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Flexible to Rigid Nanoporous Polyurethane-Acrylate (PUAC) Type Materials for Structural and Thermal Insulation Applications

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(54) **FLEXIBLE TO RIGID NANOPOROUS POLYURETHANE-ACRYLATE (PUAC) TYPE MATERIALS FOR STRUCTURAL AND THERMAL INSULATION APPLICATIONS**

C08F 222/26 (2013.01); *C08J 9/28* (2013.01);
C07C 2602/42 (2017.05); *C08F 2222/1066*
(2013.01); *C08J 2205/026* (2013.01); *C08J*
2335/02 (2013.01)

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(58) **Field of Classification Search**

CPC *C07C 271/28*; *C07C 69/74*; *C07C 69/757*;
C07C 269/02; *C07C 69/753*; *C07C 67/30*;
C07C 2602/42; *C08F 122/26*; *C08F*
122/10; *C08F 222/10*; *C08F 222/26*;
C08F 2222/1066; *C08J 9/28*; *C08J*
2335/02

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See application file for complete search history.

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(*) Notice: Subject to any disclaimer, the term of this patent is extended or adjusted under 35 U.S.C. 154(b) by 0 days. days.

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Related U.S. Application Data

(63) Continuation of application No. 14/208,892, filed on Mar. 13, 2014, now Pat. No. 9,550,846.

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Brannon Sowers & Cracraft PC

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C07C 69/753 (2006.01)
C07C 269/02 (2006.01)
C08J 9/28 (2006.01)
C08F 122/26 (2006.01)
C08F 122/10 (2006.01)
C08F 222/26 (2006.01)
C08F 222/10 (2006.01)

(57) **ABSTRACT**

Novel urethane-acrylate (UAC) Star monomers and polyurethane-acrylate (PUAC) aerogel polymers derived therefrom are described herein, along with other novel, related monomers and polymers. Also described herein are processes for preparing the UAC Star monomers, the PUAC aerogel polymers, and the other related monomers and polymers. The PUAC and related polymers herein are useful in various applications including in structural and thermal insulation.

(52) **U.S. Cl.**

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69/753 (2013.01); *C07C 69/757* (2013.01);
C07C 269/02 (2013.01); *C08F 122/10*
(2013.01); *C08F 122/26* (2013.01); *C08F*
222/10 (2013.01); *C08F 222/1006* (2013.01);

