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Selective Area Grown Semiconductor-Superconductor Hybrids: A Basis for Topological Networks

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We introduce selective area grown hybrid InAs/Al nanowires based on molecular beam epitaxy, allowing arbitrary semiconductor-superconductor networks containing loops and branches. Transport reveals a hard induced gap and unpoisoned $2e$ -periodic Coulomb blockade, with temperature dependent $1e$ features in agreement with theory. Coulomb peak spacing in parallel magnetic field displays overshoot, indicating an oscillating discrete near-zero subgap state consistent with device length. Finally, we investigate a loop network, finding strong spin-orbit coupling and a coherence length of several microns. These results demonstrate the potential of this platform for scalable topological networks among other applications.

Majorana zero modes (MZMs) at the ends of one-dimensional topological superconductors are expected to exhibit non-Abelian braiding statistics [1, 2], providing naturally fault-tolerant qubits [3, 4]. Proposed realizations of braiding [5, 6], interference-based topological qubits [7–9] and topological quantum computing architectures [10] require scalable nanowire networks. While relatively simple branched or looped wires can be realized by specialized growth methods [11, 12] or by etch- and gate-confined two-dimensional hybrid heterostructures [13–16], selective area growth [17] enables deterministic patterning of arbitrarily complex structures. This allows complex continuous patterns of semiconductor-superconductor hybrids and topological networks.

Following initial theoretical proposals [18, 19], a number of experiments have reported signatures of Majorana zero modes (MZMs) in hybrid semiconductor-superconductor nanowires [11], including zero-bias conductance peaks [15, 16, 20–26] and Coulomb blockade peak spacing oscillations [27, 28]. To date, experiments have used individual vapor-liquid-solid (VLS) nanowires [20–25] or gate-confined two-dimensional heterostructures [15, 16]. Within these approaches, constructing complex topological devices and networks containing branches and loops [5–10] is a challenge. Recently, branched and looped VLS growth has been developed toward this goal [12, 29].

In this Letter, we investigate a novel approach to the growth of semiconductor-superconductor hybrids that allows deterministic on-chip patterning of topological superconducting networks based on SAG. We characterize key physical properties required for building Majorana networks, including a hard superconducting gap, induced in the semiconductor, phase-coherence length of several microns, strong spin-orbit coupling, and Coulomb block-

ade peak motion compatible with interacting Majoranas. Overall, these properties show great promise for SAG-based topological networks.

Selective area growth was realized on an InP substrate with a pre-patterned SiO₂ mask. InAs wires were grown by molecular beam epitaxy (MBE), with one facet of the triangular cross-section covered with *in-situ* MBE grown Al [Fig. 1(a-c)]. The material growth is described in detail in Ref. [17]. Data from three devices are presented.

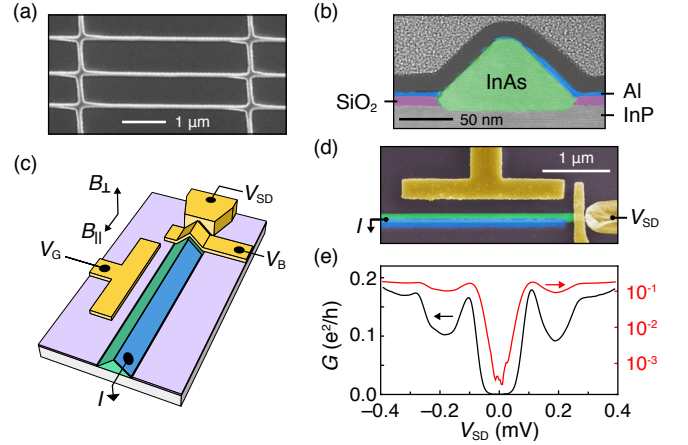


FIG. 1. (a) Scanning electron micrograph of a SAG hybrid network. (b) False-colored annular dark field scanning transmission electron micrograph of a nanowire cross-section displays InP substrate, InAs (green) nanowire, Al (blue) shell and SiO₂ (purple) mask. (c) Measurement set-up for device 1 showing the gate voltages and orientations of magnetic fields used in the measurements. (d) False-colored electron micrograph of device 1. (e) Differential conductance, G , as a function of source-drain bias, V_{SD} , in linear (black) and logarithmic (red) scales shows a hard superconducting gap.

