



# The Importance of Research on Materials

**Nitrides seem to be good for everything**

**Dilute Nitrides**

(Ga,In)(N,P,As)

cubic

**Conventional Nitrides**

AlN, GaN, InN

hexagonal (cubic)

**Ferromagnetic Nitrides**

TE- and RE-doped

hexagonal (cubic)



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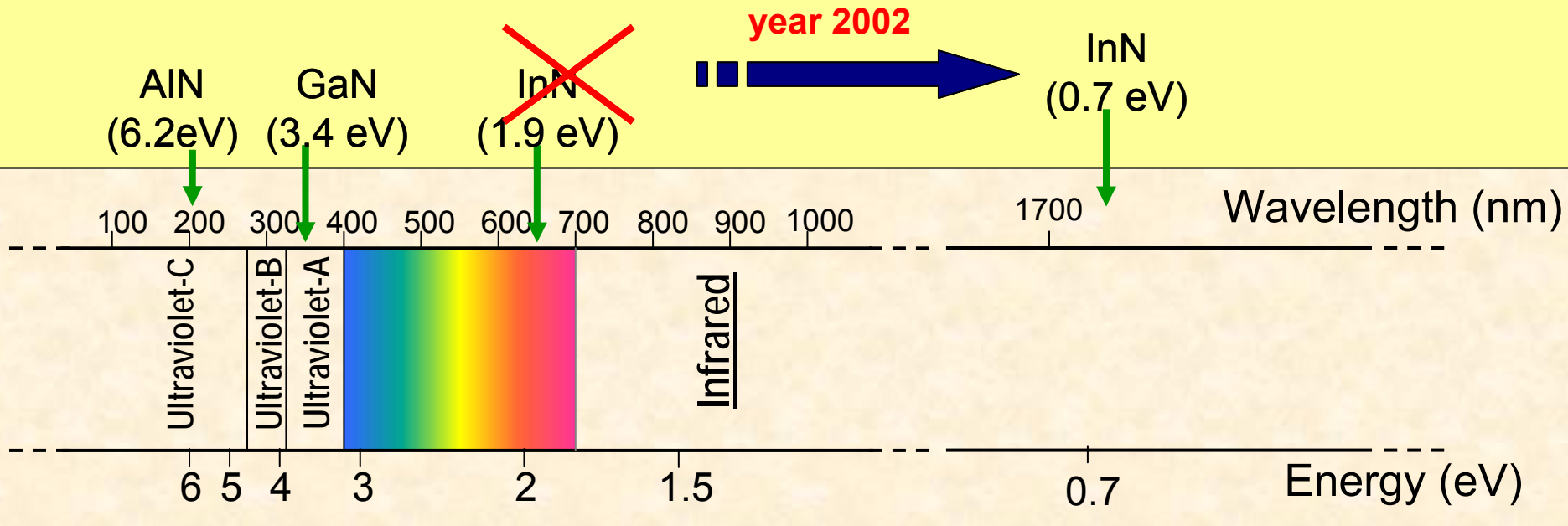
hexagonal (cubic)

**Ferromagnetic Nitrides**

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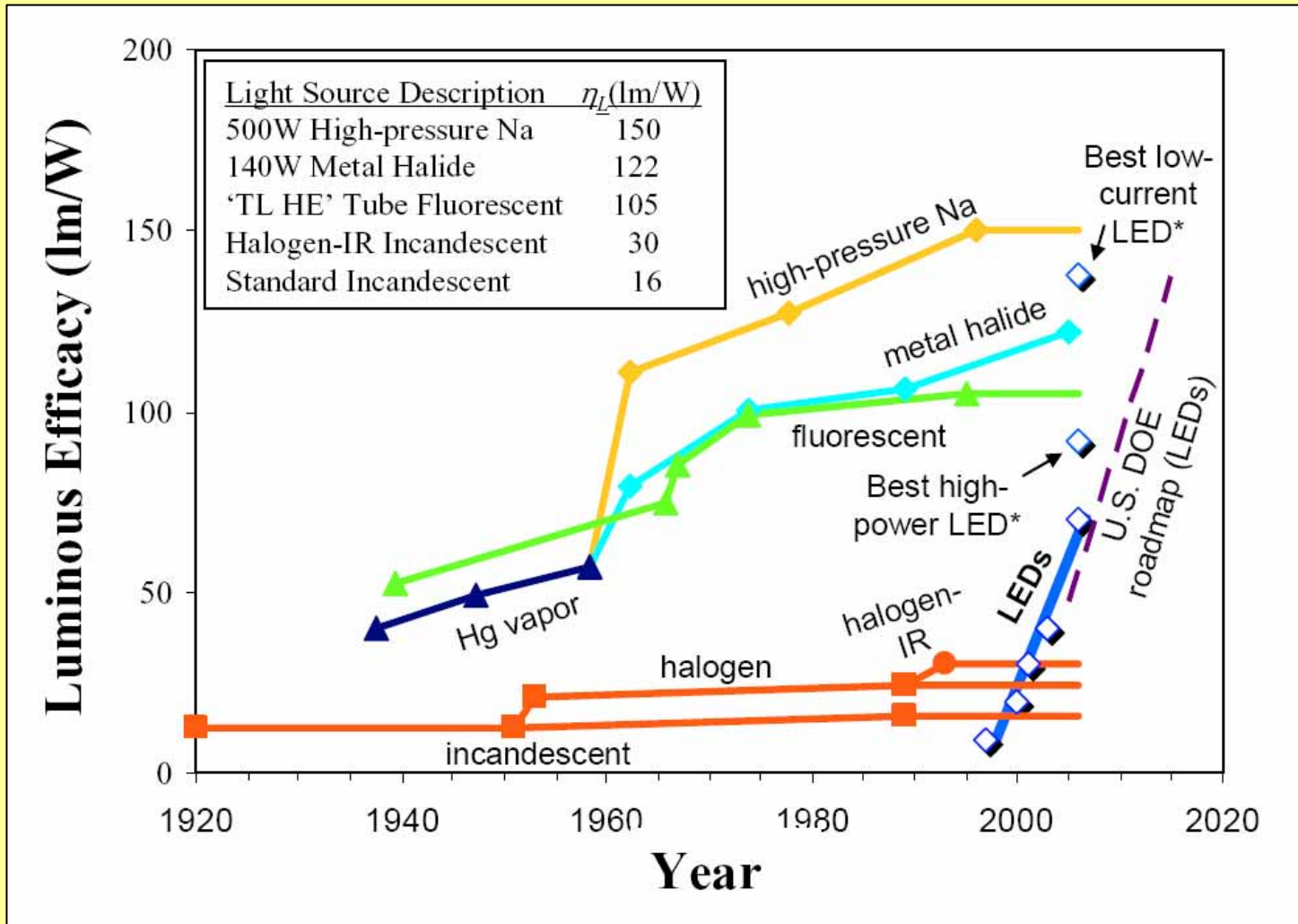
hexagonal (cubic)

# Conventional III-Nitrides

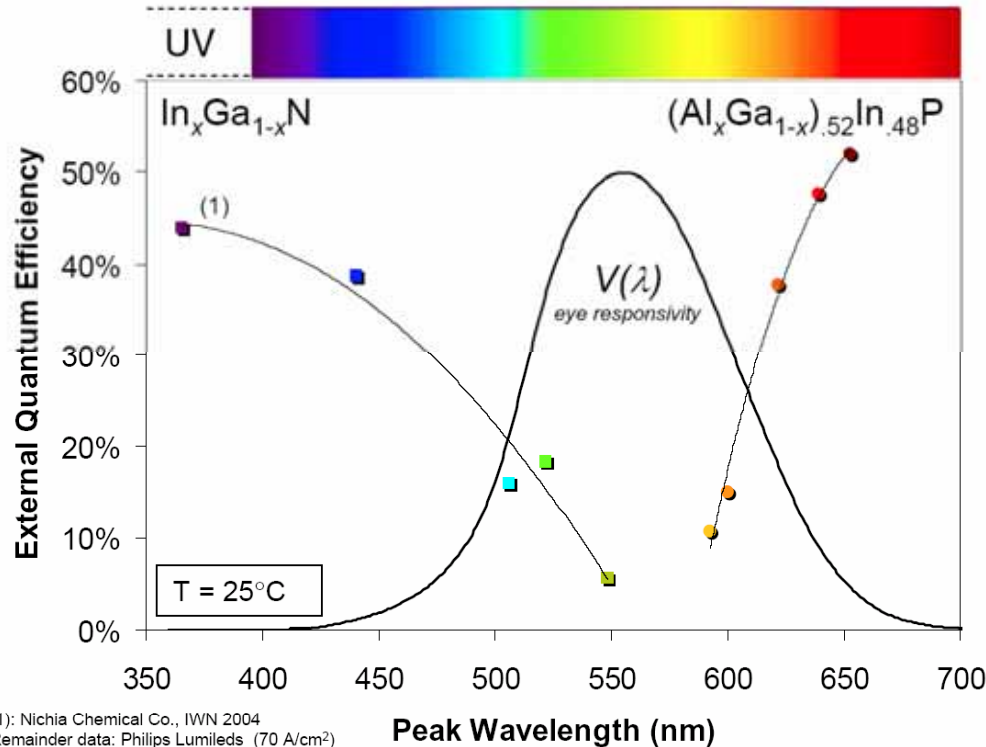


- Direct-gap semiconductors covering wide range 0.7-6.2 eV
  - IR-Blue-UV optical emitters/detectors
- Strong chemical bonding - high stability/resistance
  - high temperature/power electronics

# Luminous efficacy of light sources



# State-of-Art: High-Power LEDs (1 W)



## III-P:

- Fundamental band-structure limitations

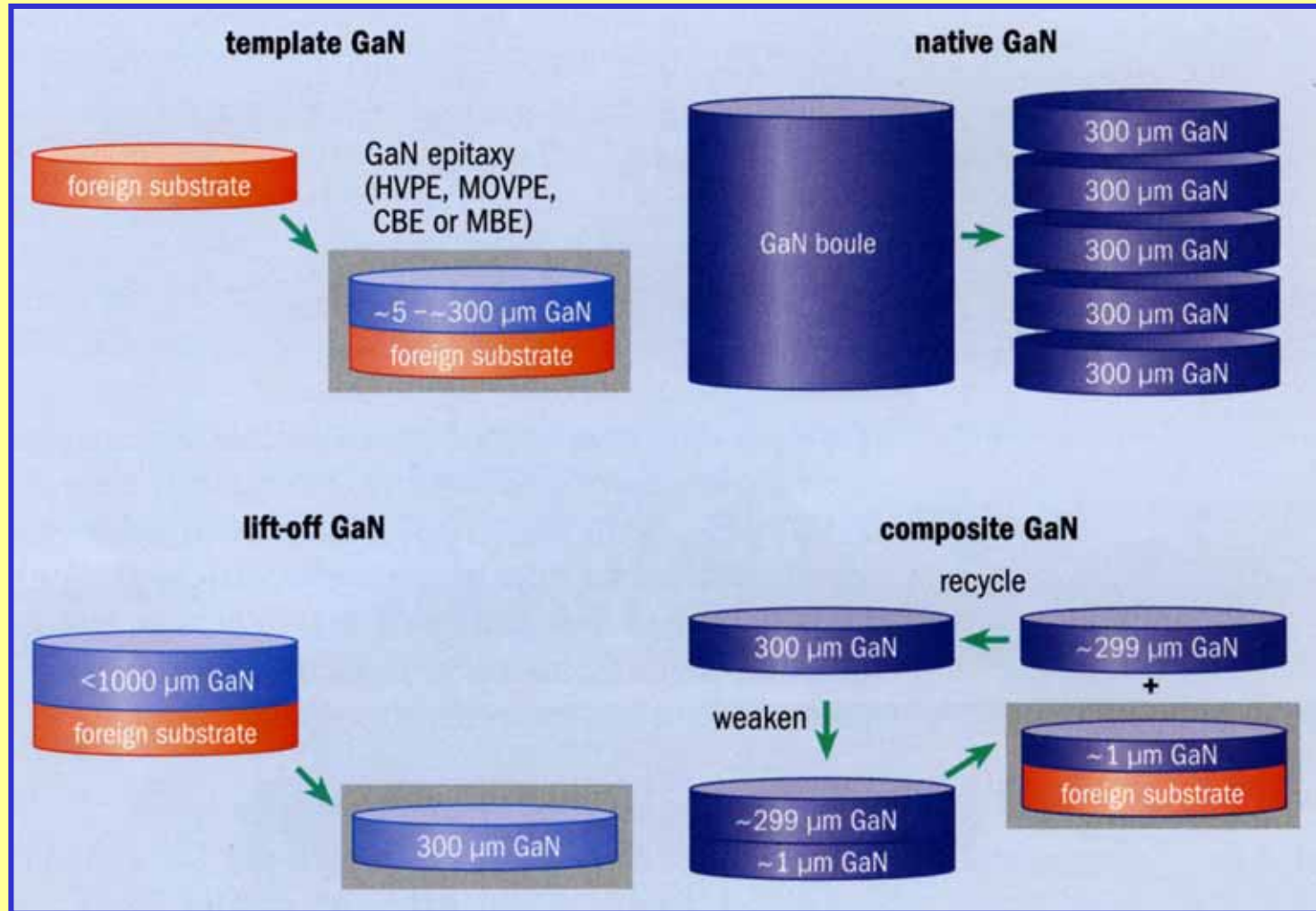
## III-N:

- IQE (~450 nm) ~50 %
- IQE (~520 nm) ~20 %

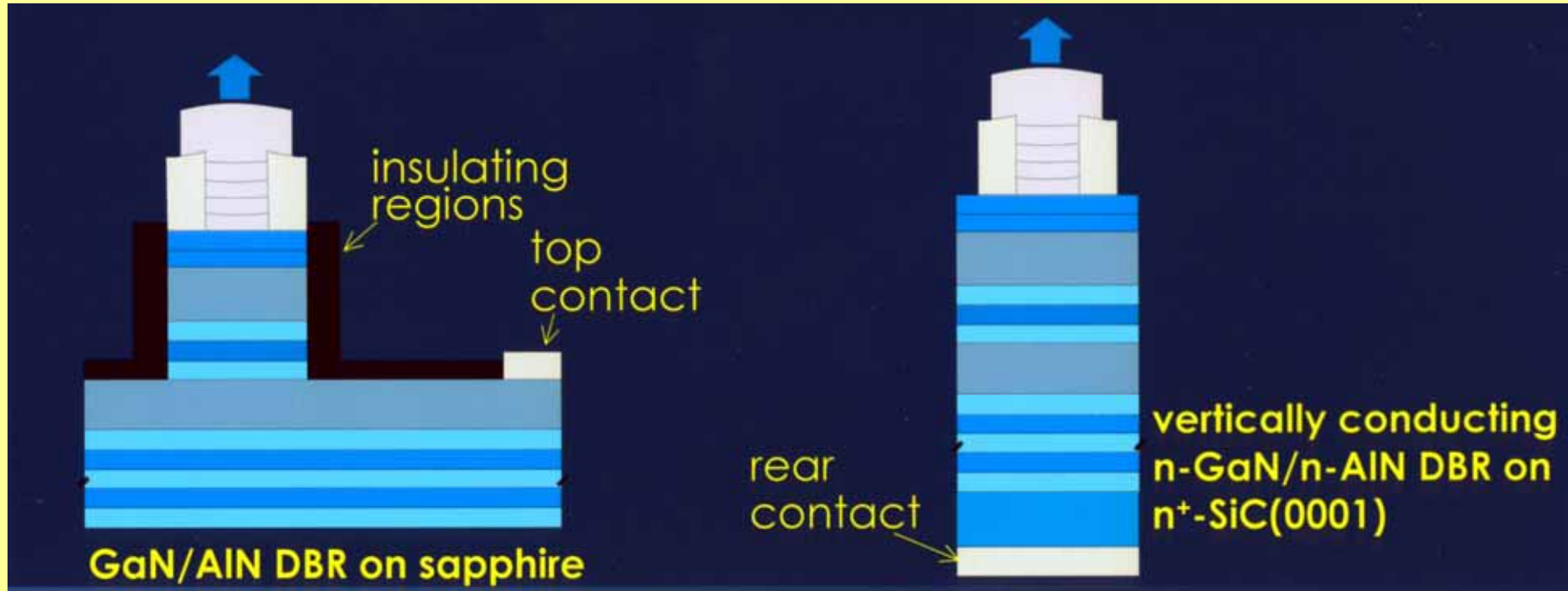
- External Quantum Efficiency:  $EQE = \text{Internal QE} \times \text{Extraction Efficiency}$   

$$EQE = P_{\text{optical}} / (E_{\text{ph}} \times I_f)$$
- Power Conversion/Wallplug Efficiency:  $PCE$  or  $WPE = P_{\text{optical}} / (V_f \times I_f)$
- Luminous Efficacy (lumens/Watt):  $LE = PCE \times V(\lambda)$

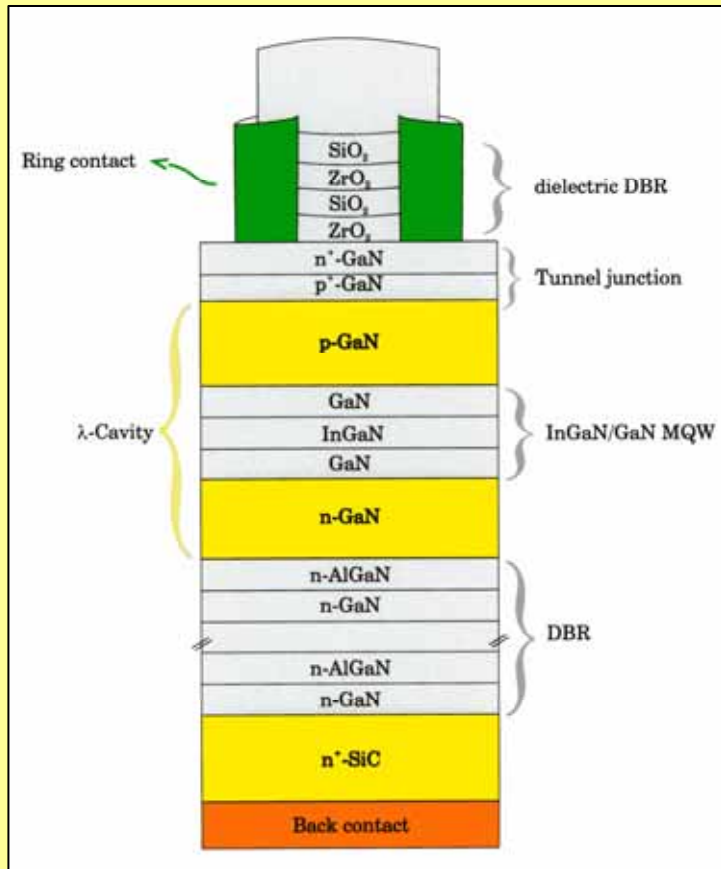
# GaN substrate technology



# Vertical cavity surface emitter (VCSEL)



# Vertical cavity surface emitter (VCSEL)

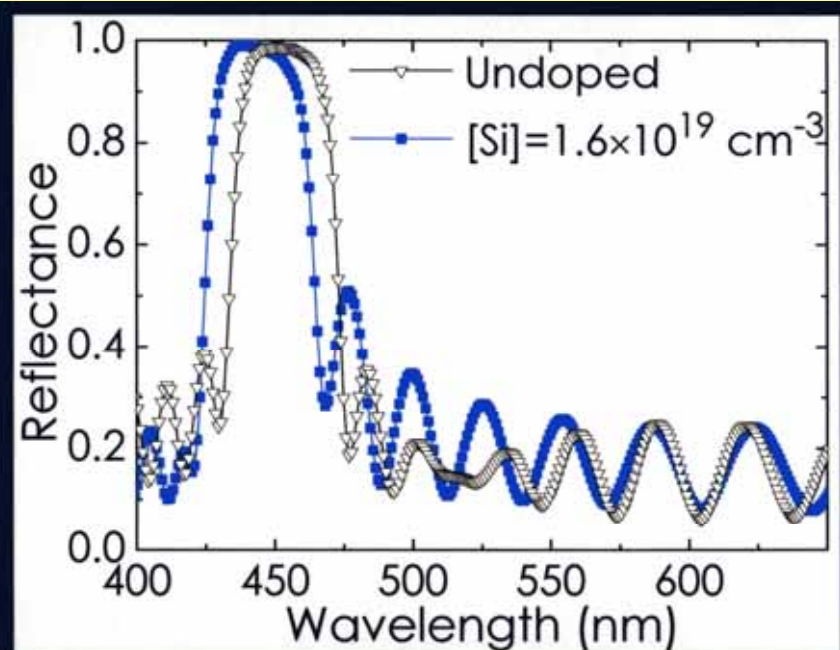
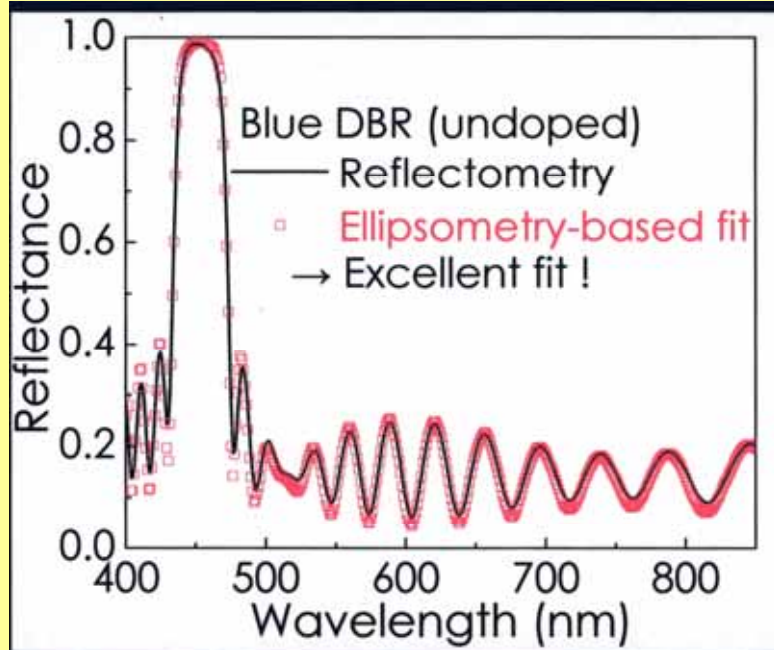