16 Claims, 17 Drawing Sheets

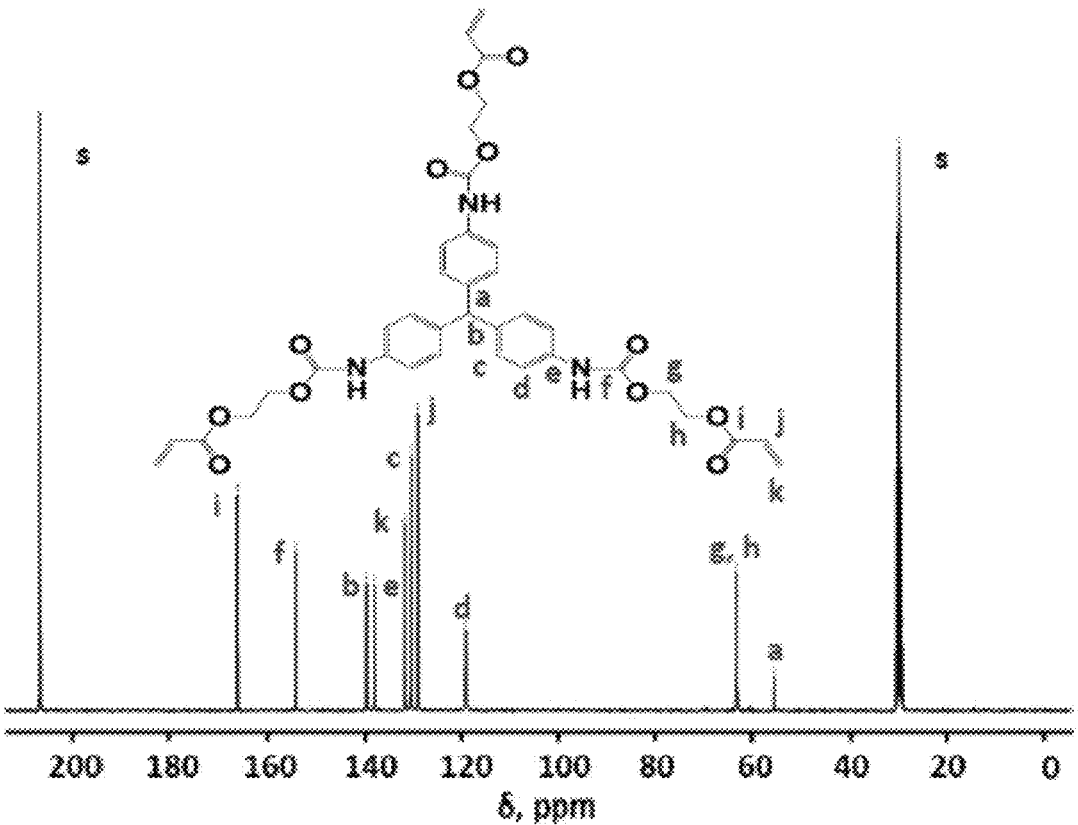


Figure 1

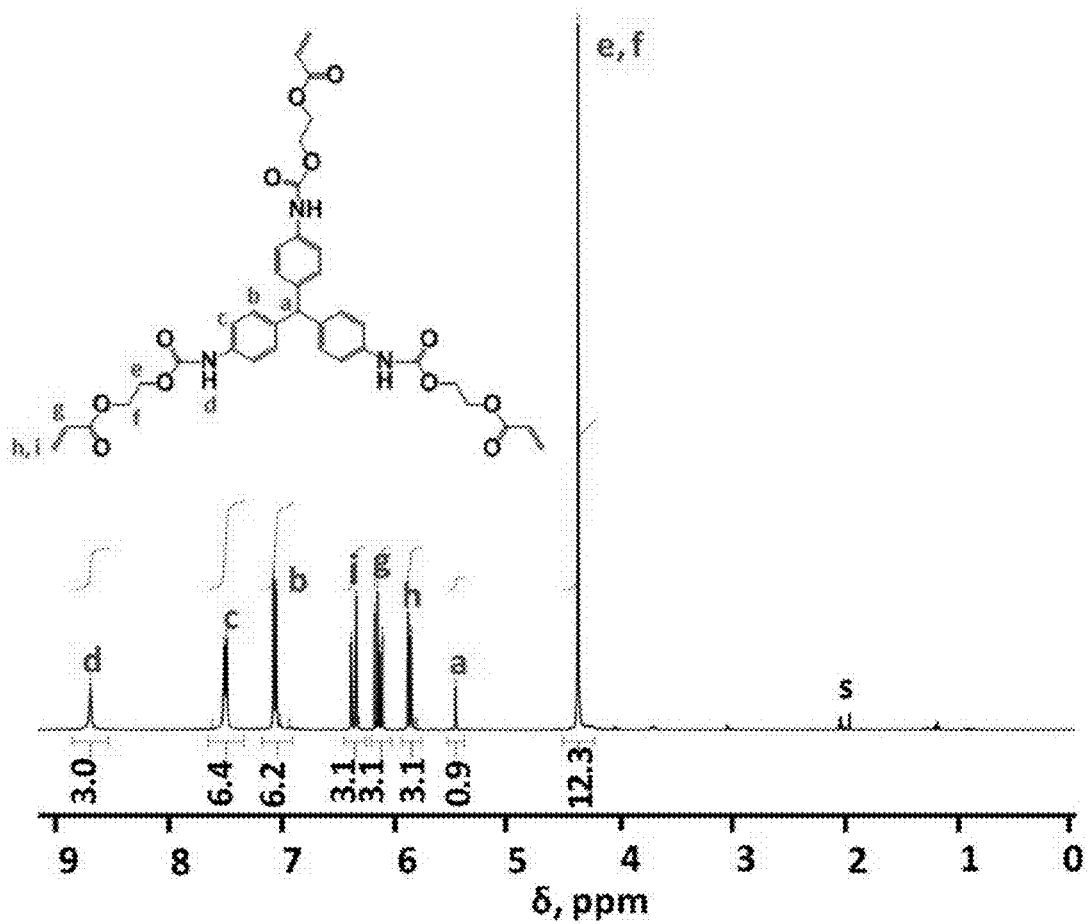


Figure 2

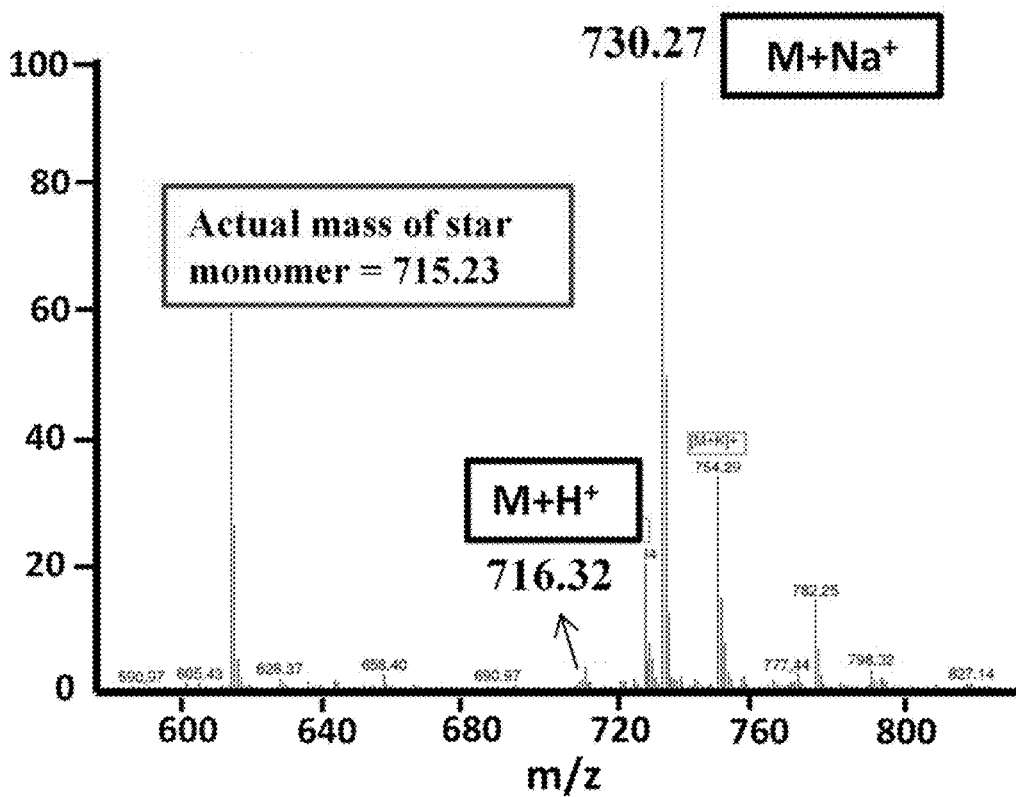


Figure 3

