

The deposition of strontium and zinc Co-substituted hydroxyapatite coatings

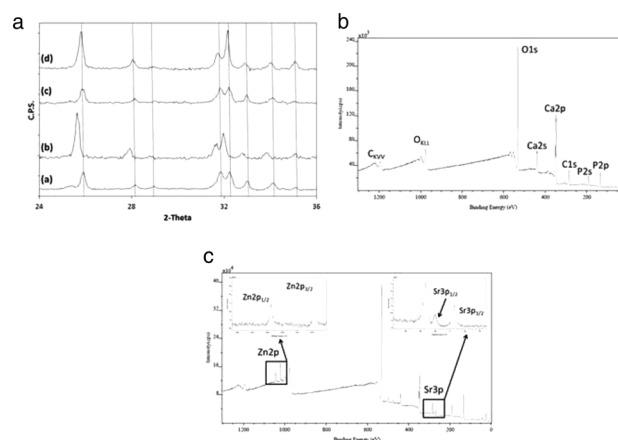
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Abstract The in vitro and in vivo performance of hydroxyapatite (HAp) coatings can be modified by the addition of different trace ions, such as silicon (Si), lithium (Li), magnesium (Mg), zinc (Zn) or strontium (Sr) into the HAp lattice, to more closely mirror the complex chemistry of human bone. To date, most of the work in the literature has considered single ion-substituted materials and coatings, with limited reports on co-substituted calcium phosphate systems. The aim of this study was to investigate the potential of radio frequency magnetron sputtering to deposit Sr and Zn co-substituted HAp coatings using Fourier transform infrared spectroscopy (FTIR), X-ray diffraction (XRD) and X-ray photoelectron spectroscopy (XPS). The FTIR and XPS results highlight that all of the Sr, Zn and Sr-Zn co-substituted surfaces produced are all dehydroxylated and are calcium deficient. All of the coatings contained HPO_4^{2-} groups, however; only the pure HAp coating and the Sr substituted HAp coating contained additional CO_3^{2-} groups. The XRD results highlight that none of the coatings produced in this study contain any other impurity CaP phases, showing peaks corresponding to that of ICDD file #01-072-1243 for HAp, albeit shifted to lower 2θ values due to the incorporation of Sr into the HAp lattice for Ca (in the Sr and Sr-Zn co-substituted surfaces only). Therefore,

the results here clearly show that RF magnetron sputtering offers a simple means to deliver Sr and Zn co-substituted HAp coatings with enhanced surface properties.

Graphical Abstract (a) XRD patterns for RF magnetron sputter deposited hydroxyapatite coatings and (b)-(d) for Sr, Zn and Sr-Zn co-substituted coatings, respectively. The XPS spectra in (b) confirms the presence of a HA sputter deposited coating as opposed to (c) XPS spectra for a Sr-Zn co-substituted sputter deposited coating.



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1 Introduction

Hydroxyapatite (HAp, $\text{Ca}_{10}(\text{PO}_4)_6(\text{OH})_2$) is a highly valuable bone repair and regeneration material because of its similarity to the inorganic phase of human bone. It is bioactive, osteoconductive and has the ability to form a direct chemical bond with human bone [1–3]. In addition to

the bulk form, HAp has been widely used as a coating material for orthopaedic implants in a variety of applications, and typically the release of Ca^{2+} and PO_4^{3-} ions into the surrounding environment leads to the development of excellent interfacial strength between the implant and bone [3–6]. However, chemically pure synthetic HAp has a Ca/P ratio of 1.67, whereas biological apatite has variable stoichiometry and can contain various different calcium phosphate (CaP) phases and ionic substitutions, resulting in a non-stoichiometric multi-element substituted material [7]. As such, when designing CaP biomaterials for clinical use one proposed route would be to introduce ionic substitutions into HAp in order to mimic the complex chemistry of human bone and thereby improve the biological performance of such materials, both in vitro and in vivo [8, 9].

A wide range of different ions such as Sr, Zn, Ag, Mg, CO_3^{2-} , F and Si (amongst others) have been strategically incorporated into the structure of synthetic HAp, replacing Ca (or PO_4^{3-}), as highlighted in Table 1 [10–48]. The substitution of such ions into pure HAp alters the crystal structure (crystallinity and crystal size) and also changes the properties of the material; namely the thermal and phase stability, its solubility, surface reactivity and its ability to be absorbed [49–51]. Particular attention is focused on several biologically active ions as Zn and Sr, which are naturally present in native bone tissue and can lead to advantageous effects on the biological response of osteoblasts. Sr and Zn both promote the osteoblast response and stimulate formation of new bone more readily than pure HAp [8, 9, 51, 52]. Sr is a beneficial trace element in human body and is shown to play an important role in bone formation. Sr stimulates a range of biological effects on the bone, such as anti-resorptive activity, osteoclast apoptosis and osteoblast stimulation [6, 8, 9, 49, 53–55]. These effects increase bone formation and reduce bone resorption, as well as improving bone strength, bone healing and microarchitecture and could potentially help in the treatment of osteoporosis [54, 56–58]. Zn is known to be an essential trace element in the body [9]. Zn is known to stimulate bone formation [8, 9, 59], but it also inhibits bacterial growth at the surgical site and improves the wound healing process [8, 52]. In addition, Zn also is known for its anti-inflammatory properties within the human body [58, 59], and its deficiency has been reported to reduce bone density and ductility, resulting in an increased probability of fracture [8, 9].

To date, there is a significant body of research dealing with single ion substituted CaP materials (as highlighted in Table 1). However, there are increasing numbers of reports on the use of co-substituted materials systems in both bulk powder form and coatings [10, 13, 14, 24, 26, 32, 58–62]. It has been suggested in the literature that co-substituted CaP systems may add significant value when considered for use in vivo (when compared to pure HAp and single ion

Table 1 Examples of different ions that have been substituted in CaP materials

Ion	Substitutes for	References
Cu^{2+}	Ca^{2+}	[10, 11]
Ag^+	Ca^{2+}	[12, 13]
Mg^{2+}	Ca^{2+}	[12, 13]
Zn^{2+}	Ca^{2+}	[10, 14]
Sr^{2+}	Ca^{2+}	[7, 14]
SiO_4^{4-}	PO_4^{3-}	[15, 16]
La^{3+}	Ca^{2+}	[17, 18]
Al^{3+}	Ca^{2+}	[17, 19]
$\text{Fe}^{2+/3+}$	Ca^{2+}	[14, 17]
F^-	OH^-	[10, 20]
SeO_3^{2-}	Ca^{2+}	[21]
Ce^{3+}	Ca^{2+}	[22]
Cl^-	OH^-	[23, 24]
Na^+	Ca^{2+}	[25, 26]
K^+	Ca^{2+}	[27, 28]
Ba^{2+}	Ca^{2+}	[29, 30]
CO_3^{2-}	OH^- or PO_4^{3-}	[26, 31]
Mn^{2+}	Ca^{2+}	[32]
Co^{2+}	Ca^{2+}	[33, 34]
Ga^{3+}	Ca^{2+}	[35]
Ti^{4+}	Ca^{2+}	[36, 37]
Eu^{3+}	Ca^{2+}	[38]
Nb^{5+}	PO_4^{3-}	[39]
Cd^{2+}	Ca^{2+}	[40, 41]
Y^{3+}	Ca^{2+}	[40, 42]
Li^+	Ca^{2+}	[43]
SO_4^{2-}	PO_4^{3-}	[44]
Ta^{5+}	Ca^{2+}	[45]
$\text{Sm}^{2+/3+}$	Ca^{2+}	[46]
Gd^{2+}	Ca^{2+}	[46]
In^{3+}	Ca^{2+}	[47]
Bi^{3+}	Ca^{2+}	[47]
Zr^{4+}	Ca^{2+}	[48]