- Improve reflectivity of III-Nitride Bragg reflector
- Compare strained AlN/GaN DBR with lattice-matched (Al,In)N/GaN DBR
- Reduce series resistance of doped Bragg reflector
- Design and realize p-type carrier injection scheme via tunneling junction

**III-Nitride based microcavities to study the formation of exciton polaritons for Bose Einstein condensation**



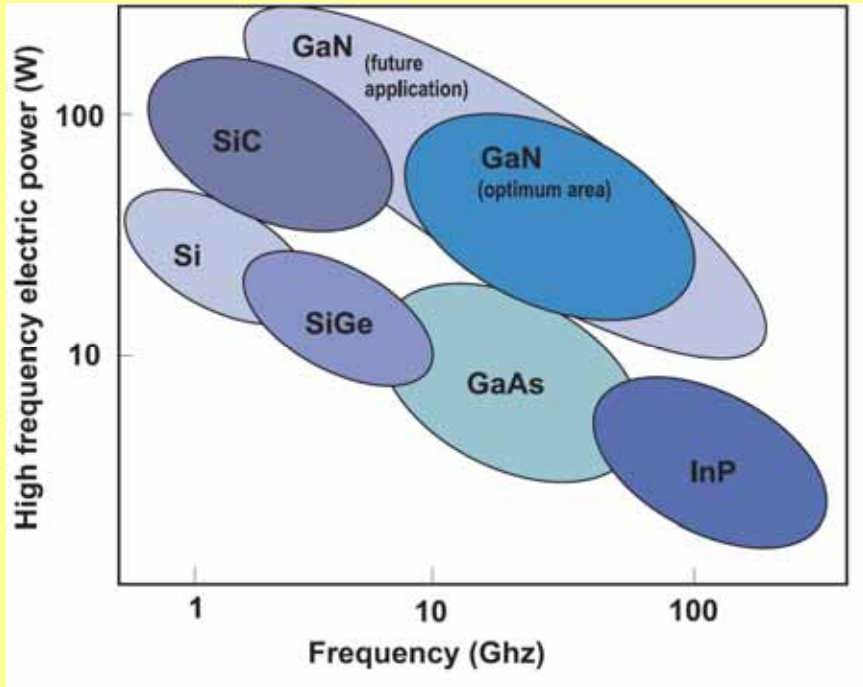
# Reflectance of AlN/GaN DBR



[Si]	R	$\lambda$ (nm)	FWHM (nm)	$d_{\text{AlN}}$ (nm)	$d_{\text{GaN}}$ (nm)
Undoped	0.99	454	41	57	46
$1.6 \times 10^{19} \text{ cm}^{-3}$	>0.99	445	41	56	45

➡ Si-doping does not degrade R

# GaN electron devices



## Heterostructure FETs (HFETs, HEMTs)

- High breakdown electric field
- High electron density
- High saturation drift velocity
- High power output, high frequency and high temperature operation

## Earlier problems:

- Collapse of drain current
- Large gate leakage current

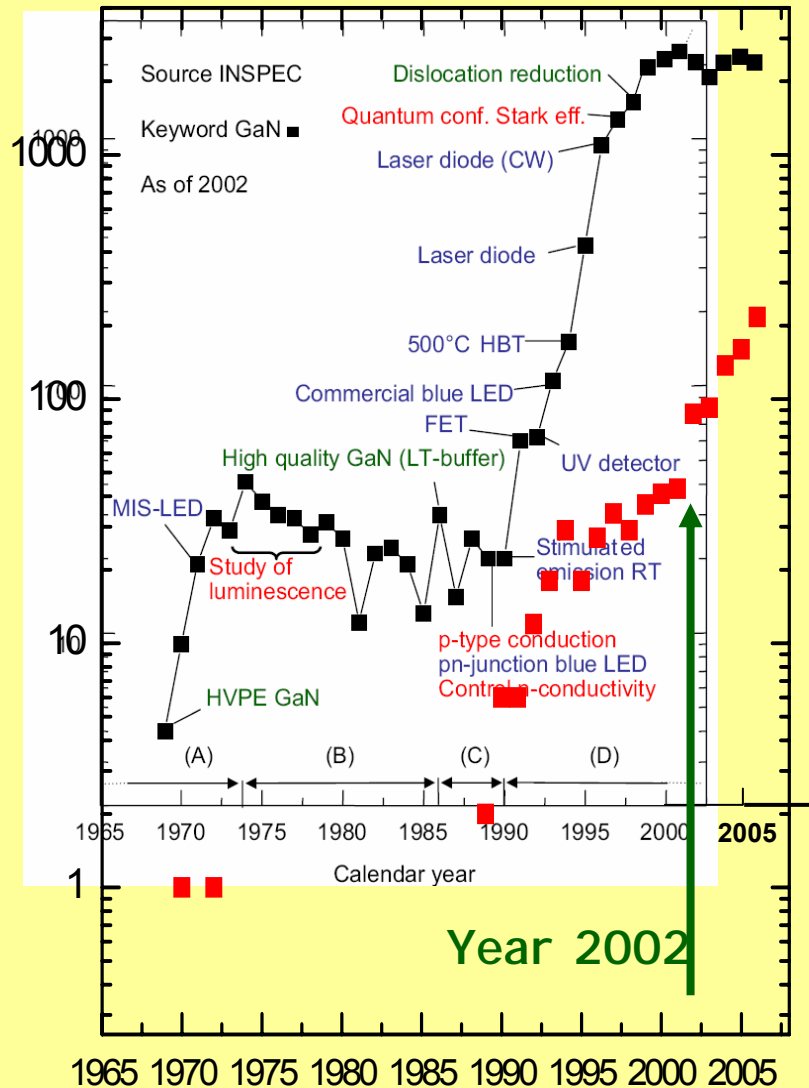
Compare (Al,Ga)N/GaN with (Al,In)N/GaN

Operation of high-power HFETs in 2 GHz band is satisfactory

Concerns: Reliability, production yield, cost, Issues of substrate material: SiC, Si, AlN

Needs for high frequency operation: High power output, high efficiency, high degree of linearity, low power consumption

# Number of publications per year



In 2006:

~ **1.800** publications with  
**GaN** as keyword in title

~ **200** publications with  
**InN** as keyword in title

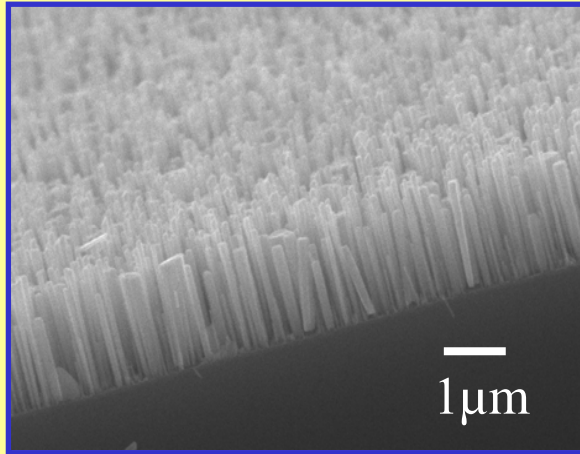


## To do list for InN films

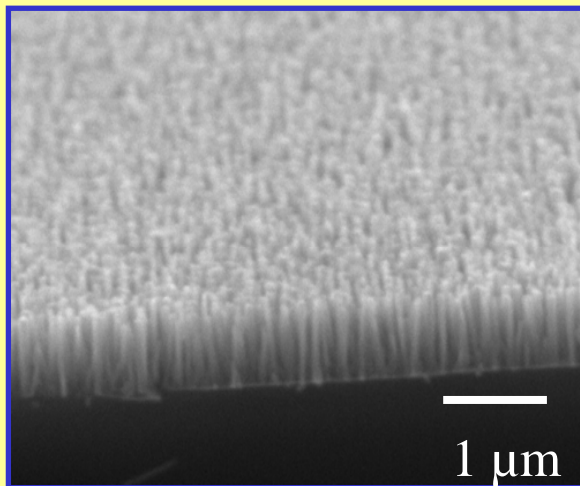
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- **Improve structural properties of InN films by optimizing growth conditions and proper selection of substrate**
- **Understand and control the intrinsic electron accumulation on polar InN surfaces**
- **Grow InN films on nonpolar surfaces**

# InN nanorods (I)

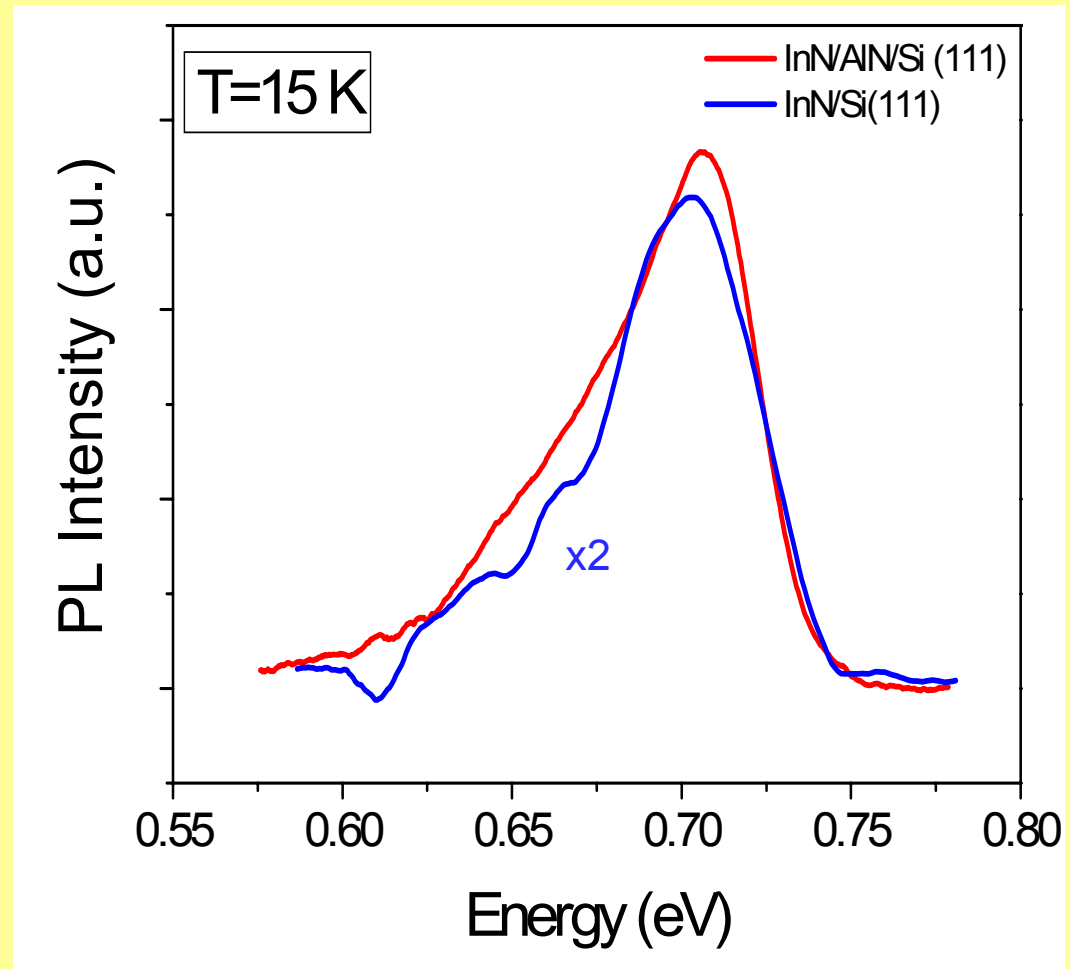


InN/AIn/Si (111)



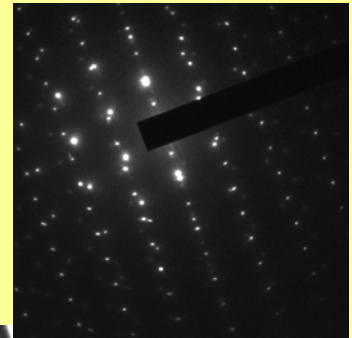
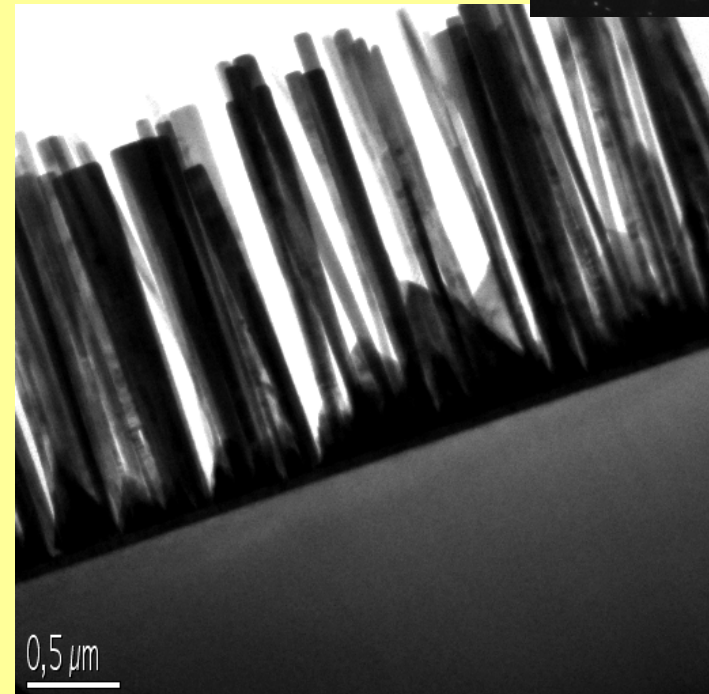
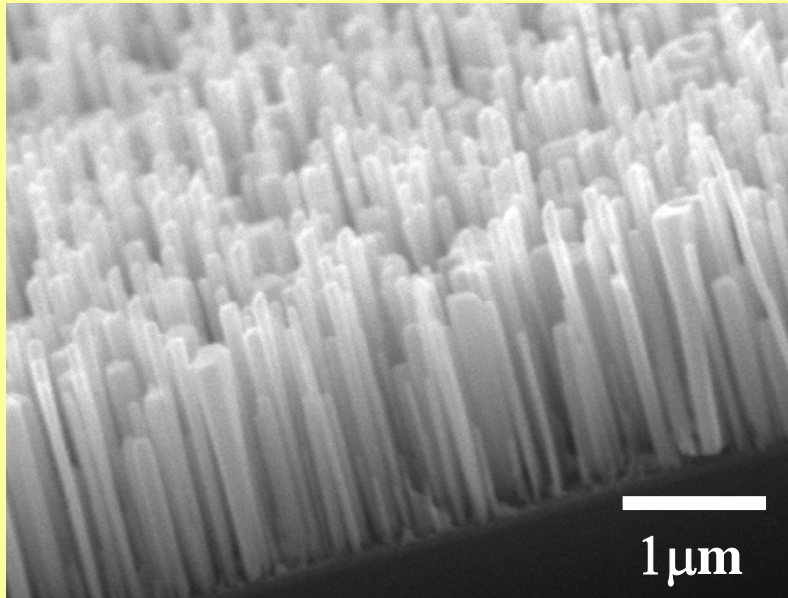
InN /Si (111)

