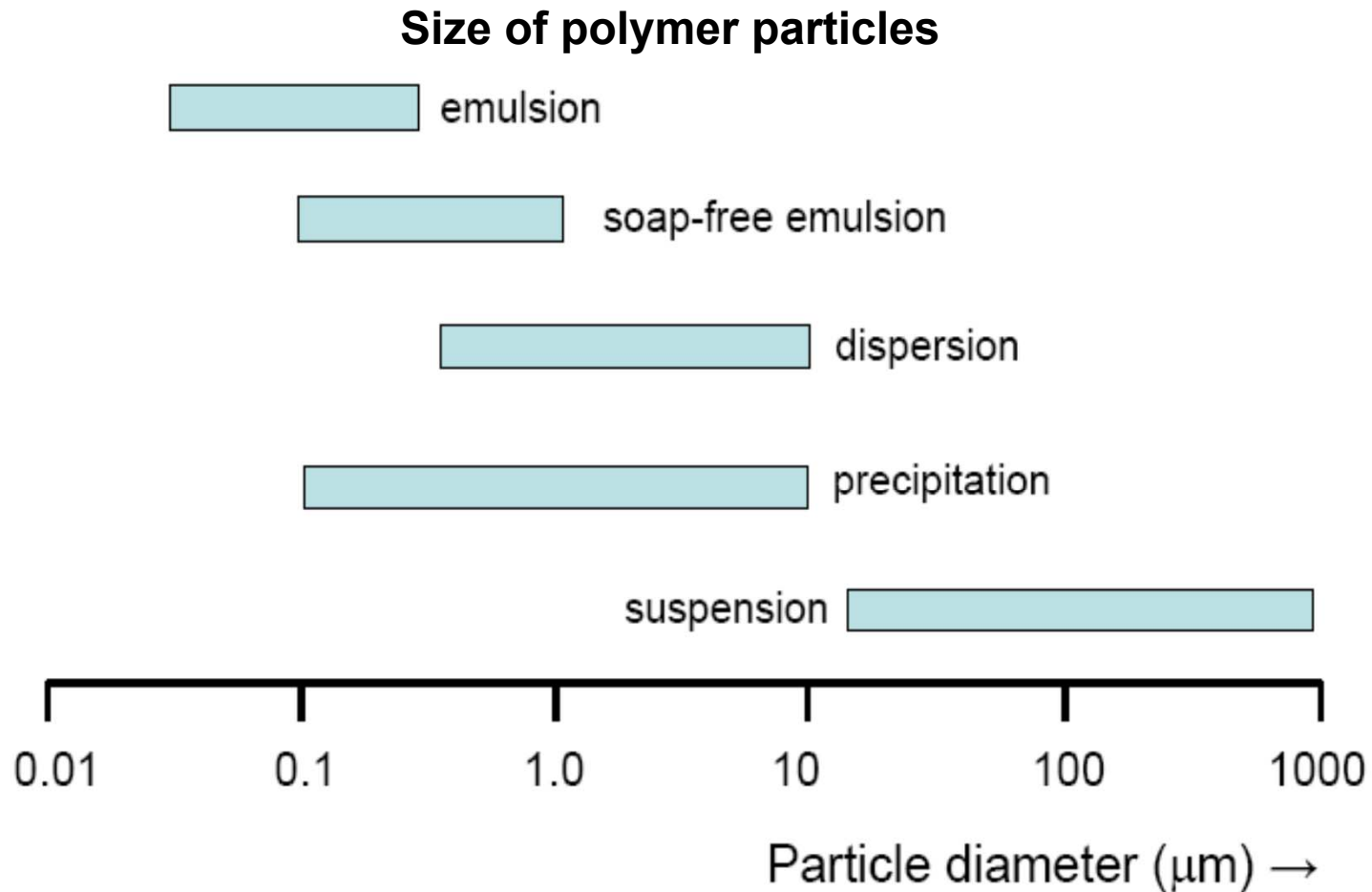
Advantages:	Disadvantages:
<ul style="list-style-type: none">➤ Lower viscosities → better control of the gel effect➤ Better temperature control➤ Polymer solution can be used for some applications (coatings)	<ul style="list-style-type: none">➤ Removal and recycling of solvents➤ Toxicity/flammability of solvents➤ Chain transfer to solvent

Used for poly(vinyl acetate) [methanol, *tert*-butanol],
ethylene [*e.g.*, cyclohexane]

Polymerization of the monomer in the gas-phase
Catalyst is immobilized on particle (mainly metal catalysts)
Only for gaseous monomers (e.g. ethylene, butadiene)



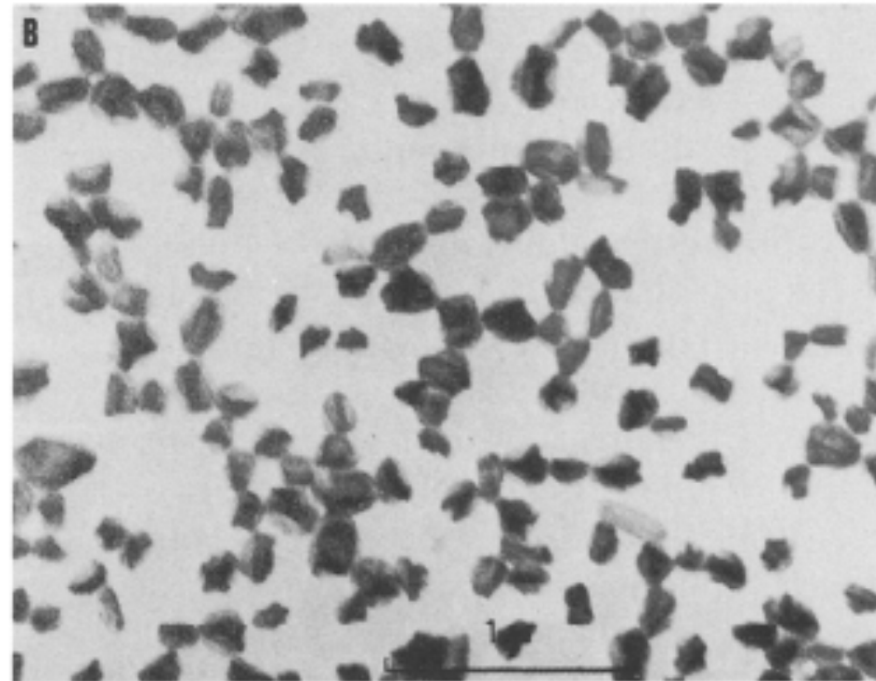
Heterogeneous polymerization: two phase system, polymer particles are formed immiscible with liquid phase



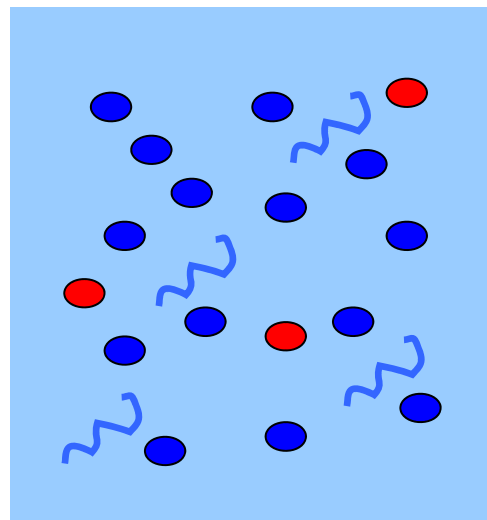
Polymerization of the monomer in the presence of a solvent or in bulk
Polymer is not soluble in the monomer or the solvent
Monomer and initiator are soluble
Monomer does not swell the polymer

—————→ **polymer precipitates** (irregularly shaped particles)

used for polymerization
of acrylonitrile in water



Monomer, initiator and stabilizer are soluble in the solvent
Polymer is not soluble in the monomer or the solvent
Precipitated polymer forms stabilized particles, swollen by monomer
Polymerization in particles (monodisperse particles)



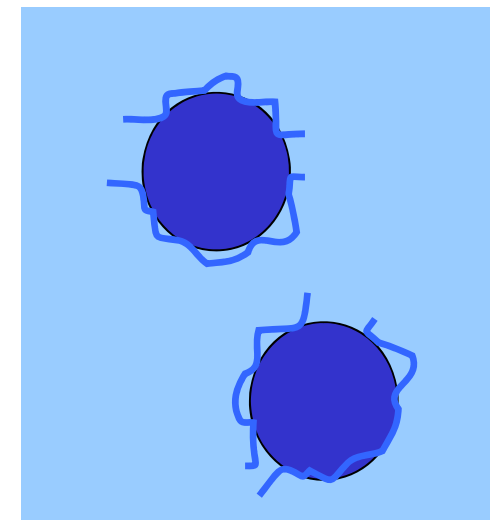
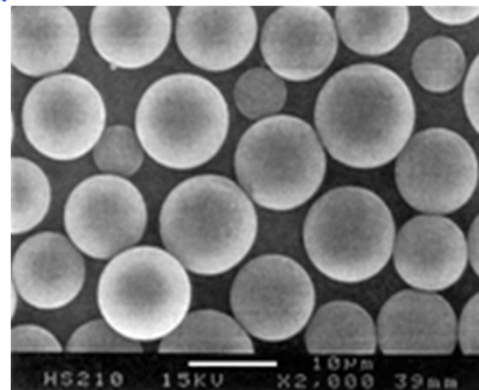
e.g. polystyrene or PMMA
in hydrocarbons or alcohols

Polymerization →

● monomer

● initiator

~ stabilizer

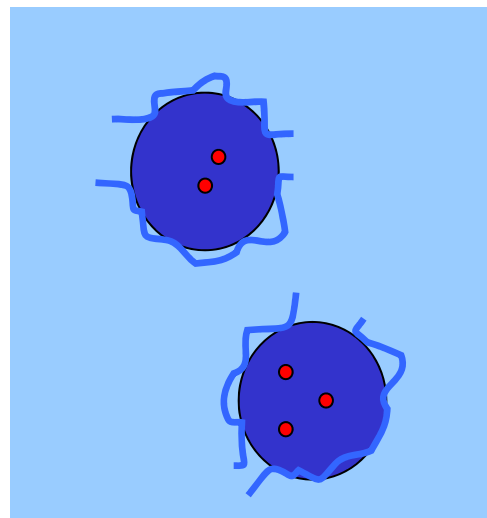


0.1-
10 μm

SEM of polystyrene in
 scCO_2 ,
AIBN as initiator
and fluorinated polyacrylate
as stabilizer

A. I. Cooper, *J. Mater. Chem.* **2000**, *10*, 207-234.

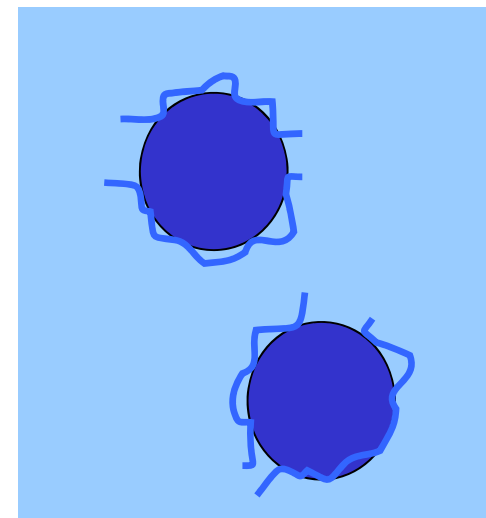
Monomer and initiator are not soluble in the solvent
Polymer is not soluble in the monomer or the solvent
Colloidal stabilizer (e. g. polyvinylalcohol)
Monomer forms stabilized droplets, initiator is dissolved in the monomer droplets
Polymerization in particles (monodisperse particles)
“Mini-bulk polymerization” → same kinetics
Improved heat transfer



monomer droplets

Polymerization
→

• initiator



polymer beads

10-
500 μm

Advantages:	Disadvantages:
<ul style="list-style-type: none">➤ Low viscosity➤ Good heat transfer➤ Easy separation of polymer and liquid phase (filtration)➤ Product: Polymer beads (easy to handle)	<ul style="list-style-type: none">➤ Mostly only batch reactor➤ Reactor wall fouling

Used for polystyrene, poly(vinyl chloride), PMMA
Ion exchange resins



www.indiamart.com.

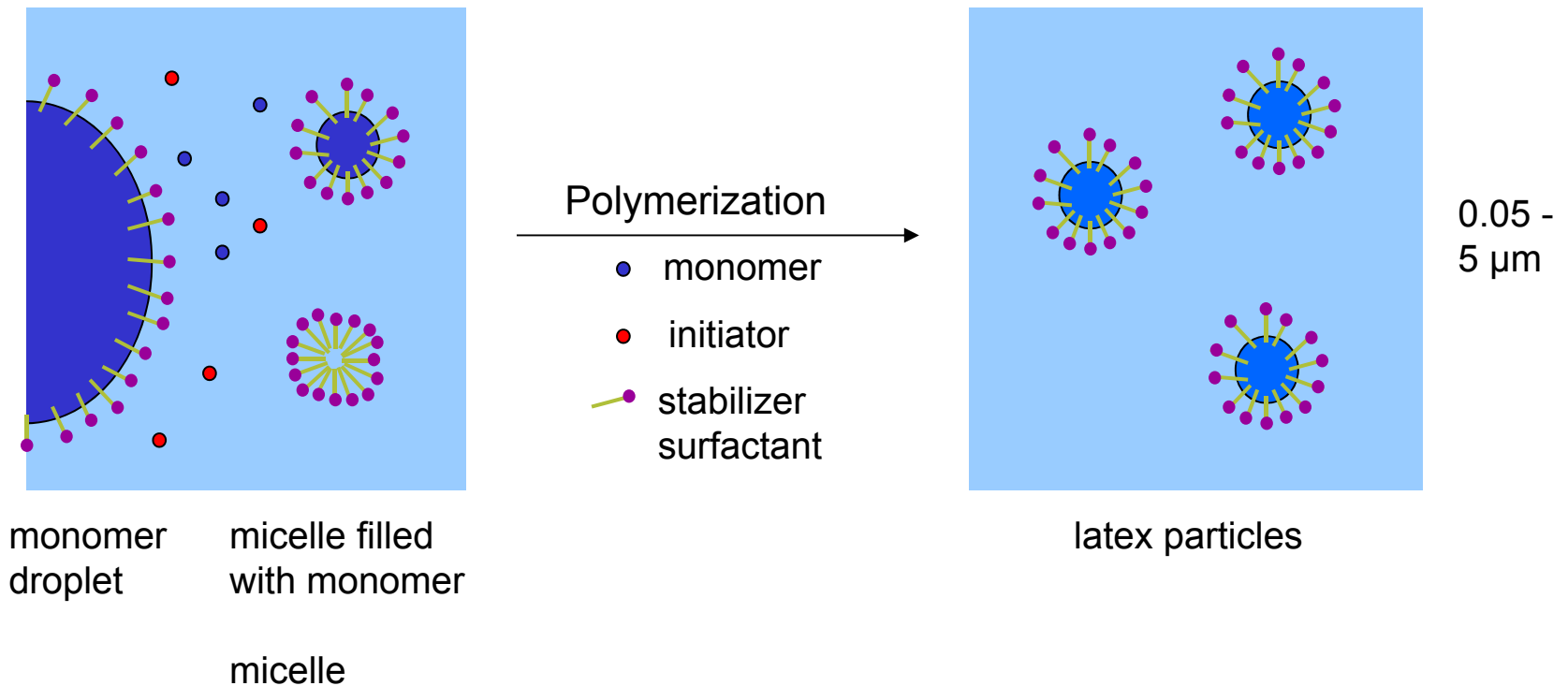
Monomer and polymer are not soluble in the solvent (water)

Initiator is soluble in the solvent

Monomer droplets are dispersed in the non-solvent, micelle forming surfactant

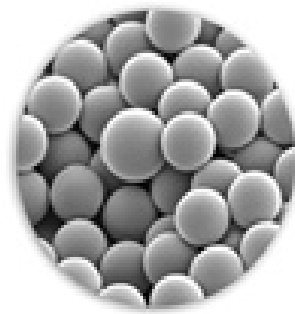
Polymerization takes place in nucleated polymer particles

Large monomer droplets, micelles and micelles filled with monomer



Advantages:	Disadvantages:
<ul style="list-style-type: none">➤ Low viscosity➤ Good heat transfer➤ Easy separation of polymer and liquid phase (filtration)➤ High molar masses➤ Direct application of latices	<ul style="list-style-type: none">➤ Mostly only batch reactor➤ Reactor wall fouling➤ Separation costs➤ Surfactant is necessary

Used for poly(vinyl chloride), polyvinylacetate, coatings, etc.
Application for latices: Paintings and coatings, glues and adhesives
textiles (fleece binder, textile coating)

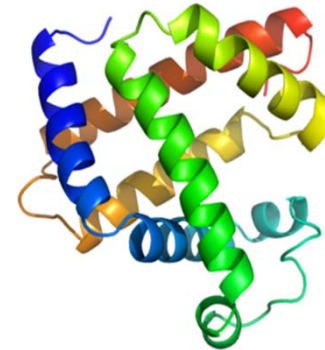
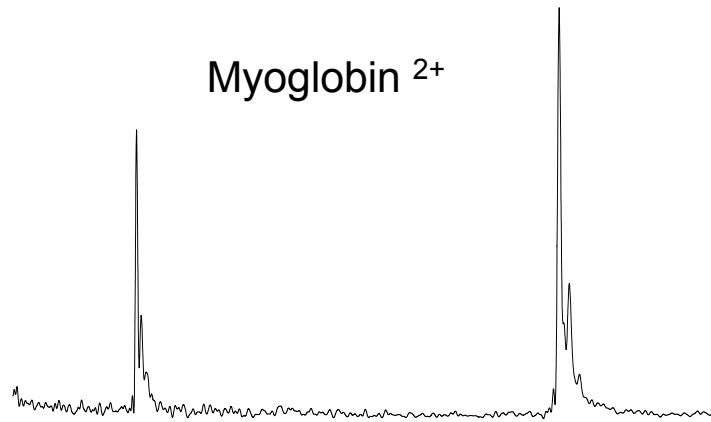