substituted systems). A number of different deposition methods have already been investigated to produce co-substituted Ca-P coatings, including electrochemical deposition [19, 34], plasma spraying [63], sol-gel methods [32] and co-blasting [64]. Of the alternative coating technologies available, radio frequency (RF) magnetron sputtering has been shown to be particularly useful for the deposition of CaP coatings due to the ability of the technique to provide greater control of the coating's properties and improved adhesion between the substrate and the coating [65–70]. With respect to producing co-substituted CaP coatings, RF magnetron sputtering allows co-deposition of multiple different target materials

simultaneously and therefore provides an alternative and simple method to produce co-substituted CaP coatings [71–73]. Several reports in the literature highlight how it has already been used to deposit Sr or Si substituted HAp coatings [70–73]. However, to date, to the best of the authors' knowledge, there is no available information on the utilisation of RF magnetron sputtering as a method to deposit Sr and Zn co-substituted CaP surfaces, with the primary objective of creating a surface whereby the concentrations of the different ion substitutions can be carefully controlled in line with requirements.

The present work was undertaken in order to study the deposition of Sr and Zn co-substituted Ca-P coatings from a custom designed RF magnetron sputtering facility utilising two sputtering targets (referred to as sources). Sr and Zn were chosen for this study as their combined application in a CaP coating could add significant value to such surfaces when compared to applying the ion substitutions separately. In particular, the influence of different target configurations on the properties of the Sr and Zn co-substituted sputter deposited coatings produced at a low discharge power level (150 W) were investigated. A low discharge power level was chosen for this study as the quality and consistency of the targets used could be guaranteed throughout the sputter deposition runs. All of the coatings produced were characterised after post-deposition annealing to 500 °C using Fourier Transform Infrared Spectroscopy (FTIR), X-Ray Diffraction (XRD) and X-Ray Photoelectron Spectroscopy (XPS).

2 Materials and methods

2.1 Synthesis and characterization of pure and ion-substituted HAp powders

HAp and HAp substituted with 10wt% Sr ([Sr/Ca + Sr]) and 10wt% Zn ([Zn/Ca + Zn]) powders were synthesized through a standard neutralization route. The experimental ratios of Ca/P or (Ca + M)/P (where M = Sr or Zn) were maintained at 1.67. Starting suspensions were prepared by dissolving Ca(NO₃)₂·4H₂O (puriss. p.a., Sigma-Aldrich, Germany) and Sr(NO₃)₂ (puriss. p.a., Sigma-Aldrich, Germany) or Zn(NO₃)₂·6H₂O (98%, reagent-grade, Sigma-Aldrich, Germany) in deionised H₂O. An aqueous solution of H₃PO₄ (75%, puriss, Sigma-Aldrich, Germany) was utilised as a source of (PO₄³⁻) and was added to the starting suspensions at a slow addition rate (~0.75 ml/min). The pH of the synthesis mixture was kept around 9, using 25% ammonia solution. The synthesis temperature was kept constant at 45 °C. The precipitates obtained were left overnight to mature, filtered, and then washed with deionised H₂O. Finally, the precipitates were dried at 105 °C for

~24 h, milled and selectively sieved to obtain powders with particle size less than 100 µm. The HAp powders were labelled HAp, 10SrHAp and 10ZnHAp, respectively.

The chemical constitutions of the prepared powders were determined by atomic absorption spectrometry (AAS) and energy dispersive X-ray spectrometry (EDX) utilising the following equipment: Varian SpektrAA 880 and Mira/LMU/Inca Energy 350, respectively. The molar ratios of (Ca + Sr)/P and (Ca + Zn)/P were calculated from the results obtained. Different functional groups were determined using Fourier transform infrared spectroscopy (FT-IR, Varian 800, Scimitar Series). The spectra were recorded in the range of 400–4000 cm⁻¹ with spectral resolution 4 cm⁻¹ with 30 scans co-added. The phase composition and crystallinity was analyzed by X-ray diffraction (XRD) utilising a Rigaku Ultima+ XRD system operating at 40 KV and 30 mA using Cu Kα (λ = 1.5406 Å) radiation. Scans was recorded over the range from 2θ = 5° to 60°. For the phase identification the International Centre for Diffraction Data (ICDD) was used (card #01-072-1243 for HAp). The Brunauer–Emmett–Teller (BET) method was used to determine the specific surface area (SSA) of precipitated powders by N₂ adsorption (Quadrachrome SI Kr, Quantachrome). The samples were degassed at 40 °C for 24 h prior the analyses. The values of mean sizes of particles of as-synthesized powders were estimated from the N₂ adsorption isotherms using the BET particle diameter (*d*_{BET}) from the following Eq. (1) by assuming the primary particles to be spherical:

$$d_{BET} = \frac{6}{(q \cdot SSA)}, \quad (1)$$

where *q* is theoretical density of HAp—3.16 g/cm³ (according to ISO 13175-3:2012 Implants for surgery—Calcium phosphates—Part 3: Hydroxyapatite and beta-tricalcium phosphate bone substitutes).

2.2 Production and characterization of pure and ion-substituted HAp coatings

For this study coupons of chemically pure titanium (cpTi), Titanium International Ltd. (15 mm × 15 mm × 0.5 mm) were abraded using a succession of 800, and 1200 grade SiC papers. The coupons were twice sonicated for 10 min consecutively in acetone, isopropyl alcohol and distilled deionised water. The abraded coupons were then dried thoroughly in a convection oven at 70 °C for 12 h. Radio Frequency (RF) magnetron sputtering of the HAp and substituted HAp coatings was performed using a cluster of two high vacuum Torus 3 M sputtering sources in a custom designed system (Kurt J. Lesker Ltd, USA) each operating with a 13.56 MHz RF generator and an impedance matching network (Huettinger, GmbH, Germany). The coating

nomenclature was defined as highlighted in Table 2. After sputter deposition, the Ca-P coatings were thermally annealed. The samples were subjected to a ramp rate of 5 °C per minute to 500 °C (from room temperature) with a soak time of 2 h and a ramp rate of 5 °C per minute back down to room temperature.

Fourier transform infrared (FTIR) spectroscopy of the samples was carried out using a BIORAD FTS 3000MX Excalibur series instrument with a PIKE diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS) accessory. X-ray diffraction (XRD) of the samples was carried out using a Bruker D8 Discover Diffractometer fitted with a Gobel Mirror. A Cu K α X-ray radiation ($\lambda = 1.540 \text{ \AA}$) source was employed. For the grazing incidence angle XRD studies of the Ca-P coatings on the titanium coated silicon substrates the tube angle was set to 0.75°. Scans were recorded between $2\Theta = 24^\circ$ to 36° . X-ray photoelectron spectroscopy (XPS) of the samples were carried out using a Kratos Axis Ultra DLD spectrometer. Spectra were recorded by employing monochromated Al K α X-rays ($h\nu = 1486.6$ electron volts (eV)). Sample charging effects on the measured BE positions were corrected by setting the lowest BE component of the C1s spectral envelope to 285.0 eV, i.e., the value generally accepted for adventitious carbon surface contamination [68]. Photoelectron spectra were further processed by subtracting a linear background and using the peak area for the most intense spectral line of each of the detected elemental species to determine the % atomic concentration. Three areas were analysed for each sample.

