A method to quantitatively evaluate Hamaker constant using the jump-into-contact effect in Atomic Force microscopy

Soma Das, P. A. Sreeram and A. K. Raychaudhuri

DST Unit for Nanoscience.

S. N. Bose National Centre for Basic Sciences, Block JD, Sector III, Kolkata 700098, India.

E-mail: soma@bose.res.in, sreeram@bose.res.in, arup@bose.res.in

Abstract. We find that the "jump-into-contact" of the cantilever in the Atomic Force Microscope (AFM) is caused by an inherent instability in the motion of the AFM cantilever. The analysis is based on a simple model of the cantilever moving in a non-linear force field. We show that the "jump-into-contact" distance can be used to find the interaction of the cantilever tip with the surface. In the specific context of the attractive van-der-Waals interaction, this method can be realized as a new method of measuring the Hamaker constant for materials. The Hamaker constant is determined from the deflection of the cantilever at the "jump-into-contact" using the force constant of the cantilever and the tip radius of curvature, all of which can be obtained by measurements. The results have been verified experimentally on a sample of cleaved mica, a sample of Si wafer with natural oxide and a silver film, using a number of cantilevers of different spring constants. We emphasize that the method described here is applicable only to surfaces that have van-der-Waals interaction as the tip-sample interaction. We also find that the tip to sample separation at the "jump-into-contact" is simply related to the cantilever deflection at this point and this provides a method to exactly locate the surface.

PACS numbers: 68.37.Ps, 34.20.Gj, 34.50.Dy

1. Introduction

Atomic Force Microscope (AFM) is one of the most widely used tools in nanoscience and nanotechnology. Since its discovery, AFM [1] has emerged as a very powerful tool in the characterization of various properties of materials at the nanometer scale. This is primarily because AFM can not only image with atomic resolution but it can also measure interatomic forces which are of the order of pico Newtons or even much less. These capabilities made AFM a versatile enabling tool in nanotechnology. One of the standard experiments performed by AFM is the measurement of the force-distance curves [2, 3] which measures the force of interaction between the tip and the substrate. In this measurement, the cantilever deflection (d) is measured as a function of the separation of the tip and the sample (z) and the force of interaction is the product of the deflection 'd' of the cantilever and the spring constant ' k_c ' of the cantilever. (Note: The force obtained in this manner is not exactly the force between the tip and the sample, since the effective spring constant of the cantilever can be modified by the elastic deformation of the surface of the sample and the tip when they are in contact with each other. Hence, in our study, we will consistently use the concept of deflection of the cantilever instead of the force.). In the measurement of the force-distance curve, d is measured from its equilibrium position (in the absence of any external force), when it is at a distance 'h' from the sample (the substrate) as shown in figure 1a. The measured force-distance curve, shown schematically in figure 1b generally shows

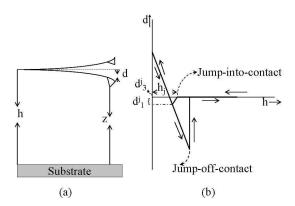


Figure 1. Schematic diagram of AFM tip and sample assembly (a) and force-distance curves (b). The dotted line in (a) marks the equilibrium position of the cantilever in absence of an external force. "d" is positive when measured upward. The arrows in (b) show the direction of motion of the cantilever.

hysteresis. The approach curve shows a "jump-into-contact" (JC) and the retraction part of the curve shows the "jump-off-contact". The concept of "jump-off-contact" has been used extensively in the past as a quantitative measure of the adhesion force [4, 5]. One of the most ubiquitous explanations for the "hysteresis-like" behaviour observed in the force-distance curves in AFM is based on the presence of adhesion forces, due to a layer of water on the surface of the sample and the tip [6]. In contrast, however not much