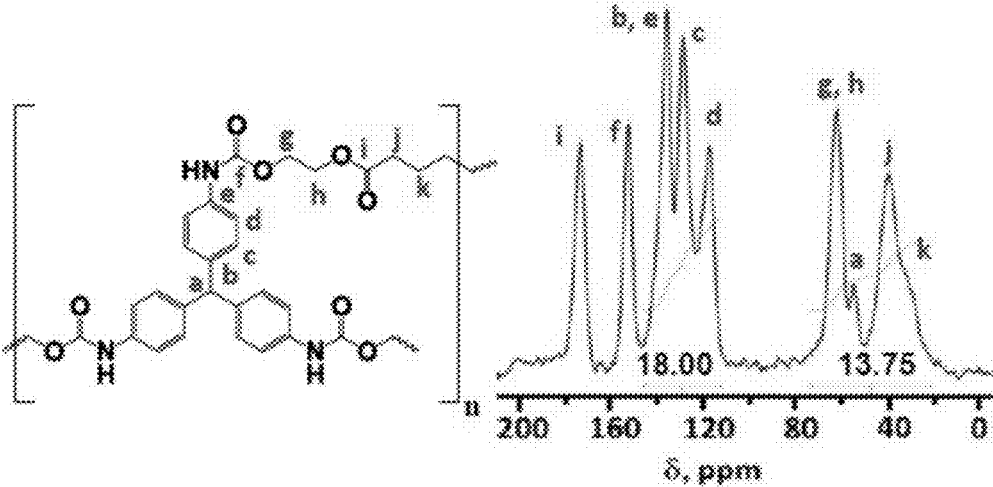


Figure 4

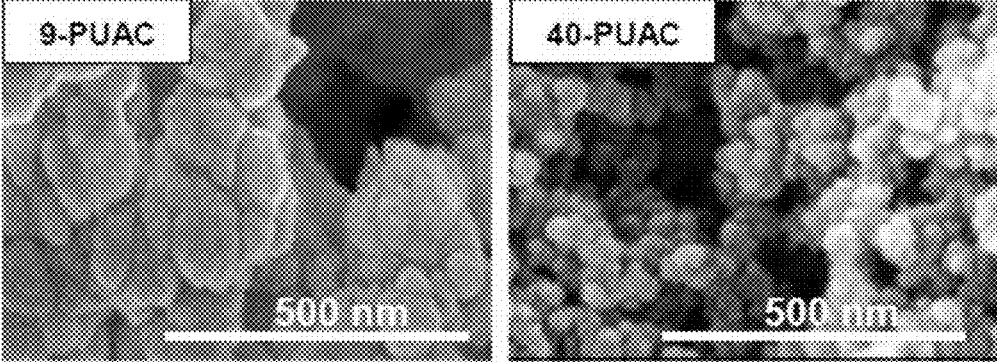


Figure 5

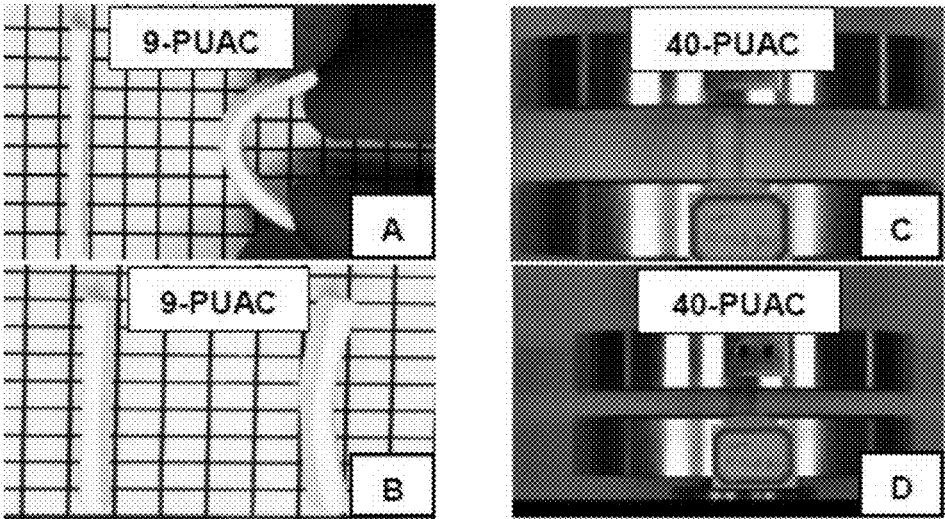


Figure 6

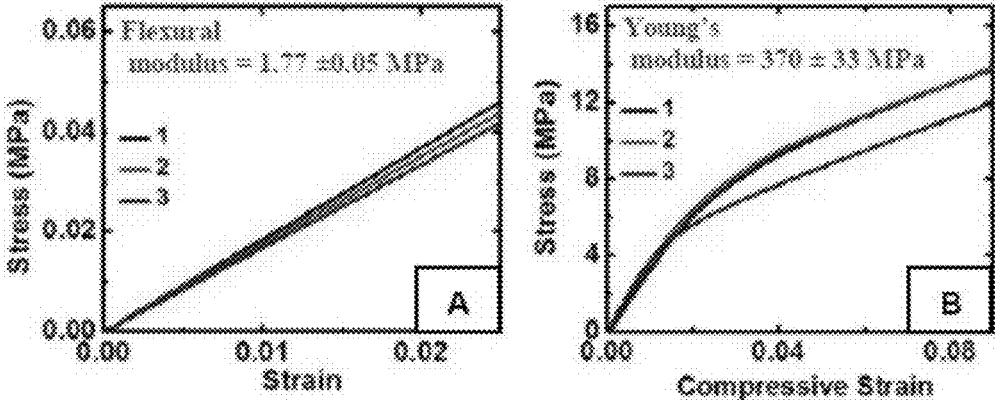