## PL Spectra of InN grown with and without buffer layer

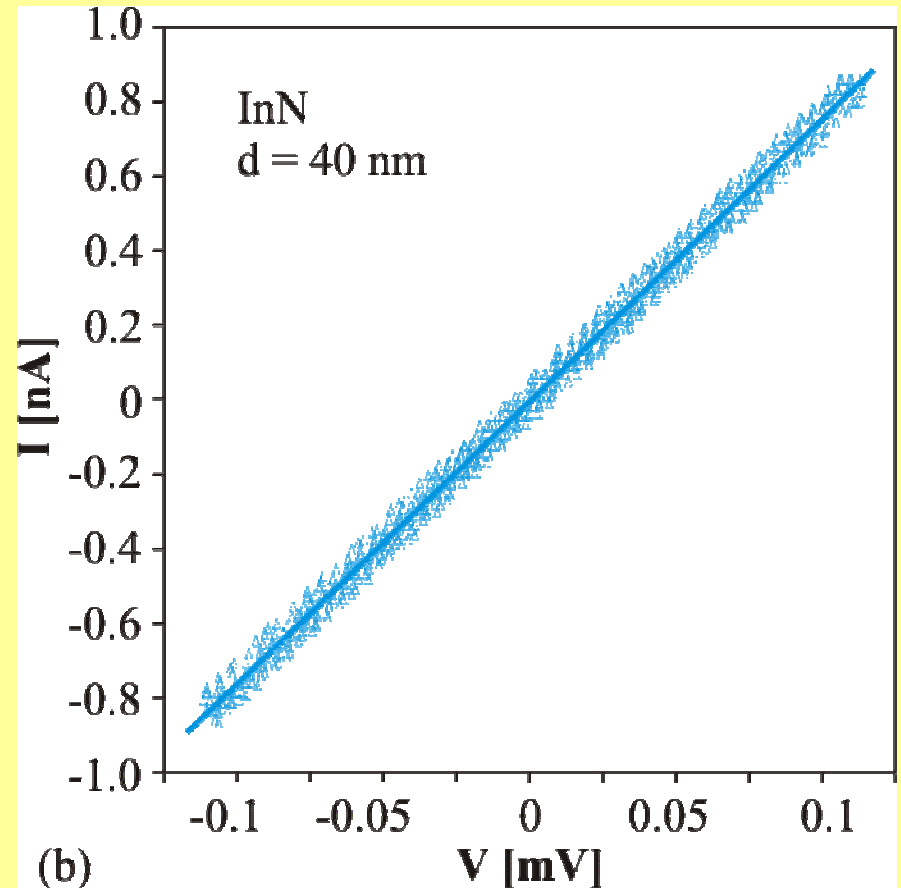
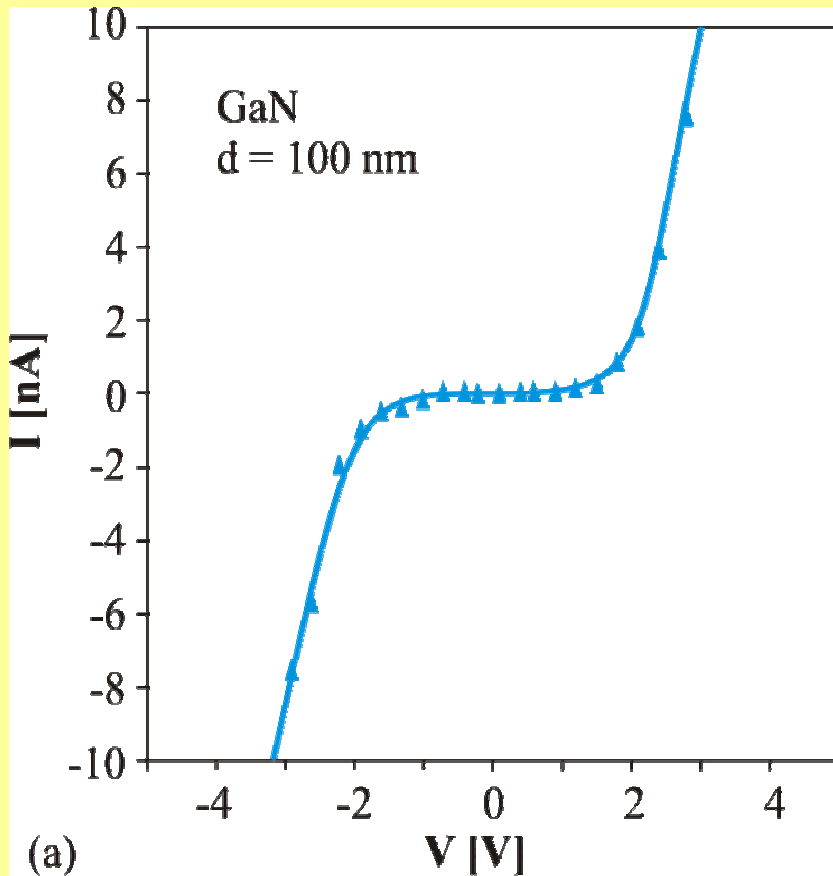


# InN nanorods (II)

- SEM and TEM photographs of InN nanorods
- SAD pattern reveal epitaxial alignment



# InN nanorods (III)



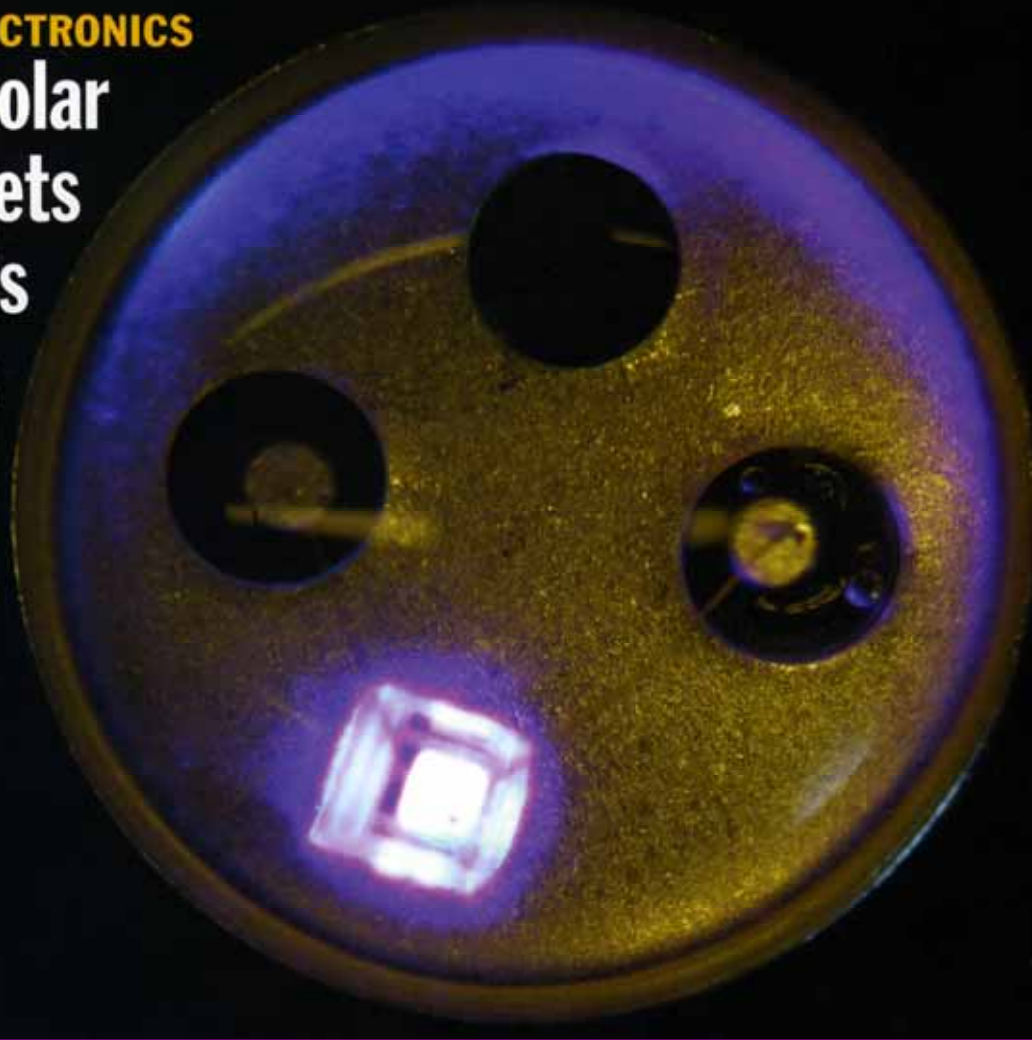
*E.Calleja et al. to be published.*

Higher conductivity for the InN nanocolumns than for GaN. Related with the tendency of InN to have a very high n-type residual concentration ( $10^{18} \text{ cm}^{-3}$ - $10^{20} \text{ cm}^{-3}$ )

# Nonpolar GaN

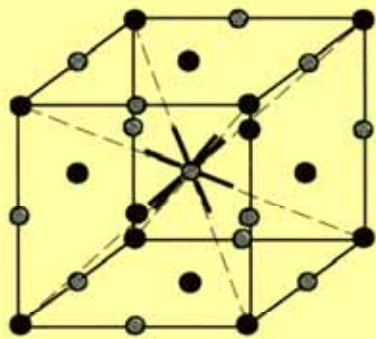
**OPTOELECTRONICS**

**Non-polar  
GaN gets  
into its  
stride**

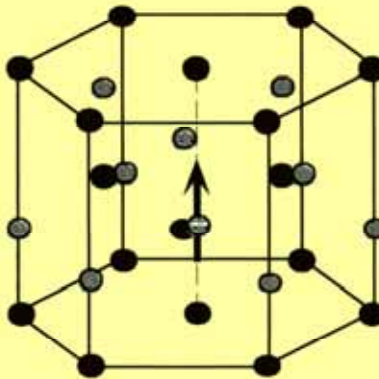




# Origin and consequences of electrical polarization



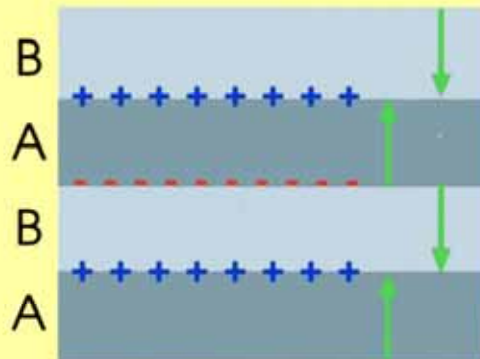
Zincblende GaN



Wurtzite GaN

**Zincblende:** 4 polar (111) axes  
→ piezoelectric along (111)  
→ not pyroelectric

**Wurtzite:** 1 polar (0001) axis  
→ piezoelectric along (0001)  
→ pyroelectric along (0001)

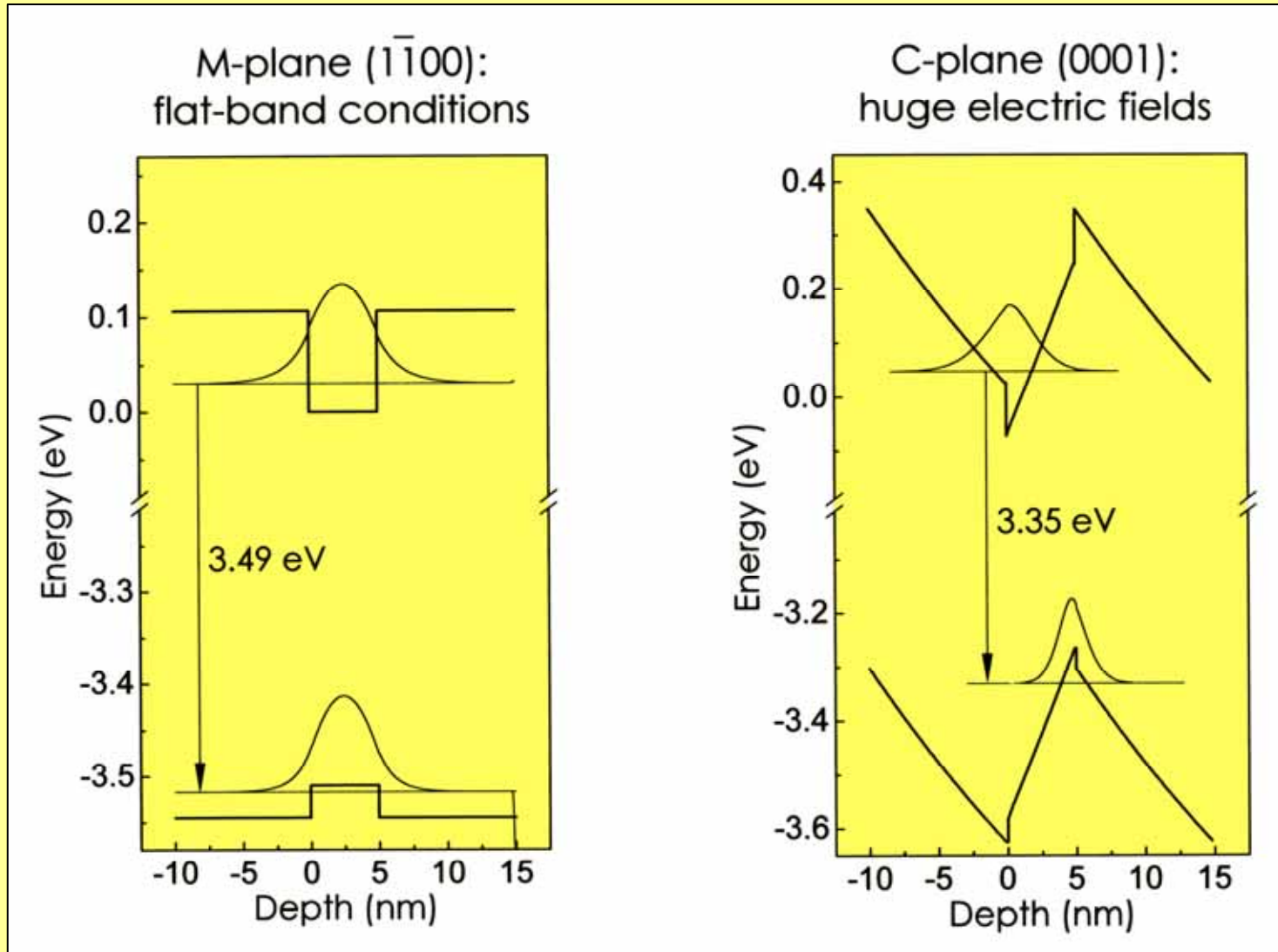


**Changes** in electrical polarization **along** growth direction

→ Formation of **interface charges**

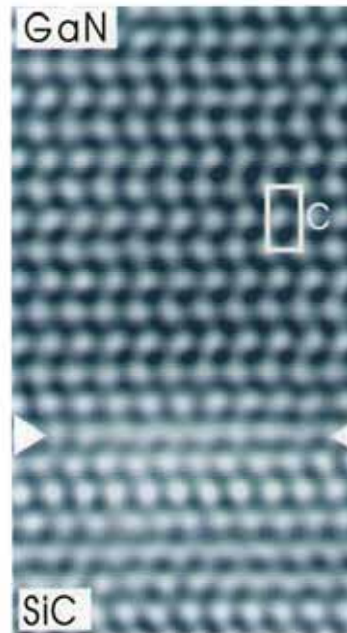
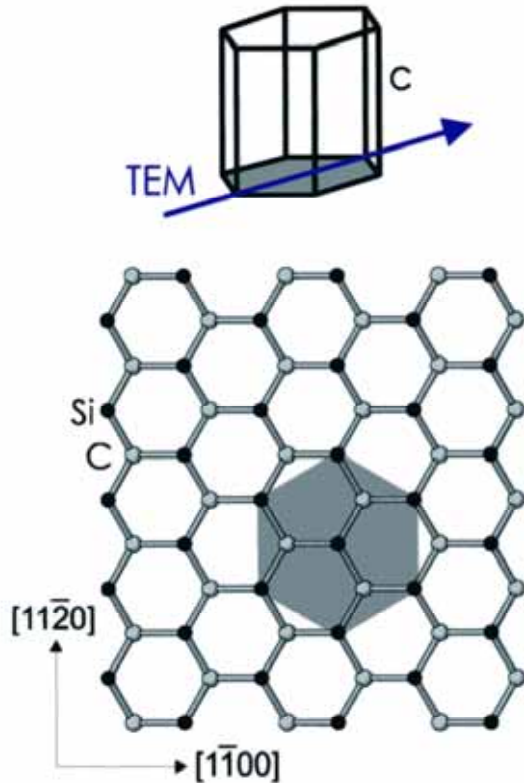
→ **Electric fields** along growth direction

# Band diagrams of GaN/(Al,Ga)N multiple quantum wells

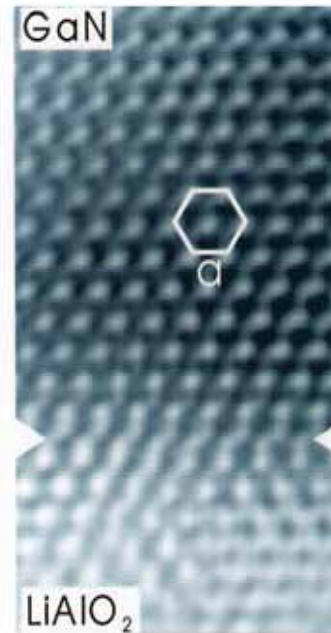
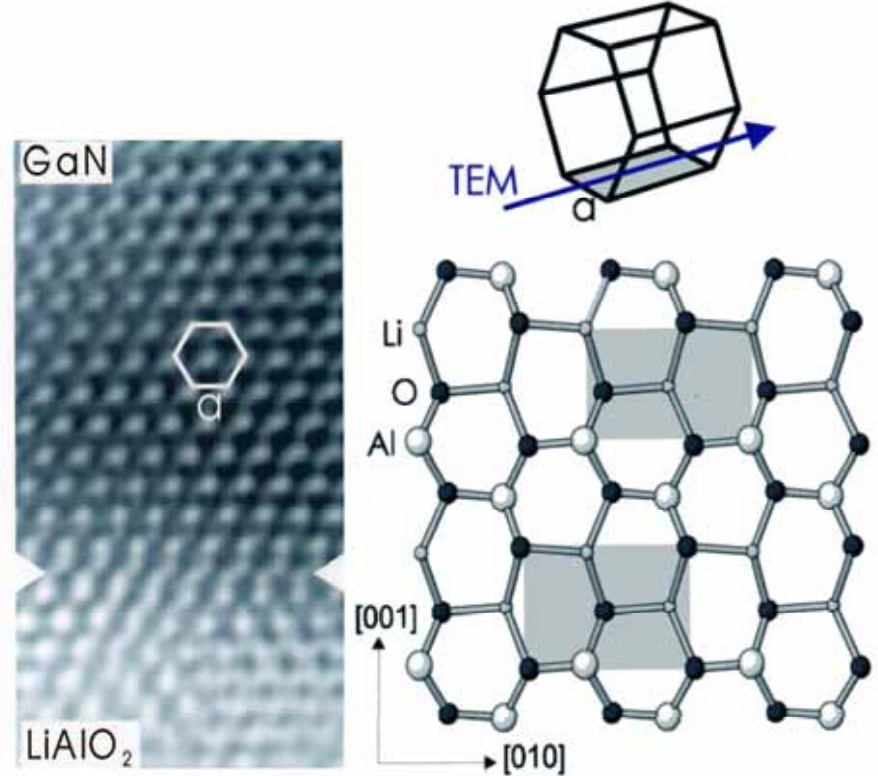


# Substrate with small mismatch ( $\gamma$ -LiAlO<sub>2</sub>) for M-plane GaN

Conventional C-plane orientations:  
GaN(0001)/6H-SiC(0001)



New M-plane orientations:  
GaN( $1\bar{1}00$ )/ $\gamma$ -LiAlO<sub>2</sub>(100)