**polystyrene
latex particles**

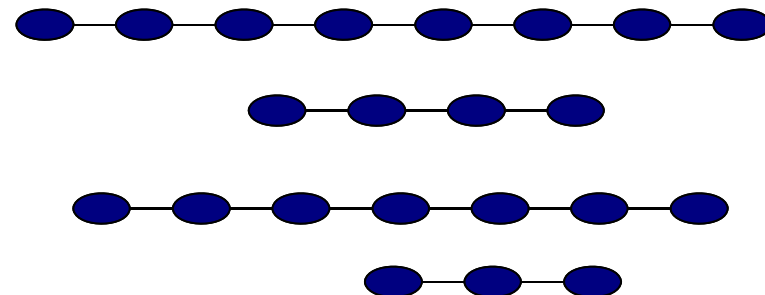
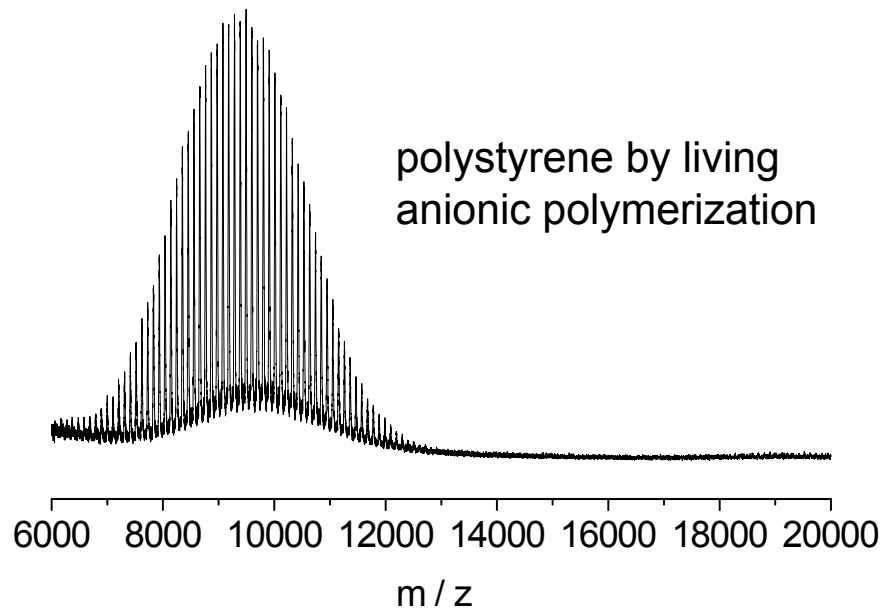
	Suspension polymerization	Emulsion polymerization
Monomer-water ratio	1 : 1 – 4 : 1	Typically < 2: 1
Initiator	Oil soluble	Water soluble
Stabilizing colloid/surfactant	< 0.1 weight%	1 – 5 weigth% > critical micellation concentration
Locus of polymerization	Stabilized colloid monomer droplets	Polymer particles (micelles)
Final particle size	10 – 500 μm	0.05 – 5 μm
Molecular weight	\approx bulk	\geq bulk

MALDI-TOF MS

Myoglobin⁺



Synthetic polymers do not have a specific molar mass. They have a molar mass distribution.



$$M_n = \frac{\sum_i N_i M_i}{\sum_i N_i} = \frac{\sum_i w_i}{\sum_i \frac{w_i}{M_i}}$$

N_i – number of polymer-chains with weight M_i

w_i – weigthfraction of polymerchains with weight M_i

This value is defined as the summation of the molar mass of a chain times the number of moles having that weight, divided by the total moles of polymer.

The number average molar mass of a polymer can be determined by
Osmometry,
End-group titration,
Colligative properties (cryoscopy, ebulliometry),
SEC,
MALDI-TOF-MS.

$$M_w = \frac{\sum_i N_i M_i^2}{\sum_i N_i M_i} = \frac{\sum_i w_i M_i}{\sum_i w_i}$$

N_i – number of polymer-chains with weight M_i

w_i – weight fraction of polymerchains with weight M_i

Weigh a number of polymer molecules, add the squares of these weights, and then divide by the total weight of the molecules.

The weight average molar mass of a polymer can be determined by
Light scattering (LS),
Small angle neutron scattering (SANS),
Sedimentation velocity,
SEC,
MALDI-TOF-MS.

$$M_z = \frac{\sum_i N_i M_i^3}{\sum_i N_i M_i^2} = \frac{\sum_i w_i M_i^2}{\sum_i w_i M_i}$$

N_i – number of polymer-chains with weight M_i

w_i – weight fraction of polymer chains with weight M_i

The sedimentation average molar mass of a polymer can be determined by Sedimentation equilibrium.

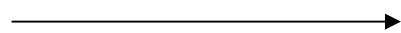
$$M_\eta = \left(\frac{\sum N_i M_i^{1+\alpha}}{\sum N_i M_i} \right)^{\frac{1}{\alpha}}$$

usually:

$$0.5 \leq \alpha \leq 1 \longrightarrow M_n \leq M_\eta \leq M_w$$

$$\alpha = -1 \rightarrow M_\eta = M_n$$

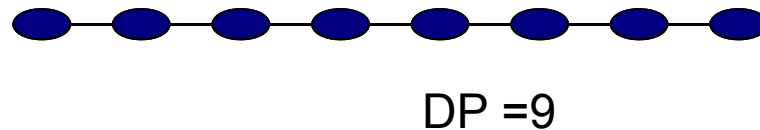
$$\alpha = 1 \rightarrow M_\eta = M_w$$



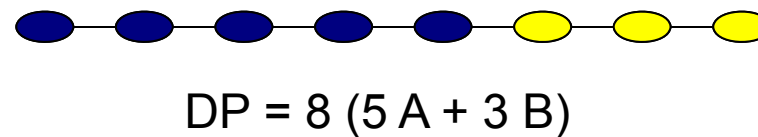
detailed information will follow:
Viscosimetry

$$DP = \frac{M_{\text{macromolecule}}}{M_{\text{repeating unit}}}$$

Number of building blocks per macromolecule



different building blocks → determination of each fraction to obtain DP



Example 1

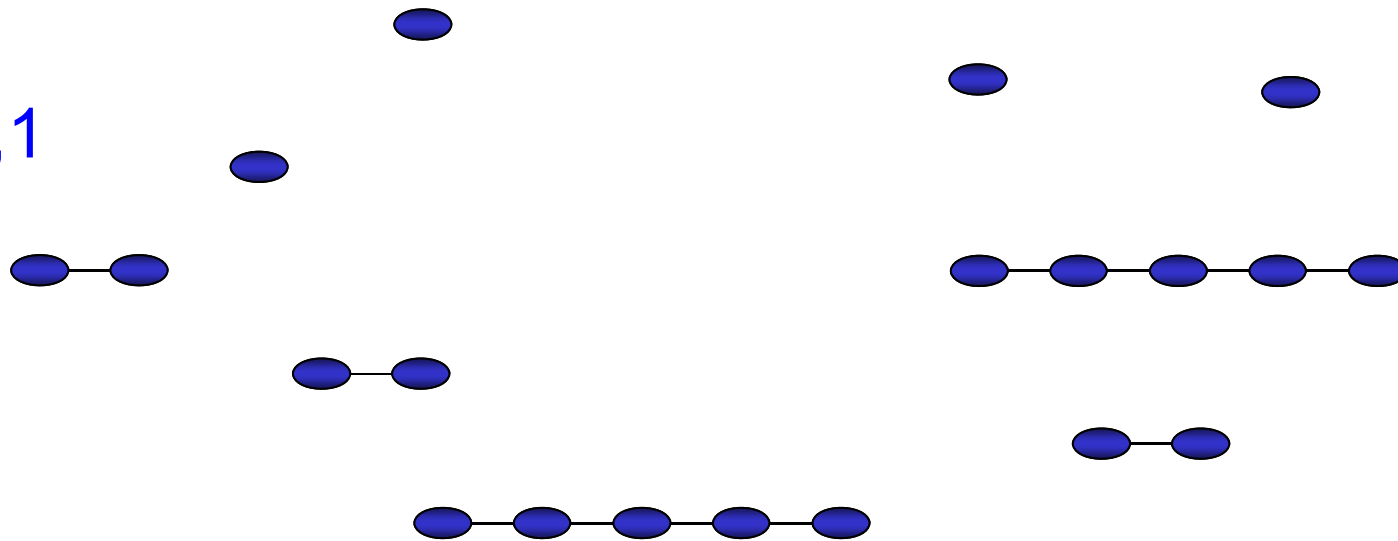


Sample with the following polymers
(number is equal to a certain molar mass or repeating unit):

1,1,1,1

2,2,2

5,5



Calculate M_n , M_w and M_z

Example 1



Number average molar mass

$$M_n = \frac{\sum_i N_i M_i}{\sum_i N_i} \quad M_n = \frac{(4 \times 1) + (3 \times 2) + (2 \times 5)}{4 + 3 + 2} = \frac{4 + 6 + 10}{9} = \frac{20}{9} = 2.22$$

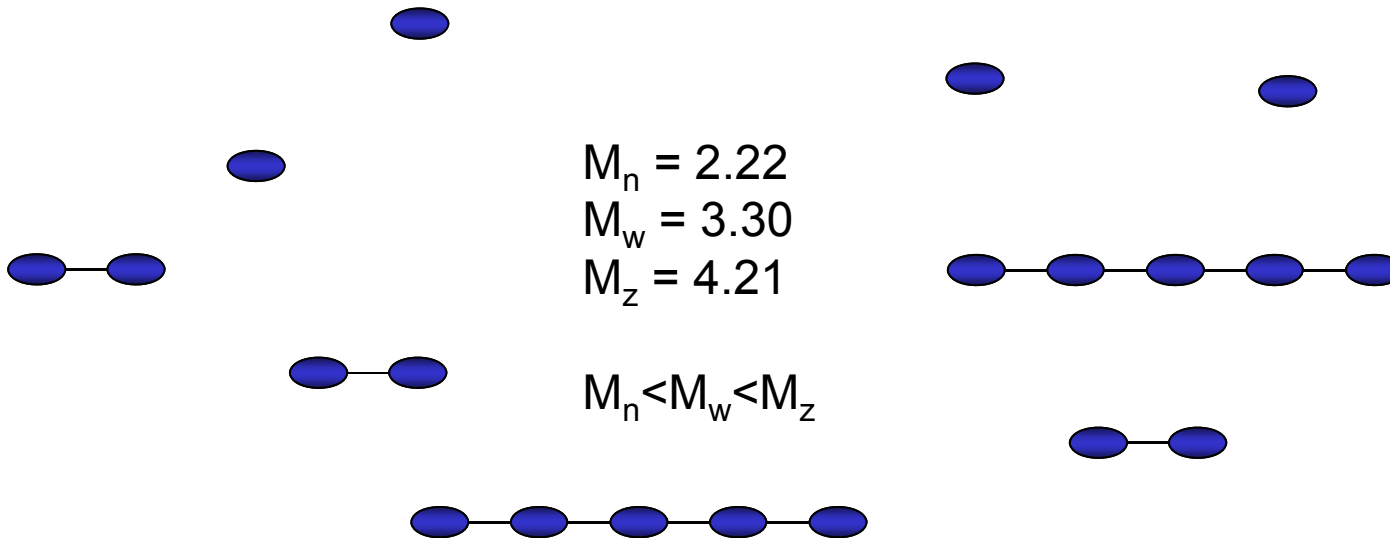
Weight average molar mass

$$M_w = \frac{\sum_i N_i M_i^2}{\sum_i N_i M_i} \quad M_w = \frac{(4 \times 1^2) + (3 \times 2^2) + (2 \times 5^2)}{(4 \times 1) + (3 \times 2) + (2 \times 5)} = \frac{4 + 12 + 50}{4 + 6 + 10} = \frac{66}{20} = 3.30$$

Example 1

Sedimentation average molar mass

$$M_z = \frac{\sum_i N_i M_i^3}{\sum_i N_i M_i^2} \quad M_z = \frac{(4 \times 1^3) + (3 \times 2^3) + (2 \times 5^3)}{(4 \times 1^2) + (3 \times 2^2) + (2 \times 5^2)} = \frac{4 + 24 + 250}{4 + 12 + 50} = \frac{278}{66} = 4.21$$

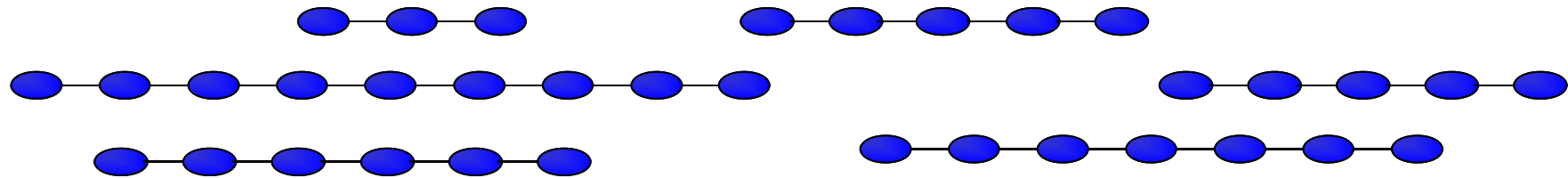


Example 2

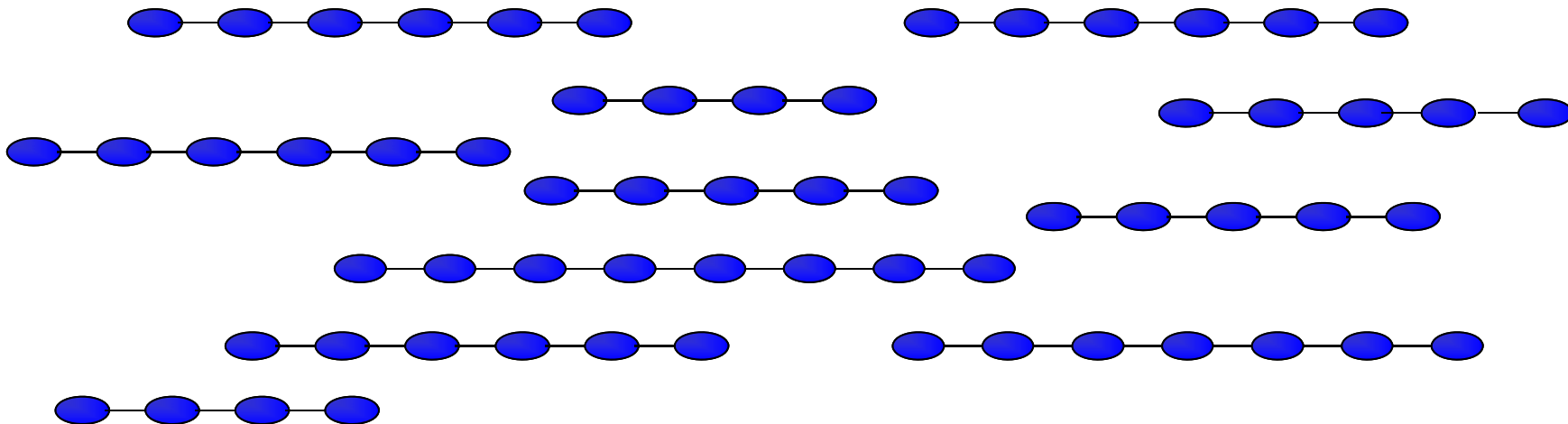


More realistic:

M	0	1	2	3	4	5	6	7	8	9	10
N	0	0	0	1	2	5	4	2	1	1	0



Calculate M_n , M_w and M_z



Example 2



More realistic:

M	0	1	2	3	4	5	6	7	8	9	10
N	0	0	0	1	2	5	4	2	1	1	0

$$M_n = \frac{(1 \times 3) + (2 \times 4) + (5 \times 5) + (4 \times 6) + (2 \times 7) + (1 \times 8) + (1 \times 9)}{1 + 2 + 5 + 4 + 2 + 1 + 1} =$$

$$M_n = \frac{3 + 8 + 25 + 24 + 14 + 8 + 9}{16} = \frac{91}{16} = 5.69$$

$$M_w = \frac{(1 \times 3^2) + (2 \times 4^2) + (5 \times 5^2) + (4 \times 6^2) + (2 \times 7^2) + (1 \times 8^2) + (1 \times 9^2)}{(1 \times 3) + (2 \times 4) + (5 \times 5) + (4 \times 6) + (2 \times 7) + (1 \times 8) + (1 \times 9)} =$$

$$M_w = \frac{9 + 32 + 125 + 144 + 98 + 64 + 81}{91} = \frac{553}{91} = 6.07$$

Example 2



More realistic:

