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Investigation of the properties of alkali-activated slag mixes involving the use of nanoclay and nucleation seeds for 3D printing

Biranchi Panda^{1,3}, Shaoqin Ruan², Cise Unluer^{2,4*}, Ming Jen Tan¹

¹ Singapore Centre for 3D Printing, School of Mechanical & Aerospace Engineering, Nanyang Technological University, 50 Nanyang Avenue, Singapore 639798

² School of Civil and Environmental Engineering, Nanyang Technological University, 50 Nanyang Avenue, Singapore 639798

³ Department of Mechanical Engineering, Indian Institute of Technology Guwahati, Guwahati, India

⁴ School of Engineering, University of Glasgow, G12 8LT, Glasgow, United Kingdom

* Corresponding author. E-mail address: Cise.Unluer@glasgow.ac.uk

Abstract:

This study investigated the properties of alkali activated slag (AAS) binders formulated for extrusion-based 3D printing. The fresh properties of AAS mixes were tailored through the use of nanoclay (NC) and nucleation seeds. The printability criteria employed were the ease of extrusion (extrudability) and the stability of the layered structure (buildability). Introduction of 0.4% NC in AAS mixes led to improved thixotropic properties due to the flocculation effect, which accounted for the extrudability and shape fidelity of the binder. Inclusion of 2% hydromagnesite seeds in this mix design provided additional nucleation sites for the increased precipitation of hydrate phases, resulting in denser microstructures. This enhanced the hydration reaction and improved the structural build-up rate necessary for large-scale 3D printing. The developed AAS mix containing 0.4% NC and 2% hydromagnesite seeds was used in the printing of an actual 3D structure to demonstrate its feasibility to be used in 3D printing applications.

Keywords: Alkali-activated slag; 3D printing; rheology; hydration; strength

1. Introduction

Extrusion-based additive manufacturing, also known as 3D printing, enables the mould-free fabrication of complex customized parts, which cannot be easily processed by other conventional manufacturing methods [1-4]. This technology has been successfully applied in aerospace, automotive and biomedical fields, while it is still being researched in the construction sector. The expected benefits of 3D concrete printing are higher productivity, shorter construction periods, higher geometrical freedom and more efficient use of natural resources [5]. The use of 3D printing can also present advantages in terms of reduced costs in the case of complex structures, in comparison to the conventional construction methods.

A pivotal approach in the success of 3D printing in construction is the use of Building Information Modelling (BIM). Since BIM already serves as a rich source of geometric information for existing structures, on site 3D concrete printing will eventually need scheduling and assembly sequence information to maintain safety and productivity [6, 7]. Previous studies [8] that shared this vision proposed a shift to a digital construction organization by combining existing technologies such as rapid digital mapping, BIM, digital collaboration, internet of things and design to construction. To further digitalise the construction industry, digital twin technology can be used to continuously monitor progress against the schedule laid out in the BIM model. Furthermore, approaches such as the digital twin technology can be combined with 3D concrete printing to reduce the volume of trial and error testing, reduce defects and shorten the time between the design and production processes [9].

While the use of cement-based materials in 3D printing presents several advantages, certain challenges need to be resolved for a successful printing process. To be used for printing applications, the rheology of cement-based mixes must meet certain requirements. Accordingly, during extrusion, the material must be fluid to prevent any blocking, bleeding or segregation. However, once they are printed, each layer must be able to harden quickly to support the superposed layers [10]. Another parameter to be considered is the time gap between printed layers. Long time gaps can provide an adequate mechanical strength for supporting the weight of the subsequently deposited layers, while short time gaps ensure optimized bonding strength.

32 Therefore, for successful printing, a narrow process window in terms of material yield stress exists,
33 as shown in Fig. 1, where a schematic of extrusion-based additive manufacturing of cement-based
34 mixes is demonstrated. Accordingly, the rheology of the developed mixes needs to be adjusted to
35 achieve a minimum yield stress for smooth extrudability. After extrusion, the yield stress must
36 evolve faster than the stress acting on the bottom layer to avoid strength-based failures.
37 Alternatively, materials with a high yield stress can result in poor interlayer bonds, despite their
38 ability to support additional layers.

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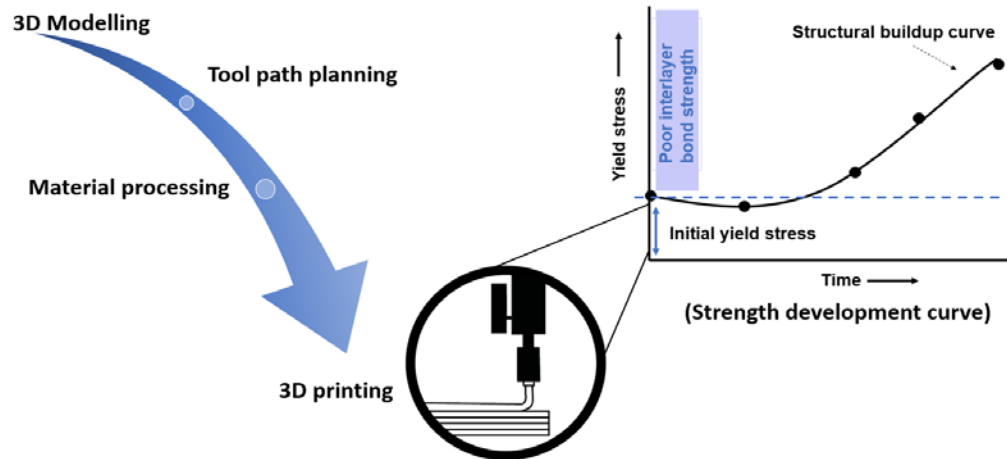
40 Incorporation of reinforcement into the concrete matrix during the printing process is one of the
41 most challenging issues needed to be dealt with to enable structural applications of 3D printing
42 technology [11, 12]. In terms of materials, most of the printable mixtures contain ordinary Portland
43 cement (PC) as the prime binder material due to its inherent thixotropic properties that originate
44 from the combined effect of interparticle, gravitational and inertial forces [13-17]. However, as
45 the production of PC accounts for 5-7% of the total anthropogenic CO₂ emissions, alternative
46 cementitious materials are being investigated for their suitability to be used in 3D printing. Recent
47 studies [18-21] have demonstrated the use of cementitious industrial by-products to reduce the
48 carbon footprint of PC in 3D printing applications. As a part of these initiatives, the importance of
49 proper rheology control for smooth extrusion and higher buildability properties was highlighted.