Device 1 [Fig. 1(d)] consists of a single barrier at the end of a $4\ \mu\text{m}$ wire, defined by a lithographically patterned gate adjacent to a Ti/Au contact where the Al has been removed by wet etching. This device allowed density of states measurement at the end of the wire by means of bias spectroscopy, to investigate the superconducting proximity effect in the InAs. Evolution of Coulomb blockade in temperature and magnetic field was studied in Device 2 [Fig. 2(b)]—a hybrid quantum dot of length $1.1\ \mu\text{m}$ defined by two Ti/Au gates adjacent to etched-Al regions. The barrier voltages V_B were used to create tunneling barriers. The chemical-potential in the wires was tuned with gate voltage V_G . Device 3 was a micron-size square loop [Fig. 4(a)] with fully removed Al, which was used to extract phase coherence lengths from weak antilocalization (WAL) and Aharonov-Bohm (AB) oscillations. Standard ac lock-in measurements were carried out in a dilution refrigerator with a three-axis vector magnet. See Supplemental Material [30] for more detailed description of the growth, fabrication and measurement setup.

Differential conductance, G , in the tunneling regime, as a function of source-drain bias, V_{SD} , for Device 1 [Fig. 1(e)] at $V_G = -9.2\ \text{V}$ reveals a gapped density of states with two peaks at $V_{SD} = 110\ \mu\text{eV}$ and $280\ \mu\text{eV}$. We tentatively identify the two peaks with two populations of carriers in the semiconductor, the one with a larger gap residing at the InAs-Al interface and with a smaller at the InAs-InP. The zero-bias conductance is ~ 400 times lower than the above-gap conductance, a ratio exceeding VLS nanowire [12, 31, 32] and 2DEG devices [14], indicating a hard induced gap. We note, however, that co-tunneling through a quantum-dot or multichannel tunneling can enhance this ratio [33].

Transport through a Coulomb island geometry [Fig. 2] at low temperatures shows $2e$ -periodic peak spacing as a function of V_G . Coulomb diamonds at finite bias yield a charging energy $E_C = 60\ \mu\text{eV}$ (see Fig. S1 in Supplemental Material [30]), smaller than the induced gap, $\Delta^* \sim 100\ \mu\text{eV}$, as seen in Fig. 1(e). The zero-bias Coulomb blockade spacing evolves to even-odd and finally to $1e$ -periodic peaks with increasing temperature, T . The $2e$ to $1e$ transition in temperature does not result from the destruction of superconductivity, but rather arises due to the thermal excitation of quasiparticles on the island, as investigated previously in metallic islands [2, 35] and semiconductor-superconductor VLS nanowires [36].

A thermodynamic analysis of Coulomb blockade peak spacings is based on the difference in free energies, $F = F_O - F_E$, between even and odd occupied states. We consider a simple model that assumes a single induced gap Δ^* , not accounting for the double-peaked density of states in Fig. 1(e). At low temperatures ($T \ll E_C, \Delta^*$), F approaches Δ^* . Above a characteristic poisoning temperature, T_p , quasiparticles become thermally activated and F decreases rapidly to zero. For $F(T) > E_C$, Coulomb

