

# Removing polymeric coatings with nanostructured fluids: influence of substrate, nature of the film, and application methodology

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The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest

### *Author contribution statement*

M.B, M.A., and D.C. contributed to the conception and design of the work, to the acquisition, the analysis, and the interpretation of experimental data.

R.G. and P.B. revised the work critically giving a substantial intellectual contribution, and ultimately provided the final approval for its publication.

### *Keywords*

Polymer coatings, cleaning, nanostructured fluids, Acrylics, vinyls, Nonionic surfactants, Hydrogels, dewetting

### *Abstract*

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Cleaning is one of the most important and delicate operations in the conservation of cultural heritage, and, if not correctly performed, may irreversibly damage works of art. The removal of aged or detrimental polymeric coatings from works of art is a common operation in conservation, and nanostructured fluids (NSFs), such as aqueous swollen micelles and oil-in-water (o/w) microemulsions, are used as an alternative to non-confined organic solvents that pose a series of non-negligible drawbacks. NSFs effectiveness in removing polymeric coatings has been thoroughly demonstrated in the last decades, while their cleaning mechanism is still under investigation. The present work deepens the knowledge on the removal mechanisms of NSFs, studying the interaction of a four-component NSF with four different types of acrylic and vinyl polymer films cast from solutions or aqueous polymer latexes on three substrates (glass, marble and polystyrene) with different hydrophilicity and wettability. NSFs were applied either as non-confined or confined in cellulose poultices (traditionally employed by conservators), or in highly retentive chemical gels, observing the influence of the confining matrix on the removal process. It was found that the NSF/polymer film interaction is greatly dependent on the film structure and composition. Films formed from solvent solutions can be swollen by water/organic solvents mixtures or dewetted when a surfactant is added to the cleaning fluid; films formed from polymer latexes, on the other hand, are generally swollen even just by water alone, but poorly dewet. The substrate also plays an important role in the removal of polymer films formed from solutions, for instance the removal of an acrylic polymer from polystyrene could be achieved only through highly selective cleaning using NSF-loaded chemical hydrogels. These results can be key for conservators, providing innovative solutions to face new challenges in art preservation.

### *Contribution to the field*

The development of innovative materials for conservation of cultural heritage poses a twofold challenge: on one side they need to meet the expectations and need of conservators and restorers from a purely applicative standpoint, while, on the other, their physico-chemical interaction with artworks' materials need to be thoroughly understood. This paper provides not previously reported results on the removal of polymeric coatings by means of nanostructured fluids. The different behavior of polymer films cast from solvents' solutions with respect to films formed from aqueous polymer latexes is explored and described for the very first time, here, improving the knowledge about physico-chemical properties of such materials. Finally, the influence of the substrate on polymer films removal is systematically studied for the first time, and it was shown that, when the affinity between the polymer coating and the substrate it is laid on is particularly high, the optimal way to selectively and completely remove it is to confine a nanostructured fluid into a highly-retentive hydrogel, which enhance the control on the cleaning action. We believe that both conservators and scientists can find their interest in reading this paper.

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In review

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13 **Keywords: polymer coatings, cleaning, nanostructured fluids, acrylics, vinyls, nonionic**  
14 **surfactants, hydrogels, dewetting**

15 **Abstract**

16 Cleaning is one of the most important and delicate operations in the conservation of cultural heritage,  
17 and, if not correctly performed, may irreversibly damage works of art. The removal of aged or  
18 detrimental polymeric coatings from works of art is a common operation in conservation, and  
19 nanostructured fluids (NSFs), such as aqueous swollen micelles and oil-in-water (o/w)  
20 microemulsions, are used as an alternative to non-confined organic solvents that pose a series of non-  
21 negligible drawbacks. NSFs effectiveness in removing polymeric coatings has been thoroughly  
22 demonstrated in the last decades, while their cleaning mechanism is still under investigation. The  
23 present work deepens the knowledge on the removal mechanisms of NSFs, studying the interaction  
24 of a four-component NSF with four different types of acrylic and vinyl polymer films cast from  
25 solutions or aqueous polymer latexes on three substrates (glass, marble and polystyrene) with  
26 different hydrophilicity and wettability. NSFs were applied either as non-confined or confined in  
27 cellulose poultices (traditionally employed by conservators), or in highly retentive chemical gels,  
28 observing the influence of the confining matrix on the removal process. It was found that the  
29 NSF/polymer film interaction is greatly dependent on the film structure and composition. Films  
30 formed from solvent solutions can be swollen by water/organic solvents mixtures or dewetted when a  
31 surfactant is added to the cleaning fluid; films formed from polymer latexes, on the other hand, are  
32 generally swollen even just by water alone, but poorly dewet. The substrate also plays an important  
33 role in the removal of polymer films formed from solutions, for instance the removal of an acrylic  
34 polymer from polystyrene could be achieved only through highly selective cleaning using NSF-

35 loaded chemical hydrogels. These results can be key for conservators, providing innovative solutions  
36 to face new challenges in art preservation.

37

## 38 **1. Introduction**

39 Cleaning of works of art generally consists in the selective removal of materials that promote the  
40 degradation of the artifacts or alter their readability and appearance. Among these materials, aged or  
41 detrimental polymeric coatings are often found on works of art, and their removal is a common  
42 operation in art conservation. Synthetic polymers have been largely employed in the traditional  
43 restoration practice as varnishes, adhesives, protectives and consolidating agents. The presence of  
44 polymeric coatings on the surface of porous inorganic substrates (wall paintings, stone, mortars)  
45 drastically reduces water permeability and enhances the degradation induced by salts, up to  
46 consistent loss of the artifacts' surface layers (Carretti and Dei, 2004; Giorgi et al., 2010). Graffiti  
47 and vandalism are other well-known examples where selective removal of polymeric coatings must  
48 be carried out on artistic surfaces (Apostol et al., 2011; Baglioni et al.; Giorgi et al., 2017; Sanmartín  
49 et al., 2014). Other cases include the removal of aged pressure sensitive tapes from paper artworks  
50 (Bonelli et al., 2018), or of discolored and cracked varnishes from paintings (Baglioni et al., 2018a;  
51 Burnstock and Kieslich, 1996).

52 The vast amount of different solvent-sensitive materials found in classic and contemporary art poses  
53 continuous challenges to the safe removal of detrimental coatings (Kavda et al., 2017). Traditionally,  
54 restorers and conservators rely on the use of organic solvents to dissolve or swell unwanted materials.  
55 Solvents are typically applied either as non-confined, using cotton swabs, or thickened in viscous  
56 polymeric solutions and solvent gels (Baglioni et al., 2012c; Burnstock and Kieslich, 1996;  
57 Burnstock and White, 2000). However, these methods exhibit poor control and scarce selectivity, or  
58 involve the presence of residues from the cleaning system. In addition, health concerns arise from the  
59 toxicity of most solvents used in restoration.

60 Alternatively, nanostructured fluids (NSFs) such as aqueous swollen micelles and oil-in-water (o/w)  
61 microemulsions were proposed in the late 1980s for the removal of hydrophobic matter from porous  
62 inorganic substrates (Borgioli et al., 1995). Their effectiveness in removing polymeric coatings from  
63 different types of surfaces has been thoroughly demonstrated in the last decades (Baglioni et al.,  
64 2010, 2012b, 2015a, 2015b, 2016, 2018c; Carretti et al., 2003, 2007). These NSFs are water-based  
65 systems where limited amounts of solvents are found either in the dispersed (as nano-sized droplets  
66 stabilized by surfactants) or in the continuous phase of the fluids. The large surface area developed  
67 by the nano-sized droplets, and the synergistic action of solvents and surfactants, are responsible for  
68 enhanced cleaning power, while the presence of the continuous aqueous phase limits the re-  
69 deposition of detached hydrophobic matter on hydrophilic surfaces. Besides, being water-based,  
70 aqueous NSFs have reduced toxicity as opposed to traditional solvents. A further improvement was  
71 represented by the confinement of NSFs into retentive physical or chemical gels able to gradually  
72 release the fluids at the gel-artifact interface, maximizing control over the cleaning action (Baglioni  
73 et al., 2015c; Chelazzi et al., 2018).

74 While these systems represent the most advanced cleaning tools currently available to conservators,  
75 the interaction mechanism of NSFs with polymeric coatings is still being investigated.

76 The removal of high molecular weight macromolecules does not follow the rules of classical  
77 detergency. Studies performed on model samples, i.e. glass slides coated with an acrylate copolymer,

78 have shown that dewetting takes place when water-based NSF's are put in contact with the polymeric  
79 coating (Baglioni et al., 2017, 2018b; Raudino et al., 2015, 201, 2017). The film swells and detaches  
80 from the surface. This behavior was also observed on mortar samples coated with the same polymer  
81 (Baglioni et al., 2018b). The nature of both the organic solvents and the surfactant(s) included in the  
82 cleaning formulation was found to be of fundamental importance in the whole process. The solvents  
83 swell the polymer and lower its glass transition temperature, increasing the mobility of the polymer  
84 chains; the surfactant reduces the interfacial tension, kinetically promoting the detachment of the  
85 polymer film from the surface, and initiating the dewetting process that eventually breaks down the  
86 coating into separate polymer droplets.

87 Despite providing fundamental insights, previous studies on this subject depicted only a partial  
88 picture of the interaction mechanisms between NSF's and polymeric coatings. Namely, all the studies  
89 were conducted only on a single type of polymer, i.e. Paraloid B72® (poly(ethyl methacrylate/methyl  
90 acrylate, 70:30) cast from solution. Only glass or mortars were considered as substrates. Besides,  
91 simplified NSF's formulations were used to isolate the contribution of individual components in the  
92 system.

93 The present work aims to deepen the knowledge on the removal mechanisms of NSF's, studying the  
94 interaction of a four-component NSF with four different types of polymer films cast either from  
95 solutions or aqueous emulsions. In fact, it is well known that films of polymers cast from solutions  
96 have different physical structures than those cast from emulsions (Chevalier et al., 1992; Steward et  
97 al., 2000; Winnik, 1997), and polymers emulsions contain several additives; these can be crucial  
98 factors in determining the interaction process with the NSF's. Three substrates were considered (glass,  
99 marble and polystyrene) with different hydrophilicity and wettability. Finally, NSF's were applied  
100 either as non-confined or confined in cellulose poultices (traditionally employed by conservators), or  
101 in highly retentive chemical gels, observing the influence of the confining matrix on the removal  
102 process.

103 The NSF is composed of water, an alcohol ethoxylate nonionic surfactant, 2-butanol (BuOH), and 2-  
104 butanone (methyl ethyl ketone, MEK), and is representative of formulations actually employed in the  
105 conservation practice for the cleaning of real works of art.

106 The three selected substrates were coated with the four different polymers, obtaining a set of 12  
107 samples, which were then exposed to the NSF, investigating the interaction mechanism for each  
108 combination. The effects of the NSF on the films were studied by means of optical microscopy and  
109 micro-reflectance infrared Fourier Transform spectroscopy 2D mapping of the areas of interest,  
110 which provides spatial resolution down to the micron-scale.

## 111 **2. Materials and methods**

### 112 **2.1. Chemicals**

113 C<sub>9-11</sub>E<sub>5,5</sub> alcohol ethoxylate (Berol 266, AkzoNobel), sodium dodecylsulfate (SDS, Sigma-Aldrich,  
114 purity ≥ 99%), butanone (MEK, Sigma-Aldrich, purity 99%), 2-butanol (BuOH, Sigma-Aldrich,  
115 purity 99%), ethanol (EtOH, Fisher Chemical, purity ≥ 99%), white spirit (White Spirit, Fidea), D<sub>2</sub>O  
116 (EurisoTop, 98%), and deuterated butanone (d-MEK, C<sub>4</sub>H<sub>3</sub>D<sub>5</sub>O, Sigma-Aldrich, purity 98%) were  
117 used as received, without further purification. Water was purified with a Millipore Milli-Q gradient  
118 system (resistance > 18 MΩ cm).

119

120

121 **2.2. Polymers formulations**

122 The four polymeric materials selected are widely used in the traditional restoration practice. They  
 123 include: poly(vinyl acetate) (PVAc), commercially known as PVA K40®, dissolved in ethanol;  
 124 PVAc as an aqueous emulsion, commercially known as Vinavil NPC®; poly(butyl methacrylate)  
 125 (PBMA) dissolved in white spirit, commercially known as Plexisol P550®; poly(ethyl  
 126 acrylate/methyl methacrylate) (PEA/PMMA) 40:60 as an aqueous emulsion, commercially known as  
 127 Plextol B500®. **Table 1** reports the main properties of the formulations. All the products were  
 128 applied on the substrates as a 10% (w/w) solution/emulsion.

129 **Table 1.** Main properties and sample names of the four commercial polymers used.

Commercial name	Polymer nature	Physical appearance	Dry matter content	Applied as	Sample name <sup>a</sup>	Film density (g/cm <sup>3</sup> ) <sup>b</sup>	T <sub>g</sub> (°C) <sup>c</sup>
PVA K40®	PVAc	Transparent grains	-	10% solution in EtOH	V <sub>S</sub>	1.27	31
Vinavil NPC®	PVAc	Aqueous emulsion	52%	10% aqueous emulsion	V <sub>E</sub>	1.05	16
Plexisol P550®	PBMA	40% solution in white spirit	40%	10% solution in white spirit	A <sub>S</sub>	1.08	28
Plextol B500®	PEA/PMMA	Aqueous emulsion	50%	10% aqueous emulsion	A <sub>E</sub>	1.03	30

130 <sup>a</sup>V = vinyl; A = acrylic; S = polymer applied as a solution; E = polymer applied as an emulsion. <sup>b</sup>The density was  
 131 calculated as described in Section 2.3. <sup>c</sup>The glass temperature (T<sub>g</sub>) was measured by means of DSC analyses according to  
 132 the procedure described in Section 2.2.1.

133

134 **2.2.1. Differential scanning calorimetry (DSC) for the glass temperature (T<sub>g</sub>) determination**

135 DSC measurements were performed on a DSC Q1000 from TA Instruments on small samples (2-5  
 136 mg) of the four dry films, according to the following procedure: equilibration at -50 °C; heating ramp  
 137 from -50°C to 60°C at 10 °C/min; cooling ramp from 60 °C to -50 °C 10°C/min; heating ramp from -  
 138 50°C to 60°C at 10 °C/min. The first heating/cooling ramp was used to equilibrate the sample, while



139 the second heating ramp is the one used for the T<sub>g</sub> determination. Each sample was run at least twice,  
140 in order to check for reproducibility of the measurement.

141

### 142 **2.3. Substrates**

143 The three different substrates selected for this work, i.e., glass, marble and polystyrene, were selected  
144 as they exhibit different wettability and are representative of artistic substrates frequently found in  
145 classic or contemporary art production. Glass slides of  $5 \times 5 \times 0.3 \text{ cm}^3$ , marble tiles of  $5 \times 5 \times 1 \text{ cm}^3$ ,  
146 and polystyrene slides of  $5 \times 5 \times 0.2 \text{ cm}^3$  were used. The four different filming materials (polymers  
147 solutions or emulsions) were then laid manually on the top surfaces of each series of samples. 150  $\mu\text{l}$   
148 of each polymer formulation ( $V_S$ ,  $V_E$ ,  $A_S$ ,  $A_E$ ) was carefully and homogeneously spread on a  $4 \times 5$   
149  $\text{cm}^2$  area on the surface of glass, marble and polystyrene samples. A reference area was thus left  
150 untreated. The specimens were left drying for 30 days, and then the dry mass of the films was  
151 evaluated. The density of the dry films was then measured, by drying for 4 days in a ventilated oven  
152 at 60 °C weighed amounts of polymer solutions/emulsions that were poured into graduated vials. The  
153 results, reported in **Table 1**, made possible to calculate that the obtained films had an average  
154 thickness of approximately 5-7  $\mu\text{m}$ , assuming that they are homogeneously spread all over the treated  
155 surface.

156

### 157 **2.4. Attenuated total reflectance Fourier-transform infrared spectroscopy**

158 ATR measurements were carried out with a Thermo Nicolet Nexus 870 spectrometer equipped with a  
159 liquid nitrogen-cooled MCT detector and a single reflection diamond crystal ATR unit. Spectra were  
160 recorded in the 4000-650  $\text{cm}^{-1}$  range (128 scans, 4  $\text{cm}^{-1}$  resolution). ATR measurements were  
161 performed on small samples of spray-can paints' dried films, removed from the glass support.