M	0	1	2	3	4	5	6	7	8	9	10
N	0	0	0	1	2	5	4	2	1	1	0

$$M_z = \frac{(1 \times 3^3) + (2 \times 4^3) + (5 \times 5^3) + (4 \times 6^3) + (2 \times 7^3) + (1 \times 8^3) + (1 \times 9^3)}{(1 \times 3^2) + (2 \times 4^2) + (5 \times 5^2) + (4 \times 6^2) + (2 \times 7^2) + (1 \times 8^2) + (1 \times 9^2)} =$$
$$M_z = \frac{27 + 128 + 625 + 864 + 686 + 512 + 729}{553} = \frac{3571}{553} = 6.46$$

$$M_n = 5.69$$

$$M_w = 6.07$$

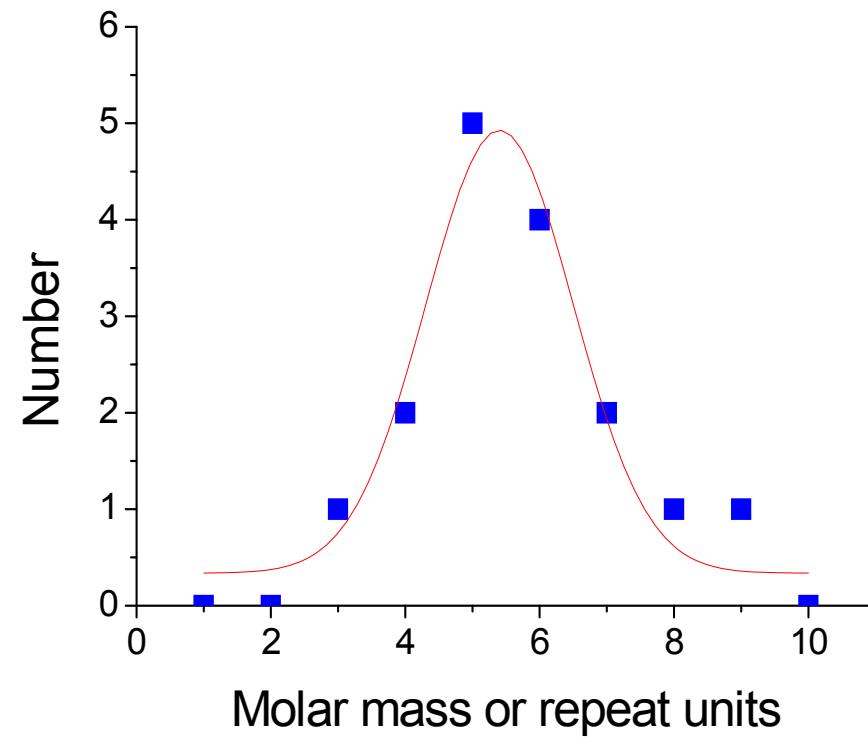
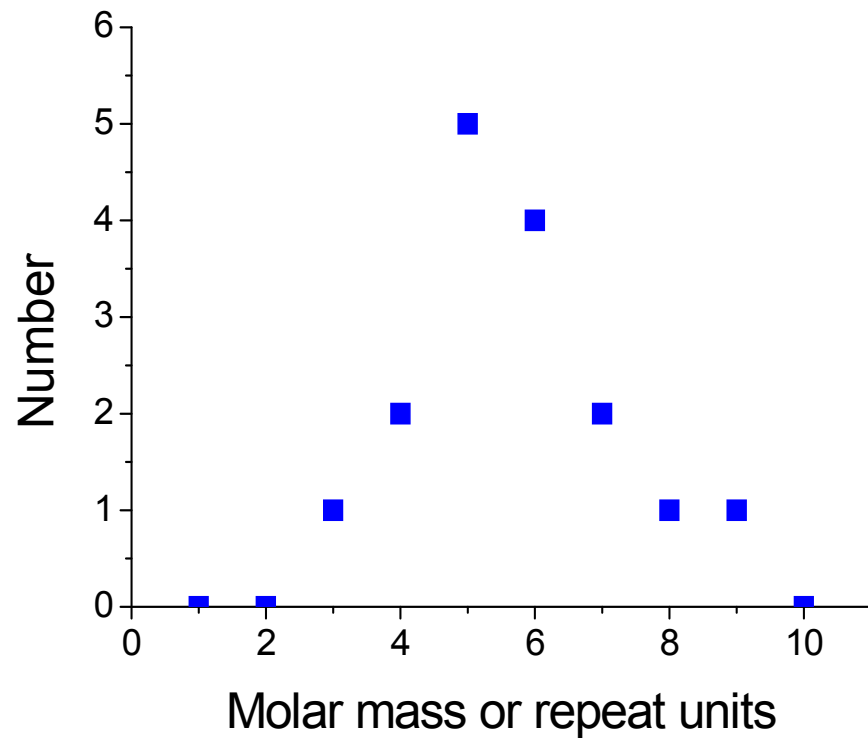
$$M_z = 6.46$$

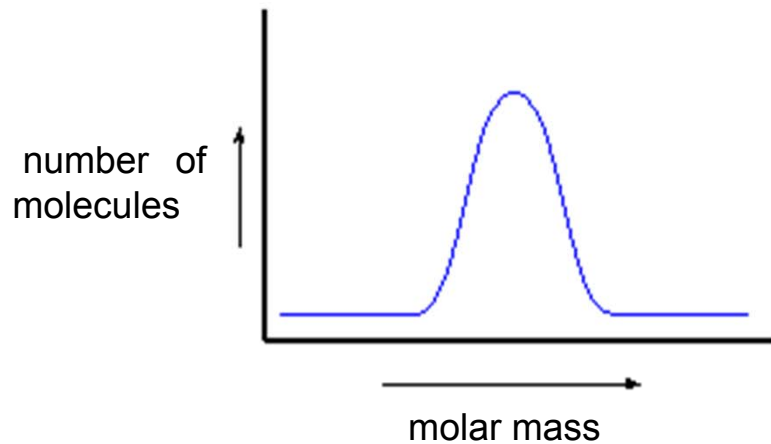
$$M_n < M_w < M_z$$

$$M_n = 5.69$$

$$M_w = 6.07$$

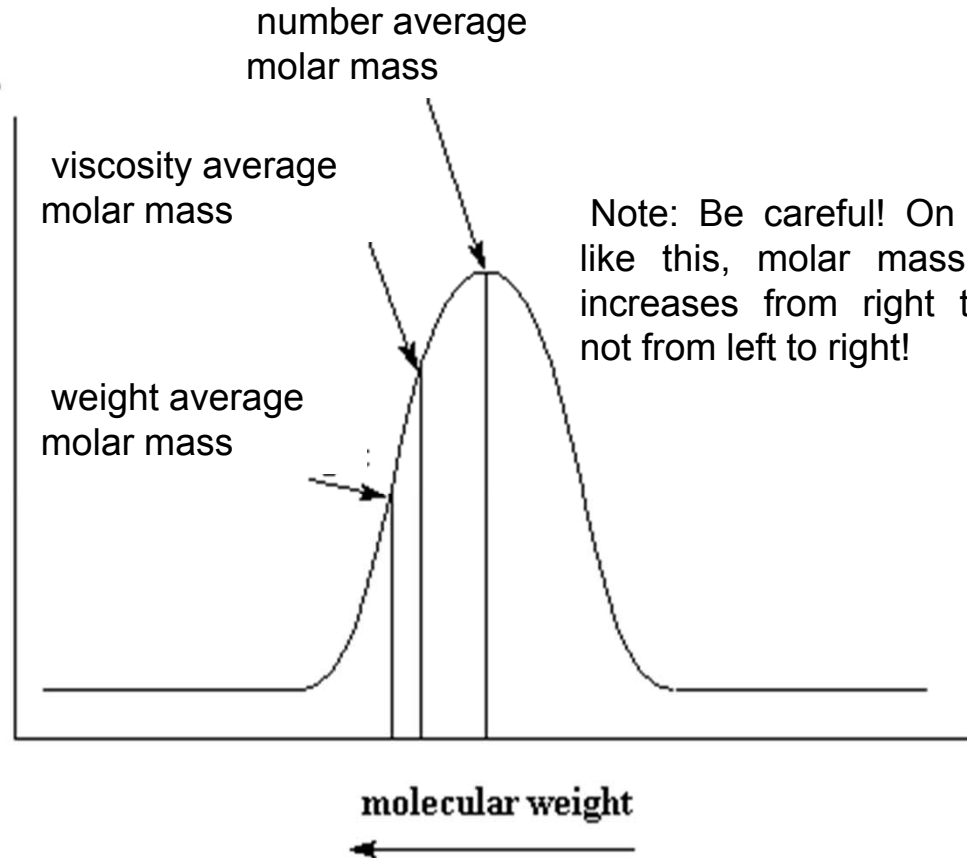
$$M_z = 6.46$$





$$M_n \leq M_\eta \leq M_w \leq M_z$$

number molecules of



Note: Be careful! On a plot like this, molar mass often increases from right to left, not from left to right!

The dispersity (\bar{D}) / polydispersity index (PDI) is the ratio of the **weight average molar mass** to the **number average molar mass**.

The \bar{D} / PDI indicates the distribution of individual molar masses in a batch of polymers.

$$PDI = \frac{M_w}{M_n}$$

Example 1:

$$PDI = \frac{M_w}{M_n} = \frac{3.30}{2.22} = 1.49$$

Example 2:

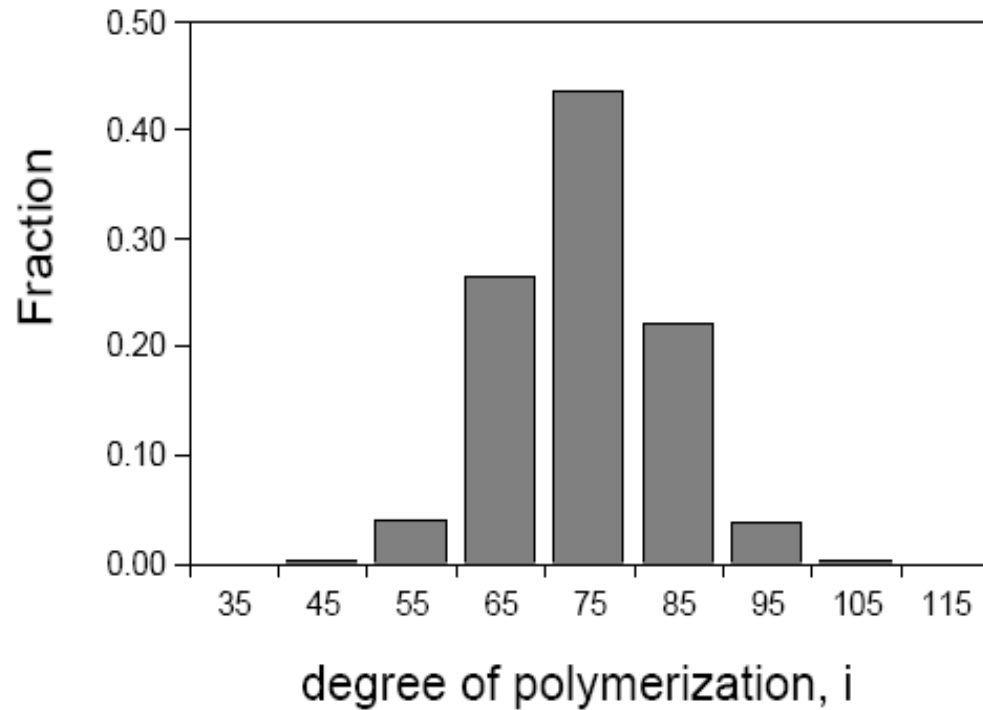
$$PDI = \frac{M_w}{M_n} = \frac{6.07}{5.69} = 1.07$$

Comparison of two polymers with same M_n

P1	M	46	47	48	49	50	51	52	53	54
	N	2	1	4	5	3	8	0	4	1
P2	M	11	21	34	40	50	65	72	83	93
	N	3	6	5	6	3	8	4	4	2

	M_n	M_w	DPI
P1	50	50.09	1.002
P2	50	61.28	1.226

Polymer with $DP_n = 75$ and PDI 1.01 \rightarrow nearly monodisperse?



only 5% of the chains $DP = 75$
 only 47% have $DP = 75 \pm 5$

method	type	range	average
membrane osmometry	absolut	$10^4 - 10^6$	M_n
endgroup analysis	absolut	$< 5 \cdot 10^4$	M_n
vaporpressure osmometry	absolut	$< 2 \times 10^4$	M_n
ebulliometry / cryoscopy	absolut	$< 5 \cdot 10^4$	M_n
light scattering	absolut	$> 5 \cdot 10^3$	M_w
MALDI-TOF-MS ESI-TOF-MS	absolut	up to 10^6	M_n, M_w
analytical ultracentrifuge	absolut	100 to $5 \cdot 10^8$	M_w, M_z
viscosimetry in diluted solutions / viscosimetry of melting	relative	$> 10^2$	M_n
gelpermeation chromatography size exclusion chromatography	relative	$< 5 \cdot 10^6$	M_n, M_w

Analysis of endgroups by:

Titration

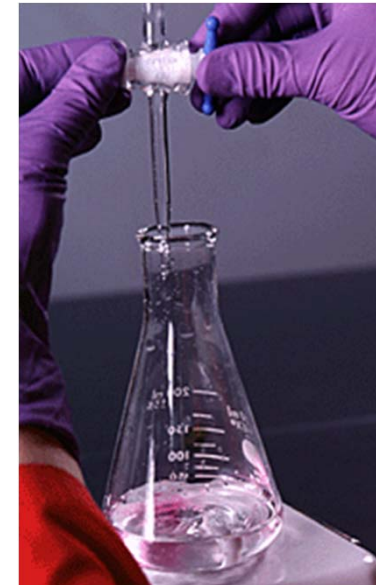
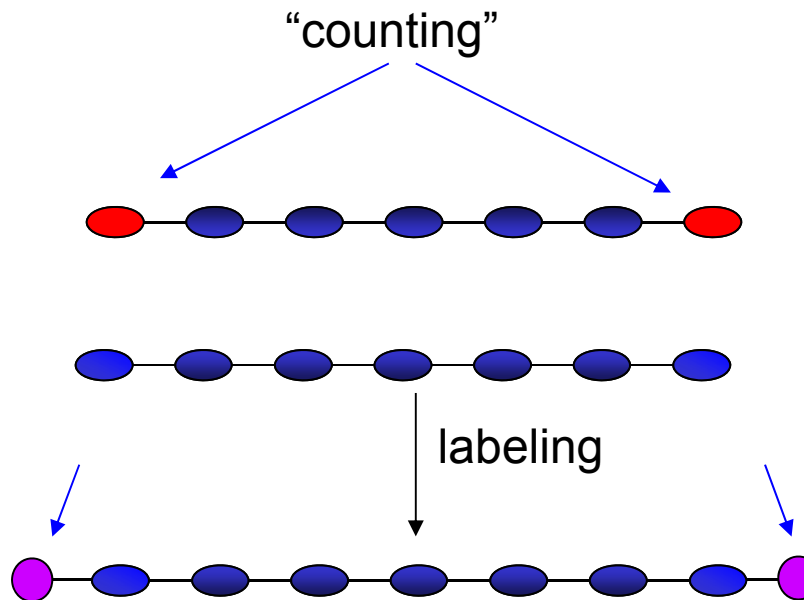
NMR-spectroscopy

UV/vis-spectroscopy

IR-spectroscopy

Radiochemical

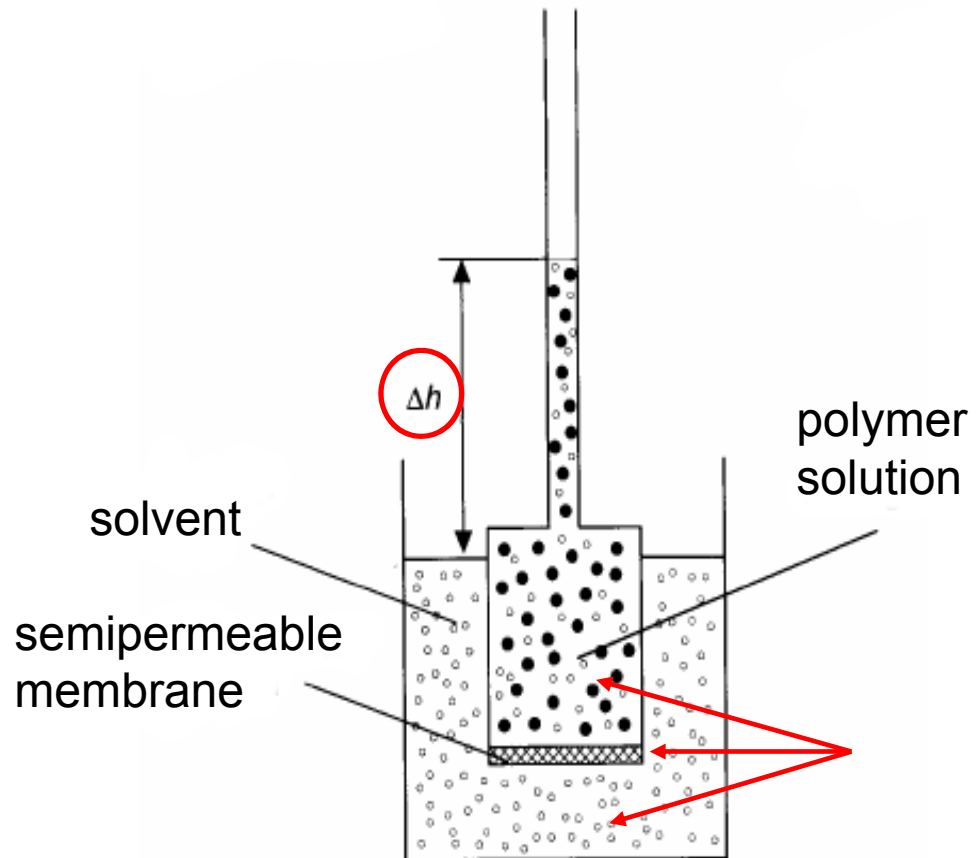
Labeling of endgroups



Most common method ($\rightarrow M_n$): titration (chemical reactions on endgroup)
Number of reacted end groups can be quantified by spectroscopy or pH measurement
Number of endgroups in known amount of polymer $\rightarrow M_n$
Difficulties: solubility, viscosity, steric hindrance, mechanisms of initiation and termination (\rightarrow which endgroup? one or two?)
Best suited in the range from 5,000 to 10,000 molar mass

Photo source: <http://www.dartmouth.edu/~chemlab/techniques/titration.html>.