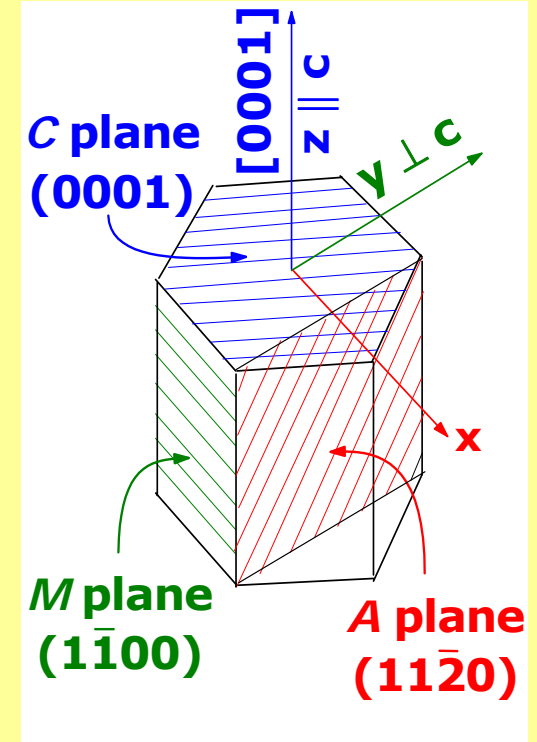
# Non-polar GaN

## M-plane GaN films grown on $\gamma$ -LiAlO<sub>2</sub>

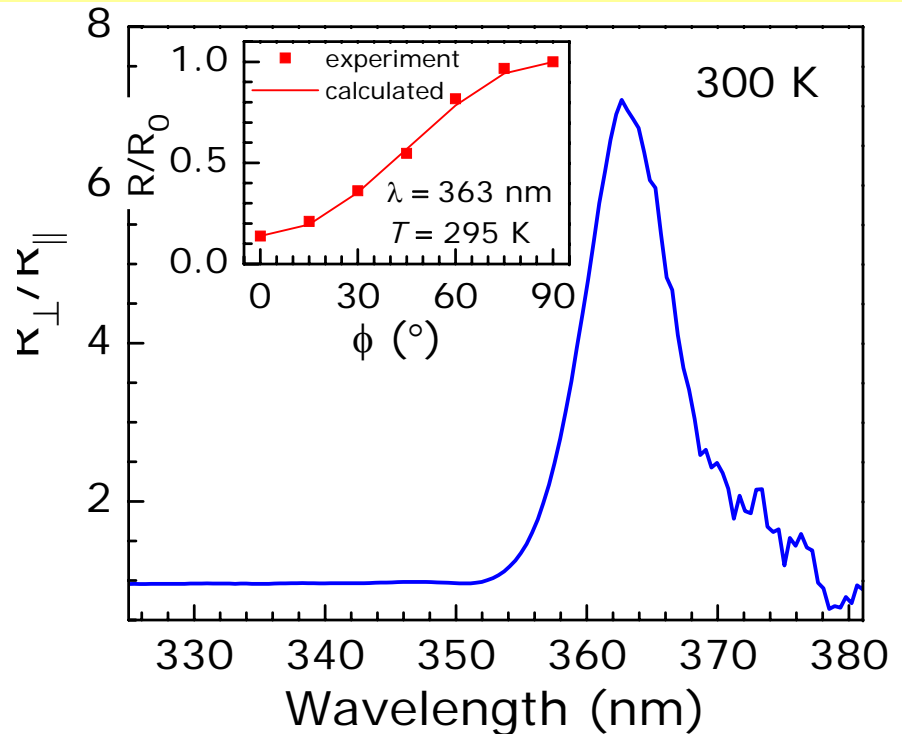
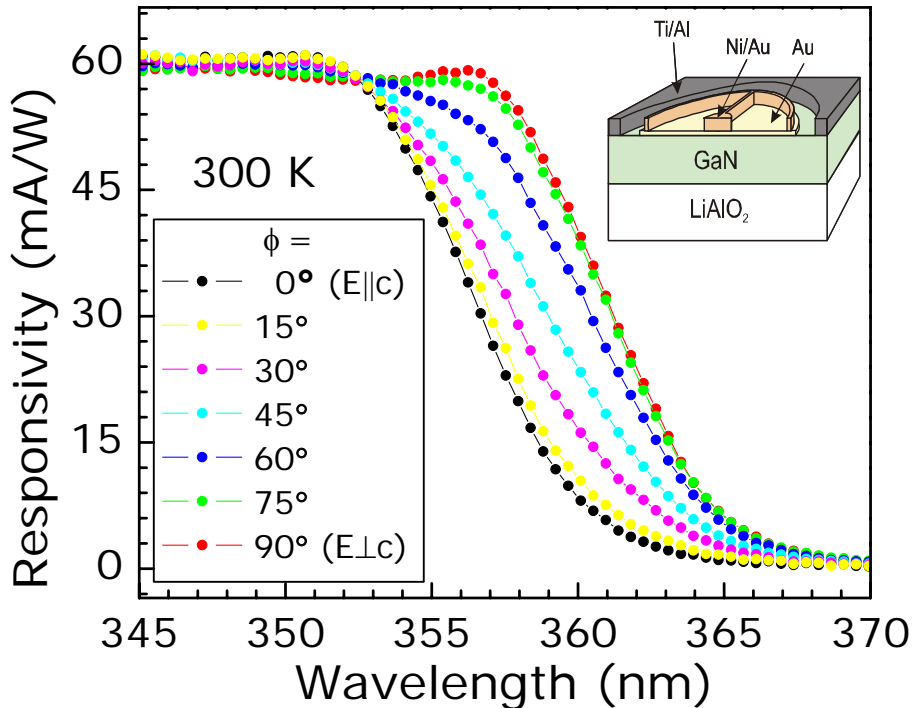
- no internal electrostatic fields in QWs
- highly anisotropic strain
- enhanced optical anisotropy (refractive index, polarization)

### Here:

- linear dichroism
- polarization filtering
- demonstration of polarization-sensitive photodetectors
- two-color Bragg reflector (DBR) based on linear birefringence

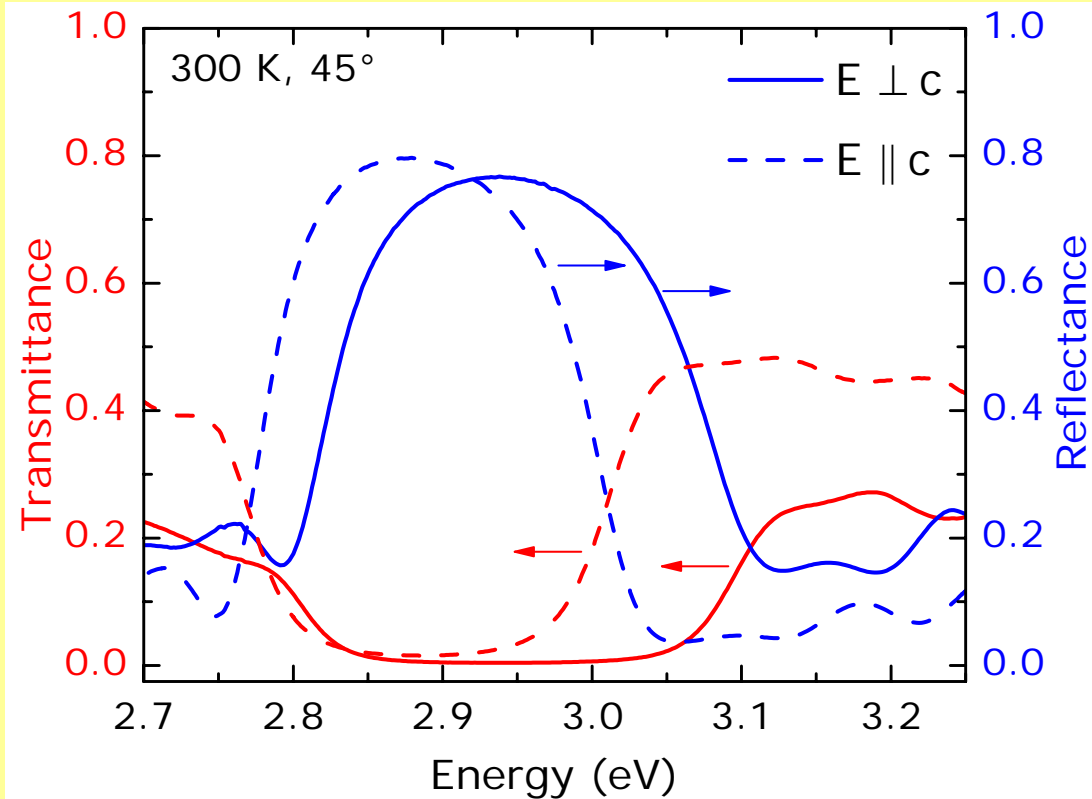
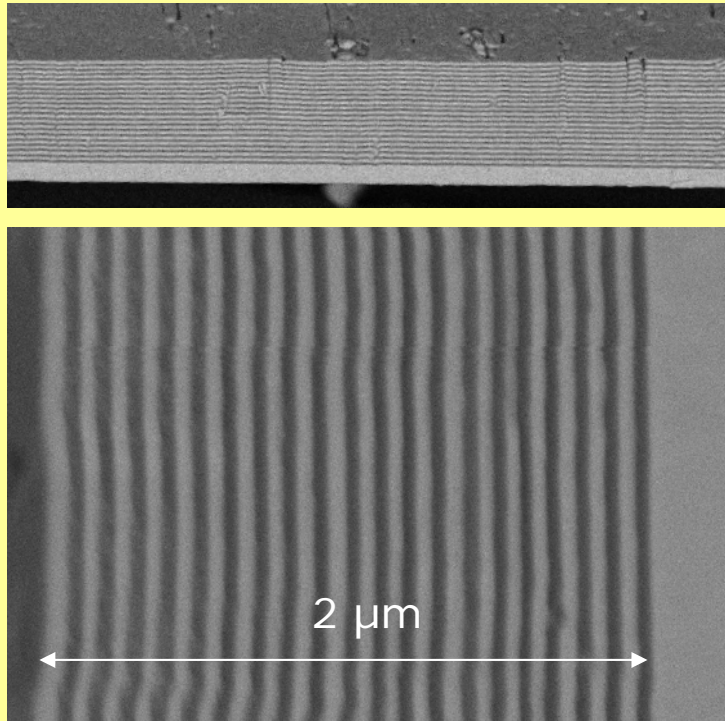


# M-plane GaN photodetectors



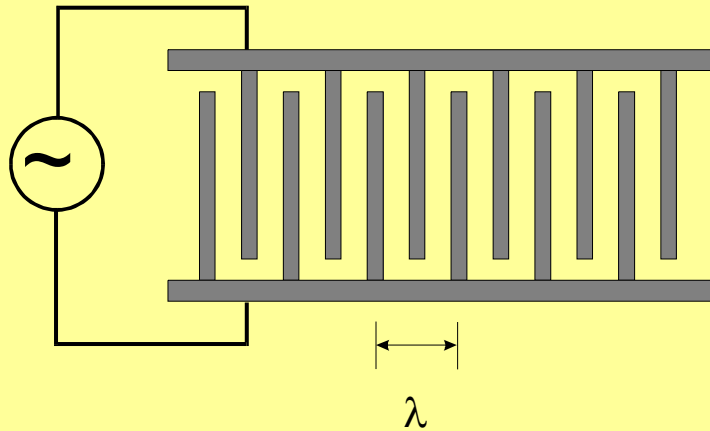
- Photocurrent measurements for semitransparent Schottky diodes, maximum responsivity  $R=60 \text{ mA/W}$
- Maximum contrast in responsivity  $>7$  at 363 nm
- $R = R_{\parallel} \cos^2 \Phi + R_{\perp} \sin^2 \Phi$

# Linear birefringence: two-color M-plane DBR



- 20-period GaN/AlN Bragg reflector on ~300 nm GaN buffer layer on LiAlO<sub>2</sub>
- maximum reflectivity 80–90% limited by interface roughness
- stopband shifts by about 70 meV due to linear birefringence
- maximum contrast between parallel and perpendicular polarization at 3.05 eV

# High frequency SAW devices



SAW  
➔

$$f = v/\lambda$$

For high frequency (f),

small wavelength ( $\lambda$ )

or **large velocity (v)**

	<b>AlN</b>	<b>GaN</b>	<b>GaAs</b>	<b>LiNbO<sub>3</sub></b>
<b>SAW velocity (m/s)</b>	<b>5790</b>	<b>3690</b>	<b>2870</b>	<b>3490-3890</b>
<b>Electromechanical coupling coefficient (%)</b>	<b>0.25</b>	<b>0.13</b>	<b>0.064</b>	<b>4.8</b>

# AlN/SiC acoustic heterostructures

AlN is typically grown on substrates such as sapphire, SiC, and Si.

For acoustic applications,

**What are the influences of velocity mismatch?**

Confinement of acoustic waves in the slow-velocity overlayer.



Reflection from the high-velocity substrate (cf. Snell's law).

SAW transmission is mediated by guided modes.

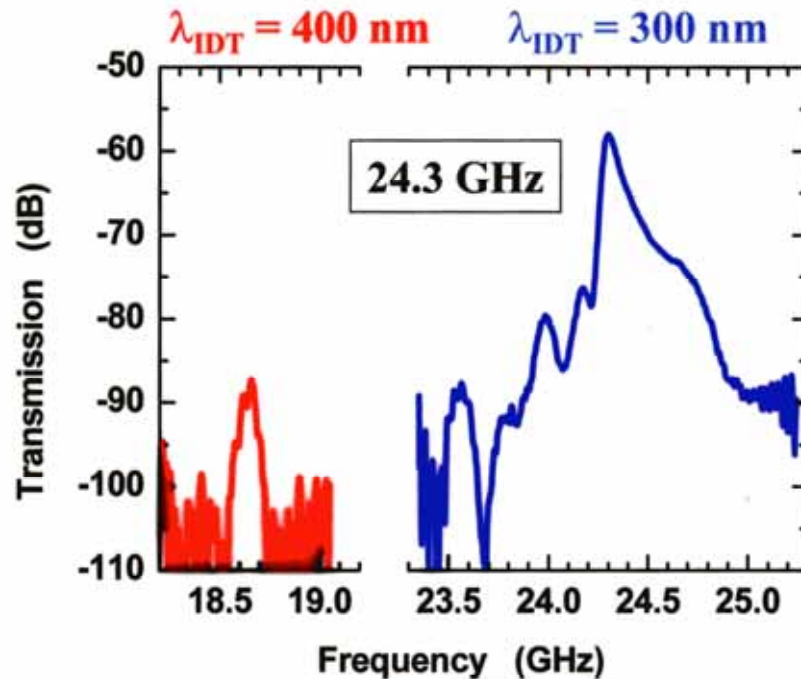
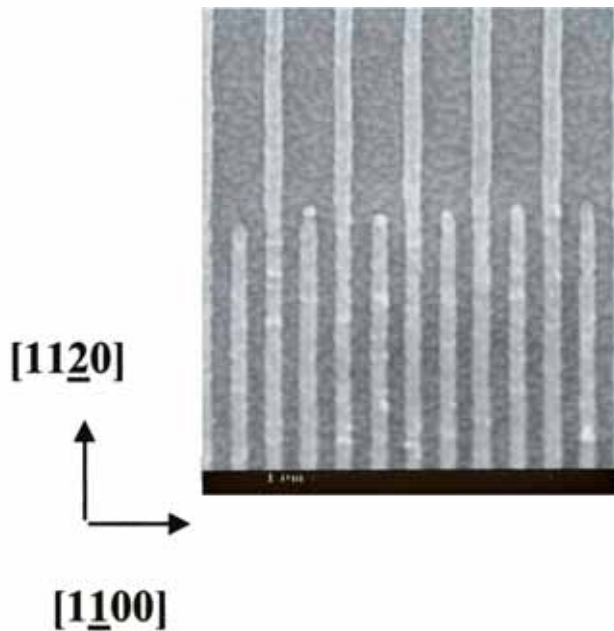
[ The guiding is absent if the AlN layer is grown on sapphire or Si because  $v_{SAW}(\text{substrate}) < v_{SAW}(\text{AlN})$ . ]



# Superhigh frequency operation

**Our strategy:** Electron-beam lithography to reduce  $\lambda_{\text{IDT}}$ .  
AlN/SiC heterostructures for large  $v_{\text{SAW}}$ .

Al (30 nm thick) electrodes



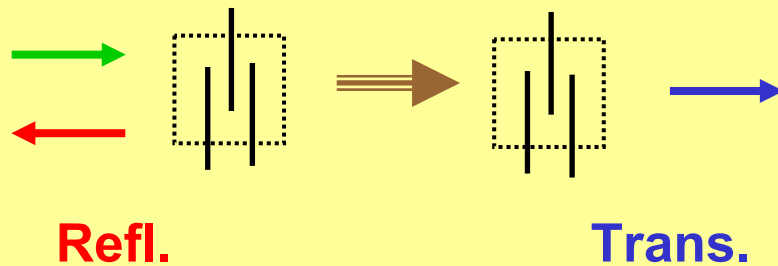
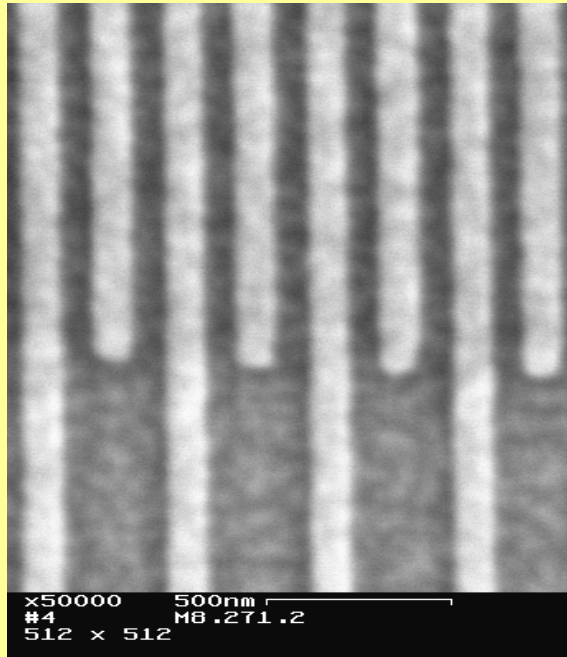
MBE-grown AlN (250 nm thick) layer on 4H-SiC(0001).

