



## RESEARCH ARTICLE

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# Hardness and Yield Strength of CO<sub>2</sub> Ice Under Martian Temperature Conditions

**Key Points:**

- We have measured the hardness of CO<sub>2</sub> ice in Martian temperatures
- We derived the corresponding yield strength values
- Under Martian conditions CO<sub>2</sub> ice is slightly weaker than H<sub>2</sub>O ice

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**Abstract** Although ice fracturing and deformation is key to understanding some of the landforms encountered in the high-latitude regions on Mars and on other icy bodies in the solar system, little is known about the mechanical characteristics of CO<sub>2</sub> ice. We have measured the hardness of solid CO<sub>2</sub> ice directly in the laboratory with a Leeb hardness tester and calculated the corresponding yield strength. We have also measured the hardness of water ice by the same method, confirming previous work. Our results indicate that CO<sub>2</sub> ice is slightly weaker, ranging between Leeb ~200 and 400 (~10 and 30 MPa yield strength, assuming only plastic deformation and no strain hardening during the experiment), for typical Martian temperatures. Our results can be used for models of CO<sub>2</sub> ice rupture (depending on the deformation timescales) explaining surface processes on Mars and solar system icy bodies.

**Plain Language Summary** We have measured the hardness and calculated the strength of dry (carbon dioxide) ice. It is slightly weaker than water ice in the temperature range that can be found on Mars. Our results can be used in models of how surface features on Mars and on other icy surfaces in the solar system are formed.

## 1. Introduction

The surfaces of numerous solar system bodies are wholly or partly covered with water and/or carbon dioxide ice whose dynamics play a key role in the shaping of planetary landforms. Responding to gravitational, tidal, and other forces, ice may flow, or fracture, affecting the morphology of surfaces composed of both volatile and nonvolatile material. While the mechanical properties of water ice, whose dynamical behavior affects large portions of the Earth, are reasonably well understood, very little attention has been paid to the mechanical properties of CO<sub>2</sub> ice, which plays an important role on bodies throughout the solar system. On Mars, CO<sub>2</sub> covers the southern pole and CO<sub>2</sub> ice seems to be responsible for numerous enigmatic landforms that can be found in regions such as the so-called cryptic terrain (Kieffer et al., 2000) or the Swiss-cheese terrain (e.g., Malin et al., 2001). Specifically, quantifying the dynamic behavior of CO<sub>2</sub> ice is important to address questions relating to the processes resulting in araneiforms. These are channel-like features scoured in the polar regolith by hypothesized gas flow under a layer of CO<sub>2</sub> ice (Kieffer, 2007). When this gas pressure exceeds the—unknown—rupture strength of the overlying ice, gas and regolith can be ejected onto the surface, creating plumes and playing an important role in dust transportation on Mars. CO<sub>2</sub> ice strength is therefore a key parameter in understanding these processes, but, to date, whenever hardness/strength data were needed to explain CO<sub>2</sub> ice-related phenomena on Mars, authors have had to find work-arounds. Modeling gas activity of CO<sub>2</sub>-covered surfaces on Mars, Portyankina et al. (2010) pointed out that considerable uncertainties are related to the poorly constrained mechanical characteristics of CO<sub>2</sub> ice and so they base their model on water ice parameters. Portyankina et al. (2012) took the same approach and point out that the fracturing part of their model “is highly uncertain mostly because of the lack of measurements of the mechanical properties of CO<sub>2</sub> ice.” Some experimental evidence does exist: Nye et al. (2000) were able to conclude, based on Durham et al’s (1999) deformation experiments of CO<sub>2</sub> ice, that Mars’s south polar cap is unlikely to consist of pure CO<sub>2</sub> ice because it would not be mechanically stable. Clark and Mullin (1976) measured the flow law for solid CO<sub>2</sub> ice and found it to be one tenth to one third as strong as water ice under equivalent Martian conditions. However, similar to Durham et al’s (1999) experiments, this represents the material behavior under low strain rate viscous flow, rather than the tensile, yield, or compressive strength values more relevant for sudden events such as fracturing. Beyond Mars, understanding the strength of CO<sub>2</sub> ice is relevant to the mechanical properties of icy regoliths on the surfaces and interiors of outer solar system objects such as comets and Kuiper belt objects.

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We have measured the hardness of compact CO<sub>2</sub> ice as a function of temperature, using a commercial Leeb hardness tester. We discuss the results of this test (a local impact hardness in the high strain rate regime) in the context of material deformation models at different scales and Martian temperature conditions and the similar existing measurements made for water ice by Henderson et al. (2019). Our results can be used to inform models of CO<sub>2</sub> ice-related landscape evolution on Mars and provide important parameters for understanding the mechanical behavior of CO<sub>2</sub>-covered surfaces elsewhere in the solar system.