Membrane osmometry



Determination of M_n

Two chambers, separated
by semipermeable membrane

Polymer solution \leftrightarrow pure solvent

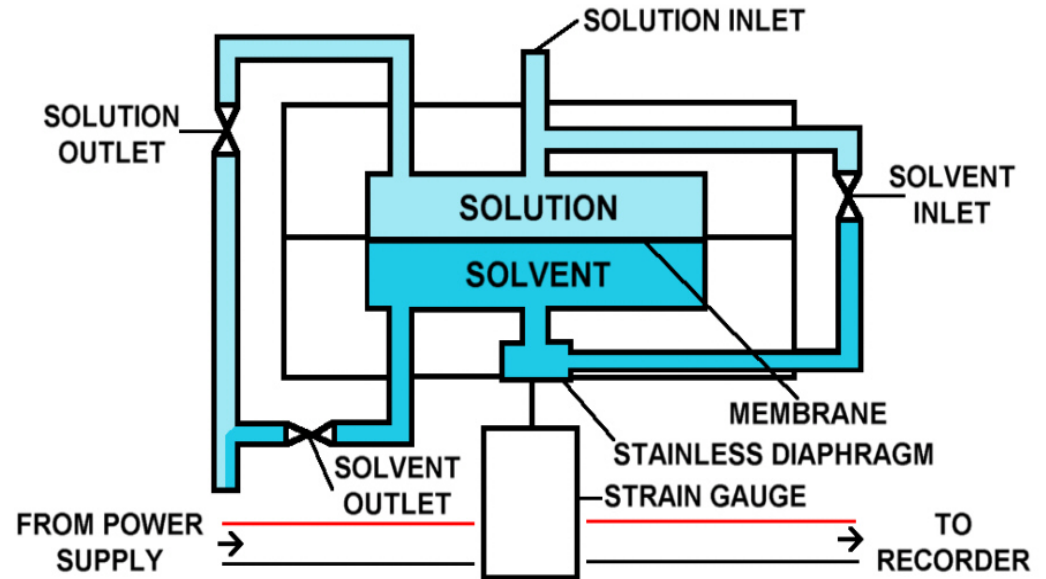
Solvent diffusion of solvent into
the polymer solution

Δh proportional to osmotic pressure

Tieke, www.gonotec.com/img/GER/bild_osmomat90.gif



membrane osmometer



http://www.uicinc.com/SysSheet_HTMLs/Model230.htm

$$\Pi = \rho \cdot g \cdot \Delta h \quad \text{correlation osmotic pressure and } \Delta h$$

can be measured

osmotic pressure \rightarrow molecular weight (analogue van't Hoff: equation for gases)

$$\Pi \cdot V = n \cdot R \cdot T$$

$$\Pi \cdot V = R \cdot T \cdot \frac{m}{M}$$

only for infinite
diluted solutions

$$\Pi = \left(\frac{m}{V} \right) \cdot \frac{R \cdot T}{M} = \frac{c \cdot R \cdot T}{M} = \frac{c \cdot R \cdot T}{M_n}$$

monodisperse
polydisperse
sample

Π – osmotic pressure
 ρ – density of solution
 g – acceleration of gravity
 (Erdbeschleunigung)
 V – volume of solution

T – temperature [K]
 n = number of mols (polymer)
 R – gas constant (83143 mbar mL/mol K)
 M – molar mass

$$\frac{\Pi}{c} = \left(\frac{R \cdot T}{M_n} \right) + A_2 \cdot c + A_3 \cdot c^2 + \dots$$

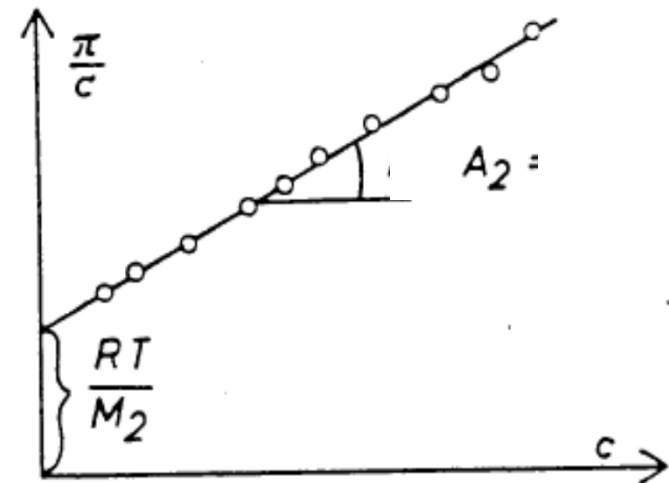
A_2 – virial coefficient
polymer – solvent interaction

approximation

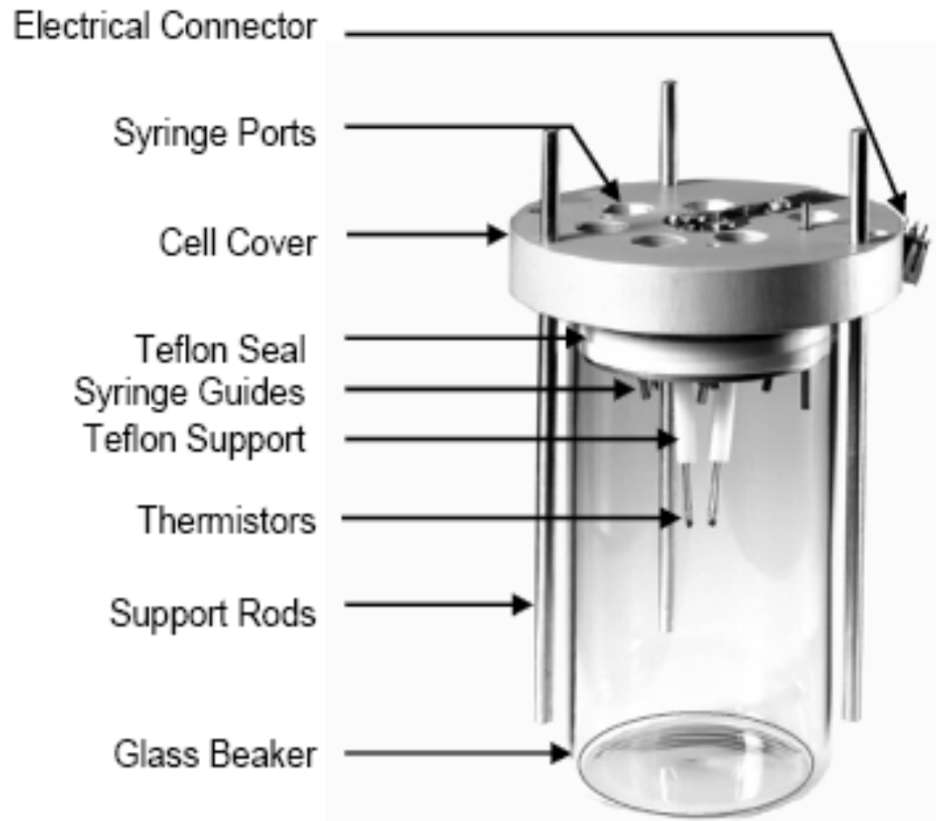
real solutions

measurement of osmotic pressure
for different concentrations

$$\bar{M}_n = \frac{RT}{(\pi/c)_{c \rightarrow 0}}$$



Vapor pressure osmometry



Vapor pressure osmometry – theory I

$$\frac{p_1}{p_1^0} = x_1 = 1 - x_2 \quad \text{Raoult's law}$$

$$1 - \frac{p_1}{p_1^0} = \frac{\Delta p}{p_1^0} = x_2 = \frac{n_2}{n_1 + n_2}$$

$$\frac{\Delta p}{p_1^0} \approx \frac{n_2}{n_1} \quad \text{very diluted solutions}$$

p_1 – vapor pressure of solution

p_1^0 – vapor pressure of solvent

x_1 – mole fraction of solvent

Δp – relative vapor pressure lowering (pure solvent \rightarrow solution)

n – number of mols

Vapor pressure osmometry – theory II

Measurement in saturated (solvent) atmosphere

Droplet of polymer solution is placed on thermistor

Beginning: the droplet has the same temperature as the surrounding

Droplet: lower vapor pressure → condensation of solvent → heat of condensation

Temperature difference between droplet and surrounding

$$\frac{\Delta T_{th}}{c} = \frac{R \cdot T^2}{L_1 \cdot \rho_s} \cdot \frac{1}{M_n}$$

ΔT_{th} – temperature difference theory

c – concentration of polymer solution

R – gas constant

T – temperature [K]

L_1 – vaporization heat of solvent (per g)

ρ_s – density of solution

→ Clausius-Clapeyron

Tieke; Lechner, Gehrke, Nordmeier.

Vapor pressure osmometry – theory III

Thermal contact between droplet and surrounding:

$$\Delta T_{\text{exp}} = \text{const.} \cdot \Delta T_{\text{th}}$$

$$\frac{\Delta T_{\text{exp}}}{c} = \text{const.} \cdot \frac{R \cdot T^2}{L_1 \cdot \rho_s} \cdot \frac{1}{M_n}$$

$$\frac{\Delta T_{\text{exp}}}{c} = K_E \cdot \frac{1}{M_n}$$

↑
calibration with known
model substance

ΔT_{th} – temperature difference (theory)

ΔT_{exp} – experimental difference

c – concentration of polymer solution

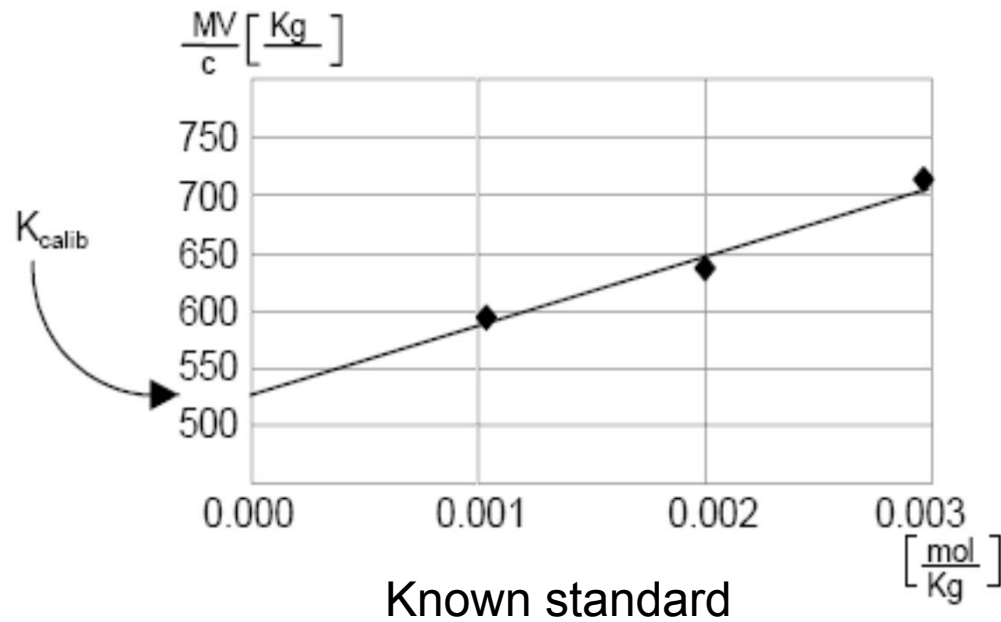
R – gas constant

T – temperature [K]

L_1 – vaporization heat of solvent (per g)

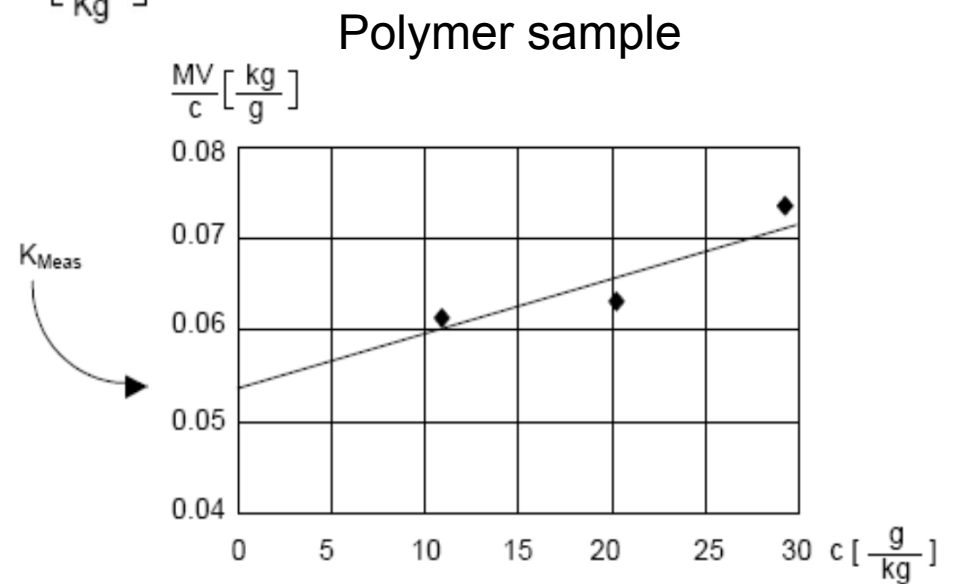
ρ_s – density of solution

K_E – constant (calibration)



MV – measured value

$$M \left[\frac{\text{g}}{\text{mol}} \right] = \frac{K_{\text{calib}}}{K_{\text{Meas}}}$$



Polymer solutions are viscous. Viscosity is influenced by:

Solvent

Kind of polymer

Molar mass of polymer

Concentration of polymer

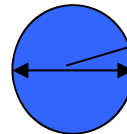
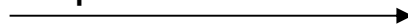
Temperature

Theory – Kuhn's Law



ideal coil

approximation
sphere



d – proportional to
radius of gyration $\sqrt{\langle s^2 \rangle}$

$$\langle s^2 \rangle = K_{geometry} \cdot N = K_{geometry} \cdot \frac{M_{coil}}{M_{monomer}} = K^2 \cdot M \quad \longrightarrow \quad \sqrt{\langle s^2 \rangle} = K \cdot \sqrt{M}$$

M_{coil} – molar mass of the coil

$M_{monomer}$ – molar mass of the monomer

$$\rho_{sphere} = \frac{m}{\frac{1}{6}\pi \cdot d^3} \Rightarrow \frac{M}{\sqrt{M}^3} \cdot K_{\rho} = \frac{K_{\rho}}{\sqrt{M}} = \rho_{equ.}$$

Tieke; Lechner, Gehrke, Nordmeier.

$$\eta_{rel} = \frac{\eta}{\eta_0} = \left(\frac{\text{viscosity of solution}}{\text{viscosity of solvent}} \right) = 2.5 \cdot \varphi + 1$$

according to Einstein the **relative viscosity** η_{rel} of a solution with spherical particles depends only on **volume ratio** φ of the particles and the solution

$$\varphi = \frac{V}{V_L} = \frac{\text{volume of dispersed phase}}{\text{volume of solution}}, V = \frac{m}{\rho} \Rightarrow V_{equ.} = \frac{m}{\rho_{equ.}}$$

problem: Is m the mass of the polymer plus solvent shell or the mass of the coil without solvent shell?

—————→ solution: We use equivalent spheres, which behave like the coil.

$$\varphi = \frac{V}{V_L} = \frac{\text{volume of dispersed phase}}{\text{volume of solution}}, V_{equ.} = \frac{m}{\rho_{equ.}}$$

c – concentration
 $V_{equ.}$ – equ. volume
 $\rho_{equ.}$ – equ. density
 m – mass

$$\varphi = \frac{\frac{m}{\rho_{equ.}}}{V_L} = \frac{m}{V_L} \cdot \frac{1}{\rho_{equ.}} = \frac{c}{\rho_{equ.}}$$

$$\eta_{rel} = 2.5 \cdot \varphi + 1 = \frac{2.5}{\rho_{equ.}} \cdot c + 1$$

Einsteins viscosity law:

$$\eta_{sp} = \eta_{rel} - 1 = \frac{\eta - \eta_0}{\eta_0} = \frac{2.5}{\rho_{equ.}} \cdot c \quad \text{specific viscosity } \eta_{sp}$$

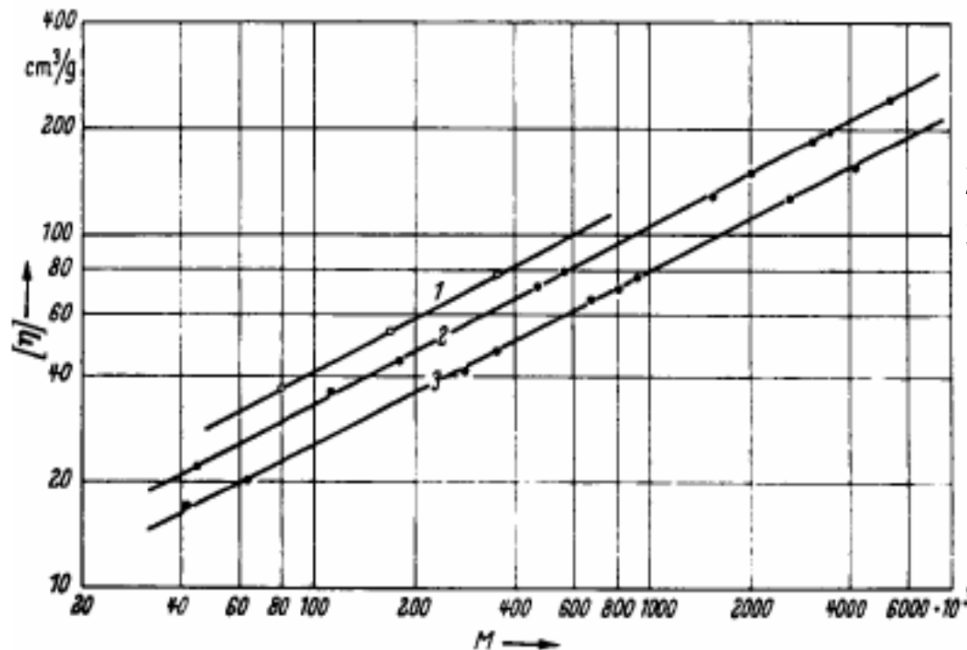
$$\eta_{red} = \frac{\eta_{sp}}{c} = \frac{2.5}{\rho_{equ.}} \quad \text{reduced viscosity } \eta_{red}$$

