

# SCIENTIFIC REPORTS



OPEN

## Presence of Li Clusters in Molten LiCl-Li

Augustus Merwin<sup>1</sup>, William C. Phillips<sup>1</sup>, Mark A. Williamson<sup>2</sup>, James L. Willit<sup>2</sup>, Perry N. Motsegood<sup>2</sup> & Dev Chidambaram<sup>1</sup>

Received: 22 October 2015

Accepted: 08 April 2016

Published: 05 May 2016

Molten mixtures of lithium chloride and metallic lithium are of significant interest in various metal oxide reduction processes. These solutions have been reported to exhibit seemingly anomalous physical characteristics that lack a comprehensive explanation. In the current work, the physical chemistry of molten solutions of lithium chloride and metallic lithium, with and without lithium oxide, was investigated using *in situ* Raman spectroscopy. The Raman spectra obtained from these solutions were in agreement with the previously reported spectrum of the lithium cluster,  $\text{Li}_8$ . This observation is indicative of a nanofluid type colloidal suspension of  $\text{Li}_8$  in a molten salt matrix. It is suggested that the formation and suspension of lithium clusters in lithium chloride is the cause of various phenomena exhibited by these solutions that were previously unexplainable.

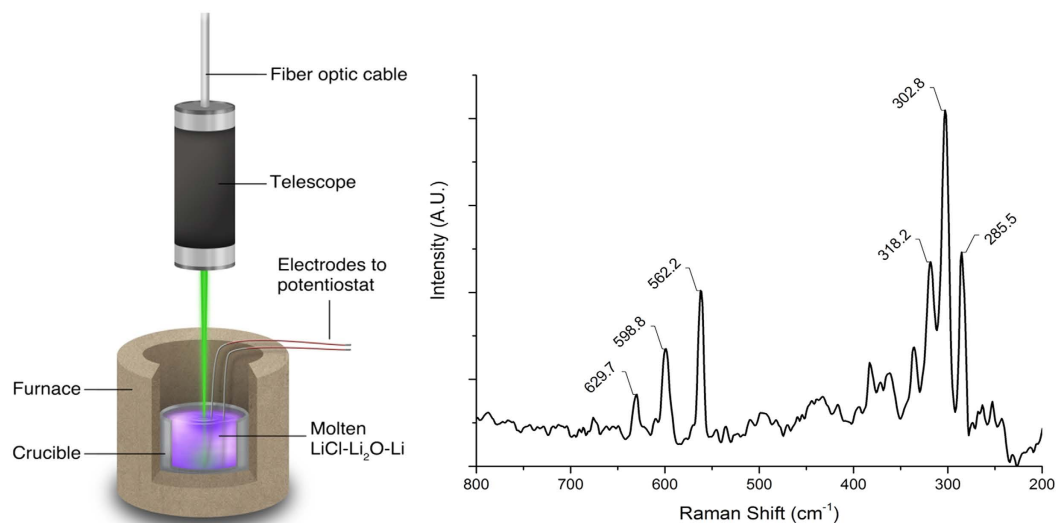
Electrolytic reduction in molten LiCl-Li<sub>2</sub>O electrolytes is commonly used in the conversion of TiO<sub>2</sub><sup>1</sup>, SiO<sub>2</sub><sup>2</sup>, Ta<sub>2</sub>O<sub>5</sub><sup>3</sup>, Nb<sub>2</sub>O<sub>5</sub><sup>4</sup>, and UO<sub>2</sub> (oxide nuclear fuel) to their base metals<sup>5–8</sup>. In several of these processes, notably the reduction of actinide oxides, Li<sup>+</sup> is unavoidably co-reduced with the desired metal oxide, and as a result, elemental Li (Li) is formed and dispersed into the molten salt electrolyte as the process proceeds<sup>9,10</sup>. The dissolution of Li in the electrolyte results in a loss of current efficiency in these processes and therefore is of significant interest. Despite the importance of this dispersion phenomenon, molten solutions of Li and LiCl are not well understood. The interface of Li and molten solutions of LiCl, with and without the presence of either Li<sub>2</sub>O or KCl, has been the subject of extensive research<sup>11–20</sup>. Even so, a knowledge gap still exists in the understanding of the true nature of the molten solutions. Previous research has demonstrated the following phenomena that seemingly are unexplainable:

- Dispersion of Li in LiCl is associated with the formation of a “metal fog” that cannot be explained by physical dissolution<sup>11,12</sup>.
- The reported values of the solubility limit of Li in LiCl measured by different methods of analyses vary significantly and are in disagreement<sup>13–19</sup>.
- The electrical conductivity exhibited by LiCl-Li solutions under metal saturated conditions is unexpectedly low<sup>19</sup>.
- Electrochemical measurements of Li in the presence of LiCl appear as if the thermodynamic activity of Li is significantly lower than unity<sup>13</sup>.
- An intermediate electrochemical potential, between that of  $\text{Li}|\text{Li}^+$  and that of the electrode material, is observed when Li disperses from an electrode<sup>20</sup>.