50

51 In line with these initiatives, this study aims to investigate the use of alkali-activated slag (AAS)
52 mixes in 3D printing applications to contribute to the development of a sustainable built
53 environment. To enable this, the prepared AAS formulations included nanoclay due to its
54 thixotropic properties [23]. Furthermore, nucleation seeds (i.e. hydromagnesite) were also
55 included to improve the rheological and mechanical properties of AAS [24, 25]. Accordingly,
56 previous research [26] has shown that the use of seeds can provide additional nucleation sites for
57 the increased precipitation of hydrate phases, thereby enhancing the rate and degree of the
58 hydration reaction. To study the effect of these additives, nanoclay was first introduced into AAS
59 mixtures for improved printability. An initial assessment of the fresh properties led to the
60 determination of the mix with the highest yield stress and lowest viscosity. Once this mix was
61 determined, different dosages (i.e. 1-2% by mass of slag) of hydromagnesite seed was introduced
62 into the mix design. X-ray diffraction (XRD) and field emission scanning electron microscopy

63 (FESEM) were employed to analyse the formation of hydration products and investigate the
64 microstructural development at the end of the curing process.

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68 **Fig. 1** Schematic of extrusion-based additive manufacturing of cement-based mixes

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71 **2. Materials and Methodology**

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73 **2.1 Raw materials and sample preparation**

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75 The primary material utilized in this study was ground granulated blast-furnace slag (GGBS),
76 provided by EnGro Corporation (Singapore). The chemical and physical properties of GGBS are
77 shown in Table 1. The alkali activator was solid sodium metasilicate ($\text{Na}_2\text{SiO}_3 \cdot 5\text{H}_2\text{O}$), obtained
78 from VWR (Singapore). Nanoclay (NC), a thermal treated, highly purified attapulgite clay
79 supplied by Actigel (USA), was used to improve the thixotropy of the AAS binder. As can be seen
80 from Fig. 2(a), NC particles agglomerated into clusters of different size composed of a pleated
81 spherical morphology. Hydromagnesite ($4\text{MgCO}_3 \cdot \text{Mg}(\text{OH})_2 \cdot 4\text{H}_2\text{O}$) seeds, obtained from Fisher
82 Scientific (UK) with a specific surface area of $43.5 \text{ m}^2/\text{g}$, were used as nucleation seeds to stimulate
83 the nucleation and growth of hydration products. The SEM image of hydromagnesite (Fig. 2(b))
84 demonstrated these seeds as spherical agglomerations with a diameter of $\sim 1\text{-}7 \mu\text{m}$, composed of

85 ~0.5 μm diameter disks [24]. Fine aggregates with a maximum particle size of 1.18 mm was used
 86 in a saturated surface dry (SSD) condition to formulate the AAS mortars.

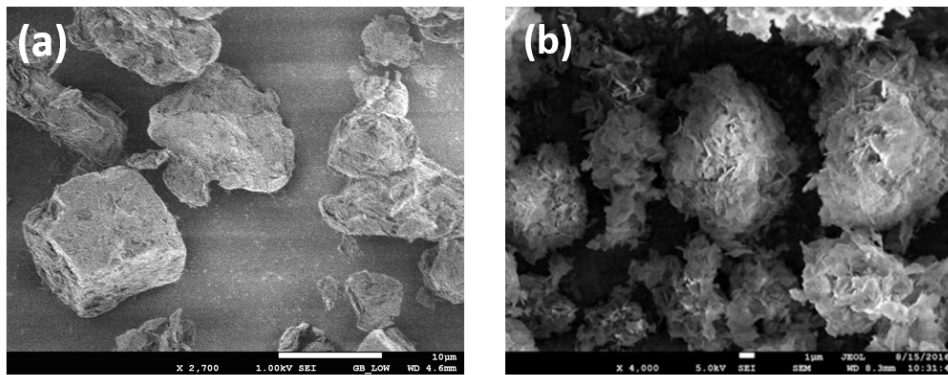
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88 **Table 1** Chemical composition and physical properties of GGBS.

89

	Chemical composition (%)							Physical properties	
	SiO ₂	Al ₂ O ₃	CaO	MgO	SO ₃	TiO ₂	LOI	Specific gravity (g/cm ³)	Blaine surface area (m ² /g)
GGBS	29.65	15.56	39.37	7.54	4.32	1.75	4.0	2.85	> 300

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92

93 **Fig. 2** SEM images of (a) nanoclay and (b) hydromagnesite seed [20]

94

95 The AAS used in this study was composed of only GGBS and 10% activator (i.e. by mass of slag)
 96 in line with the findings of a previous study [26]. Mixtures were prepared using water to binder
 97 (w/b) ratios of 0.35 and 0.40 for all pastes and mortars (i.e. at a sand/binder ratio of 0.83),
 98 respectively. AAS mortars without NC were prepared by adding water to the slag and sand mixture
 99 and mixed until a homogenous blend was obtained. For the preparation of mixes containing NC,
 100 NC was first blended into the predetermined amount of water for 3-4 minutes, after which it was
 101 added into the slag and sand mix and further mixed for another 2 minutes to ensure effective
 102 dispersion. For mixes involving the use of seeds, the seeds were first dispersed in half of the
 103 required total water and added into the mix, followed by the addition of NC, which was mixed
 104 with the remaining water.