Figure 7

Characterization of Polyurethane-acrylate Aerogels

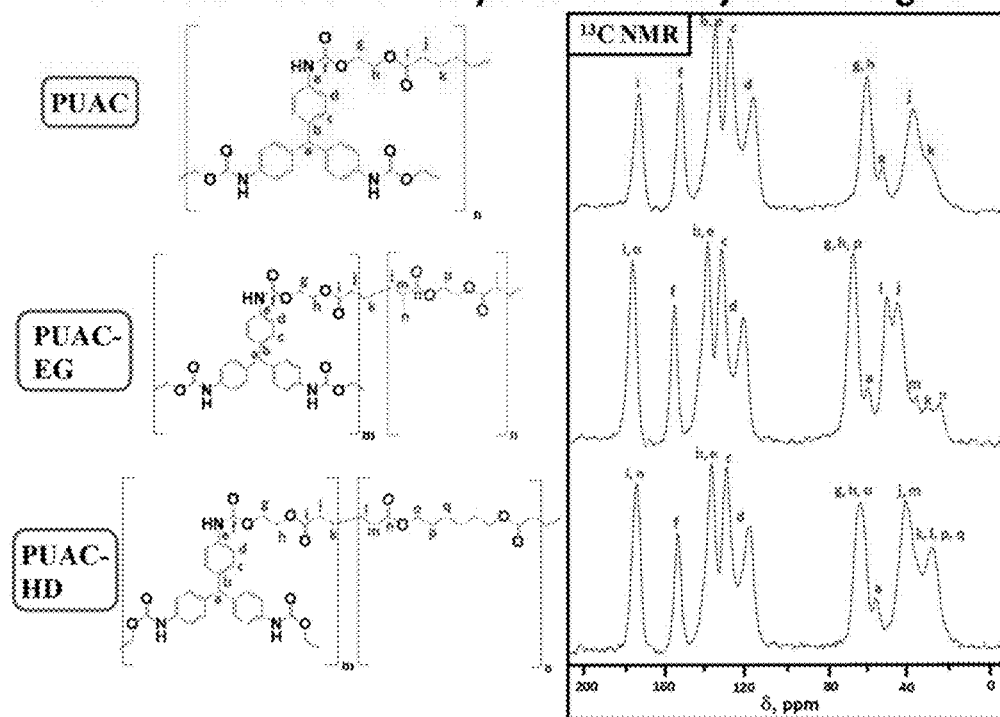


Figure 8

Characterization of Polyurethane-acrylate Aerogels

