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Abstract	This paper examines w BioMEMS integrating potassium hydroxide (alcohol (IPA) additive coupled process. 3D s interferometric profilo to measure the reflecti aimed to find optimal BioMEMS. TMAH etc	wet and dry fabrication of vertical micro-mirrors in (110) silicon for use in an innovative gripping and micro force sensing functionalities. Wet anisotropic chemical etching in (KOH) and tetramethyl ammonium hydroxide (TMAH) with and without isopropanol was examined. Deep Reactive Ion Etched samples were produced using inductive urface roughness of samples was examined using scanning electron microscope, ometer and atomic force microscopy. An optic fiber displacement sensor was exploited wity of uncoated or coated samples with evaporated metallic thin film. The research fabrication technique for fabricating vertical micro-mirrors in polymer based ched silicon samples with surface roughness $R_a = 15.1$ nm showed highest reflectivity		

of all structures fabricated, reflectivity was more than doubled by adding a 10 nm layer of evaporated aluminum coating.

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3D surface topography and reflectivity of anisotropic etched 2 silicon micromirrors for BioMEMS 3

R. E. Mackay · N. Lionis · H. R. Le 4

5 Received: 8 August 2011 / Accepted: 28 September 2011 6 © Springer-Verlag 2011

7 **Abstract** This paper examines wet and dry fabrication of 8 vertical micro-mirrors in (110) silicon for use in an inno-9 vative BioMEMS integrating gripping and micro force 10 sensing functionalities. Wet anisotropic chemical etching 11 in potassium hydroxide (KOH) and tetramethyl ammonium 12 hydroxide (TMAH) with and without isopropanol alcohol 13 (IPA) additive was examined. Deep Reactive Ion Etched 14 samples were produced using inductive coupled process. 15 3D surface roughness of samples was examined using 16 scanning electron microscope, interferometric profilometer 17 and atomic force microscopy. An optic fiber displacement 18 sensor was exploited to measure the reflectivity of uncoa-19 ted or coated samples with evaporated metallic thin film. 20 The research aimed to find optimal fabrication technique 21 for fabricating vertical micro-mirrors in polymer based 22 BioMEMS. TMAH etched silicon samples with surface 23 roughness $R_a = 15.1$ nm showed highest reflectivity of all 24 structures fabricated, reflectivity was more than doubled by 25 adding a 10 nm layer of evaporated aluminum coating. 26

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1 Introduction

MEMS (Micro-electro-mechanical systems) technology 28 was introduced a few decades ago and MOEMS (micro-29 opto-electro-mechanical systems) during early 1990s, but it 30 was not until late 1990s when their reliability and effec-31 tiveness made them commercially viable. Since then, the 32 market of MEMS and especially MOEMS has experienced 33 an exponential growth in demand as more sectors in 34 industry require their capabilities. The automotive industry 35 is a characteristic example of where MEMS are used 36 extensively nowadays. Mostly in the form of sensors, 37 MEMS are implemented in many parts of a modern auto-38 mobile. An example is the accelerometers used to detect a 39 collision and inflate an airbag (Matsunaga and Esashi 2002). 40 As the market grows MEMS is being introduced to new 41 42 fields, one rapidly expanding field is BioMEMS (Biological micro-electro-mechanical systems) (Grayson et al. 2004; 43 Bashir 2004). Currently researchers are working on a pro-44 ject developing polymer micro-grippers with an optical 45 micro force sensor (Fig. 1) (Mackay and Le 2008; Mackay 46 47 et al. 2011). This is an exciting example to incorporate BioMEMS with MOEMS to fulfill dual requirements of 48 49 micro object handling and micro force sensing. This project aims to characterise the mechanical properties of the epi-50 thelium tissue. The mechanical characterization of tissues 51 will help scientists to understand fundamental cell physi-52 53 ology. With respect to cancer mechanical properties of normal cells could be compared to those of abnormal cells. 54 This could lead to new early diagnostic tools and therapies 55 in the treatment of colon cancer (Suresh 2007). However, 56 57 the silicon mirror requires through wafer etching to leave a free standing silicon mirror. 58