## 2. Method

The Leeb rebound hardness test method is a dynamic test procedure that was developed as a flexible alternative to traditional hardness testing methods such as the Vickers method (Smith & Sandland, 1925). These methods usually require fixed workstations and can cause damage to the test object. The use of a portable Leeb hardness tester usually does not result in significant damage to the object of interest. Originally designed for metal quality control testing, Leeb testers are now also frequently used in geomorphological research (see, e.g., Alberti et al., 2013; Aoki & Matsukura, 2008) and, due to their easy operation in the field and the very short time of contact between the tester and the sample, qualify for hardness measurements on icy surfaces. The Leeb hardness test uses the rebound method. This method relies on deriving sample hardness from the loss of energy of a small impactor. A 3–5 mm diameter impactor with defined weight impinges on the surface of the test object at a defined speed of a few meters per second. The impact creates a plastic deformation of the surface, that is, an indentation, causing the impactor to lose part of its original kinetic energy. The velocities before and after the impact are measured using a contactless method. A spring-driven piston plunges toward the surface of the sample. The piston moves through a coil, causing an electric current with a voltage proportional to the velocity of the piston. The Leeb hardness (or Leeb number) HL is defined as

$$HL = 1,000 \frac{|v_i|}{|v_r|} = 1,000C_r \quad (1)$$

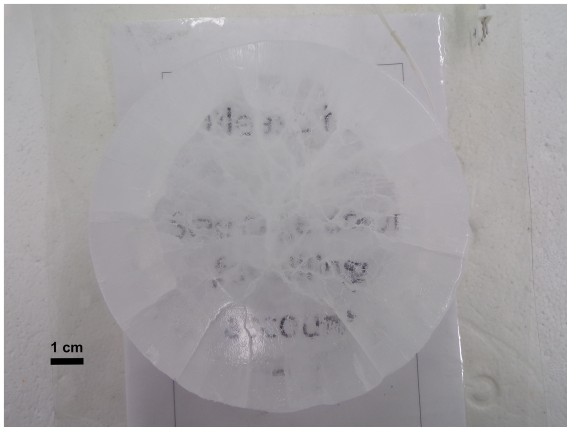
(Leeb, 1979), where  $v_i$  and  $v_r$  are the impact and rebound velocities, respectively.  $C_r$  is the coefficient of restitution. The impact and rebound velocities are measured automatically.

We measured CO<sub>2</sub> ice hardness using a mobile Leeb hardness tester (Sauter GmbH, model Sauter HK-D). The main unit can be operated using seven different impact devices with various impactor sizes and energies, depending on the expected mechanical properties of the sample material. The so-called D-type sensor with an impact energy of 11 mJ is designed for general testing conditions. Henderson et al. (2019) chose this sensor for their hardness measurements of water ice. However, we found that the C-type sensor is a better choice for CO<sub>2</sub> ice measurements. Slabs of CO<sub>2</sub> ice usually have more cracks than H<sub>2</sub>O ice samples and therefore break more easily. Because of these cracks, the sample areas suitable for hardness measurements are restricted. The C-type sensor has a lower impact energy of 2.7 mJ and a lighter tungsten carbide impactor (3 g vs. 5 g). Thus, significant damage to or breaking of the CO<sub>2</sub> ice sample becomes less likely. Furthermore, the lower impact energy of the C-type sensor allows shorter distances between two indentation centers and between an indentation center and the edge of tested sample compared to the D-type sensor, making it more suitable for smaller surface areas. In general, the C-type sensor is designed for small, light, and thin samples.

In our experiments we focus on the properties of CO<sub>2</sub> ice and the differences between CO<sub>2</sub> and H<sub>2</sub>O ice. For comparison of our own data with Henderson et al. (2019) values for water ice, water ice hardness was measured with a Leeb tester using both the C-type and the D-type sensors.

### 2.1. Sample Preparation

We condensed CO<sub>2</sub> ice from the gas phase into a cylindrical tank whose base was cooled by liquid nitrogen and whose top was heated to a temperature above 193 K to prevent clogging of the gas inlet, as described by Kaufmann and Hagermann (2017). As a consequence of the directional ice deposition caused by the thermal gradient inside the tank, we obtained clear slab ice showing no granularity (Figure 1). In some instances, patterns of cracks near the outer surfaces of the samples (which might be due to thermal expansion) could be interpreted as the delineations of individual crystals. These features were usually larger than 2 mm, a size similar to or greater than our impactor diameter. Six CO<sub>2</sub> ice slabs with a density of  $1,513 \pm 75 \text{ kg/m}^3$ , diameter of  $12 \pm 0.8 \text{ cm}$ , and an average height of  $4.36 \pm 1 \text{ cm}$  were used as samples. Portyankina et al. (2019) found that the clear slab ice as used in our measurements is the most likely type



**Figure 1.** Example of a CO<sub>2</sub> ice block used as sample for the hardness measurements.

crack-free ice slabs. Compact, transparent areas suitable for Leeb hardness measurements are usually smaller and rarer than in H<sub>2</sub>O ice samples. As a result, the C-type sensor is more suitable for hardness measurements in CO<sub>2</sub>; due to its lower impact energy, further cracking is avoided and the minimum surface areas required for accurate measurements can be smaller. We carried out hardness measurements on top of a crack in a sample in order to ascertain how cracks affect the hardness values measured. We found the values to be much lower than in the compact, clear regions of the sample. Sometimes, the effect would be such that hardness values could not be obtained at all because the instrument returned a measurement error.