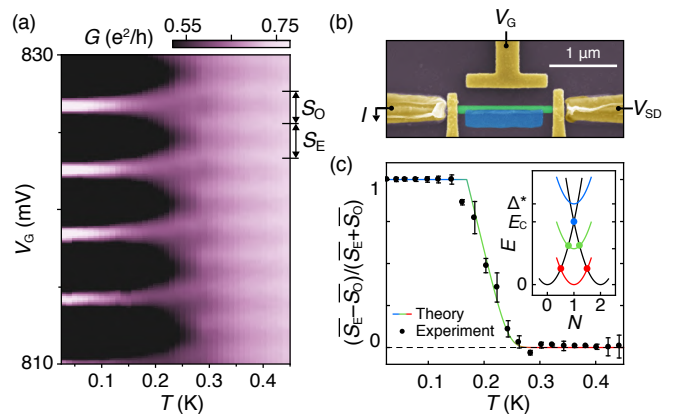


FIG. 2. (a) Conductance, G , of Device 2 as a function of applied gate voltage, V_G , and temperature, T . A characteristic $2e$ to $1e$ transition occurs around $200\ \text{mK}$. The color scale was adjusted for better visibility. (b) False-colored electron micrograph of Device 2. (c) Normalized even-odd peak spacing difference, $(\overline{S_E} - \overline{S_O})/(\overline{S_E} + \overline{S_O})$, from the measurements shown in (b) as a function of T . The error-bars were estimated using standard deviation of the peak spacing. The theoretical fit corresponds to an induced superconducting gap $\Delta^* = 190\ \mu\text{eV}$. Inset: Energy, E , of the device as a function of normalized gate voltage, N . Black (colour) parabolas correspond to even(odd)-parity ground state. Transport occurs at the charge degeneracy points indicated by filled circles.

peaks are $2e$ periodic with even peak spacings, $S_E \propto E_C$, independent of T . For $F(T) < E_C$, odd states become occupied, and the difference in peak spacing, $S_E - S_O$, decreases roughly proportional to F . A full analysis following Ref. [36] (see Supplemental Material [30]) yields the peak spacing difference

$$\begin{aligned} S_E - S_O &= \frac{2}{\eta e} \min(E_C, F) \\ &= (S_E + S_O) \min(1, F/E_C), \end{aligned} \quad (1)$$

where η is the dimensionless gate lever arm measured from Coulomb diamonds.

Figure 2(c) shows the measured even-odd difference in peak spacing, $(\overline{S_E} - \overline{S_O})/(\overline{S_E} + \overline{S_O})$, averaged over 4 peaks in Device 2, along with Eq. (1). Thermodynamic analysis shows an excellent agreement with the peak spacing data across the full range of temperatures. The fit uses an independently measured E_C , with the induced gap as a single fit parameter, yielding $\Delta^* = 190\ \mu\text{eV}$, a reasonable value that lies between the two density of states features in Fig. 1(e). The island remains unpoisoned below $T_p \sim 250\ \text{mK}$.

The evolution of Coulomb blockade peaks with parallel magnetic field, B_{\parallel} , is shown in Fig. 3(a). In this data set, peaks show even-odd periodicity at zero field due to a gate-dependent gap or a bound state at energy E_0 less than E_C . A subgap state results in even-state spacing proportional to $E_C + E_0$ and odd-state spacing $E_C - E_0$

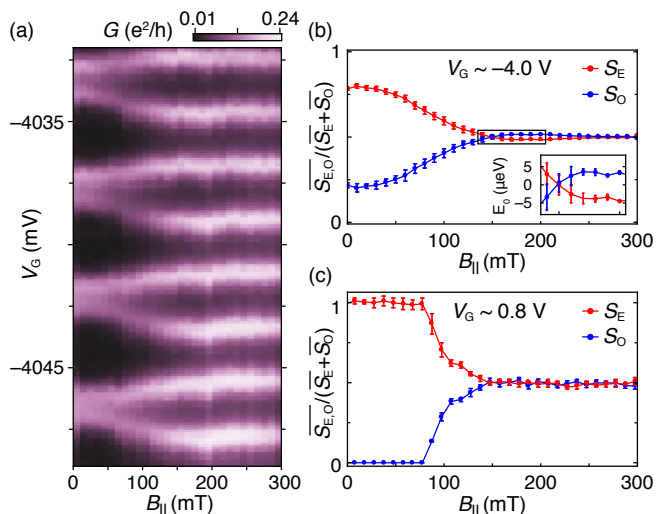


FIG. 3. (a) Conductance as a function of V_G and B_{\parallel} from Device 2, taken at $V_G \sim -4.0$ V, shows even-odd peak spacings at zero field transit to $1e$ spacing when the field is increased. (b) Normalized even and odd Coulomb peak spacings, $\overline{S_{E,O}}/(\overline{S_E} + \overline{S_O})$, from the measurements shown in a, as a function of B_{\parallel} . Inset: zoom-in of the peak spacing overshoot with amplitude of $7 \mu\text{eV}$ at $B_{\parallel} = 170$ mT. (c) Same as (b), but taken at $V_G \sim 0.8$ V. At positive gate voltage, the peaks become evenly spaced above $B_{\parallel} = 150$ mT. The error-bars in (a) and (b) were estimated using the standard deviation of the peak position.