attention has been paid to the phenomena of "jump-into-contact" (JC), except early papers that pointed out the basic causes for existence of such a phenomena [2, 7]. In this paper, we revisit the issue of "jump-into-contact" again and present a new approach to "jump-into-contact" phenomenon in the force-distance curves of AFM. This is done in order to investigate whether it can be used to obtain quantitatively some of the microscopic parameters of tip-sample interaction and thus can be made an useful tool. We find that JC is a generic manifestation of the fundamental instability in the motion of the cantilever in a non-linear force field. A simple model is used to understand this instability and obtain a quantitative measure of not only the distance ' h_i ' at which the JC should occur but also how much should be the magnitude of the deflection of the cantilever at the JC. These measures are directly related to parameters of the force field. We have performed experiments to verify some of the predictions of our theory. In this paper we investigate this phenomena in the specific context of the van-der-Waals interaction and from the measured deflection of the cantilever at the JC we determined the Hamaker constant using the known parameters like the radius of curvature of the tip (R_t) and the cantilever spring constant (k_c) . In addition, our investigation also gives us a practical way to locate the actual distance of the cantilever from the surface. The phenomena is completely governed by the attractive part of the tip-sample interaction. An important outcome of the investigation is the observation of the independence of the JC on the actual elastic forces that make the tip-sample contact interaction. Since the JC can be measured at different spots on a given surface, this method gives us a tool to obtain a map of the tip-surface interaction as measured by parameter like Hamaker constant (in the context of van-der-Waals interaction) with a spatial resolution offered by a typical AFM. The spatial resolution is an added advantage over other methods of determining the Hamaker constant like the surface force apparatus.

The paper is organized as follows. In Section 2, we introduce our simple spring-ball model for the motion of the cantilever. We solve the static force equation analytically to locate the instability that makes the JC to happen. In Section 3, we show experimental data of force-distance curves to measure the JC distance, experimentally verify some of the theoretical predictions and obtain the Hamaker constant. We also discuss the extent of uncertainty in the data and compare the relative merits of this method vis-a-vis other methods of determining Hamaker constant. In section 4, we present our conclusions.

2. Theoretical Modelling

The AFM is a nonlinear system. Our aim here is to use a simple model which could explain the feature seen in experiments. We model the motion of a cantilever by a spring-ball system. The inherent nonlinearity of the cantilever due to its finite dimensions have not been introduced into our calculation, in order to keep things simple. Thus, we write the equation of motion of the cantilever as,

$$m\ddot{d}(t) + \eta \dot{d}(t) + kd(t) = f_{ts}(h + d(t)). \tag{1}$$

Here, m is the mass of the cantilever, η is the friction constant, k_c is the spring constant, d(t) is the deflection of the cantilever measured from its equilibrium position in the absence of any external force, h is the distance between the sample and the tip when the tip is in the equilibrium position (in the absence of any external force) and $f_{ts}(h+d)$ is the atomic force between the tip and the sample at the instantaneous position of the tip and t represents time. In case of the static (or quasi-equilibrium) experiment d(t) = d, where d is the deflection of the cantilever at which it comes to rest. The dynamic equation will reduce to a simple static equation of the form,

$$k_c d = f_{ts}(h+d), (2)$$

One can take a generalized force field for $f_{ts}(h+d)$ and obtain a solution to the Equation (2) that will give the parameters of the interaction potential. In order to have a definite result that can be verified by experiment, we investigated the specific case of van-der-Waals interaction between the tip and the surface and an elastically deformable surface for the contact force. The subsequent results obtained are thus specific to the van-der-Waals interactions. The tip-sample force is modelled by a combination of van-der Waals force at large tip-sample distances (h) which is essentially attractive and by the Dejarguin-Muller-Toporov(DMT) [8, 9] force which is a combination of the attractive van-der-Waals-like force (except that it is h-independent) and the repulsive forces arising due to elastic interaction between the tip and the sample. Thus, formally, the force is given by,

$$f_{ts}(z) = \begin{cases} -\frac{HR_t}{6z^2} & \text{for } z > a_0, \\ -\frac{HR_t}{6a_0^2} + \frac{4}{3}E^*\sqrt{R_t}(a_0 - z)^{3/2} & \text{for } z \le a_0. \end{cases}$$
(3)

Here, z = h + d, a_0 is an intermolecular distance, H is the Hamaker constant, which depends on the material of the tip and the sample and also on the intervening medium. E^* is the effective elastic modulus between the tip and the sample. Note that the form of the van-der-Waals force is chosen for a sphere-plate geometry, which is close to the real situation in an AFM experiment. In this paper, we will only concentrate on the regime where $z > a_0$, where the force is purely a van-der-Waals force. We will see below that the JC is mainly determined by the attractive part of the interaction.