Because our aim was to obtain hardness values for CO<sub>2</sub> as a function of temperature, we measured under the controlled conditions described below. However, as accurate temperature control is very resource intensive, we also made use of temperatures occurring naturally during the CO<sub>2</sub> slab production process or temperature regimes readily available in a freezer (which explains why our temperature error bars vary). Slab temperatures were measured using an RTD sensor placed approximately in the center of the sample. We were able to obtain hardness values over a temperature range from ~100 to 190 K.

Temperature control was achieved by placing the sample slab on an aluminum cold plate cooled to 186 K using LN<sub>2</sub> before the sample was placed on top. The sample was placed inside a Perspex tube, and the gap between sample and inner wall of the tube was filled with precooled glass beads, before the cold plate (and with it the sample) was slowly cooled further to 92.8 K (see Figure 2). A Styrofoam box covering the perspex tube was used to reduce frost formation by minimizing the flow of warm air onto the sample. Nevertheless, a

thin layer of frost building up during the cooling process could not be avoided. The frost was wiped off before each measurement, leaving a minimal amount of frost on the surface. Since the diameter of the impactor and the depth of the crater obtained are much larger than the frost grain size and the height of the layer, the influence of frost on the hardness values is negligible. During cooling and subsequent measurements, the samples usually lost ~3% in volume and became less translucent, although samples tended to remain intact and could easily be removed and handled without breaking.

For hardness measurements of H<sub>2</sub>O ice at temperatures of about 80 K a more simple, less time consuming approach for cooling the sample was chosen. For these measurements the sample was placed in a metal container with the gap between the sample and the inner wall of the container filled with glass beads. The container was then immersed directly into a bath of LN<sub>2</sub>. Hardness measurements were carried out as soon as the sample temperature had stabilized. Since hardness values obtained using a Leeb hardness tester for H<sub>2</sub>O ice have already been published (Henderson et al., 2019), measurements were only made in three temperature ranges: 243–247, 192–207, and 77–82 K, with the same causes for



**Figure 2.** Setup for our CO<sub>2</sub> ice measurements. The cooling plate is connected via the pipes shown in the right lower corner to a LN<sub>2</sub> Dewar and a digital temperature control unit to stabilize the temperature. The image was taken after the measurements were done.

**Table 1**  
*Hardness Values for CO<sub>2</sub> Ice at Different Temperatures*

Temperature (K)	Leeb hardness (HL) mean $\pm$ st. dev.	No. of measurements
92.8 $\pm$ 1.8	359.1 $\pm$ 24.4	12
105.5 $\pm$ 0.3	384.6 $\pm$ 25.8	16
131.3 $\pm$ 2.9	315.4 $\pm$ 44.3	10
137.0 $\pm$ 1.2	288.3 $\pm$ 33.4	12
143.9 $\pm$ 3.1	304.5 $\pm$ 30.6	18
149.0 $\pm$ 2.1	277.1 $\pm$ 31.2	9
165.0 $\pm$ 1.5	220.2 $\pm$ 26.0	11
179.0 $\pm$ 0.8	200.1 $\pm$ 16.4	10
182.3 $\pm$ 2.2	197.2 $\pm$ 13.0	38
186.9 $\pm$ 4.0	196.2 $\pm$ 18.4	100

the variation in temperature uncertainties as described above. For comparison with Henderson et al., 2019's results, a few H<sub>2</sub>O ice values were measured using both the C-type and D-type sensors.

The reason for the difference in experimental procedure was that the H<sub>2</sub>O approach was not suitable for CO<sub>2</sub> ice because we found that nitrogen steam affected all samples considerably when it penetrated cracks in the ice slabs, resulting in softening or disintegration of the slabs. CO<sub>2</sub> slabs tended to have more cracks than H<sub>2</sub>O slabs, although softening and disintegration could also be observed in H<sub>2</sub>O samples with a high number of cracks.

