

# Redox-dependent Franck-Condon blockade and avalanche transport in a graphene-fullerene single-molecule transistor

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## Abstract

We report transport measurements on a graphene-fullerene single-molecule transistor. The robust architecture where a functionalised C<sub>60</sub> binds to graphene nanoelectrodes results in strong electron-vibron coupling and weak vibron relaxation. Using a combined approach of transport spectroscopy, Raman spectroscopy and DFT calculations, we demonstrate center-of-mass oscillations, redox-dependent Franck-Condon blockade and a transport regime characterized by avalanche tunnelling in a single-molecule transistor.

Coupling between electronic and vibrational degrees of freedom in single-molecule devices can lead to transport properties very different from those of metal/semiconductor nanostructures.<sup>1</sup> Charge transfer can excite vibrational modes,<sup>2</sup> or vibrons, and strong electron-vibron coupling leads to suppression of tunnel current at low bias.<sup>3,4</sup> Theory further predicts super-

Poissonian current noise, characterised by giant Fano factors, for molecular junctions with weak vibron-relaxation owing to avalanche-type charge transfer.<sup>5</sup> Here, we demonstrate a robust architecture of a single-molecule transistor where a functionalised C<sub>60</sub> binds to graphene nano-electrodes via  $\pi - \pi$  interactions. The stability and weak vibron-relaxation of our system enables us to investigate redox-dependent electron-vibron coupling and avalanche transport.

We perform transport measurements of graphene-fullerene single-molecule transistors fabricated using feedback-controlled electroburning.<sup>6</sup> The molecule bridging the graphene nano-gap is a pyrene functionalised C<sub>60</sub> bisadduct synthesised through a 1,3 dipolar cycloaddition using 1-pyrene carboxaldehyde and N-methylglycine (Supplementary section 1.1).<sup>7</sup> While the functionalisation results in different isomers, steric considerations mean only the cis-2 and cis-3 isomers are expected to bridge the graphene nano-gap. Density Functional Theory (DFT) calculations of the relaxed graphene-fullerene-graphene structure confirm that the cis-3 isomer forms the most stable configuration without steric hindrance, as shown in Figure 1a. Iso-surfaces of the frontier orbitals of gas phase cis-3 isomer are shown in Figure

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1b.<sup>8</sup> The wavefunction of the lowest unoccupied molecular orbital (LUMO) level is mostly localized in the central C<sub>60</sub>, whereas for the highest occupied molecular orbital (HOMO) level, the wavefunction is extended to the pyrene anchors. Comparison of the iso-surfaces of frontier orbitals for cis-2 and cis-3 isomers further show that the wavefunction delocalization of the HOMO levels are more pronounced in the cis-3 isomer, indicating that it has a better  $\pi$ - $\pi$  interaction with the graphene electrodes and is therefore more likely to bridge the junction (Supplementary Figure S11). In pristine C<sub>60</sub>, DFT predicts LUMO dominated transport which was verified with thermoelectric experiments.<sup>9</sup> In contrast, our DFT calculations predict that charge transport in the bisadduct is more likely to be HOMO dominated.<sup>10,11</sup>

Figure 1c shows the current  $I$  as a function of bias  $V_b$  and gate  $V_g$  voltage of a single-molecule transistor (device A), measured at 20 mK. The addition energies between adjacent redox states, measured from the height of the Coulomb diamonds, are  $E_{\text{add}} \sim 160$  meV for all charge-state transitions. We attribute the absence of odd/even oscillations of  $E_{\text{add}}$ , previously observed for two-fold spin shell systems,<sup>6,12</sup> to orbital degeneracies of the C<sub>60</sub> molecule.<sup>13</sup> Calculations have shown that the interactions within each shell in C<sub>60</sub> are almost constant, with only a slight increase of the electron-electron on-site Coulomb interaction in the middle of each shell where electrons with opposite spins begin to fill.<sup>13</sup> We observe a small increase of  $\sim 10$  meV per electron added from charge state  $N$  to  $N + 2$  which we attribute to the on-site Coulomb interactions. In the constant interaction model  $E_{\text{add}}$  is given by the sum of the charging energy  $E_C$ , due to Coulomb interactions of the electrons in the molecule, and the energy gap  $\Delta_{\text{HL}}$  between the HOMO and LUMO.<sup>14</sup> For a two-fold degenerate system,  $E_{\text{add}}$  oscillates between  $E_C$  and  $E_C + \Delta_{\text{HL}}$ , while for systems with higher degeneracies,  $E_{\text{add}} = E_C$  until a level is completely filled. From the fact that  $E_{\text{add}}$  here is constant for three charge-state transitions, we infer that electrons are filling exclusively a HOMO that is at least four-fold degenerate. This is consistent

