



THERMAL AND ELECTRICAL BEHAVIOUR OF THE SUPERPROTONIC CONDUCTOR PHASE IN Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5}

Noura NOUIRI, Khaled JAOUADI*, Tahar MHIRI, Nabil ZOUARI

Laboratoire physico-chimie de l'état solide, Faculté des Sciences de Sfax, BP 1171, 3038 Sfax, Tunisie

nouirinoura@yahoo.fr

Laboratoire physico-chimie de l'état solide, Faculté des Sciences de Sfax, BP 1171, 3038 Sfax, Tunisie

khaledjaouadi@yahoo.fr

Laboratoire physico-chimie de l'état solide, Faculté des Sciences de Sfax, BP 1171, 3038 Sfax, Tunisie

taharmhiri@yahoo.fr

Laboratoire physico-chimie de l'état solide, Faculté des Sciences de Sfax, BP 1171, 3038 Sfax, Tunisie

bizrirl@yahoo.fr

ABSTRACT

Crystals of a new compound with a superprotonic phase transition $Rb_2(HSeO_4)_{1.5}(H_2AsO_4)_{0.5}$ (noted RbHSeAs), were synthesized by slow evaporation of an aqueous solution at room temperature. The differential scanning calorimetric analyses showed two endothermic peaks at 465 K and 566 K. The last peak corresponds to the decomposition of the material. The first transition was characterized by several techniques (impedance spectroscopy, complex modulus, Raman and X-ray diffraction powder depending on temperature). ac impedance measurements revealed that, upon heating, the compound undergoes at ~ 443 K a sharp increase in conductivity from a low temperature protonic phase to a superprotonic conductivity phase. The activation energies calculated from the modulus (ΔE_f) and impedance (ΔE_{σ}) spectra respectively are approximately equal, suggesting that the transport properties in this material above and below the superprotonic phase transition (443 K) are probably due to an H⁺ protons hopping mechanism.

Indexing terms/Keywords

Inorganic materials; Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5}; Thermal behavior; Raman spectroscopy; Superprotonic phase transition

Council for Innovative Research

Peer Review Research Publishing System

Journal: Journal of Advances in Chemistry

Vol. 10, No. 7 editorjaconline@gmail.com www.cirjac.com

2967 | Page



1. INTRODUCTION

The compounds of the general formula $M_2(HXO_4)_{(2-x)}(H_2AO_4)_x$, where (M = K, Rb and Cs; X= S, Se and A = P, As) are of interest because their high-temperature phase exhibits an unusually high conductivity ranging from 7×10^{-4} to $37 \times 10^{-3} \Omega^{-1} \text{ cm}^{-1}$ [1-8]. Hence they belong to the family of superionic-protonic conductors, which are of great interest because of their potential use in various electrochemical devices such as batteries, fuel cells, electrochemical sensors and electrochemical reactors [9-12]. The most interesting result was obtained for the title compound because its superconducting transition temperature is lowered to 443 K.

In our previous work, we have studied the superprotonic transport and relaxation properties and we describe the thermal behaviour by differential scanning calorimetric and thermo-gravimetric analyses of the rubidium selenate-arsenate solid acid family $Rb_2(HSeO_4)_{1.5}(H_2AsO_4)$. Accordingly, a detailed analysis of the frequency and temperature dependence of the ac conductivity data is necessary in order to characterize the microscopic mechanisms and the accompanying relaxation phenomena of the charge carrier transport.

2. EXPERIMENTAL

2.1. Crystal growth

Crystals of RbHSeAs were grown from aqueous solutions containing rubidium carbonate Rb_2CO_3 , selenic acid H_2SeO_4 and arsenic acid H_3AsO_4 in molar ratio (Rb:Se:As = 4:3:1), as reported earlier [13], according to the following reaction:

2 Rb₂CO₃ + 3 H₂SeO₄ + H₃AsO₄ \longrightarrow 2 Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5} + 2 H₂O + 2 CO₂

Slow evaporation of water under ambient conditions yielded transparent and colorless single crystals of Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5}. The formula of this material was determined by chemical analysis and confirmed by structural refinement [13].