[27] (see Supplemental Material [30]), giving

$$\begin{aligned} S_{E,O} &= \frac{1}{\eta e} [E_C \pm \min(E_C, E_0)] \\ &= \frac{S_E + S_O}{2} [1 \pm \min(1, E_0/E_C)]. \end{aligned} \quad (2)$$

Figure 3(b) shows the B_{\parallel} dependence of even and odd peak spacings, $\overline{S_{E,O}}/(\overline{S_E} + \overline{S_O})$, extracted from the data in Fig. 3(a), giving an effective g -factor of ~ 13 . Even and odd peak spacings become equal at $B_{\parallel} = 150$ mT, then overshoot at higher fields with a maximum amplitude corresponding to $(7 \pm 1) \mu\text{eV}$. At more positive gate voltages [Fig. 3(c)], where the carrier density is higher, peaks are $2e$ -periodic at zero field, then transition through even-odd to $1e$ -periodic Coulomb blockade without an overshoot, with an effective g -factor of ~ 31 .

Overshoot of peak spacing, with S_O exceeding S_E , indicates a discrete subgap state crossing zero energy [27, 37], consistent with interacting Majorana modes. The overshoot observed at more negative V_G is quantitatively in agreement with the overshoot seen in VLS wires of comparable length [27]. The absence of the overshoot and the increase of the g -factor at positive V_G is consistent with the gate-tunable carrier density in VLS wires [38].

To demonstrate fabrication and operation of a simple SAG network, we investigate the coherence of electron transport in the loop structure shown in Fig. 4(a), with

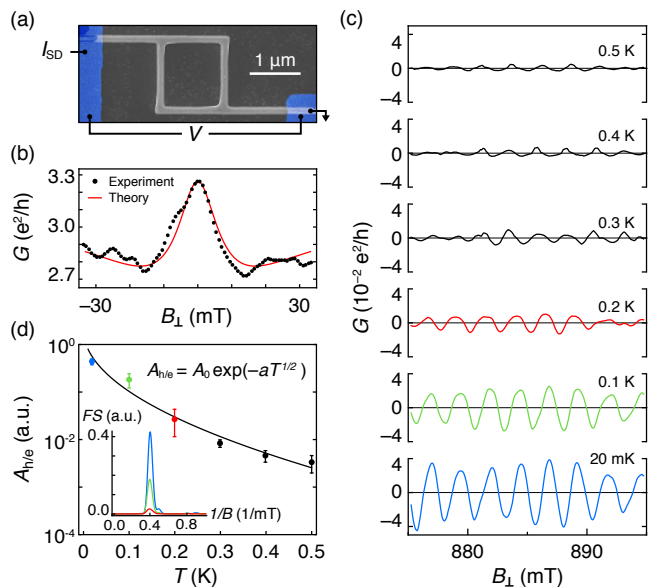


FIG. 4. (a) Electron micrograph of Device 3 with false-colored epitaxial Al contacts (blue) and overlaid 4-wire measurement setup. (b) Conductance, G , as a function B_{\perp} . Red curve is a theoretical magnetoconductance displaying the weak antilocalization effect in a system with spin-orbit length, $l_{SO} \sim 0.4 \mu\text{m}$, and phase-coherence length, $l_{\phi}^{\text{WAL}} \sim 1.2 \mu\text{m}$. (c) Aharonov-Bohm oscillations around $B_{\perp} \sim 900$ mT at different temperatures. (d) Amplitude of the h/e oscillations as a function of T . The exponential fit corresponds to the base-temperature phase-coherence length of $l_{\phi}^{\text{AB}} = 3.9 \mu\text{m}$. The error-bars correspond to the standard deviation between 4 data sets. Inset: Fourier spectra, FS , of the interference signal at 20 mK (blue), 100 mK (green) and 200 mK (red) from the measurements shown in (b).

the Al layer completely removed by wet etching. Conductance as a function of perpendicular magnetic field, B_{\perp} , shows a peak around zero magnetic field, characteristic of WAL [Fig. 4(b)]. A fit to a theoretical model for disordered quasi one-dimensional wires with strong spin-orbit coupling [39] yields an electron phase-coherence length $l_{\phi}^{\text{WAL}} \sim 1.2 \mu\text{m}$, and a spin-orbit length $l_{SO} \sim 0.4 \mu\text{m}$.