For the observation of JC, we work in the region of attractive interaction and take the force on the right hand side of Equation (2) to be the van-der-Waals force. This is justified because, we will see below that the "jump-into-contact" distance is usually much larger than a_0 . From Equations (2) and (3) after some simple manipulations, we obtain the equation for the deflection (d) as,

$$k_c d(d+h)^2 + \frac{HR_t}{6} = 0. (4)$$

Rewriting $\tilde{d} = d/h$ and $\tilde{a} = HR_t/6k_ch^3$, we get,

$$\tilde{d}(1+\tilde{d})^2 + \tilde{a} = 0. \tag{5}$$

The three solutions of this equation are given by,

$$\tilde{d}_{1} = -\frac{b_{2}}{3} + (S+T)$$

$$\tilde{d}_{2} = -\frac{b_{2}}{3} - \frac{1}{2}(S+T) + \frac{\sqrt{3}}{2}i(S-T)$$

$$\tilde{d}_{3} = -\frac{b_{2}}{3} - \frac{1}{2}(S+T) - \frac{\sqrt{3}}{2}i(S-T)$$
(6)

where,

$$S = \left(R + \sqrt{D}\right)^{1/3}$$

$$T = \left(R - \sqrt{D}\right)^{1/3}$$

$$R = \frac{9b_1b_2 - 27b_0 - 2b_2^3}{54}$$

$$D = Q^3 + R^2$$

$$Q = \frac{3b_1 - b_2^2}{9}$$
(7)

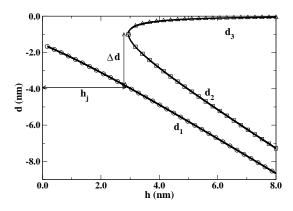


Figure 2. Plot of solutions given by Equation (6) as a function of tip-sample distance (h) for the parameters mentioned in the text. The open circles $(d_1 = \tilde{d}_1.h)$ and open triangles $(d_3 = \tilde{d}_3.h)$ represent stable solutions. The open squares $(d_2 = \tilde{d}_2.h)$ represent the unstable solution. Here only the real part of the solutions in shown.

and b_0 , b_1 and b_2 are the coefficients of $(\tilde{d})^0$, $(\tilde{d})^1$ and $(\tilde{d})^2$ in Equation (5). Here we want to mention that Equation (6) is valid only for the real values of S and T defined in Equation (6). For complex values, the expressions for Equation (6) will change. It can be easily seen that, $b_0 = \tilde{a}$, $b_1 = 1$ and $b_2 = 2$. The distance \tilde{d}_1 has only a real part, while the solutions \tilde{d}_2 and \tilde{d}_3 are either both real or complex conjugate of each other, depending on the parameters of the equation. The actual deflection (d) is obtained by multiplying the solution by the corresponding tip-sample distance (h). Figure 2 shows the solutions of the actual deflection (d) as a function of the tip-sample distance (h) for $HR_t=2.26 \times 10^{-27} \text{ N.m}^2$ and $a_0=0.15 \text{ nm}$. Of the three solutions, the solution given by the open circles corresponds to $d_1 = \tilde{d}_1.h$ while the open square and triangle

