

# Evaluation of Self-Field Distributions for Bi2223 Tapes with Oxide Barriers Carrying DC Transport Current

Tomohide Makihara, Ryoji Inada, Akio Oota, Shusaku Sakamoto, Chengshan Li, and Pingxiang Zhang

**Abstract**—For AC loss reduction in Bi2223 tapes subjected to an AC external magnetic field, twisting the superconducting filaments and/or introducing oxide layers as highly resistive barriers around each filament are required. However, the structure of barrier tape becomes very complex and longitudinal uniformity of both transport property and tape structure could be easily deteriorated. To improve the current transport capability and its longitudinal uniformity for barrier tape for AC use, simple and non-destructive techniques to characterize them should be indispensable. In this paper, we examined the self-field distributions for Bi2223 tapes with oxide barriers carrying DC transport current by a scanning Hall-probe microscopy (SHM). Non-twisted and twisted 19-filamentary tapes with SrZrO<sub>3</sub> + Bi2212 barriers were prepared by powder-in-tube process. The distributions of self-field in perpendicular to the broader face of a tape carrying DC current below critical current  $I_c$  was measured at 77 K by SHM with an active area of 50  $\mu\text{m} \times 50 \mu\text{m}$ , at 0.5 mm away from the tape surface. Based on the measured self-field distributions, longitudinal non-uniformity of transport properties and the presence of local defects to obstruct the current transport in barrier tapes were investigated.

**Index Terms**—Bi2223 tapes, oxide barriers, transport current, self-field, Hall-probe

## I. INTRODUCTION

REDUCTION of AC losses in silver-sheathed Bi2223 tapes is one of the key issues for practical AC power devices such as transmission cables, motors and transformers. The large loss generation in an AC external magnetic field is attributed to strong electromagnetic coupling among the superconducting filaments. This interfilamentary coupling is caused by the low resistivity of pure silver matrix and/or direct connection between the filaments through interfilamentary bridging [1]. Particularly, due to geometrical anisotropy of

Bi2223 tape, both hysteresis loss of superconductor and coupling loss of matrix part in a perpendicular field become much higher than in a parallel field and the conditions for filament decoupling becomes more restrictive [2], [3]. In order to achieve significant loss reduction in an AC external field with various directions against broader face of tape, therefore, not only twisting the filaments with a proper pitch length but also introducing oxide layers among the filaments as highly resistive barriers should be indispensable [4]–[8].

To date, several research works have been reported for the fabrication and characterization of Bi2223 tapes with interfilamentary oxide barriers, and noticeable loss reduction in a perpendicular field with power-grid frequency range has been already demonstrated [5]–[8]. However, the structure of a barrier tape becomes very complex and the longitudinal uniformity of both transport property and tape structure could be easily deteriorated. To improve the current transport capability and its longitudinal uniformity for barrier tape with low AC losses, simple and non-destructive techniques to characterize the locally degraded part in a tape should be strongly required.

Scanning Hall-probe Microscopy (SHM) is known to be a useful tool for magnetic and non-destructive evaluation of a variety of structural materials such as stainless steel and low carbon steel utilized in industry [9], [10]. Moreover, SHM has been also applied to characterize magnetic field distributions for Bi2223 mono- or multifilamentary tapes [11]–[15] or coated conductors [16], [17]. In this paper, we examined the self-field distributions for Bi2223 tapes with oxide barriers carrying DC transport current by SHM. Based on the experimental results, longitudinal non-uniformity of transport properties and the presence of local defects to obstruct the current transport in barrier tapes were investigated.

## II. EXPERIMENTAL

Bi2223 tapes with interfilamentary oxide barriers were fabricated by a conventional powder-in-tube (PIT) method. SrZrO<sub>3</sub> with its mean grain size below 1  $\mu\text{m}$  was selected as barrier materials because of its compatibility with Bi2223 superconductors. Moreover, additional Bi2212 powder corresponding to 15wt% was mixed with SrZrO<sub>3</sub> to improve its ductility for cold working [8]. The precursor powders were packed into pure Ag tube with an outer diameter of 9.5 mm and a wall thickness of 0.75 mm. Then, the composite was deformed into a hexagonal cross-sectional shape by drawing.