2.2. Thermal behavior

The differential scanning calorimetry analysis was performed with a METTLER DSC131 instrument for temperatures ranging from 298 to 673 K with a heating rate of 5 K.min⁻¹. A polycrystalline sample of 34.72 mg was placed in hermetic aluminum cell in nitrogen atmosphere.

Thermo-gravimetric measurements were performed with a SETARAM (TG-ATD 92) thermo flex instrument under flowing air, with a heating rate of 5 K.min⁻¹ from 298 to 673 K. A powdered sample, 57.28 mg, was spread evenly in a large platinum crucible to avoid mass effects.

Temperature-dependent X-ray powder diffraction (TDXD) was performed with a powder diffractometer combining the curved-position-sensitive detector (CPS120) from INEL and a high temperature attachment from Rigaku. The detector was used in a semi-focusing arrangement by reflection (Cu K α 1 radiation, λ = 1.5406 Å) as described elsewhere [14].

2.3. Impedance measurements

The analysis of the frequency response of ac conductivity data was used to determine the superprotonic behaviour of $Rb_2(HSeO_4)_{1.5}(H_2AsO_4)_{0.5}$. Complex impedance measurements were performed using a compressed pellet of ~ 8 mm diameter and 0.5 mm thickness sintered at 298 K. Silver paint was evenly applied on both sides of the pellet for better electrical contact, the sample was then held between two spring-loaded electrodes. The electrical properties were collected under stagnant air atmosphere and determined by impedance and modulus method using a frequency response analyser (Hewlett - Packard 4192 A LF). The impedance $|Z^*|$ and phase angle θ were measured with a computer interfaced. The frequency range was 100 Hz - 13 MHz and measurements were carried out at constant temperature ranging from 295 K to 503 K and maintained at ±1 K accuracy by a Herrmann - Moritz 28480 controller for half an hour before collecting data.

2.4. Raman spectroscopy

Raman spectra of the crystals selected in a cell were obtained in the range 300 - 523 K on a computerized RTI 30 Dilor triple monochromator. The 514 nm radiation from a spectra-physics model 2000 argon ion laser was used for excitation with a powder between 0.1 and 0.2 W. A home-built furnace was employed for high-temperature experiments. The resolution of the spectrometer was between 0.5 and 2 cm⁻¹ and the accuracy of the temperature measurements was about 2 K.

3. RESULTS AND DISCUSSION

3.1. Structural properties

The structural properties [13] of the crystals were characterized by X-ray single analysis (performed at room temperature), which revealed that (RbHSeAs) crystallizes in space group P -1 with lattice parameters: a = 7.6950(5)Å, b = 7.9317(5)Å, c = 8.0392(4)Å, α = 70.249(4)°, β = 87.851(4)° and γ = 86.622(4)°. This compound has a unit cell volume 460.92(7) Å³ and two formula units per cell, giving a calculated density of 3.302. The structure was solved from 3519 independent reflections and refined with 110 parameters yielded weighted residuals of 0.0898 and 0.0502 based on F² and F values, respectively. The crystal structure of (RbHSeAs) projected onto the *bc* plane is shown in Fig. 1. This



structure is characterized by $HSeO_4^-$ and disordered $(H_xSe/As)O_4^-$ tetrahedra connected to dimmer via hydrogen bridges. These dimmers are linked and stabilized by an additional hydrogen bonds (O-H-O) and hydrogen bonds (O-H-··O) to built a chains of dimmers, which are parallel to the [0 1 0] direction Fig. 1. The rubidium atoms have a nine fold coordination.