For H<sub>2</sub>O samples the hardness sensor was precooled when making measurement in the higher temperature ranges. In case of CO<sub>2</sub> ice samples, hardness values obtained with precooled sensor did not noticeably differ from measurements using a “warm” sensor; all values obtained were within the same error range. In order to ascertain that precooling the sensor does not affect the performance of the Leeb hardness tester, a simple series of experiments was carried out using a copper plate (20.5  $\times$  20.5  $\times$  1.5 cm) as a sample. The plate was kept at a constant temperature of 295 K. The sensor was cooled in three different ways: (i) the tip of the sensor was immersed in LN<sub>2</sub>, (ii) only the sensor's impact body was cooled with LN<sub>2</sub>, and (iii) the whole tester was stored in a freezer at 253 K for 35 min. The hardness values measured show no noticeable changes in comparison to the values measured with a sensor kept at room temperature. Average values are almost identical, with a standard deviation of 2.37 for the C-type sensor and 0.66 for the D-type sensor. As expected, the values obtained with the C-type sensor tend to be higher, in this case  $\sim$ 21%.

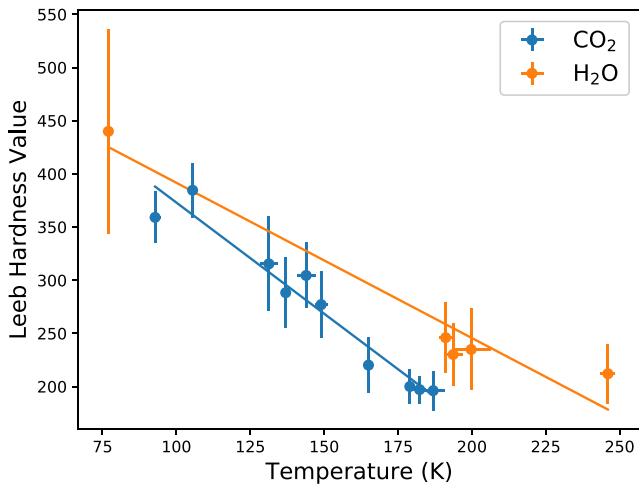
### 3. Results

The results of our measurements of Leeb hardness of CO<sub>2</sub> and H<sub>2</sub>O ice are shown in Tables 1 and 2, respectively. Listed values were obtained using the C-type sensor unless otherwise noted and are graphically summarized in Figure 3. Comparing the values measured for CO<sub>2</sub> ice with both types of sensor at different temperatures, values measured with the C-type sensor are about 19% higher. In the case of H<sub>2</sub>O ice this factor is slightly larger; values obtained with the C-type values are approximately 32% higher than the D-type values. Visual inspection of the ice surface after each experiment revealed small indentations of roughly the size of the impactor, to have formed, but no new fracturing occurred.

**Table 2**  
*Hardness Values for H<sub>2</sub>O Ice at Different Temperatures*

Temperature (K)	Leeb hardness (HL) mean $\pm$ st. dev.	No. of measurements
79.4 $\pm$ 1.1	440.0 $\pm$ 96.0	50
191.1 $\pm$ 2.3	246.0 $\pm$ 33.1	37
193.7 $\pm$ 3.1	230.2 $\pm$ 29.5	33
199.8 $\pm$ 6.5	234.9 $\pm$ 38.4	60
246.0 $\pm$ 2.4	212.1 $\pm$ 28.0	67





**Figure 3.** Leeb hardness test values for CO<sub>2</sub> and H<sub>2</sub>O ice at different temperatures with the C-type sensor.

We found an approximately linear decrease in Leeb hardness with increasing temperature for both CO<sub>2</sub> and H<sub>2</sub>O ice, as can be seen in Figure 3. An increase in hardness with decreasing temperature (away from the melting/sublimation point) is expected for icy materials and is fully in line with the H<sub>2</sub>O ice hardness results of Henderson et al. (2019). The best fit relation for CO<sub>2</sub> is  $HL = -2.089T + 582$ , demonstrating a steeper temperature relation than water ice. CO<sub>2</sub> also has a larger thermal expansion coefficient than water ice (Mangan et al., 2017), potentially showing that its physical properties are more temperature dependent. When considering the uncertainties in Figure 3, however, the differences are not large.

#### 4. Discussion

Hardness, as quantified by the Leeb values, represents the ability of a material to withstand indentation by plastic deformation. It can also be expressed in MPa and various other dimensionless scales (Vickers, Brinell, etc.) related to each other by empirical conversion tables. A material's hardness is itself dependent on other, more intrinsic, properties such as its elastic stiffness (Young's modulus) and the stress at which it transitions

from an elastic to plastic response (yield strength). The relation between these properties, and between yield strength and ultimate breaking strength, under compressive (such as in a hardness test), shear or tensile stresses, is complex and material dependent.