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T. Makihara, R. Inada, and A. Oota are with Toyohashi University of Technology, 1-1 Hibarigaoka, Tempaku-cho, Toyohashi, Aichi 4418580 Japan (e-mail: inada@ee.tut.ac.jp).

S. Sakamoto is with Kisarazu National College of Technology, 2-1-1 Kiyomidai-Higashi, Kisarazu, Chiba 2920041 Japan.

C.S. Li and P.X. Zhang are with Northwest Institute for Nonferrous Metal Research, P.O. Box 51, Xi'an Shaanxi 710016 P.R. China.

The outside surface of the monocoil wire was coated by the oxide paste with the thickness of  $\sim 60 \mu\text{m}$ . After the heat treatment at  $550^\circ\text{C}$  for 24 h in air to decompose and evaporate the organic binder, 19 pieces of coated monocoil wire were stacked into honeycomb structure and packed into Ag-Mg alloy tube with an outer diameter of 15.5 mm and a wall thickness of 0.75 mm. The composites were drawn to a diameter of 1.83 mm and then twisted. Finally, the round wires were formed into tape shapes by flat rolling and sintered with an intermediate rolling. Typical transverse cross sectional view and specifications of barrier tapes are shown in Fig. 1 and Table 1. Cross sectional sizes and lengths for fully reacted barrier tapes are approximately  $3.7 \text{ mm} \times 0.23 \text{ mm}$  and 0.8–0.9 m, respectively. For comparison, the non-twisted tapes without barriers were also prepared by using same fabrication procedure.

TABLE I SPECIFICATIONS OF Bi2223 TAPES WITH  $\text{SrZrO}_3 + \text{Bi2212}$  BARRIER

Sample	T1	T2
Number of filaments	19	19
Tape cross section	$3.70 \text{ mm} \times 0.23 \text{ mm}$	$3.70 \text{ mm} \times 0.23 \text{ mm}$
Tape length	0.9 m	0.8 m
Twist pitch length $L_t$	Not twisted	10 mm
Fraction of SC filaments	24%	23%
Maximum $J_c$	$24.6 \text{ kA/cm}^2$	$19.2 \text{ kA/cm}^2$
Averaged $J_c$	$21.2 \text{ kA/cm}^2$	$17.3 \text{ kA/cm}^2$
Standard deviation of $J_c$	7%	9%

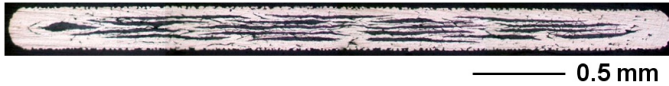


Fig. 1. Typical transverse cross sectional view of 19-filamentary Bi2223 tapes with  $\text{SrZrO}_3 + \text{Bi2212}$  barriers.

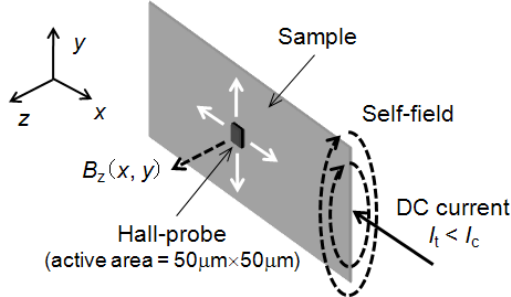


Fig. 2. A schematic diagram for measuring self-field  $B_z$  distributions on a tape by Hall-probe. The definition of coordinates is also shown.  $x$  and  $y$  directions are corresponding to longitudinal and lateral directions for the tape, respectively.

Before measuring self-field distributions, transport critical currents  $I_c$  at 77 K and self-field were measured at every 10 mm section along a length of each tape, by a conventional DC four probe method with an electric field criterion of  $1 \mu\text{V/cm}$ . For  $I_c$  measurement without soldering voltage taps on a tape, contact-type voltage taps with a separation of 10 mm were used. The critical current density  $J_c$  was determined from  $I_c$  and transverse cross-sectional area of Bi2223 filaments.