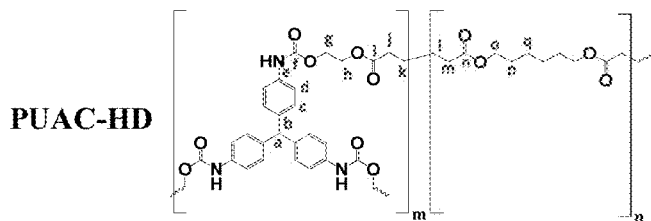
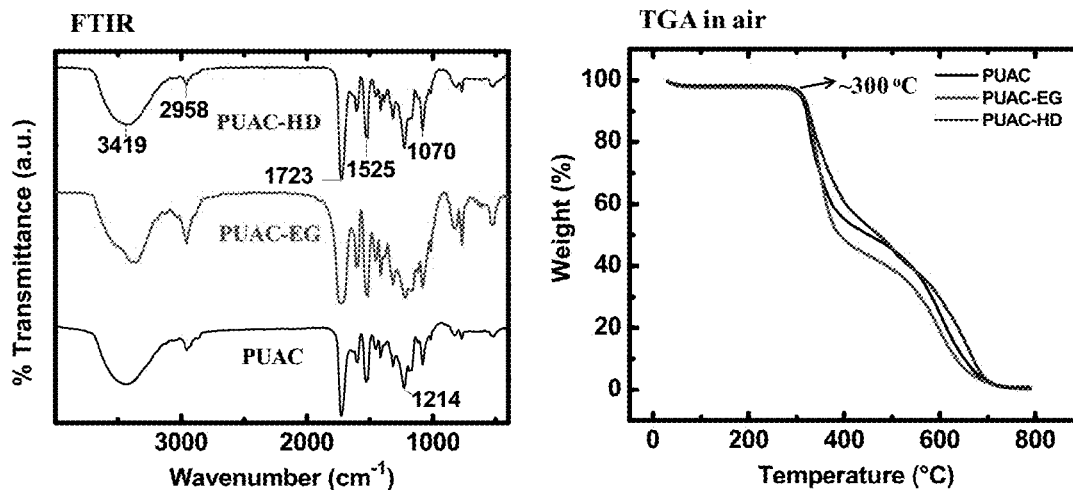
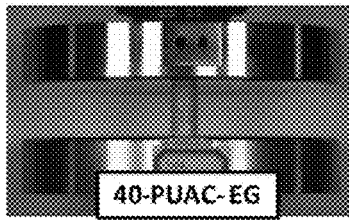
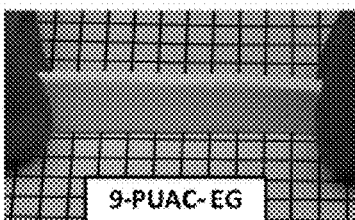
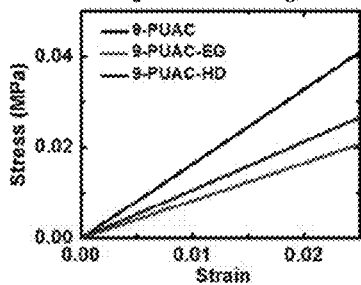