Staudinger:

$$\lim_{c \rightarrow 0} \frac{\eta_{sp}}{c} = [\eta] = \frac{2.5}{\rho_{equ.}} = \frac{2.5}{K_{\rho}} \cdot M^{0.5} = K_{\eta} \cdot M^{0.5}$$

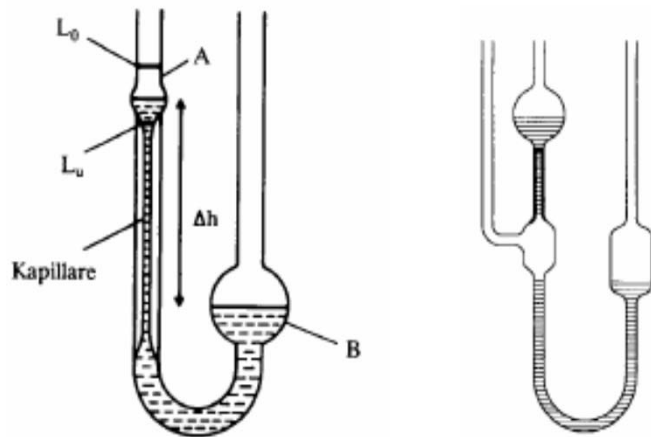
$$\frac{K_{\rho}}{\sqrt{M}} = \rho_{equ.}$$

Kuhn's Law



- 1 – cellulose tricaprylate in γ -phenylpropylalcohol (321 K)
- 2 – polyisobutylene in benzene (297 K)
- 3 – polystyrene in cyclohexane (307 K)

These values are only obtained under these special conditions (right solvent and temperature) polymer \rightarrow “ideal coil”



Measurement of viscosity

Determination of the time in which the solution flows through a capillary

Ostwald-, and Ubbelohde-Viscosimeter

Law of Hagen-Poiseuille:

$$\eta = \frac{\pi \cdot r^4 \cdot \Delta p \cdot t}{8 \cdot l \cdot V}$$

$$\Delta p = \rho \cdot g \cdot h$$

- r – diameter of capillary
- Δp – pressure difference
- V/t – volume which flows in a certain time through capillary
- l – length of capillary
- h – height difference

polymer solution

$$t = \frac{8 \cdot \eta \cdot l \cdot V}{\pi \cdot \rho \cdot r^4 \cdot g \cdot h}$$

$$t_0 = \frac{8 \cdot \eta_0 \cdot l \cdot V}{\pi \cdot \rho_0 \cdot r^4 \cdot g \cdot h}$$

pure solvent

$$t = \frac{8 \cdot \eta \cdot l \cdot V}{\pi \cdot \rho \cdot r^4 \cdot g \cdot h}$$

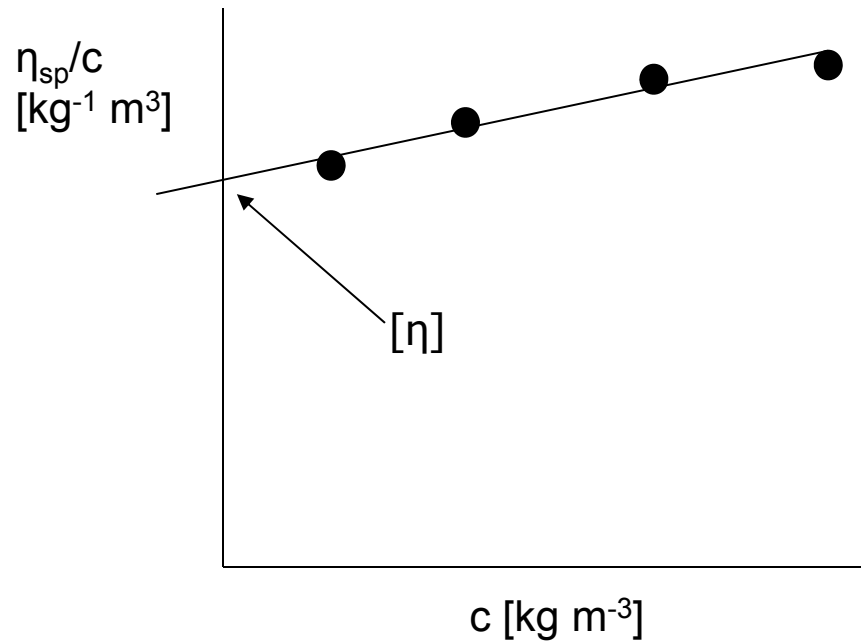
$$t_0 = \frac{8 \cdot \eta_0 \cdot l \cdot V}{\pi \cdot \rho_0 \cdot r^4 \cdot g \cdot h}$$



diluted solution: $\rho = \rho_0$

$$\eta_{rel} = \frac{\eta}{\eta_0} = \frac{\rho \cdot t}{\rho_0 \cdot t_0} \approx \frac{t}{t_0};$$

$$\eta_{sp} = \frac{\eta - \eta_0}{\eta_0} = \frac{t - t_0}{t_0} = \eta_{rel} - 1$$

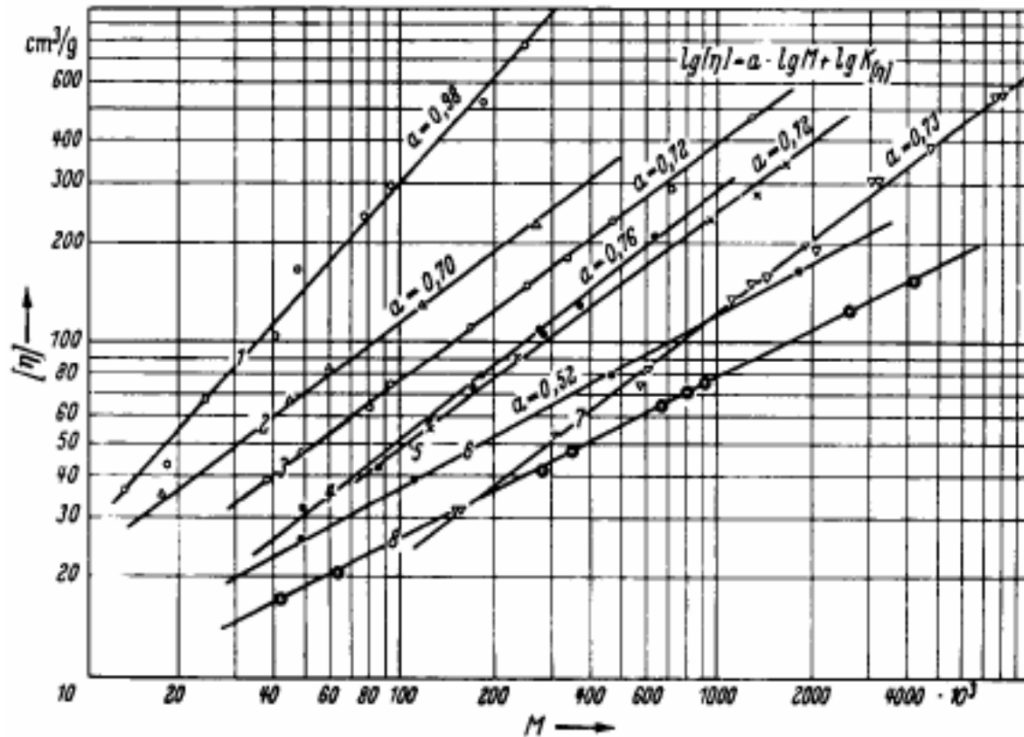


graphical analysis:

$$\frac{\eta_{sp}}{c} = [\eta] + K[\eta]^2 c \quad (\text{Huggins})$$

empirical equation to fit the measured values

Mark-Houwink



[η]-M-plots in double logarithmic scale

- 1 (○) Cellulosenitrate in acetone at 293 K;
 - 2 (Δ) Polycarbonate in THF at 293 K;
 - 3 (⊙) Polyisobutylene in Cyclohexane at 303K;
 - 4 (●) Polypropylene isotactic in α-Chlornaphthaline at 418K;
 - 5 (×) Polystyrol in Toluol bei 295 K;
 - 6 (⊕) Polyisobutylen in Benzol bei 298 K;
 - 7 (∇) PMMA in Aceton bei 293 K;
 - 8 (⊖) Polystyrol in Cyclohexan in 307 K
- (according to G.V. Schulz, Meyerhoff, Flory, Zimm)

➔
$$[\eta] = K_{\eta} \cdot M^{0.5}$$

not all polymers obey this equation!

➔
$$[\eta] = K_{\eta} \cdot M^a$$

Mark-Houwink

Polymer	Solvent	T ^{*)}	K _η ^{**)}	a
natural rubber	Toluene	25	5,00 · 10 ⁻²	0,67
Cellulosenitrate	Acetone	25	0,25 · 10 ⁻²	1,00
Polycaprolactame	m-Kresol	20	0,73 · 10 ⁻²	1,00
	Sulfuric acid	25	2,90 · 10 ⁻²	0,78
Polyethylene (HD)	Tetrahydronaphthaline	120	2,36 · 10 ⁻²	0,78
	α-Chlornaphthaline	125	4,30 · 10 ⁻²	0,67
Polyethylene (LD)	Decaline	70	3,87 · 10 ⁻²	0,74
	Xylene	75	1,35 · 10 ⁻²	0,63
Polyisobutylene	Benzene	24	8,30 · 10 ⁻²	0,50
	Benzene	60	2,60 · 10 ⁻²	0,66
	Toluene	20	2,60 · 10 ⁻²	0,64
Styrene-Butadiene Rubber	Toluene	25	5,20 · 10 ⁻²	0,67
Polystyrene	Benzene	25	1,13 · 10 ⁻²	0,73
	Methylethylketone	25	3,90 · 10 ⁻²	0,58
Polymethylmethacrylate	Benzene	20	0,83 · 10 ⁻²	0,76
	Acetone	20	0,39 · 10 ⁻²	0,73
Polyvinylacetate	Aceton	30	1,76 · 10 ⁻²	0,68
Polyvinylpyrrolidone	Water	30	1,40 · 10 ⁻²	0,70
Polycarbonate	Tetrahydrofurane	20	3,99 · 10 ⁻²	0,70

Mark-Houwink constants

$$[\eta] = K_{\eta} \cdot M^a$$

*) temperature in °C

**) ([η] in mL/g)

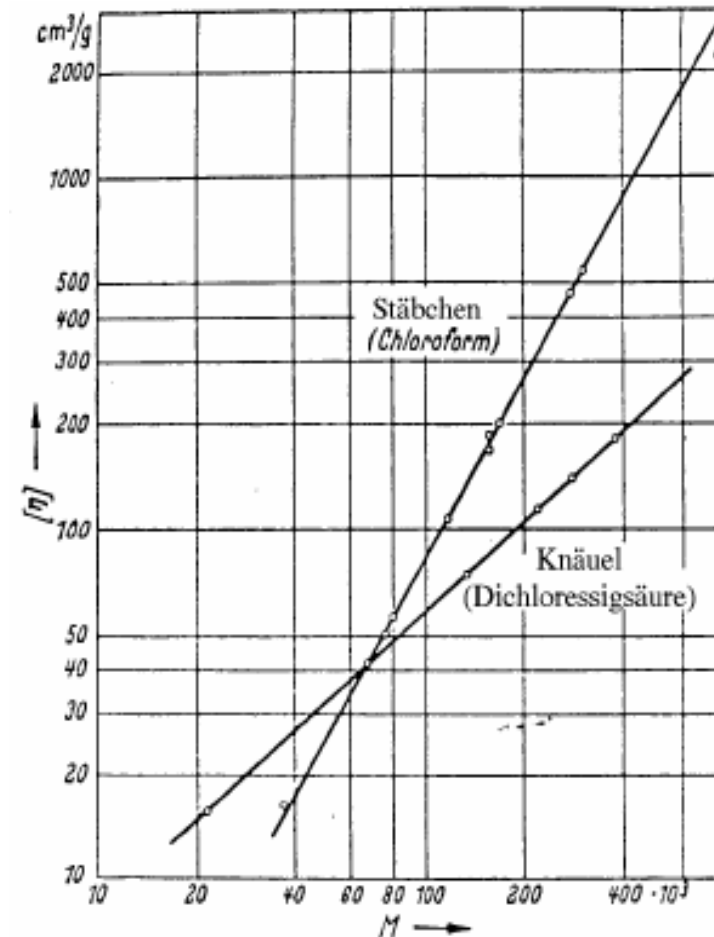
Mark-Houwink constants

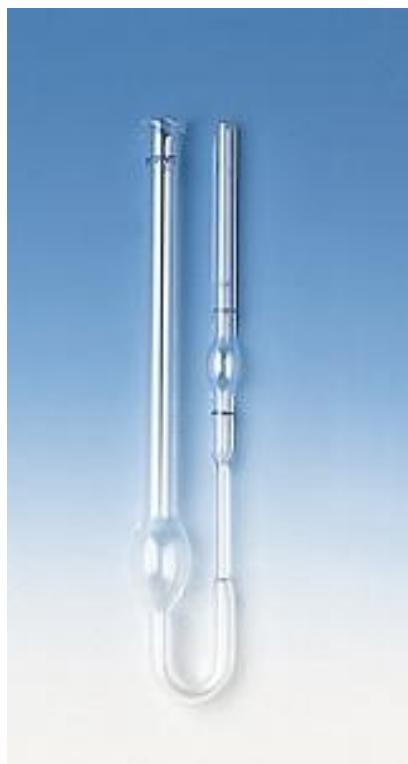
a	shape
0	Hard sphere
0.5	Ideal coil
0.6-0.8	Real coil
≈1	Some cellulose derivatives
1.3-1.4	Rigid rod like polymers
2	Ideal rod

$$[\eta] = K_{\eta} \cdot M^a$$

a – measure for the shape of the macromolecule

poly(benzylglutamate) in chloroform
 → helical structure → rigid rod
 in dichloro acetic acid → coil

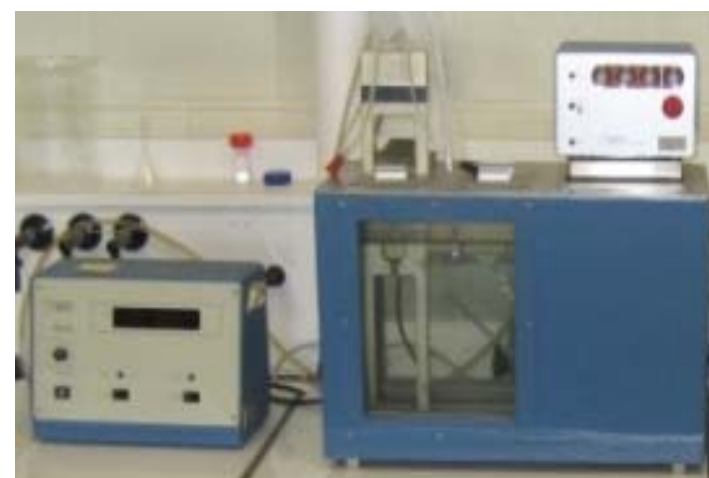




Ostwald-viscosimeter



Ubbelohde-viscosimeter



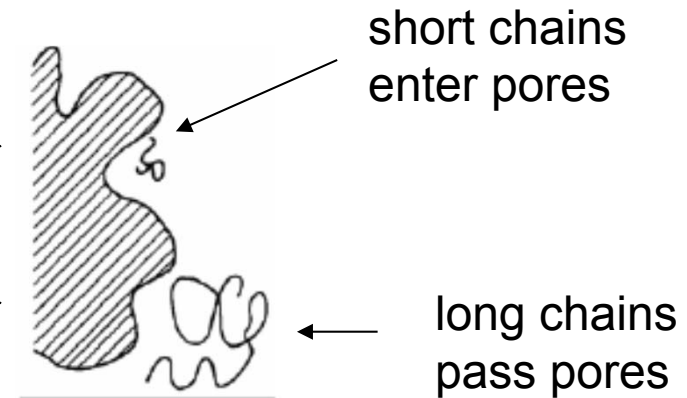
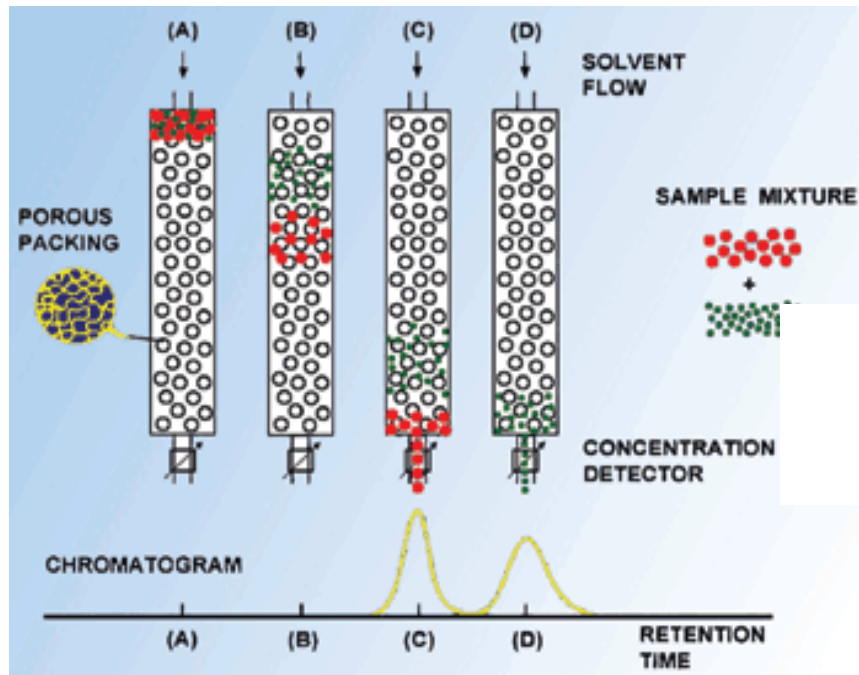
Ubbelohde-viscosimeter
with thermostat