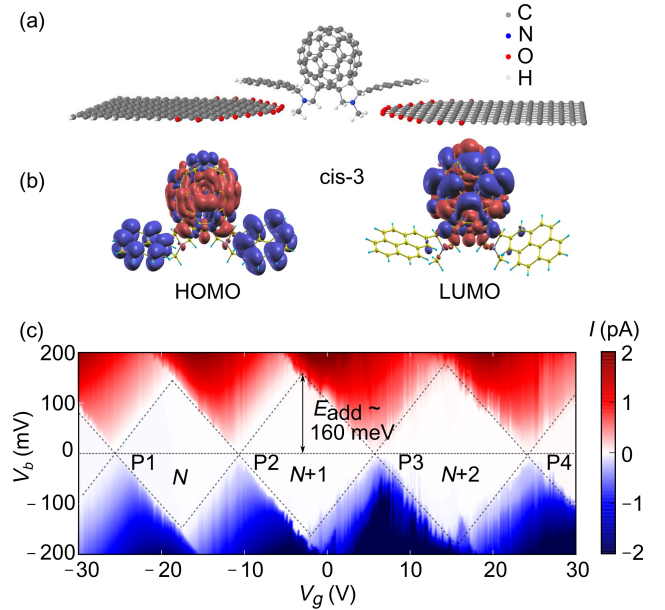


Figure 1: (a) DFT calculations showing the relaxed structure of the cis-3 isomer bridging the graphene nano-gap. The electrodes are fabricated using feedback-controlled electroburning of chemical vapour deposition grown graphene transferred to a heavily doped silicon substrate with a 300 nm thick silicon oxide layer, which also serves as a back-gate that electrostatically modulates the chemical potential of the molecule. The molecules are deposited on the electrodes from a chloroform solution (Methods). (b) DFT simulations of the iso-surfaces of the LUMO and HOMO of cis-3 isomer. (c) Current stability diagram of device A. All measurements are performed at 20 mK unless otherwise stated. Within each diamond, the system is in Coulomb blockade (CB) and the charge on the molecule is stable. The molecular redox state changes by one between adjacent diamonds.

with our DFT prediction of HOMO dominated transport and calculations of the Kohn-Sham HOMO and LUMO energies, which show that the HOMO (-4.59 eV) and HOMO-1 (-4.60 eV) levels are almost degenerate in energy, in contrast to the LUMO (-3.12 eV) and LUMO+1 (-2.86 eV) levels (Supplementary section 2.3).

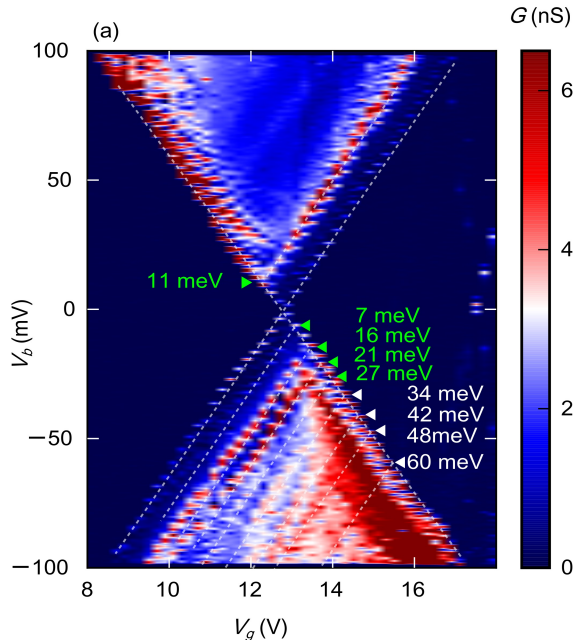


Figure 2: (a) Conductance stability diagram of the  $N + 1$  to  $N + 2$  transition (P3 in Figure 1c) obtained by taking the numerical derivative of the current measured. Excited states are indicated by the white and green arrows. Details on how the excited states are determined are included in Supplementary section 1.5.

Figure 2 shows a map of the differential conductance  $G$  of charge transition  $N + 1$  to  $N + 2$  of device A taken with a higher resolution (P3 in Figure 1c). The gate potential for the charge transition is different for repeated measurements due to a common hysteretic effect likely related to charge traps in the oxide. The effect occurs when the gate potential is swept quickly across a large range, but is otherwise stable when swept slowly. We observe lines corresponding to excited state transitions at energies  $\geq 33$  meV (white arrows) that are in excellent agreement with the intrinsic vibrational modes of pristine  $C_{60}$ ,<sup>15</sup> as summarized in Table 1. We also observe excited states at energies  $< 33$  meV (green arrows), below the lowest vi-

brational mode of the pristine  $C_{60}$ . The asymmetry of the excited states on the bias polarity can be explained through a rate equation model involving asymmetric tunnelling barriers, which was previously observed in the orbital excited state spectrum of a single donor in silicon. When the excited state relaxations are fast compared to the tunneling rates, an electron can tunnel on to the molecule through any number of excited states that lie within the bias window but only leave through the ground state as the molecule relaxes. In the limit where the molecule-electrodes coupling is very asymmetric, current steps are not expected for one bias polarity.<sup>16</sup>

A comparison of the transport spectroscopy and Raman spectra of pristine  $C_{60}$  and bisadduct (Table 1) indicates that these excited state transitions have a vibrational nature. To identify the mechanical motion of these additional low-lying vibrational modes, we use DFT results where we analyse the mechanical degrees of freedom of the bisadduct in the limit where the atomic mass of the pyrene anchor groups approach infinity (Supplementary section 2.6). In this limit, the anchor groups are effectively clamped and the modes corresponding to internal vibrations of the pyrene are filtered from the vibrational spectrum. DFT calculations in this limit reveal six vibrational modes at energies  $< 33$  meV, with two sets of three modes each corresponding to the translational and rotational center-of-mass motion of the central  $C_{60}$  with respect to the anchor groups (Supplementary Figure S13a, animations in Supplementary Information).