3.2. Calorimetric and thermo-gravimetric analyses

The results of the high-temperature behavior of (RbHSeAs) are presented in Fig.2. The differential scanning calorimetric (DSC) revealed that (RbHSeAs) undergoes structural changes over the temperature 465 K which are unrelated to decomposition; its weight showed by the thermo-gravimetric curve (TGA) is stable to 480 K. This endothermic peak was also characterized by impedance and modulus measurements and was attributed to the superprotonic phase transition of (RbHSeAs) materials. The high-temperature phase of $Rb_2(HSeO_4)_{1.5}(H_2AsO_4)_{0.5}$ is thus 'superprotonic' in nature .

The decomposition of (RbHSeAs) occurs in two stages, Fig. 2. The first and the second weight loss regions occur over the temperature ranges 510 - 545 K and 570 - 650 K, respectively. Approximately 5 % of the mass is lost, which corresponds to 1.25 molecules of water per formula unit or complete dehydration. Those two weight losses are accompanied by one endothermic peak in the DSC curve at 566 K.

In order to get further information about the first phase transition revealed by the differential scanning calorimetric (DSC), we have undertaken X-ray powder diffraction, performed as a function temperature at T = 298 K, T = 473 K and T = 503 K (Fig. 3). We select the θ interval which shows significant changes with temperature variation. The results shown in Fig. 3, demonstrate that the compound Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5} reveal a structural transition upon heating to 473 K. Indeed, upon heating the samples from 298 to 473 K, the X-ray powder show the appearance of some reflections and the disappearance of others indicating sever changes in structure. This confirms the first transition observed at T = 465 K by



the DSC which is attributed to the superprotonic phase transition. The third powder diffractogram (at T = 503 K) contains the same reflections observed in that at T = 473 K, indicates that the title compound takes place at much higher temperatures, with an onset of approximately 510 K.



Figure 2: The differential scanning calorimetry (DSC) and the thermogravimetry (TG) of Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5}









3.3. ac conductivity behavior

Four different formalisms are generally employed for analyzing the ac response of materials namely: the complex dielectric constant ϵ^* , complex electric modulus M*, complex impedance Z* and complex admittance G*. We have used the Z*, ϵ^* and M* representations to analyse the conductivity behaviour.

The ionic conductors (ICs) with point defects lead to a conductivity ranging up to $10^{-5} \Omega^{-1}$ cm⁻¹ whereas the superionic conductors (SICs) result in a conductivity of at least $10^{-4} \Omega^{-1}$ cm⁻¹. The main difference between these two groups of materials concerns the activation energy (ΔE_{σ}): in the case of SICs, ΔE_{σ} is lower than 0.4 eV while in ICs, values varying between 0.6 and 1.2 eV are usually observed [15]. The superionic conductors have thus a high conductivity far below the melting point. This fundamental difference is due principally to the particular structures of SICs.

Temperature dependence of the protonic conductivity of (RbHSeAs) was determined from the analysis of the impedance spectroscopy at different temperatures. Complex plane plots of the complex impedance diagrams, - Z" versus Z', at selected temperatures are shown in Fig. 4(a and b) in phases low and high temperatures, respectively. These impedance spectra show one non-ideal semicircle arc, which was attributed to the bulk boundary properties and whose center is displaced below the real axis. This shows that (RbHSeAs) follows the Cole-Cole law and their extrapolation gives rise to an [$\alpha(\pi/2)$] dispersion angle where $\alpha = 0.133$ is an empirical parameter ($0 \le \alpha \le 1$) proportional to the degree of deviation from the Debye model. The bulk ohmic resistance relative to each experimental temperature is the intercept on the real axis of the zero-phase angle extrapolation of the highest-frequency curve. The observed non-ideal semicircle was modeled using an equivalent circuit that contains one sub-circuit which consists of a resistance Rg, a capacitance Cg and a constant phase element (CPE) connected in parallel which represents the grain response of the sample. These curves also show the temperature dependence of the resistance proving the superprotonic conduction properties of (RbHSeAs).