162

### 163 **2.5. NSF preparation**

164 The NSF selected for this work is composed as follows (w/w): H<sub>2</sub>O, 65.9%; C<sub>9-11</sub>E<sub>5,5</sub>, 3.5%; BuOH,  
165 9.7%; MEK, 20.9%. In order to better understand the role of each component in the interaction with  
166 the polymer coatings, other liquid systems were also used, obtained removing some of the  
167 components from the complete NSF formulation. Namely, a water/C<sub>9-11</sub>E<sub>5,5</sub> surfactant solution, and a  
168 water/BuOH/MEK solvents mixture were used. In both systems, the ratio between each component  
169 was the same as in the complete NSF formulation. It is worth noting that both BuOH and MEK are  
170 partly water-miscible (12.5% and 24% at 20°C, respectively (Verschueren, 2001)) and this made  
171 possible to obtain a single-phase stable mixture of water and solvents even in the absence of  
172 surfactants.

173

### 174 **2.6. Small-Angle Neutron Scattering**

175 Small-angle neutron scattering (SANS) experiments were performed on the spectrometer V4 (Bensch-  
176 Helmholtz Zentrum Berlin). Two different configurations were employed (i.e., sample-to-detector

177 distances,  $SD = 2$  or  $8$  m) to cover a range of wave vectors  $q$  ( $q = (4\pi/\lambda)\sin(\theta/2)$ , where  $\lambda$  is the  
178 wavelength of the incident neutron beam and  $\theta$  the scattering angle) from  $0.007$  to  $0.28 \text{ \AA}^{-1}$ . For each  
179 configuration a  $6 \text{ \AA}$  neutron wavelength was used and the wavelength resolution,  $\Delta\lambda/\lambda$ , was less than  
180 10%. Samples were contained in  $1$  mm thick quartz cells and kept at  $20 \pm 2 \text{ }^\circ\text{C}$  during the  
181 measurements. The scattering intensity was corrected for the empty cell contribution, transmission,  
182 and detector efficiency and was normalized to the absolute scale by direct measurement of the  
183 intensity of the incident neutron beam. The integration of the normalized 2D intensity distribution  
184 with respect to the azimuthal angle yielded the 1D scattering intensity distribution,  $I(q)$ , in  $\text{cm}^{-1}$ . The  
185 reduction of the data was performed using standard BENSIC procedures for small-angle isotropic  
186 scattering. The background from the incoherent scattering coming from each sample was determined  
187 from analysis of the Porod asymptotic limit and subtracted from the normalized spectra.  
188 Experimental data normalized to absolute scale were fitted using Igor routines (NCNR\_SANS\_  
189 package\_6.011) (Kline, 2006) available from NIST, National Institute for Standard and Technology,  
190 Gaithersburg, MD, running on Igor Pro© (Wavemetric Inc., Lake Oswego, Oregon; Version 6.22).

191

## 192 2.7. NSF/polymer interaction experiments

### 193 2.7.1. Immersion tests

194 The aim of these tests was to investigate the direct interaction between cleaning fluids and polymer  
195 films, without the influence of a medium (i.e., cellulose pulp poultice or gel). To this purpose, the  
196 coated specimens were partly immersed in  $32$  ml of the selected fluids. The area of the polymer film  
197 exposed to the fluid action was kept constant and fixed to  $1 \times 5 \text{ cm}^2$ . At selected time intervals, up to  
198 10 minutes, the specimens were extracted from the fluid and let dry, and then visual inspection of  
199 possible changes in the film was carried out. At  $t = 0$  min and  $t = 10$  min the superficial  
200 micromorphology of the films (after drying) was investigated by optical microscopy and FTIR 2D  
201 micro-reflectance mapping. Finally, after 10 min of incubation with each liquid system, a removal  
202 test was performed on all the samples, using a cotton swab soaked with water, in order to check the  
203 removability of the polymeric coatings.

204

### 205 2.7.2. Confocal laser scanning microscopy (CLSM)

206 Confocal Microscopy experiments were performed on a Leica TCS SP8 confocal microscope (Leica  
207 Microsystems GmbH, Wetzlar, Germany) equipped with a  $63\times$  water immersion objective. The four  
208 polymers were stained with Coumarin 6, which was dissolved/dispersed in the liquid solutions or  
209 latexes. Then,  $2\text{--}4 \text{ }\mu\text{m}$ -thick polymer films were obtained by spin coating coverglasses for  $60$  s at  
210  $1000$  rpm. Coumarin 6 was excited with the  $488$  nm laser line of an argon laser. The emission of the  
211 dye was acquired with a PMT in the range  $498\text{--}530$  nm. CLSM experiments were performed to  
212 monitor the interaction of the different polymer films with the NSF. Briefly,  $50 \text{ }\mu\text{l}$  of the unlabeled  
213 liquid phase were put in contact with the coumarin 6-labeled polymer coated coverglass, and the  
214 morphological variations of the polymeric film were monitored over time.

215

### 216 2.7.2. Cleaning tests

217 Two sets of cleaning tests were performed, in order to evaluate the influence on the application  
218 method of the NSF, and the effect of the supporting material (i.e., cellulose pulp poultice or chemical  
219 hydrogel) on the interaction between the NSF and the polymeric coating. Traditional cellulose pulp  
220 poultices were prepared by mixing 35 mg of cellulose pulp (Arbocel®, Zecchi, Firenze) and 160  $\mu\text{L}$   
221 of NSF. The obtained poultices were applied, interposing a Japanese paper sheet (Zecchi, Firenze)  
222 between the compress and the specimen surface, on 1  $\text{cm}^2$  areas of each sample for 15 min. After  
223 removing the poultices, a gentle mechanical action was performed on the treated surface, by means of  
224 wet cotton swabs. Poly(hydroxyethyl methacrylate/poly vinylpyrrolidone) semi-interpenetrated  
225 networks (pHEMA/PVP SIPNs) are highly retentive hydrogels, previously proposed in combination  
226 with NSFs for the cleaning of works of art (Baglioni et al., 2018a), and prepared as described  
227 elsewhere (Domingues et al., 2013a, 2013b). Gels having a size of  $1 \times 1 \times 0.2 \text{ cm}^3$  were used, able to  
228 upload ca. 160  $\mu\text{L}$  of NSF, i.e., the same amount of cleaning fluid loaded in the poultices. The gel was  
229 applied for 15 min, and at the end of the treatment a gentle mechanical action was performed on the  
230 coating residues.

231

## 232 **2.8. Contact angle measurements**

233 The contact angle of 5  $\mu\text{L}$  sessile droplets of Milli-Q water on the selected material surface was  
234 measured with a Rame-Hart Model 190 CA Goniometer. The three substrates and the four polymer  
235 films were investigated. The equilibrium contact angle was measured in at least eight different areas,  
236 and the average value and standard deviation was evaluated.

237

## 238 **2.9. Optical microscopy**

239 A Reichert Zetopan 353–890 microscope was used to collect micrographs. The instrument was  
240 coupled with a Nikon Digital Sight DS-Fi2 camera. The NIS-ELEMENTS software was used to  
241 capture and edit images.

242

## 243 **2.10. Fourier-transform infrared spectroscopy – 2D Micro-reflectance mapping**

244 The Fourier transform infrared (FTIR) 2D imaging of the treated surfaces was carried out using a  
245 Cary 620-670 FTIR microscope, equipped with an FPA  $128 \times 128$  detector (Agilent Technologies).  
246 The spectra were recorded directly on the surface of the samples in reflectance mode, with open  
247 aperture and a spectral resolution of  $8 \text{ cm}^{-1}$ , acquiring 128 scans for each spectrum. Each analysis  
248 produces an IR map of  $700 \times 700 \mu\text{m}^2$  ( $128 \times 128$  pixels), with a spatial resolution of  $5.5 \mu\text{m}$  (i.e.  
249 each pixel has dimensions of  $5.5 \times 5.5 \mu\text{m}^2$  and is associated to an independent spectrum). In each  
250 map, the intensity of a characteristic peak of each polymer, e.g. the C=O ester stretching typical of  
251 both acrylics and vinyls, at about  $1730 \text{ cm}^{-1}$ , was shown with a chromatic scale, following the order  
252 red > yellow > green > blue.

253

## 254 **3. Results and discussion**

255 **3.1. Polymer films characterization**

256 The four polymeric coatings were characterized by means of FTIR-ATR and contact angle  
 257 measurements. **Table 2** reports the values for the contact angle of water droplets laid on the surface  
 258 of the polymer films, or of the selected substrates (glass, marble, polystyrene). The substrates exhibit  
 259 a range of contact angles passing from hydrophilic (glass) to hydrophobic surfaces (polystyrene).  
 260 Regarding the polymers, it can be noticed that the acrylics are more hydrophobic than the vinyls, and  
 261 the films deriving from polymer latexes are more hydrophilic than those coming from solutions. This  
 262 is likely due to the presence of surfactants, stabilizers and other polar additives in the latexes  
 263 emulsions, which affect the wettability of the film.

264 **Table 2.** Contact angle of water droplets on the surface of the four selected polymeric coatings, and of the three types of  
 265 substrates.

Sample	Contact angle	Substrate	Contact angle
V <sub>S</sub>	57.8 ± 0.8°	glass	35.3 ± 3.1°
V <sub>E</sub>	39.1 ± 1.9°	marble	61.7 ± 1.8°
A <sub>S</sub>	78.2 ± 3.1°	polystyrene	77.0 ± 1.6°
A <sub>E</sub>	62.5 ± 2.1°		

266

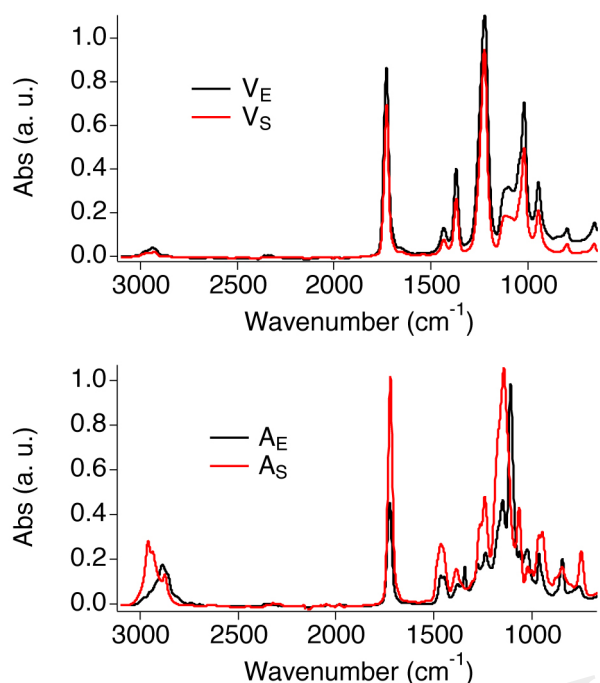
267 **Figure 1** shows the FTIR-ATR spectra of the four polymer films. The spectra of the two vinyl films  
 268 match closely, confirming that the same type of polymers is present in both films. The main bands  
 269 were assigned as follows (Doménech-Carbó et al., 2001; Learner and Institute, 2004): C-H stretching  
 270 (2928 cm<sup>-1</sup> and a shoulder at 2972 cm<sup>-1</sup>); C=O stretching (intense and narrow peak at 1727 cm<sup>-1</sup>);  
 271 asymmetric stretching of the C-O group (1222 cm<sup>-1</sup>, distinctive of PVAc); C-H in-plane bending  
 272 (1428 cm<sup>-1</sup> and 1371 cm<sup>-1</sup>); C-H out-of-plane bending (1117 cm<sup>-1</sup>); C-O symmetric stretching (1018  
 273 cm<sup>-1</sup>); C-C stretching (944 cm<sup>-1</sup>); C-H rocking (792 cm<sup>-1</sup>). Besides, the spectrum of V<sub>E</sub> shows a broad  
 274 band at 3335 cm<sup>-1</sup>, which is likely due to the OH stretching of poly(vinyl alcohol) (PVA), i.e., one of  
 275 the main additives of the Vinavil NPC® aqueous latex.

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277

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279



280

281 **Figure 1.** ATR spectra of the four polymeric films selected.

282

283 The spectra of  $A_S$  and  $A_E$  show slightly different features in the fingerprint and CH stretching  
 284 regions, as expected given the different type of acrylate polymers found in those films.

285 The main bands of the acrylic films were assigned as follows (Doménech-Carbó et al., 2001; Learner  
 286 and Institute, 2004; Pintus and Schreiner, 2011): C-H stretching ( $A_S$ : peaks at 2958, 2933, 2873  $\text{cm}^{-1}$ ;  
 287  $A_E$ : peaks at 2887  $\text{cm}^{-1}$  and 2860  $\text{cm}^{-1}$ ); C=O stretching (1720  $\text{cm}^{-1}$ ); C-H in-plane bending (1465 and  
 288 1380  $\text{cm}^{-1}$ ;  $A_E$  shows an additional peak is visible at 1343  $\text{cm}^{-1}$ ); C-O stretching ( $A_S$ : peaks at 1240  
 289 and 1065  $\text{cm}^{-1}$ ;  $A_E$ : peaks at 1279, 1241, 1109, and 1022  $\text{cm}^{-1}$ ); C-C stretching (965 and 946  $\text{cm}^{-1}$ );  
 290 C-H rocking (840 and 750  $\text{cm}^{-1}$ ).

291

### 292 3.2. NSF Characterization

293 The selected NSF for this work is composed of water, a nonionic surfactant and two solvents, i.e.,  
 294 BuOH and MEK, which are partly miscible with water. The structure of this NSF was never studied  
 295 before, thus acquiring information on the micelles' size and shape, and on the location of each  
 296 component in the NSF, was deemed a preliminary step to help understanding the interaction of the  
 297 fluid with the polymeric films. To this aim, SANS measurements were performed on four  $\text{D}_2\text{O}$  based  
 298 samples: 1) the  $\text{D}_2\text{O}/\text{C}_{9-11}\text{E}_{5.5}$  binary mixture; 2) the  $\text{D}_2\text{O}/\text{C}_{9-11}\text{E}_{5.5}/\text{BuOH}$  ternary system; 3) the  
 299 complete NSF formulation, i.e.,  $\text{D}_2\text{O}/\text{C}_{9-11}\text{E}_{5.5}/\text{BuOH}/\text{MEK}$ ; 4) the complete NSF formulation with  
 300 deuterated d-MEK fully replacing the regular MEK. The analysis of samples 3 and 4 represents a  
 301 contrast variation experiment, in which everything is kept constant from one sample to the other,  
 302 except for its contrast, by changing the SLD of one or more chemicals (in this case, exchanging MEK  
 303 with d-MEK).

304 **Figure 2-top** shows the SANS profiles for the four samples analyzed. The data were fitted according  
 305 to two different models. The supramolecular aggregates, in the case of the binary surfactant/water  
 306 and the water/surfactant/BuOH systems, were modeled as non-interacting polydisperse core-shell  
 307 spheres, defined by two contrasts, *i.e.* bulk/shell and shell/core. On the other hand, the best fitting for  
 308 the complete NSF was obtained by modeling the micelles as non-interacting prolate core-shell  
 309 ellipsoidal particles, again defined by a double contrast. The scattering length density (SLD) of bulk,  
 310 shell and core, *i.e.*, respectively,  $\rho_{\text{bulk}}$ ,  $\rho_{\text{shell}}$  and  $\rho_{\text{core}}$ , were calculated according to the SLD for  
 311 neutrons of each chemical included in the formulations, as reported in **Table 3**. For globular micelles  
 312 of homogeneous scattering length density, the total scattered intensity  $I(q)$  ( $\text{cm}^{-1}$ ) is given by (Liu et  
 313 al., 1995; Sheu and Chen, 1988):

$$314 \quad I(q) = N_p V_p^2 \Delta\rho^2 P(q) S(q) + \text{bkg}_{\text{inc}} \quad (1)$$

315 where  $N_p$  is the number density of the scattering particles ( $\text{cm}^{-3}$ ),  $V_p$  is the volume ( $\text{cm}^3$ ),  $\Delta\rho$  is the  
 316 contrast term ( $\text{cm}^{-2}$ ),  $P(q)$  is the form factor and  $S(q)$  is the structure factor. In this case  $S(q) = 1$ , as  
 317 the particles were considered to be non-interacting. In the case of spherical core-shell aggregates, the  
 318 particle scattering intensity is expressed as follows (Kline, 2006):

$$319 \quad I(q) = \frac{\phi}{V_p} \left[ (\rho_{\text{core}} - \rho_{\text{shell}}) \frac{3V_c j(qr_c)}{qr_c} + (\rho_{\text{shell}} - \rho_{\text{bulk}}) \frac{3V_p j(qr_s)}{qr_s} \right]^2 \quad (2)$$

320 where  $j(x)$  is a spherical Bessel function and is expressed as:

$$321 \quad j(x) = \frac{(\sin x - x \cos x)}{x^2} \quad (3)$$

322 and where  $\phi$  is the volume fraction of the micellar phase,  $V_c$  is the core volume,  $r_c$  is the core radius,  
 323  $r_s = r_c + t$  ( $t$  is the shell thickness). Since this model takes into account a polydisperse core, which  
 324 follows the Schultz distribution, the form factor calculated in equation (2) is normalized by the  
 325 average particle volume:

$$326 \quad \langle V \rangle = \frac{4\pi}{3} \langle r_c^3 \rangle \quad (4)$$

327 where:

$$328 \quad \langle r_c^3 \rangle = \frac{(z+2)(z+3)}{(z+1)^2} \langle r_c \rangle \quad (5)$$

329 and  $z$  is the width parameter of the Schultz distribution (Degiorgio et al., 1985):

$$330 \quad z = \frac{1}{\left(\frac{\sigma}{\langle r_c \rangle}\right)^2} - 1 \quad (6)$$

331 being  $\sigma^2$  the variance of the distribution. The polydispersity index (PDI), reported in **Table 4** is  
 332 defined as  $\sigma/\langle r_c \rangle$  (see equation (6)) and its value is comprised between 0 and 1.