Mode	Energy (meV)			
	Transport (this work)	Raman (this work)	DFT (this work)	$C_{60}$ (Ref. <sup>15</sup> )
CM $T_y$	7		8	
CM $T_x$			10	
CM $R_z$	11	13	12	
CM $T_z$	16	16	16	
CM $R_x$	21	19	19	
CM $R_y$	27	25	21	
$H_g(1)$	34	33	32, 33, 34	33
$T_{2u}(1)$	42			43
$G_u(1)$	42			43
$H_g(2)$	48	55		53
$A_g(1)$	60	61		61

Table 1: Comparison between measured excitation energies from transport and Raman spectroscopy, calculated values and intrinsic vibrational modes of pristine C<sub>60</sub>. Transport measurements are performed at 20 mK in vacuum with the molecule anchored to graphene electrodes. Raman spectroscopy is performed at room temperature with the molecules drop-casted on to gold coated silicon substrates to enhance the Raman scattering. DFT calculations are performed with the condition where the degrees of motion of the anchor groups are completely restricted in order to elucidate the center-of-mass motions. In the actual system, the pyrene groups are expected to exhibit small motions. The values for the vibrational modes of pristine C<sub>60</sub> are obtained from.<sup>15</sup> We find that the discrepancies between our measurements fall within the range of values reported in ref. 18, which are obtained from experiments and *ab initio* theories reported by different groups and can differ by up to 10 meV.

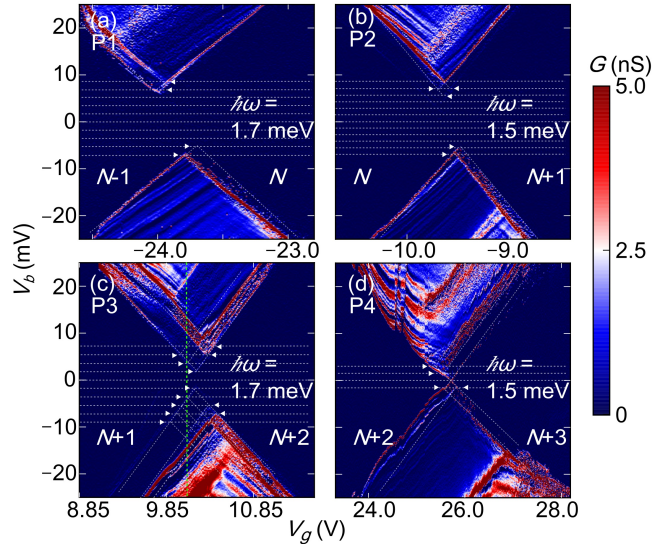


Figure 3: (a-d) Conductance stability diagrams for transitions  $N-1$  to  $N+3$ . The low bias current suppression is due to FC blockade in systems with strong electron-vibron coupling. Sets of integer spaced vibrational state lines (white arrows) are observed in the low bias regime. The vertical green line in (c) is the gate voltage at which the large current noise discussed in Figure 5 is measured. The energy of a single vibrational quantum,  $\hbar\omega$ , is extracted for all 4 charge transitions (Supplementary section 1.5). These energy spacings are determined by first fitting the peaks of the excited state transitions at two different  $V_g$  to a thermally broadened Lorentzian in order to extract the peak position and slopes.  $\hbar\omega$  is then determined by the intersection of the excited state lines with the Coulomb diamond edges.

Figure 3 shows low energy excitations observed for all four charge transitions (P1-4 in Figure 1c) with spacings of 1.7 meV (P1), 1.5 meV (P2) 1.7 meV (P3) and 1.5 meV (P4) (white arrows). Low energy excitations are similarly observed in measurements of device B ( $\sim 2$  meV, Supplementary Figure S7a). Lines with different slopes, corresponding to different capacitive couplings, are probably not related to the energy spectrum of the molecule. They can be from a different Coulomb diamond related to a separate quantum dot located close to the device or associated with density of states fluctuations in the graphene electrodes. Similar fluctuations have also been observed in silicon SETs.<sup>17</sup> To account for the nature of the low energy excitations ( $\sim 2$  meV), we analyse the vibrational modes of the bisadduct using DFT where all the mechanical degrees of freedom of the atoms are allowed (Supplementary section 2.6). The lowest non-zero mode is at 1.7 meV, consistent with our measurements (Supplementary Figure S13b). This mode is due to the combined center-of-mass motions of the central C<sub>60</sub> and pyrene anchors moving perpendicularly relative to each other, with the central