The temperature dependence of the conductivity between 333 and 503 K is represented in Fig. 5, in a log(σ T) vs 1000/T plot. Below 443 K, the protonic conductivity can be described well by the Arrhenius relation: $\sigma T = \sigma_0 \exp(-\Delta E_{\sigma}/kT)$ (1) with an activation energy $\Delta E_{\sigma} = 0.54 \text{ eV}$, where σ_0 is the pre-exponential factor, ΔE_{σ} is the activation energy for ion migration and k is Boltzmann's constant. At the temperature 443 K, the conductivity jumped from $1.52 \times 10^{-4} \ \Omega^{-1} \text{cm}^{-1}$ to $0.87 \times 10^{-3} \ \Omega^{-1} \text{cm}^{-1}$ at 473 K, characterizing the superprotonic conduction phase of the salt. The activation energy decreases from 0.54 eV at low temperature to 0.32 eV in the superionic phase, also such behavior shows the superprotonic conduction of $\text{Rb}_2(\text{HSeO}_4)_{1.5}(\text{H}_2\text{ASO}_4)_{0.5}$ compound [15]. The transition observed at 443 K, corresponds to the structural transformation between the triclinic phase and the superprotonic phase. Therefore, this transition which revealed by thermo-differential measurement, is well related to the hydrogen bonds in which the proton sublattice reaching the 'quasi-liquid' state where protons of the SeO₄⁻² and AsO₄⁻³ tetrahedral ions contribute to the unusually high conductivity as in the case of CsHSO₄ [16]. This behavior is characteristic of a plastic phase and implies a 'free' rotation of HSeO₄⁻ ions on given sites. In the superionic phase a 'quasi-liquid' state is manifested and both protons and rubidium ions are disordered while in low temperature, the disorder is essentially confined to the protons. The drastic increase in conductivity by almost ten orders of magnitude on going from 443 K to 473 K, is thus related to the high disorder of both sublattices.



An analysis of the ion conductivity relaxation process in (RbHSeAs) has been undertaken in the complex electric modulus formalism, ($M^* = M' + jM''$ (2)). This formalism is useful in determining the charge carrier parameters such as the conductivity relaxation time [17]. For a temperature and a frequency given, the real part, M', and the imaginary part, M'', of the M* complex modulus have been calculated from the complex impedance data ($Z^* = Z' - jZ''$ (3)) by the following relations: $M' = \omega C_0 Z''$ (4) and $M'' = \omega C_0 Z''$ (5). Isothermal frequency spectra, during heating process of the real, log M' and the normalized M''/M''max imaginary part of the electric modulus for (RbHSeAs) versus log f are displayed in Fig. 6 and Fig. 7, respectively. It is noticed that for all temperatures given, the real modulus M' tends to a frequency-independent constant M'[∞] at high frequencies. At low frequencies, it decreases sharply (Fig. 6), which indicates that the electrode polarization phenomena make a negligible contribution to M* and may be ignored when the electric data are analysed in this form [17].

The M"/M"max spectrum relative to a given temperature shows an asymmetrical peak approximatively centered in the dispersion region of M' (Fig 7). The region to the left of the peak is where the H⁺ protons are mobile over long distances whereas the region to the right is where the ions are spatially confined to their potential wells. It is clear from this figure that the maximum of the asymmetric peak shifts toward higher frequencies as the temperature is increased. The frequency range where the peak occurs is indicative of the transition from long-range to short-range mobility at increasing frequency and is defined by the condition $\omega \tau_{\sigma} \approx 1$ where τ_{σ} is the most probable proton relaxation time [18].

The M"/M"max curves are asymmetric, in agreement with the non - exponential behavior of the electrical function, which is well described by the empirical stretched exponential Kohlrausch function $\varphi(t) = \exp\left[-\left(t \ / \tau_{\sigma}\right)^{\beta}\right]$ (6) where β (0 < β < 1) is the stretched exponential parameter, [19-21]. The full width at half-maximum (FWHM) of the M"/M"max



spectrum is greater than the breadth of a Debye – peak (1.14 decades) [22] and it results with an average value of β = 0.84 for the Kohlrausch parameter. This value of β clearly indicates that the relaxation is non-exponential.