3 Results

3.1 Powder results

The main characteristics of the precursor powders are summarised in Tables 3 and 4. The experimental amounts

Table 2 Target configuration for sputter deposition

Coating nomenclature	Target 1 powder	Target 2 powder
HAp	HAp	HAp
SrHAp	10SrHAp	10SrHAp
ZnHAp	10ZnHAp	10ZnHAp
Sr-ZnHAp	10SrHAp	10ZnHAp

Table 3 Chemical analysis of the Sr- and Zn-substituted HAp powders. *M – substituting cation (Sr for Sr-substituted HAp, Zn for Zn-substituted HAp)

Powder Sample	M/(Ca+M) in starting solution	M concentration measured by AAS, wt. %	M/(Ca+M) in solid products
10ZnHAp	0.10	7.10 ± 0.60	0.086
10SrHAp	0.10	5.40 ± 0.50	0.070

of Zn and Sr in the powders, evaluated through AAS are reported as a function of the Zn and Sr content in the starting solution. It was found that not all of into Sr and Zn synthesis media added were substituted into the lattice of HAp. It implies that some of Zn and Sr ions remain in the precursor solution after precipitation. The differences between the theoretical and measured Sr and Zn concentrations might be due to Sr or Zn complexes $\text{Sr}(\text{NH}_3)_4^{2+}$ or $\text{Zn}(\text{NH}_3)_4^{2+}$ formed with the ammonia used for adjusting pH of the synthesis media [74]. Thus, Sr and Zn substitutions were inhibited, and the resulting materials were Ca-deficient. Given that the FTIR and XRD results (discussed below) did not show the presence of any by-products in the as-synthesized powders, it is most likely that the complexes were removed by washing the precipitate. Notably, the ratio between the two cations Sr/(Ca + Sr) and Zn/(Ca + Zn) in the solid products are slightly lower than those in the precursor solutions. EDX results, as reported in Table 4, shows that 10SrHAp and 10ZnHAp powders obtained by the precipitation processes have a (Ca+Sr)/P and (Ca + Zn)/P ratio < 1.67. Thus, the powders were found to be sub-stoichiometric apatite [75]. The Ca/P ratio of the HAp powder was 1.69, slightly higher than that expected for stoichiometric HAp [1, 68].

FTIR spectra of the as-synthesized HAp and Sr-, Zn-substituted HAp powders are shown in Fig. 1a–c, respectively. All of them show the characteristic absorption bands expected for the HAp powder. The absorption peaks observed at 3570 and 633 cm^{-1} are assigned to the vibrational modes of the [OH] group [62, 68]. However, the resolution and intensity of the absorption peak of the structural [OH] group at 633 cm^{-1} observed in the as-synthesized 10SrHAp and 10ZnHAp, were significantly diminished when compared to the HAp powder, highlighting the loss of OH^- ions from the unit cell and the

Table 4 Stoichiometry analysis of the HAp and the Sr- and Zn-substituted HAp powders.

Sample	Expected values		Calculated values	
	(Ca+M*)/P	Ca/P	(Ca+M*)/P	Ca/P
HAp	–	1.67	1.69	1.69
10SrHAp	1.67	1.59	1.53	1.46
10ZnHAp	1.67	1.54	1.51	1.41

*M substituting cation (Sr for Sr-substituted HAp, Zn for Zn-substituted HAp)

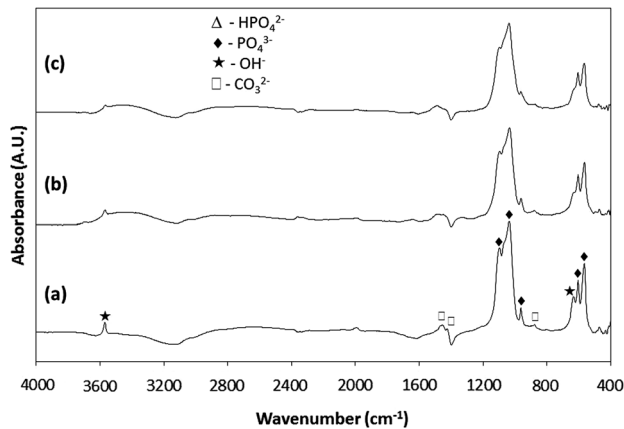


Fig. 1 FT-IR spectra of the as-synthesized **a** HAp, **b** 10SrHAp and **c** 10ZnHAp powders

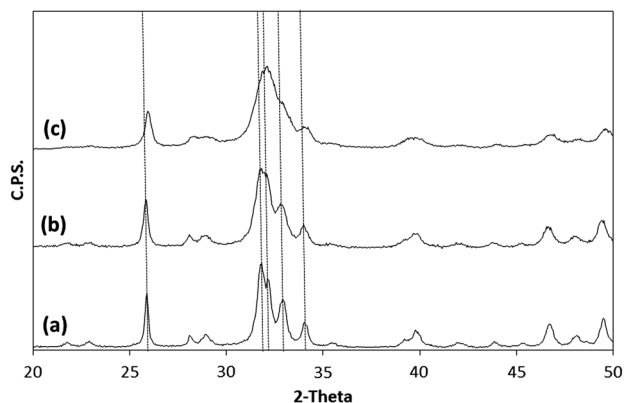


Fig. 2 XRD patterns of as-synthesized **a** HAp, **b** 10SrHAp and **c** 10ZnHAp powders

apparent substitution of Sr or Zn into the lattice of HAp [55]. The absorption band observed from 1100 to 1000 cm^{-1} and absorption peaks at 906, 602, 565 and 475 cm^{-1} are assigned to the $[\text{PO}_4]$ groups [62, 69]. The broad (weak) stretching band from 3600 to 3400 cm^{-1} and a bending peak at 1620 cm^{-1} are attributed to the adsorbed H_2O molecules [60, 62, 66]. Small peaks related to $[\text{CO}_3]$ groups were detected at 1450, 1414 cm^{-1} and 875 cm^{-1} [72, 73, 76]. The peak observed at 875 cm^{-1} may also be associated with HPO_4^{2-} groups. [72, 73]. However, as no other HPO_4^{2-} were detected in these synthesised materials, the peak at 875 cm^{-1} is more likely a consequence of related to CO_3^{2-} groups. The XRD results for the as-synthesized HAp and Sr-, Zn-substituted powders as shown in Fig. 2a–c, respectively. The peak positions and relative peak intensities for all of the synthesised powders correspond closely to those indicated in the International Centre for Diffraction Data (ICDD) file #01-072-1243 for HAp [77]. It was notable that all the synthesised powders exhibited significant peak broadening, however, no additional CaP

Table 5 BET data of the as-synthesized powders

Sample	SSA, m^2/g	d_{BET} , nm
HAp	55.67 ± 0.58	34.11 ± 0.04
10SrHAp	70.31 ± 0.59	27.00 ± 0.02
10ZnHAp	95.80 ± 1.28	19.82 ± 0.03

phases were detected in any of the diffraction pattern obtained. In the case of the 10SrHAp substituted powder, the peak positions in the diffraction pattern are shifted to slightly lower 2-theta (2θ) values, whereas, the peak positions are relatively unchanged for the 10ZnHAp powder. The specific surface area (SSA) of precipitated powders (m^2/g) was observed to increase significantly with the substitution of Ca by both Sr and Zn in the lattice, as highlighted in Table 5. This corresponds to the expected decrease in the estimated mean microcrystallite size (d_{BET} , nm) for the substituted materials when compared to the HAp powder, as reported in Table 5. These would be in line with expectations for these materials.

3.2 Coating results

The CaP coatings produced from the HAp, and the different 10SrHAp and 10ZnHAp target combinations were analysed using FTIR, XRD and XPS, to assess the surface properties after thermal annealing at 500 °C. The as-deposited CaP and substituted CaP coatings were not analysed in this study as previous work by the authors has shown that these surfaces are amorphous and require thermal annealing in order to produce crystalline coatings [67–69, 72, 73]. Therefore, all coatings that are described here are those that have been annealed to 500 °C only.