Upon suppressing WAL with a large perpendicular field periodic conductance oscillations are observed [Fig. 4(c)] with period $\Delta B_{\perp} = 2.5$ mT corresponding to h/e AB oscillations with area $1.7 \mu\text{m}^2$, matching the lithographic area of the loop. The oscillation amplitude, $A_{h/e}$, measured from the power spectral band around h/e [Fig. 4(d), inset] was observed to decrease with increasing temperature as seen in Fig. 4(d). Taking $A_{h/e} \propto \exp[-L/l_{\phi}^{\text{AB}}(T)]$ with the loop circumference $L = 5.2 \mu\text{m}$ and $l_{\phi}^{\text{AB}} \propto T^{-1/2}$ for a diffusive ring [40], a fit of the logarithmic amplitude $\log(A_{h/e}) = \log(A_0) - aT^{1/2}$ yields $\log(A_0) \sim 0.7$ and $a \sim 9 \text{K}^{-1/2}$ (Fig. 4d), giving a base-temperature phase-coherence length $l_{\phi}^{\text{AB}}(20 \text{mK}) \sim 4 \mu\text{m}$.

The discrepancy between the extracted l_{ϕ}^{WAL} and l_{ϕ}^{AB} has previously been observed in an experiment on GaAs/AlGaAs-based arrays of micron-sized loops [41]. It has been argued theoretically that WAL and AB interference processes are governed by different dephasing mechanisms [40]. As a result, l_{ϕ}^{WAL} and l_{ϕ}^{AB} have different temperature dependences.

Our results show that selective area grown hybrid nanowires are an excellent platform for scalable Majorana networks. The hard induced superconducting gap and $2e$ -periodic Coulomb oscillations imply strongly suppressed quasiparticle poisoning. The overshoot of Coulomb peak spacing in a parallel magnetic field indicates the presence of a discrete low-energy state and is a signature of MZMs. The measured SAG-based network exhibits strong spin-orbit coupling and phase-coherent transport. Furthermore, the ability to design arbitrary hybrid wire networks—a requirement for realizing topological quantum information processing—is readily achievable in SAG. Future work on SAG-based hybrid networks will focus on spectroscopy, correlations, interferometry, and manipulation of MZMs.

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[1] A. Y. Kitaev, *Physics-Uspekhi* **131**, 130-136 (2001).
 [2] N. Read, and D. Green, *Phys. Rev. B* **61**, 10267 (2000).
 [3] C. Nayak, S. H. Simon, A. Stern, M. Freedman, and S. Das Sarma, *Rev. Mod. Phys.* **80**, 1083 (2008).
 [4] S. Das Sarma, M. Freedman, and C. Nayak, *NJP Quantum Information* **1**, 15001 (2015).
 [5] J. Alicea, Y. Oreg, G. Refael, F. von Oppen, and M. P. A. Fisher, *Nat. Phys.* **7**, 412 (2011).
 [6] D. Aasen, Michael Hell, R. V. Mishmash, A. Higginbotham, J. Danon, M. Leijnse, T. S. Jespersen, J. A. Folk, C. M. Marcus, K. Flensberg, and J. Alicea, *Phys. Rev. X* **6**, 031016 (2016).
 [7] K. Flensberg, *Phys. Rev. Lett.* **106**, 090503 (2011).
 [8] S. Plugge, A. Rasmussen, R. Egger, and K. Flensberg, *New J. Phys.* **19**, 012001 (2017).
 [9] S. Vijay, and L. Fu, *Phys. Rev. B.* **94**, 235446 (2016).