59 Deep etching through silicon wafers has been a problem for many years; the use of these deep etched structures as 60

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Fig. 1 Design concept for the gripping and force measurement system $% \left({{{\mathbf{F}}_{{\mathbf{F}}}}_{{\mathbf{F}}}} \right)$

mirrors for MOEMS remains a considerable challenge. 61 62 Through wafer etching must be incorporated in the fabri-63 cation of micro-grippers to allow a mirror to be created for 64 displacement sensing. Wet anisotropic etching is the lowest 65 cost and most commonly adopted solution; however geometries are significantly limited by crystal orientation 66 (Agarwal 2007). LIGA can be used to create vertical 67 68 mirrors; however this requires deep X-ray lithography 69 which is not available in the majority of fabrication units. 70 Deep reactive ion etching allows for complex geometries to 71 be fabricated through wafers, however this generally 72 results in high sidewall roughness although some groups 73 have managed to fabricate mirrors using this etching 74 technique alone (Marxer et al. 1997). A number of groups 75 have amalgamated dry and wet etch techniques to produce 76 complex geometries with smooth sidewalls (Agarwal 2007; 77 Yun et al. 2006). The wafers are orientated and patterned as 78 if for wet etching, however the first step is DRIE followed 79 by a polishing step in a wet etchant to produce {111} 80 planes (Price 1973). However, this increases the manu-81 facturing costs.

82 DRIE can produce non-vertical sidewalls which can 83 have spherical deviation which along with high roughness 84 cause optical losses in MOEMS. DRIE was developed 85 from RIE, a dry isotropic etching process using SF₆ radi-86 cals to etch silicon. One DRIE technique utilizes two gases 87 to create an anisotropic etch; SF_6 is used as the etchant due 88 to fluorine atoms reacting with the silicon substrate, C_4F_8 is 89 used to passivate the sidewalls between each SF_6 cycle to 90 allow deep holes to be etched in silicon. The cycles of 91 etching and passivation cause a curtaining effect to occur 92 on the sidewalls of the wafer creating high surface 93 roughness (Craciun et al. 2002). Deep etches cause large 94 defects to occur at the top of the sidewall which has been

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subjected to a large number of etching and passivation 95 cycles. 96

97 Wet anisotropic etching is a lower cost process and requires simple experimental setup which results in high 98 99 etch rates with smooth walls being fabricated with a high level of anisotropy. Wet chemical etched structures are 100 limited to specific geometries dictated by crystal orienta-101 tion of the specific silicon wafer type, i.e. (110). Three 102 main types of chemicals are used for anisotropic etching of 103 silicon; alkaline metal hydroxides (i.e. potassium hydrox-104 105 ide (KOH) or sodium hydroxide (NaOH)) are relatively cheap, non-toxic and result in low roughness sidewalls; 106 diamine based etchants (i.e. ethylenediamine pyrocatecol 107 (EDP)) require complex etch apparatus, have short shelf 108 life and produce highly toxic gases during the reaction with 109 silicon; quaternary ammonium hydroxides (i.e. tetramethyl 110 ammonium hydroxide (TMAH)) has excellent selectivity, 111 is non-toxic, however it is more expensive than alkaline or 112 diamine etchants but can be doped with silicic acid and 113 ammonium persulfate to increase sidewall smoothness 114 (Biswas and Kal 2006). Etching of (110) silicon gives high 115 anisotropy due to the etch rates of the different planes 116 $\{110\}$ etches faster than $\{100\}$ which is faster than $\{111\}$ 117 by a ratio of 400:200:1 at 85°C (Kendall 2003). 118

Isopropanol alcohol (IPA) has been added to alkaline 119 and TMAH etchants to help improve sidewall surface fin-120 ish. IPA has no active role in the reaction between etchant 121 122 and silicon (Williams and Muller 1996), however it does reduce the reaction rate, therefore lowering sidewall 123 roughness. Palik et al. (1983) used Raman spectroscopy to 124 understand the reactions occurring during alkaline etching 125 of silicon (Eq. 1) 126