In the first instance, the FTIR spectrum of the coatings produced from the HAp targets were indicative of crystalline HAp, as shown in Fig. 3a. Well resolved P-O stretching vibrations were observed as expected between 1100–950 cm^{-1} . O-P-O bending vibrations are also present between 620–560 cm^{-1} , as shown in Table 6 [67–69]. Hydrogen phosphate bands (HPO_4^{2-}) can also be observed at 1117 and 584 cm^{-1} [72, 73]. A weak O-H librational band is observed around 632 cm^{-1} as a very weak shoulder, with a further peak associated with O-H groups observed at 3568 cm^{-1} . This peak may be associated with O-H stretching groups within the film [67–69]. The absence of strong OH functional groups, commonly observed at approximately 632 cm^{-1} and 3568 cm^{-1} , indicate a degree of dehydroxylation within the CaP crystal structure deposited from the HAp targets under the conditions employed here. Furthermore, weak CO_3^{2-} bands are observed between 1550–1400 cm^{-1} and 820–890 cm^{-1} [72, 76]. The peak observed at 584 cm^{-1} may also be due to the presence of HPO_4^{2-}

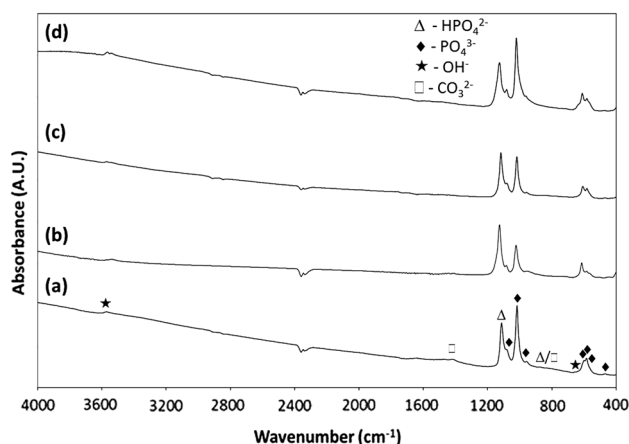


Fig. 3 FTIR spectra for **a** HAp coating, **b** SrHAp coating, **c** ZnHAp coating and **d** Sr-ZnHAp coating (all annealed to 500 °C)

Table 6 FTIR peak assignments for the HAp, SrHAp, ZnHAp and Sr-ZnHAp co-substituted coatings

Peak assignment	HAp	SrHAp	ZnHAp	Sr-ZnHAp
HPO ₄ ²⁻ (P-O-H)	1117	1122	1120	1124
PO ₄ ³⁻ (P-O)	1080	1082	1078	1079
	1018	1020	1020	1020
PO ₄ ³⁻ (P-O)	961	961	962	964
PO ₄ ³⁻ (O-P-O)	601	611	605	605
HPO ₄ ²⁻ (P-O-H)	584	584	584	586
PO ₄ ³⁻ (O-P-O)	566*	565*	566*	565*
PO ₄ ³⁻ (O-P-O)	471***	475***	470***	471***
CO ₃ ²⁻ (C-O)	1550–1400**	1550–1400**	–	–
CO ₃ ²⁻ (O-C-O)/	875***	880***	–	–
HPO ₄ ²⁻ (P-O-H)				
OH ⁻ (O-H)	3568	3570	3569	3570
OH ⁻ (O-H)	632**	–	–	631*

*weak shoulder, **very weak shoulder, ***weak band

groups [72, 73]. For those coatings produced from the two 10SrHAp target configurations (SrHAp), the phosphate bands (PO₄²⁻) and hydrogen phosphate bands (HPO₄²⁻) bands are observed in largely similar peak positions as would be expected for SrHA, with shifting to slightly higher wavenumbers for most peaks observed [71–73]. Again, the OH stretching vibration at 3570 cm⁻¹ are poorly resolved and the OH⁻ librational band typically expected at around 632 cm⁻¹ is absent as can be seen in Fig. 3b and Table 6. Similar results are also seen for the ZnHAp coatings as was observed for the SrHAp coatings. However, no obvious peaks for carbonate or the OH⁻ librational vibration can be detected, as shown in Fig. 3c and Table 6. The co-substituted Sr-ZnHAp coating FTIR spectrum is again very similar to those observed for the SrHAp and ZnHAp surfaces, however, a weak shoulder indicative of OH⁻

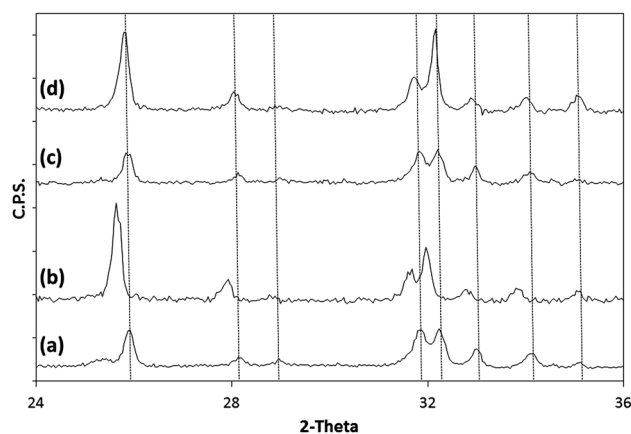


Fig. 4 XRD patterns for **a** HAp coating, **b** SrHAp coating, **c** ZnHAp coating and **d** Sr-ZnHAp coating (all annealed to 500 °C)

librational band is again present at around 631 cm⁻¹, as shown in Fig. 3d and Table 6.

The peaks observed in the XRD patterns for the Ca-P coatings derived from the HAp targets, as shown in Fig. 4a, correspond closely to those in the ICDD file #01-072-1243 for HAp. Furthermore, the intensity of the 002 reflection at 25.9° 2θ is higher than would be expected for a HAp coating, which suggests that this coating may have a 002 preferred orientation and would be line with expectations for coatings deposited under the conditions employed here [72, 73]. The XRD pattern for the SrHAp coatings is shown in Fig. 4b, however, as expected for a Sr-substituted material, the peak positions are shifted to slightly lower 2-theta (2θ) values [71–73]. No significant peak broadening was observed for this particular coating. However, there was a notable change in the relative peak intensities of the 211 and 112 peaks with the addition of Sr to the coatings. For the ZnHAp coatings, the peak positions and intensities are very similar to those obtained for the HAp coatings produced here, with an obvious preferred 002 orientation again observed as shown in Fig. 4c. The co-substituted Sr-ZnHAp coating XRD pattern, as shown in Fig. 4d is again very similar to those observed for the SrHAp coatings, with a pronounced 002 preferred orientation and a clear difference in the relative peak intensities of the 211 and 112 peaks. In the case of all the HAp, SrHAp, ZnHAp and Sr-ZnHAp coatings, no significant amorphous background hump was observed. In addition to this no other Ca-P phases were detected in the XRD pattern for any of the coatings.