189  $C_{60}$  moving out-of-plane while the anchors move 238  
190 in-plane (animation in Supplementary Informa- 239  
191 tion). Our findings are consistent with a pre- 240  
192 vious report that the binding energy of pyrene 241  
193 to graphene is much stronger out-of-plane com- 242  
194 pared to in-plane, such that the pyrene an- 243  
195 chors can more easily slide on the graphene elec- 244  
196 trodes.<sup>18</sup> 245

197 The Coulomb diamond edges do not intersect 246  
198 at zero bias for P1 to 3 (Figure 3a-c) and in de- 247  
199 vice C (Supplementary Figure S7b). The cur- 248  
200 rent suppression at low bias remain unchanged 249  
201 with the application of a magnetic field, ruling 250  
202 out a magnetic origin (Supplementary Figure 251  
203 S6)<sup>19</sup> and the size of the gaps correspond to an 252  
204 integer spacing of the energy excitations. We 253  
205 attribute these features to Franck-Condon (FC) 254  
206 blockade that occurs for strong electron-vibron 255  
207 coupling,<sup>5</sup> illustrated in Figure 5a. **Electrons** 256  
208 **tunnelling onto the molecule shifts the equi-** 257  
209 **librium coordinates of the harmonic oscillator** 258  
210 **wavefunctions for charge states  $N$  and  $N + 1$**  259  
211 **by an amount proportional to the dimension-** 260  
212 **less electron-vibron coupling constant  $\lambda$ . For** 261  
213 **strong coupling,  $\lambda \gg 1$ , the equilibrium coordi-** 262  
214 **ates of the  $N + 1$  charge state is greatly dis-** 263  
215 **placed from that of the  $N$  charge state (Figure** 264  
216 **5a). As a result, the transition rate between** 265  
217 **the vibronic ground states of charge states  $N$**  266  
218 **and  $N + 1$ , determined by the overlap of the** 267  
219 **vibron wavefunctions, is exponentially reduced** 268  
220 **with  $\lambda$ , leading to current suppression at low** 269  
221 **bias. The FC blockade is lifted when the ap-** 270  
222 **plied bias is sufficiently large for transitions** 271  
223 **from the vibronic ground state to higher ex-** 272  
224 **cited states (Figure 5c), where the transition** 273  
225 **rates are higher from increased overlap of the** 274  
226 **vibronic wavefunctions. When electrons are** 275  
227 **added to the molecule, we observe a decrease** 276  
228 **of the FC gap from P2 to P4, such that for** 277  
229 **P4 (Figure 3d), the FC blockade is completely** 278  
230 **lifted. The FC gap, i.e. the threshold voltage at** 279  
231 **which the FC blockade is lifted when it matches** 280  
232 **a multiple of the vibrational quantum  $n\hbar\omega$ , is** 281  
233 **proportional to  $\lambda^2\hbar\omega$ .<sup>5</sup> Additional electrons on** 282  
234 **the molecule shift the equilibrium positions of** 283  
235 **the harmonic oscillator wavefunctions, leading** 284  
236 **to a smaller displacement of the equilibrium co-** 285  
237 **ordinates between charge states  $N+2$  and  $N+3$ ,** 286

i.e. a smaller  $\lambda$ . The higher transition rate for  
the vibronic ground states from the increased  
wavefunction overlap allows for current in the  
previously FC blockade region. Redox depend-  
ent electron-vibron coupling is possibly due to  
Jahn-Teller (JT) distortion of the molecule,<sup>20</sup>  
which is particularly strong in molecules with a  
large orbital degeneracy and high spatial sym-  
metry such as  $C_{60}$ .<sup>21</sup> The geometrical JT dis-  
tortion lowers the overall ground state energy  
which is directly proportional to  $\lambda$ , as derived  
in ref. 23. Our observation of HOMO degener-  
acy and a charge state dependent  $\lambda$  is therefore  
in line with these predictions.

Using the set of evenly spaced excitations of  
P3 (Figure 3c) and fitting them to the rate  
equation model,<sup>22</sup> we determine  $\lambda = 3.0$  for  
the charge transition  $N + 1$  to  $N + 2$  (Supple-  
mentary section 1.5). We simulate the stability  
diagram for an asymmetric molecule-electrode  
tunnel coupling, dependent on the contact ge-  
ometry of the molecule to the electrodes,  $t_L =$   
 $0.2t_R$  (Figure 4b), which shows good qualitative  
agreement with our measurements (Figure 4a).  
The electroburnt graphene nano-electrodes are  
not likely to be symmetric on the atomic scale  
which can lead to asymmetric positioning of the  
molecule across the junction. Furthermore, our  
DFT calculations show that the pyrene anchors  
are not symmetric relative to the central  $C_{60}$   
(Supplementary section 2.2).