 $Si + 2H_2O + 2OH^- = Si(OH)_2O_2^- + H_2O + 2H_2$ (1)

The etchant must be mixed mechanically to ensure 128 striation does not occur giving different etch rates 129 throughout the solution (Palik et al. 1983); also if IPA is 130 present this does not readily dissolve in solution, mixing 131 ensures even concentration of IPA throughout the entire 132 solution. Hydrogen bubb = eadily form on the wafer 133 surface, as seen from Eq. 1 Seidel et al. 1990), agitation of 134 the etchant helps remove hydrogen bubbles which can act 135 as a 'pseudo' mask stopping small areas from being etched, 136 increasing surface roughness, due to the formation of 137 hillocks (Yang et al. 2005). Mechanical agitation ensures 138 hydrogen bubbles are removed quickly from the surface of 139 the silicon being etched. 140

Both DRIE and wet chemical etching rely on a large141number of variables to ensure smooth, defect free, vertical142sidewalls are obtained. Wet chemical etching depends on143type of etchant, concentration of etchant, temperature,144mixing rate, additives and alignment to {111} plane. DRIE145variables include pressure and flow rate of etchant and146

147 passivation chemicals. RF power, distribution of reactive 148 fluorine species and concentration and distribution of waste 149 products. Finally wet etching can cause stiction of the 150 envisaged free hanging MEMS structure, which must be 151 taken into account when designing for BioMEMS and 152 MOEMS, whilst DRIE avoids this due to the dry etchants 153 being used.

154 In total, 42 wet etching experiments were carried out 155 using KOH, KOH + IPA, TMAH and TMAH + IPA. 156 Deep reactive ion etched samples were also examined which were etched using inductive coupled plasma process. 158 Sample reflectivity was studied using Philtec D6 fibre optic displacement sensor. Uncoated bare Si samples were tested 160 along with samples coated with a 20 nm layer of evaporated Au-Pd and 10 nm Al. Samples were examined using light microscopy and SEM. Surface profiling was done using a Dektak (Veeco) surface profiler and Zygo 3D interferometric profilometer.

165 **2** Experimental procedure

166 Silicon wafers (110) p-type (resistivity 1–5 ohm cm) single side polished with a 76 mm diameter and thicknesses of 167 168 381 µm were used for anisotropic etching experiments. 169 Wafers were thermally oxidized in air to create a 1 µm 170 SiO₂ insulating layer. The wafer was spin coated with 171 Shipley 1813 photoresist of about 1.2 µm; this was baked 172 on a hotplate at 115°C for 3 min. Two photomask patterns 173 were used throughout the experiments (Fig. 2). The wafer 174 was aligned to the photomask so the {111} plane was 175 parallel to the mirror pattern. The wafer was exposed to the 176 mask pattern using an OAI J500 photo aligner. The wafer 177 was exposed for 4 s and the photoresist developed by 178 immersion in MICROPOSIT MF 321 diluted with deion-179 ized H₂O at a ratio of 1:3 respectively at room temperature 180 for 20 s. The sample was then placed in buffered hydro-181 fluoric acid (BHF) for 7 min 30 s until the oxide was 182 removed leaving bare silicon. Remaining photoresist was 183 removed by rinsing in acetone. The thickness of the oxide 184 layer was verified using a Dektak surface profiler.