The XPS results for the SrHAp, ZnHAp and Sr-ZnHAp coatings are highlighted in Fig. 5a–d, respectively and Table 7. For the HAp coating, shown in Fig. 5a, peaks corresponding to Ca2s (438.6 eV), Ca2p_{3/2} (347.5 eV), Ca2p_{1/2} (351.0 eV), P2p (133.5 eV), P2s (191.0 eV), O1s (531.5 eV), O Auger_{KLL} (764.0 eV) and Ca Auger_{LMM}

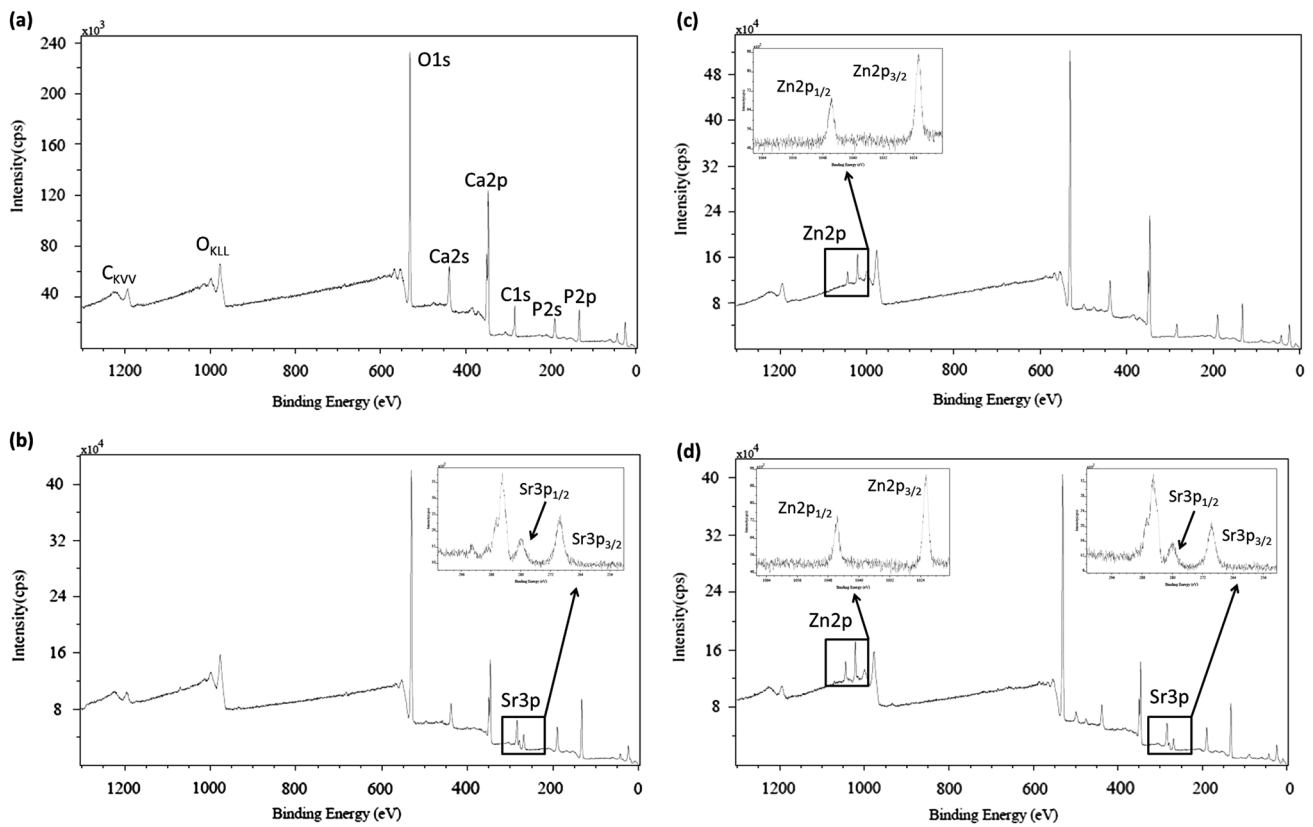


Fig. 5 XPS wide energy survey scans (WESS) for **a** HAp coating, **b** SrHAp coating, **c** ZnHAp coating and **d** Sr-ZnHAp coating (all annealed to 500 °C)

Table 7 Comparative XPS data for the different HAp and substituted HAp coatings

Sample	Peak positions – B.E. (eV)						Quantification ratios		
	C1s	O1s	Ca2p _{3/2}	P2p _{3/2} / Sr3d _{5/2}	Sr3p _{3/2}	Zn2p _{3/2}	Ca/P	Ca+M/P	Ca+M/Ca
HAp	285.0	531.5	347.5	133.5	–	–	1.60 ± 0.01	–	–
SrHAp	285.0	531.5	347.5	133.5	269.5	–	1.38 ± 0.19	(Sr) 1.45 ± 0.22	(Sr) 1.05 ± 0.02
ZnHAp	285.0	531.3	347.5	133.5	–	1022.5	1.16 ± 0.08	(Zn) 1.20 ± 0.08	(Zn) 1.03 ± 0.01
Sr-ZnHAp	285.0	531.1	347.5	133.5	269.4	1022.4	1.39 ± 0.12	(Sr) 1.47 ± 0.13 (Zn) 1.42 ± 0.12 (Sr+Zn) 1.50 ± 0.14	(Sr) 1.06 ± 0.01 (Zn) 1.02 ± 0.01 (Sr+Zn) 1.08 ± 0.01

*M substituting cation (Sr for Sr-substituted HAp, Zn for Zn-substituted HAp as indicated in the Table)

(964.0 eV) are clearly observed and correspond closely to those reported in the literature for HAp [67–69, 72, 73]. No other elemental species were detected at least at the detection limits of the instrument (~0.1 atomic % concentration). The Ca/P ratio of the custom synthesised HAp derived coating was calculated at 1.60 ± 0.01 and is slightly lower than that expected for stoichiometric HAp, however, it is line with expectations for coatings produced under the conditions employed here [67–69]. The peak positions highlighted in Fig. 5b and in Table 7 for the SrHAp coating were all in line with those expected for Sr-substituted HAp,

with clear evidence for the presence of Sr highlighted by the presence of the Sr3p_{3/2} at 269.5 and Sr3p_{1/2} at 280.1 eV [72, 73]. The Ca/P and Ca + Sr/P ratios for the SrHAp coatings were 1.38 ± 0.19 and 1.45 ± 0.22, respectively (as shown in Table 7). The spectral envelope observed at 133.5 eV consists of an overlap of the Sr3d and P2p peaks because the Sr3d_{5/2} (133 ± 0.5 eV), Sr3d_{3/2} (135 ± 0.5 eV) and the P2p (132–133 eV) lines are located so close together. For the ZnHAp coating, the presence of Zn in the coating is confirmed due to the presence of the Zn3p_{3/2} at 1022.5 and Zn3p_{1/2} at 1045.4 eV as shown in Fig. 5c [78]. The Ca/P

and Ca+Zn/P ratios for the SrHAp coatings were 1.16 ± 0.08 and 1.20 ± 0.08 , respectively (as shown in Table 7). In comparison, the wide energy survey scan for the Sr and Zn co-substituted HAp coating (Sr-ZnHAp), highlighted in Fig. 5d, shows the presence of peaks for both Sr (Sr3p_{3/2} at 269.4 and Sr3p_{1/2} at 280.1 eV) and Zn (Zn3p_{3/2} at 1022.4 and Zn3p_{1/2} at 1045.2 eV) [72, 73, 78]. The Ca/P, Ca + Sr/P, Ca + Zn/P and Ca + Sr + Zn/P ratios for the Sr-ZnHAp coatings were 1.39 ± 0.12 , 1.47 ± 0.13 , 1.42 ± 0.12 and 1.50 ± 0.14 , respectively. If the Ca + M/Ca ratios (where M = Sr or Zn substituting ion) are also compared for the different Sr and Zn substituted coatings (as shown in Table 7), it appears that the Sr levels are comparable between the SrHAp and Sr-ZnHAp surfaces. A similar trend was observed for the Zn levels between the ZnHAp and Sr-ZnHAp surfaces. For the Sr, Zn and Sr-Zn substituted coatings, no other elemental species were detected within the detection limits of the instrument.