correspond to $d_2 = \tilde{d}_2.h$ and $d_3 = \tilde{d}_3.h$ respectively. Note that as the tip-sample distance is reduced, the solutions corresponding to d_2 and d_3 approach each other and they meet at one point (for example, at $h \approx 2.9$ nm in figure 2). For tip sample distances below this both these solutions become complex (in figure 2 the real part only is shown). It is necessary to note that the solutions d_1 and d_3 are stable solutions, while d_2 is unstable. This has been checked by finding the sign of the derivative of Equation (2) with respect to d at each value of h. We denote the point where the solutions d_2 and d_3 meet as the "jump-into-contact" point. This is the limit of stability for the solution d_3 which defines the motion of the cantilever for the approach curve till this point. If the tip-sample distance (h) is reduced beyond this point of stability, there is only one real solution available (d_1) and the system will jump into the stable solution given by d_1 . This defines the "jump-into-contact". It must be noted here that this jump has occurred in the attractive regime and we do not take recourse to any adhesion forces for explaining the phenomena. We also emphasize here that on the retract path the cantilever dynamics follow the solution given by d_1 until it jumps back to the solution given by d_3 at the "jump-off-contact" point. The solutions of the cubic equation given by Equation (6) have a number of interesting features. For example, let us consider the point where the "jump-into-contact" occurs in our model. At this point $d_2 = d_3$ and the discriminant D is exactly equal to zero. If we denote the tip-sample distance at this point by ' h_j ' and corresponding values of R, Q as R_j , Q_j , we get the equation,

$$R_j + \sqrt{Q_j^3 + R_j^2} = R_j - \sqrt{Q_j^3 + R_j^2}$$
 (8)

which leads to the equation,

$$Q_i^3 = -R_i^2 \tag{9}$$

Replacing the expressions for Q_j and R_j from Equation (7), and putting in the values of b_0 , b_1 and b_2 , we get,

$$\frac{HR_t}{6k_ch_j^3} = \frac{4}{27} \tag{10}$$

Equation (10) can be used to find the Hamaker constant (H) because the tip radius R_t and the cantilever spring constant k_c are known and h_j is experimentally measureable. However, the problem arises because the postion of the surface being not known exactly, the absolute value of h_j has a large uncertainty. Below we show that the magnitude of the jump of the cantilever at JC is simply related to h_j and this fact can be used to calculate the Hamaker constant (H) with high degree of confidence which is limited by the magnitude of the uncertainty in determination of k_c and R_t both of which, however, are experimentally measurable [10, 11].

At the JC, there are only two distinct real solutions to the cubic equation since the solutions corresponding to d_2 and d_3 are degenerate. Subtracting d_3 from d_1 , and again putting in the values of b_0 , b_1 and b_2 , we get the jump of the cantilever (Δd) at JC as,

$$\Delta d = d_3^j - d_1^j = -h_j \tag{11}$$

where d_1^j and d_3^j are the deflections at the "jump-into-contact" point corresponding to the two solutions. Equations (10) and (11) lead to a practical way of calculating the Hamaker constant from the deflection-displacement curves. Determination of Hamaker constant from equations (10) and (11) will need knowledge of k_c and R_t of a given cantilever which can be obtained from experiment. Alternatively, we note that if h_j is measured for a material with known Hamaker constant, this can be used to calibrate (k_c/R_t) ratio of a given cantilever, which in turn can be used to find an unknown Hamaker constant. Given the practical difficulties in knowing k_c and R_t exactly this may be a more practical method. Note also that Equation (11) is itself independent of the material of the tip, sample and the intervening medium. The above mentioned process also indicates that one can obtain a precise method of shifting the raw data obtained from the AFM measurements to properly locate the surface.

3. Experimental verification

The data has been taken using an Atomic force microscope (Model CP II) from Veeco [12] on freshly cleaved mica, on Si wafer with natural oxide and on a silver metal film. We have used three different cantilevers for taking data for a given surface in order to vary (k_c/R_t) ratio. The cantilevers used had Si₃N₄ tips and spring constants (radius of curvature) of 0.03 N/m (R_t = 30 nm), 0.1 N/m (R_t = 35 nm) and 0.9 N/m (R_t=50 nm). We have found out the radius of curvature of the tip from the images taken by a Field Emission Gun Scanning Electron Microscope (FEG-SEM). The medium between the tip and sample for all the experiments was air at room temperature and the rates of data collection were 0.5 Hz and 0.1 Hz. We have repeatedly taken the force-distance curves using the same three cantilevers mentioned above. The reproducibility of the data confirms that there was no damage of the tip of the cantilever during the experiment and this is also corroborated by FEG-SEM image.