Previous attempts to explain these properties have led to extensive theoretical research on the existence of “hyperlithiated” compounds, such as Li<sub>2</sub>Cl<sup>20–23</sup>. While mass spectrometry experiments<sup>23,24</sup> have shown the presence of hyperlithiated compounds *in vacuo*, there has been no evidence of their existence in a fused phase. Alternatively, theoretical work has postulated the formation of lithium dimers, Li<sub>2</sub>, in the molten LiCl matrix, as a rationale for explaining the properties of LiCl-Li<sup>25,26</sup>. Recently, suspensions of nanoparticles in other molten salts have been investigated for a wide variety of applications due to their unique physical properties<sup>27</sup>. Similarly, experimental work by Nakajima *et al.* suggested that the dispersion of Li in LiCl is the sum of two separate processes, i.e., physical dissolution and colloidal suspension<sup>15–18</sup>. In these studies, micron-sized particles of metallic Li were observed in quenched LiCl-Li. However, it also was noted that the metallic species would require an emulsifying agent to be suspended in the ionic fluid. While it was proposed that impurities, such as Li<sub>2</sub>O and

<sup>1</sup>Materials Science and Engineering, University of Nevada, Reno 1664 N. Virginia St. Reno, MS0388, NV 89557, USA.

<sup>2</sup>Nuclear Chemical Engineering Department, Nuclear Engineering Division Argonne National Laboratory, Argonne, IL 60439, USA. Correspondence and requests for materials should be addressed to D.C. (email: dcc@unr.edu)



**Figure 1.** (Left) Schematic diagram of the experimental setup used for the measurement of the Raman features of LiCl-Li<sub>2</sub>O with electrochemically generated Li. (Right) Raman spectrum of LiCl-Li<sub>2</sub>O-Li at 923 K obtained after reducing the equivalent of 1 wt% Li from LiCl-3 wt% Li<sub>2</sub>O. The spectrum was recorded using a 10-mW, 532-nm laser focused vertically onto the surface of the molten solution. The spectrum was comprised of three fundamental features at 285.5, 302.8, and 318.2 cm<sup>-1</sup>, with overtones of decreasing intensity at approximately integer multiples of these Raman shifts.

Li<sub>3</sub>N, act as emulsifying agents, it also was noted that the concentration of dispersed Li in LiCl was not highly dependent on the concentration of either Li<sub>2</sub>O or Li<sub>3</sub>N.

Validation of the previously discussed hypotheses is experimentally challenging due to the highly reactive nature of molten solutions that contain metallic Li and LiCl. Furthermore, *ex situ* experimental techniques are not reliable because the phase stability of mixtures of LiCl and Li is temperature dependent<sup>13,14</sup>. In the current research, we used Raman spectroscopy for the *in situ* characterization of molten mixtures of LiCl, Li<sub>2</sub>O, and Li at 923 K in an attempt to understand the nature of these solutions.