For the impact of a sphere onto a plate, where the solid plate only undergoes plastic deformation and no strain hardening or significant fracturing, the coefficient of restitution typically has a power law dependence on impact velocity and can be related to the dynamic yield strength  $Y$  by Johnson (1987) and Henderson et al. (2019)

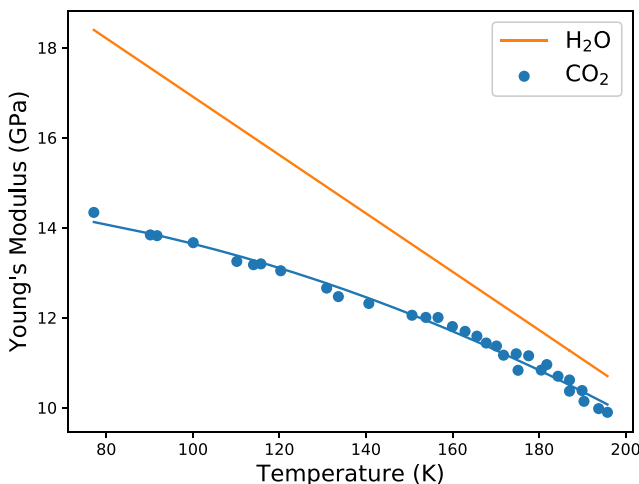
$$Y = \left[ \frac{HL}{1000} \times \frac{E^*}{3.1} \frac{v^{1/4} \rho^{1/8}}{1} \right]^{8/5} \quad (2)$$

Here,  $v$  is the impact speed (1.34 and 2 m s<sup>-1</sup> for the C-type and D-type sensors, respectively),  $\rho$  is the target material density, and  $E^*$  is the reduced Young's modulus, given by

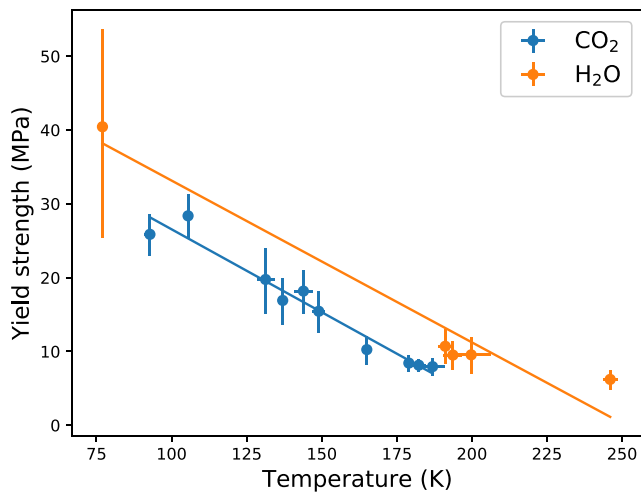
$$E^* = \left[ \frac{1 - \nu_I^2}{E_I} + \frac{1 - \nu_T^2}{E_T} \right]^{-1} \quad (3)$$

with  $\nu_I = 0.18$  and  $E_I = 621$  GPa Poisson's ratio and Young's modulus of the tungsten carbide impactor (Henderson et al., 2019). For the CO<sub>2</sub> ice target,  $\nu_T = 0.544$  (Yamashita & Kato, 1997), and we obtain Young's modulus,  $E_T$ , from the temperature-dependent longitudinal wave velocity  $v_L$ , measured by Yamashita and Kato (1997), and density, calculated in equation (1) of Mangan et al. (2017), with the relation  $E = \rho v_L^2$ . The resulting Young's modulus over the relevant temperature range is shown in Figure 4, with a best fit curve of  $E = -1.361 \times 10^5 T^2 + 2.985 \times 10^6 T + 1.47 \times 10^{10}$ . For comparison, water ice at the same temperature (Gold, 1958) is  $\sim 1.1$ – $1.3$  times stiffer. This agrees with the results of Clark and Mullin (1976), who found CO<sub>2</sub> ice more likely to flow than water ice under the same stress.

With the above parameters we can calculate yield strengths of our ice samples. To do this, we must assume that they behave in a purely plastic way during the impact and that no strain hardening occurs, that is, the power law dependence of coefficient of restitution holds. For the former, the formation of small indentations in the sample of a similar size to the test impactor (3–5 mm) suggests plastic deformation, and no visible fractures are observed. For the latter there is no existing experimental evidence for CO<sub>2</sub> ice, while for water ice at higher temperatures,



**Figure 4.** Young's modulus with temperature for water ice (Gold, 1958) and CO<sub>2</sub> ice (calculated from Yamashita & Kato, 1997, and Mangan et al., 2017).