In Fig. 5, we introduce the temperature dependence of log (fp), where $f_p = 1/2\pi\tau_{\sigma}$ (7) is the frequency relative to M"max peak corresponding to the bulk relaxation. For all given temperatures, an Arrhenius - type law is shown with a noticeable break observed at about 443 K. Above this temperature the activation energy value is 0.29 eV. This result confirms the presence of the superprotonic transition already observed by measurements of conductivity, D.S.C. and T.G.A. approximately at the same temperature. At low and high temperatures, both lines of the conductivity log (σ T) and the modulus peak maxima log (fp) observed in the temperature studied are quasi - parallel, the activation energies



deduced from the impedance (ΔE_{σ}) and modulus (ΔE_{f}) spectra are very close ($\Delta E_{\sigma(I,T)} = 0.54 \text{ eV}$; $\Delta E_{f(I,T)} = 0.55 \text{ eV}$), suggesting that the H⁺ protons transport in (RbHSeAs) is probably due to a hopping mechanism [23].

On the other hand, the value of the full width at half-maximum (FWHM) of peaks (Fig. 7) corresponding to various temperatures is approximately close to an average value of 1.35 decades. Consequently, β may be considered as independent of temperature in the studied temperature range. The value of the β parameter, smaller than 1, can be attributed to the existence of a distribution of relaxation times in the compound. Such an interpretation has been adopted for many solid electrolytes [24, 25]. On the contrary, recent studies based on the overlapping of the log M' vs log f and log M' vs log f plots obtained for various temperatures have shown that this interpretation was questionable [18, 26].

3.4. Vibrational studies

3.4.1. Assignments of Raman bands at room temperature

To gain more information on the superprotonic phase transition, we have undertaken a vibrational study using Raman scattering. The tentative assignments of the vibrations are based on a comparison of Raman spectra of (RbHSeAs) with other spectra known from the literature [3, 4, 27, 28].

Raman spectra of both internal and external modes of polycrystalline samples of $Rb_2(HSeO_4)_{1.5}(H_2ASO_4)_{0.5}$ were recorded at different temperatures 300, 453 and 523 K (Figs 8 (a, b)). This study was restricted to the 10–1200 cm⁻¹ spectral region because it included the lattice vibrations and u(X-OH) and u(X-O) (where X = As, Se) stretching modes, which are the most sensitive to phase transition and crystal changes [29]. The observed Raman bands at selected temperatures and their assignments are given in Table 1. In the Raman spectrum, the weak peaks at low frequencies (<80 cm⁻¹) are assigned to translational vibrations of the Rb⁺, SeO₄²⁻ and AsO₄³⁻ ions and those at 95, 164 cm⁻¹ involve respectively the deformations and elongations of the hydrogen bonds $\delta(O...O)$ and v(O...O). The bands in the 282-304 cm⁻¹ and 320-417 cm⁻¹ range are assigned to symmetric and antisymmetric deformation vibration of the XO₄ group respectively (where X = As, Se), and those in the range 744-860 cm⁻¹ and 924-933 cm⁻¹ (Fig. 8) are assigned to symmetric and antisymmetric elongation vibration of the XO₄ group respectively.

3.4.2. Superprotonic phase transition

The spectral evolution illustrated in Figs 8 (a, b) characterizes clearly the superprotonic phase transition at about T = 453 K. Its transition temperature is close to that found by differential scanning calorimetry, X-ray powder diffraction and conductivity measurements. This transition is characterized by spectral changes in the spectral regions studied. Thus, the following changes are noted: (1) Both the internal and external modes show progressive broadening (Figs 6(a, b)). The large broadening of the external modes, namely that of SeO4⁻² and AsO4⁻³ motions (T), indicates weakening of the hydrogen bonds and the occurrence of a libration of H(Se/As)O4⁻².(2) The lines assigned to $v_1(XO_4)$ stretching decrease strongly in intensity at 744cm⁻¹ and in width at half height at 810 cm⁻¹.