185 Wet etching experiments were carried out in a Teflon 186 beaker (Fig. 3); a magnetic stirrer was placed in the bottom 187 of the beaker to agitate the chemicals as a reaction rate 188 controlled process requires a constant diffusion rate and 189 also avoids stratification of the etchant to maintain even 190 etching across the wafer. A small Teflon guard was placed 191 over the stirrer to avoid collision between this and the sil-192 icon wafers being etched. The speed of the stirrer was set to 193 a constant speed of 250 rpm when aqueous solutions of 194 KOH or TMAH were used. However, for solutions with IPA 195 added, the speed was set to 500 rpm to ensure proper mix 196 and distribution of the alcohol into the solution. The beaker



Fig. 2 Mask patterns used for back etching on (110) silicon wafers a early mask used for initial experiments; b mask designed for polymer micro-gripper system

was placed in a water bath which was placed directly onto a 197 hotplate. The hot plate temperature controller was used in 198 199 order to set the etching temperature. A probe was placed in the etchant to sense the temperature within the solution and 200 provide feedback to the controller in order for the temper-201 202 ature to remain constant. The hotplate was set at a temperature two times larger than the etching temperature, due 203 to the etchant being in a Teflon beaker immersed in a water 204 bath, which gave a temperature tolerance to the etchant of 205 \pm 5°C. Outside of this range the etching temperature either 206 207 would not be reached or it would be exceeded. Evaporation 208 of the solution was an issue so the plastic beaker was sealed 209 with a cap which featured a small hole for the temperature probe which was placed directly into the solution in order to 210 provide the necessary feedback. 211

KOH solutions were made by dissolving KOH pellets in 212 213 DI H₂O; this will give KOH concentration $\pm 5\%$ due to the absorption of moisture into the KOH pellets (Powell 2001). 214 TMAH solutions were made by mixing 20 wt% TMAH 215 solution with DI H₂O. 4% IPA was added to solutions 216 which contained the alcohol. Wet chemical experiments 217 were carried out using 20 wt% KOH, 25 wt% KOH, 218 30 wt% KOH, 25 wt% KOH + IPA, 30 wt% KOH + 219 IPA, 20 wt% TMAH, 10 wt% TMAH, 13 wt% TMAH and 220 221 13 wt% TMAH + IPA.

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Fig. 3 Experimental setup for silicon etching experiments a Overview of experimental setup b teflon beaker for silicon etching

222 DRIE samples were obtained from the Scottish Micro-223 electronics Centre Edinburgh. These samples were produced 224 on 76 mm Si wafers with 200 µm thickness. SF₆ is used for 225 etching the silicon; this is an anisotropic process however it 226 causes an initial undercut in the silicon. The C_4F_8 is used to 227 passivate the sidewall to stop the area being isotropically 228 etched during the next etching cycle. This helps restrain 229 isotropy but this cannot be completely eliminated as a small 230 amount of undercutting occurs in every etch cycle.

231 2.1 Reflectivity tests

Reflectivity tests were carried out in order to relate surface
roughness with optimal reflectivity of samples. Samples
tested were polished and unpolished silicon, 30 wt%
KOH + IPA, 25 wt% KOH + IPA, 13 wt% TMAH,
13 wt% TMAH + IPA, 13 wt% TMAH coated with thermally evaporated AI, DRIE and DRIE then coated with
sputtered Pd/Au.

A 3D micromanipulator by Kleindiek Nanotechnik was used to hold and manipulate samples, whilst a specially designed fibre optic holder was used to ensure the displacement sensor was parallel to the chip (Fig. 4). This allowed for movement of samples in 3 axes with step sizes ranging from 0.25 to 500 nm. Precise alignment between the {110} sidewall and fibre optic displacement sensor

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Fig. 4 Reflectivity testing setup

(Philtec D6) is paramount to retrieve accurate results,
sidewall thickness is 381 μm. The fibre optic displacement
sensor had wavelength 670 nm.246
247

Static experiments were used to test optimal reflectivity 249 of samples whilst dynamic testing examined the range of 250 voltages that could be acquired when the chip was dis-251 placed cyclically over ±288 µm. Nanocontrol software 252 supplied with the micromanipulator was used to create a 253 macro program to run a series of displacement loops. 254 255 A Labview program was used to retrieve optimal voltages and displacement voltages. 256