4 Discussion

The present work was undertaken in order to study the deposition of Sr and Zn substituted and co-substituted CaP coatings onto titanium substrates using RF magnetron sputtering.

Before sputtering the HAp and 10SrHAp and 10ZnHAp target materials (both 10% substituted with Sr and Zn, respectively) were analysed using FTIR, XRD EDX and AAS. The purity, crystallinity and stoichiometry (Ca/P ratio of 1.69) of the HAp powder was as expected and confirmed the presence of a slightly calcium rich, carbonated HAp powder. Despite this, the XRD results highlighted that no other CaP phases were detected in the custom synthesised HAp powder. Due to the presence of the significant CO₃²⁻ peaks between 1550–1400 cm⁻¹ and at 875 cm⁻¹ for the HAp powders, this indicates substitution of both the OH⁻ and PO₄³⁻ groups by CO₃²⁻ (A-B type substitution) [76]. As no other significant HPO₄²⁻ peaks were observed in the FTIR spectra for the HAp powder, it would be suggested that the peak at 875 cm⁻¹ was due to the presence of CO₃²⁻ in this powder sample. Therefore, the results here would suggest the HAp powder formula to be Ca₁₀[(PO₄)_{6-x}(CO₃)_x][(OH)_{2-x}(CO₃)_x] [76]. By comparison, the purity and crystallinity of the 10SrHAp powder were largely as expected and confirmed the presence of significant levels of Sr-substitution within the HAp lattice. However, the EDX results clearly show that the 10SrHAp powder synthesised here have a (Ca+Sr)/P ratio of 1.53 (much less than the expected value of 1.67). This might be due to the generation of crystal defects from the ion substitution process [75]. The corresponding FTIR spectra highlighted that the presence of Sr in the HAp lattice resulted in significant peak broadening

with respect to the PO₄³⁻ peaks between 1200–900 cm⁻¹, which is indicative of Sr incorporation in the HAp lattice [72, 73]. The Sr-substituted HAp powder also became more dehydroxylated as a result of the presence of Sr, with the OH- stretching and librational bands in the FTIR spectra around 3570 cm⁻¹ and 632 cm⁻¹ becoming much less prominent. Peak broadening and dehydroxylation are indicative of Sr incorporation of the HAp lattice. [72, 73, 79]. This usually results in structural disorder along the *c-axis* and it is well documented that a lack of hydroxylation within the HAp lattice affects the degree of atomic ordering in HAp [80, 81]. Small peaks related to [CO₃] groups were also detected at 1450, 1414 cm⁻¹ and 875 cm⁻¹, which is indicative of CO₃²⁻ groups, which are partially substituting for the PO₄³⁻ groups in the structure of HAp (B-type carbonated HAp) [58, 76]. Despite this, no significant HPO₄²⁻ peaks were observed in the FTIR spectra for the 10SrHAp powders as might have been expected for a Sr-substituted CaP material [72, 73]. Typically, Sr substitution for Ca in the HAp lattice results in an increase in both CO₃²⁻ and HPO₄²⁻ groups (especially CO₃²⁻ groups) due to the increased lattice strain caused by the larger Sr ions [79, 82]. The XRD results for the Sr-substituted powder also highlighted that due to the incorporation of Sr into the HAp lattice, the peak positions shifted to slightly lower 2-theta (2θ) values. The peaks also exhibited significant broadening as a consequence of Sr substitution, as can be observed in Fig. 2. This would be in line with expectations and is due to substitutional strain in the lattice [83, 84]. The XRD results also highlighted that no other CaP phases were detected in any of the Sr-substituted powders. Interestingly, for the Sr-substituted HAp powder used here to produce the sputtering targets, no pronounced 002 preferred orientation was observed in the diffraction pattern. Previous studies on Sr-substituted apatites have shown that as the Sr content in HAp increases, so does *c-axis* orientation [71–73].

Similar observations are made for the 10ZnHAp powder when compared to that of the 10SrHAp powder, whereby the FTIR spectrum shows the powder material to be dehydroxylated (due to the diminished OH- peaks) and containing carbonate ions due to the presence of weak CO₃²⁻ at 1450, 1414 cm⁻¹ and 875 cm⁻¹, which is again indicative again of B-type carbonated HAp (albeit 10ZnHAp here). The 10ZnHAp powder is also calcium deficient, with EDX giving a Ca+Zn/P ratio of only 1.51, much lower than that expected in this case (as highlighted in Table 4). The XRD results for the 10ZnHAp powders were as expected, with peaks corresponding closely to that of HAp (ICDD File# 01-072-1243). No phase impurities were detected in the 10ZnHAp powder, however, significant peak broadening was observed and a typical random crystallite orientation was observed for both the 10SrHAp and 10ZnHAp powders synthesised here and consequently used to produce the

sputtering targets in this body of work. The XRD results for the 10SrHAp and 10ZnHAp powders corroborate those results highlighted from the EDX and FTIR analyses confirming that the 10SrHAp and 10ZnHAp powders contains appreciable levels of Sr and Zn, with Sr and Zn substituting for Ca in the HAp lattice. These results also suggest that the substituted powder synthesised for this study are non-stoichiometric, i.e. Ca-deficient apatite containing CO_3^{2-} substitutions, with CO_3^{2-} only substituting for PO_4^{3-} (B-type carbonated HAp) [75, 76]. In both cases this would suggest a possible $\text{Ca}_{10-x-y-z}(\text{M})_y(\text{PO}_4)_{6-x-z}(\text{CO}_3)_z(\text{OH})_{2-x}$, (where M is either Sr or Zn substituting for Ca in the HAp lattice), as this allows for both a Ca/P and Sr+Ca/P lower than 1.67. Further to this, these findings also indicate increasing structural disorder or reduction in crystallite sizes as a consequence of Zn or Sr substituting for Ca in the HAp lattice. It was reported that the incorporation of a small amount of Zn into the HAp structure induces an evident reduction in the degree of crystallinity of the apatite phase, which leads to the inability of the structure to host greater amounts of Zn [85]. These observations are supported by results of BET analysis, summarized in Table 5. The SSA measurements show a significant reduction of BET particle sizes due to the incorporation of Zn into the apatite crystal structure. However, in the case of 10SrHAp powders, the results of BET analysis show an increased particle size when compared to that of 10ZnHAp. It is well documented that the Zn, Ca, and Sr ionic radii are, 0.074, 0.099, and 0.112 nm, respectively [62, 74]. The difference between the Sr and Ca ionic radii is noticeably lower as is the difference between the Zn and the Ca ionic radii. Therefore, the Zn-containing apatite is characterized by considerably higher strains and distortions of the structure [75].