Two-dimensional distributions of self-field  $B_z$  in perpendicular to the broader face of a tape carrying DC transport current  $I_t < I_c$  were measured by SHM [5], [6].

Schematic diagram for self-field measurement is shown in Fig. 2. SHM was equipped with a micro-Hall probe with an active area of  $50 \mu\text{m} \times 50 \mu\text{m}$  on a movable  $x$ - $y$  stage and a sample holder facing the probe. The probe was used to measure the magnetic field in perpendicular to the tape surface. The Hall voltages were measured using a nanovoltmeter by scanning the probe on the  $x$ - $y$  stage with finite steps of 0.2 mm in both  $x$ - and  $y$ -directions, at a fixed distance of 0.5 mm away from a tape surface. Consequently, two-dimensional  $B_z$  distributions can be visualized in a  $x$ - $y$  plane on a tape. The measurements were carried out at 77 K. It is noted that the time elapsed for each scan in our SHM was about 0.2 s and the scan was started right after applying DC current.

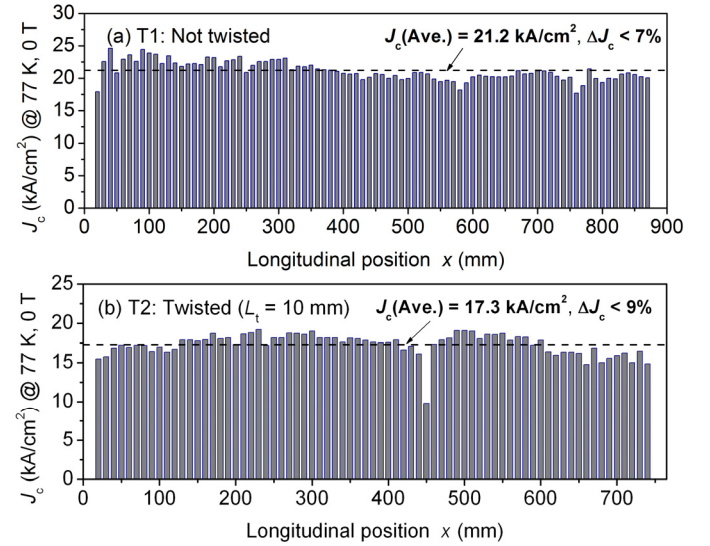


Fig. 3. Longitudinal distributions of critical current density  $J_c$  at 77 K and self-field for Bi2223 tapes with interfilamentary  $\text{SrZrO}_3 + \text{Bi2212}$  barriers: (a) without twisting and (b)  $L_t = 10 \text{ mm}$ . Measurements were carried out at every 10 mm section along each tape length.

### III. RESULTS AND DISCUSSION

Fig. 3 shows longitudinal  $J_c$  distributions for non-twisted barrier tape T1 and twisted one T2 with  $L_t = 10 \text{ mm}$  at 77 K and self-field. Maximum value, averaged value and standard deviation of  $J_c$  for each tape are also summarized in Table 1. As can be seen, averaged  $J_c$  for twisted tape T2 is approximately 20% lower than non-twisted one T1. The longitudinal uniformity of both barrier tapes was confirmed to be within 10%. Non-twisted tape T1 shows  $J_c > 18 \text{ kA/cm}^2$  in whole part along a tape length. Since averaged  $J_c$  for tapes without barriers prepared by same fabrication process was ranged in 23–24  $\text{kA/cm}^2$ , degradation of  $J_c$  induced by introducing  $\text{SrZrO}_3 + \text{Bi2212}$  barriers among the filaments was suppressed within 15%. Only for twisted barrier tape T2, sudden drop of  $J_c < 10 \text{ kA/cm}^2$  was clearly observed around  $x = 450 \text{ mm}$ . In this part, generation of small transversal crack with the size of  $\sim 0.5 \text{ mm}$  was observed in sheath part at one side of tape edge.