## Results

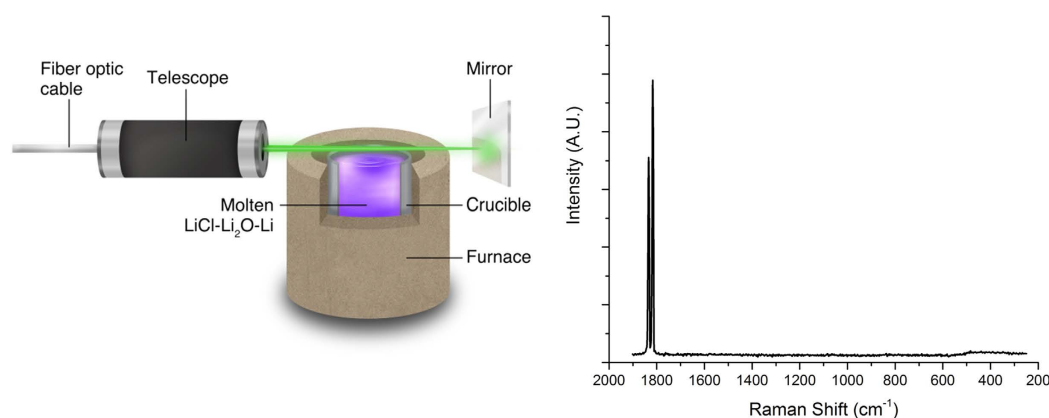
Figure 1 shows the *in situ* Raman spectrum of molten LiCl-Li<sub>2</sub>O-Li (923 K) after electrochemically reducing an equivalent of 1 wt% of the melt to metallic lithium as well as a schematic depiction of the experimental setup. It was observed that the primary features at 285.5, 302.8, and 318.2 cm<sup>-1</sup> exhibited overtones of decreasing intensity with increasing Raman shift. Spectra exhibiting the Raman features in Fig. 1 were observed to deviate minimally over the course of 90 minutes. The spectra showed no dependence on the crucible material (Mo and Ta) in which the melts were contained. Additionally, similar Raman spectra were observed in melts of LiCl-Li without the addition of Li<sub>2</sub>O, and the intensity of the spectral features was observed to have a linear dependence on the power of the excitation laser (Supporting Information Figs SI 1–3).

An additional experiment was conducted to confirm that the spectrum shown in Fig. 1 was characteristic of the molten metal/molten salt phase of LiCl-Li<sub>2</sub>O-Li and not of the vapor. To characterize the vapor phase that existed above the mixture, the excitation laser was maintained parallel to the surface of the melt, approximately 5 mm above the fluid/vapor interface, and it was reflected by a stainless steel mirror. The spectrum recorded in this manner, shown in Fig. 2, exhibited high intensity Na fluorescence lines at 1818.1 and 1835.3 cm<sup>-1</sup> (589 and 589.6 nm)<sup>28</sup>. For the sake of clarity, identical features were omitted from the spectrum in Fig. 1; however, the full spectral range of this spectrum is provided in the Supporting Information (Fig. SI 4).

## Discussions

The Raman spectrum of molten LiCl-Li<sub>2</sub>O-Li at 923 K, shown in Fig. 1, is in agreement with the spectrum of the Li cluster, Li<sub>8</sub>, previously reported by Kornath *et al.*<sup>29</sup>. The minor discrepancies between the spectra found in the literature and this study are attributed to the difference in the temperatures used in the experiments; the spectra given in this study being recorded 908 K higher than those in the published work. Despite the vast difference in the temperatures used in the experiments, the symmetry of the two spectra were nearly identical and they consisted of three primary features centered on the A<sub>1</sub> breathing mode of Li<sub>8</sub> at 302.8 cm<sup>-1</sup> recorded at 923 K in the current study, and reported at 295.3 cm<sup>-1</sup> when observed at 15 K. Additionally, the overtones in the spectrum obtained in this study were in agreement with the spectra obtained from Li<sub>8</sub> by Kornath *et al.*<sup>29</sup>.

Spectroscopy of the vapor phase that exists above the LiCl-Li melt was conducted to confirm that the reported spectrum was a characteristic of the fused phase. The spectrum in Fig. 2, obtained with the laser passing horizontally above the surface of the melt, shows only the fluorescence lines from Na and none of the Raman features that have been suggested to be characteristic of Li<sub>8</sub>. LiCl has a lower free energy of formation than NaCl at 923 K, and,



**Figure 2.** (Left) Schematic depiction of the experimental setup used to characterize the vapor phase that existed above the molten LiCl-Li. The excitation laser was maintained horizontally 5 mm above the surface of the LiCl-Li melt and reflected by a stainless steel mirror. (Right) Recorded spectrum of the vapor existing above the surface of LiCl-Li melt maintained at 923 K. The intense features spanning 1800 to 1900  $\text{cm}^{-1}$  are characteristic of the fluorescence of Na from NaCl, which is found as a contaminant in LiCl.

as a result, metallic Li will displace Na from NaCl (found as contaminant in LiCl) in the melt to form LiCl and  $\text{Na}^{30}$ . Furthermore, Na has a vapor pressure of 7.43 kPa at 923 K and is expected to vaporize in significant quantities<sup>31</sup>. Therefore, as a result of the reaction between Li and impurity NaCl, the vapor existing above the LiCl-Li melt consists primarily of Na. Observation of just the Na fluorescence signal in Fig. 2, without the Raman features of interest, demonstrated that the vapor phase does not contain detectable quantities of Li clusters, and that the Raman spectrum shown in Fig. 1 is characteristic of molten LiCl-Li.