After sputtering the HAp, 10SrHAp and 10ZnHAp materials onto the titanium substrates (using various different target configurations, as highlighted in Table 2) distinct changes were observed in their surfaces properties as determined using FTIR, XRD, and XPS analyses. The coatings in each case were annealed to 500 °C after deposition due to the as-deposited coatings being amorphous in nature. This is a consequence of the low sputtering power employed in this study (150 W) [67–69, 72, 73]. Others have shown that crystalline coatings can be produced by increasing the RF power [65]. However, for the purposes of this study a lower deposition power is deliberately employed as coatings produced using low power have been shown to contain only hydroxyapatite and no other additional Ca-P phases or by-products [67–69]. This would be in line with expectations of the various international standards that cover hydroxyapatite powders and coatings (ASTM 1185-03(2014), ASTM 1609—(08)2014 and ISO 13779 (Parts 2 and 3)). Furthermore, no impurities were detected on the surface of any of these coatings (as

confirmed by the analyses undertaken here). The HAp derived coatings annealed to 500 °C has a Ca/P ratio of 1.60 ± 0.01 , which is slightly lower than would be expected for a sputter deposited coating under the conditions employed here [67–69]. However, it should be noted that previous studies where Sr was sputter deposited onto titanium surfaces, three sputtering targets rather than two (as utilised here) were employed, and the source HAp was custom synthesised, as compared to the commercially available HAp powders used previously by the authors [68–70, 72, 73]. This may account for the slightly lower coating stoichiometry observed here for the HAp coatings. The XRD results for the same coatings produced from the two HAp targets also highlighted the expected 002 preferred *c*-axis orientation, as determined by XRD analyses. Typically, 002 preferred orientation in HAp results in an elevated Ca/P ratio [82]. To add to this, the FTIR results highlight that these coatings produced from the HAp precursor powders contain trace amounts of CO_3^{2-} within the lattice, due to the presence of peaks between $1550\text{--}1400\text{ cm}^{-1}$ and 875 cm^{-1} , as highlighted in Fig. 3a and Table 6. There is evidence for the presence of significant HPO_4^{2-} functional groups due to the strong peaks at 1122 and 582 cm^{-1} (and possibly the peak at 875 cm^{-1}). Furthermore, the HAp derived coatings are also dehydroxylated. Typically, it is expected that if HAp is dehydroxylated, then the charge balance within the crystal structure must therefore be provided by other means. HAp is understood to undergo carbonate (CO_3^{2-}) substitution within the crystal lattice at multiple sites [76, 83]. Most notably, A type substitution (for OH^- groups), B type substitutions (for PO_4^{3-} groups), or both (AB) substitution are acknowledged as providing for charge balance in the lattice structure of HAp [76, 83]. The presence of $\text{Ca}_{10-x-y/2}[\text{HPO}_4]_x(\text{PO}_4)_{6-x-y}(\text{CO}_3)_y(\text{OH})_{2-x-y}$ species could be suggested for this coating, however, a combination of calcium deficient hydroxyapatite (CDHA) material $[\text{Ca}_{10-x}[(\text{PO}_4)_{6-x}(\text{HPO}_4)_x](\text{OH})_{2-x}]$ and A-B substituted materials cannot be ruled out $[\text{Ca}_{10}[(\text{PO}_4)_{6-x}(\text{CO}_3)_x][(\text{OH})_{2-x}(\text{CO}_3)_x]]$, which may help to explain the lower Ca/P ratio obtained for this surface via XPS analysis. Typically, CDHA results in a material that has a Ca/P ratio less than 1.67 [80].

In comparison, the XPS results for the SrHAp coatings were in line with expectations for a Sr-substituted HAp coating and are similar to those reported previously [72, 73]. Peaks for $\text{Sr}3p_{3/2}$ and $\text{Sr}3p_{1/2}$ can be clearly observed around 269.6 and 280.1 eV in addition to the expected peaks for a CaP material [72, 73]. Figure 5b shows a representative wide energy survey spectra and high resolution spectra for the SrHAp coatings. The Ca/P ratios (and corresponding Ca + Sr/P ratio) for the same coatings were 1.38 ± 0.19 and 1.45 ± 0.22 , respectively, as reported in Table 7. The Ca/P ratio reduces significantly for the SrHAp coatings when compared to the HAp surfaces, as might

have been expected with the incorporation of Sr for Ca in the HAp lattice. This is corroborated by the Ca+Sr/P ratio as reported in Table 7. Furthermore, the Sr content of the coatings deposited in previous studies were lower than that of the original precursor target material, and the same trend is observed in this study [72, 73]. The corresponding FTIR spectra highlighted that for the SrHAp coating there was significant peak broadening for the PO_4^{3-} peaks between 1200–900 cm^{-1} . These peaks are also seen to shift to slightly lower wavenumbers and are dehydroxylated, both characteristics which are indicative of Sr substituting for Ca in the HAp lattice [72, 73]. CO_3^{2-} peaks are also detected in the FTIR spectrum for this coating, with the peaks positions observed indicating substitution of both the OH^- and PO_4^{3-} groups by CO_3^{2-} (A-B type substitution) [76]. Further to this, HPO_4^{2-} bands were also clearly observed in the FTIR spectra for SrHAp coatings. It is expected that as Sr substitutes for Ca in the HAp lattice an increase in both CO_3^{2-} and HPO_4^{2-} groups (especially CO_3^{2-} groups) should be observed, due to the increased lattice strain caused by the larger Sr ions as highlighted previously [86, 87]. The corresponding XRD results for SrHAp coatings highlight that they have peak positions that are shifted to slightly lower 2-theta (2θ) values with addition of Sr, as can be observed in Fig. 4. In addition the XRD results also show that the SrHAp coatings exhibit significant preferred 002 *c-axis* orientation, when compared to the HAp coatings and the 10SrHAp precursor powder. The intensity of the 002 peak (and therefore the 002 *c-axis* orientation) also appears to increase relative to the other peaks (namely the 211 peak). There is also a change in the relative peak intensities of the 211 and 112 peaks with increasing Sr content, as shown in Fig. 4b for the SrHAp surfaces. These observations would be in line with previous studies on Sr-substituted HAp, whereby it has been shown that as the Sr content in HAp increases, so does *c-axis* orientation [72, 73]. Considering all the results obtained here, this could suggest a formula such as $\text{Ca}_{10-x-y/2}[\text{HPO}_4]_x(\text{PO}_4)_{6-x-y}(\text{CO}_3)_y(\text{OH})_{2-x-y}$ for the SrHAp coating, given the presence of both the CO_3^{2-} and HPO_4^{2-} groups, the apparent dehydroxylation of the coatings and the low Ca/P and C+Sr/P ratios.

In comparison, the XPS results for the ZnHAp surfaces confirmed the presence of Zn in these coatings as highlighted by WESS and Table 7. The Ca/P ratios (and corresponding Ca+Sr/P ratio) for the same coatings were 1.16 ± 0.08 and 1.20 ± 0.08 , respectively, as reported in Table 7. The Ca/P ratio is significantly lower for the ZnHAp coatings when compared to both the HAp and SrHAp surfaces. The FTIR results for the ZnHAp coatings indicate that the coatings are dehydroxylated, have obvious HPO_4^{2-} groups, but in this case, contain no real trace of any CO_3^{2-} groups, as was observed for the HAp and SrHAp coatings. The XRD pattern obtained for this coating type is largely similar

to that of HAp, with some preferred orientation observed, albeit to the lesser extent than the SrHAp surfaces. The preferred orientation observed in sputter deposited coatings, such as those deposited here, is well documented and in this case for the ZnHAp coatings the preferred orientation may be a consequence of the deposition parameters employed, as opposed to the incorporation of Zn into the HAp lattice [67–69]. No evidence for preferred orientation in Zn substituted apatites has been found in the previous literature [78, 85]. Some do report shifting of the peaks to higher 2θ values in the XRD patterns for Zn substituted materials; however, that was not observed here [88]. Typically the substitution of Zn into HAp results in a decrease of both *a* and *c* lattice parameters; however, varying points of view on this do exist in the literature [78, 85] and this is beyond the scope of this particular study. Therefore, the results here would indicate that the formula for the ZnHAp coating may be $\text{Ca}_{10-x-y}(\text{Zn})_y(\text{HPO}_4)_x(\text{PO}_4)_{6-x}(\text{OH})_{2-x}$, given that this allows for low Ca/P and Ca+Zn/P ratios and the apparent lack of CO_3^{2-} in the lattice. Furthermore, the apparent lack of hydroxylation, as observed in the FTIR spectrum, provide clear evidence (along with the XRD and XPS) that Zn has substituted for Ca in the ZnHAp coatings.