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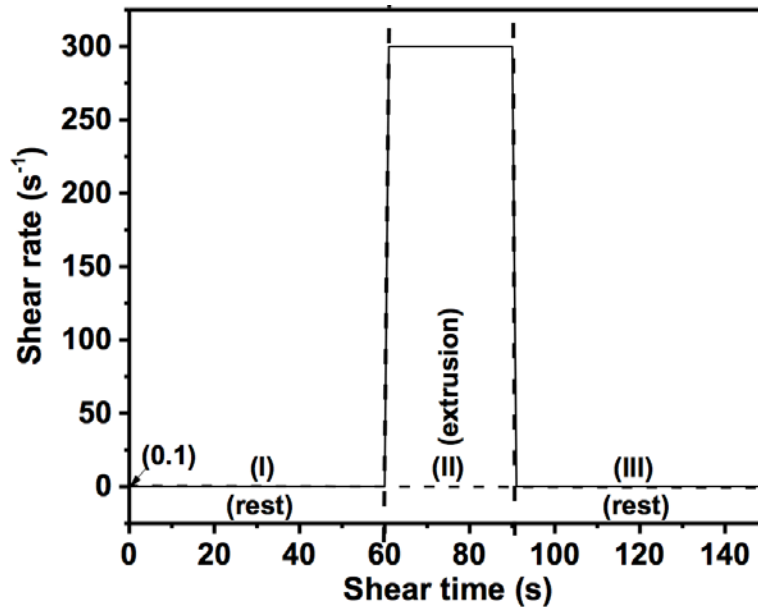
2.2 Methodology

2.2.1 Static yield stress

A commercial rheometer (Anton-Paar MCR 102) was used to measure the static yield stress of the prepared AAS mortars. To initiate the study, freshly prepared AAS mortars were loaded in a measuring cup. Stress growth test was then performed by applying deformation at a constant shear rate of 0.1 s^{-1} . The shear stress progressively developed to a maximum value, followed by its decline to reach an equilibrium value. The static yield stress was defined as the peak shear stress value [27].

2.2.2 Thixotropy (shear thinning and viscosity recovery)

Thixotropy is an important property of printable materials, which can be characterized by a high viscosity at low stress and vice versa [39]. While there are various methods to quantify thixotropy, the “viscosity recovery” test was used to measure the thixotropy of AAS mixtures in this paper. Shear thinning property was measured by applying a constant shear rate of 300 s^{-1} , while viscosity recovery was measured by following a three-stage protocol, as previously described in [28]. The three stages and their respective shear rates and shearing timings were decided by mimicking the concrete printing process, where the state of the material starts from rest (i.e. at hopper) followed by high shear (i.e. extrusion) and finally ends at rest (i.e. on print bed). Fig. 3 shows the schematic of the protocol used for the evaluation of viscosity recovery. In addition to thixotropy, the structural build-up rate was also calculated from the evolution of static yield stress after 0, 5, 10 and 15 minutes of rest, which was used to assess the buildability property of the AAS mortars.



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135 **Fig. 3** Viscosity recovery protocol for measuring the thixotropy of AAS mortars

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138 2.2.3 3D printing

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140 The printability criteria employed in this study were: (i) ease of extrusion through a rectangular
141 nozzle (i.e. extrudability) and (ii) stability of the layered printed filament (i.e. buildability). A 4-
142 axis gantry printer [29] was used to test the extrudability of AAS mortars using a 30 (L) x 15 (W)
143 mm nozzle. The printer was connected to a screw pump and the print speed was varied from 60 to
144 100 mm/sec at a constant flow rate to find the optimum speed for better shape stability of the
145 filaments. This was followed by the printing of a hollow cylinder with the optimum combination
146 of flow rate and speed to demonstrate the buildability of the AAS mortar.

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149 2.2.4 Microstructural characterization (XRD and FESEM)

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151 Samples extracted from AAS pastes were stored in acetone to stop hydration, followed by vacuum
152 drying in preparation for XRD and FESEM analyses. XRD was recorded on a Philips PW 1800
153 spectrometer using Cu K α radiation (40 kV, 30 mA), with a scanning rate of 0.04° 2 θ /step from

154 10 to 70° 2θ. FESEM was carried out with a Zeiss Evo 50 microscope to investigate the
155 morphologies of the hydration products. The vacuum dried samples were mounted onto aluminium
156 stubs using double-sided adhesive carbon disks and coated with gold before FESEM analysis.

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159 **3. Results and Discussion**

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161 **3.1 Effect of nanoclay on extrusion rheology**