**SAW propagation is isotropic in the C-plane of hexagonal crystals.**

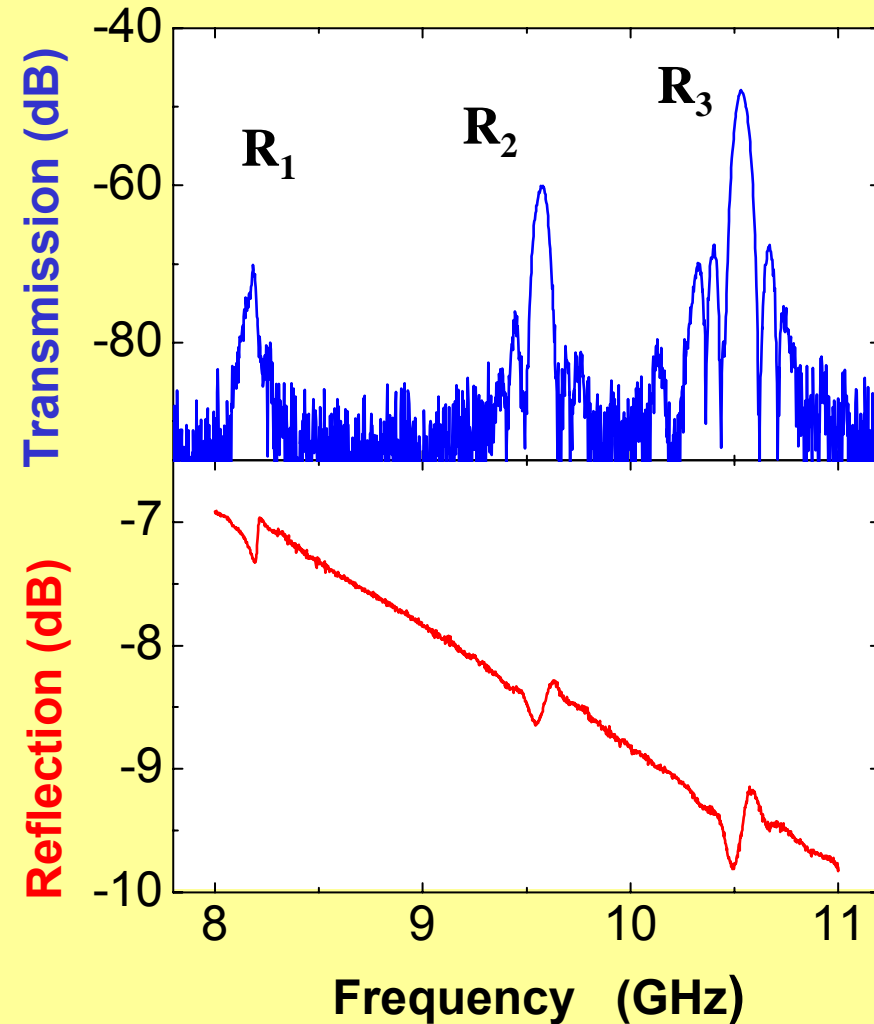
# Transmission characteristics

Al electrodes  
on  
AlN/SiC

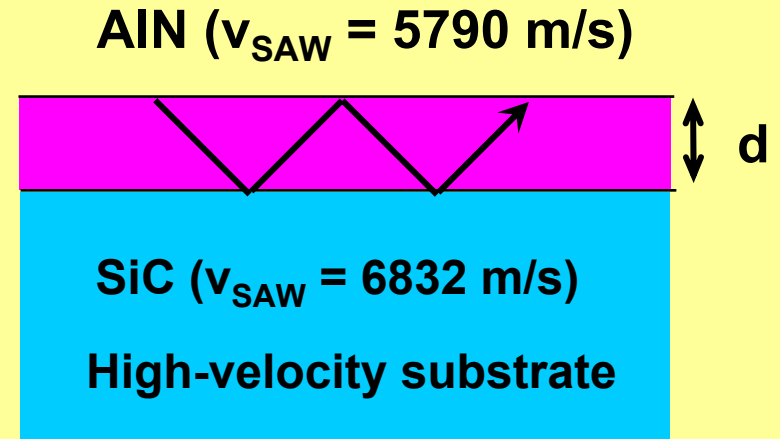
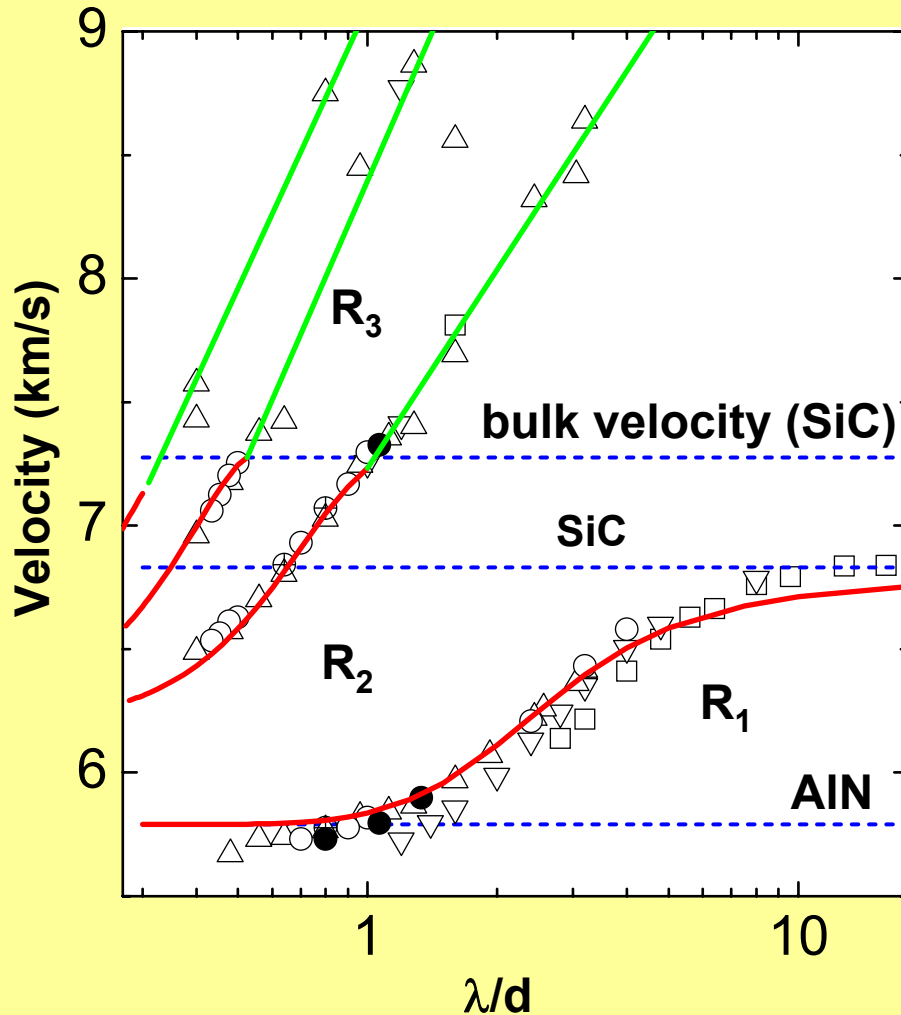
( $\lambda = 400$  nm)



$\lambda = 700$  nm,  $d = 1.25$  mm



# SAW dispersion (hard-supported layer)



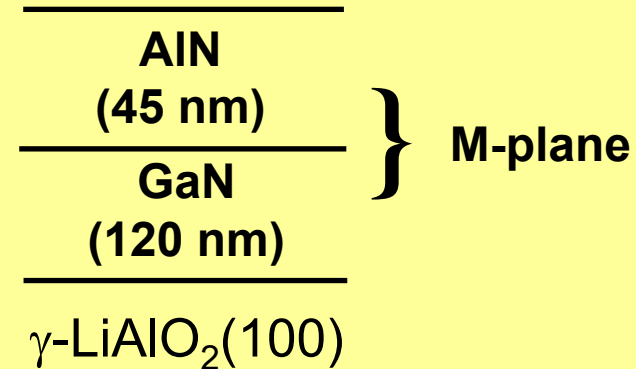
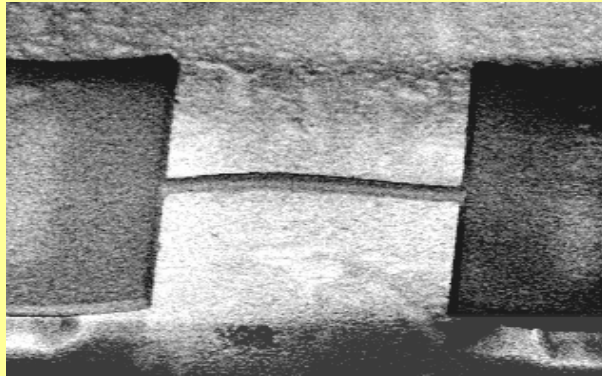
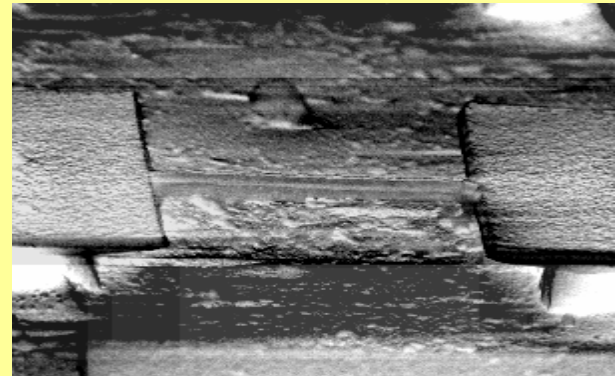
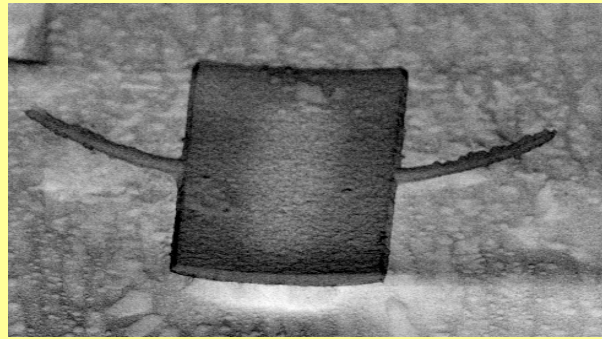
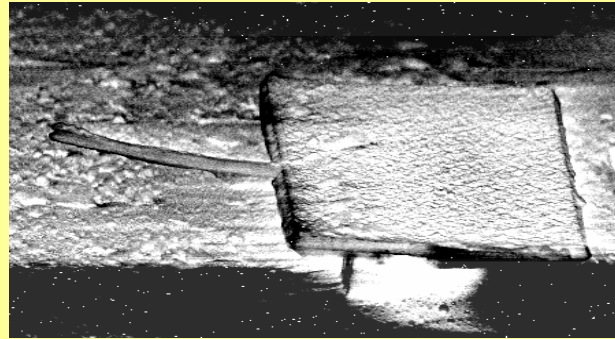
Confinement of acoustic wave  
in low-velocity layer

( "Quantized" modes  
in acoustic well )

# AlN/GaN/g-LiAlO<sub>2</sub> for MEMS/NEMS

// [0001]

// [11 $\bar{2}$ 0]

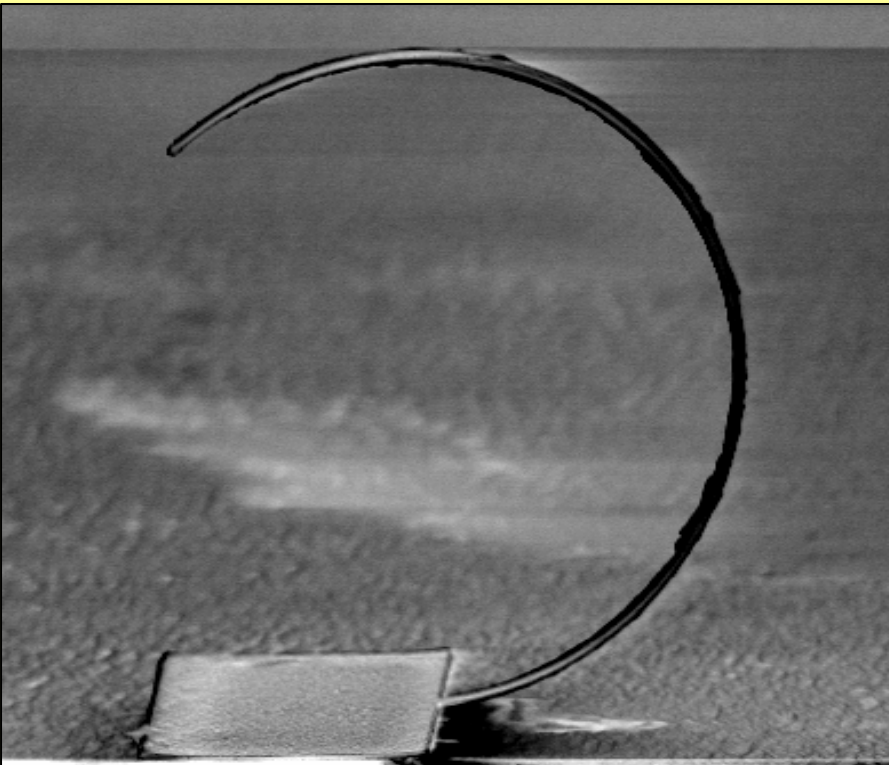


**Advantages of AlN**  
stiffness  
durability

	Width (mm)	Length (mm)
Cantilevers	0.2	4
Beams	0.2	6

Mismatch	$\Delta a/a$ (%)	$\Delta c/c$ (%)
AlN/GaN	2.4	3.9
GaN/ $\gamma$ -LiAlO <sub>2</sub>	-1.7	-0.3
AlN/ $\gamma$ -LiAlO <sub>2</sub>	0.7	3.6

# Strain relaxation at AlN-GaN interface



**Width**                    1 mm  
**AlN thickness**        45 nm  
**GaN thickness**       120 nm  
  
**Radius of loop**       11 mm

## Strain estimated from self-rolling

d(AlN) (nm)	d(GaN) (nm)	[1120̄]		[0001]	
		radius (mm)	$\Delta a/a$ (%)	radius (mm)	$\Delta c/c$ (%)
45	120	11	1.0	...	$\ll 1$
70	110	14	0.7	39	0.3
350	515	276	0.2	...	...



**Relaxation for thick AlN layers**



**Almost relaxed along the c axis**



**Nitrides seem to be good for everything**

**Dilute Nitrides**

(Ga,In)(N,P,As)

cubic

**Conventional Nitrides**

AlN, GaN, InN

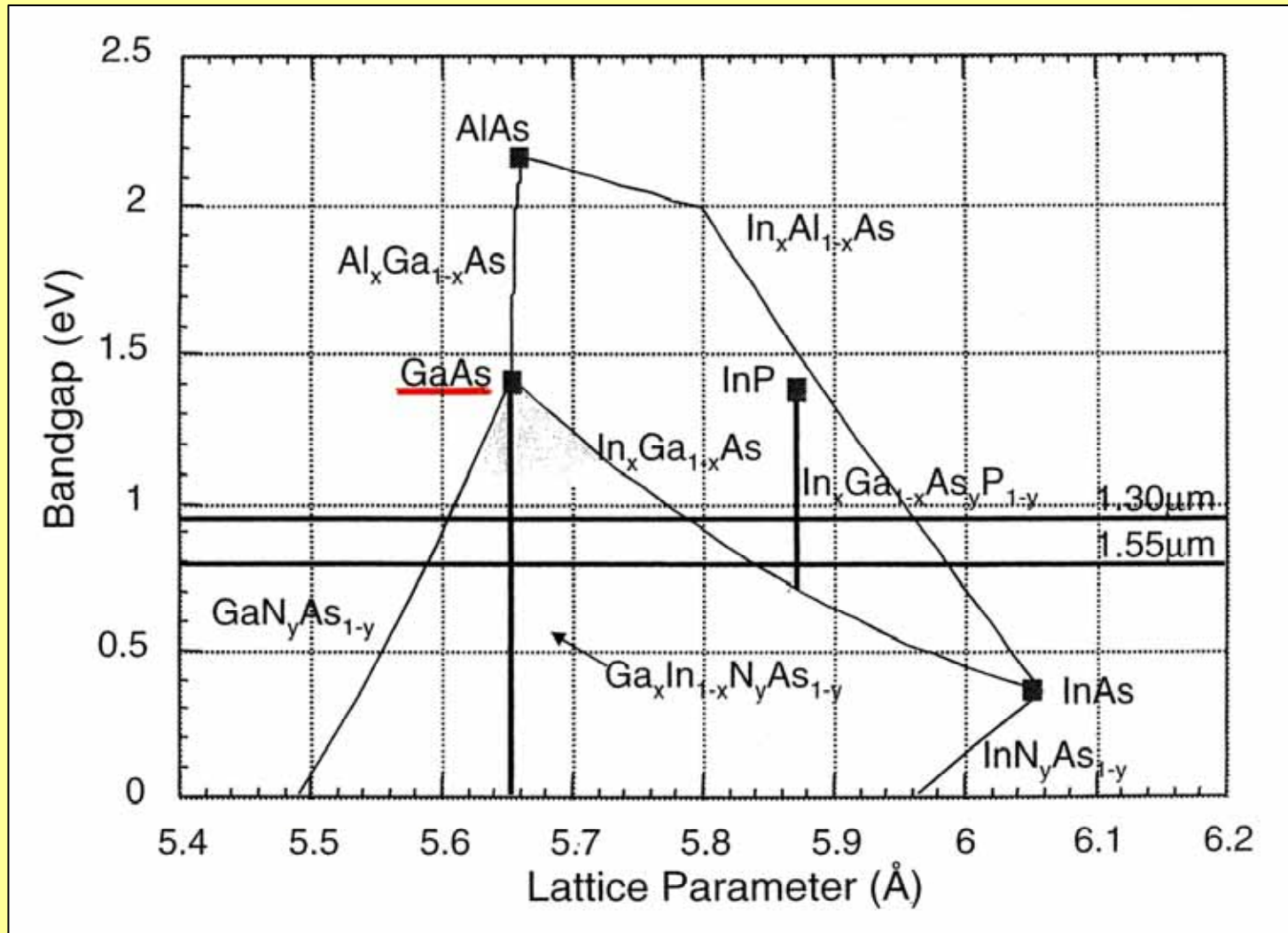
hexagonal (cubic)

**Ferromagnetic Nitrides**

TE- and RE-doped

hexagonal (cubic)

# Bandgap of dilute Nitrides





# **(Ga,In)(N,P,As) alloys lattice-matched to GaAs, GaP or Si (Dilute Nitrides)**

- **Infrared light emitter**
- **Multijunction solar cells**
- **Monolithic integration of Ga(N,P,As)/GaP heterostructure lasers with Si-CMOS circuits („Silicon Photonics“)**

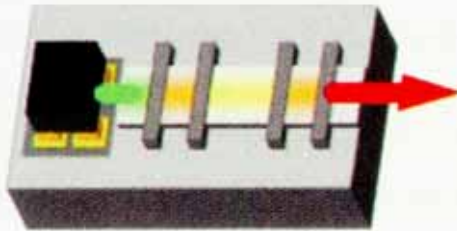
## **Problem areas for growth of these metastable alloys**

- **Large differences in atomic radii and electronegativities of constituent elements**
- **Large differences in bond strengths of constituent binaries**
  - **Tendency of alloy clustering enhanced**
- **Sound models about surface kinetics and growth mechanisms do not exist**
  - **Empirical optimization of growth conditions**
    - **Low substrate temperature for 2D growth**
    - **2D growth mode stabilized by surfactants?**
- **Post-growth annealing required to improve internal quantum efficiency**
  - **Point defects, antisite defects, ion damage ?**
  - **Deep electron and/or hole traps ?**