Based on the longitudinal  $J_c$  distribution for each tape, we measured self-field distributions at some selected regions with their section lengths of 40 mm by SHM. Fig. 4 shows two-dimensional distributions of self-field  $B_z$  in perpendicular to the wider face of non-twisted barrier tape T1 at  $x = 280\text{--}320$

mm with higher  $I_c$  ( $= 47\text{--}48$  A) and  $x = 440\text{--}480$  mm with lower  $I_c$  ( $= 38\text{--}41$  A). In both graphs, the direction of  $B_z$  is inversed at tape center ( $y = 0$  mm) as a border and magnitude of  $B_z$  is represented by light and shade of gray color. The transport current  $I_t$  was set to be  $\sim 50\%$  of  $I_c$  monitored at each measurement section of 40 mm. The time dependence of  $B_z$  at a specific position was checked by applying  $I_t = 0.5I_c$  for several hours to confirm  $B_z$  did not vary during the measurement. This indicates that the energy dissipation with increasing time by bias current is not caused during  $B_z$  measurement. As shown in Fig. 4(a), the distribution of  $B_z$  at higher  $I_c$  section seems to be nearly uniform along a tape length so that there are no large defects to obstruct the current transport in this section. On the other hand, for lower  $I_c$  section in Fig. 4(b), the contour lines for  $B_z$  are distorted at  $x = 460\text{--}470$  mm with the lowest  $I_c$  ( $= 37.8$  A) part in this section. Since the lateral  $B_z$  distributions at fixed longitudinal position  $x$  are strongly influenced by field penetration inside the filamentary core of a tape, distorted contour lines of  $B_z$  at  $x = 460\text{--}470$  mm along a tape length is attributed to the presence of locally degraded region.

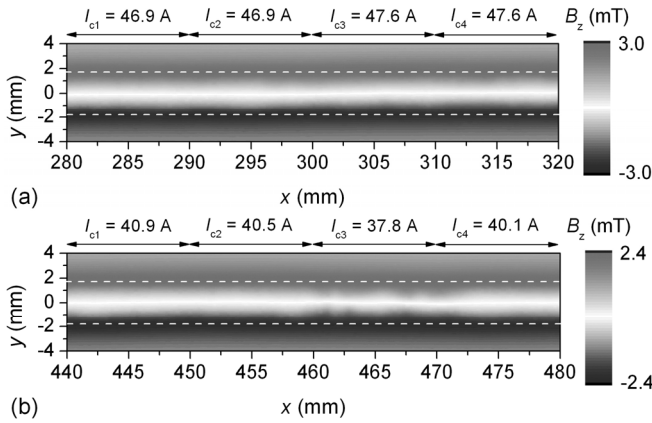


Fig. 4. Contour maps for self-field  $B_z$  on non-twisted barrier tapes T1 at (a)  $x = 280\text{--}320$  mm with high  $I_c$  and (b)  $x = 440\text{--}480$  mm with low  $I_c$ . White dashed lines represent the positions of both tape edges. Magnitude of transport currents  $I_t$  for each data are 24 A for (a) and 19 A for (b), respectively. Note that the direction of  $B_z$  becomes opposite at tape center ( $y = 0$  mm) as a border.

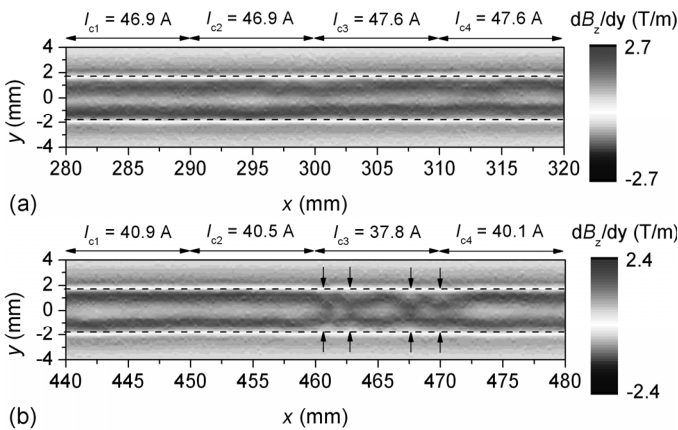


Fig. 5. Contour maps for the differentials of self-field  $dB_z/dy$  on non-twisted barrier tapes T1 at (a)  $x = 280\text{--}320$  mm with high  $I_c$  and (b)  $x = 440\text{--}480$  mm with low  $I_c$ . Black dashed lines represent the positions of both tape edges. Magnitude of transport currents  $I_t$  for each data are 24 A for (a) and 19 A for (b), respectively. The positions indicated by pairs of vertical arrows in (b) correspond to locally degraded region.