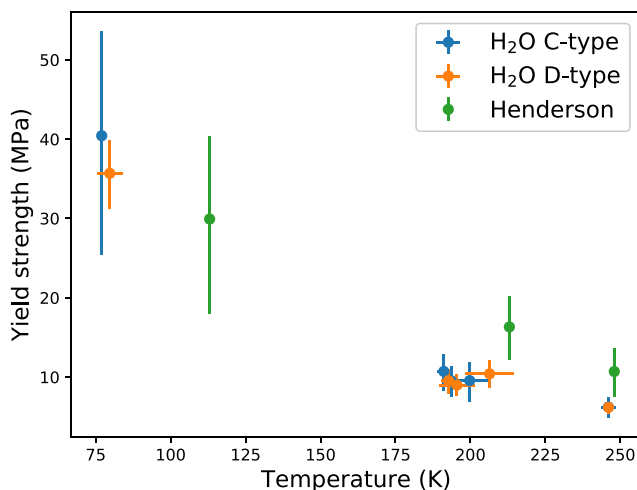


**Figure 5.** Calculated yield strengths for CO<sub>2</sub> and H<sub>2</sub>O ice. The best fit line for CO<sub>2</sub> ice is  $Y(\text{Pa}) = -2.247 \times 10^5 T + 4.902 \times 10^7$ .

respond to Brinell hardness values  $H < 130$ , suggesting an ultimate compressive strength of  $\sigma < 50$  MPa when using the above scaling relation. This is similar to the  $\sigma \sim 20\text{--}60$  MPa estimate above, despite the uncertainty of using a conversion table for a different material. The relationship between tensile, shear, and compressive strength is also complex and material dependent, but in most materials it is found that compressive strength exceeds shear strength and that this in turn exceeds tensile strength. Rocks and geological materials, for example, typically have compressive strengths  $\sim 10$  times larger than tensile (Sheorey, 1997), which suggests an ultimate CO<sub>2</sub> ice tensile strength of  $\sim 2\text{--}6$  MPa from our results or  $< 5$  MPa from the steel scaling relationship. This is in broad agreement with the suggested  $\sim 2$  MPa tensile strength of water ice under Martian conditions used in previous works (Mellon, 1997).

As a further comparison, we show calculated yield strengths for our H<sub>2</sub>O samples, taken with both the C-type and D-type sensors, alongside Henderson et al. (2019) values for slowly crystallized water ice in Figure 6. We find a very good match between sensor types, validating our use of the C-type sensor for the CO<sub>2</sub> ice, as well as broad agreement with Henderson et al. (2019) for H<sub>2</sub>O ice, bearing in mind the large uncertainties in both experiments.

Thus, we can conclude by saying that at typical Martian temperatures, CO<sub>2</sub> ice is less hard than H<sub>2</sub>O ice, as well as being less stiff (lower Young's modulus Gold, 1958 and larger creep coefficients Clark & Mullin, 1976; Durham et al., 1999) and with a similar or slightly lower yield strength (this work). It would therefore



**Figure 6.** Calculated yield strengths for H<sub>2</sub>O ice using two different sensor types and from Henderson et al. (2019).

Higa et al. (1998) find that the coefficient of restitution does not follow a power law. In light of the lack of further data concerning CO<sub>2</sub> ice, and for ease of comparison with other works, we follow the method of Henderson et al. (2019) in assuming no strain hardening and in using the Johnson (1987) equation. We must advise caution when using the following results for absolute values. Nevertheless, relative values between the two ice types should always be applicable.

Figure 5 shows the results of equation (2). Our CO<sub>2</sub> ice samples have an estimated compressive yield strength of between  $\sim 10$  and 30 MPa over the temperature range of  $\sim 100\text{--}200$  K, with a similar temperature dependence to water ice but  $\sim 8$  MPa smaller magnitude. Yield strength is typically smaller by a few times than ultimate strength  $\sigma$  (at which fractures will occur), so our results suggest a compressive fracturing strength for CO<sub>2</sub> ice of  $\sigma \sim 20\text{--}60$  MPa. For comparison with an alternate method, the scaling relation  $\sigma = 0.383H$  has been used previously for water ice (Epifanov, 2004), to relate ultimate strength,  $\sigma$ , to dynamic hardness,  $H$ , measured by the Brinell method. According to typical conversion tables for steel (e.g., ShapeCUT Steel, 2016), our Leeb hardness values corre-

spond to Brinell hardness values  $H < 130$ , suggesting an ultimate compressive strength of  $\sigma < 50$  MPa when using the above scaling relation. This is similar to the  $\sigma \sim 20\text{--}60$  MPa estimate above, despite the uncertainty of using a conversion table for a different material. The relationship between tensile, shear, and compressive strength is also complex and material dependent, but in most materials it is found that compressive strength exceeds shear strength and that this in turn exceeds tensile strength. Rocks and geological materials, for example, typically have compressive strengths  $\sim 10$  times larger than tensile (Sheorey, 1997), which suggests an ultimate CO<sub>2</sub> ice tensile strength of  $\sim 2\text{--}6$  MPa from our results or  $< 5$  MPa from the steel scaling relationship. This is in broad agreement with the suggested  $\sim 2$  MPa tensile strength of water ice under Martian conditions used in previous works (Mellon, 1997).