Interestingly, the observed Raman spectrum of LiCl-Li does not exhibit the predicted vibrational modes for  $\text{Li}_2\text{Cl}^{20,21}$ . However, it was noted that the original work by Hébant *et al.* hypothesized the existence of  $\text{Li}_2\text{Cl}$  precisely at the interface between liquid Li and LiCl, while the spectrum reported in this study were specifically recorded on the surface of the bulk fluid. Additionally, the Raman mode of the  $\text{Li}_2$  dimer at  $349 \text{ cm}^{-1}$  was not observed in the molten LiCl-Li<sup>32,33</sup>.  $\text{Li}_2$  has been detected under a variety of conditions via Raman spectroscopy, and, previously, it was theorized to be the probable form of Li complexes in molten LiCl-Li<sup>25,26</sup>. Importantly, however, the dimer molecule possesses  $D_{\infty h}$  symmetry<sup>34,35</sup>, while the  $\text{Li}_8$  cluster possesses a complex hypertetrahedral  $T_d$  geometry<sup>29,36–38</sup>. The axially-symmetric nature of  $\text{Li}_2$  would prevent the dimer from exhibiting a dipole moment and therefore would be immiscible with the pure electrolyte. Alternatively, the asymmetric electronic structure of  $\text{Li}_8$  may enable the suspension of these clusters in an ionic fluid. Colloids of lithium clusters are known to form in solid LiF and  $\text{Li}_2\text{O}$  crystals when subjected to sufficient irradiation<sup>39–41</sup>. Furthermore,  $\text{Li}_8$  clusters have been observed to be stable in LiF at temperatures up to 1143 K<sup>42</sup>. These reports demonstrated that Li clusters are stable in ionically-bound systems at temperatures exceeding those of the current study.

The variance in reported values of the solubility limit of Li in molten LiCl is suspected to derive from the colloidal suspension of Li clusters in molten LiCl-Li. The solubility limit of Li in LiCl detected by thermal analysis has been reported to be  $0.5 \pm 0.2 \text{ mol}\%$  at 913 K<sup>14</sup>, while electrochemical analysis quantified the limit to be 1.8 mol% at 923 K<sup>13,19</sup>, and chemical analysis of LiCl-Li quenched at 923 K has been reported to contain greater than 3 mol% Li<sup>15–18</sup>. Should Li clusters exist in molten solutions of LiCl-Li, a well-defined solubility limit may not exist due to the dispersion mechanism of colloidal suspension in addition to physical dissolution. In this case, the quantity of Li that may be suspended or dispersed under a given set of conditions would be highly dependent upon experimental factors such as thermally induced mixing of the melt or mechanical agitation.

The conglomeration of Li atoms in the form of clusters may explain why the  $\text{F}^-$  center model notably overestimates the electrical conductivity of the LiCl-Li system while accurately predicting such properties for alternative solutions of alkali metal-alkali halide salts<sup>19,43,44</sup>. The  $\text{F}^-$  center model used to describe electron mobility in metal-salt solutions operates on the assumption that each metal atom acts as an electron donor to the electronic structure of the molten system<sup>45–48</sup>. Under this assumption, the melt should exhibit a rapid increase in electrical conductivity with the inclusion of a small concentration of electron donor atoms. Alternatively, if Li atoms suspended in molten LiCl-Li form clusters their valance electrons would be confined to the clusters instead of extending into shared electron states of the melt as a whole. This effect can therefore account for the consumption of what would be “free” electrons under metal saturated conditions, resulting in a suppression of the electrical conductivity of the melt<sup>43,45–49</sup>. Similarly, the  $\text{F}^-$  center model does not apply to polyvalent metal-salt solutions such as Bi-BiI<sub>3</sub><sup>50,51</sup>. In a manner analogous to the proposed formation of Li clusters, metal salt solutions containing transition metals form abnormally reduced complexes referred to as subhalides. The formation of these subhalides localizes what would otherwise be delocalized electrons causing the electron mobility in the melt to be lower than that predicted by the  $\text{F}^-$  center model.