It is known that a differential of self-field  $B_z$  along a lateral direction  $dB_z/dy$  for a tape-form conductor strongly correlates to the sheet current density flowing in parallel to a tape length at the observation point [12], [16]. Therefore, two-dimensional distribution for  $dB_z/dy$  on a tape surface roughly corresponds to that for sheet current density. The comparison of contour maps for  $dB_z/dy$  on non-twisted barrier tape T1 at both higher and lower  $I_c$  sections are shown in Fig. 5, which is derived from the data for  $B_z$  in Fig. 4. For both measurement sections,  $dB_z/dy$  becomes larger near both tape edges, so that current is mainly flowing in outer layer of filamentary core. This behavior is consistent with conventional Bi2223 multifilamentary tapes without barriers [12], and suggests that resistive barriers among the filaments have no notable effect on self-field penetration into filamentary core under current transport. In higher  $I_c$  section at  $x = 280\text{--}320$  mm in Fig. 5(a), the distributions for  $dB_z/dy$  along a tape length are nearly uniform. On the other hand, in lower  $I_c$  section at  $x = 420\text{--}460$  mm in Fig. 5(b), the contour lines for  $dB_z/dy$  are distorted along a tape length at  $460\text{--}470$  mm with the lowest  $I_c$  ( $= 37.8$  A). At several positions indicated by the pairs of vertical arrows, the current flowing regions from tape edges to center are more extended. At these positions, several defects were intermittently generated along a tape length during tape fabrication and current transport capability was intermittently deteriorated. We believe that such local degradation could be eliminated by optimizing the conditions for flat rolling.

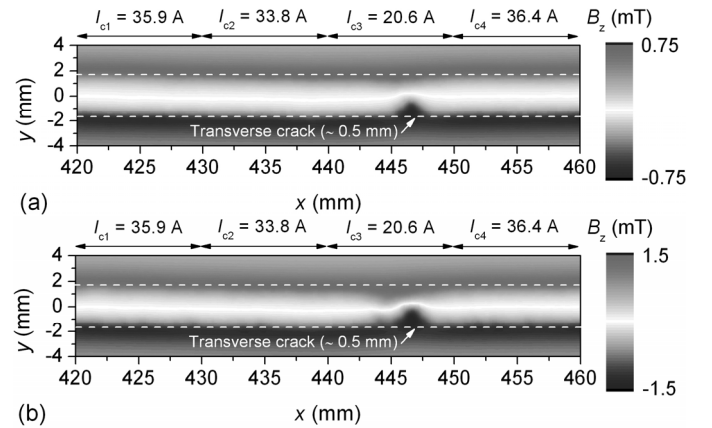


Fig. 6. Contour maps for self-field  $B_z$  on twisted barrier tapes T2 at  $x = 420\text{--}460$  mm at (a)  $I_t = 5$  A and (b)  $I_t = 10$  A. White dashed lines represent the positions of both tape edges. Note that the direction of  $B_z$  becomes opposite at tape center ( $y = 0$  mm) as a border.

The self-field  $B_z$  distribution was also evaluated for twisted barrier tape T2. Although not only barrier introduction but also filament twisting with a proper pitch length should be indispensable for reducing losses under an AC field in perpendicular to a tape face [5]–[8], filament twisting is another important factor for degrading transport property and its uniformity along a tape length [15]. Fig. 6 shows the two-dimensional  $B_z$  distribution on twisted barrier tape T2 with  $L_t = 10$  mm at  $x = 420\text{--}460$  mm, under different fixed transport current  $I_t = 5$  A and 10 A. It is noted that  $I_c$  monitored at this 40 mm section was estimated to be 22 A, which is mainly determined by the lowest  $I_c$  part at  $x = 440\text{--}450$  mm. As mentioned above, a small transversal crack with its length of

$\sim 0.5$  mm at sheath part near one tape edge is observed at  $x = 447$  mm. As can be seen, lateral field penetration is extended from both tape edges to center with increasing  $I_t$ . Near the crack position, contour lines for  $B_z$  are distorted remarkably and lateral field penetration becomes deeper than other position.