[10] T. Karzig, C. Knapp, R. M. Lutchyn, P. Bonderson, M. Hastings, C. Nayak, J. Alicea, K. Flensberg, S. Plugge, Y. Oreg, C. M. Marcus, and M. H. Freedman, *Phys. Rev. B.* **95**, 235305 (2017).
 [11] P. Krogstrup, N. L. B. Ziino, W. Chang, S. M. Albrecht, M. H. Madsen, E. Johnson, J. Nygård, C. M. Marcus, and T. S. Jespersen, *Nat. Mat.* **14**, 400 (2015).
 [12] S. Gazibegovic, D. Car, H. Zhang, S. C. Balk, J. A. Logan, M. W. A. de Moor, M. C. Cassidy, R. Schmits, D. Xu, G. Wang, P. Krogstrup, R. L. M. Op het Veld, K. Zuo, Y. Vos, J. Shen, D. Bouman, B. Shojaei, D. Pennachio, J. S. Lee, P. J. van Veldhoven, S. Koelling, M. A. Verheijen, L. P. Kouwenhoven, C. J. Palmstrøm, and E. P. A. M. Bakkers, *Nature* **548**, 434 (2017).
 [13] J. Shabani, M. Kjaergaard, H. J. Suominen, Y. Kim, F. Nichele, K. Pakrouski, T. Stankevic, R. M. Lutchyn, P. Krogstrup, R. Feidenhans'l, S. Kraemer, C. Nayak, M. Troyer, C. M. Marcus, and C. J. Palmstrøm, *Phys. Rev. B* **93**, 155402 (2016).
 [14] M. Kjaergaard, F. Nichele, H. J. Suominen, M. P. Nowak, M. Wimmer, A. R. Akhmerov, J. A. Folk, K. Flensberg, J. Shabani, C. J. Palmstrøm, and C. M. Marcus, *Nat. Commun.* **7**, 12841 (2016).
 [15] H. J. Suominen, M. Kjaergaard, A. R. Hamilton, J. Shabani, C. J. Palmstrøm, C. M. Marcus, and F. Nichele, *Phys. Rev. Lett.* **119**, 176805 (2017).
 [16] F. Nichele, A. C. C. Drachmann, A. M. Whiticar, E. C. T. O'Farrell, H. J. Suominen, A. Fornieri, T. Wang, G. C. Gardner, C. Thomas, A. T. Hatke, P. Krogstrup, M. J. Manfra, K. Flensberg, and C. M. Marcus, *Phys. Rev. Lett.* **119**, 136803 (2017).
 [17] F. Krizek, J. E. Sestoft, P. Aseev, S. Marti-Sanchez, S. Vaitiekėnas, L. Casparis, S. A. Khan, Y. Liu, T. Stankevič, A. M. Whiticar, A. Fursina, F. Boekhout, R. Koops, E. Uccelli, L. P. Kouwenhoven, C. M. Marcus, J. Arbiol, P. Krogstrup, arXiv:1802.07808 (2018).
 [18] R. Lutchyn, J. D. Sau, and S. Das Sarma, *Phys. Rev. Lett.* **105**, 077001 (2010).
 [19] Y. Oreg, G. Refael, and F. von Oppen, *Phys. Rev. Lett.* **105**, 177002 (2010).
 [20] V. Mourik, K. Zuo, S. M. Frolov, S. R. Plissard, E. P. A. M. Bakkers, and L. P. Kouwenhoven, *Science* **336**, 1003 (2012).
 [21] M. T. Deng, C. L. Yu, G. Y. Huang, M. Larsson, P. Caroff, and H. Q. Xu, *Nano Lett.* **12**, 6414 (2012).
 [22] A. Das, Y. Ronen, Y. Most, Y. Oreg, M. Heiblum, and H. Shtrikman, *Nat. Phys.* **8**, 887 (2012).
 [23] Ö. Gül, H. Zhang, J. D. S. Bommer, M. W. A. de Moor, D. Car, S. R. Plissard, E. P. A. M. Bakkers, A. Geresdi, K. Watanabe, T. Taniguchi, and Leo P. Kouwenhoven, *Nat. Nanotech.* <https://doi.org/10.1038/s41565-017-0032-8>, (2018).
 [24] M. T. Deng, S. Vaitiekėnas, E. B. Hansen, J. Danon, M. Leijnse, K. Flensberg, J. Nygård, P. Krogstrup, and C. M. Marcus, *Science* **354**, 6319 (2016).
 [25] H. Zhang, C.-X. Liu, S. Gazibegovic, D. Xu, J. A. Logan, G. Wang, N. van Loo, J. D. S. Bommer, M. W. A. de Moor, D. Car, R. L. M. Op het Veld, P. J. van Veldhoven, S. Koelling, M. A. Verheijen, M. Pendharkar, D. J. Pennachio, B. Shojaei, J. S. Lee, C. J. Palmstrøm, E. P. A. M. Bakkers, S. Das Sarma, L. P. Kouwenhoven, arXiv:1710.10701 (2017).
 [26] J. E. Sestoft, T. Kanne, A. N. Gejl, M. von Soosten,