Gel permeation chromatography - GPC

Size exclusion chromatography - SEC

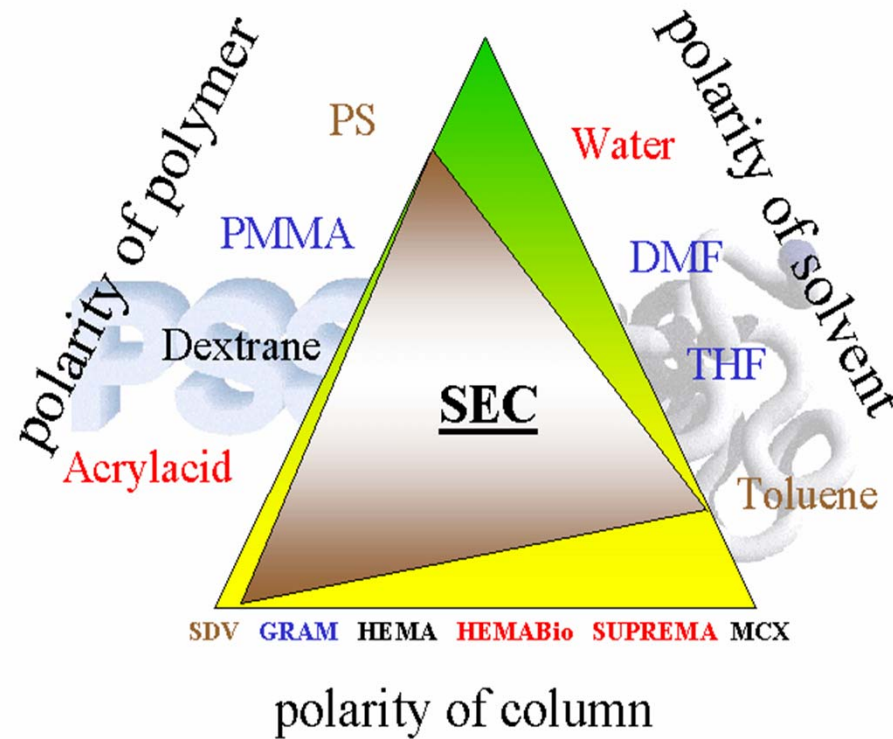
Principle:



short chains → long retention time
 long chains → shorter retention time

chromatogram: signal of a detector
 as a function of volume

No absolute method!
 Calibration is necessary.



Choice of column and solvent depends on polymer, which will be analyzed.

<http://www.polymer.de/bilder/secani.gif>

SEC-columns

Material:

Polystyrene (crosslinked with divinylbenzene)

Silica

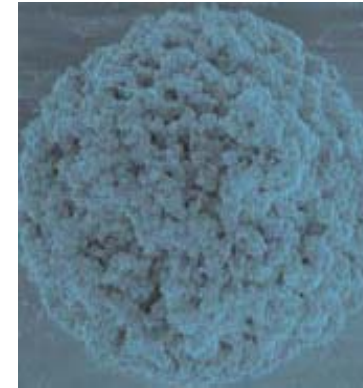
Glucose attached to DVB polymer

Modified DVB polymer (-OH, -SO₃H, -F, -NH₂)

Monolithic columns

thin columns → analytical SEC

wider columns → preparative SEC



www.polymerlabs.com.

Detectors:

UV/vis spectrometer: Polymer or some functional groups absorb light
very sensitive and selective detector
one specific wavelength (254 nm → aromatic groups in the polymer)

Diode array: Detection by several diodes (light is dispersed in front of diodes → UV-vis spectrum of eluted polymer)

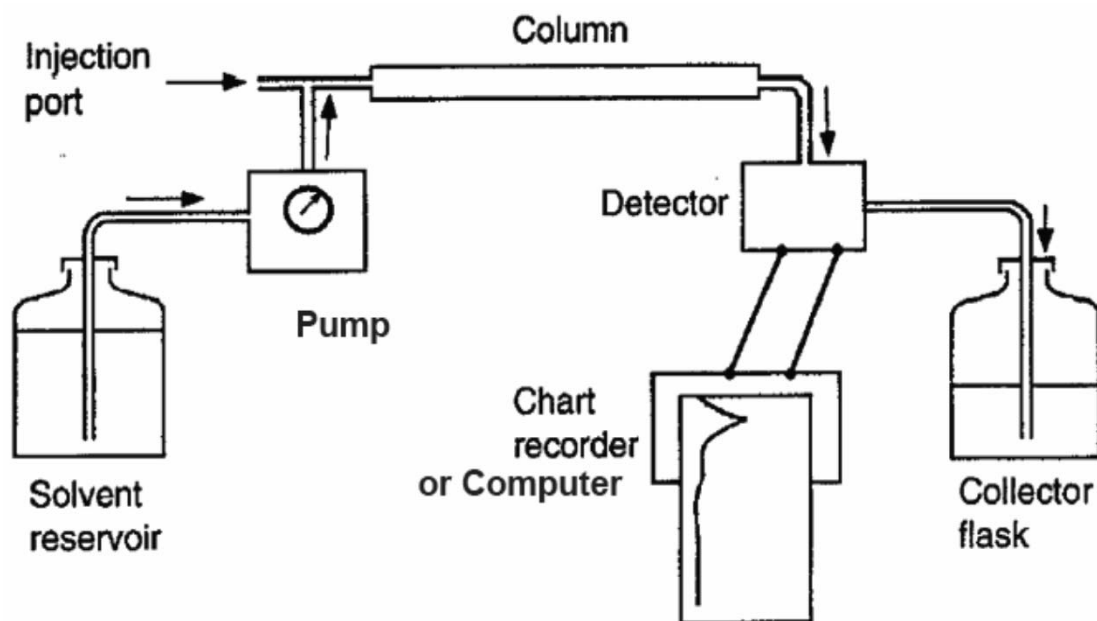
Differential-refractometer RI: Refractive index of solution and the one of the pure solvent are compared; applicable to most polymers; strong dependence on the pressure and temperature → exact temperature of column and detector

Viscosity detector: Universal calibration

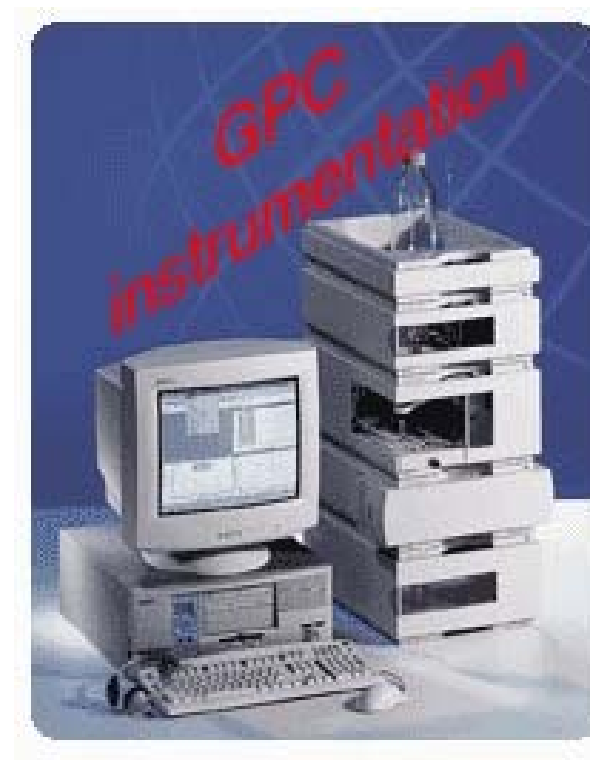
Light scattering LS: Very sensitive for high molar mass compounds
RI proportional to c , LS proportional to $c \cdot M$

Evaporative light scattering detector ELSD: Solvent emerging from column is evaporized in a stream of air, solute does not evaporate and is nebulized → formation of small droplets → scattering of light → amount of scattered light is proportional to concentration of material

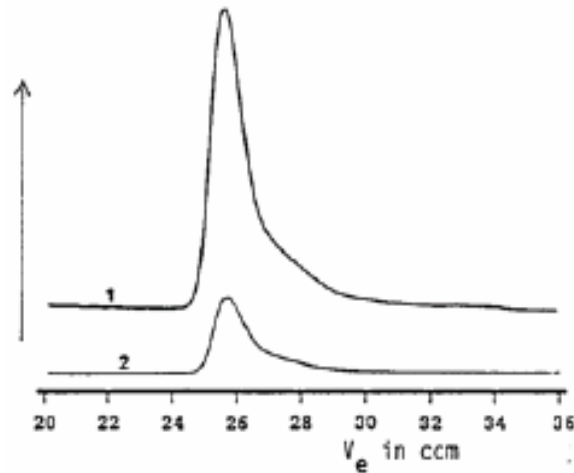
SEC-setups



schematic set-up

real set-up:
3 columns
autosampler

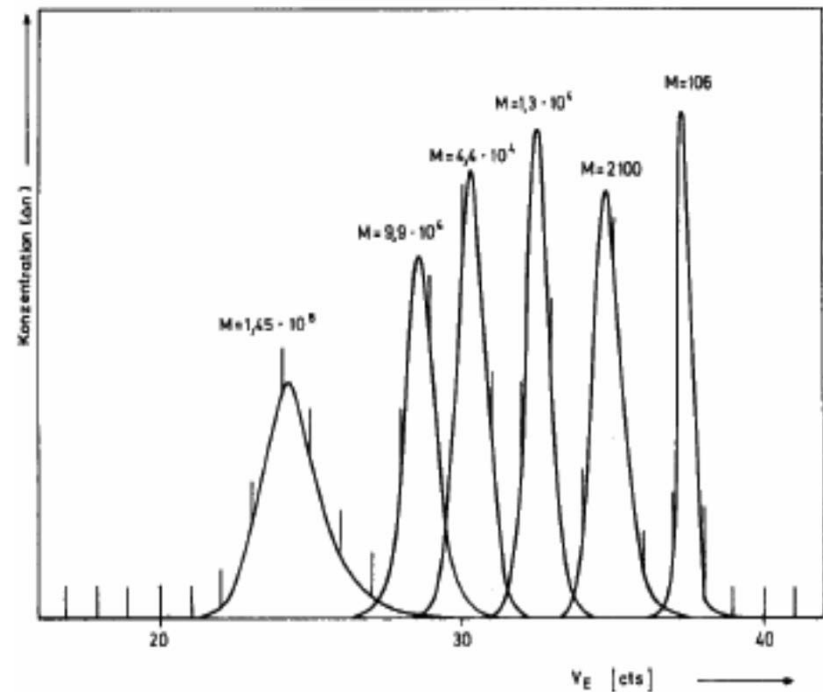
Calibration



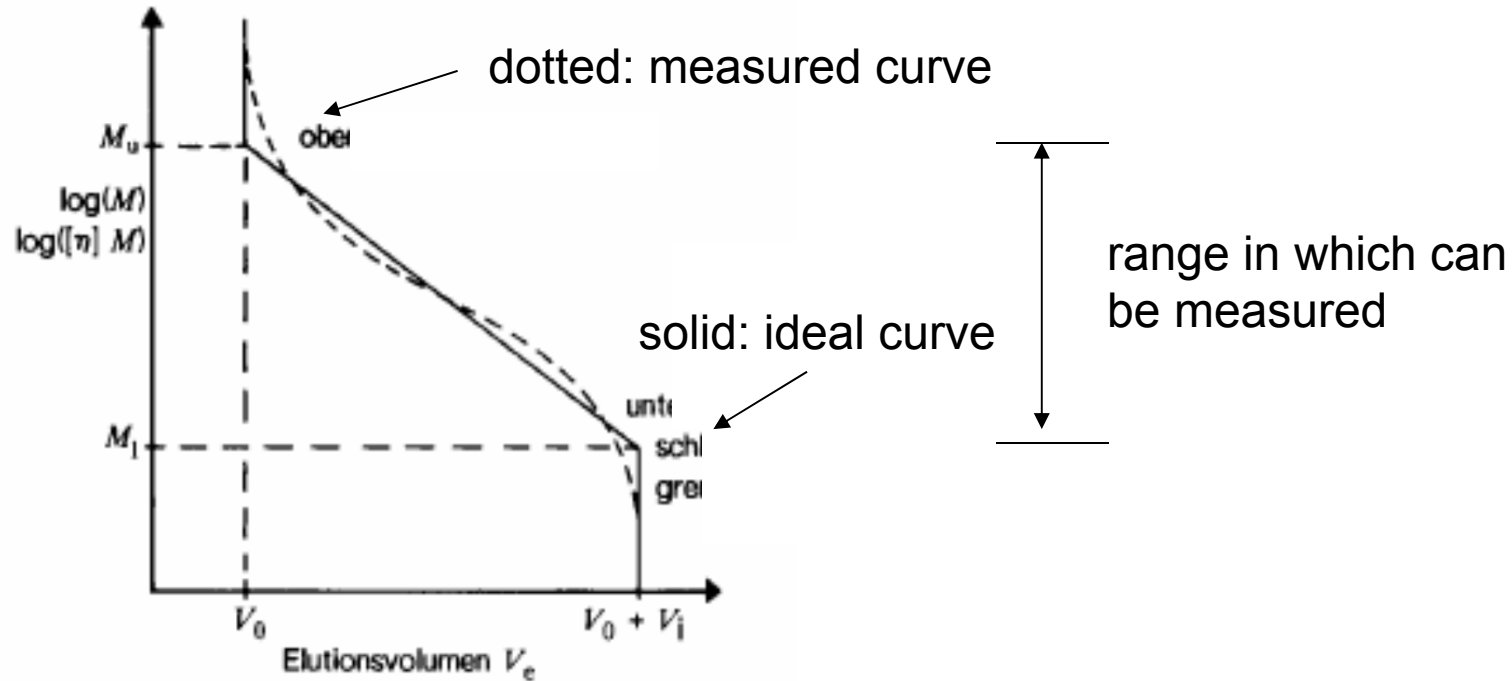
detector signal is proportional
to polymer concentration
 $c(1) > c(2)$
→ calibration

well known polymers (narrow distribution) are used as standards
molar mass is compared with eluted volume V_E

SEC curves of different polystyrenes and ethylbenzene in THF

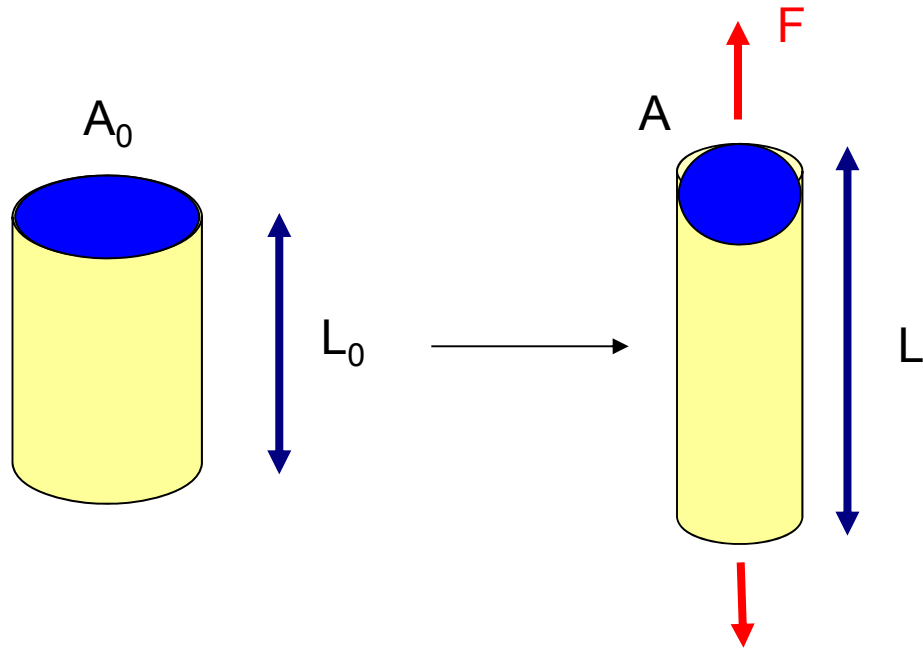


Calibration



upper exclusion limit: no more pores accessible
 lower exclusion limit: all pores accessible

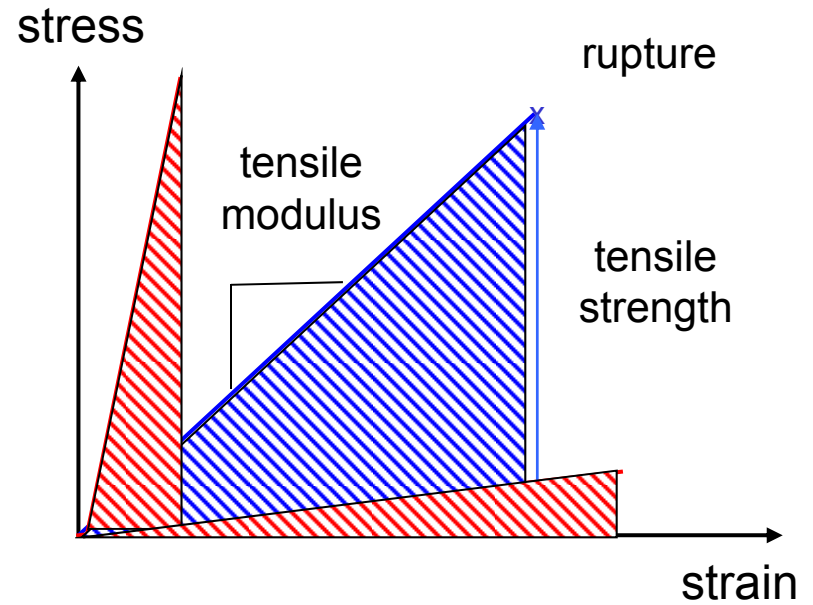
Calibration is only valid for the polymer used for calibration!