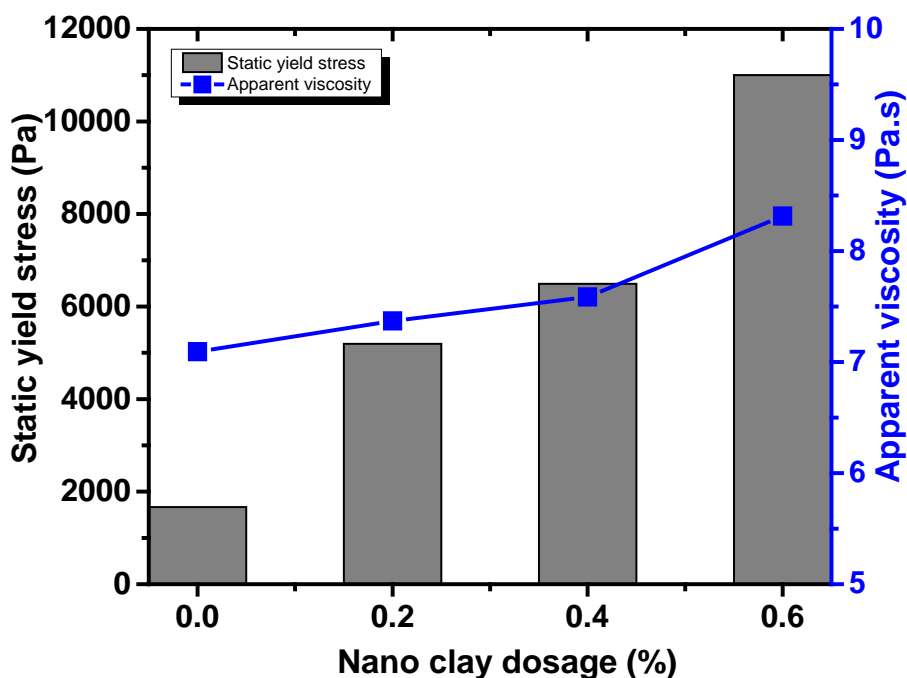
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163 Printability is defined as the ability of a mixture to be extruded (i.e. extrudability) and maintain its
164 structural integrity when built in layers (i.e. buildability) [10, 32]. In this respect, yield stress is an
165 important rheological parameter that influences printability. Similarly, viscosity of a mixture is
166 also related to its extrudability. Therefore, the yield stress and viscosity values obtained from a
167 rotational rheometer were synergistically considered to analyse the printability of the prepared
168 AAS mixes. Fig. 4 shows the effect of the inclusion of different amounts of NC on the static yield
169 stress and apparent viscosity of AAS mortars. AAS mixes usually demonstrate low yield stress
170 values due to the absorption of silicate anions (i.e. from the activator) on the slag surfaces, which
171 results in strong double-layer repulsive forces between the slag particles, thereby causing particle
172 separation and a low yield stress [31]. However, when compared to the control mix (0% NC), an
173 increase in the NC content led to a higher static yield stress. Accordingly, the inclusion of a small
174 amount of NC (0.4%) was sufficient to increase the yield stress by higher than three times when
175 compared to the control mix. Therefore, the addition of NC enabled rapid flocculation, making it
176 an ideal thixotropic material for 3D printing applications.

177

178 While the use of NC increased the yield stress of AAS mixes, the apparent viscosity remained
179 relatively constant under different NC contents. This could be useful in providing a smooth
180 extrusion and deposition of the prepared mix, maintaining the shape fidelity of the filaments. Such
181 unique behaviour of NC is attributed to the flocculation phenomena that controls the rheology,
182 depending on the applied shear forces at different stages of concrete printing [32]. Accordingly,
183 NC particles carry a negative and positive charge on their opposite ends. They tend to associate
184 with each other by electrical attraction when the material is at rest. NC also presents a higher initial

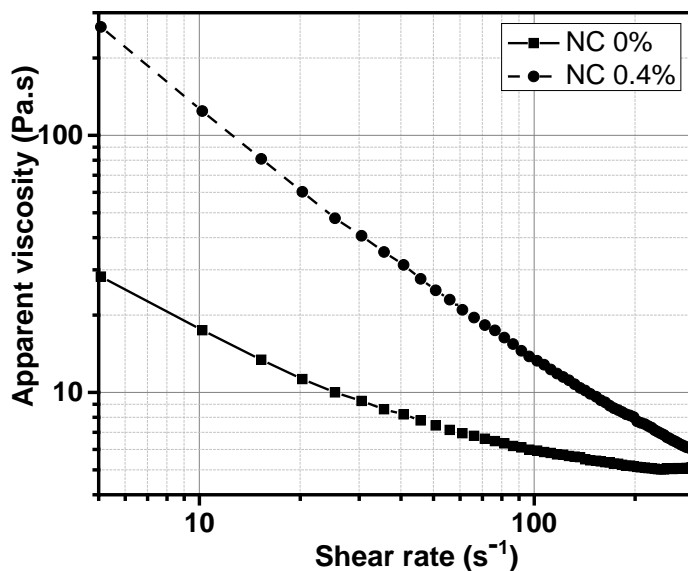
185 viscosity at rest, which then decreases under strong pre-shearing as the structure breaks down [33].
 186 Amongst the mixes prepared, since the addition of 0.6% NC resulted in the maximum yield stress
 187 that was measurable by the rheometer, this was the highest amount of NC introduced into the
 188 prepared AAS mortars in this study. However, as it provided the highest yield stress within the
 189 extrudable limit (< 8 KPa), a NC dosage of 0.4% was selected as the optimum amount for
 190 subsequent analyses. Alternatively, the addition of 0.6% NC led to the extrusion of a discontinuous
 191 filament, which could negatively affect the interlayer bond strength due to the high yield stress of
 192 the mixture.
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 196 **Fig. 4** Static yield stress and apparent viscosity as a function of NC dosage
 197

198 Fig. 5 shows the flow curves over a range of shear rates ($1-300$ s^{-1}) for AAS mortars with and
 199 without the addition of 0.4% NC. A shear-thinning behaviour was observed in both samples, which
 200 indicated increased viscosities with decreasing shear rates. When compared to the control mix, the
 201 NC modified mix resulted in higher viscosities at low shear rates, which could be helpful in
 202 maintaining the shape of the filament and support more layers. Alternatively, both mixtures
 203 demonstrated comparably low viscosities at high shear rates, which could enable a smooth