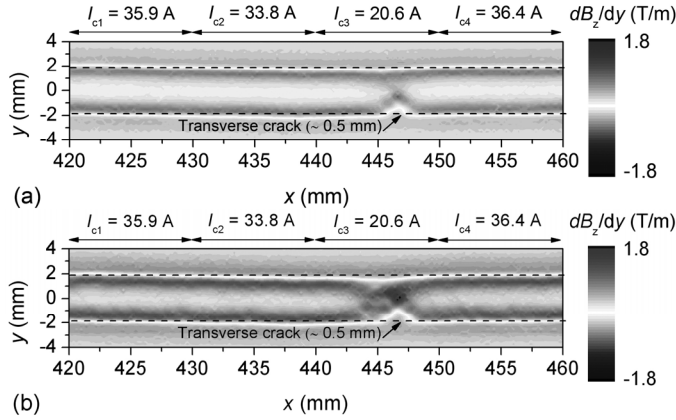


Fig. 7. Contour maps for the differentials of self-field  $dB_z/dy$  on twisted barrier tapes T2 at  $x = 420\text{--}460$  mm at (a)  $I_t = 5$  A and (b)  $I_t = 10$  A. Black dashed lines represent the positions of both tape edges.

For qualitative investigation of the current distributions for twisted barrier tape T2, we also calculated  $dB_z/dy$  distribution based on  $B_z$  distribution in Fig. 6. The two-dimensional  $dB_z/dy$  distributions for tape T2 at  $I_t = 5$  A and 10 A are shown in Fig. 7. As well as the case for non-twisted tape T1 in Fig. 5,  $dB_z/dy$  becomes larger near both tape edges at high  $I_c$  part, indicating that current is mainly flowing in outer layer of twisted filamentary core. It is also evident that the region with large  $dB_z/dy$  is extended from both tape edges to center with increasing  $I_t$ . While at  $x = 440\text{--}450$  mm with the lowest  $I_c$  ( $= 20.6$  A) in this measurement section,  $dB_z/dy$  shows quite complex distributions. Around  $x = 447$  mm with a small transversal crack in sheath part,  $dB_z/dy$  becomes zero near one side of tape edge, suggesting that several filaments positioned here are damaged by the crack generation in sheath part and have no contribution for current transport. At  $I_t = 10$  A, as shown in Fig. 7(b),  $dB_z/dy$  at  $x = 447$  mm has the maximum near tape center so that all filaments positioned here are saturated by self-field. Although the size of crack in sheath part was not so large ( $\sim 0.5$  mm), its influence on local current transport capability is quite large. The processing parameters during both twisting and flat rolling must be precisely controlled to avoid such a crack generation in sheath part.

#### IV. CONCLUSION

We evaluated self-field distributions on Bi2223 tapes with interfilamentary  $\text{SrZrO}_3 + \text{Bi2212}$  as resistive barriers under DC current transmission, by using a scanning Hall-probe microscopy (SHM). Non-twisted and twisted barrier tapes with their lengths of  $\sim 1$  m were used for measurements. Although the longitudinal uniformity of transport  $J_c$  measured at every 10 mm sections for both tapes were within 10%, sudden drop of  $J_c$  was observed only in twisted tape. Self-field distributions on both barrier tapes are significantly distorted in lower  $I_c$  section. From the measurement for non-twisted barrier tape, it was

confirmed that several defects with its longitudinal size of 1–2 mm were intermittently generated at lower  $I_c$  part. On the other hand, at the lowest  $I_c$  part in twisted tape, the filaments near one tape edge were damaged by twisting and had no contributions for current transport. Such information given by SHM has crucial importance to analyze the obstacles for current transport in barrier tape with low AC losses. In future work, we are going to examine the relations between deformation parameters during twisting and rolling process and self-field distributions systematically to improve  $J_c$  and scalability for barrier tapes.

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