At elevated temperatures  $T$ , the thermal equi-  
librium population of vibrational excited states,  
with exponentially greater transition rates, are  
increased,<sup>3,4</sup> and transitions at low bias can now  
proceed via these states. In device A the FC  
gap is lifted at 3 K (Figure 4c,d), and device  
C at 4 K (Supplementary Figure S7b,c). We  
extract and fit the temperature dependence of  
 $I \propto 1/k_B T \times 1/(exp(\hbar\omega/k_B T) - 1)^{3,4}$  at  $V_g = 7$   
V and  $V_b = 7.2$  mV in Figure 4c and obtained  
a value of  $\hbar\omega = 1.7$  meV (Supplementary Fig-  
ure S8), accordant with the value measured in  
Figure 3c. The increase of peak intensity with  
temperature is in direct contrast with the case  
for which a Fermi-Dirac distribution is relevant,  
where a decrease is expected,  $I \propto 1/k_B T$ .<sup>3</sup>  
These observation confirms the bosonic nature  
of the transition and is consistent with our in-

287 terpretation of the FC model.

288 Low energy excitations (3 to 7 meV) were  
 289 previously observed for a pristine C<sub>60</sub> single-  
 290 molecule transistor connected to gold electro-  
 291 des.<sup>2</sup> Those excitations were attributed to  
 292 nanomechanical oscillations arising from the  
 293 center-of-mass motion of the C<sub>60</sub> and no FC  
 294 blockade was observed. This is due to the dif-  
 295 ferent coupling mechanisms for internal molec-  
 296 ular vibrations and center-of-mass oscillations.  
 297 While vibrations couple directly to the electron,  
 298 oscillations instead couple through displace-  
 299 ment dependent tunneling matrix elements.<sup>22</sup>  
 300 At energies below 10 meV, it has been shown  
 301 that the coupling strength for oscillations is in-  
 302 sufficient to induce FC blockade.<sup>4</sup> The observa-  
 303 tion of FC blockade from the 1.7 meV excitation  
 304 in our device is consistent with this interpre-  
 305 tation and our DFT calculations which showed  
 306 that the 1.7 meV mode is related to a combined  
 307 vibration of the central C<sub>60</sub> and the pyrene an-  
 308 chor groups.

309 Strong electron-vibron coupling and weak vi-  
 310 bron relaxation can further lead to avalanche  
 311 transport characterised by strong current fluctu-  
 312 ations.<sup>5</sup> These fluctuations are the result of  
 313 occasional charge transfer events exciting the  
 314 molecule from its vibrational ground state and  
 315 setting off an avalanche of electrons tunneling  
 316 via vibrational excited states that have greater  
 317 wavefunction overlap, as shown in Figure 5a-f.  
 318 An avalanche is terminated when the molecule  
 319 returns to its ground state by tunneling induced  
 320 de-excitation or dissipative relaxation. Figure  
 321 5g shows that avalanche transport, in contrast  
 322 to field-induced bi-stabilities,<sup>23</sup> leads to fluctu-  
 323 ations that persist over the entire bias range  
 324 corresponding to the vibrational 0 ↔ 2 tran-  
 325 sition. For  $\beta eV_b < 2\hbar\omega$  ( $\beta \sim 0.5$  given by  
 326 the ratio of the source capacitance and the  
 327 total capacitance), the vibrational transitions  
 328 0 ↔ 0 and 0 ↔ 1 are suppressed due to FC  
 329 blockade (a,d). At  $2\hbar\omega \leq \beta eV_b < 3\hbar\omega$ , the  
 330 0 ↔ 2 transition enters the bias window (Figure  
 331 5b,e). While the transition rate is higher, the  
 332 system typically experiences a long wait time  
 333 with no current while it remains in the vibron  
 334 ground state. For weak vibron relaxation, oc-  
 335 casional electron inelastic tunnelling excites the