333 In the case of monodisperse non-interacting prolate ellipsoids, on the other hand, when modeling  
 334 asymmetric micelles with a core-shell scattering length profile,  $P(q)$  included in equation (1) is  
 335 usually calculated as an orientationally-averaged normalized form factor,  $\bar{P}(q)$ . First, the orientation-

336 dependent form factor  $F(q, \mu)$ , is defined as follows (where  $\mu$  is the cosine between the direction of  
 337 the symmetry axis of the ellipsoid and the  $Q$  vector):

$$338 \quad F(q, \mu) = f(\rho) \frac{3j(u)}{u} + (1 - f(\rho)) \frac{3j(v)}{v} \quad (7)$$

339 where  $j(x)$  is the same spherical Bessel function defined in eq. (3) and  $u$  and  $v$  are expressed as:

$$340 \quad u = q[\mu^2 a^2 + (1 - \mu^2) b^2]^{1/2} \quad (8)$$

$$341 \quad v = q[\mu^2 (a + t)^2 + (1 - \mu^2) (b + t)^2]^{1/2} \quad (9)$$

342 which define the geometrical shape of the micelles, and where  $f(\rho)$  contains the contrast calculation,  
 343 and  $a$ ,  $b$ ,  $t$  are the geometrical parameters of the ellipsoid, i.e. the major semi-axis, the minor semi-  
 344 axis, and the shell thickness, respectively.

345  $\bar{P}(q)$  is then calculated as follows (Kotlarchyk and Chen, 1983):

$$346 \quad \bar{P}(q) = \int_0^1 d\mu |F(q, \mu)|^2 \quad (10)$$

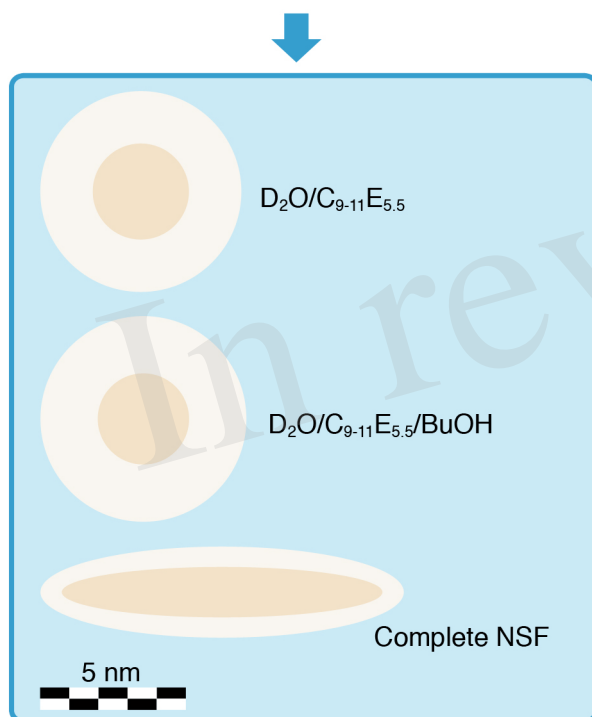
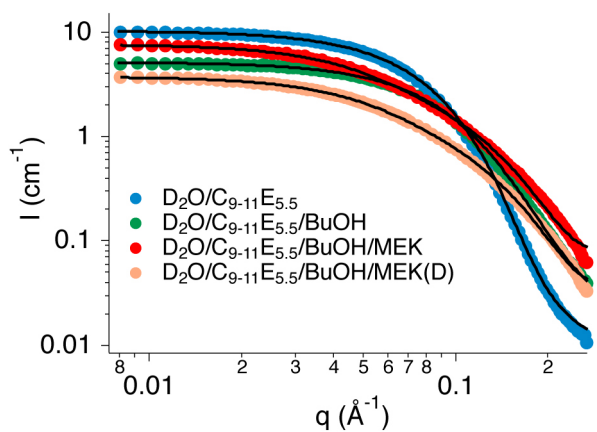
347

348 **Table 3.** SLD for neutrons scattering of the chemicals included in the NSF.

Chemical formula/structure	Compound/molecular group	SLD ( $10^{-6} \text{ \AA}^{-2}$ )
D <sub>2</sub> O	Heavy water	6.39
CH <sub>3</sub> (CH <sub>2</sub> ) <sub>8-10</sub> (OCH <sub>2</sub> CH <sub>2</sub> ) <sub>5.5</sub> OH	C <sub>9-11</sub> E <sub>5.5</sub> <sup>a</sup>	0.37
CH <sub>3</sub> (CH <sub>2</sub> ) <sub>8-10</sub> <sup>a</sup>	C <sub>9-11</sub> E <sub>5.5</sub> apolar tail	-0.41
HO(CH <sub>2</sub> CH <sub>2</sub> O) <sub>5.5</sub>	C <sub>9-11</sub> E <sub>5.5</sub> polar head	0.97
C <sub>4</sub> H <sub>10</sub> O	BuOH	-0.33
C <sub>4</sub> H <sub>3</sub> D <sub>5</sub> O	d-MEK	3.66
C <sub>4</sub> H <sub>8</sub> O	MEK	0.17

349 <sup>a</sup>For the SLD calculation a C<sub>10</sub> aliphatic chain was considered.

350



351

352 **Figure 2.** Top) SANS profiles of the four samples analyzed. The best fitting curves are represented as continuous black  
 353 lines. Bottom) The cartoon illustrates the structural evolution (in scale) of the three formulations investigated, i.e., the  
 354 binary water/surfactant mixture, the water/surfactant/alcohol system, and the complete NSF.

355

356 Even though the measured samples included  $D_2O$  instead of  $H_2O$ , the structural picture emerging  
 357 from the analysis of SANS data can be safely transferred to the  $H_2O$ -based NSF, apart from possible  
 358 slight changes (Baglioni et al., 2012a).

359 **Table 4** shows the main fitting results, while **Figure 2-bottom** contains a cartoon depicting the size  
 360 and shape evolution of micelles following the addition of each component to the formulation. The  
 361 size of  $C_{9-11}E_{5.5}$  micelles in  $D_2O$  was perfectly consistent with the length of the surfactant molecule  
 362 and with the results of previous SAXS measurements performed on similar systems (Baglioni et al.,  
 363 2017), where a  $r + t$  total micelle radius of about  $35 \text{ \AA}$  had been obtained. A polydispersity index of  
 364 0.2 for the size of the hydrophobic core fits such systems, which usually have a PDI in the 0.1 - 0.5



365 range. According to fitting results, the structure and size of the micelles do not change significantly  
 366 after the addition of BuOH (see **Figure 2-bottom**). In fact, BuOH was found to be partitioned  
 367 between the micellar and the aqueous bulk phase in a 30:70 ratio, meaning that most of the solvent is  
 368 mixed with water. The fraction solubilized in the micelles is preferentially located in the shell,  
 369 replacing D<sub>2</sub>O hydration molecules, and thus affects the micellar size and shape only slightly. The  
 370 main effect of the inclusion of BuOH is to double the polydispersity of the core radius with respect to  
 371 the binary water/surfactant system, even though the value is perfectly suitable for a micellar solution.

372 The results of the contrast variation experiments performed on the complete NSF formulation  
 373 showed that the inclusion of a significant amount of MEK in the system completely alters both the  
 374 size and shape of the aggregates (see **Figure 2-bottom**), as the result of a sphere-to-rod transition.  
 375 This behavior was observed for similar systems containing the same surfactant (Baglioni et al., 2014)  
 376 and is likely due to the system getting closer to its cloud point. In fact, micelles are known to grow  
 377 and assume elongated shapes close to the clouding temperature, as observed in the present case. C<sub>9</sub>-  
 378 <sub>11</sub>E<sub>5.5</sub> has a cloud point of about 55-60 °C, which is known to be lowered by the interaction of this  
 379 surfactant with MEK (Baglioni et al., 2014). The inclusion of MEK in the system does not alter  
 380 significantly the partition coefficient of BuOH ( $P_{\text{BuOH}}$ ) between the micelles and the water phase,  
 381 while MEK is mainly dissolved in the bulk phase ( $P_{\text{MEK}} = 0.17$ ).

382 Overall, SANS analysis clarified that the structure of the NSF is that of rod-like micelles, mainly  
 383 composed of the sole surfactant, dispersed in a water/solvents mixture, indicating that the system is  
 384 close to its cloud point. This structural information has relevant implications in terms of cleaning  
 385 power of the NSF, as detergency is known to be maximized when these systems are close to the  
 386 cloud point (Baglioni et al., 2014; Holmberg, 2002; Holmberg et al., 2002; Stubenrauch, 2008).  
 387 Therefore, high cleaning effectiveness was expected from this NSF formulation.

388

389 **Table 4.** Fitting results of the SANS data acquired on the NSF.

Fitting parameter	D <sub>2</sub> O/C <sub>9-11</sub> E <sub>5.5</sub>	D <sub>2</sub> O/C <sub>9-11</sub> E <sub>5.5</sub> /BuOH	Complete NSF
$r$ (Å)	16.7 ± 0.3	15.7 ± 0.2	-
$t$ (Å)	17.9 ± 0.1	19.6 ± 0.1	9.4 ± 0.4
PDI	0.20 ± 0.01	0.40 ± 0.02	-
$a$ (Å)	-	-	53.0 ± 2.0
$b$ (Å)	-	-	7.2 ± 0.3
$P_{\text{BuOH}}$	-	0.30 ± 0.03	0.28 ± 0.02
$P_{\text{MEK}}$	-	-	0.17 ± 0.01

390

391

392 **3.3. NSF/polymer interaction**

393 The interaction between the NSF and the four polymer films was initially investigated simply by  
 394 immersing the coated specimens into four different liquid systems, i.e., water, water/surfactant,  
 395 water/solvents, and the complete NSF. The micromorphology of the film was observed by optical  
 396 microscopy at different times during the total 10 minutes of incubation of the specimens in the  
 397 liquids, while 2D micro-FTIR mapping was performed on the polymer surface before exposure to the  
 398 fluids and after 10 minutes of incubation. Finally, a removal test was performed on the polymers  
 399 using wet cotton swabs, in order to check their removability.

400

401 **Table 5.** Results of immersion tests performed on polymer-coated glass, marble and polystyrene specimens. Both the film  
 402 appearance and its removability after 10 min of incubation in the four different liquid systems are summarized in the  
 403 table, following analysis with optical microscopy and 2D micro-FTIR mapping.

	Glass		Marble		Polystyrene		
	Film appearance	Removability <sup>a</sup>	Film appearance	Removability <sup>a</sup>	Film appearance	Removability <sup>a</sup>	
V <sub>S</sub>	H <sub>2</sub> O	No change	±	No change	x	No change	x
	H <sub>2</sub> O/C <sub>9-11</sub> E <sub>5.5</sub>	Swollen	o	Swollen	±	Swollen	±
	H <sub>2</sub> O/BuOH/MEK	Dewetted	o	Dewetted	o	Dewetted	o
	NSF	Dewetted	o	Dewetted	o	Dewetted	o
A <sub>S</sub>	H <sub>2</sub> O	No change	x	No change	x	No change	x
	H <sub>2</sub> O/C <sub>9-11</sub> E <sub>5.5</sub>	Swollen	o	Swollen	x	Swollen	x
	H <sub>2</sub> O/BuOH/MEK	Dewetted	o	Partly dewetted	±	Swollen	x
	NSF	Dewetted	o	Dewetted	o	Swollen	x
V <sub>E</sub>	H <sub>2</sub> O	Swollen	o	Swollen	o	Partly dewetted	o
	H <sub>2</sub> O/C <sub>9-11</sub> E <sub>5.5</sub>	Swollen	o	Swollen	o	Partly	o

						dewetted	
	H <sub>2</sub> O/BuOH/MEK	Swollen	o	Swollen	o	Swollen	±
	NSF	Swollen	o	Swollen	o	Partly dewetted	o
-----							
	H <sub>2</sub> O	Swollen	o	Swollen	o	Swollen	o
	H <sub>2</sub> O/C <sub>9-11</sub> E <sub>5.5</sub>	Swollen	o	Swollen	o	Swollen	o
<b>A<sub>E</sub></b>	H <sub>2</sub> O/BuOH/MEK	Swollen	o	Swollen	o	Swollen	±
	NSF	Swollen	o	Swollen	o	Swollen	o

404 <sup>a</sup>After 10 min of incubation in the liquid systems, the removability of the coatings was checked performing a gentle  
 405 mechanical action with a cotton swab soaked with water; x = no removal, ± = partial removal, o = complete removal.

406

407 Looking at the results of the immersion tests, illustrated in **Figures 3- 6** and summarized in **Table 5**,  
 408 a clear difference in the behavior of samples emerges, i.e. the polymer films deriving from latexes  
 409 (V<sub>E</sub> and A<sub>E</sub>) are preferentially swollen and only in few cases partly dewetted, whereas the films  
 410 deriving from solutions are unaffected by just water, are swollen by the surfactant solution, and are  
 411 partly or completely dewetted when solvents are included in the liquid systems. For instance, **Figures**  
 412 **3 and 5** show how for chemically identical polymers (both PVAc) the type of the film plays a key  
 413 role in determining the behavior when the film is exposed to the same NSF. 2D microFTIR mapping  
 414 was crucial to confirm the location and distribution of the polymers on the micron-scale. In  
 415 particular, when a film is dewetted (see **Figures 3 and 4**) it was shown that no polymer residues are  
 416 present (above the instrumental detection limit) outside the droplets on the substrate surface.