Figure 9

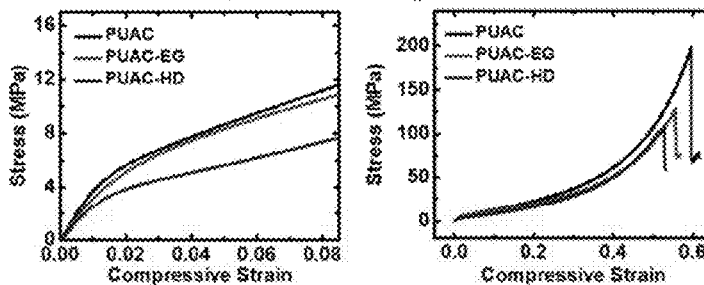
Mechanical Characterization of Polyurethane-acrylate Aerogels



3-point bending



Quasi-static compression

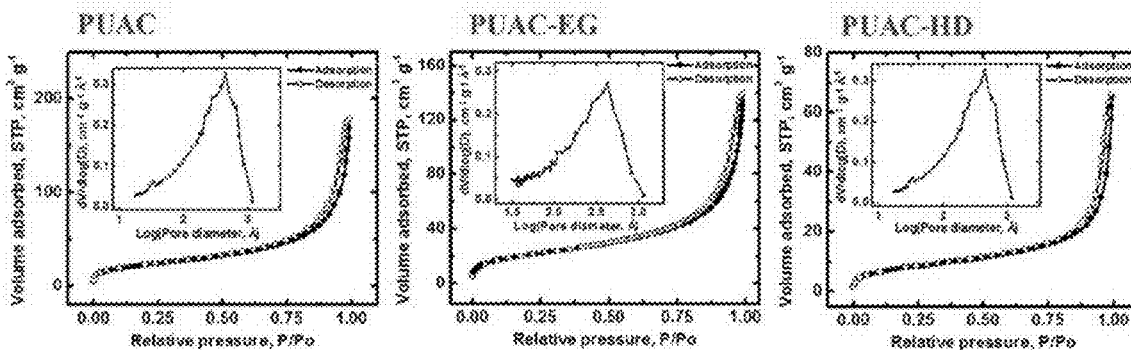


samples	bulk density ($\rho_b, \text{g cm}^{-3}$)	Flexural modulus (MPa)
9-PUAC	0.135±0.004	1.774±0.049
9-PUAC-EG	0.139±0.003	0.656±0.014
9-PUAC-HD	0.132±0.010	1.013±0.075

samples	bulk density ($\rho_b, \text{g cm}^{-3}$)	Young's Modulus (E, MPa)	specific energy abs. ($\Gamma_{abs}, \text{J g}^{-1}$)
40-PUAC	0.662±0.004	370±33	45±3
40-PUAC-EG	0.616±0.009	316±47	38±7
40-PUAC-HD	0.627±0.012	255±23	47±1

Figure 10

9% w/w Concentration of Chemical Precursors



samples	linear shrinkage (%)	bulk density, ρ_b (g cm ⁻³)	skeletal density, ρ_s (g cm ⁻³)	porosity, Π (%void space)	BET surface area, σ (m ² g ⁻¹)	particle radius, r (nm)
PUAC	18	0.135±0.004	1.331±0.007	90	81	28
PUAC-EG	17	0.139±0.003	1.340±0.006	90	74	30
PUAC-HD	15	0.132±0.010	1.321±0.009	90	29	78