Figure 3 shows a typical AFM deflection (d) versus displacement (h) curve for a freshly cleaved Mica sample and a Si₃N₄ tip of spring constant 0.1 N/m in air. The data has been plotted as deflection versus distance. The arrows in figure 3 indicate the direction of motion of the cantilever (approach and retract). The JC region is highlighted in the inset of figure 3. From figure 3 we can see that the "jump-into-contact" occurs at tip-sample separation (h) of approximately 2.9 nm which is the attractive regime of the force distance curves, since $a_0 \approx 0.15$ nm [8, 9]. The "jump-into-contact" occurs at larger values of h for smaller k_c . For instance, for $k_c \approx 0.03$ N/m, the jump occurs at $h_j \approx 3.5$ nm. Thus the parameters of the attractive potential are good enough to determine the JC.

In figure 4 we plot the observed h_j^{-3} (h_j is jump into contact distance) as a function of the quantity (k_c/R_t) ratio. The ratio (k_c/R_t) are the physical parameters of the cantilevers used. We have taken three cantilevers of same composition but with different k_c and R_t to achieve three different (k_c/R_t) ratio for a given surface material. The main uncertainty in determination of H from the experiment arises from these two parameters.

A new method ...

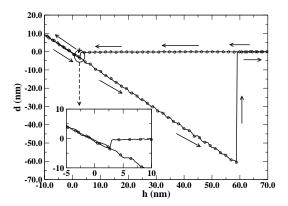


Figure 3. Approach and retract curves of the deflection (d) versus the displacement (h) of the tip of the microcantilever for Mica using a Si₃N₄ tip (free motion spring constant k=0.1 N/m) in air. The arrows indicate the direction of motion of the cantilever (approach and retract). The JC region is highlighted in the inset.

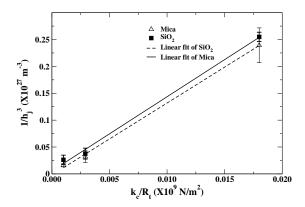


Figure 4. Plot of the observed jump into contact distance h_j^{-3} as a function of the quantity k_c/R_t . The solid line and the dashed line are the best fits of the experimental data on mica and Si wafer with natural oxide respectively.

One can use the parameters given by the manufacturer's data but a better alternative is to experimentally determine them. From Equation (10), it can be seen that since the Δd or $h_j \propto k_c^{-1/3}$, thus it is advisable to use a softer cantilever (low k_c) so that Δd or h_j are larger leading to less uncertainty in determination of these quantities. The graph according to Equations (10) and (11) give a straight line and the inverse of the slope gives the Hamaker constant H. The data have been taken in ambient within a glove chamber. The error bar in the data have been obtained by repeated data taking on the same surface and with the same tip and it shows the extent of variance one would expect in such experiments. The reproducibility of the data also indicates that the tip, used in the experiment, did not get damaged during the collection of the data. From our experiment we obtain $H \approx (0.64 \pm 0.07) \times 10^{-19}$ J for mica, $H \approx (0.66 \pm 0.27) \times 10^{-19}$ J for SiO₂ and $H \approx (3.73 \pm 0.89) \times 10^{-19}$ J for silver. A summary of results obtained is shown in Table 1. This can be compared with calculated values of 1.28 ×