417 Polymer coatings formed from aqueous emulsions are significantly more sensitive to the action of  
 418 water (a well known issue when emulsion-based acrylic paint layers are exposed to aqueous cleaning  
 419 fluids(Murray et al., 2002; Ormsby and Learner, 2009; Willneff et al., 2014)), and tend to be swollen.  
 420 This is justified by the presence of a non-negligible amount of hydrophilic additives and surfactants  
 421 in the polymer emulsions, which remain in the film after drying and are able to interact with water  
 422 molecules, favoring the swelling of the film. The presence of hydrophilic additives can also explain  
 423 why the swollen films tend to dewet less easily. The energetic balance of dewetting can be described  
 424 by the spreading coefficient S, which for a polymer film on a glass surface, immersed in a liquid, is  
 425 defined as (Baglioni and Chelazzi, 2013):

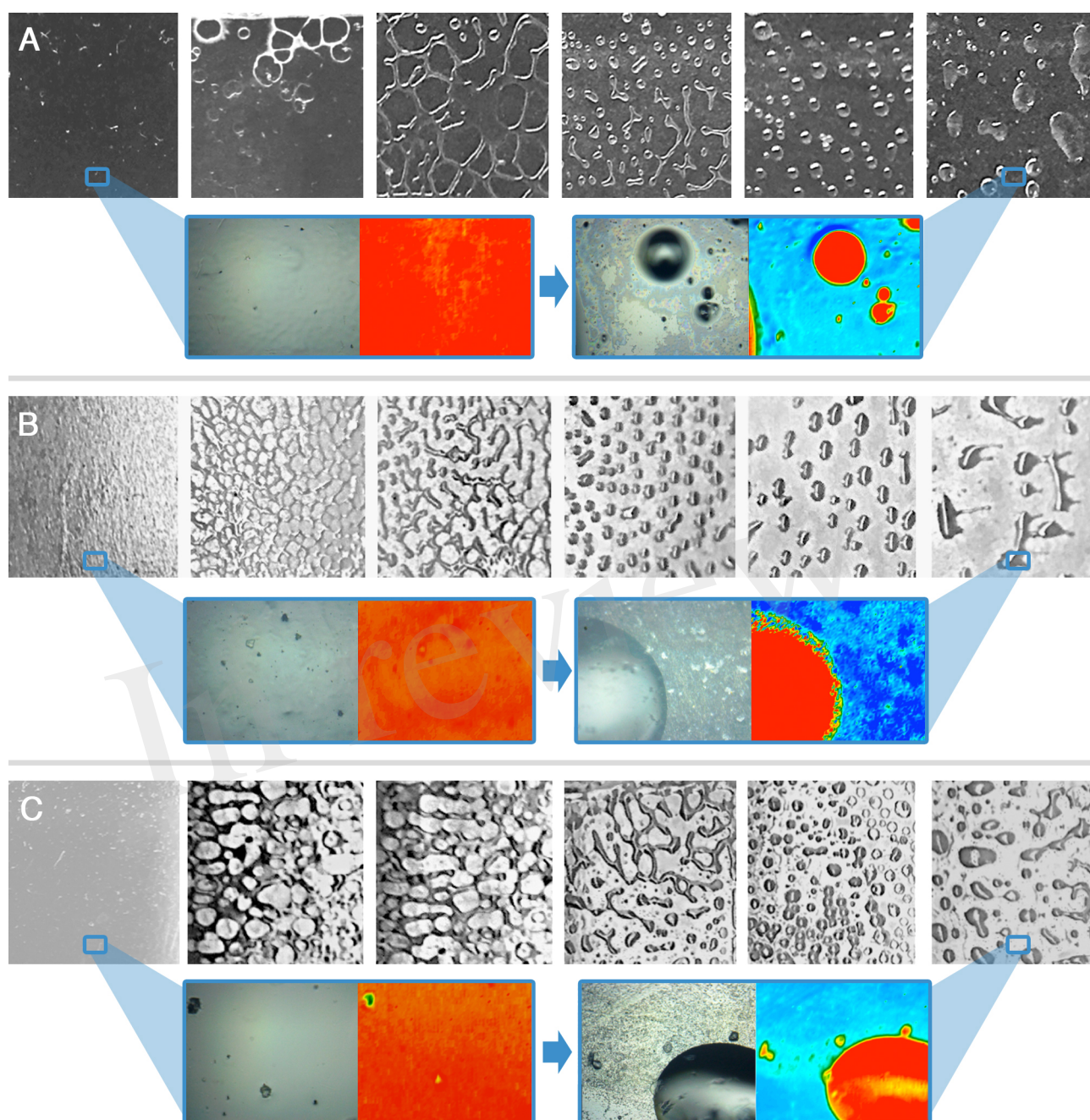
$$426 \quad S = \gamma_{LG} - \gamma_{PG} - \gamma_{LP} \quad (10)$$

427 where  $\gamma_{LG}$  is the interfacial tension between glass and the liquid,  $\gamma_{PG}$  is the interfacial tension between  
 428 glass and the polymer, and  $\gamma_{LP}$  is the interfacial tension between the liquid and the polymer. When S  
 429 is negative, dewetting is energetically favored and occurs spontaneously unless an activation energy  
 430 barrier hinders the process kinetically. Hydrophilic additives in the polymer film might lower the  
 431 values of both  $\gamma_{PG}$  and  $\gamma_{LP}$ , making S less negative for a given liquid/polymer/glass set. On the other

432 hand, the presence of surfactant additives may lower the glass transition temperature ( $T_g$ ) of the  
433 polymer films (as in the case of  $V_E$  – see Table 1), making polymer chains more mobile, thus  
434 possibly decreasing the energy costs related to the formation of new interfacial regions during the  
435 detachment of the film, and lowering the activation energy necessary to initiate dewetting (Baglioni  
436 et al., 2017, 2018b; Montis et al., 2019).

437 Overall, in the case of the polymer films from emulsions investigated here, the films' thermodynamic  
438 stability seems to prevail on the kinetic drive of the dewetting process. However, the swollen films  
439 are softened and easily removable from glass and marble even with water and water/  $C_{9-11}E_{5.5}$ ,  
440 meaning that the adhesion to the solid surface was sensibly reduced, while in the case of polystyrene  
441 the removal is slightly more difficult. When a polymer film formed from a solvent solution is  
442 exposed to a cleaning fluid, a more diversified behavior is observed. Water and water/  $C_{9-11}E_{5.5}$  are  
443 partly or completely ineffective in removing the films. Instead, dewetting processes are induced by  
444 water/solvents mixtures or, more efficiently, by the synergistic action of solvents and surfactants  
445 (Baglioni et al., 2017, 207, 2018b; Gentili et al., 2012; Xu et al., 2012). This is clearly exemplified by  
446  $A_S$  on marble, where water alone produces no visible effects on the polymer. The nonionic surfactant  
447 micellar solution is able to induce some swelling of the film, but not sufficient to grant its easy  
448 removal. A water/solvents mixture, on the other hand, produces a partial dewetting pattern, and the  
449 film can be partly removed using some mechanical action. This means that solvents are able to swell  
450 the polymer to such an extent that the  $T_g$  is lowered below room temperature, starting the dewetting  
451 process. However, it is only with the synergistic action of a surfactant, which lowers the interfacial  
452 tension at the liquid/substrate and liquid/polymer interfaces, that dewetting proceeds further, leading  
453 to easy and complete polymer removal. The substrate's chemical nature is a key factor. Increasing the  
454 hydrophobicity of the substrate, the affinity of the polymer film with the substrate increases, leading  
455 to less efficient polymer removal, culminating with the limit case of  $A_S$  on polystyrene, which  
456 resulted completely irremovable.

457



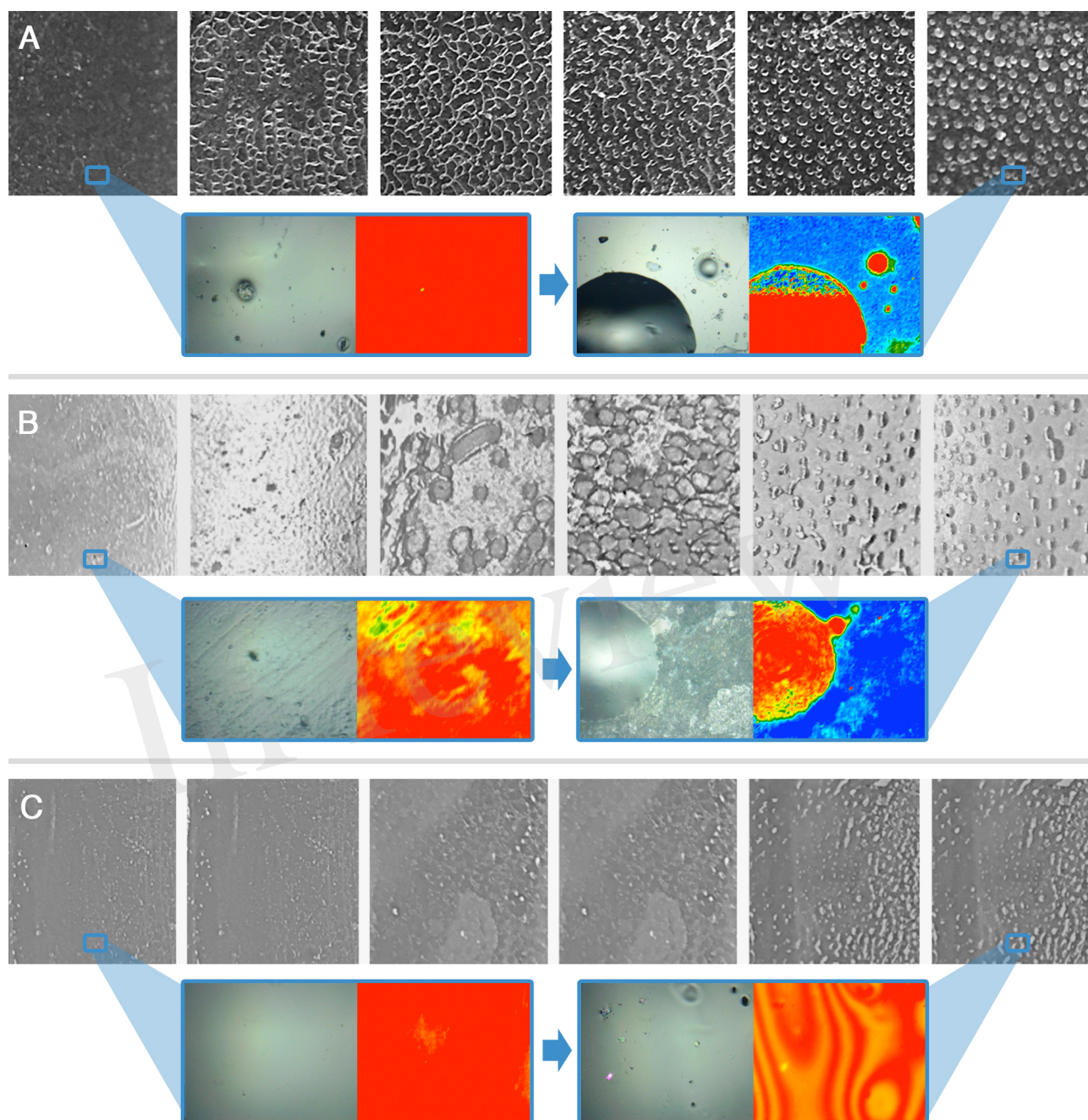
458

459 **Figure 3.** Immersion tests results for the  $V_S$ -coated glass (A), marble (B) and polystyrene (C) specimens. The frames, left  
 460 to right, are taken at  $t = 0$  s, 10 s, 30 s, 1 min, 5 min, 10 min. For  $t = 0$  s and  $t = 10$  min a representative 2D micro-  
 461 reflectance FTIR map is reported, together with its relative micrograph in visible light. In each map, the intensity of the  
 462 C=O ester stretching at about  $1730\text{ cm}^{-1}$  is shown with a chromatic scale, following the order red > yellow > green > blue.  
 463 Visible light micrographs are approximately  $1\text{ cm}^2$  large; the FTIR mapping areas have a size of  $400\text{ }\mu\text{m} \times 550\text{ }\mu\text{m}$ .

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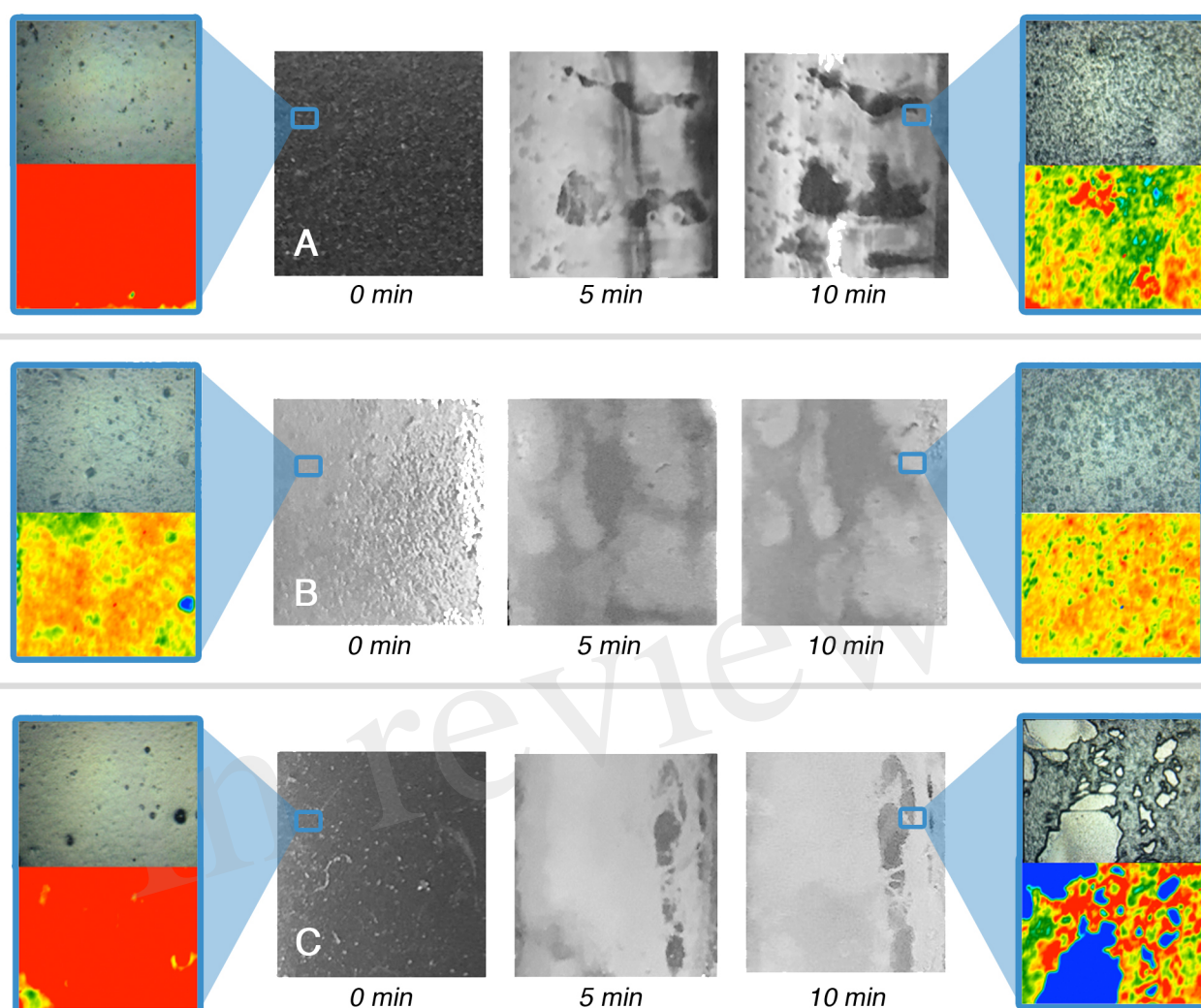
467

468 **Figure 4.** Immersion tests results for the  $A_S$ -coated glass (A), marble (B) and polystyrene (C) specimens. The frames, left  
 469 to right, are taken at  $t = 0$  s, 10 s, 30 s, 1 min, 5 min, 10 min. For  $t = 0$  s and  $t = 10$  min a representative 2D micro-  
 470 reflectance FTIR map is reported, together with its relative micrograph in visible light. In each map, the intensity of the  
 471 C=O ester stretching at about  $1730\text{ cm}^{-1}$  is shown with a chromatic scale, following the order red > yellow > green > blue.  
 472 Visible light micrographs are approximately  $1\text{ cm}^2$  large; the FTIR mapping areas have a size of  $400\text{ }\mu\text{m} \times 550\text{ }\mu\text{m}$ .

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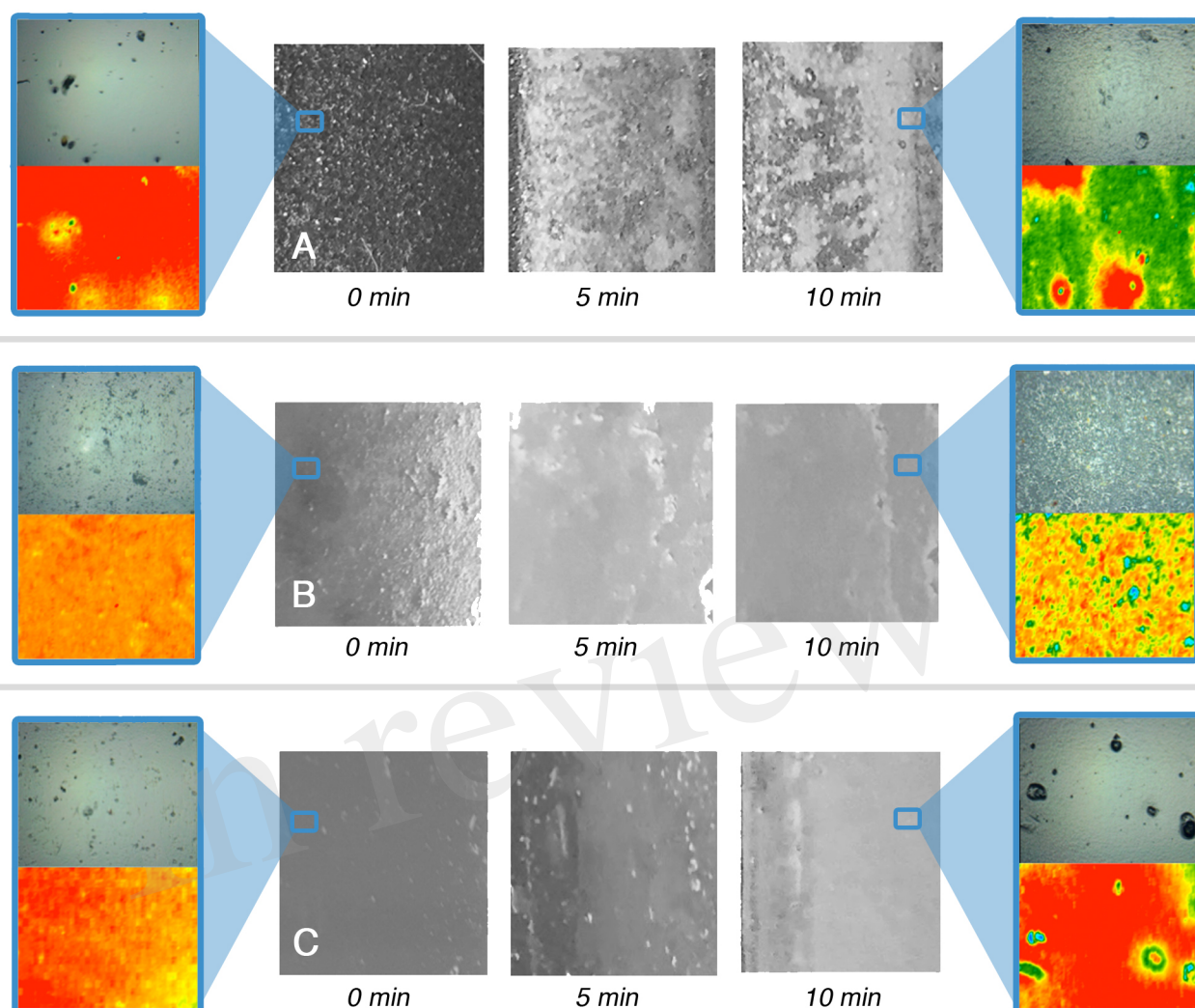
476

477 **Figure 5.** Immersion tests results for the  $V_E$ -coated glass (A), marble (B) and polystyrene (C) specimens. For  $t = 0$  s and  $t$   
478  $=10$  min a representative 2D micro-reflectance FTIR map is reported, together with its relative micrograph in visible  
479 light. In each map, the intensity of the C=O ester stretching at about  $1730\text{ cm}^{-1}$  is shown with a chromatic scale, following  
480 the order red > yellow > green > blue. Visible light micrographs are approximately  $1\text{ cm}^2$  large; the FTIR mapping areas  
481 have a size of  $400\text{ }\mu\text{m} \times 550\text{ }\mu\text{m}$ .