204 concrete flow without any discontinuity in the filament deposition. These results have
205 demonstrated the role of NC in contributing to the development of more rigid AAS mixes via the
206 flocculation of clay, resulting in mixes with high viscosities at low shear rates and vice versa.
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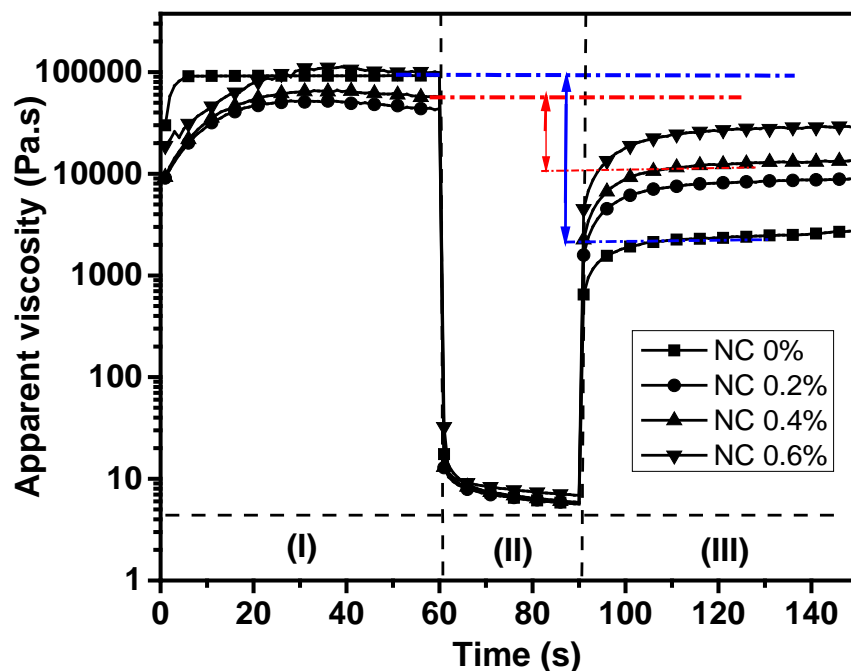
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210 **Fig. 5** Viscosity as a function of shear rate for the control (0% NC) and 0.4% NC modified AAS
211 mortars

212

213 In order to be ideal for extrusion based printing purposes, a material should be highly thixotropic
214 [10, 34]. In this respect, a low viscosity under a high shear force and the recovery of the viscosity
215 to its original value after the removal of the shear force are preferable. Therefore, it is important
216 to know how fast a deposited AAS mortar can recover its initial viscosity before the next layer is
217 printed. Accordingly, Fig. 6 shows the viscosity recovery curves of AAS mortars including
218 different amounts of NC. Although the initial viscosity (i.e. at stage I) of the control sample was
219 higher, its recovery ability after strong shearing (i.e. at stage III) was much lower. This difference,
220 shown by the blue arrow mark (Fig. 6), was an indication of the low recovery property of the
221 control sample, which would negatively affect its shape retention after deposition as the viscosity
222 after extrusion reduced due to poor recovery ability. On the other hand, mixes including NC
223 revealed significantly better recovery properties, as shown by the red arrow mark (Fig. 6). The
224 improved recoveries of these samples were attributed to the thixotropic property of NC. Amongst

225 the prepared samples, AAS including 0.4% NC was chosen for the assessment of its printability
226 and structural build-up property due to its extrudable nature and 25% higher recovery ability than
227 the control mix (0% NC).
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231 **Fig. 6** Effect of the NC on the viscosity recovery of AAS mixtures

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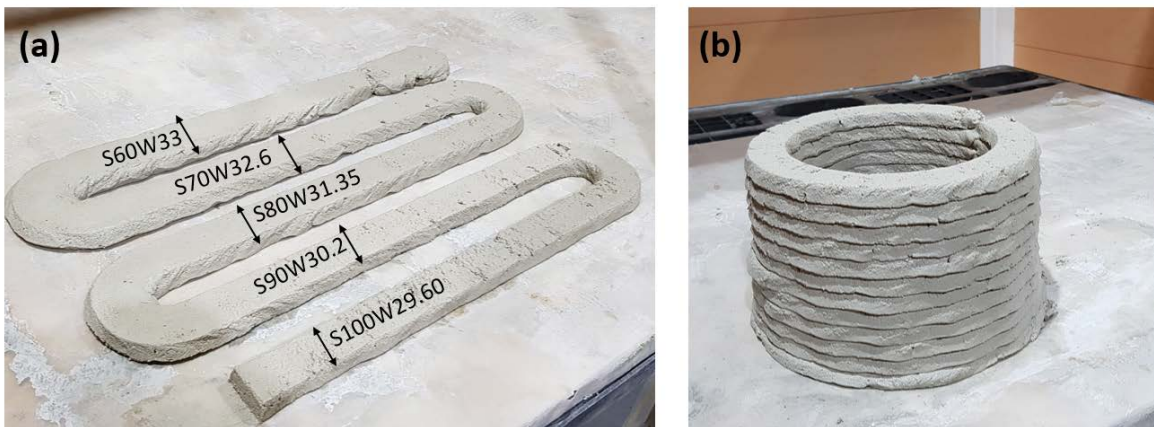
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234 3.2 3D printing

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236 3D printing of cementitious materials is a complex process, during which material properties, part
237 design and process parameters interact with each other and define the characteristics of the final
238 product. Process parameters like printing speed, nozzle shape and flow rate have a strong impact
239 on the dimensional accuracy and mechanical properties of the final product [3, 29]. Fig. 7(a) shows
240 the effect of printing speed (i.e. ranging between 60-100 mm/sec) on the dimensional accuracy of
241 the filaments, while flow rate was kept constant. The printing process revealed that the printing
242 speed and flow rate were dependant on each other. Accordingly, at each flow rate, there was an
243 optimum speed that enabled the production of filaments with the same dimension as the nozzle

244 inlet. An increase or decrease in the speed beyond this optimum value would cause extra or
245 insufficient material deposition, respectively. Out of the trials performed, setting the printing speed
246 to 90 mm/s resulted in extrudates with a width (W) that was equivalent to the nozzle dimension
247 (30 mm). Therefore, to demonstrate the possibility of using the developed AAS mortars in 3D
248 printing applications, this speed was used in the printing of a cylinder with a diameter of 20 cm.
249 Accordingly, the AAS mix containing 0.4% NC was printed up to 15 layers without the
250 deformation of the bottom layer, as shown in Fig. 7(b). The deformation during printing can also
251 be simulated by using optical techniques based on digital photogrammetry or terrestrial laser
252 scanning, as discussed in previous studies [34].
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256 **Fig. 7** 3D printing of AAS mix containing 0.4% NC, showing the effect of: (a) printing speed on
257 filament dimension and (b) buildability of several layers