 $10^{-19}~{
m J}$ for mica [13], $1.21~{
m \times}~10^{-19}~{
m J}$ for SiO₂ [13] and $2.9~{
m \times}~10^{-19}~{
m J}$ for silver [14], using Si₃N₄ as the tip material with vacuum as the intervening medium . A similar calculation with water as the intervening medium gives 0.245×10^{-19} J for Mica [13], $0.207 \times 10^{-19} \text{ J for SiO}_2$ [13] and $1.39 \times 10^{-19} \text{ J for silver}$ [15] with Si₃N₄ as the tip material. These values have been obtained from calculations using full Lifshitz theory [16, 17, 18] for the individual materials. The values we have obtained experimentally lie between the Hamaker constant values for vacuum as the intervening medium and water as the intervening medium, suggesting the influence of relative humidity of air in calculating the Hamaker constant. The relative humidity in the glove chamber during measurement was typically $\sim 55\%$ (for mica and SiO₂) and $\sim 33\%$ for silver. Since the JC data are routinely obtained while one takes the force-distance curve it is easy to obtain a very important physical parameter from the same experiment. We point out that the utility of JC data to obtain a quantitative measure of the physical parameter like the Hamaker constant is novel and the uncertainty in determination of h_j can be eliminated by measurement of Δd and the (k_c/R_t) ratio. As mentioned before, a known tip-surface system (known Hamaker constant) can be used to calibrate a given cantilever (k_c/R_t) ratio using Equations (10) and (11) if no direct measurement of k_c and R_t are available. This calibration can also be used to find Hamaker constant for unknown surface. We also emphasize that since for a given cantilever, (k_c/R_t) ratio is fixed a map of JC on an inhomogeneous surface can generate a map of Hamaker constant.

There are quite a few methods of measuring Hamaker constant [19]. These methods include direct force measurements using surface force apparatus [20] and atomic force microscope [3, 21, 22, 23] where the full force-distance curve is fitted to a model of the van-der-Waals equation. The other methods are based on measuring physical properties of materials like dielectric constant [24, 25]. In general, Lifshitz theory [16, 17, 18] is widely used for calculating Hamaker constants from the dielectric constant of materials. We note that, the reported values of Hamaker constants for the same material obtained from different methods showed considerable variations [19]. The earlier works in obtaining Hamaker constant from Atomic force microscope were mainly based on fitting the attractive part of the approach curve with the expression for the van der-Waals force. In that method the main problem was the presence of the "jump-into-contact". In our method we have actually used the "jump-into-contact" to find the interaction constant.

The advantage of measuring Hamaker constant using the method described in this paper is that we can find the "jump-into-contact" distance from experimental force-distance curves using Equation (11) easily. No numerical fit to the complete force-distance curve is necessary. Here we also want to emphasize that this method has the advantage of mapping the Hamaker constant in an inhomogeneous system whereas it is not possible using surface force apparatus because it does not have the spatial resolution. Two important points have to be noted in this context - first, the experimental force-distance curves should be taken properly for approach of the cantilever motion with close measurements near the JC and second, the radius of the tip (R_t) and the spring

Materials	Atomic Force Microscope	Surface Force Apparatus	Full Lifshitz Theory
	(using "jump-into-contact") (10^{-19} J)	(10^{-19} J)	(10^{-19} J)
Mica	0.65 ± 0.07	1.35, 0.22 [19]	1.28, 0.245 [13]
	$(air, R_H \sim 55\%)$	(vacuum), (water)	(vacuum), (water)
$Si0_2$	0.62 ± 0.27	50-60 [19]	1.21, 0.207 [13]
	$(air, R_H \sim 55\%)$	(vacuum)	(vacuum), (water)
Silver	3.73 ± 0.89	2.9 [14]	2.4 [26], 1.39 [15]
	$(air, R_H \sim 33\%)$	(air)	(vacuum), (water)

Table 1. Values of Hamaker Constant obtained by our method and its comparison with Lifshitz theory. We also show experimental values obtained by surface force apparatus.

constant of the cantilever (k_c) have to be found out with least uncertainty if absolute data have to be obtained.

We point out that Equation (11) is also a very important outcome of this work. This gives us a way to determine the actual position of the surface. In AFM measurements there is indeed a problem in evaluating the absolute value of the tip-sample separation (h). The JC is a special point at which the distance (h) is equal to $|\Delta d|$, which thus can be appropriately fixed. Once this is fixed the position of origin of h (the sample) can be located. We emphasize that the analysis above is applicable only to surfaces that have van-der-Waals interaction as the tip-sample interaction because of the specific type of tip-sample interaction used. However, the method is general enough and can be used with other tip-sample interaction as well. The fact that nonlinear force field introduces an instability that leads to "jump-into-contact" is a conclusion of general validity.