The presence of Li clusters in molten LiCl-Li may additionally elucidate the unattributed electrochemical phenomena exhibited by these solutions. As mentioned previously, electrochemical measurements of Li in contact

with molten LiCl exhibits two distinct electrochemical potentials<sup>20,21</sup>. Furthermore, the alteration to the Li|Li<sup>+</sup> open circuit potential that occurs with varying concentrations of Li in the melt is not Nernstian; it behaves as if the activity of the reduced form, Li is not unity and changes based on concentration<sup>13</sup>. These facts suggest that molten solutions of LiCl and Li contain additional Li complexes other than LiCl and Li; a hypothesis that has been debated without confirmation for decades<sup>15,20,21,46,48,50–52</sup>. It was noted that Li clusters possess significantly different ionization potentials than metallic Li<sup>36,37</sup>; therefore they are expected to exhibit thermodynamic activity that is different from the metallic phase. As a result, the anomalous electrochemical properties of LiCl-Li can be attributed to the simultaneous existence of multiple, variable-activity Li phases. The multiple electrochemical potentials exhibited by Li, as well as the appearance of Li not maintaining unit activity are hypothesized to be due to the presence Li clusters and metallically bonded Li in physical contact with the molten solution concurrently.

## Methods

Anhydrous 99.999 wt% purity LiCl, 99.9 wt% purity Li<sub>2</sub>O, and 99.9 wt% purity Li were procured from VWR Scientific. Melts were contained in Mo or Ta crucibles, and were maintained at 923 ± 10 K throughout all of the experiments. The experiments were conducted in an Ar atmosphere glove box containing less than 5 ppm O<sub>2</sub> and 2 ppm H<sub>2</sub>O.

Molten solutions of LiCl-Li<sub>2</sub>O-Li were generated electrochemically via electrolysis of Li<sub>2</sub>O from molten LiCl-3 wt%-Li<sub>2</sub>O. Electrolysis was conducted using a coil of Pt as the working anode and a coil of stainless steel alloy 316L as the cathode at a cell voltage of 3.2 V, analogous to the cell potential utilized in the electrolytic reduction of UO<sub>2</sub><sup>9,52</sup>. Polarization was conducted in this manner until sufficient charge was passed through the cell to reduce an equivalent of 1 wt% of the melt to metallic Li. The electrodes were maintained in the melt for one hour following electrolysis prior to removal for characterization using Raman spectroscopy. The LiCl-Li melts were prepared in subsequent experiments by directly adding metallic Li to molten LiCl. In all cases, Li<sub>2</sub>O and/or Li were added after drying the LiCl at 823 K to remove residual H<sub>2</sub>O and suppress the formation of LiOH.

Raman spectroscopic measurements were conducted *in situ* using a Thermo-Scientific DXR spectrometer with a custom fiber optic probe procured from InPhotonics. A 10 mW, 532 nm laser beam was passed through the fiber optic cable and telescope before being focused on the surface of the melt. The telescope functioned as the incident and receiving optic for the laser light incident on and reflecting off of the molten solution. In alternative experiments, the laser beam was propagated horizontally, approximately 5 mm above the LiCl-Li mixture, and it was reflected by a metallic surface to characterize the vapor phase that existed on top of the melt. The reported spectra were an average of eight consecutively recorded spectra, each of which was recorded for eight seconds<sup>53</sup>.

## Conclusions

The Raman spectra of molten LiCl-Li<sub>2</sub>O-Li and LiCl-Li were recorded *in situ* at 923 K. A Raman active phase forms in the fused salt when Li is electrochemically reduced from Li<sub>2</sub>O in LiCl, and the recorded spectrum is consistent with previously-reported characteristic spectrum of the lithium cluster Li<sub>8</sub>. These spectra were seen to be characteristic of the bulk fluid rather than the vapor phase that existed above the melt, and it was observed to be stable over a 90-minute period. The presence of a colloidal suspension of lithium clusters in the molten salt may explain the anomalous physical behavior of LiCl-Li solutions.