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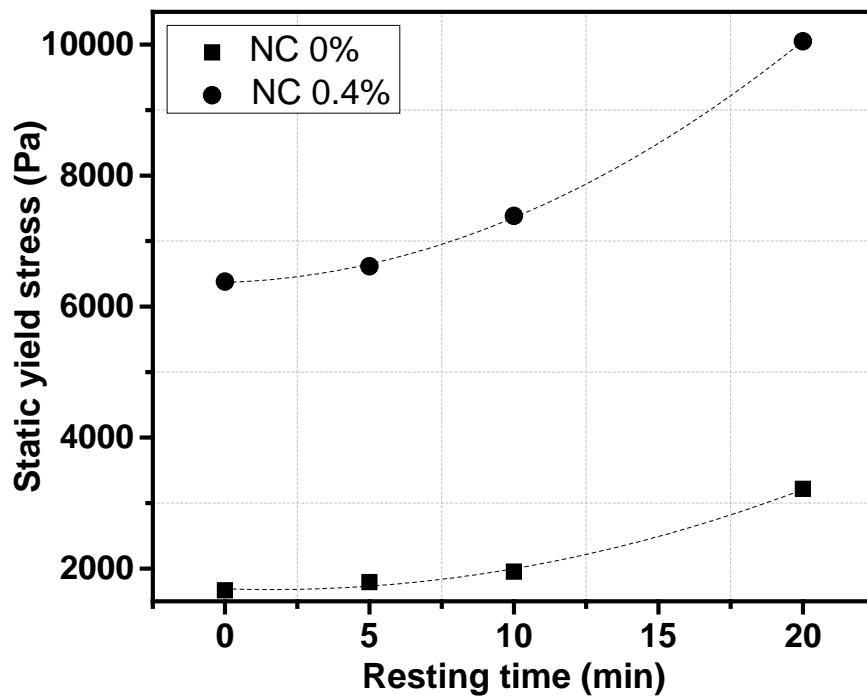
260 **3.3 Time-dependent rheological properties**

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262 The rheology of the material changes with time according to binder chemistry and presence of
263 additives. The evolution of rheology with time is an important parameter for the printing process
264 since fresh concrete is deposited in a layer-by-layer manner. This necessitates the material to
265 possess fresh properties that can provide a 3D printed structure with structural stability [35].
266 Depending on the printing volume and speed, the material can be tailored to gain sufficient strength

267 before the printing of subsequent layers. The early strength of the material can be indirectly
268 measured from its yield strength. Accordingly, Fig. 8 shows the yield stress values of the control
269 AAS mix and the AAS mix containing 0.4% NC at different time intervals, obtained directly from
270 the stress growth test. The inclusion of NC was found to increase the static yield stress of AAS
271 mixes, revealing values that were three times higher than those of the control mix. This
272 improvement in the yield stress was associated with the flocculation of NC particles, as discussed
273 earlier in Section 3.1.

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277 **Fig. 8** Evolution of the static yield stress of AAS mortars with and without 0.4% NC

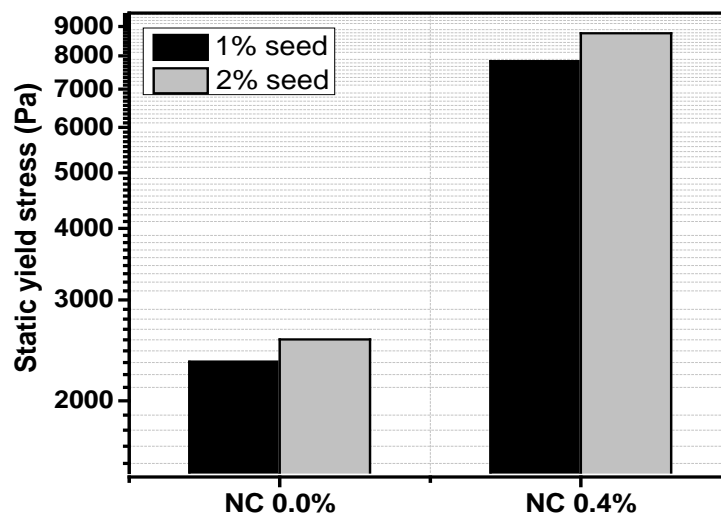
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279 Alternatively, the rate of change of yield stress over time, also known as the structural build-up
280 rate, was not significantly affected by the addition of clay. AAS mix containing 0.4% NC
281 progressively gained yield stress, similar to the control sample. This observation was in line with
282 the findings of previous studies [32, 33], where the addition of NC was reported to have an
283 immediate impact on the yield stress, but have little or no influence on the rate of change of the
284 yield stress over time. To overcome this issue, an accelerator can be added into AAS mixtures,
285 which will enhance the build-up rate by accelerating the chemical (i.e. hydration) reaction.