482

483

484



485

486 **Figure 6.** Immersion tests results for the  $A_E$ -coated glass (A), marble (B) and polystyrene (C) specimens. For  $t = 0$  s and  $t = 10$  min a representative 2D micro-reflectance FTIR map is reported, together with its relative micrograph in visible  
 487 light. In each map, the intensity of the C=O ester stretching at about  $1730\text{ cm}^{-1}$  is shown with a chromatic scale, following  
 488 the order red > yellow > green > blue. Visible light micrographs are approximately  $1\text{ cm}^2$  large; the FTIR mapping areas  
 489 have a size of  $400\text{ }\mu\text{m} \times 550\text{ }\mu\text{m}$ .  
 490

491

492 The interaction between the complete NSF and the four polymer films was also monitored through  
 493 CLSM imaging, as described in Section 2.7.2. Homogeneous and reproducible 2-4  $\mu\text{m}$  thick films on  
 494 coverglasses were obtained by spin coating. **Figure 7** summarizes the results of the investigation,  
 495 which were in perfect agreement with what was observed during the immersion tests reported above.  
 496  $A_S$  and  $V_S$ , i.e. the polymer films cast from solutions, were completely and quickly dewetted by the  
 497 NSF, while  $A_E$  and  $V_E$ , i.e. the films cast from polymer latexes, were just swollen. **Figure 7** shows  
 498 that some small cracks and/or holes (of few microns) are visible in the  $V_E$  and  $A_E$  swollen films,  
 499 which nonetheless maintain their overall coherence on a macro-scale. These holes possibly form in  
 500 correspondence of previous film defects, which can be present on such thin films. Overall, the  
 501 behavior of polymer films cast from aqueous latexes can be seen as the physical process stopped at  
 502 the very early stages of dewetting. This is in agreement with what observed during previous

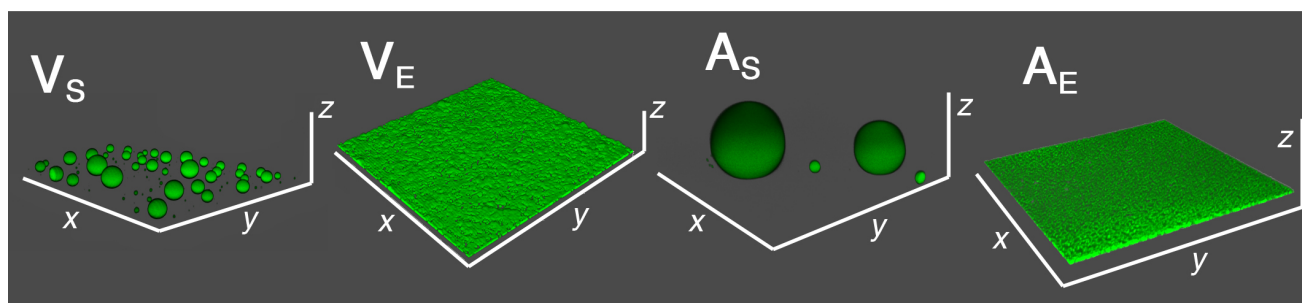
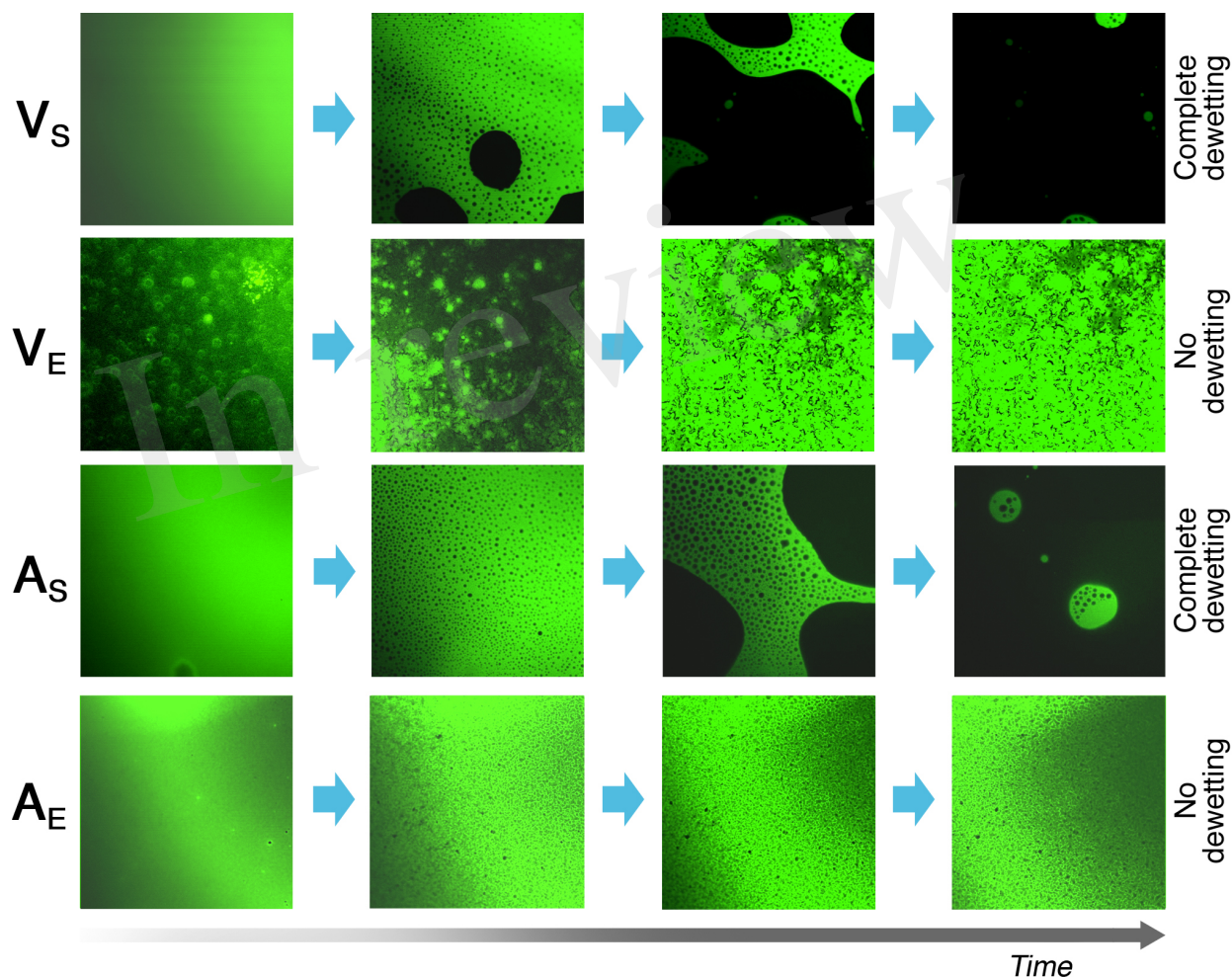


503 experiments, and seems to enforce the hypothesis that the amphiphilic additives in these films play a  
 504 key role in inhibiting the dewetting process, which otherwise would likely occur.

505 The inhomogeneous aspect of the  $V_E$  and  $A_E$  films at  $t = 0$  is due to the fact that the hydrophobic  
 506 fluorescent dye was not evenly distributed in the aqueous polymer latex. As the films is swollen by  
 507 the penetration of the organic solvents, the dye evenly spread through the film.

508 CLSM investigations, overall, allowed to confirm the results of the immersion tests on macroscopic  
 509 samples, where the polymer thickness could not be directly measured and not accurately controllable.

510



511

512 **Figure 7.** Results of CLSM experiments on 2-4  $\mu\text{m}$  thick spin-coated polymer films on coverglasses. The films were  
513 exposed to the action of 50  $\mu\text{l}$  of the complete NSF, and their morphology was observed over time. The time scale is not  
514 the same for all samples, i.e.,  $V_S$  and  $A_S$  completely dewetted from the coverglass in less than 10 s, while  $V_E$  and  $A_E$  films  
515 were still intact after 10 minutes of incubation with the NSF. In the bottom grey box, the 3D reconstruction of the four  
516 polymers' morphology after 10 minutes is reported. The different behavior between polymer films coming from solvent  
517 solutions or aqueous emulsions is clearly evidenced. Each side of the CLSM micrographs is 150  $\mu\text{m}$  long.  $x$  and  $y$  axes in  
518 the 3D reconstructions have the same length.

519

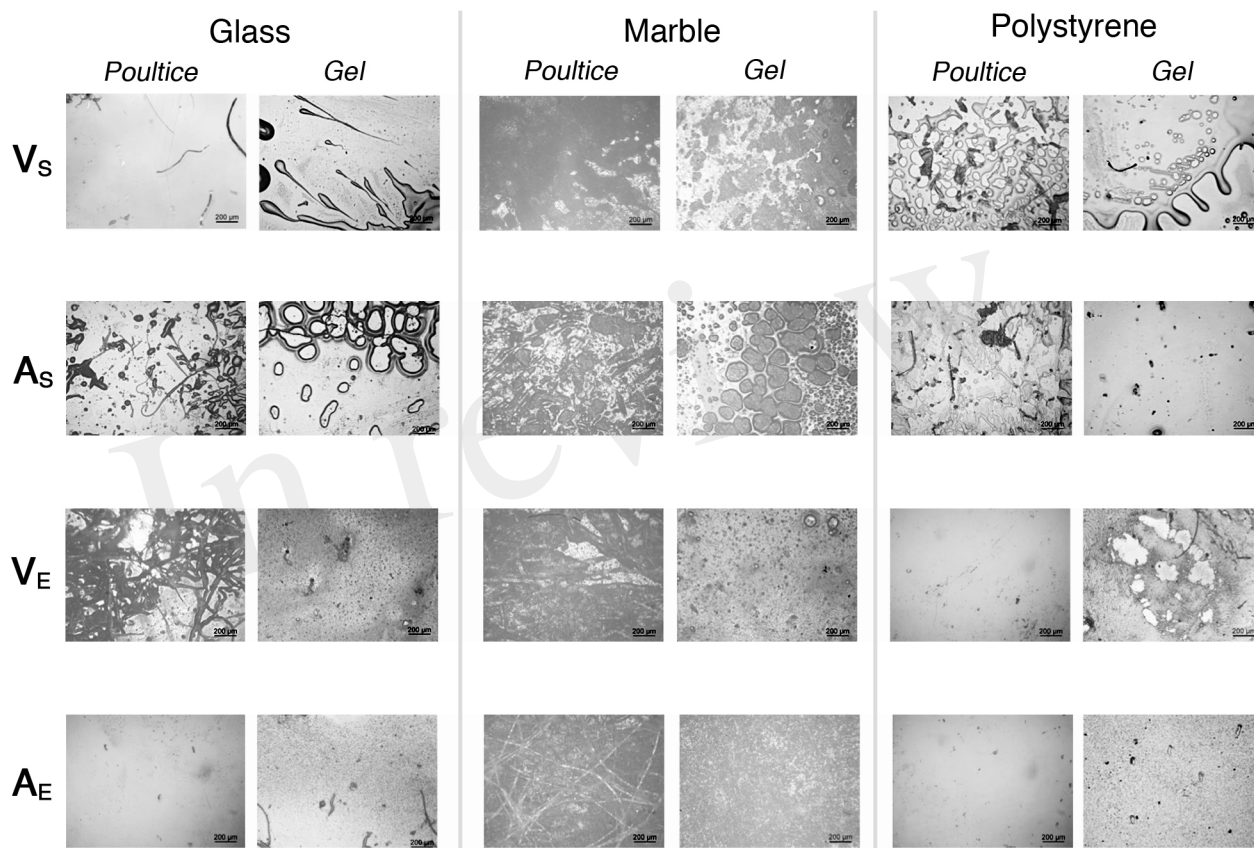
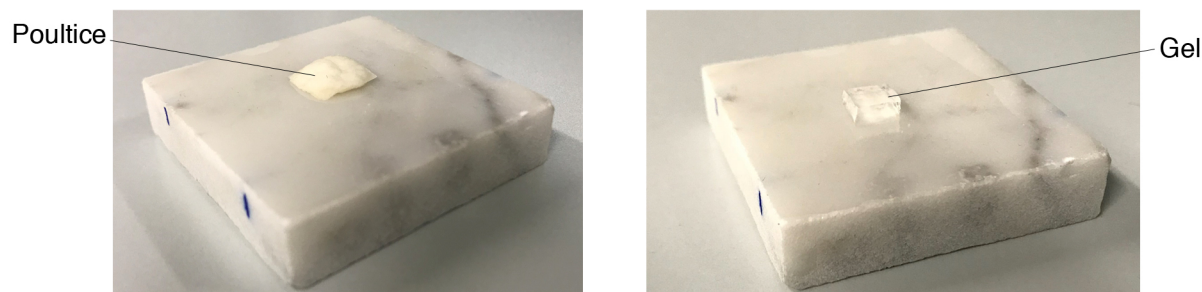
520 Finally, the influence of the application methodology on the outcome of removal tests was evaluated.  
521 To this aim, the same amount of NSF was uploaded respectively in traditional cellulose pulp  
522 poultices and in pHEMA/PVP chemical hydrogels. The latter have been characterized and assessed  
523 in the last years, and their use for the removal of unwanted materials from water-sensitive substrates  
524 has been thoroughly reported (Baglioni et al., 2018a; Bonelli et al., 2018; Domingues et al., 2013a).  
525 Recently, SAXS and rheology studies showed that these gels act as “sponges”, able to load different  
526 NSFs without being altered or dramatically alter the properties of the fluids (Baglioni et al., 2018a).

527 The two systems were applied for 15 minutes on the surface of the films, and then micrographs of the  
528 treated area were taken before checking the coating removability via gentle mechanical action with a  
529 wet cotton swab. As visible in **Figure 8**, the areas treated with the NSF-loaded poultice are generally  
530 more inhomogeneous than the ones treated with the NSF-loaded hydrogel. Moreover, several  
531 cellulose fibers were spotted on the samples that were in contact with the poultice, indicating the  
532 permanence of paper or cellulose pulp residues on the treated areas. Dewetting patterns could be  
533 clearly highlighted in the areas of  $A_S$  on glass and marble treated with the NSF-loaded hydrogel,  
534 indicating a more controlled and reliable cleaning action. In any case, after the removal of either the  
535 compress or the gel, complete and easy removal could be obtained via a gentle mechanical action  
536 using wet cotton swabs, for all the specimens except  $A_S$  on polystyrene. In this case, the acrylic  
537 polymer could not be removed after 15 minutes of application of either the NSF-loaded poultice or  
538 hydrogel (**Figures 9A and 9B**). We hypothesized that in this case the application time was too long,  
539 causing the migration of solvents through the AS coating up to the polymer/substrate interface, where  
540 they interacted with polystyrene creating a sort of joint or adhesion layer between the polymer and  
541 the substrate. This issue is easily overcome by using multiple applications of shorter length, assuming  
542 the confining matrix of the NSF is enough retentive to release gradually the uploaded fluid on short  
543 time scales, which is the case of the pHEMA/PVP gels. In fact, repeated shorter applications (2 +2 +2  
544 minutes) of the NSF loaded in the hydrogel allowed complete removal of the coating (as confirmed  
545 by the 2D microFTIR mapping), while uneven polymer residues were left from the same application  
546 using cellulose poultices (see **Figures 9C-F**) The cartoon in **Figure 9** summarizes the removal  
547 process in the two cases: using the highly retentive hydrogel it is possible to perform a gradual  
548 action, which proceeds layer by layer, controlling the interaction of the NSF with the substrate. This  
549 approach was recently used to perform highly selective removal of overpaintings and vandalism  
550 (Giorgi et al., 2017).