Figure 8(a): Raman spectra in the range 10–600 cm⁻¹ for Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5} at various temperatures



T = 300K Raman (cm ⁻¹)	T = 453K Raman (cm ⁻¹)	T =5 23K Raman (cm ⁻¹)	Assignment
61	60	-	Lattice vibrations of
77	75	74	Rb^+ , SeO ₄ ²⁻ and AsO ₄ ³⁻
95	96	95	δ _{O···O}
164	165	-	<i>V</i> _O _O
282	284	283	<i>v</i> ₂ (XO ₄)
304	304	305	
320	320	321	
342	341	346	<i>v</i> ₄ (XO ₄)
373	373	374	
417	417	419	
744	745	743	
764	764	767	
810	810	812	<i>v</i> ₁ (XO ₄)
860	857	857	
924	924	923	<i>v</i> ₃ (XO ₄)
933	932	-	

Table1: Raman bands and their assignments of Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5} at selected temperatures



Figure 8(b): Raman spectra in the range 600–1200 cm⁻¹ for Rb₂(HSeO₄)_{1.5}(H₂AsO₄)_{0.5} at various temperatures



4.CONCLUSION

The physical properties and phase transitions of the $Rb_2(HSeO_4)_{1.5}(H_2AsO_4)_{0.5}$ compound were examined by different methods. The differential scanning calorimetry and thermo-gravimetric analyses (D.S.C-T.G.A.) showed that (RbHSeAs) undergoes structural changes over the temperature 465 K which is unrelated to decomposition; its weight showed by the thermo-gravimetric curve (T.G.A) is stable to ~ 480 K. This endothermic peak was also characterized by X-ray powder diffraction verses temperature and was attributed to the superprotonic phase transition of (RbHSeAs). The decomposition of $Rb_2(HSeO_4)_{1.5}(H_2AsO_4)_{0.5}$ occurs in two stages. Approximately 5 % of the mass is lost, which corresponds to 1.25 molecules of water per formula unit or complete dehydration. The weight loss is accompanied by one endothermic peak in the DSC curve at 566 K.

The analysis of the ac conductivity data and the modulus formalism support these results. From ac impedance measurements, both conductivity and modulus spectra of this compound were analysed at low and high frequencies in the 403 – 483 K temperature range. The relaxation conductivity is well described by the empirical, stretched exponential Kohlrausch function φ (t) = exp [- (t / $\tau_{\sigma})^{\beta}$]. The value of the β parameter ($0 < \beta < 1$) represents for the conductivity relaxation the departure from the simple exponential ($\beta = 1$). The f_p frequency relative to M"_{max} (f_p = 1/2 $\pi\tau_{\sigma}$) is defined by the condition $\omega\tau_{\sigma} = 1$, where τ_{σ} is the most probable ion relaxation time. f_p increases with increasing temperature and the temperature dependence of f_p is of Arrhenius type (f_p = f_p⁰ exp(- $\Delta E_f / kT$). Information about the charge carrier transport mechanism was obtained by comparison of ΔE_f with ΔE_{σ} . Below 443 K, the activation energies for the (RbHSeAs) compound calculated from the modulus and impedance spectra respectively, are approximately equal, suggesting that the protonic transport is probably due to a hopping mechanism. On the other hand, in the temperature range studied, β may be considered as independent of temperature. The value of this parameter, smaller than 1, shows the existence of a distribution of relaxation times in the (RbHSeAs) compound and confirms the validity of this model for proton conductivity.