- J. S. Yodh, D. Sherman, B. Tarasinski, M. Wimmer, E. Johnson, M.-T. Deng, J. Nygård, T. S. Jespersen, C. M. Marcus, P. Krogstrup, arXiv:1711.06864 (2017).
- [27] S. M. Albrecht, A. P. Higginbotham, M. Madsen, F. Kuemmeth, T. S. Jespersen, J. Nygård, P. Krogstrup, and C. M. Marcus, *Nature* **531**, 7593 (2016).
- [28] D. Sherman, J. S. Yodh, S. M. Albrecht, J. Nygård, P. Krogstrup, and C. M. Marcus, *Nat. Nanotechnol.* **12** 212 (2017).
- [29] F. Krizek, T. Kanne, D. Razmadze, E. Johnson, J. Nygård, C. M. Marcus, and P. Krogstrup, *Nano Lett.* **17**, 6090 (2017).
- [30] See Supplemental Material for supporting information.
- [31] W. Chang, S. M. Albrecht, T. S. Jespersen, F. Kuemmeth, P. Krogstrup, J. Nygård, and C. M. Marcus, *Nat. Nanotechnol.* **10**, 232 (2015).
- [32] H. Zhang, Ö. Gül, S. Conesa-Boj, M. P. Nowak, M. Wimmer, K. Zuo, V. Mourik, F. K. de Vries, J. van Veen, M. W. A. de Moor, J. D. S. Bommer, D. J. van Woerkom, D. Car, S. R. Plissard, E. P. A. M. Bakkers, M. Quintero-Pérez, M. C. Cassidy, S. Koelling, S. Goswami, K. Watanabe, T. Taniguchi, and L. P. Kouwenhoven, *Nat. Commun.* **8**, 16025 (2017).
- [33] C. W. J. Beenakker, *Phys. Rev. B* **46**, 12841 (1992).
- [34] M. T. Tuominen, J. M. Hergenrother, T. S. Tighe, and M. Tinkham, *Phys. Rev. Lett.* **69**, 1997 (1992).
- [35] P. Lafarge, P. Joyez, D. Esteve, C. Urbina, and M. H. Devoret, *Phys. Rev. Lett.* **70**, 994 (1993).
- [36] A. P. Higginbotham, S. M. Albrecht, G. Kiršanskas, W. Chang, F. Kuemmeth, P. Krogstrup, T. S. Jespersen, J. Nygård, K. Flensberg, and C. M. Marcus, *Nat. Phys.* **11**, 1017 (2015).
- [37] C.-K. Chiu, J. D. Sau, and S. Das Sarma, *Phys. Rev. B.* **96**, 054504 (2017).
- [38] S. Vaitiekėnas, M.-T. Deng, J. Nygård, P. Krogstrup, and C. M. Marcus, arXiv:1710.04300 (2017).
- [39] C. Kurdak, A. Chang, A. Chin, and T. Chang, *Phys. Rev. B.* **46**, 6846 (1992).
- [40] T. Ludwig and A. D. Mirlin, *Phys. Rev. B.* **69**, 193306 (2004).
- [41] M. Ferrier, A. C. H. Rowe, S. Guéron, H. Bouchiat, C. Texier, and G. Montambaux, *Phys. Rev. Lett.* **100**, 146802 (2008).

Supplemental Material:

Selective Area Grown Semiconductor-Superconductor Hybrids: A Basis for Topological Networks

SAMPLE PREPARATION

The InAs nanowires with a triangular cross-sections were selectively grown by MBE along the [100] direction on a semi-insulating (001) InP substrate with a pre-patterned (15 nm) SiO₂ mask [1]. A thin (7 nm) layer of Al was grown *in-situ* at low temperatures on one facet by angled deposition, forming an epitaxial interface with InAs. For the fabrication of the devices, Al was selectively removed using electron-beam lithography and wet etch (Transene Al Etchant D, 50 °C, 10 s). Normal Ti/Al (5/120 nm) ohmic contacts were deposited after *in situ* Ar milling (RF ion source, 15 W, 18 mTorr, 5 min). A film of HfO₂ (7 nm) was applied via atomic layer deposition at 90 °C before depositing Ti/Au (5/100 nm) gate electrodes.