551

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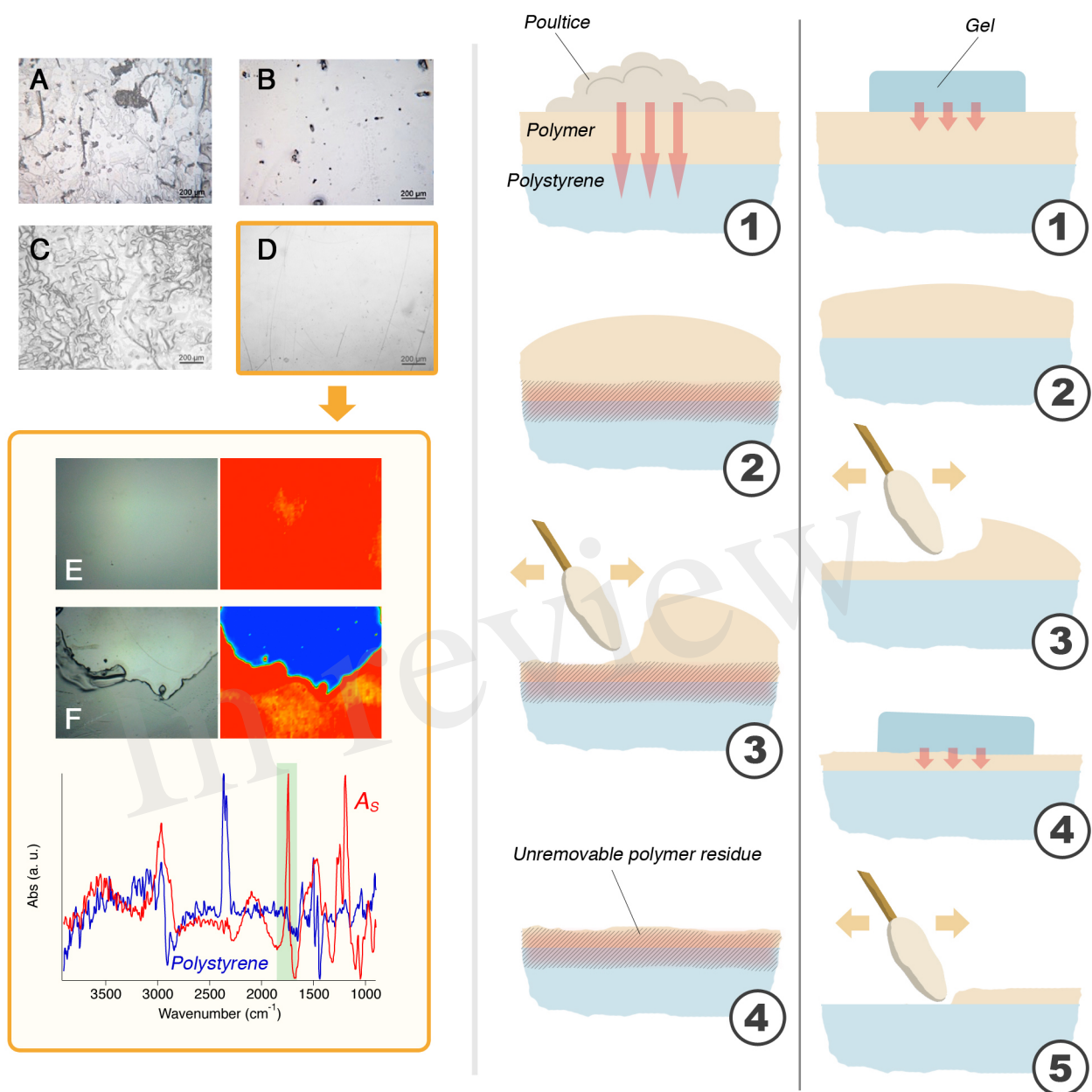
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554

555 **Figure 8.** Results of the cleaning tests performed comparing the effect of the same amount of NSF loaded respectively on  
 556 a traditional cellulose pulp poultice and on an innovative highly retentive chemical hydrogel. Top) Marble specimens  
 557 during the tests. Bottom) Selected representative micrographs taken after 15 minutes of application of both the poultice  
 558 and the gel. Each micrograph is  $1.2 \times 1.6 \text{ mm}^2$  large.

559



560

561 **Figure 9.** A) The result of 15 minutes of application of the NSF-loaded poultrice on  $A_S$  laid on polystyrene; B) The result  
 562 of 15 minutes of application of the NSF-loaded gel on  $A_S$  laid on polystyrene; C) The result of 2 min + 2 min + 2 min of  
 563 application of the NSF-loaded poultrice on  $A_S$  laid on polystyrene; D) The result of 2 min + 2 min + 2 min of application  
 564 of the NSF-loaded gel on  $A_S$  laid on polystyrene. This is the only case that resulted in the complete removal of the  
 565 polymer coating, as confirmed by the comparison between the micro-reflectance FTIR maps collected before (E) and  
 566 after (F) the cleaning with the NSF-loaded hydrogel (the micro-picture is taken at the border between of the cleaned area,  
 567 seen in blue). The cartoon on the right illustrates the mechanism proposed to explain the observed results. With the  
 568 poultrice (left), the solvents migrate through the polymer (1) and attack the polystyrene substrate, creating a junction or an  
 569 enhanced adherence between the polymer film and the substrate (2). When a mechanical action is performed (3), an  
 570 uneven layer of unremoved polymer layer is left stuck to the surface of polystyrene. Using the gel (right), the solvents  
 571 migration is limited by the retention properties of the scaffold (1). The outer layers of the polymer coating are, then,  
 572 slightly swollen (2) and easily removed by means of a gentle mechanical action (3). The gel is applied again (4) and the  
 573 removal proceeds layer by layer, in a selective way, which allows for the complete and safe removal of the polymer film  
 574 (5). Micrographs A, B, C and D are  $1.2 \times 1.6 \text{ mm}^2$  large; micrographs E and F (and relative FTIR maps) have a size of  
 575  $400 \mu\text{m} \times 550 \mu\text{m}$ .

576

577 **4. CONCLUSIONS**

578 This study focused on unveiling some key aspects of the NSF/polymer coatings interaction  
579 depending on several different factors: i) the chemical nature and, thus, hydrophilicity of the  
580 substrate; ii) the chemical nature and physical structure of the polymeric films to be removed; iii) the  
581 influence of the application methodology on the cleaning outcome. A series of systematic tests was  
582 performed and a coherent and clear picture emerged. A water/C<sub>9-11</sub>E5.5/BuOH/MEK NSF was  
583 selected for this study and was firstly characterized by means of SANS measurements, which showed  
584 that rod-like nonionic micelles are dispersed in a water/BuOH/MEK mixture close to the system  
585 cloud point. These features possibly make this NSF particularly effective in polymer removal. Then,  
586 glass, marble, and polystyrene specimens were coated with four different polymers, including two  
587 vinyls and two acrylics, each applied either as a solvent solution or as an aqueous emulsion. It was  
588 found that the NSF/polymer film interaction is greatly dependent on the film structure and  
589 composition. Films formed from solvent solutions can be swollen by water/organic solvents mixtures  
590 or dewetted when a surfactant is added to the cleaning fluid; films formed from polymer latexes, on  
591 the other hand, are generally swollen even just by water, but they tend not to dewet. This happens  
592 independently from the chemical nature of the polymer, and is a direct consequence of its structure  
593 and composition, which includes a significant amount of amphiphilic additives. These substances  
594 alter the energetic balance of the liquid/polymer/solid system and stabilize the film, which does not  
595 dewet. However, these films are easily removable from the substrates, meaning that the action of the  
596 cleaning fluid induces loss of adhesion, similarly to what happens during the first stages of the  
597 dewetting process that occurs for films cast from polymer solutions. The substrate also plays an  
598 important role in the removal of polymer films formed from solutions. In this case, the more the film  
599 is affine to the substrate, the harder its removal. In the limit case, the removal of an acrylic polymer  
600 from polystyrene could be achieved only through selective cleaning action using a NSF-loaded  
601 highly retentive chemical hydrogel, which grants significantly more controlled performances than  
602 traditional cellulose pulp poultices. These results have twofold relevance: they deepen the knowledge  
603 of the physico-chemical processes that underpin phenomena of daily conservation practice, and  
604 provide conservators with innovative solutions to face new challenges in art preservation.

605

606 **Conflict of Interest**

607 The authors declare that the research was conducted in the absence of any commercial or financial  
608 relationships that could be construed as a potential conflict of interest.

609

610 **Author Contributions**

611 M.B, M.A., and D.C. contributed to the conception and design of the work, to the acquisition, the  
612 analysis, and the interpretation of experimental data.

613 R.G. and P.B. revised the work critically giving a substantial intellectual contribution, and ultimately  
614 provided the final approval for its publication.

615

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621

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625

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Figure 1.JPEG

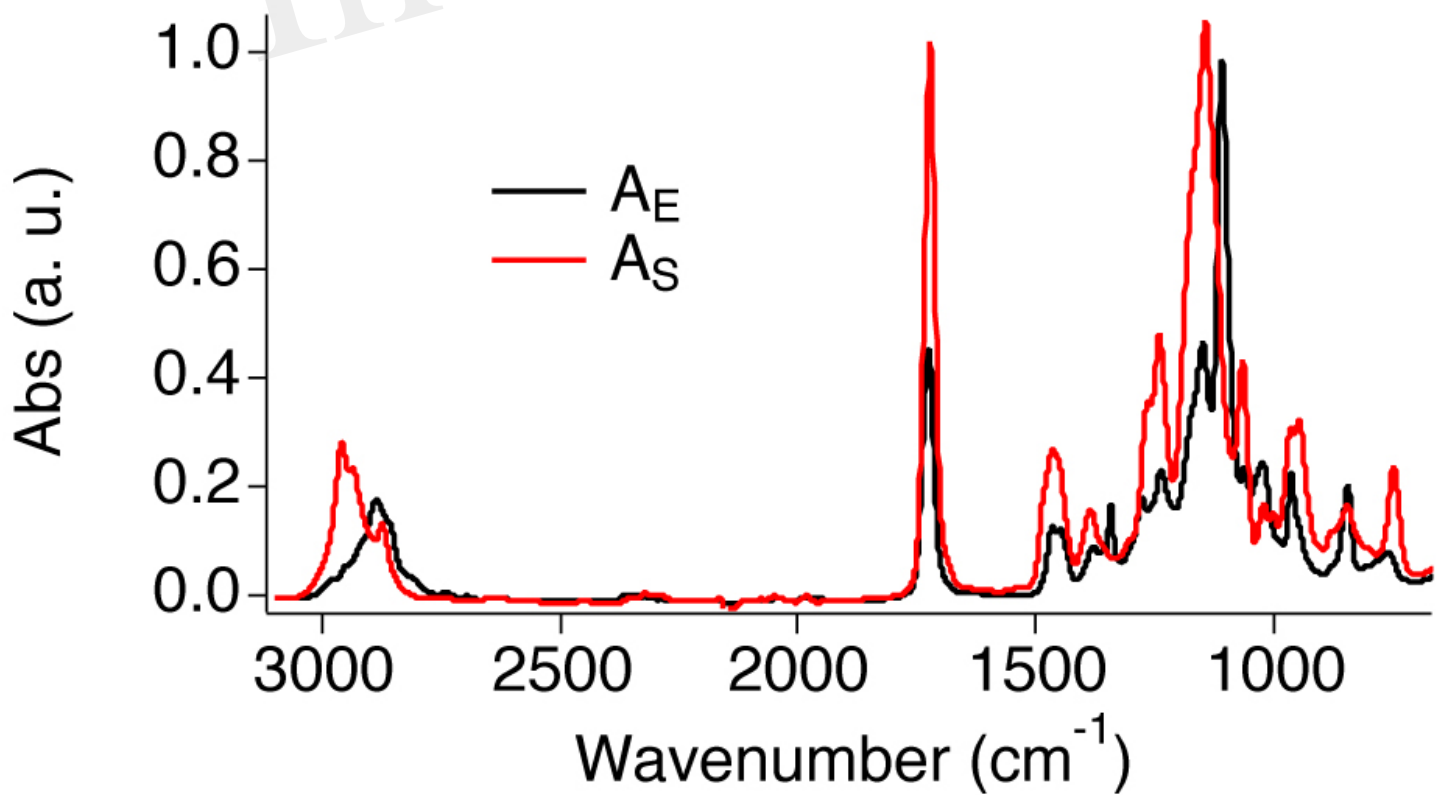
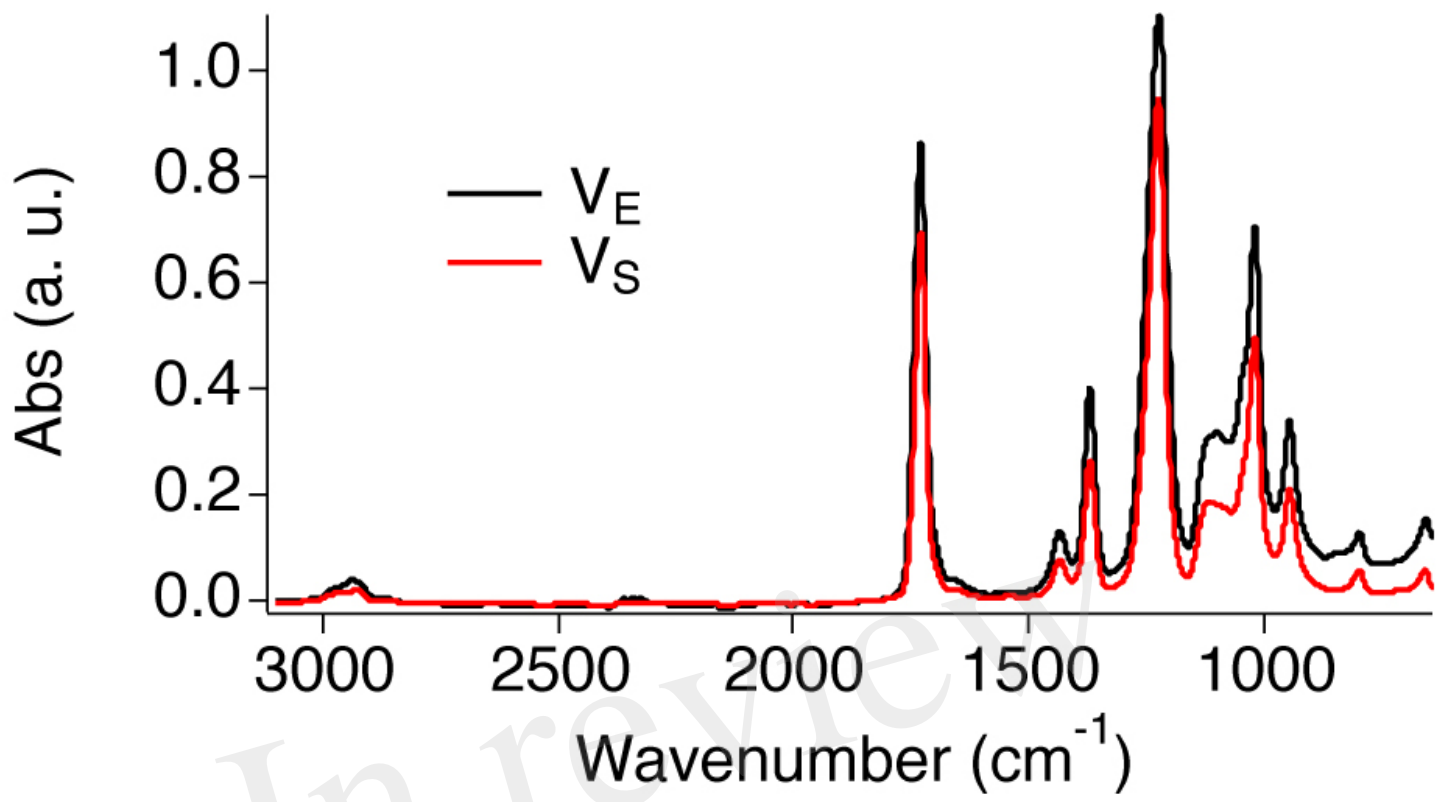