tensile load \rightarrow stretching in direction of load
 \rightarrow narrowing in perpendicular plane

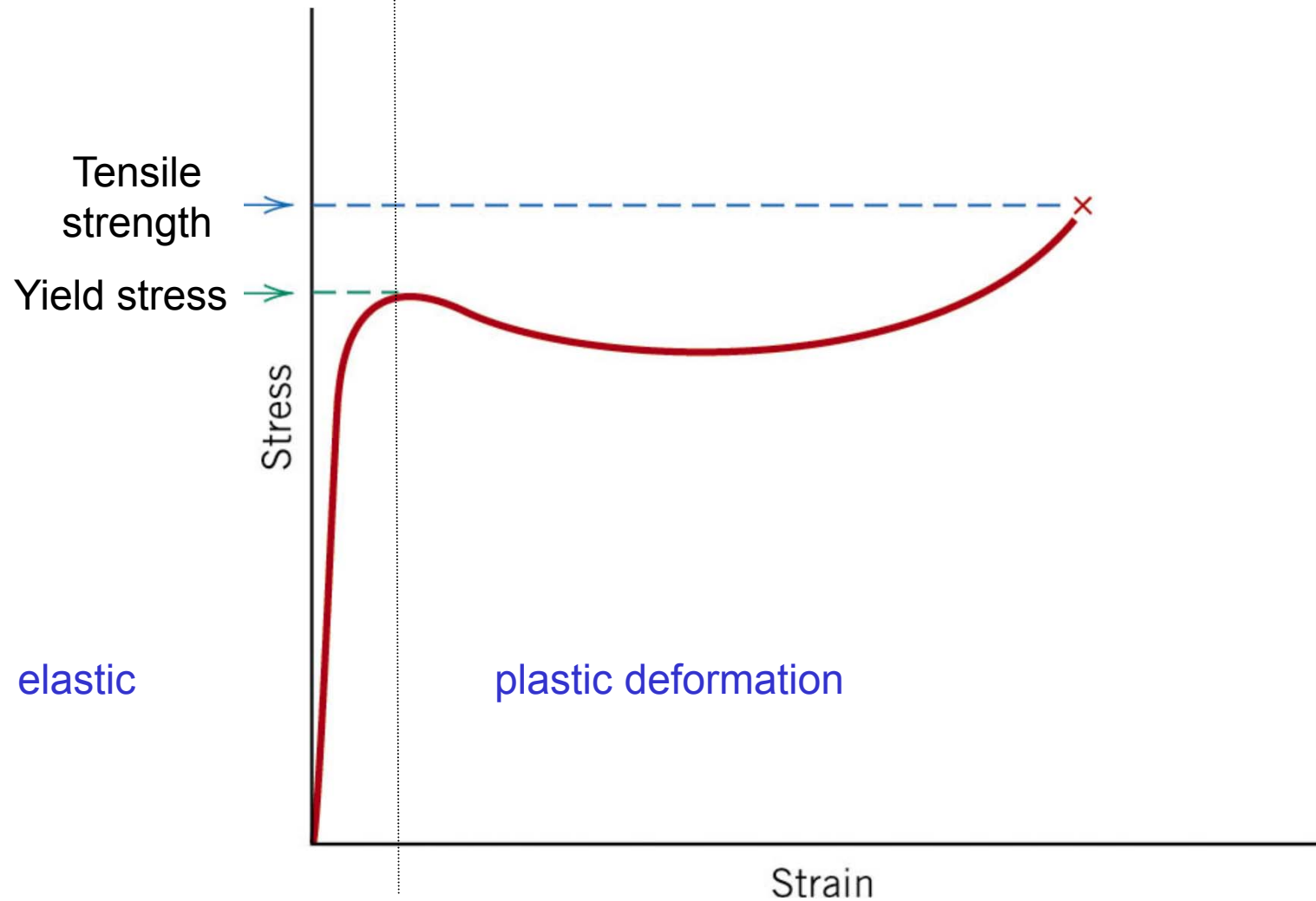
Stress = F/A
Strain = change of length/initial length

Stress-strain-curve

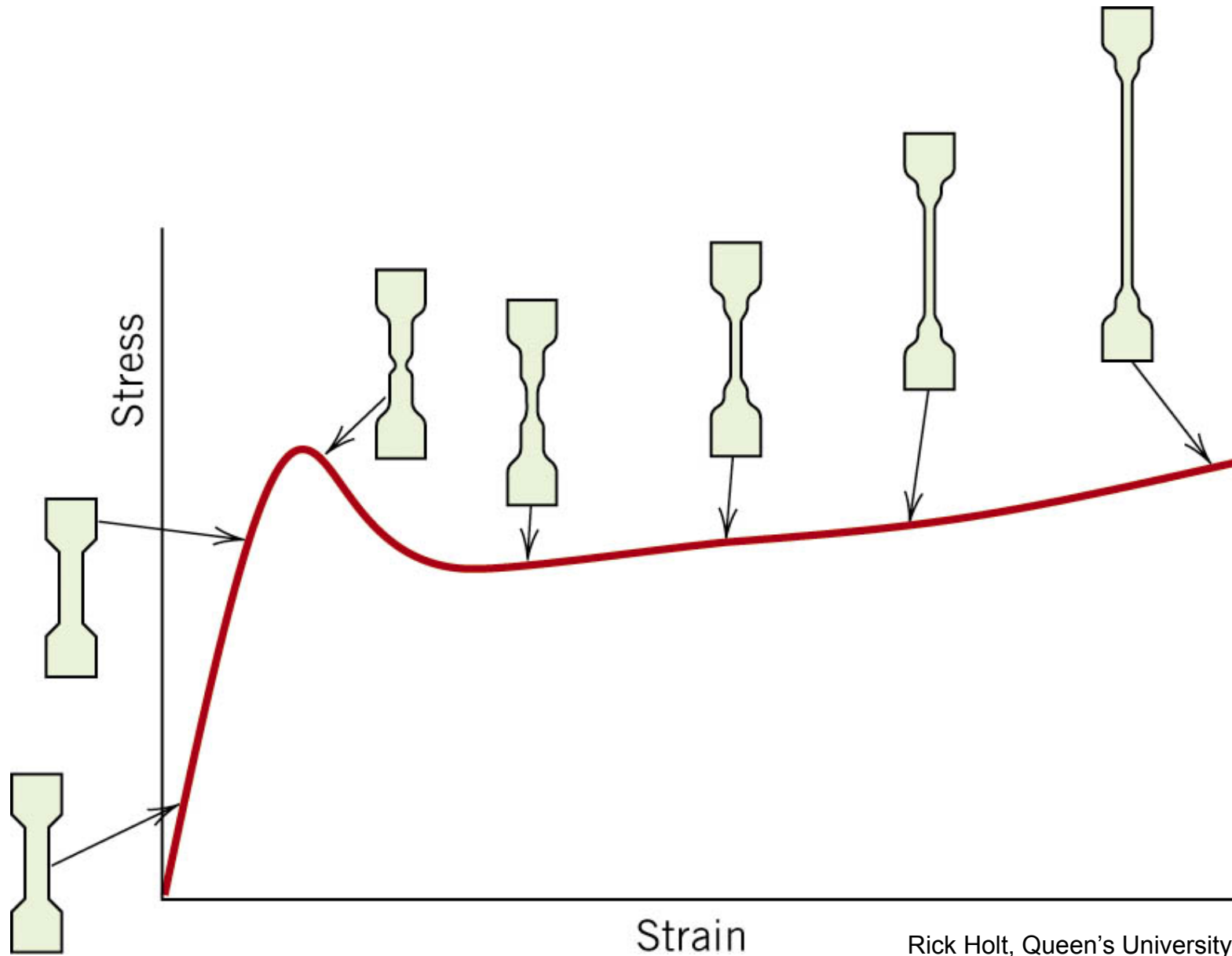


tensile modulus = Young's modulus

A real stress-strain curve of a polymer

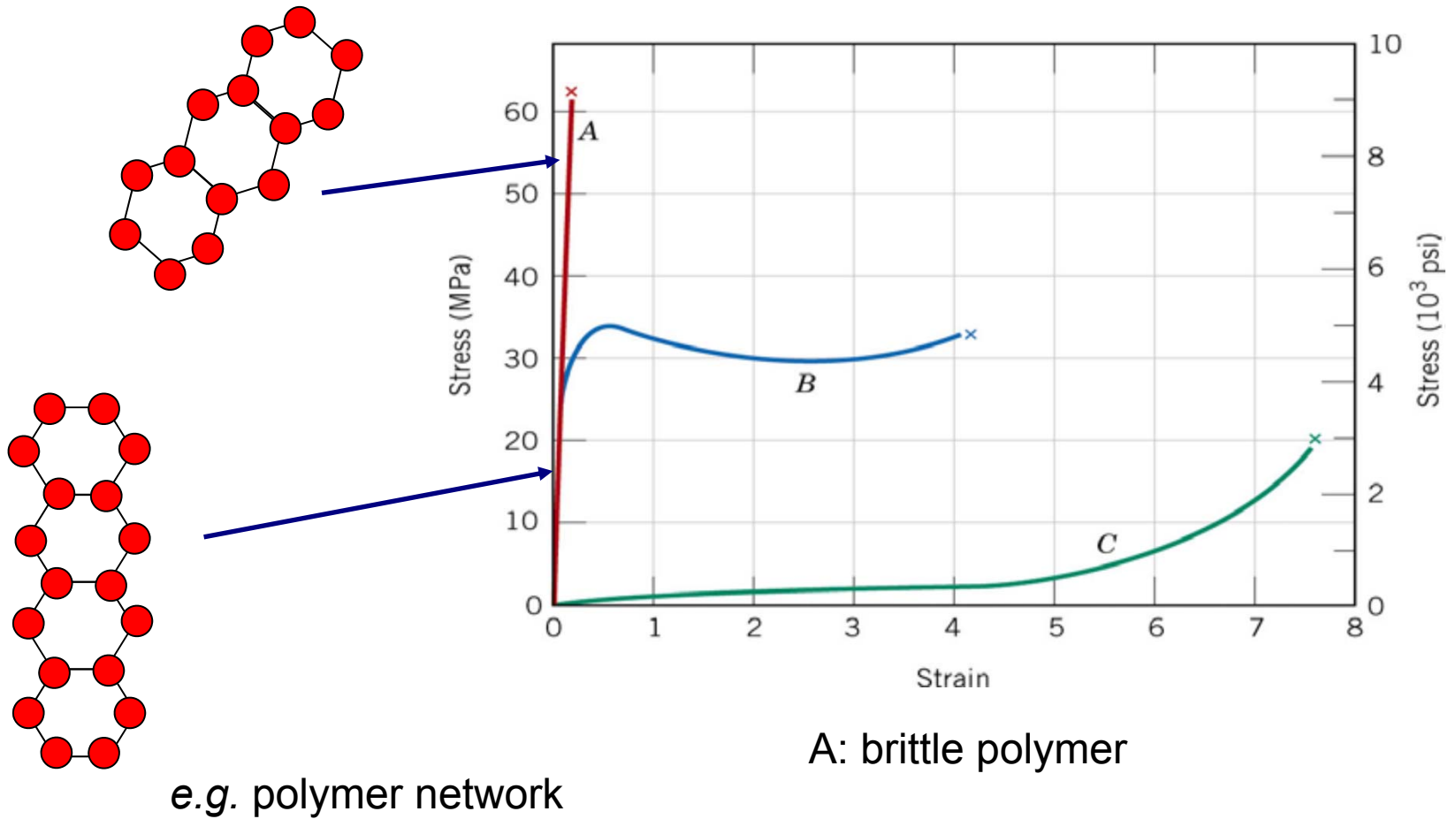


Rick Holt, Queen's University, Kingston.

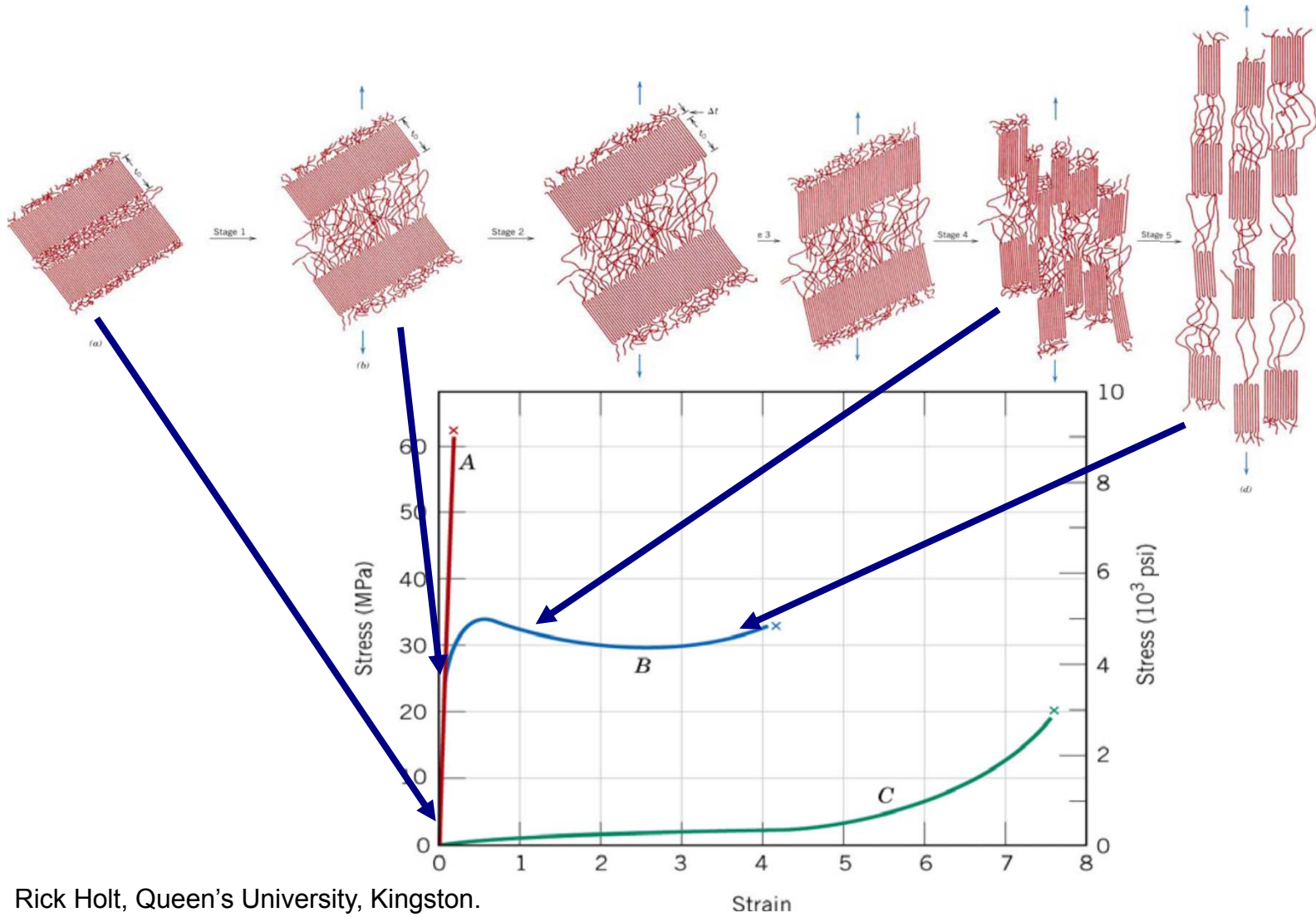


Rick Holt, Queen's University, Kingston.

What is happening on the molecular level?

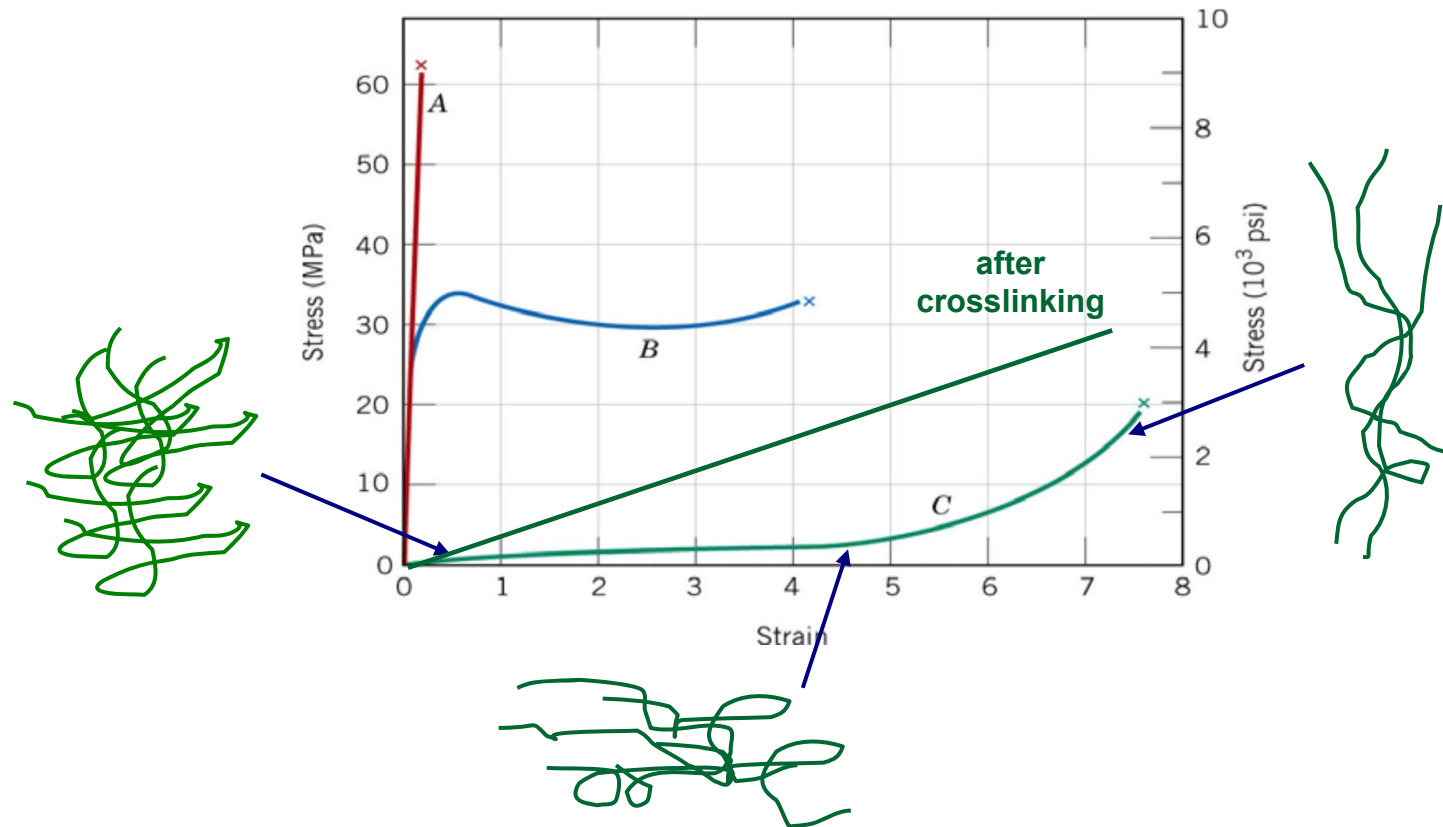


What is happening on the molecular level?