## References

- Hur, J.-M., Lee, S.-C., Jeong, S.-M. & Seo, C.-S. Electrochemical Red uction of TiO<sub>2</sub> in Molten LiCl-Li<sub>2</sub>O. *Chem. Lett.* **36**, 1028–1029 (2007).
- Lee, S.-C., Hur, J.-M. & Seo, C.-S. Silicon powder production by electrochemical reduction of SiO<sub>2</sub> in molten LiCl-Li<sub>2</sub>O. *J Ind. Eng. Chem.* **14**, 651–654 (2008).
- Jeong, S. M., Jung, J.-Y., Seo, C.-S. & Park, S.-W. Characteristics of an electrochemical reduction of Ta<sub>2</sub>O<sub>5</sub> for the preparation of metallic tantalum in a LiCl-Li<sub>2</sub>O molten salt. *J. Alloys Compd.* **440**, 210–215 (2007).
- Jeong, S. M., Yoo, H. Y., Hur, J.-M. & Seo, C.-S. Preparation of metallic niobium from niobium pentoxide by an indirect electrochemical reduction in a LiCl-Li<sub>2</sub>O molten salt. *J. Alloys Compd.* **452**, 27–31 (2008).
- Gourishankar, K., Reddy, L. & Williamson, M. In *Light Metals 2002* (ed. The Minerals, Metals & Materials Society), 1075–1082 (Warrendale, 2002).
- Sakamura, Y. & Omori, T. Electrolytic Reduction and Electrorefining of Uranium to Develop Pyroprocessing of Oxide Fuels. *Nucl. Technol.* **171**, 266–275 (2010).
- Choi, E.-Y. *et al.* Electrochemical reduction of porous 17 kg uranium oxide pellets by selection of an optimal cathode/anode surface area ratio. *J. Nucl. Mater.* **418**, 87–92 (2011).
- Herrmann, S. D., Li, S. X. & Westphal, B. R. Separation and Recovery of Uranium and Group Actinide Products From Irradiated Fast Reactor MOX Fuel via Electrolytic Reduction and Electrorefining. *Sep. Sci. Technol.* **47**, 2044–2059 (2012).
- Park, W. *et al.* An experimental study for Li recycling in an electrolytic reduction process for UO<sub>2</sub> with a Li<sub>2</sub>O-LiCl molten salt. *J. Nucl. Mater.* **441**, 232–239 (2013).
- Choi, E.-Y. *et al.* Electrochemical reduction of UO<sub>2</sub> in LiCl-Li<sub>2</sub>O molten salt using porous and nonporous anode shrouds. *J. Nucl. Mater.* **444**, 261–269 (2014).
- Takenaka, T., Shigeta, K., Masuhama, H. & Kubota, K. Influence of Some Factors upon Electrodeposition of Liquid Li and Mg. *ECS Trans.* **49**, 441–448 (2009).
- Takenaka, T., Morishige, T. & Umehara, M. In *Molten Salts Chemistry and Technology* (eds Gaune-Escard, M. *et al.*) Ch. 2, 143–148 (John Wiley & Sons, Ltd, 2014).
- Liu, J. & Poignet, J. C. Measurement of the activity of lithium in dilute solutions in molten lithium chloride between 650 °C and 800 °C. *J. Appl. Electrochem.* **20**, 864–867 (1990).
- Dworkin, A. S., Bronstein, H. R. & Bredig, M. A. Miscibility of Metals with Salts. VI. Lithium-Lithium Halide Systems. *J. Phys. Chem.-US* **66**, 572–573 (1962).
- Nakajima, T., Nakanishi, K. & Watanabe, N. Study of Emulsions in Molten Salts III. The concentration stability and particle-size distribution of dispersed lithium in molten lithium chloride. *Bull. Chem. Soc. Jpn.* **49**, 994–997 (1975).
- Nakajima, T., Nakanishi, K. & Watanabe, N. The Dispersion of Metallic Lithium in Various Molten Salts. *Nippon Kagaku Kaishi* **1975**, 617–621 (1975).