286 Previous studies [24-26] have shown that nucleation seeds such as Ca- or Mg-based
287 hydrate/carbonate phases can provide additional nucleation sites for the rapid and increased growth
288 of hydration products. Therefore, different amounts of hydromagnesite
289 ($4\text{MgCO}_3 \cdot \text{Mg}(\text{OH})_2 \cdot 4\text{H}_2\text{O}$) seeds were introduced into the AAS mix containing 0.4% NC to
290 investigate their effectiveness in improving the build-up rate and early age mechanical properties.
291

292 Fig. 9 shows the effect of 1% and 2% seed addition into the AAS mortar with and without the use
293 of NC. Along with the findings of previous studies, where it was revealed that high seed contents
294 were not effective due to dispersion issues [24, 25], only up to 2% seed addition was investigated
295 in this study. Although the seed dosage had a little effect on the yield stress of both mixes, its
296 influence on increasing the rate of change of yield stress over time, which can enhance the
297 buildability of the printed structure without the need for any additional admixtures, should be
298 evaluated before any final conclusions on the effectiveness of seeds could be made. Accordingly,
299 Fig. 10 shows a successfully printed twisted column involving the use of the AAS mix containing
300 0.4% NC and 2% hydromagnesite seeds. As a part of this approach, a slender column was printed
301 directly from the CAD model demonstrated in Fig. 10(a). This process was completed without any
302 significant bottom layer deformation, as can be seen in Fig. 10(b). While this was an indication of
303 the necessary stiffness of the AAS mix in resisting deformation imposed by top layers, its higher
304 stiffness resulted in a rough surface texture, which could be improved for better printing results.

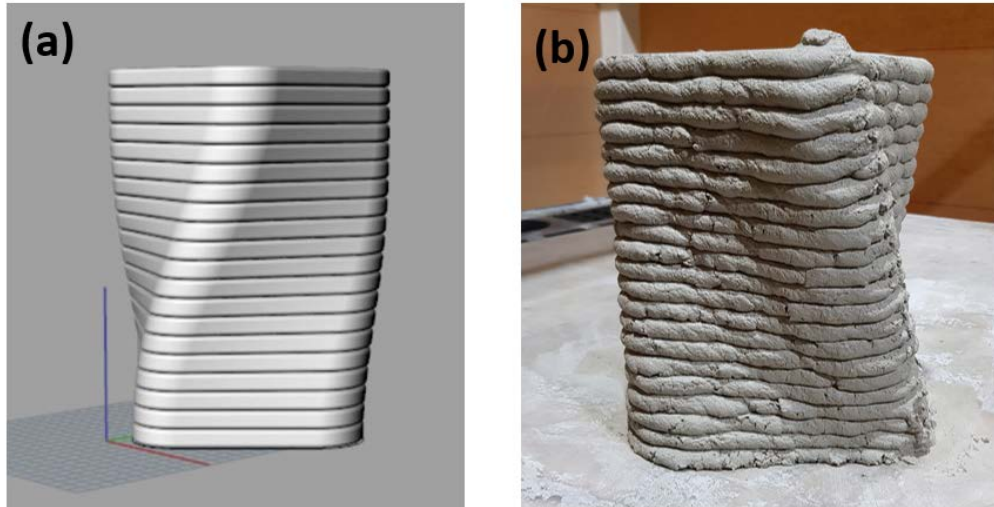
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308 **Fig. 9** Effect of nucleation seeds on the yield stress of AAS mixes containing 0.4% NC, obtained
309 just after mixing
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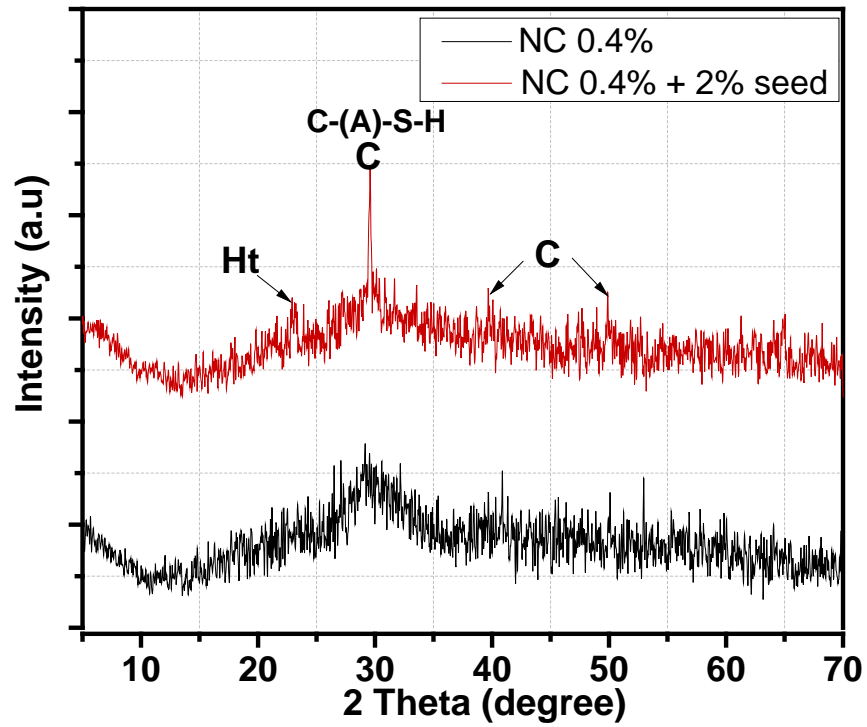
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313 **Fig. 10** 3D twisted column generated by using the AAS mix containing 0.4% NC and 2%
314 hydromagnesite seeds, showing: (a) 3D CAD model and (b) actual printed structure
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317 **3.4 Microstructural characterization**

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319 The feasibility of using the developed AAS mix in large-scale printing applications was further
320 evaluated via the assessment of its microstructural properties Fig. 11 presents the XRD patterns of
321 the AAS mix containing 0.4% NC and 0-2% hydromagnesite seeds after 14 days of curing. The
322 main hydrate phase observed in both samples was C-(A)-S-H gel (i.e. $30-31^\circ 2\theta$; PDF #00-033-
323 0306), which was in line with the findings of previous studies [37, 38]. A comparison of the XRD
324 patterns of both samples revealed the higher intensity of C-(A)-S-H in the seeded sample when
325 compared with the unseeded sample. Along with C-(A)-S-H, a small amount of calcite (CaCO_3 ;
326 PDF #01-071-3699) and hydrotalcite ($\text{Mg}_6\text{Al}_2\text{CO}_3(\text{OH})_{16}\cdot 4(\text{H}_2\text{O})$) were also observed.
327 Furthermore, the microstructures of both samples, shown in Fig. 12, revealed the denser
328 microstructure of the seeded sample in comparison to the unseeded sample. The wide-spread
329 growth of C-(A)-S-H gel on the surfaces of the rosette-like hydromagnesite particles confirmed

330 the role of these seeds in providing additional nucleation sites for the increased precipitation of
331 hydration products, which could lead to fast setting and improved early mechanical performance
332 [36, 40].