The results for the co-substituted Sr-ZnHAp coatings show many similarities to those obtained for both SrHAp and ZnHAp surfaces and indicate that both Sr and Zn have been successfully substituted for Ca in the Sr-ZnHAp lattice. The reported Ca/P ratio from the XPs analyses was 1.39 ± 0.12 , with Ca+Sr/P and Ca + Zn/P ratios of 1.47 ± 0.13 and 1.45 ± 0.12 , respectively, as reported in Table 7. These results taken in combination with the Ca + Sr + Zn/P ratio of 1.50 ± 0.14 clearly and the WESS in Fig. 5d suggests the presence of both Sr and Zn in the co-substituted coatings. Further evidence of the successful co-substitution of the Sr-ZnHAp coatings is provided by the apparent lack of hydroxylation in the FTIR results, as highlighted in Fig. 3d and Table 6. Again, as with the Zn-HAp coatings, the co-substituted surfaces have obvious HPO_4^{2-} groups in the FTIR spectra but no real evidence of any CO_3^{2-} groups. The corresponding XRD pattern in Fig. 4d also shows shifting of the peaks to slightly lower 2θ values, along with strong 002 preferred orientation, which has been shown to be a consequence of Sr substituting for Ca in the lattice. This, along with the changes in the relative peak intensities of the 211 and 112 peaks in the XRD pattern, and the lack of hydroxylation and PO_4^{3-} peak broadening from the FTIR analyses of the Sr-ZnHAp surfaces, suggest *c-axis* orientation and apparent disorder within the lattice of this coating. This is a direct consequence of simultaneous substitution of Sr and Zn for Ca in the lattice. As such, the suggested formula for the co-substituted Sr-ZnHAp coating produced here could be $\text{Ca}_{10-x-y}(\text{M})_y(\text{HPO}_4)_x(\text{PO}_4)_{6-x}$

(OH)_{2-x}, (with M denoting both Sr and Zn ions substituting for Ca in the lattice).

If the Ca+M/Ca ratios (M is the substituting metal ion) are also taken into consideration here, the levels of Sr in the SrHAp and Sr-ZnHAp surfaces are comparable (1.05 ± 0.02 and 1.06 ± 0.01 , respectively, as shown in Table 7). This is despite the difference in the number of targets used for each deposition run (two 10SrHAp targets for the SrHAp coating and one for the Sr-ZnHAp coating). A similar trend is observed for the Zn containing coatings (Ca + M/Ca), with the levels of Zn in the ZnHAp and Sr-ZnHAp surfaces reported as 1.03 ± 0.01 and 1.02 ± 0.01 , respectively. By comparison, the Sr-ZnHAp coating has a Ca + Sr + Zn/Ca ratio of 1.08 ± 0.01 , which clearly corroborates the successful incorporation of both Sr and Zn into the co-substituted coating. Clearly these results for the SrHAp, ZnHAp and Sr-ZnHAp coatings, highlighted from the XPS XRD and FTIR analyses confirm that they contain appreciable levels of Sr, Zn or Sr and Zn, with Sr and Zn substituting for Ca in the HAp lattice and that no other Ca-P phases or impurities (such as SrCO₃), were detected for any of the substituted coatings in this study. Previous similar work undertaken by Ozeki et al. highlighted the presence of additional unwanted CaP phases after sputtering from Sr-doped HAp targets [71]. As highlighted previously, the coatings produced here were deposited using distinctly different experimental conditions, and as such it might be expected that different results were observed between the two studies. Most notably, given the relatively low sputtering power employed in this study it is less likely that all of the Sr and Zn atoms have the required energy or momentum to reach the substrate surface (in line with the Sr and Zn content of the sputtering targets. This has been described previously in the literature [71–73]. It has been suggested that the incorporation of Zn into the HAp lattice distorts the crystal lattice and hinders HAp crystal growth, and this may be inferred here from the low Ca/P and Ca⁺ Zn/P ratios for the 10ZnHAp powder and ZnHAp coatings [78, 85]. This has been suggested to be a consequence of the mismatch between the size of the smaller Zn²⁺ ion (0.074nm) and that of Ca²⁺ (0.099 nm). Despite this, it is apparent from the results obtained here that RF magnetron sputtering can be successfully employed to sputter deposit ZnHAp, SrHAp and co-substituted Sr-ZnHAp coatings. Of particular interest here is the fact that in all cases here, the coatings produced had a distinct 002 preferred orientation (albeit more pronounced for the Sr containing coatings (SrHAp and Sr-ZnHAp). Previous work by others has shown that a preferred 002 orientation can enhance the cellular response in vitro, in particular for mesenchymal stem cells (MSC) [82]. This is most likely a consequence of the deposition parameters utilised here during sputtering, but this, with the combined therapeutic potential of Sr and

Zn in the co-substituted surfaces opens up new opportunities to develop coatings that have significant in vitro and in vivo capability, beyond that already available. To date, there are very few studies that have considered the use of co-substituted Sr and Zn CaP materials in the literature [14, 62, 89], with most of the work focused towards potential co-substitutions of Mg and Sr or other ion combinations for antibacterial applications [13, 19, 23, 26, 32, 58, 61, 62, 90]. To the knowledge of the authors, this is the first successful attempt to produce Sr and Zn co-substituted HAp coatings via sputtering (and also Zn substituted HAp coatings as well). Given the obvious flexibility that sputtering provides, the development of multi-ion substituted CaP coatings is a real possibility. In the case of the co-substituted coatings produced here, previous studies have highlighted the positive effect of Sr-substitution in HAp, both in vitro and in vivo through enhancing osteoblast activity (proliferation), differentiation, and reducing osteoclast activity [7, 8, 89]. Add to this the fact that Zn has been shown to stimulate bone growth and mineralisation, lowers osteoclast activity like Sr, and has possible anti-inflammatory and antibacterial properties [7, 8, 49], the possibilities of easily delivering sputter deposited coatings with these combined properties offers significant advantages over single ion substitutions. However, detailed in vitro and in vivo studies would need to be conducted on these coatings in order to verify these findings and to study the effects of different ion concentrations when co-substituted in HAp coatings.

5 Conclusions

The aim of this particular work was to study the feasibility of using RF magnetron sputtering to deposit Sr, Zn and Sr-Zn co-substituted HAp coatings. In particular, the influence of different target configurations on the properties of the substituted sputter deposited coatings produced at a low discharge power level (150 W) were investigated. The results presented herein demonstrate that RF magnetron sputtering can be used to produce co-substituted Sr-ZnHAp coatings, and that none of the coatings produced contained any other impurity CaP phases, with XRD data showing peaks corresponding to that of ICDD file #01-072-1243 for HAp, albeit shifted to lower 2θ values due to the incorporation of Sr into the HAp lattice for Ca for the SrHAp and Sr-ZnHAp coatings. All of the coatings are calcium deficient and exhibit preferred 002 orientation, which is more pronounced for the Sr-containing surfaces. These are all properties that have been highlighted in the literature as having the potential to have a positive influence on the osteoblast response. RF magnetron sputtering therefore offers a means to control the attendant surface properties of

Sr, Zn and Sr-Zn co-substituted HAp, such as the crystallinity, stoichiometry and phase purity. However, these surfaces would need to be investigated by rigorous in vitro and in vivo testing to provide further evidence of their potential to enhance osteoconduction, osteoinduction and to be osteogenic by more closely mirroring the complex chemistry of bone. This could be achieved with the potential added benefit of being anti-inflammatory and anti-bacterial, and delivering the next generation of biomaterials for orthopaedics.

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Compliance with Ethical Standards

Conflict of interest The authors declare that they have no competing interest.

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