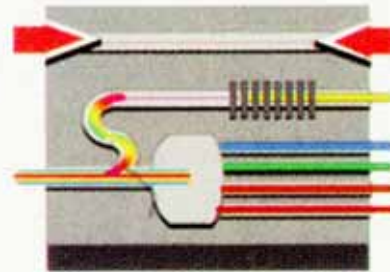


# Major building blocks for Si photonics

Light Sources

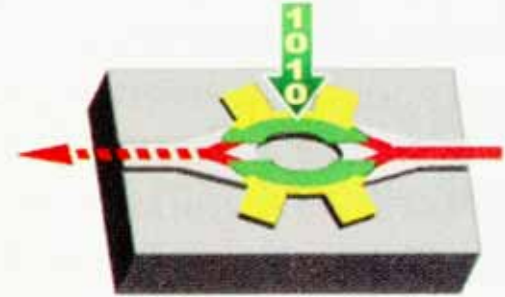


Guiding Light

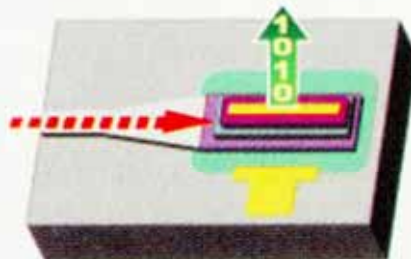


Waveguide Devices

Modulation



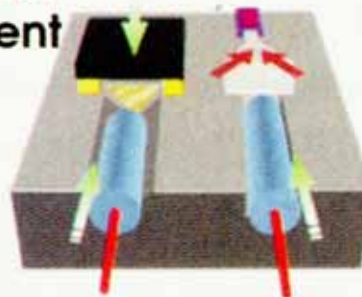
Photodetection



SiGe Photodetectors

Low-Cost Assembly

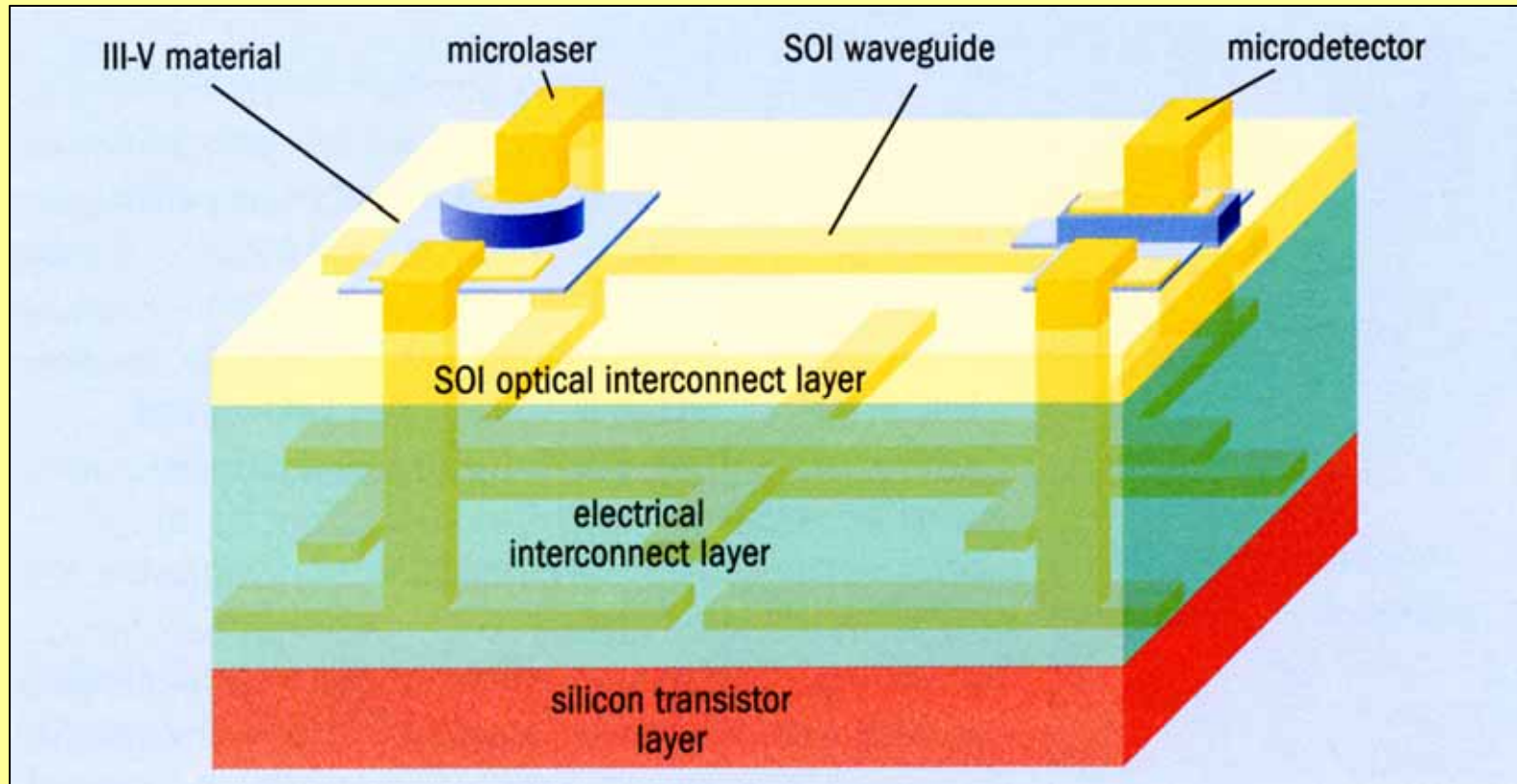
Passive Alignment



Intelligence

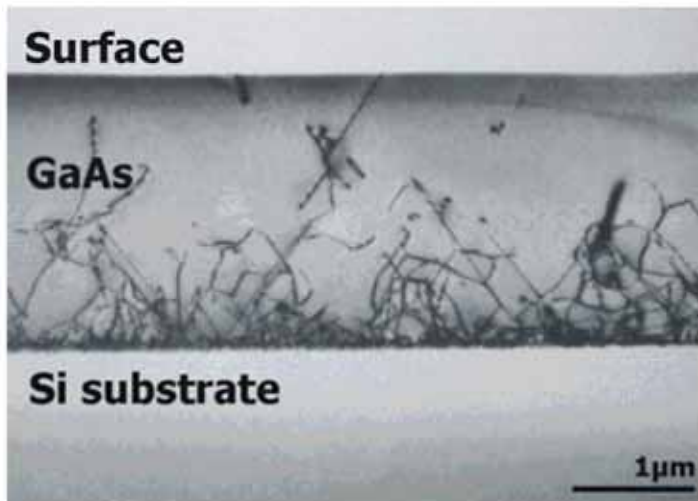


# On-chip optical interconnects

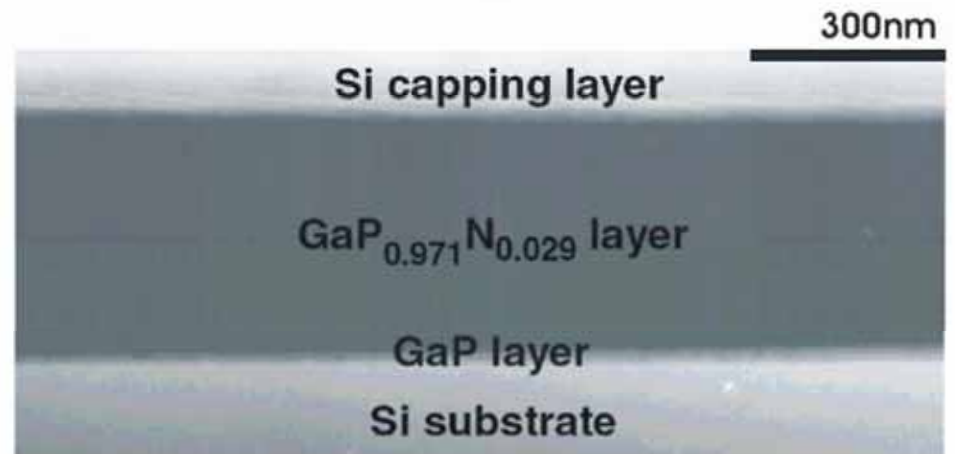


# TEM images

## GaAs/Si

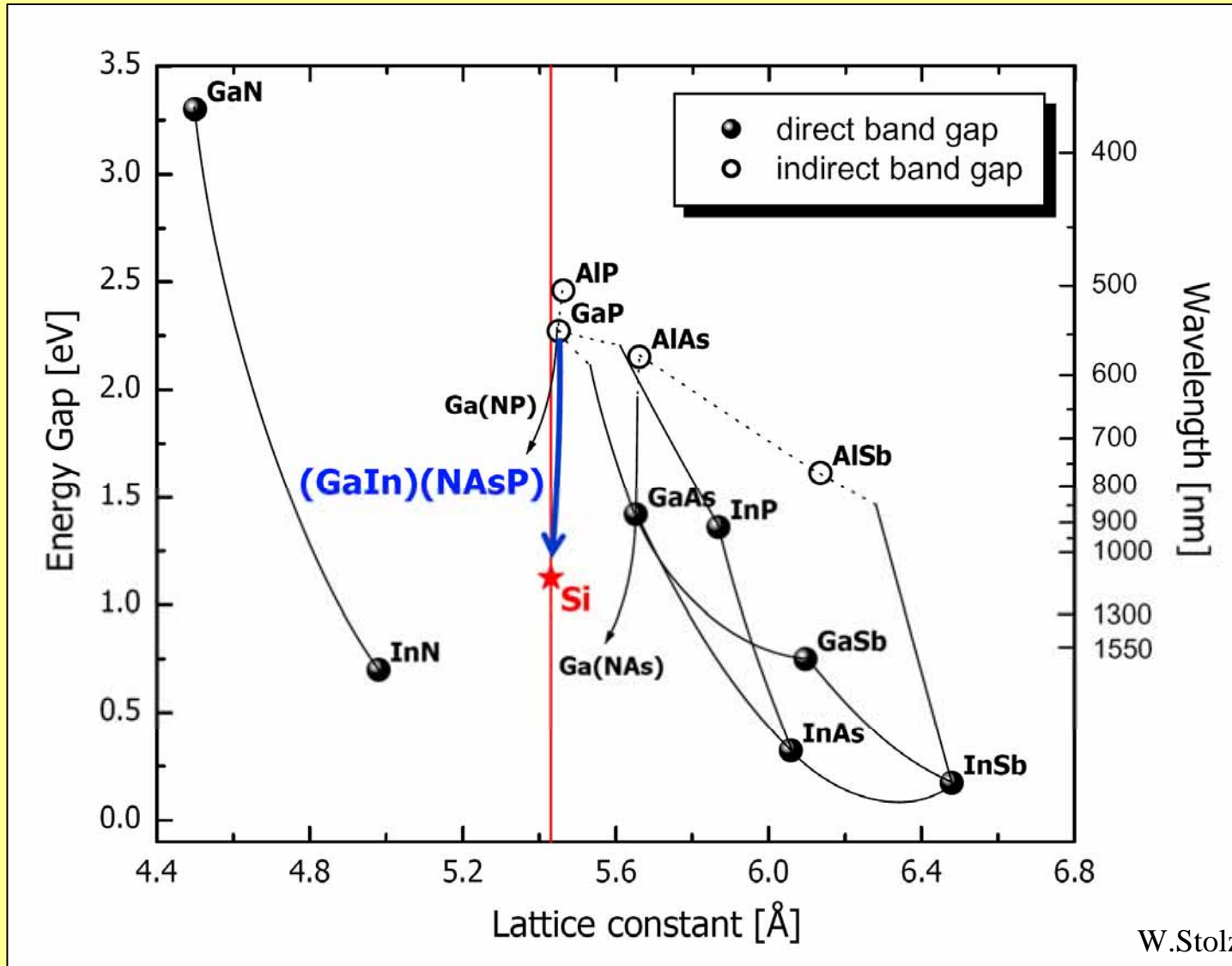


## GaP/Si

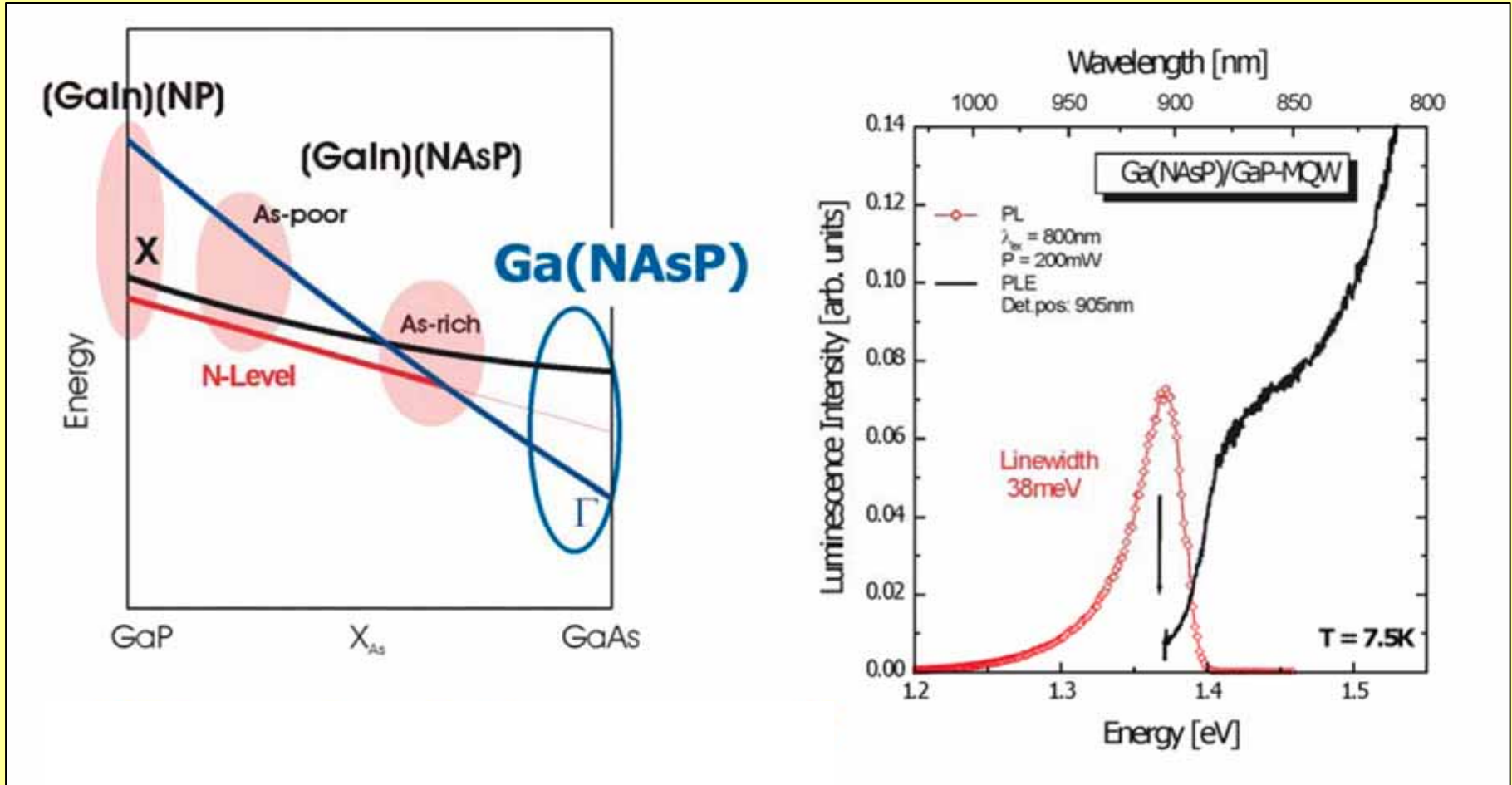


H. Yonezu and co-workers, Toyohashi Univ. Japan

# (Ga,In)(N,P,As) alloy

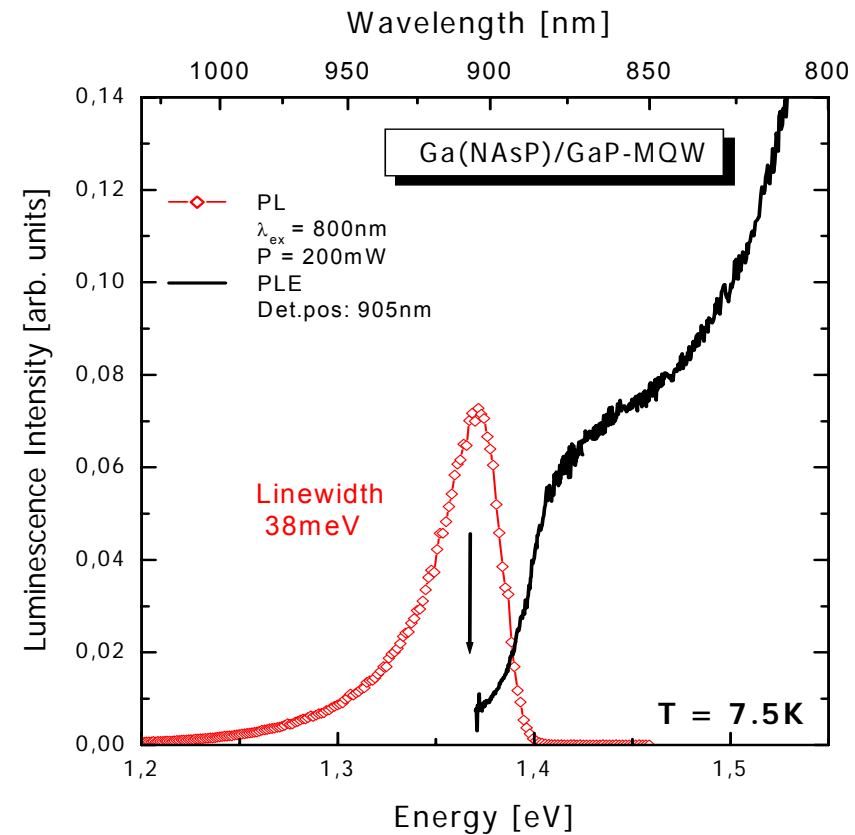
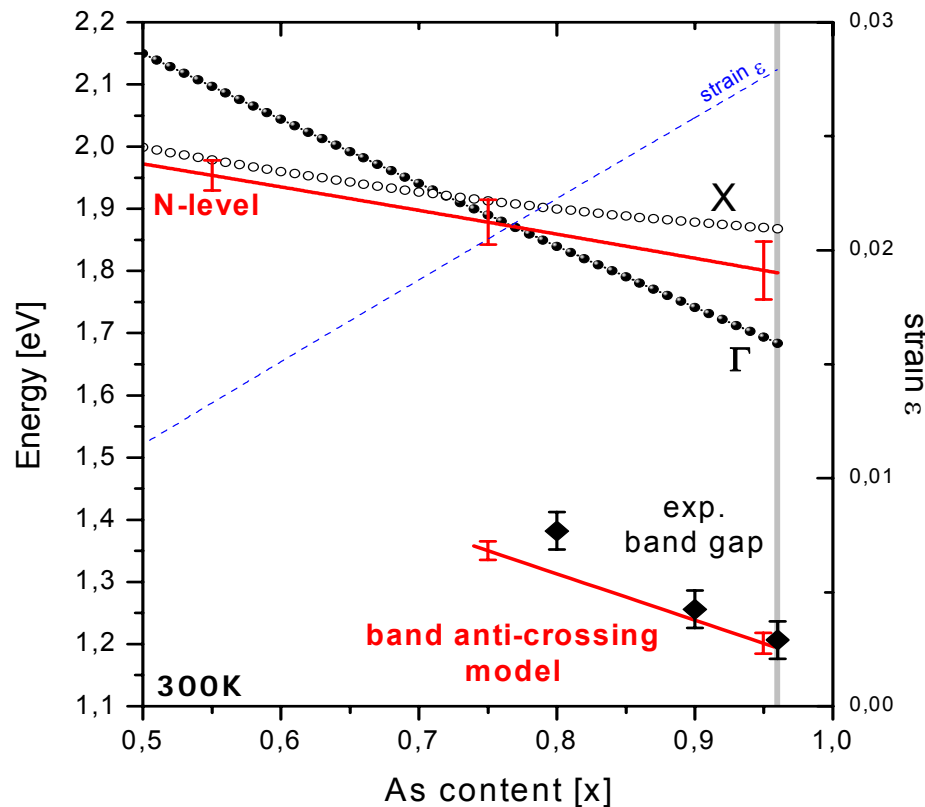


# Proof of concept



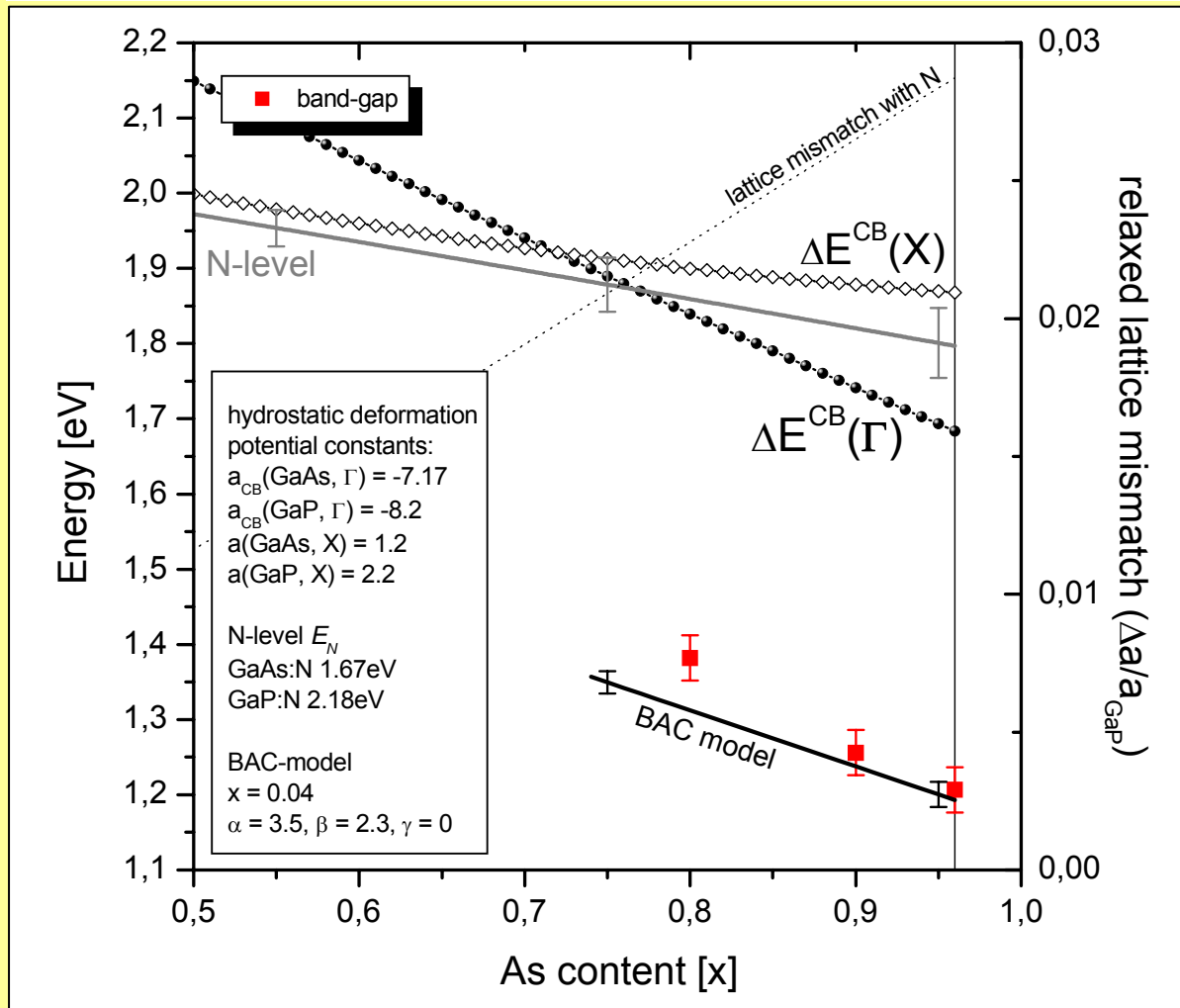
# Estimated vs experimental bandgap

Estimation: conduction band extrema  
Strained Ga(N<sub>0.04</sub>As<sub>x</sub>P<sub>1-x</sub>)