The evolution of the Raman spectrum as a function of temperature characterizes clearly the transition superprotonic phase about T = 453 K. Its transition temperature is close to that found by differential scanning calorimetry, X-ray diffraction powder and conductivity measurements.

5. REFERENCES

[1] K. Jaouadi, N. Zouari, T. Mhiri, M. Giorgi, journal of Phase Transition, 78 (2005) 627.

- [2] K. Jaouadi, N Zouari, T. Mhiri, A. Daoud, M. Jannin, journal of Alloys and Compounds, 413 (2006) 46.
- [3] N Zouari, K. Jaouadi, T. Mhiri, journal of Solid Stat Ionics, 177 (2006) 237.
- [4] K. Jaouadi, N. Zouari, T. Mhiri, M. Giorgi, journal of Physics Procedia, 2 (2009) 1185.

[5] CRI. Chisholm, SM. Haile, Acta Cryustallographica, B55 (1999) 937.

- [6] CRI. Chisholm, SM. Haile, journal of Solid Stat Ionics, 136-137 (2000) 229.
- [7] A.I. Baranov, A.V. Tregubchenko, L.A. Shuvalov, N.M. Shechagina, Sov. Phys. Solid State, 29 (1987) 1448.
- [8] M. Boubia, M. T. Averbuch-Pouchot et A. Durif, Acta crystallographica, C41 (1985) 1562.
- [9] KD. Kreuer, journal of Mater Chemistry, 8 (1996) 610.
- [10] B. Zhu, I. Albirson G. Mong, journal of Solid State Ionics, 125 (1999) 439.
- [11] B. Zhu, journal of Solid State Ionics, 125 (1999) 397.
- [12] R. Brouchet, S. Miller, M. Duclot, journal of Solid State Ionics, 145 (2001) 69.
- [13] K. Jaouadi, M. Amri, N. Zouari, T. Mhiri, S. Pechev, journal of Materials Research Bulletin, Submitted (2014).
- [14] W. Rekik , H. Naili, T. Mhiri, journal of Organometallic Chemistry 691 (2006) 4725

[15] P. Colomban, Proton Conductors: Solids, Membranes, and Gels Materials and Devices; Chemistry of Solid State Materials, Cambridge University Press, (1992).

- [16] N.G. Hainovsky, E. Hairetdinov, Izv. Ot. SSSR Ser, Khim. Nauk, 8 (1986) 33.
- [17] A. Novak, Structure Bonding 18 (1974) 177.
- [18] H. K. Patel, S. W. Martin, Phys. Rev. B 45 (1992) 10292.
- [19] F. S. Howell, R. A. Bose, P. B. Macedo, C. T. Moynihan, journal of Phys. Chem. 78 (1974) 639.
- [20] G. Williams, D. C. Watts, Trans. Faraday Soc. 23 (1970) 625.
- [21] K. L. Ngai, S. W. Martin, Phys. Rev. B 40 (1989) 10550.
- [22] S. W. Martin, C. A. Angell, journal Cryst. Solids 83 (1986) 185.
- [23] B. V. R. Chowdari, R. Gopalakrishnan, journal of Solid State Ionics 23 (1987) 225.
- [24] J. Kawamura, M. Shimoji, journal of Mat. Chem. Phys. 23 (1989) 99.



[25] C. Liu, H. G. K. Sundar, C. A. Angell, journal of Solid State Ionics, 18 (1986) 442.

[26] S. W. Martin, Appl. Phys., A 49 (1989) 239.

[27] M. Amri, K. Jaouadi, N. Zouari, T. Mhiri, F. Mauvy, S. Pechev, P. Gravereau, journal of Physics and Chemistry of Solids, 74 (2013) 737.

[28] W. Hayes, H. Loudon, Scattering of Light by Crystals, John Wiley and Sons (1978).

[29] K. Jaouadi, N. Zouari, T. Mhiri, A. Daoud, F. Romain, journal of Raman Spectrosc, 38 (2007) 344.