As a further comparison, we show calculated yield strengths for our H<sub>2</sub>O samples, taken with both the C-type and D-type sensors, alongside Henderson et al. (2019) values for slowly crystallized water ice in Figure 6. We find a very good match between sensor types, validating our use of the C-type sensor for the CO<sub>2</sub> ice, as well as broad agreement with Henderson et al. (2019) for H<sub>2</sub>O ice, bearing in mind the large uncertainties in both experiments.

Thus, we can conclude by saying that at typical Martian temperatures, CO<sub>2</sub> ice is less hard than H<sub>2</sub>O ice, as well as being less stiff (lower Young's modulus Gold, 1958 and larger creep coefficients Clark & Mullin, 1976; Durham et al., 1999) and with a similar or slightly lower yield strength (this work). It would therefore appear that CO<sub>2</sub> ice landforms should be more susceptible to fracturing than water ice features, when experiencing stresses such as from gas buildup in a solid-state greenhouse (Kieffer, 2007) or contraction from thermal cycling (Portyankina et al., 2012), the latter being especially relevant due to CO<sub>2</sub> ice's large thermal expansion coefficient relative to water ice (Mangan et al., 2017). This is, however, dependent on the rate of strain and physical scale of the relevant displacement. Our experimental results apply to a microscale, fast strain rate regime, as compared to slow processes such as glacial flow. At larger scales, grain boundaries and pre-existing fractures will reduce the calculated yield strength, and at slow strain rates, the large CO<sub>2</sub> creep values will act to diffuse stress by plastic deformation rather than fracturing. Application of our results to real features should therefore be underpinned by a full viscoelastic model of the relevant problem. Our hardness measurement, relative to water ice, does represent an important input parameter for such models. We conclude with the following simple, order of magnitude, calculation regarding such modeling. Portyankina et al. (2010) simulate the buildup of gas by the solid-state greenhouse effect in Martian araneiform features and estimate fracturing after 1–20 days at 75°S using H<sub>2</sub>O ice strengths of  $Y = 50$  kPa

to 100 MPa (note that there is a missing factor of 1,000, i.e., a missing “kilo,” in their Figure 9 and associated description, G. Portyankina; personal communication, July 2019). At  $T = 150$  K our results give  $Y \approx 15.8$  MPa, which, when inserted into the relationship displayed in their Figure 9, would lead to fracturing on the 18th day of gas buildup.

## 5. Conclusion

We have measured the hardness of solid CO<sub>2</sub> ice directly in the laboratory with a Leeb hardness tester. We have also measured the hardness of water ice by the same method, confirming previous work for water ice. Our results indicate that CO<sub>2</sub> ice is slightly weaker, with Leeb hardness values ranging between  $\sim 200$  and  $400$  ( $\sim 10$  and  $30$  MPa yield strength, assuming only plastic deformation and no strain hardening during the experiment), for typical Martian temperatures. This work supports the hypothesis that fracturing of solid CO<sub>2</sub> ice slabs (by basal sublimation or thermal stresses) plays an important role in morphological changes in the Martian polar regions. Future models of CO<sub>2</sub> ice fracturing should consider the timescales of plastic deformation when investigating surface-shaping processes.