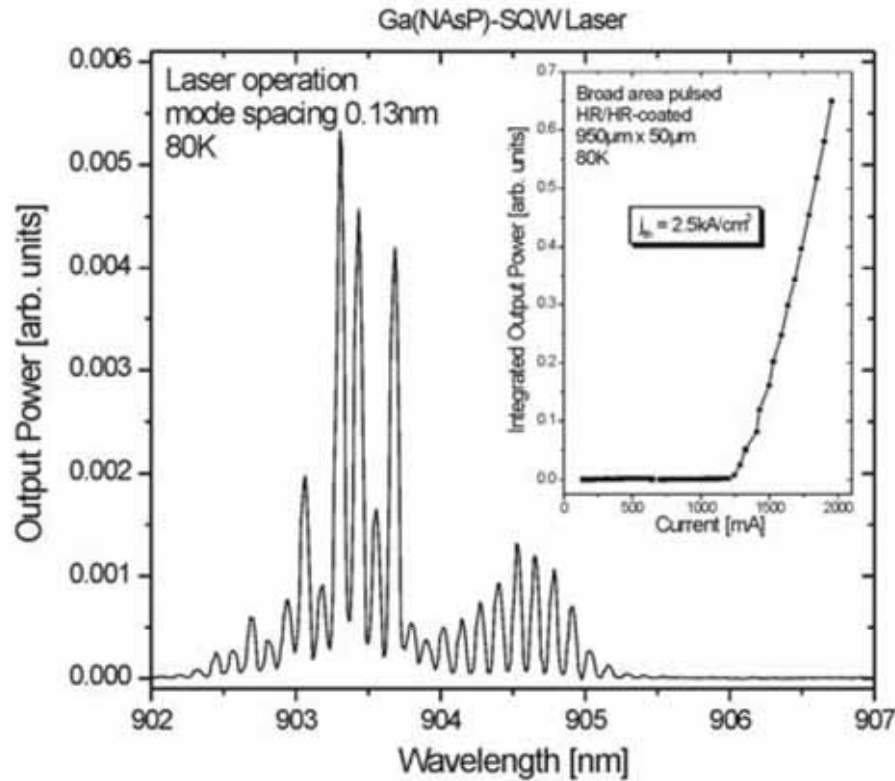
# Band anticrossing (BAC) model

$$E_- = \frac{1}{2} \cdot \left[ (E_\Gamma - \gamma x) + (E_N + \alpha x) - \sqrt{((E_\Gamma - \gamma x) + (E_N + \alpha x))^2 + 4 \cdot \beta^2 x} \right]$$



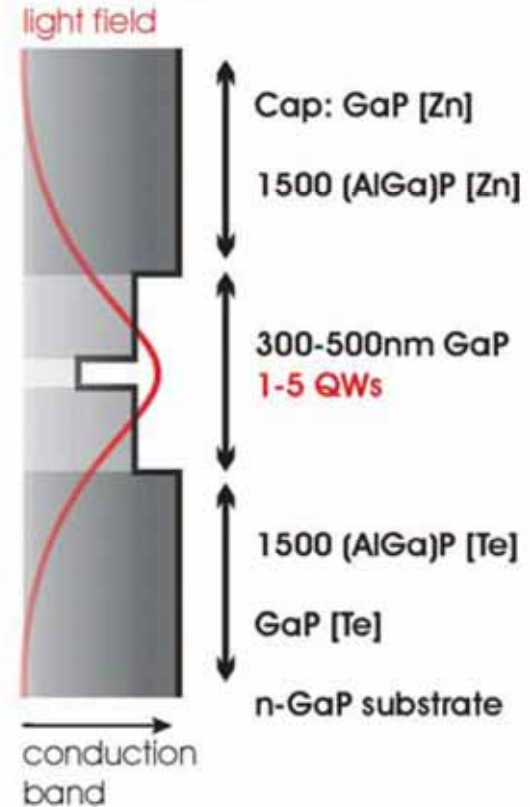
**Bandstructure  
Modification:  
Strained  
Ga(N<sub>0.04</sub>As<sub>x</sub>P<sub>1-x</sub>)**

# Ga(N,P,As) SQW laser



**Etched top-p<sup>+</sup> contact:**  
 **$j_{th} = 1.5 \text{ kA/cm}^2$**

semiconductor edge emitter







**Nitrides seem to be good for everything**

**Dilute Nitrides**

(Ga,In)(N,P,As)

cubic

**Conventional Nitrides**

AlN, GaN, InN

hexagonal (cubic)

**Ferromagnetic Nitrides**

TE- and RE-doped

hexagonal (cubic)



# Rare-earth (RE) doping of GaN

- Sharp RE intra-f-shell optical transitions allow light emission in the visible to infrared spectral range
  - Eu-doped GaN → 623 nm emission
  - Er-doped GaN → 1.55  $\mu\text{m}$  emission
- Isovalent RE<sup>3+</sup> ions on Ga lattice sites form electrically inert centers (no deep gap states)

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Ref: P. N. Favennec et al., Electron Lett. 25 (1989) 718  
Y. Q. Wang and A. J. Steckl, Appl. Phys. Lett. 82 (2003) 402  
J. S. Filhol et al., Appl. Phys. Lett. 84 (2004) 2841

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- Magnetic coupling of partially filled 4f-orbitals of RE<sup>3+</sup> ions possible  
→ weaker than d-orbitals in transition metals
- Gd has both partially filled 4f and 5d orbitals  
→ new coupling mechanism?

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Ref: M. Hashimoto et al., Jpn. J. Appl. Phys. 42 (2003) L1112  
N. Teraguchi et al., Solid State Commun. 122 (2002) 651

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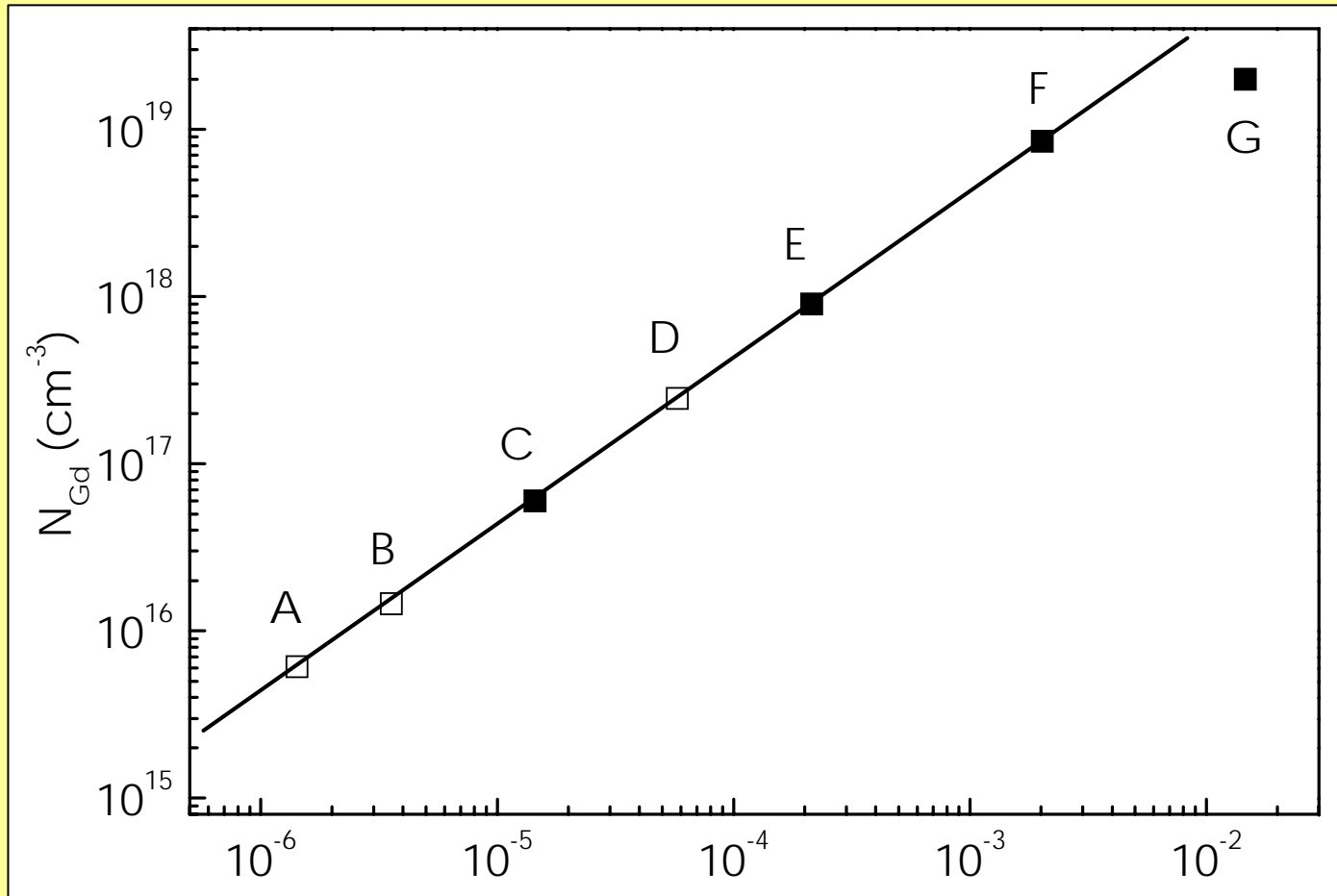
# Ferromagnetic nitrides

**The mean-field model of free holes mediating ferromagnetism via RKKY interaction have identified wide-gap semiconductors as ideal candidates to generate ferromagnetic semiconductors with high Curie temperature by doping with magnetic transition metal (TM) ions.**

**Numerous attempts with Mn, Cr, and Fe doping of GaN have yielded inconclusive results.**

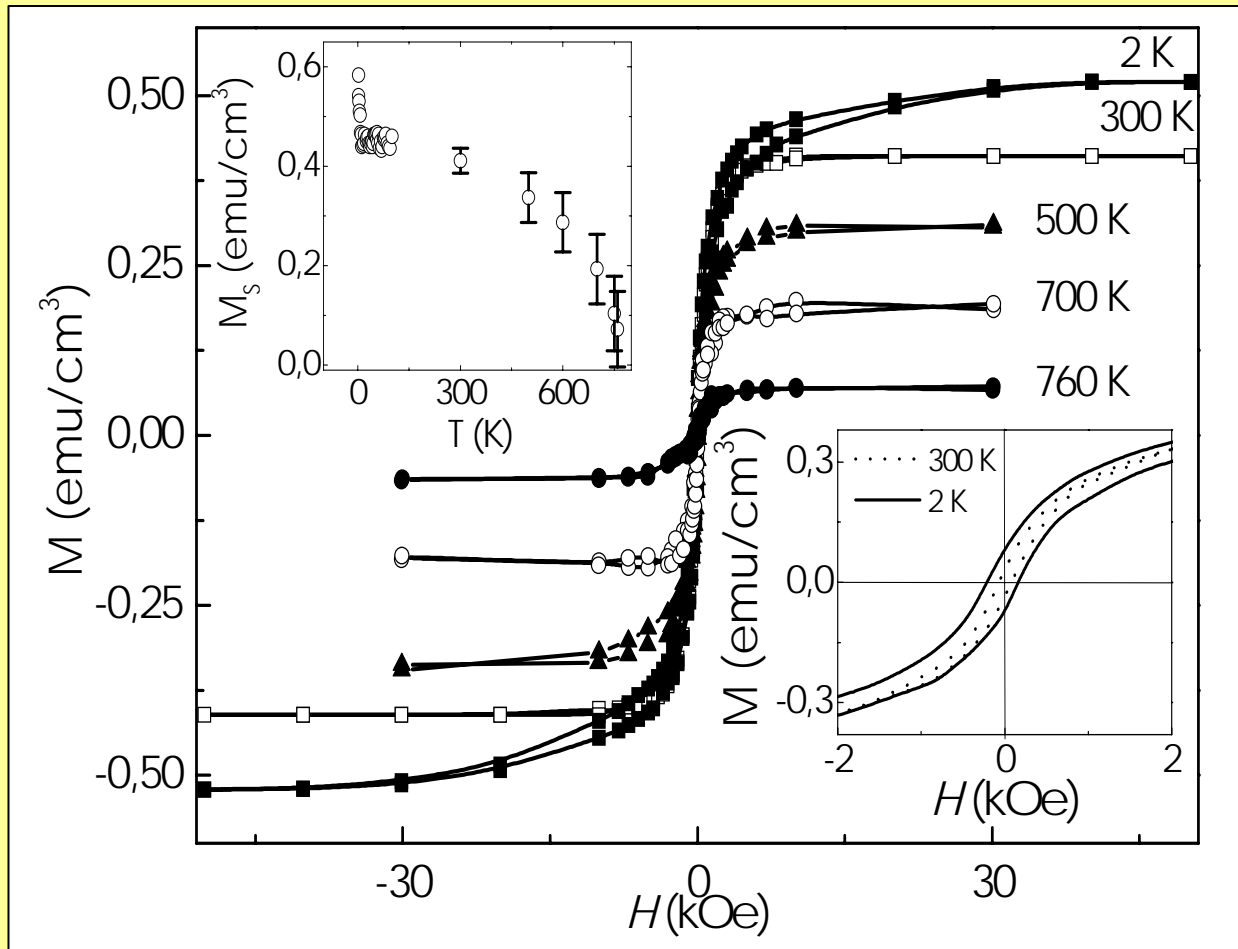
**Doping of GaN with rare earth elements (RE), like Eu, Er, Gd, is well established. Isoelectronic RE species are incorporated on Ga lattice sites.**

# Gd concentration vs Gd/Ga flux ratio



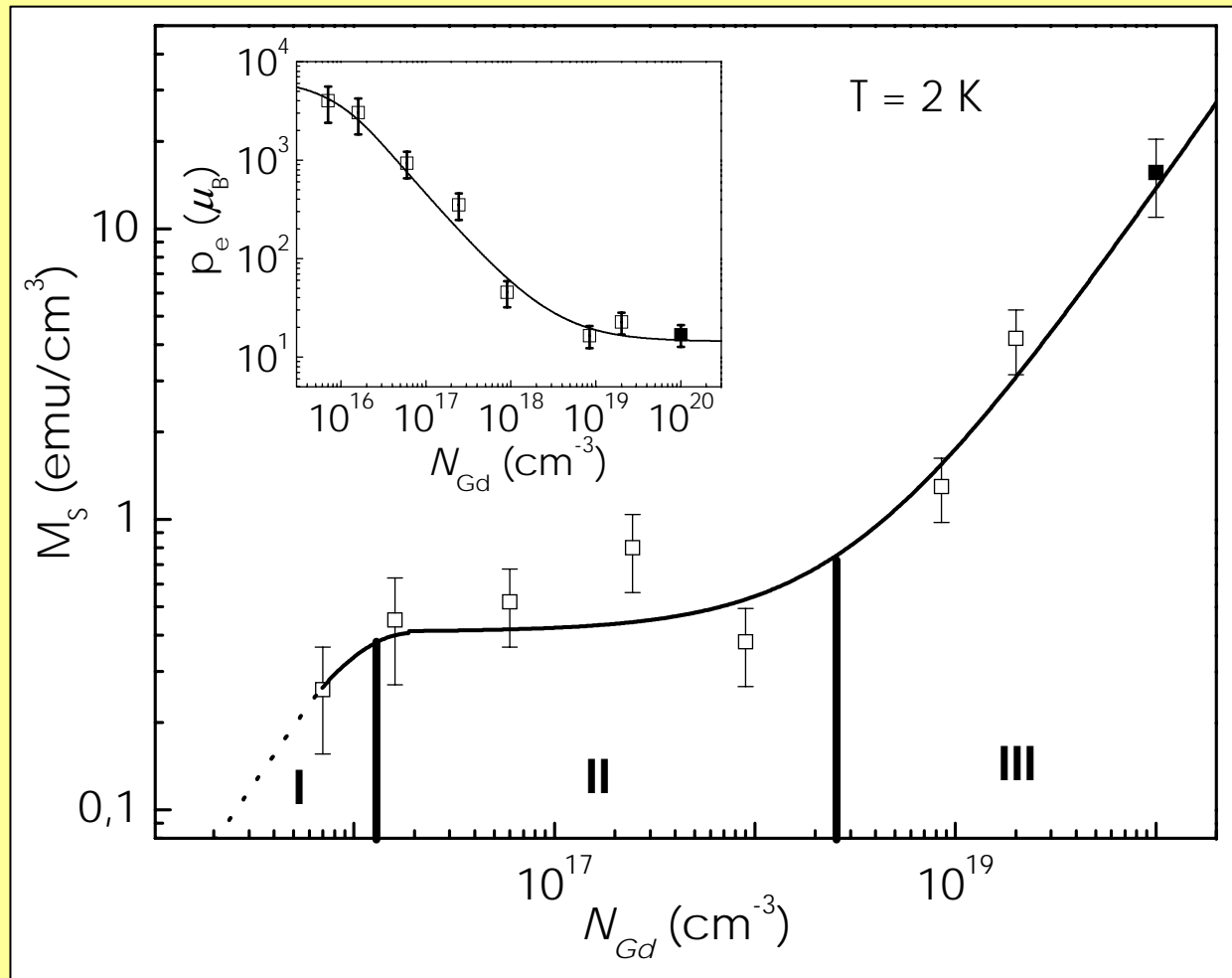
Unity sticking coefficient of Gd up to  $10^{19} \text{ cm}^{-3}$

# Magnetic hysteresis ( $[Gd] = 6 \times 10^{16} \text{ cm}^{-3}$ )



- Magnetization saturates at high fields  $\Rightarrow$  Ferromagnetism
- Superposition of two loops with different  $H_c$  and  $M_r$  at 2 K ?  
 $\rightarrow$  above 10 K phase with larger  $H_c$  and  $M_r$  disappears

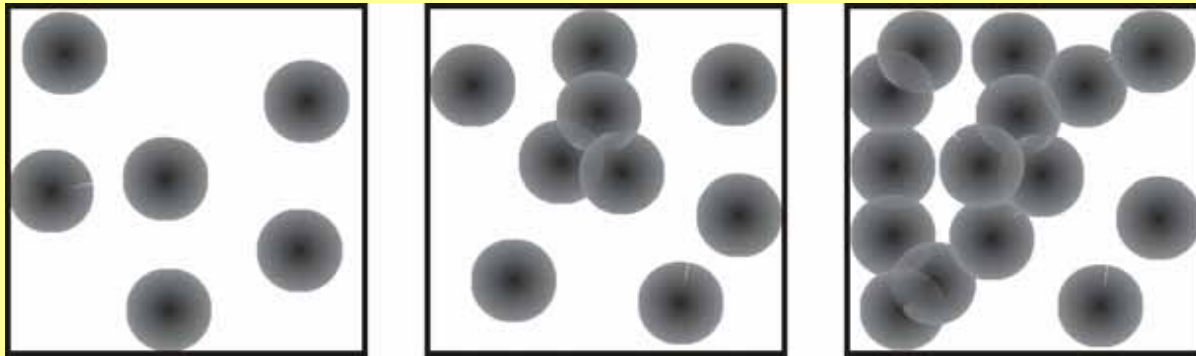
# Saturation magnetization vs [Gd]