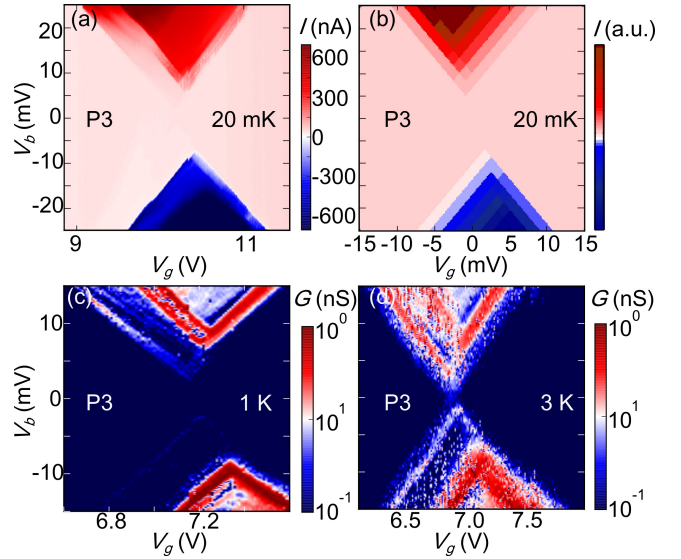


Figure 4: (a) Current stability diagram of P3 at 20 mK. (b) Simulated current stability diagram using the rate equation model with  $\lambda = 3$  and  $t_L = 0.2t_R$ , showing good qualitative agreement with (a). Conductance stability diagrams of P3 at 1 K (c) and 3 K (d). The FC gap is lifted at 3 K. Transitions due to vibron absorption are described by Bose-Einstein statistics, such that  $I_{\text{step}} \propto 1/k_B T \times 1/(exp(\hbar\omega/k_B T) - 1)$ . We extract and fit the temperature dependence of  $I$  at  $V_g = 7$  V and  $V_b = 7.2$  mV in Figure 3c and obtained a value of  $\hbar\omega = 1.7$  meV, consistent with our experimentally determined value.

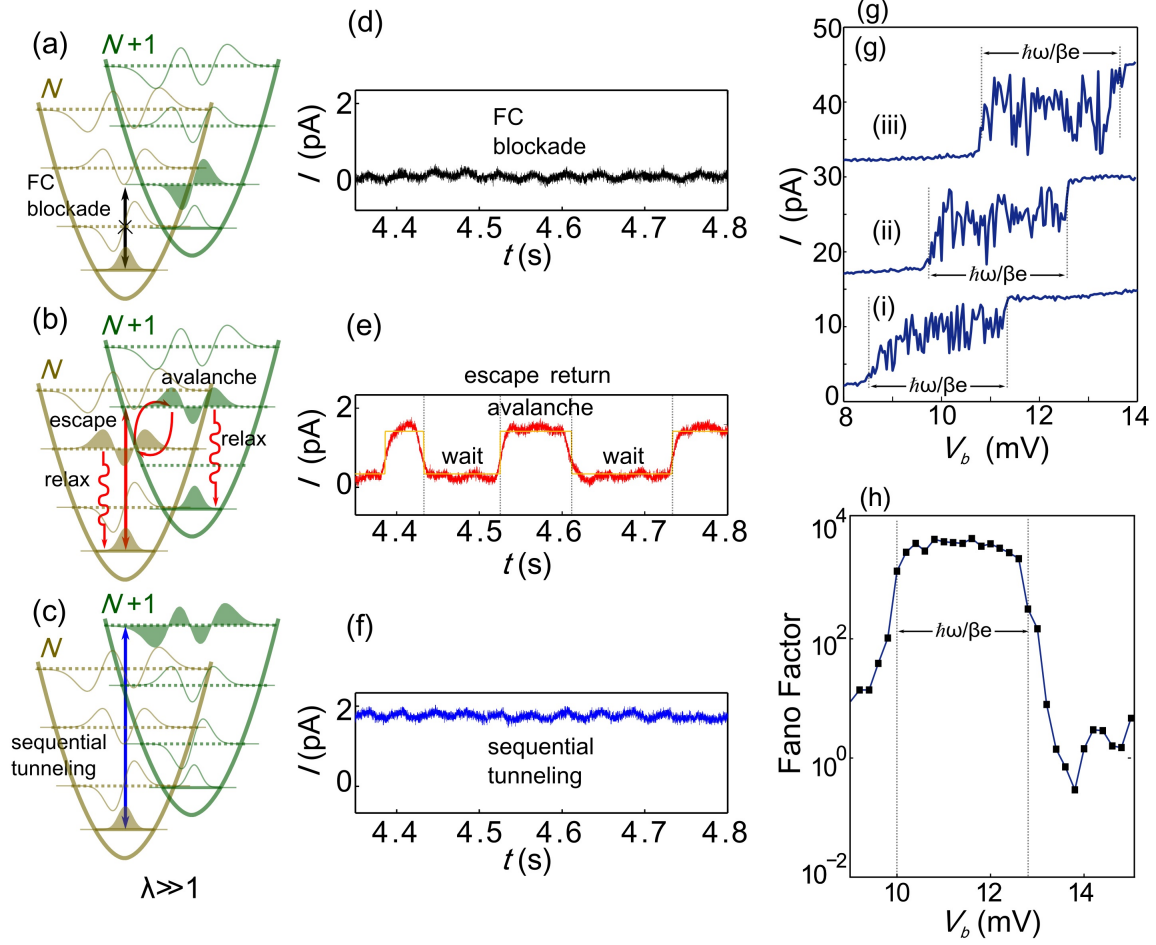


Figure 5: (a-c) Schematics for avalanche transport. (d-f) Current-time traces at  $V_b = 9, 12$  and  $14$  mV. The full current-time trace over which we perform the HMM fit was measured for 20 s with a 100 kHz sampling rate. The orange curve (e) is the HMM fit to the full current-time trace. (g)  $IV$  at  $V_g =$  (i) 10.1 V, (ii) 10.15 V and (iii) 10.2 V. Curves are offset for clarity, +15 pA (ii) and +30 pA (iii). The current noise extends over an applied bias corresponding to  $\hbar\omega/\beta e$ . (h) Fano factors of the zero-frequency noise measured as a function of  $V_b$  at  $V_g = 10.15$  V indicated by the green line in Figure 3c. Giant Fano factors  $\sim 10^3$  extend over an applied bias corresponding to  $\hbar\omega/\beta e$ , reflecting avalanche transport, before decreasing to  $\sim 1$  at higher bias in accordance with sequential tunnelling.