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## References

- Alberti, A. P., Gomes, A., Trenhaile, A., Oliveira, M., & Horacio, J. (2013). Correlating river terrace remnants using an Equotip hardness tester: An example from the Miño River, northwestern Iberian Peninsula. *Geomorphology*, *192*, 59–70. <https://doi.org/10.1016/j.geomorph.2013.03.017>
- Aoki, H., & Matsukura, Y. (2008). Estimating the unconfined compressive strength of intact rocks from Equotip hardness. *Bulletin of Engineering Geology and the Environment*, *67*, 23–29. <https://doi.org/10.1007/s10064-007-0116-z>
- Clark, B. R., & Mullin, R. P. (1976). Martian glaciation and the flow of solid CO<sub>2</sub>. *Icarus*, *27*, 215–228. [https://doi.org/10.1016/0019-1035\(76\)90005-1](https://doi.org/10.1016/0019-1035(76)90005-1)
- Durham, W. B., Kirby, S. H., & Stern, L. A. (1999). Steady-state flow of solid CO<sub>2</sub>: Preliminary results. *Geophysical Research Letters*, *26*(23), 3493–3496. <https://doi.org/10.1029/1999GL008373>
- Epifanov, V. P. (2004). Rupture and dynamic hardness of ice. *Physics - Doklady*, *49*, 86–89. <https://doi.org/10.1134/1.1686876>
- Gold, L. W. (1958). Some observations on the dependence of strain on stress for ice. *Canadian Journal of Physics*, *36*, 1265. <https://doi.org/10.1139/p58-131>
- Henderson, B. L., Gudipati, M. S., & Bateman, F. B. (2019). Leeb hardness of salty Europa ice analogs exposed to high-energy electrons. *Icarus*, *322*, 114–120. <https://doi.org/10.1016/j.icarus.2019.01.006>
- Higa, M., Arakawa, M., & Maeno, N. (1998). Size dependence of restitution coefficients of ice in relation to collision strength. *Icarus*, *133*(2), 310–320. <https://doi.org/10.1006/icar.1998.5938>
- Johnson, K. L. (1987). *Contact mechanics*: Cambridge University Press.
- Kaufmann, E., Attree, N., Bradwell, T., & Hagermann, A. (2019). CO<sub>2</sub> ice hardness and yield strength under Martian temperature conditions. University of Stirling. Faculty of Natural Sciences. Dataset <http://hdl.handle.net/11667/138>
- Kaufmann, E., & Hagermann, A. (2017). Experimental investigation of insolation driven dust ejection from Mars CO<sub>2</sub> ice caps. *Icarus*, *282*, 118–126. <https://doi.org/10.1016/j.icarus.2016.09.039>
- Kieffer, H. H. (2007). Cold jets in the Martian polar caps. *Journal of Geophysical Research*, *112*, E08005. <https://doi.org/10.1029/2006JE002816>
- Kieffer, H. H., Titus, T. N., Mullins, K. F., & Christensen, P. R. (2000). Mars south polar spring and summer behavior observed by TES: Seasonal cap evolution controlled by frost grain size. *Journal of Geophysical Research*, *105*(E4), 9653–9700. <https://doi.org/10.1029/1999JE001136>
- Leeb, D. (1979). Dynamic hardness testing of metallic materials. *NDT International*, *12*, 274–278. [https://doi.org/10.1016/0308-9126\(79\)90087-7](https://doi.org/10.1016/0308-9126(79)90087-7)
- Malin, M. C., Caplinger, M. A., & Davis, S. D. (2001). Observational evidence for an active surface reservoir of solid carbon dioxide on Mars. *Science*, *294*(5549), 2146–2148. <https://doi.org/10.1126/science.1066416>
- Mangan, T. P., Salzmann, C. G., Plane, J. M. C., & Murray, B. J. (2017). CO<sub>2</sub> ice structure and density under Martian atmospheric conditions. *Icarus*, *294*, 201–208. <https://doi.org/10.1016/j.icarus.2017.03.012>
- Mellon, M. T. (1997). Small-scale polygonal features on Mars: Seasonal thermal contraction cracks in permafrost. *Journal of Geophysical Research*, *102*, 25,617–25,628. <https://doi.org/10.1029/97JE02582>
- Nye, J., Durham, W., Schenk, P., & Moore, J. (2000). The instability of a south polar cap on Mars composed of carbon dioxide. *Icarus*, *144*(2), 449–455. <https://doi.org/10.1006/icar.1999.6306>
- Portyankina, G., Markiewicz, W. J., Thomas, N., Hansen, C. J., & Milazzo, M. (2010). HiRISE observations of gas sublimation-driven activity in Mars' southern polar regions: III. Models of processes involving translucent ice. *Icarus*, *205*, 311–320. <https://doi.org/10.1016/j.icarus.2009.08.029>
- Portyankina, G., Merrison, J., Iversen, J. J., Yoldi, Z., Hansen, C. J., Aye, K. M., et al. (2019). Laboratory investigations of the physical state of CO<sub>2</sub> ice in a simulated Martian environment. *Icarus*, *322*, 210–220. <https://doi.org/10.1016/j.icarus.2018.04.021>
- Portyankina, G., Pommerol, A., Aye, K.-M., Hansen, C. J., & Thomas, N. (2012). Polygonal cracks in the seasonal semi-translucent CO<sub>2</sub> ice layer in Martian polar areas. *Journal of Geophysical Research*, *117*, E02006. <https://doi.org/10.1029/2011JE003917>
- ShapeCUT Steel (2016). Hardness conversion table. Retrieved 2019-06-19, from <https://www.shapecut.com.au/wp-content/uploads/2016/01/ShapeCUT-Steel-Hardness-Conversion-Table.pdf>
- Shoorey, P. R. (1997). *Empirical rock failure criterion*. CRC Press.
- Smith, R. L., & Sandland, G. E. (1925). Some notes on the use of a diamond pyramid for hardness testing. *Journal of the Iron and Steel Institute*, *111*, 285–294.
- Yamashita, Y., & Kato, M. (1997). Viscoelastic properties of polycrystalline solid methane and carbon dioxide. *Geophysical Research Letters*, *24*, 1327–1330. <https://doi.org/10.1029/97GL01205>