Inset: Magnetic moment per Gd atom

# Empirical model for origins of colossal moment

**Gd atoms polarize the matrix**



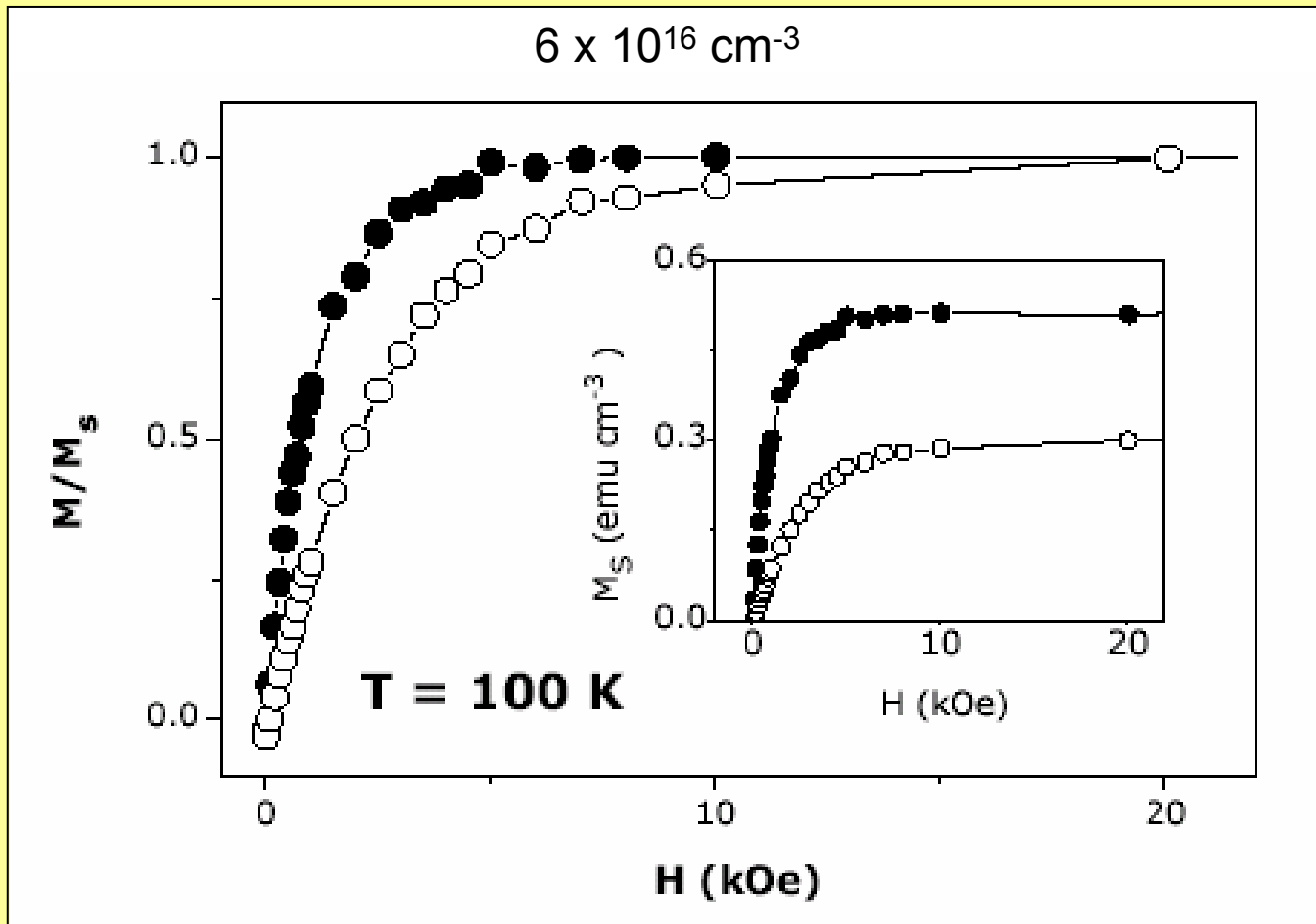
$$p_e = p_{Gd} + p_m v N_o/N_{Gd}; v = 1 - \exp(-v N_{Gd})$$

$p_e$  decreases as  $N_{Gd}$  is increased  $\rightarrow$  experimentally observed

**Overlap of spheres  $\rightarrow$  ferromagnetic coupling**

$T_c$  increases with  $N_{Gd}$   $\rightarrow$  experimentally observed

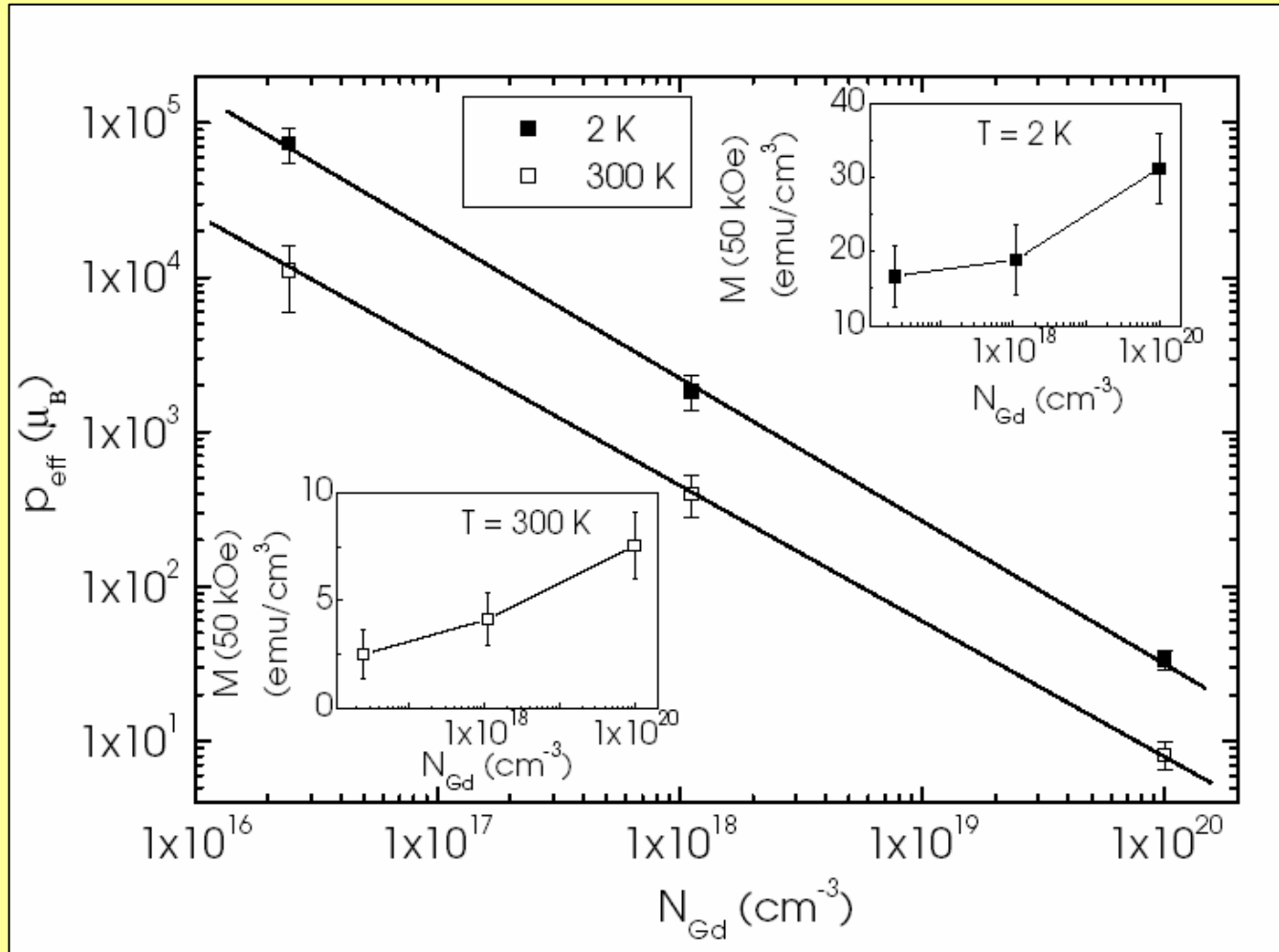
# Magnetization curves of Gd-doped GaN measured in two perpendicular directions



Saturation magnetization is smaller along hard axis

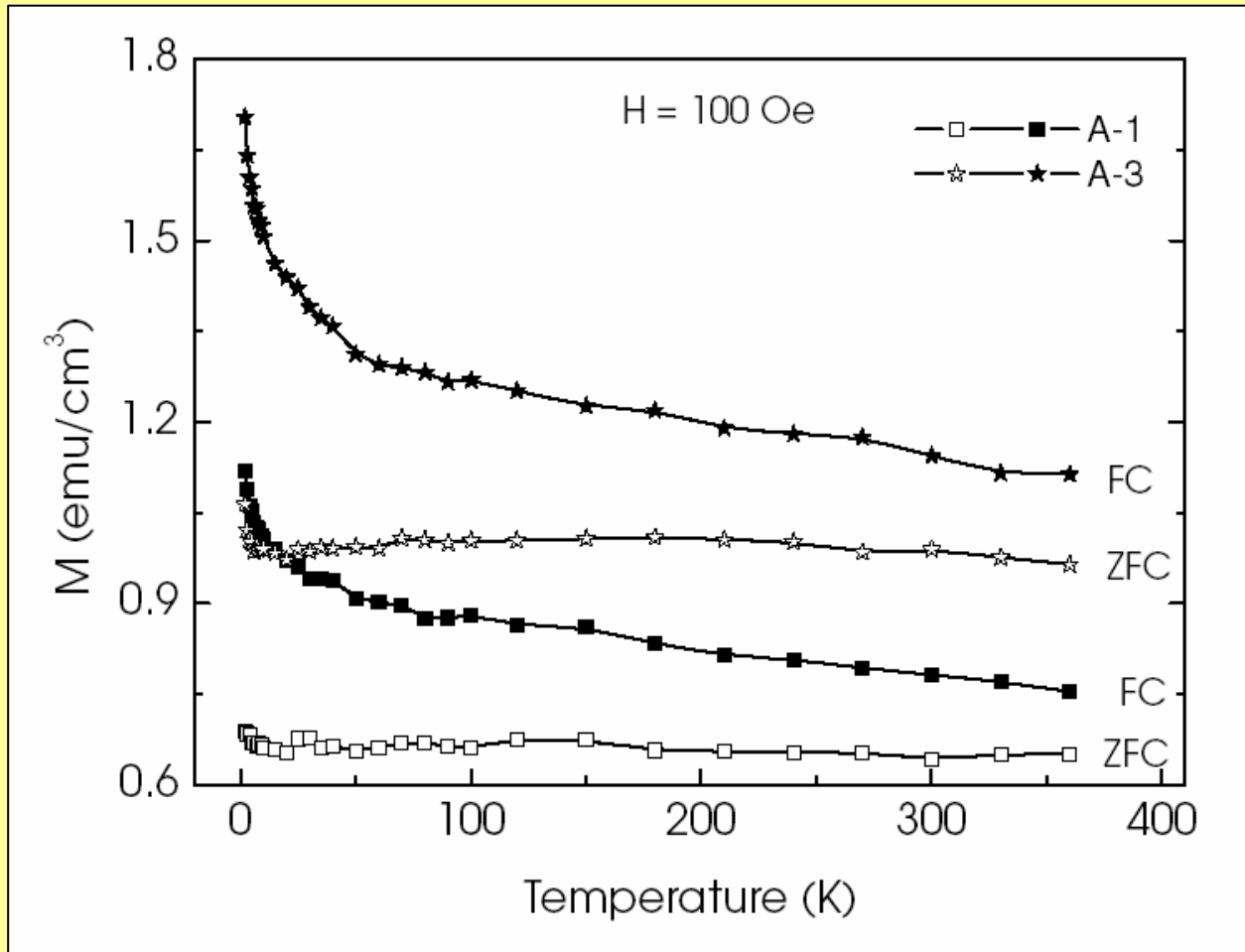


# Magnetic moment of Gd in implanted GaN



Saturation magnetization shown in insets

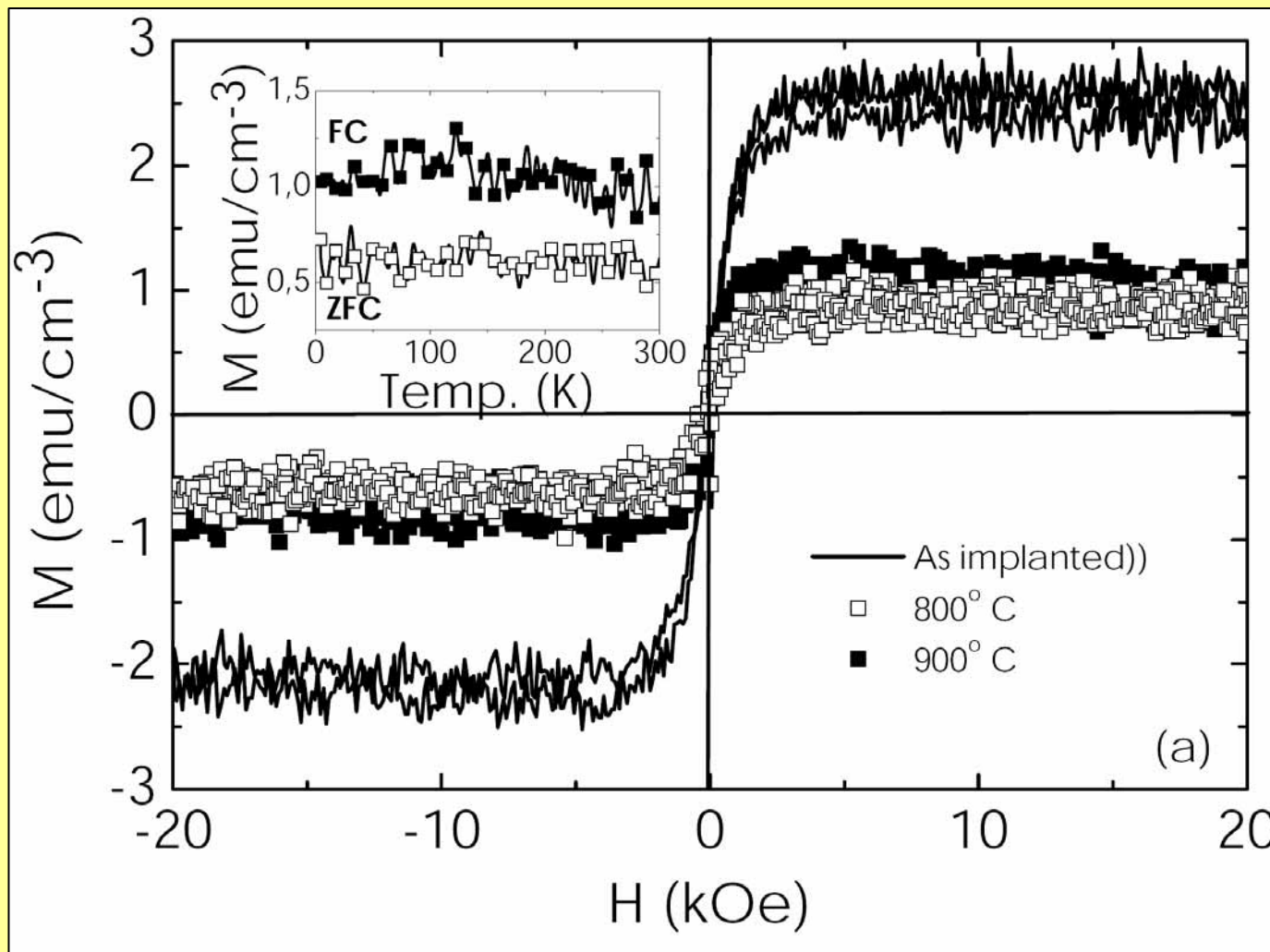
# FC and ZFC magnetization in Gd-implanted GaN



A-1  $2 \times 10^{16} \text{ cm}^{-3}$

A-3  $1 \times 10^{20} \text{ cm}^{-3}$

# Magnetic hysteresis of Gd-implanted GaN





# Results from Gd-doped GaN

Single-phase Gd-doped GaN layers show ferromagnetic behavior with in-plane easy axis and high  $T_c$

Colossal magnetic moment per Gd atom

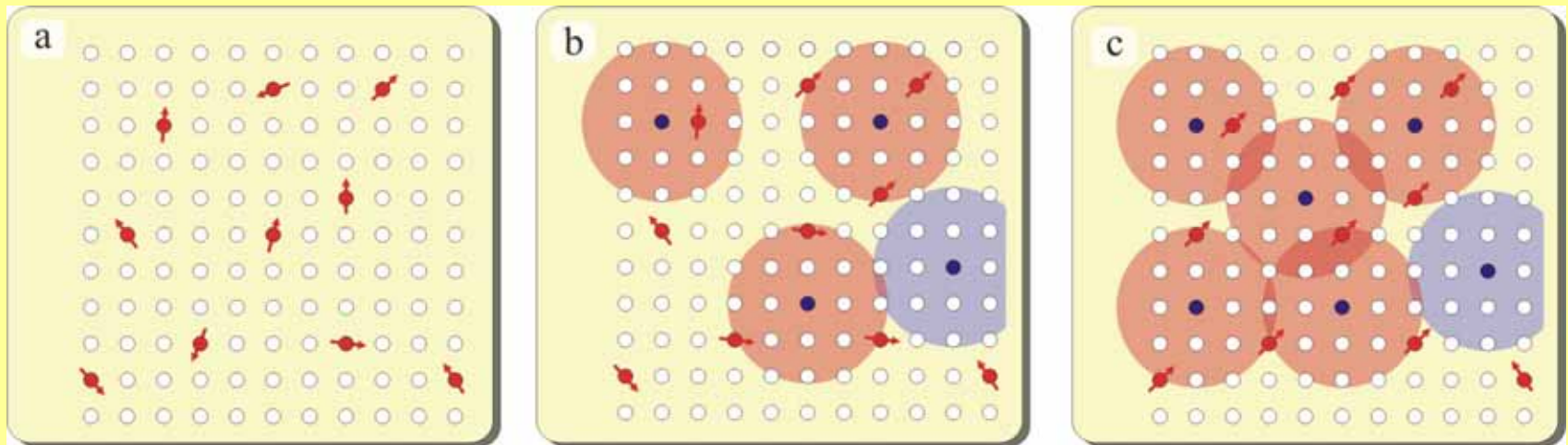
Coexistence of two ferromagnetic phases with different order temperatures

Measured saturation magnetization depends on orientation of the magnetic field

Structural defects play important role for magnitude of colossal magnetic moment per Gd atom

Empirical model based on polarization of entire GaN matrix by Gd dopants (an/or defects) can explain the observed colossal magnetic moment

# Role of defects in creating ferromagnetism in semiconductors (polaron model)





# III-Nitride nanostructures

Large number and wide variety of nano-objects consisting of III-Nitrides, including

- quantum wires
- nanowires
- nanocolumns
- nanorodes
- quantum dots
- nanodots
- etc

have been fabricated by different growth techniques.

Challenges are to have control over

- size (geometrical dimensions)
- uniformity
- placements
- interconnects (for electronic access)



# Concluding remarks



issues  
& events

## US condensed-matter community grapples with availability of crystalline samples

Crystal growing for physics measurements has fallen between the cracks in the US; without a turnaround, the country can't help but lag in the discovery of new materials and their applications.