336 molecule to a higher vibrational state, with sub- 385  
 337 sequent tunnelling events further increasing this 386  
 338 excitation. Having escaped the ground state, 387  
 339 an avalanche of tunnelling events occur. The 388  
 340 process terminates when the system returns to 389  
 341 the vibrational ground state, resulting in an- 390  
 342 other long wait time before the cycle repeats. 391  
 343 At  $\beta eV_b \geq 3\hbar\omega$ , charge transfer proceeds via 392  
 344 sequential tunnelling through the  $0 \leftrightarrow 3$  transi- 393  
 345 tion (Figure 5c,f). 394

346 To characterise the current fluctuations we 395  
 347 extract the  $0 \leftrightarrow 2$  escape and return rates by 396  
 348 fitting a current-time trace of device A mea- 397  
 349 sured at  $V_b = 12$  mV to a Hidden Markov 398  
 350 Model (HMM) assuming a two state stochas- 399  
 351 tic process (orange line in Figure 5e).<sup>24</sup> We 400  
 352 obtain an escape rate of  $15 \pm 2$  s<sup>-1</sup>, which 401  
 353 corresponds to a FC matrix element  $W_{0 \leftrightarrow 2} \sim$  402  
 354  $0.002$  and  $\lambda \sim 4$  (Supplementary section 1.9), 403  
 355 and a return rate of  $22 \pm 2$  s<sup>-1</sup>. These rates 404  
 356 place an upper-bound of  $\sim 1$  s<sup>-1</sup> on the dis- 405  
 357 sipative relaxation rate. The observation of 406  
 358 such a slow relaxation rate, compared to val- 407  
 359 ues measured in time resolved spectroscopy ( $\sim$  408  
 360  $1$  ps<sup>-1</sup>),<sup>25</sup> is indeed surprising. However, those 409  
 361 measurements are usually performed in a sol- 410  
 362 vent where dissipation to solvent ions is possi- 411  
 363 ble. Low temperature and vacuum condi- 412  
 364 tions suppresses the dissipation rate,<sup>26</sup> which 413  
 365 is mostly dependent on dissipation to the elec- 414  
 366 trodes in single-molecule junctions.<sup>10,27</sup> Fits of 415  
 367 the conductance peaks (Supplementary section 416  
 368 1.5) show that the pyrene anchors form weak 417  
 369  $\pi$ - $\pi$  stacking interactions with the graphene 418  
 370 electrode (molecule-electrode coupling  $t_0 \sim 1$  419  
 371 meV). While the in-plane thermal conductivity 420  
 372 of graphene is exceptionally large, its out-of- 421  
 373 plane thermal conductivity is limited by weak 422  
 374 van der Waals interactions between adjacent 423  
 375 planes and substrates and can possibly suppress 424  
 376 vibron dissipation to the graphene electrodes.<sup>28</sup> 425  
 377 Vibrationally induced conformational changes 426  
 378 where an anharmonic potential with local mini- 427  
 379 mas suppress dissipation is another possibility.<sup>1</sup> 428  
 380 Metastable molecular conformation can be long 429  
 381 lived on the order of milliseconds, consistent 430  
 382 with the return rates in our system, and can be 431  
 383 accessed through a vibrational excited state.<sup>29</sup> 432  
 384 Dissipation to the electrodes is dependent on

the atomistic details of the vibron density of 385  
 states at the molecule-electrode interface. The 386  
 non-equilibrium distribution of vibrons is non- 387  
 trivial and requires a self-consistent calculation 388  
 of electron-vibron dynamics. 389

Electron avalanche transport is characterized 390  
 by giant Fano factors on the order of  $10^2$ - $10^3$  oc- 391  
 ccurring in steps over the bias window ( $\hbar\omega/\beta e$ ) 392  
 corresponding to a vibron transition.<sup>5</sup> The zero 393  
 frequency Fano factor  $F = S/2e\langle I \rangle$  for the 394  
 excess noise  $S(0) - S(0)_{V_b=0}$  indicates the de- 395  
 viation from Poissonian noise, where  $F = 1$ . 396  
 The noise power density  $S(\omega)$  is defined as the 397  
 Fourier transform of the current-current correla- 398  
 tion, 399

$$S(\omega) = \int_{-\infty}^{\infty} dt e^{i\omega t} \langle \delta I(t+t') \delta I(t') \rangle_{t'}. \quad (1)$$