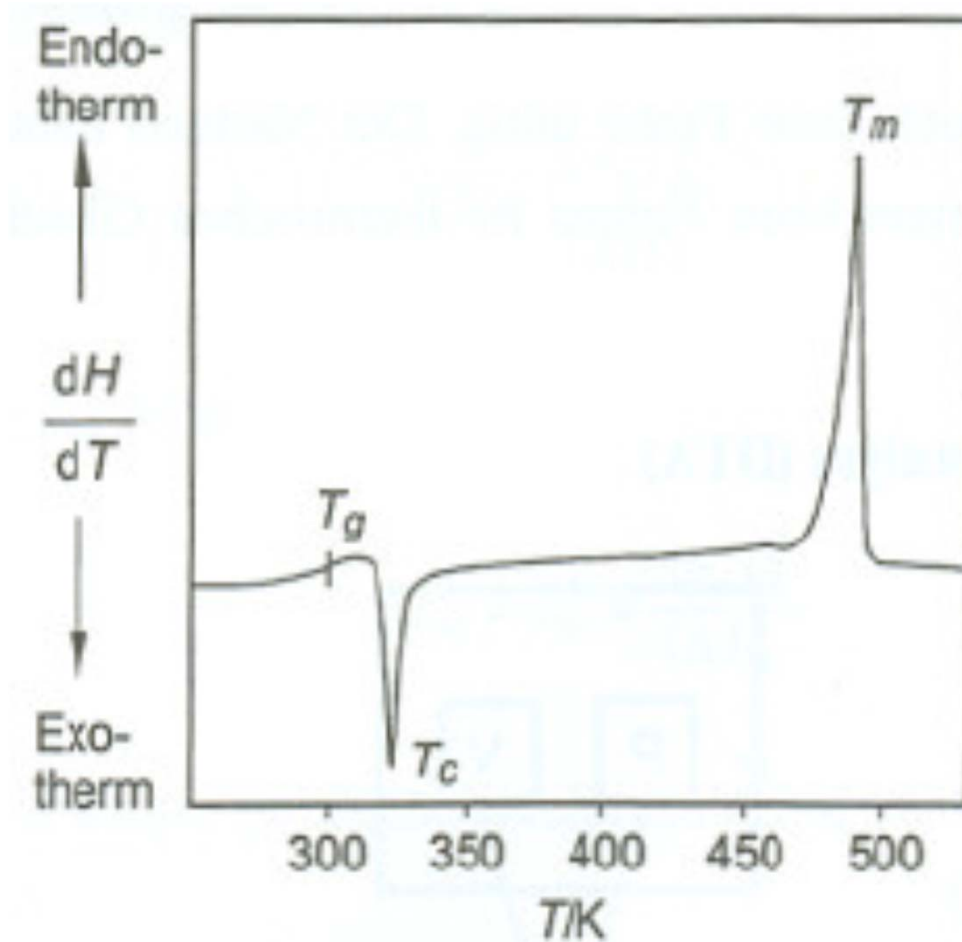


Rick Holt, Queen's University, Kingston.

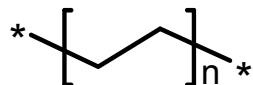
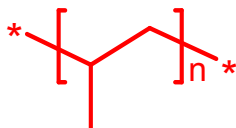
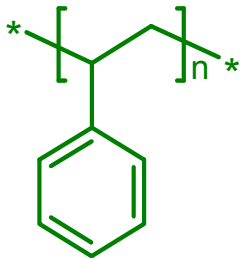
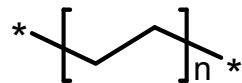
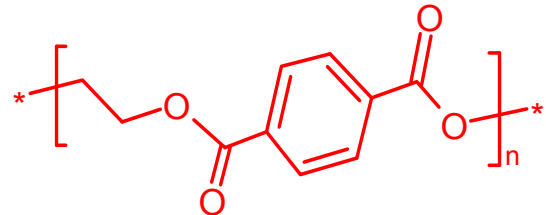
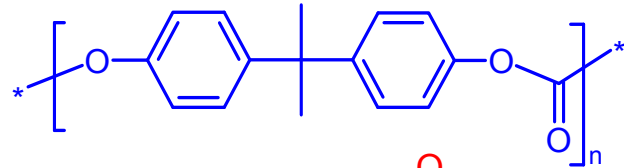
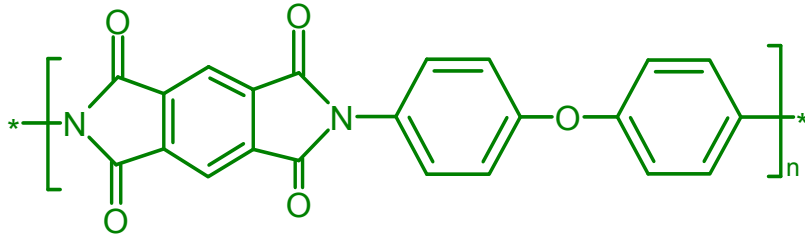
What is happening on the molecular level?



DSC Diagram



T_g – glass transition
 T_c – crystallization temperature
 T_m – melting temperature



More flexible chain → lower T_g

Lower main chain stiffness

Polyimide > 400 °C

Polycarbonate 150 °C

Polyethyleneterephthalate 70 °C

Polyethylene -120 °C

Chemical cross-linking increases T_g

Smaller side group

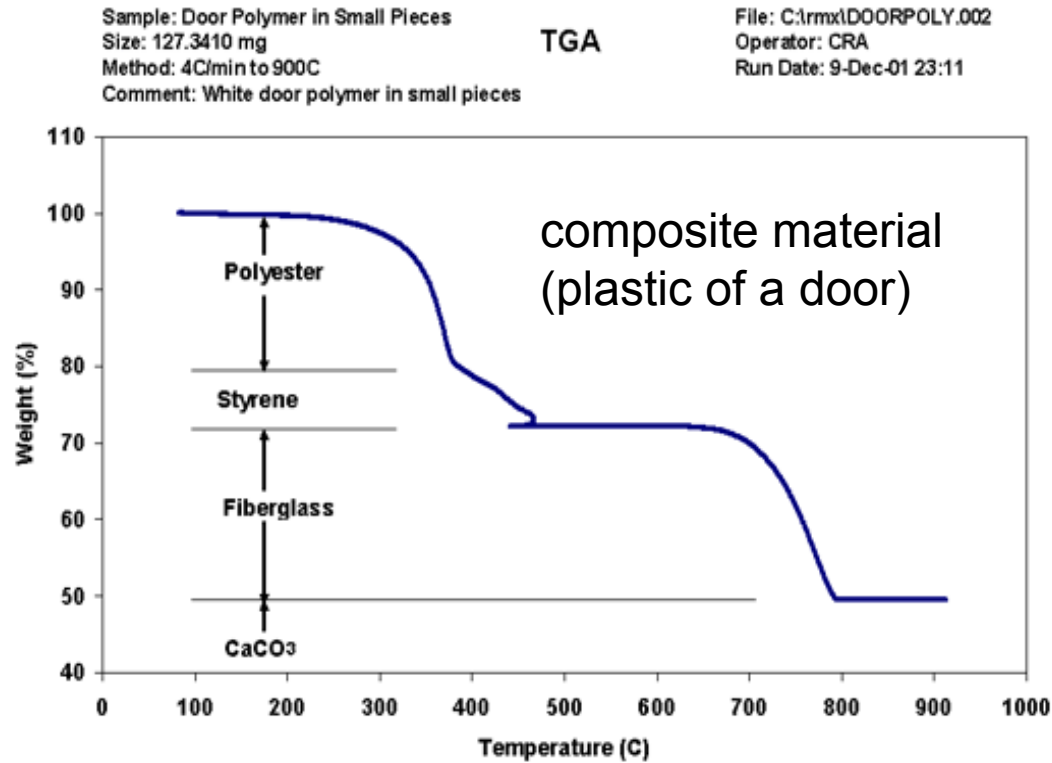
Polystyrene 100 °C

Polypropylene 15 °C

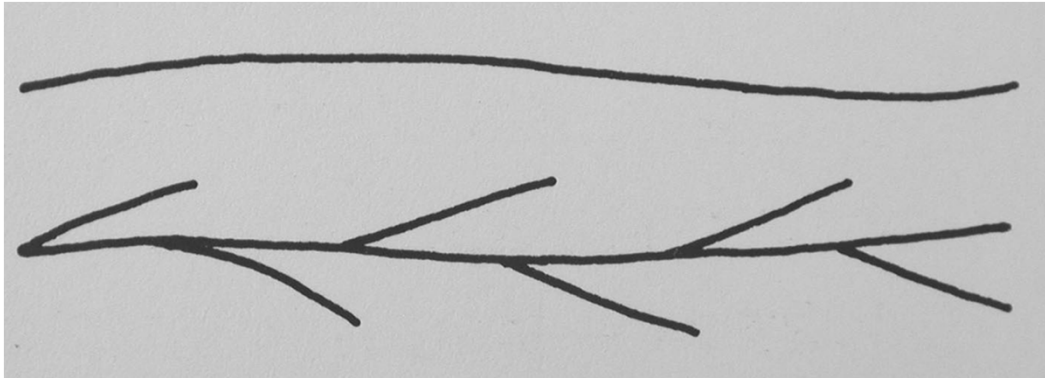
Polyethylene -120 °C

Thermogravimetric analysis (TGA)

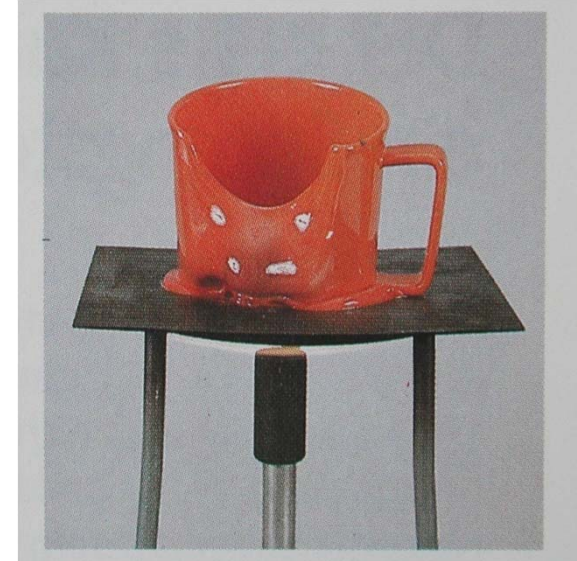
- thermal stability of polymers
- measurement weight loss against temperature
- residual solvent
- thermal decomposition



Thermoplastics

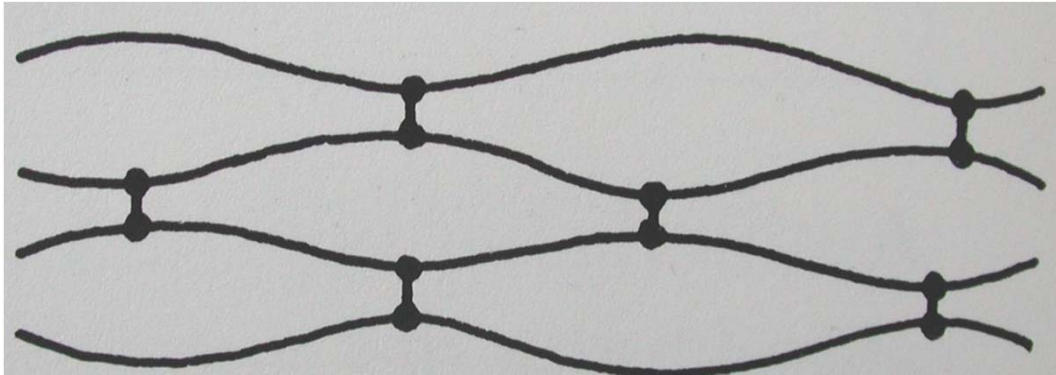


linear, branched polymers

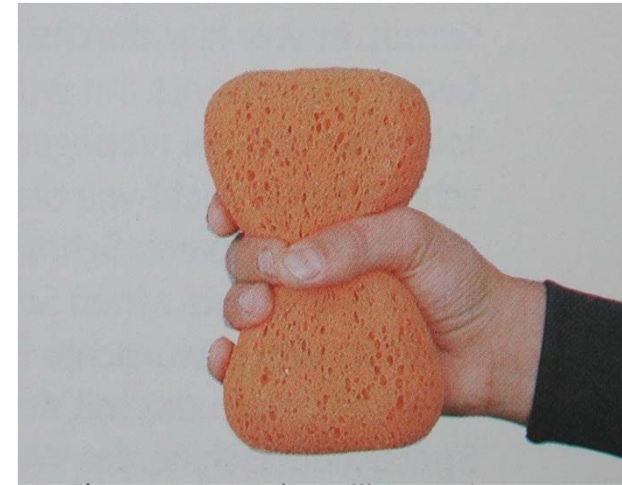


Thermoplastics above T_g and T_m are capable of flowing
Manufacture by extrusion and injection moulding

Elastomers



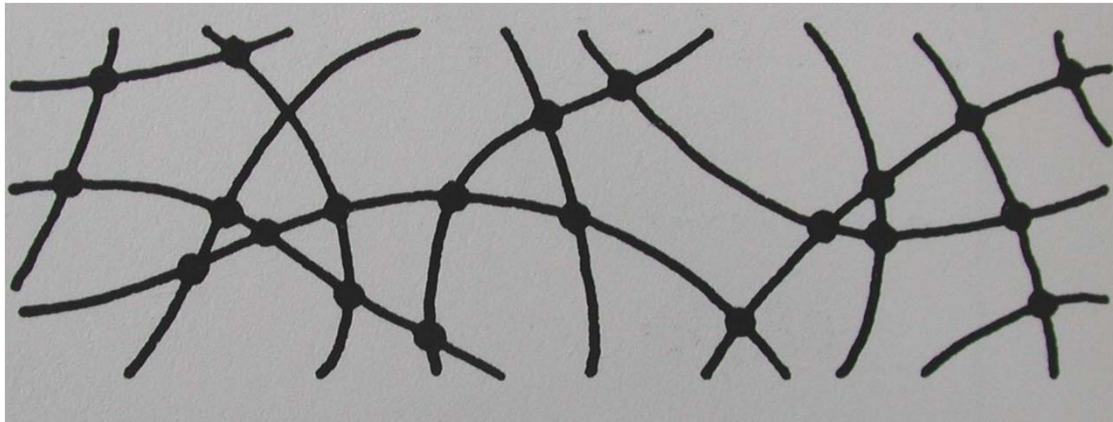
crosslinked polymers (wide meshed)



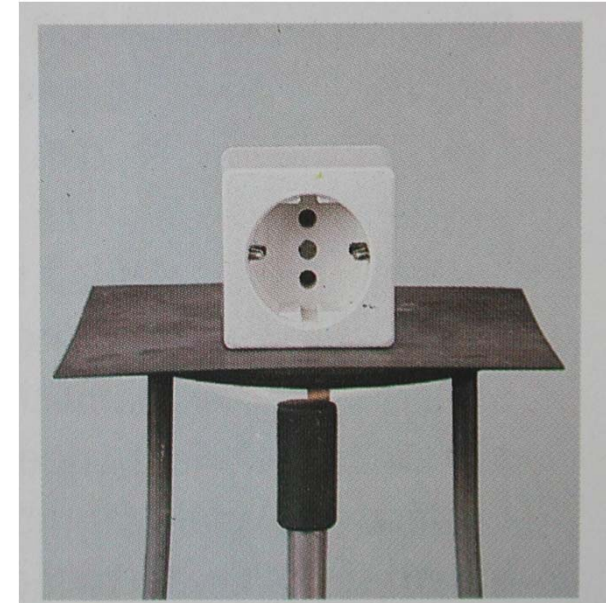
elastomers are swellable and elastic like rubber

rigid, but elastically deformable $T_g < \text{room temperature}$

Duroplastics



crosslinked polymers



Manufacture only by casting of monomers/oligomers
→ Thermal crosslinking (polymer insoluble)