3 Results and discussions

The surface finish of samples was analyzed using a Dektak258surface profiler and Zygo interferometric profilometer and259Veeco CPii atomic force microscope (AFM). Samples were260also examined using both light microscopy to help with261sample selection and then scanning electron microscope to262examine the microstructure of the etched {111} sidewalls.263

3.1 Deep reactive ion etched samples

265 Deep reactive ion etched samples showed even etching throughout the majority of the sample. Samples show striated 266 267 lines due to the process of DRIE (Fig. 5a), cycling etching 268 and passivation. AFM results showed the depth of trenches formed due to the DRIE process (Fig. 5b), the figure shows a 269 $50 \times 50 \ \mu m$ area and the resulting curtaining pattern is 270 clearly visible. Etch trenches were measured in the centre of 271 a Si wafer sidewall showed depths of \sim 7.5 µm which 272 increases surface roughness significantly. DRIE samples 273 274 varied greatly depending on etch parameters. Average 275 roughness of DRIE samples was found to be $R_a = 1,707$ nm

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Fig. 5 Sidewall surface of deep reactive ion etched samples a SEM image and b AFM scan profile

(b)

276 and rms = 2,027 nm. However the majority of samples 277 showed poor reflectivity. One sample with much lower sur-278 face roughness $R_a = 533$ nm and rms = 684 nm was 279 selected for reflectivity tests. SEM examination of DRIE samples showed large defects occurring at the top of the 280 281 sidewall where wafers were exposed to thousands of etching 282 and passivation cylces (Fig. 6). These deep trenches could 283 seriously affect reflectivity if they penetrate $>50 \ \mu m$ deep into the sidewall. An area of $150 \times 150 \ \mu m$ is needed for 284 285 reflection to the optical displacement sensor. However, the 286 majority of these defects do not penetrate beyond 10 µm.

287 3.2 KOH etched samples

288 One of the greatest difficulties the authors faced when 289 etching with KOH was using SiO₂ as a mask for KOH



Fig. 6 DRIE defects seen at the top of the wafer where the material is subjected to hundreds of etching and passivation cycles



Fig. 7 KOH etched wafers with 1 μ m SiO₂ mask layer

etched samples. Some pitting was seen on the surface of 290 291 the wafers due to uneven etching of SiO₂ across the surface of the wafer, this resulted in masked areas of silicon 292 etching significantly (Fig. 7). KOH etched samples fea-293 294 tured a number of circular trenches forming on the surface of the {111} plane when etched at low concentrations 295 <20 wt%. In order to etch vertical mirrors in (110) silicon 296 297 KOH concentrations in the range of 25-35 wt% were found to be more desirable and fewer circular trenches 298 299 appeared parallel to the {111} plane. Additions of IPA at these concentrations greatly reduced surface roughness; 300 however some pitting was still seen on the surface. The Ξ 301 302 addition of IPA reduces the etch rate due to changes is surface energy of Si. Figure 8a shows a number of hillocks, 303 which have formed due to hydrogen bubbles acting a 304 'pseudo' masks throughout the etching process. The AFM 305 results show an uneven surface with significant areas of 306 pitting and the edge of a hillock formed (Fig. 8b). 307

KOH samples showed high surface roughness, samples 308 309 etched in 25 wt% KOH at 75°C showed $R_a = 3,127$ nm. 25 wt% KOH + 4% IPA showed lower average roughness 310 of $R_a = 1,113$ nm and rms = 1,316 nm when etched at 311 75°C. Increasing KOH concentration to 30 wt% + 4% IPA 312 decreased surface roughness to $R_a = 616$ nm and rms = 313 763 nm. Increasing concentrations of KOH etching solu-314 tion and addition of IPA significantly decreases the etch 315 rate, slower etch rates show better surface finish due to less 316

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Surface Height (um)





Fig. 8 SEM image of silicon sidewall etched in 25 wt% KOH +4% IPA, 70°C, 4 h 30 min **a** SEM image of vertical mirror **b** AFM surface profile

317 defects being formed on the Si surface. High surface 318 roughness of KOH samples could be due not only to slight 319 misalignment with $\{111\}$ plane but also depth of etch 320 $>200 \mu m$ (Sato et al. 1999b).