In Figure 5h, we determine the zero frequency 400  
 Fano factors as a function of  $V_b$  (Supplementary 401  
 section 1.9), taken at  $V_g = 10.15$  V indicated 402  
 by the vertical green line in Figure 3c. At  $10$  403  
 $\leq V_b \leq 13$  mV, the Fano factors increase to an 404  
 order of  $\sim 10^2 - 10^3$ , reflecting the current fluctu- 405  
 ations due to avalanche transport. These giant 406  
 Fano factors likewise persist over an applied 407  
 bias corresponding to  $\hbar\omega$ . When  $V_b$  is sufficient 408  
 to lift FC blockade, the Fano factor is expected 409  
 to return to values of the order of 1.<sup>5</sup> At  $V_b =$  410  
 $13.4$  mV, the Fano factors decrease to  $\sim 1$  in 411  
 accordance with sequential transport.<sup>30</sup> 412

We have demonstrated a robust architecture 413  
 using functionalised C<sub>60</sub> bisadduct and 414  
 graphene electrodes to create single-molecule 415  
 transistors which enables the observation of 416  
 redox-dependent FC blockade and avalanche 417  
 transport. While doping-induced variations 418  
 in the Fermi-level of the graphene electrodes 419  
 and offset charges in the oxide can influence 420  
 the alignment of the electrochemical potential 421  
 of the molecule with the Fermi-level of the 422  
 graphene electrodes, we show that the function- 423  
 alisation of C<sub>60</sub> with pyrene anchors groups can 424  
 modify electron delocalization and energy levels 425  
 leading to HOMO dominated transport, in 426  
 contrast to previous reports of LUMO domi- 427



428 nated transport in  $C_{60}$  transistors.<sup>10</sup> Calcula- 471  
429 tions have suggested that the modification of 472  
430 electron delocalization and energies via the in- 473  
431 clusion of side group can be used to tune the 474  
432 thermopower of single-molecule junctions.<sup>31,32</sup> 475  
433 Furthermore, thermoelectric efficiency is max- 476  
434 imised when charge transport across a junction 477  
435 occurs through a single level with low vibra- 478  
436 tional heat conductance.<sup>32</sup> Therefore, the ob- 479  
437 servation of an extremely long vibron lifetime 480  
438 not only offers unique potential for the funda- 481  
439 mental study of quantized motion and thermal 482  
440 transport on the nanoscale, but is also promis- 483  
441 ing for practical applications in single-molecule 484  
442 energy-conversion devices.<sup>32</sup> 485

## 443 Methods

### 444 Device Fabrication and Measure- 490 445 ments

446 The molecular devices are fabricated on a heav- 492  
447 ily doped silicon chip with a 300 nm thick oxide 493  
448 which also serves as a back gate to modulate 494  
449 charge transport through the junction. First, 495  
450 graphene is synthesized on liquid copper sup- 496  
451 ported on a tungsten substrate using chemi- 497  
452 cal vapour deposition.<sup>33</sup> Using a wet transfer 498  
453 technique, the graphene is transferred to the 499  
454 silicon substrate with pre-patterned gold elec- 500  
455 trodes. The graphene electrodes are fabricated 501  
456 based on the method of feedback-controlled 502  
457 electroburning.<sup>34</sup> Fabricated nanogaps are typ- 503  
458 ically between 1 to 2 nm. The devices are next 504  
459 immersed in a 10  $\mu$ M chloroform solution con- 505  
460 taining the  $C_{60}$  bisadducts for 30 mins. We 506  
461 wirebond the devices to our sample puck and 507  
462 transfer them to our dilution fridge with a base 508  
463 temperature of 20 mK. All electrical measure-  
464 ments are performed using low noise, battery  
465 operated electronics, while the gate voltage was  
466 modulated with a Keithley 2400 sourcemeter.

### 467 DFT theoretical methods 509

468 To calculate the vibrational modes of the 511  
469 bisadducts, we use the Harmonic approxima- 512  
470 tion method to construct the Dynamical Ma- 513

trix  $D$ . The  $xyz$  coordinate of the two isomers of  
the bisadducts were constructed and the geome-  
try relaxation were performed with Siesta<sup>35</sup> im-  
plementation of the Density Functional Theory  
(DFT) by double- $\zeta$  polarized bases set (DZP)  
and the GGA functional with PBE parameter-  
ization to the force tolerance of 20 meV/Å. A  
real-space grid is defined with an equivalent en-  
ergy cut-off of 250 Ry. From the relaxed  $xyz$   
coordinates of the system, a set of the  $xyz$  coor-  
dinates were generated by displacing each atom  
in positive and negative  $x$ ,  $y$  and  $z$  directions  
by  $\delta q' = 0.01$  Å. The forces in three direc-  
tions  $q_i = (x_i, y_i, z_i)$  on each atom were then  
calculated by DFT with the same parameters as  
the relaxed system but without geometry relax-  
ation. These set of the force  $F_i^q = (F_i^x, F_i^y, F_i^z)$   
vectors are used to construct the Dynamical  
matrix (Supplementary section 2.1). The elec-  
tronic structure and transport calculations are  
performed using GOLLUM as described in.<sup>8</sup>

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