MEASUREMENT SETUP

The measurements were carried out with a lock-in amplifier at 173 Hz in a dilution refrigerator with a base temperature of 20 mK. For voltage bias measurements, an ac signal with an amplitude of 0.1 V was applied to a sample through a homebuilt resistive voltage-divider (1 : 17.700), resulting in $V_{AC} \sim 6 \mu\text{V}$ excitation. For current bias, we applied 1 V ac signal to a 1 GΩ resistor in series with a sample giving $I_{AC} = 2 \text{ nA}$ excitation.

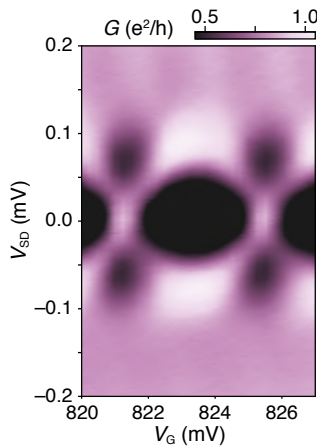


FIG. 1. Differential conductance, G , of Device 2 as a function of source-drain bias, V_{SD} , and gate voltage, V_G at zero magnetic field shows a coulomb diamond with $2E_C = 120 \mu\text{eV}$.

FREE ENERGY MODEL

The theoretical fit in Fig. 2(b) of the main text is based on a free energy model given by Eq. 1 in the main text, where the difference in free energy between odd and even occupied states is given by

$$F(T) = k_B T \ln \left[\frac{(1 + e^{-\Delta^*/k_B T})^{N_{\text{eff}}} + (1 - e^{-\Delta^*/k_B T})^{N_{\text{eff}}}}{(1 + e^{-\Delta^*/k_B T})^{N_{\text{eff}}} - (1 - e^{-\Delta^*/k_B T})^{N_{\text{eff}}}} \right],$$

with the effective number of continuum states $N_{\text{eff}} = 2V_{\text{Al}}\rho_{\text{Al}}\sqrt{2\Delta^*k_B T}$, where V_{Al} is the volume of the island and ρ_{Al} is the density of states at the Fermi energy [2]. The fit was obtained by using $V_{\text{Al}} = 2.2 \times 10^{-6} \text{ nm}^3$, consistent with Fig. 2(a) in the main text, electron density of states $\rho_{\text{Al}} = 23 \text{ eV}^{-1} \text{ nm}^{-3}$ [2] and $E_C = 60 \text{ meV}$, measured from Coulomb diamonds, with Δ^* as the single fit parameter.

DEVICE ENERGY

Energy, E , of a Coulomb blockaded device with electron occupancy, n , as a function of normalized gate voltage, N , can be defined as

$$E(N) = E_C(n - N)^2 + FN_0, \quad (\text{S1})$$

where E_C is the charging energy, F is the relative free energy and $N_0 = 0$ (1) for even (odd) parity of the device, see Fig. 2(c) in the main text, inset. Charge degeneracy points can be extracted using Eq. (S1), from which we can deduce the normalized even and odd peak spacings in units of charge, e , as

$$N_{\text{E,O}} = 1 \pm F/E_C. \quad (\text{S2})$$

The even and odd peak spacing difference in gate voltage is given by

$$S_E - S_O = \frac{E_C}{e\eta}(N_E - N_O), \quad (\text{S3})$$

with the dimensionless lever arm $\eta = E_C/eS$.

Note that in the limit of zero temperature, F is defined by the size of the induced gap, Δ^* , or, if present, by the energy of a subgap state, E_0 .

- [1] F. Krizek, J. E. Sestoft, P. Aseev, S. Marti-Sanchez, S. Vaitiekėnas, L. Casparis, S. A. Khan, Y. Liu, T. Stankevič, A. M. Whitar, A. Fursina, F. Boekhout, R. Kooops, E. Uccelli, L. P. Kouwenhoven, C. M. Marcus, J. Arbiol, P. Krogstrup, arXiv:1802.07808 (2018).

- [2] Tuominen, M. T., Hergenrother J. M., Tighe T. S., & M. Tinkham, Phys. Rev. Lett. **69**, 1997 (1992).