321 3.3 TMAH etched samples

TMAH showed much better selectivity to the SiO₂ mask 322 323 than KOH. Vertical sidewalls aligned to the {111} plane 324 were produced with fewer defects due to this high selec-325 tivity. Uniformity of the etched surface was observed to be 326 greater in TMAH than KOH. TMAH samples etched with 327 concentrations of 13 wt% showed lower surface roughness 328 than those etched at 10 and 20 wt%. The optimum etching 329 temperature was found to be 85°C.

TMAH samples had a lowest roughness of all samples produced with $R_a = 14$ nm and rms = 20 nm. Figure 9b shows an area 10×10 µm and the average surface height over this area. Two large hillocks can be seen in this area but their height is ~100 nm.

Samples etched with TMAH only were observed to be
less rough than those with the addition of 4% IPA. Addition of IPA caused striations to occur across the wafer

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Fig. 9 AFM surface profile of silicon sidewall etched in 13 wt% TMAH at 85°C a SEM image of vertical mirror b AFM surface profile

increasing surface roughness, however, the addition of IPA338did eliminate the formation of hillocks on the {111} sur-
face (Fig. 10). Hillocks formed during TMAH etches were339significantly smaller and fewer than seen in KOH etchants.341

Optimal alignment was achieved by manipulating the chip when set at a distance of 200 µm from the tip of the fiber optic displacement sensor. This distance gives the optimal voltage output from the displacement sensor. The maximum voltages were retrieved during static testing (Table 1). Silicon wafer surface samples of polished silicon and unpolished silicon were used as reference values. 343

Polished silicon showed the highest static output voltage, 1.66 V, whilst unpolished the lowest; maximum voltage output for the displacement sensor is 5 V. Polished 352

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Fig. 10 Samples etched in 13 wt% TMAH + IPA showed striation and rougher surfaces than those etched with TMAH only

353 and unpolished silicon samples were much easier to align 354 than sidewall samples as they had a much greater depth 355 >2 mm when compared to etched samples which had a 356 depth 200-381 µm. TMAH showed lowest surface rough-357 ness and highest maximum voltage, 0.83 V during static 358 tests, this increased significantly to 2.48 V once coated 359 with 10 nm Al. DRIE samples tested were those with the 360 lowest roughness ($R_a = 533$ nm). DRIE samples showed output voltage of 0.72 V which increased to 1.12 V whe 361 362 the Pd-Au coating was sputtered on the surface. TMA-363 H + IPA samples showed much lower static output voltage 364 of 0.35 V due to the striations scattering light away from the fiber optic sensor. Samples etched in 30 wt% 365 366 KOH + IPA gave an output voltage of 0.24 V whilst 25 wt% KOH + IPA gave an output voltage of 0.04 V 367 showing slower etch rates increase reflectivity. However 368 369 due to defects in KOH etched silicon arising because of 370 poor masking, reflectivity is lower than DRIE samples. 371 Samples etched with KOH only could not be tested as

Table 1Summary of optimal voltage outputs and voltage variationfor $\pm 288 \ \mu m$ of displacement

Etching technique/solution	Optimal voltage (V)	Voltage variation (V)
Polished silicon	1.66	0.26
DRIE metalized	1.12	0.1
13 wt% TMAH metalized	2.48	0.07
13 wt% TMAH	0.83	0.11
DRIE	0.72	0.17
13 wt% TMAH + IPA	0.35	0.05
30 wt% KOH + IPA	0.24	0.05
Unpolished silicon	0.15	_
25 wt% KOH + IPA	0.04	-