17. Watanabe, N., Nakanishi, K. & Nakajima, T. The Dissolution of Lithium in Molten Lithium Chloride. *Nippon Kagaku Kaishi* **1974**, 401–404 (1974).
18. Nakajima, T., Minami, R., Nakanishi, K. & Watanabe, N. Miscibility of Lithium with Lithium Chloride and Lithium Chloride - Potassium Chloride Eutectic Mixture. *Bull. Chem. Soc. Jpn.* **47**, 2071–2072 (1974).
19. Liu, J. & Poignet, J. C. Electronic conductivity of salt-rich Li–LiCl melts. *J. Appl. Electrochem.* **22**, 1110–1112 (1992).
20. Héban, P. & Picard, G. S. Electrochemical investigations of the liquid lithium/(LiCl–KCl eutectic melt) interface. Chronopotentiometric and electrochemical impedance spectroscopy measurements. *Electrochim. Acta* **43**, 2071–2081 (1998).
21. Héban, P. & Picard, G. S. Computational investigations of the liquid lithium/(LiCl–KCl eutectic melt) interface. *J. Mol. Struct. - THEOCHEM* **426**, 225–232 (1998).
22. Veličković, S. R., Djustebek, J. B., Veljković, F. M., Radak, B. B. & Veljković, M. V. Formation and ionization energies of small chlorine-doped lithium clusters by thermal ionization mass spectrometry. *Rapid Commun. Mass Spectrom.* **26**, 443–448 (2012).
23. Veličković, S. *et al.* Ionization energies of  $\text{Li}_n\text{X}$  ( $n = 2, 3$ ;  $\text{X} = \text{Cl, Br, I}$ ) molecules. *Rapid Commun. Mass Spectrom.* **20**, 3151–3153 (2006).
24. Nešković, O. M., Veljković, M. V., Veličković, S. R., Petkovska, L. T. & Perić-Grujić, A. A. Ionization energies of hypervalent  $\text{Li}_2\text{F}$ ,  $\text{Li}_2\text{Cl}$  and  $\text{Na}_2\text{Cl}$  molecules obtained by surface ionization electron impact neutralization mass spectrometry. *Rapid Commun. Mass Spectrom.* **17**, 212–214 (2003).
25. Bengtsson, L., Holmberg, B. & Ulvenlund, S. Fluorodilithium(1+) and hydroxodilithium(1+) in molten alkali-metal nitrate. *Inorg. Chem.* **29**, 3615–3618 (1990).
26. Corbett, J. D. In *Fused Salts* (ed. Sundheim, B.) Ch. 6, 341–407 (McGraw Hill, 1964).
27. Xu, J., Chen, L., Choi, H., Konish, H. & Li, X. Assembly of metals and nanoparticles into novel nanocomposite superstructures. *Sci. Rep.* **3**, doi: 10.1038 (2013).
28. Juncar, P., Pinard, J., Hamon, J. & Chartier, A. Absolute Determination of the Wavelengths of the Sodium D 1 and D 2 Lines by Using a CW Tunable Dye Laser Stabilized on Iodine. *Metrologia* **17**, 77 (1981).
29. Kornath, A., Kaufmann, A., Zoermer, A. & Ludwig, R. Raman spectroscopic investigation of small matrix-isolated lithium clusters. *J. Chem. Phys.* **118**, 6957–6963 (2003).
30. Pankratz, L. Thermodynamic Properties of Halides. *Bulletin 674 Published by United States Department of the Interior, Bureau of Mines*, (1964).
31. Makansi, M. M., Muendel, C. H. & Selke, W. A. Determination of the Vapor Pressure of Sodium. *J. Chem. Phys.* **59**, 40–42 (1955).
32. Froben, F. W. & Schulze, W. Raman Measurements of Matrix-Isolated Small Metal Clusters. *Berich Bunsen Gesell.* **88**, 312–314 (1984).
33. Briley, A., Pederson, M. R., Jackson, K. A., Patton, D. C. & Porezag, D. V. Vibrational frequencies and intensities of small molecules: All-electron, pseudopotential, and mixed-potential methodologies. *Phys. Rev. B* **58**, 1786–1793 (1998).
34. Boustani, I., Pestworf, W., Fantucci, P., Bonačić-Koutecký, V. & Koutecký, J. Systematic *ab-initio* configuration-interaction study of alkali-metal clusters: Relation between electronic structure and geometry of small Li clusters. *Phys. Rev. B* **35**, 9437–9450 (1987).
35. Ishii, S., Ohno, K., Kawazoe, Y. & Louie, S. G. *Ab-initio* GW quasiparticle calculation of small alkali-metal clusters. *Phys. Rev. B* **65**, 245109 (2002).
36. Dugourd, P. *et al.* Measurements of lithium cluster ionization potentials. *Chem. Phys. Lett.* **197**, 433–437 (1992).
37. de Heer, W. A. The physics of simple metal clusters: experimental aspects and simple models. *Rev. Mod. Phys.* **65**, 611–676 (1993).
38. Grassi, A., Lombardo, G. M., Angilella, G. G. N., March, N. H. & Pucci, R. Equilibrium geometries of low-lying isomers of some Li clusters, within Hartree–Fock theory plus bond order or MP2 correlation corrections. *J. Chem. Phys.* **120**, 11615–11620 (2004).
39. Davidson, A. T. *et al.* Vacuum ultraviolet absorption and ion track effects in LiF crystals irradiated with swift ions. *Phys. Rev. B* **66**, 214102 (2002).
40. Krexner, G., Prem, M., Beuneu, F. & Vajda, P. Nanocluster Formation in Electron-Irradiated  $\text{Li}_2\text{O}$  Crystals Observed by Elastic Diffuse Neutron Scattering. *Phys. Rev. Lett.* **91**, 135502 (2003).
41. Ibragimova, E. M., Mussaeva, M. A. & Buzrikov, S. N. Recombination gamma-luminescence at the nanometal Li – dielectric LiF interfaces. *Radiat. Phys. Chem.* **111**, 40–45 (2015).
42. Beuneu, F., Vajda, P. & Zogal, O. J. Li colloids created by electron-irradiation of LiF: A great wealth of properties. *Nucl. Instrum. Meth. B* **191**, 149–153 (2002).
43. Bronstein, H. R. & Bredig, M. A. The Electrical Conductivity of Solutions of Alkali Metals in their Molten Halides. *J. Am. Chem. Soc.* **80**, 2077–2081, (1958).
44. Freyland, W. Bulk and Surface Characteristics of Metal-Molten Salt Solutions. *J. Non-Cryst. Solids* **117**, 613–622 (1990).
45. W.W. Warren, J. In *Molten Salts: From Fundamentals to Applications* (ed. Gaune-Escard, M.) 23–46 (Kluwer Academic Publishers, 2001).
46. Nattland, D., Blanckenhagen, B. v., Juchem, R., Schellkes, E. & Freyland, W. Localized and mobile electrons in metal - molten-salt solutions. *J. Phys.: Condens. Matter* **8**, 9309 (1996).
47. Beck, U., Koslowski, T. & Freyland, W. Electronic structure of metal-molten salt solutions: Electron localization and the metal-non-metal transition. *J. Non-Cryst. Solids* **205–207**, 52–56 (1996).
48. Tosi, M. P. Nonmetal-Metal Transition in Solutions of Metals in Molten Salts. *J. Mater. Sci. Technol.* **14**, 1–8 (1998).
49. Masset, P. & Guidotti, R. A. Thermal activated (thermal) battery technology: Part II. Molten salt electrolytes. *J. Power Sources* **164**, 397–414 (2007).
50. Bredig, M. A. In *Molten Salt Chemistry* (ed. Blander, M.) 367–425 (John Wiley and Sons, 1964).
51. Grantham, L. F. & Yosim, S. J. Electrical Conductivities of Molten Bi–Bi<sub>3</sub> Solutions. *J. Chem. Phys.* **38**, 1671–1676 (1963).
52. Choi, E.-Y. *et al.* Electrochemical reduction behavior of a highly porous SIMFUEL particle in a LiCl molten salt. *Chem. Eng. J.* **207–208**, 514–520 (2012).
53. Gese, N., Pesic, B. In *2013 TMS Annual Meeting & Exhibition* (ed. TSM), 2013).