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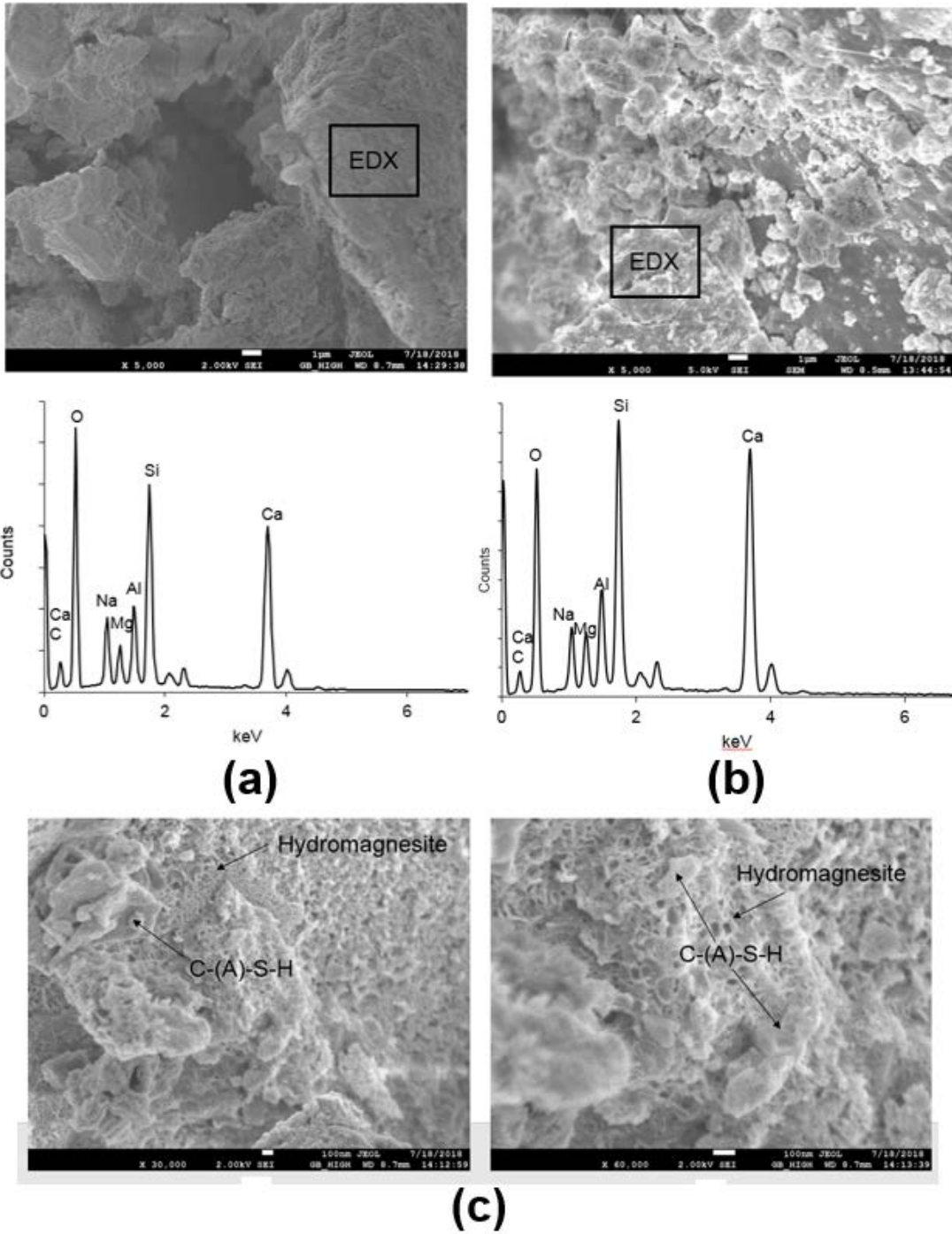


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336 **Fig. 11** XRD patterns of the AAS mix containing 0.4% NC and 0-2% hydromagnesite seeds after
337 14 days of curing (Ht: hydrotalcite and C: calcite)

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Fig. 12 SEM images and elemental composition of the AAS mix containing 0.4% NC and (a) 0% seeds and (b)-(c) 2% seeds

345 **4. Conclusions**

346

347 This paper aimed to improve the rheological properties of alkali-activated slag (AAS) binders for
348 extrusion-based 3D printing. While slag was used as the main binder, different amounts of
349 nanoclay (NC) were introduced to enable the development of extrudable and buildable (i.e.
350 printable) mixes. Selected mixes were subjected to a range of rheological measurements to
351 determine their yield stress, viscosity and thixotropy recovery required for 3D printing
352 applications. The AAS mix by itself was found to exhibit a low yield stress due to the plasticizing
353 and deflocculating effects of the silicate particles. Inclusion of 0.4% NC in these AAS mixes
354 significantly improved the initial yield stress due to the flocculation effect. However, structural
355 rebuilding rate remained unaffected even in the presence of NC and this was resolved via the
356 addition of 2% hydromagnesite seeds, which can increase the rate of the hydration reaction and
357 early strength development of AAS mixes, contributing to the buildability property necessary for
358 large-scale concrete printing.

359

360 While this study has paved the way for the development of alternative binders with improved
361 properties for 3D printing applications, further studies focusing on the optimization of the mix
362 design in terms of the NC content and seed type could enable the end users to fully harvest the
363 benefits of each additive in creating sustainable mixes with an enhanced performance.

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