Figure 2.JPEG

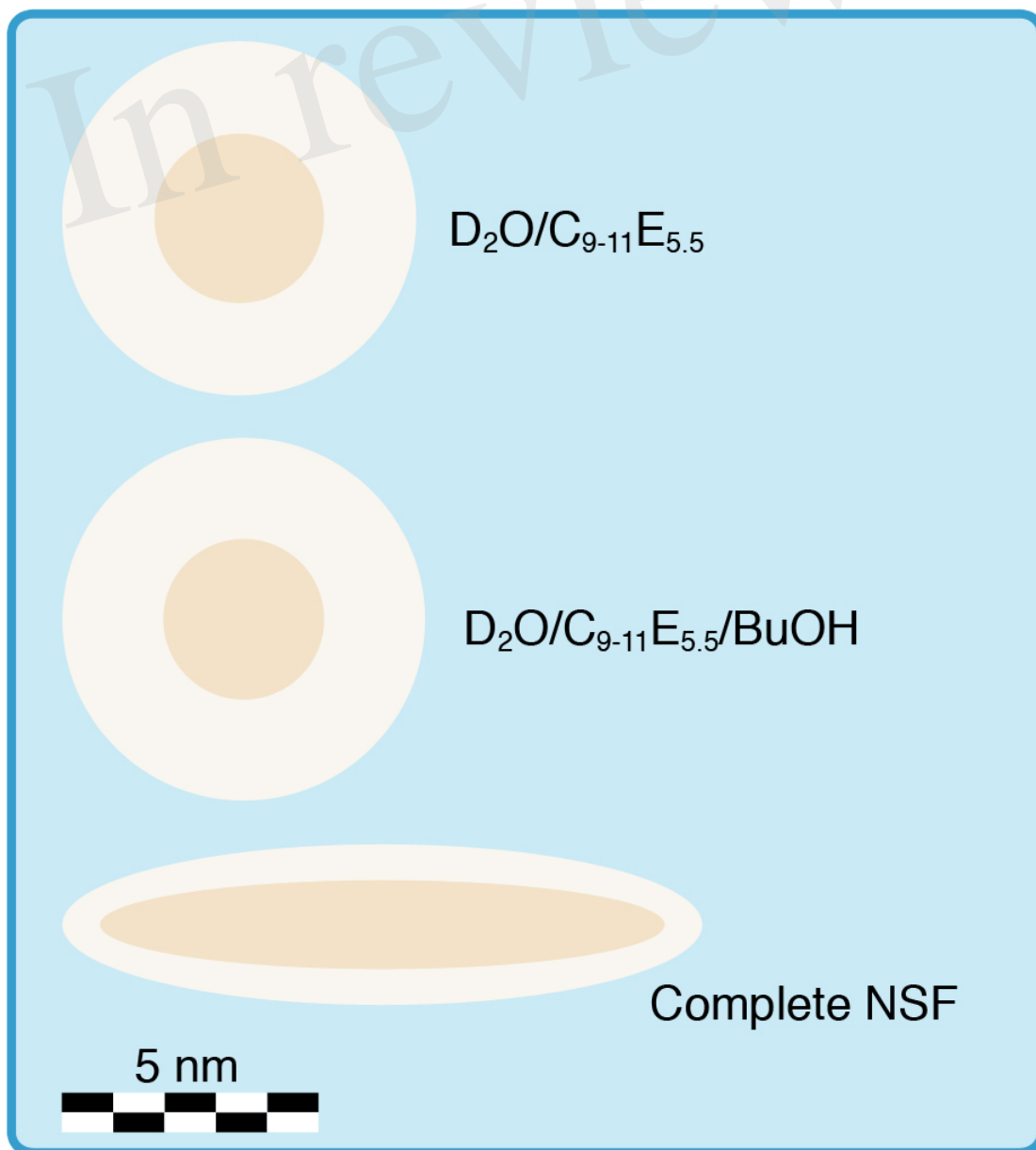
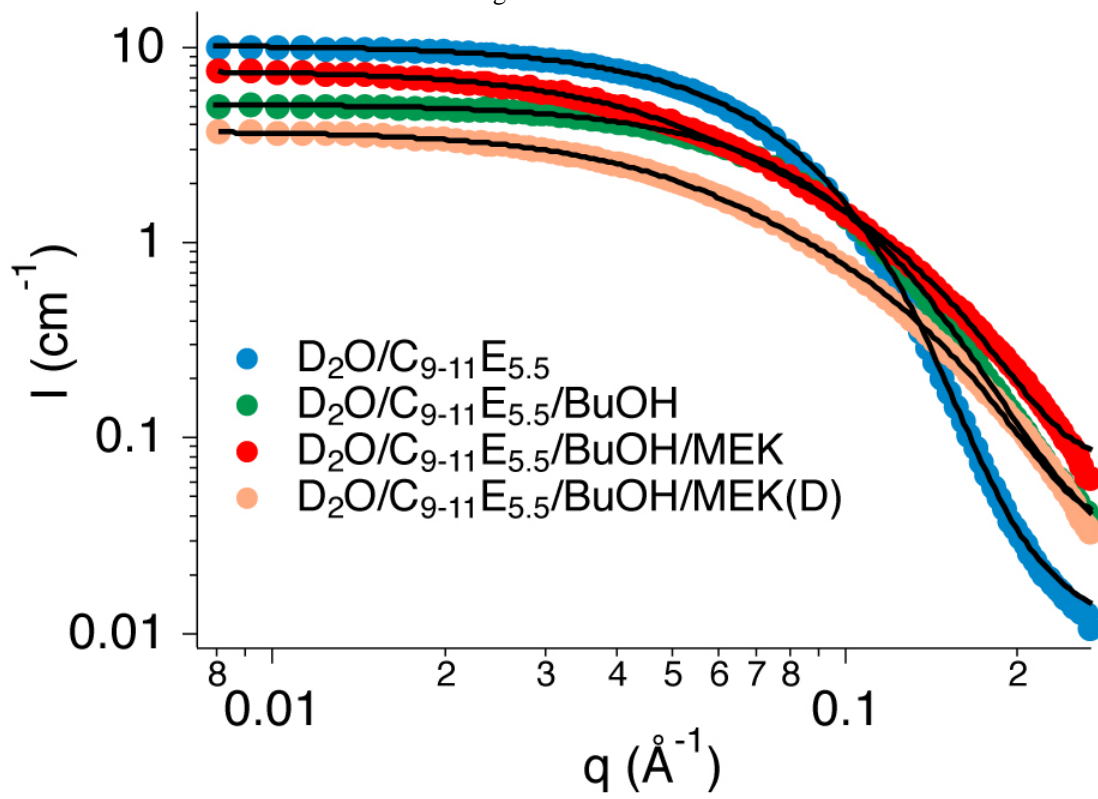