Fig. 11 Static reflectivity tests showing optical sensor output voltage vs average roughness of silicon sidewall mirror

samples did not reflect enough light to produce an output 372 373 voltage. Static output voltages were related to average roughness (R_a) measurements (Fig. 11). Polished silicon 374 375 $(R_{\rm a} = 5 \text{ nm})$ shows significantly higher output voltage when compared to TMAH samples ($R_a = 14$ nm), it is 376 believed this is due to the small area being tested and 377 378 difficulty in aligning the 200 µm thick sample to the tip of 379 the optic fiber. Sato et al. found orientation dependence of etching differs bet = the two wet etchants, within the 380 current experiments TMAH was found to be easier to align 381 to {111} orientation (Sato et al. 1999a). It can be seen that 382 DRIE samples ($R_a = 533$ nm) with lower roughness than 383 KOH + IPA samples (567 nm) showed higher output 384 voltage than the slightly rougher KOH + IPA samples. 385 Addition of highly reflective metals (i.e. Au-Pd), sputtered 386 onto the silicon sidewalls greatly increased static voltage 387 output. Au-Pd has excellent reflectivity of near 100% for 388 the wavelength of light (690 nm) being reflected. Al has 389 slightly lower reflectance around 95% for the specific 390 displacement sensor however it is much lower cost. Si 391 showed lowest reflectivity and is highly dependent on 392 surface roughness (Hashim and Salih 2005). 393

Displacement experiments were carried out using the 394 Kleindiek micromanipulator, the range of output voltage 395 difference for displacement of 566 µm is shown in Table 1. 396 Results were poorer than expected but it is believed this is 397 due to parasitic motion in the micromanipulator, resulting 398 in rotation of samples, this resulted in misalignment of the 399 wafer to the fiber optic sensor. 400

4 Conclusions

Etching of silicon in multifunctional BioMEMS represents402a complex procedure due to the number of variables and403the associated outcomes of each. Fabrication of vertical404mirror surfaces via anisotropic deep etching through wafers405for BioMEMS is achievable by controlling the etch rate406and extraction of gaseous products from the etchant. One407

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408 obvious limit of anisotropic etching is the limit of the 409 pattern to the crystal orientation of (110) wafers, it is 410 impossible to produce rectangular structures and trapezoids 411 are formed in the structure; however it is proved possible to 412 orientate the wafer so the mirror is present and perfectly 413 aligned to the {111} plane. A new method was developed 414 to examine the reflectivity of micro-engineered surface 415 using optic fiber displacement sensor and 3D piezoelectric manipulator. 416 417

The results show that concentration of both KOH and 418 TMAH is an important factor to produce optimal surfaces. 419 Generally higher percentages of KOH result in smoother walls, here the optimal is 30 wt%, this slows down the 420 421 etch rate resulting in smoother walls when examined 422 using SEM. KOH must be used with the addition of IPA, 423 to reduce surface energy of Si and cause fewer hillocks to 424 be formed, samples etched in KOH only could not be 425 used as vertical silicon mirrors. The addition of IPA to 426 TMAH solutions resulted in rougher, striated surfaces 427 being formed. For TMAH, the optimal etchant concen-428 tration was found to be 13 wt%. Temperature of etchant 429 also results in variability of smooth sidewalls, for 430 KOH + IPA lower temperatures, <70°C resulted in optimal smoothness, the lowering of temperature helps 432 slow the etch rate. For TMAH etchants, the optimal was 433 found to be 85°C.

434 The argument by previous authors that TMAH etches 435 can result in smoother sidewalls than KOH etched samples, 436 was proved. TMAH showed good results with the lowest 437 $R_{\rm a}$ recorded and highest output of uncoated micro-samples. It was also found that IPA did not improve the surface 438 439 quality when TMAH was used. KOH + IPA showed 440 higher roughness due to the formation of sizable hillocks and significant pitting in the SiO₂ mask. If KOH is to be 441 442 used to form vertical sidewalls SiN must be employed as 443 the mask.

DRIE samples showed significantly higher surface 444 445 roughness than TMAH but performed well during reflectivity results. Uncoated samples gave a reasonable static 446 output voltage from the fiber optic displacement sensor. 447 448 Coated samples of DRIE and TMAH gave the best overall 449 reflectivity results, as expected.

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