## Acknowledgements

This work was performed under the auspices of the Department of Energy (DOE) through grant DE-NE0008262, and the US Nuclear Regulatory Commission (NRC) under contracts NRCHQ-11-G-38-0039, NRC-38-10-949 and NRC-HQ-13-G-38-0027. Mr. Kenny Osborne serves as the program manager for the DOE awards and Ms. Nancy Hebron-Isreal serves as the grants program officer for the NRC awards.

## Author Contributions

D.C. conceived the work. A.M., W.C.P. and D.C. designed the experiments. A.M. and W.C.P. constructed the experimental setup. A.M. collected the data. A.M. and D.C. performed the analysis. M.A.W., J.L.W. and P.N.W. participated in discussions with A.M., D.C., A.M. and D.C. wrote the manuscript. W.C.P., M.A.W., J.L.W. and P.N.W. edited the manuscript. All authors reviewed the manuscript.



### Additional Information

**Competing financial interests:** The authors declare no competing financial interests.

**How to cite this article:** Merwin, A. *et al.* Presence of Li Clusters in Molten LiCl-Li. *Sci. Rep.* **6**, 25435; doi: 10.1038/srep25435 (2016).



This work is licensed under a Creative Commons Attribution 4.0 International License. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in the credit line; if the material is not included under the Creative Commons license, users will need to obtain permission from the license holder to reproduce the material. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>