4. Conclusions

We have studied the static deflection-distance curves for an atomic force microscope using a simple model that gives "jump-into-contact" often observed in force-distance curves of AFM as an instability of the cantilever moving in a non-linear force field. The model provides a unique method of determining the tip-sample interaction parameters. We have developed the concept specially for van-der-Waals interaction for definiteness. In this case the method gives the Hamaker constant. We find values that are comparable to the Hamaker constant measured by other methods. The model also provides a reliable criterion for locating the sample and thus shifting the raw deflection-distance data obtained from AFM by locating the distance at which the "jump-into-contact" occurs. This process removes the arbitrariness of locating the sample in AFM. The method also gives us a way to map the Hamaker constant over a surface, that may be inhomogeneous, by mapping the JC with AFM.

Acknowledgments

The authors want to thank the Department of Science and Technology, Government of India for financial support as a Unit for Nanoscience. One of the authors, (PAS) would like to thank Prof. B. Dutta Roy for very useful suggestions.

References

- [1] Binning G, Quate C F and Gerber Ch 1986 Phys. Rev. Lett. 56 930.
- [2] Cappella B and Dietler G 1999 Surface Science Reports 34 1.
- [3] Butt H J, Cappella B and Kappl M 2005 Surface Science Reports 59 1.
- [4] Cappella B, Baschieri P, Frediani C, Miccoli P and Ascoli C 1997 Engineering in Medicine and Biology Magazine IEEE 16 58.
- [5] Hao Huang Wen, Baró A M and Sáenz J J 1991 Jour. Vac. Sci. Tech. 9 1323.
- [6] Sarid D, Hunt J P, Workman R K, Yao X and Peterson C A 1998 Appl. Phys. A 66 S283.
- [7] Burnham Nancy A and Colton Richard J 1989 Jour. Vac. Sci. Tech. A 7 2906.
- [8] Israelachvili J 1991 Intermolecular and Surface Forces (Academic, Orlando, 1991).
- [9] Derjaguin B V, Muller V M and Toporov Y P 1975 Jour. Colloid Interface Sci. 53 314.
- [10] S.M. Cook, T.E. Schäffer, K.M. Chynoweth, M. Wigton, R.W. Simmonds and K.M. Lang 2006 Nanotechnology 17 2135.
- [11] Charles A Clifford and Martin P Seah 2005 Nanotechnology 16 1666.
- [12] Veeco Instruments Inc. Corporate Headquarters 100 Sunnyside Blvd. Ste. B Woodbury New York 11797-2902.
- [13] Bergström Lennart 1997 Adv. Coll. Int. Sc. **70** 125.
- [14] J. L. Parker 1992 Langmuir 8 551.
- [15] J. Visser, 1976 Adv. Col. Int. Sc. 3 331.
- [16] Lifshitz E M 1956 Sov. Phys. JETP 2 73.
- [17] Dzyaloshinskii I E, Lifshitz E M and Pitaevskii L P 1961 Adv. Phys. 10 165.
- [18] Ninham B W and Parsegian V A 1970 Jour. Chem. Phys. 52 4578.
- [19] Ackler Harold D, French Roger H and Chiang Yet-Ming 1996 Journal of Colloid and Interface Science 179 460.
- [20] Horn R G, Clarke D R and Clarkson M T 1988 Jour. Mat. Res. 3 413.
- [21] Burnham Nancy A, Dominguez Dawn D, Mowery Robert L and Colton Richard J 1990 Phys. Rev. Lett. 64 1931.
- [22] Ederth T 2001 Langmuir 17 3329.
- [23] Seog J, Dean D, Plaas A H K, Wong-Palms S, Grodzinsky A J and Ortiz C 2002 Macromolecules 35 5601.
- [24] Tabor F R S D and Winterton R H S 1969 Proc. Roy. Soc. 312 435.
- [25] Hough D B and White L R 1980 Adv. Coll. Inter. Sci. 14 3.
- [26] C J Coakley and D Tabor 1978 J. Phys. D: Appl. Phys. 11 L77