Figure 3.JPEG

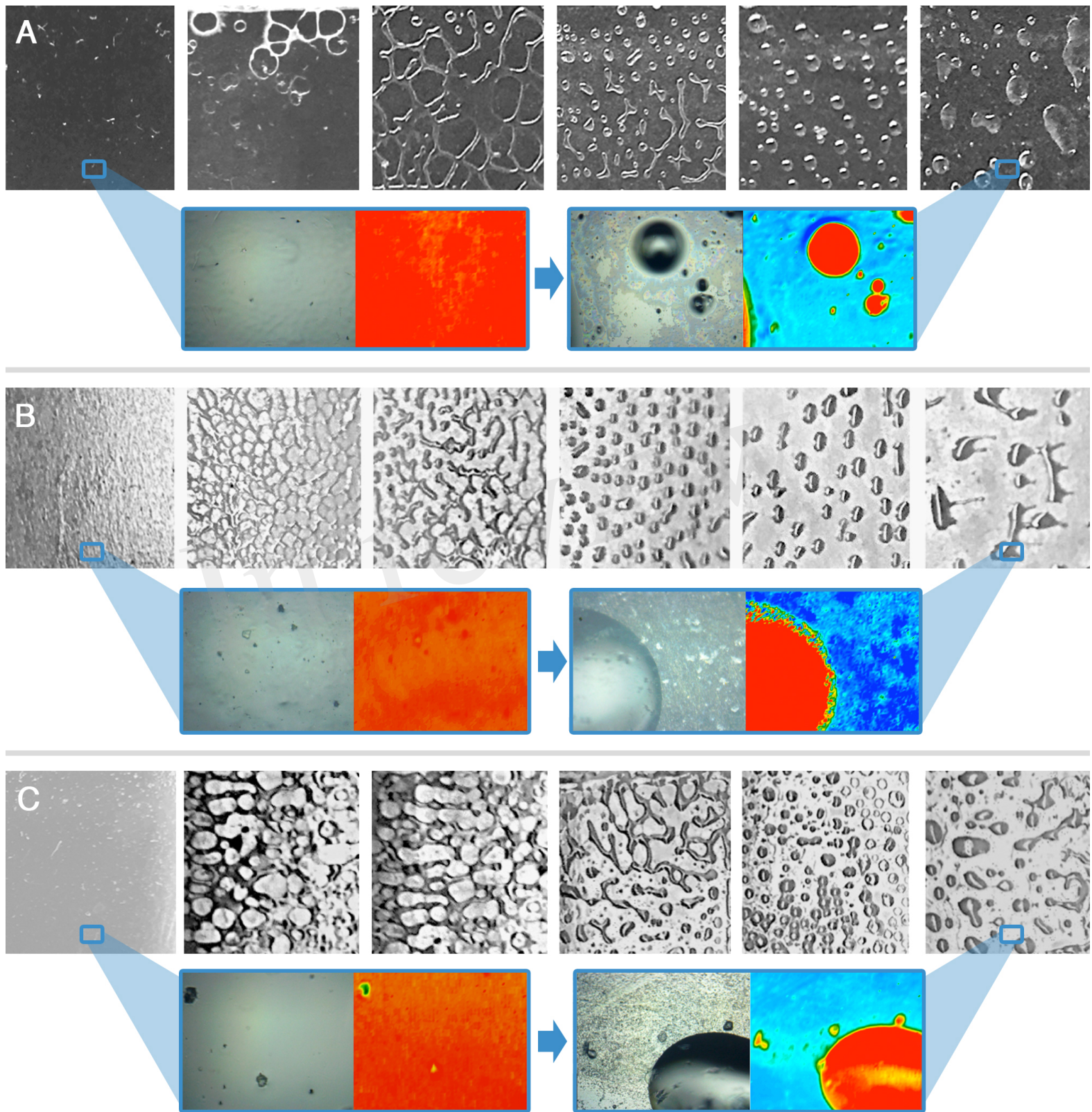


Figure 4.JPEG

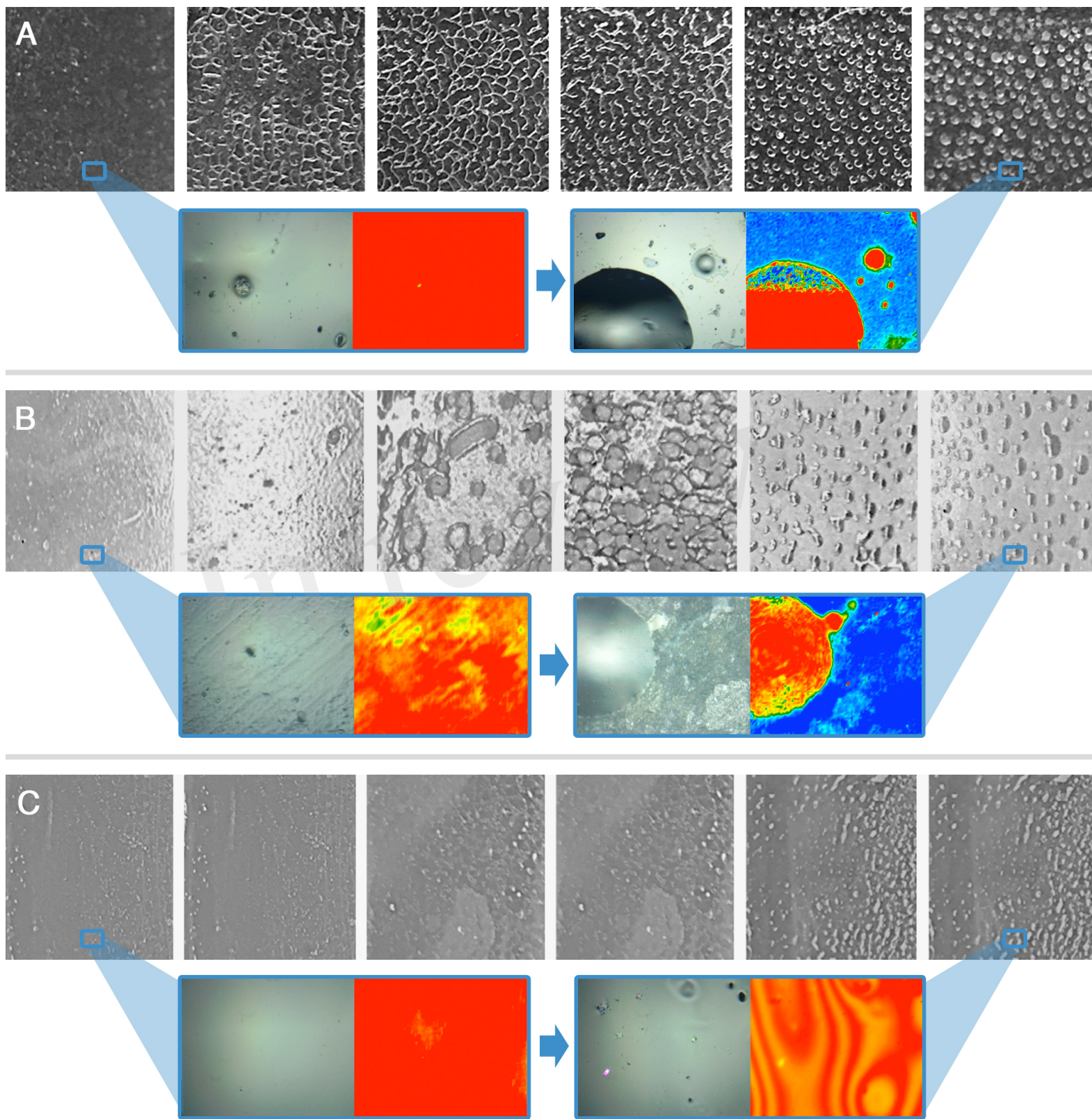


Figure 5.JPEG

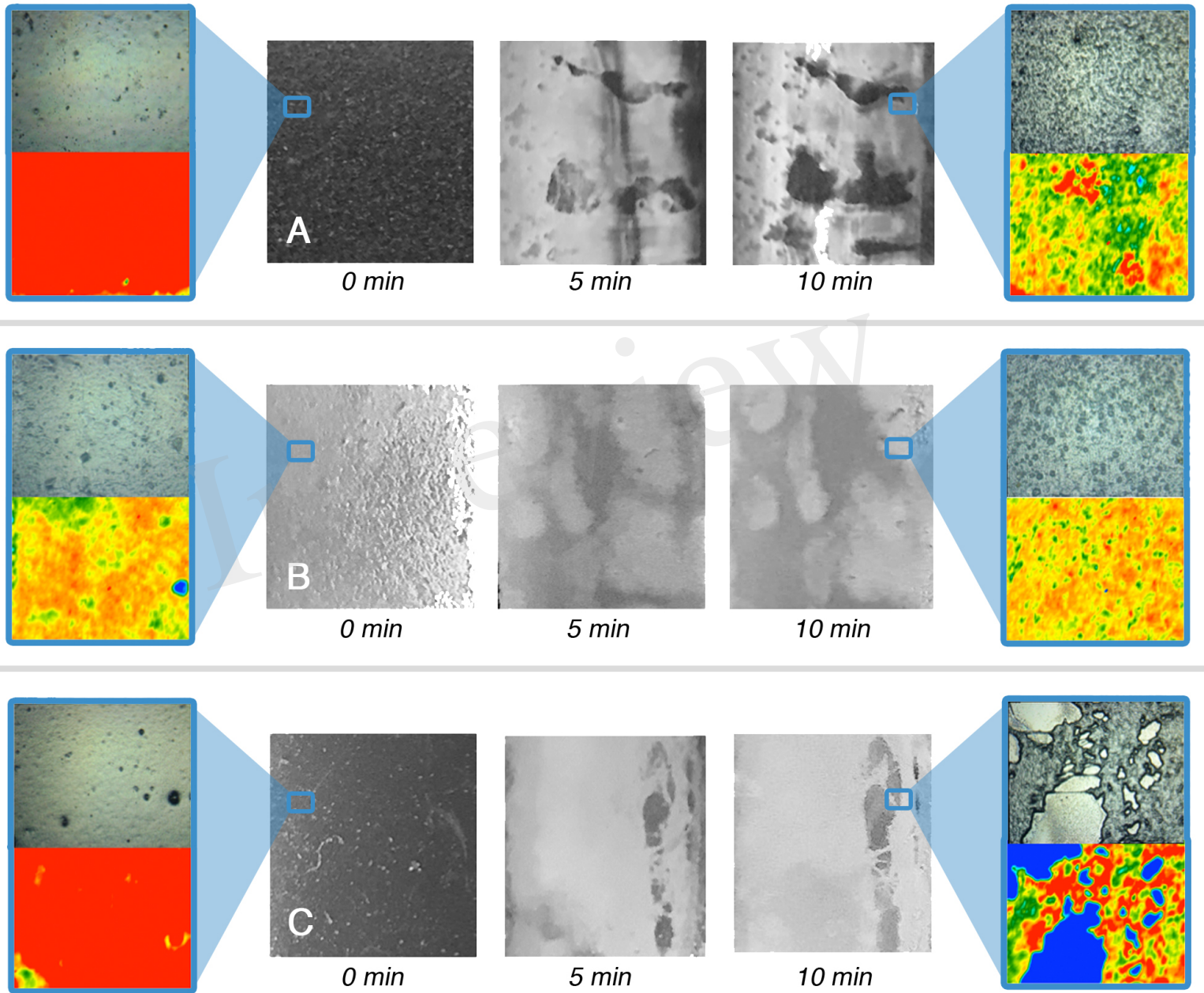


Figure 6.JPEG

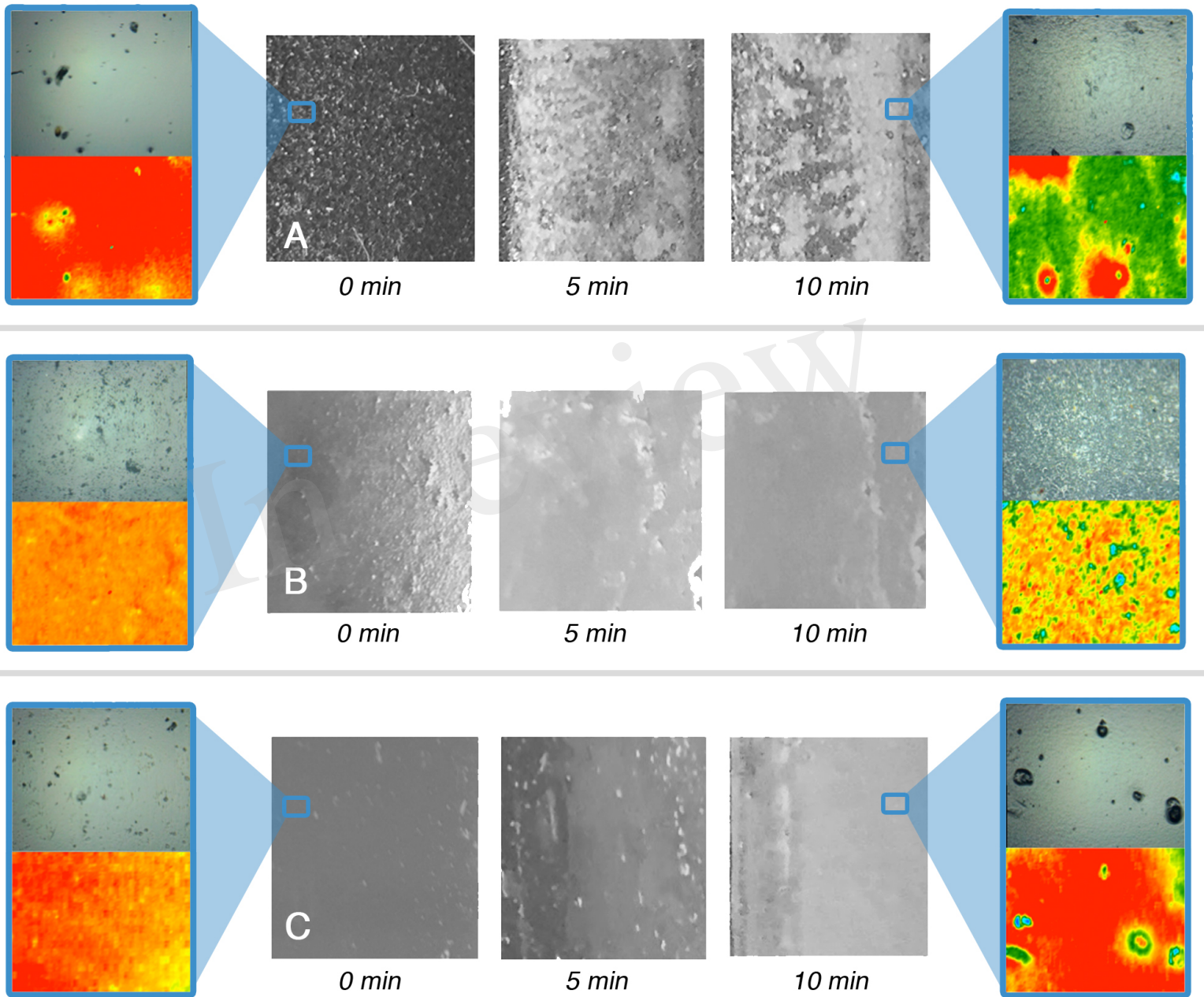




Figure 7.JPEG

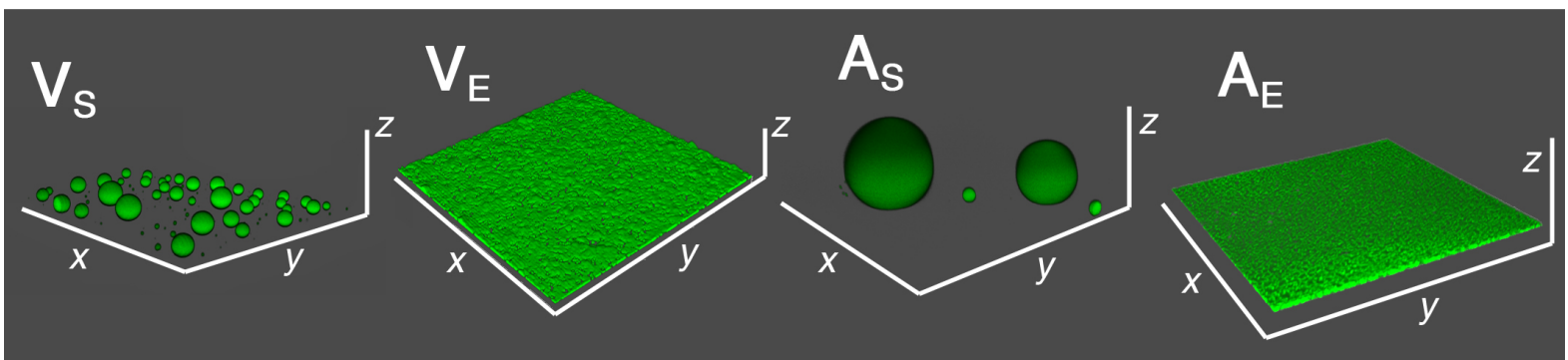
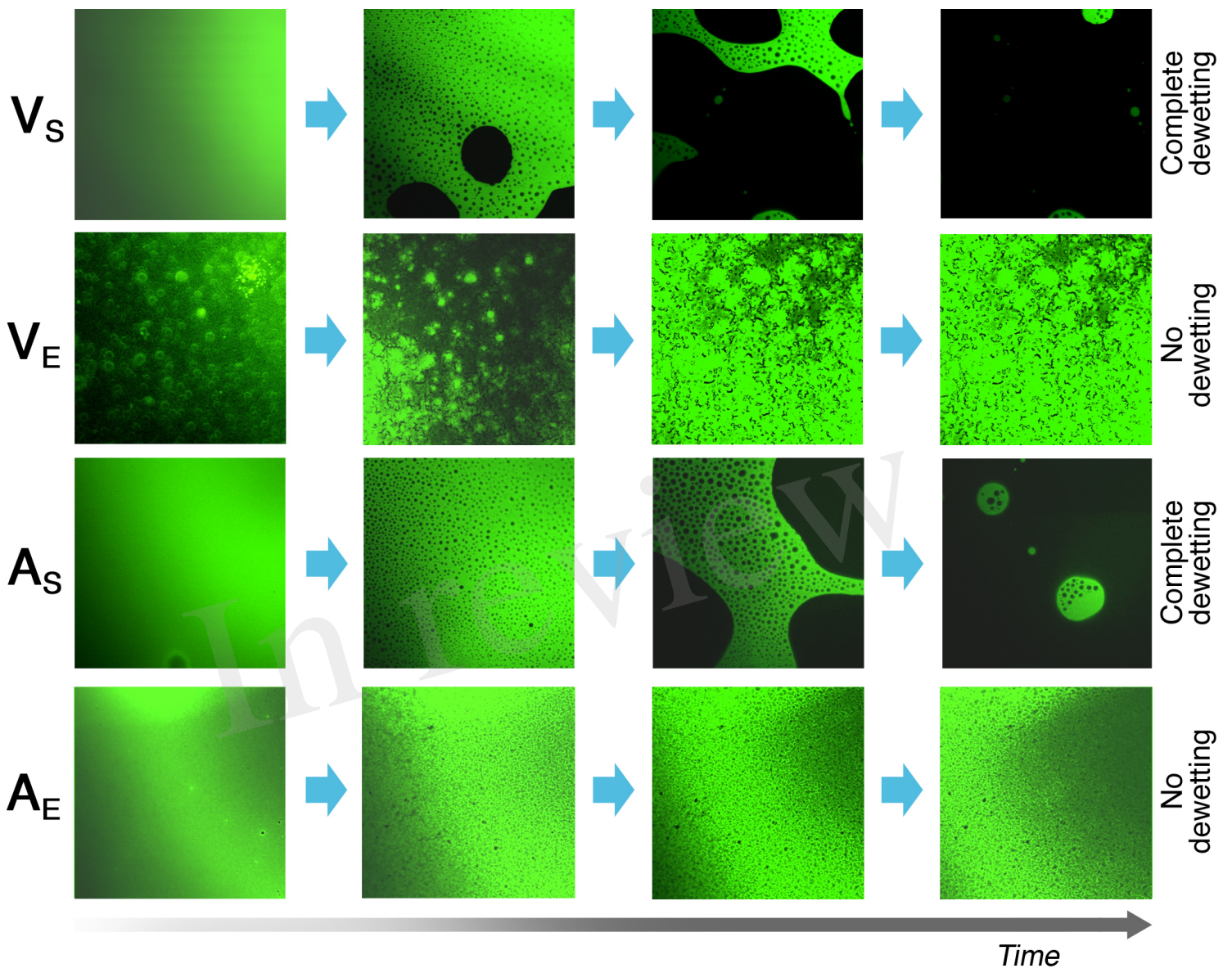


Figure 8.JPEG

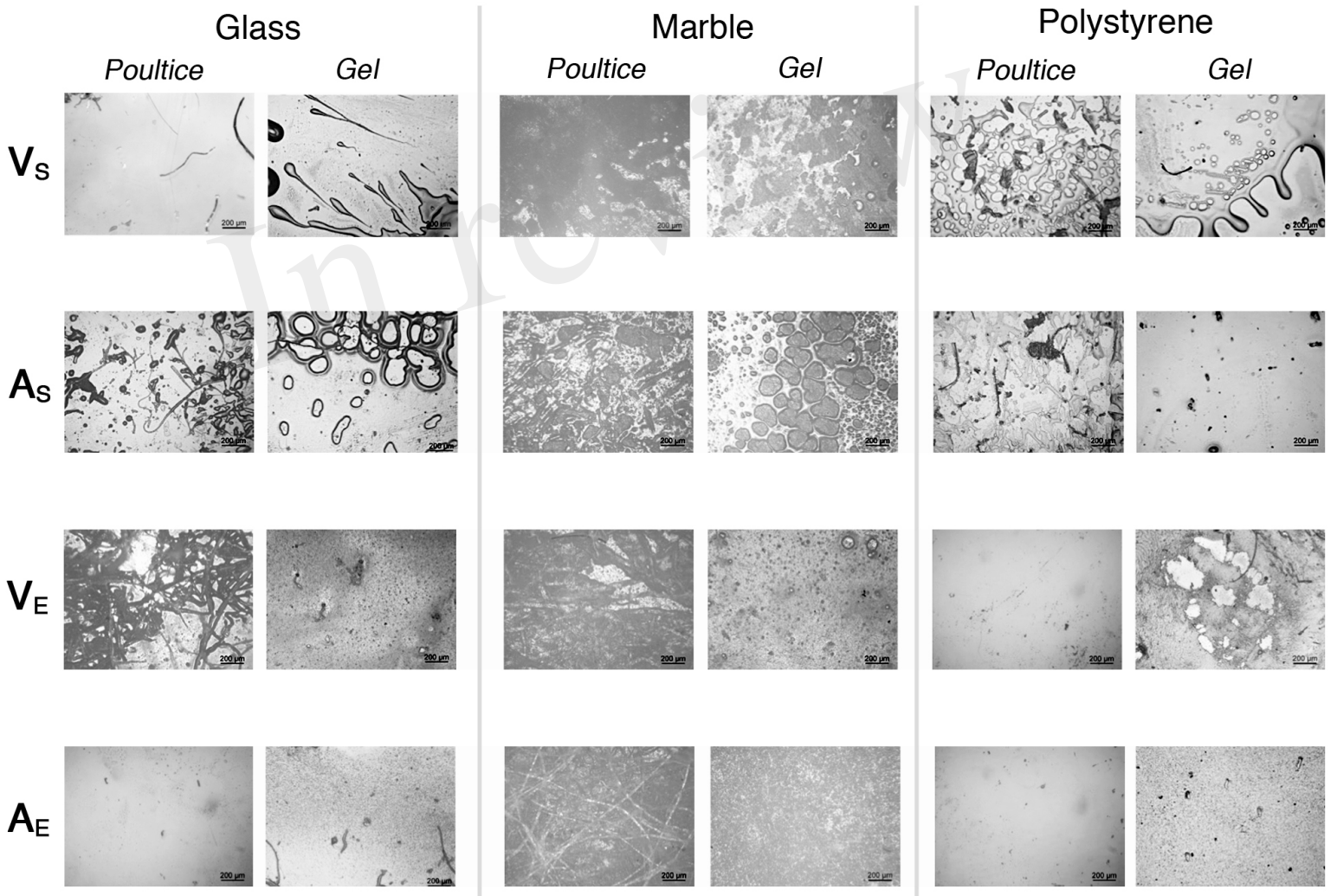
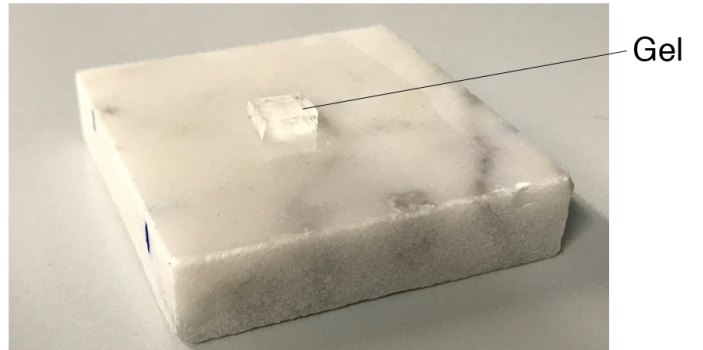
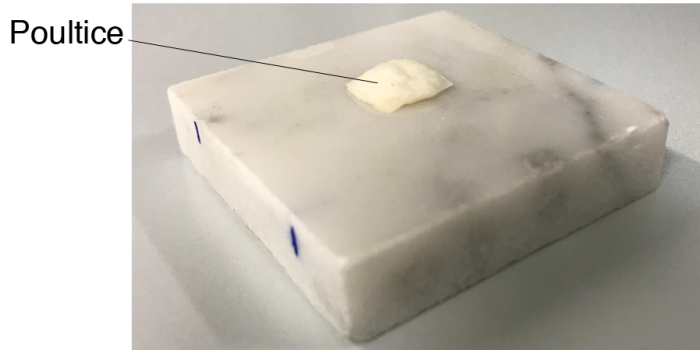


Figure 9.JPEG