Figure 11

9% w/w Concentration of Chemical Precursors

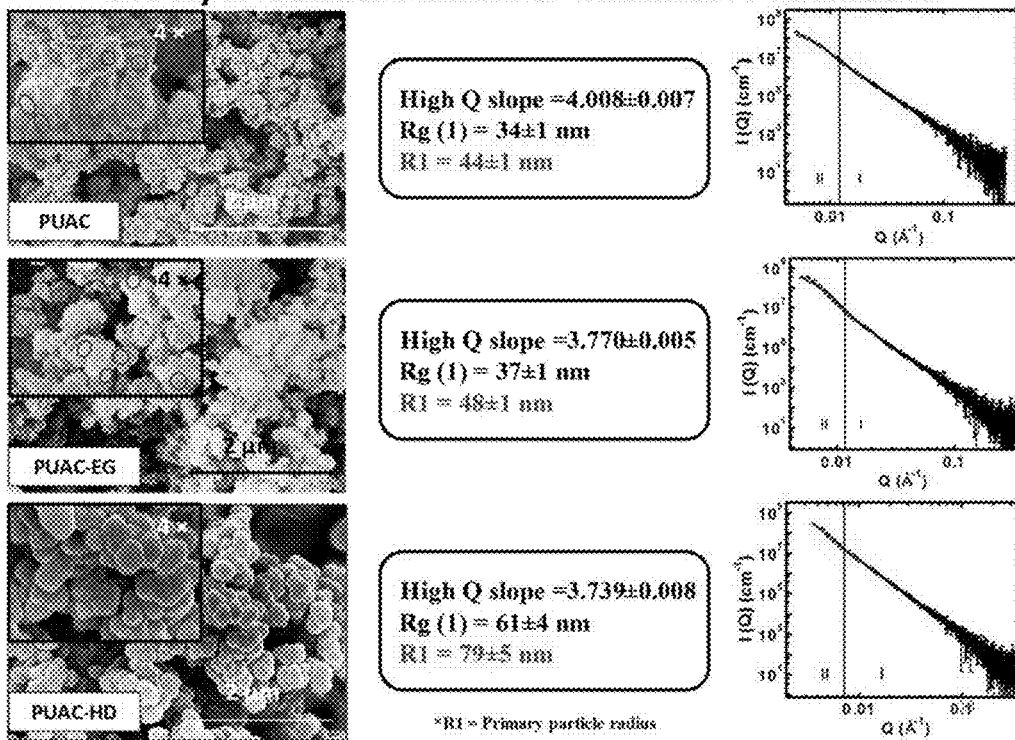
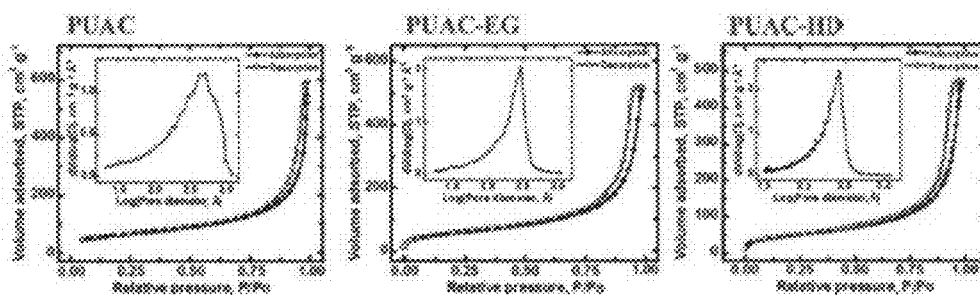


Figure 12

40% w/w Concentration of Chemical Precursors



samples	linear shrinkage (%)	bulk density, ρ_b (g cm ⁻³)	skeletal density, ρ_s (g cm ⁻³)	porosity, Π (% void space)	BET surface area, σ (m ² g ⁻¹)	particle radius, r (nm)
PUAC	9	0.662±0.004	1.314±0.004	20	217	11
PUAC-EG	8	0.616±0.009	1.309±0.003	53	208	11
PUAC-HD	8	0.627±0.012	1.308±0.004	52	175	13

Figure 13

40% w/w Concentration of Chemical Precursors

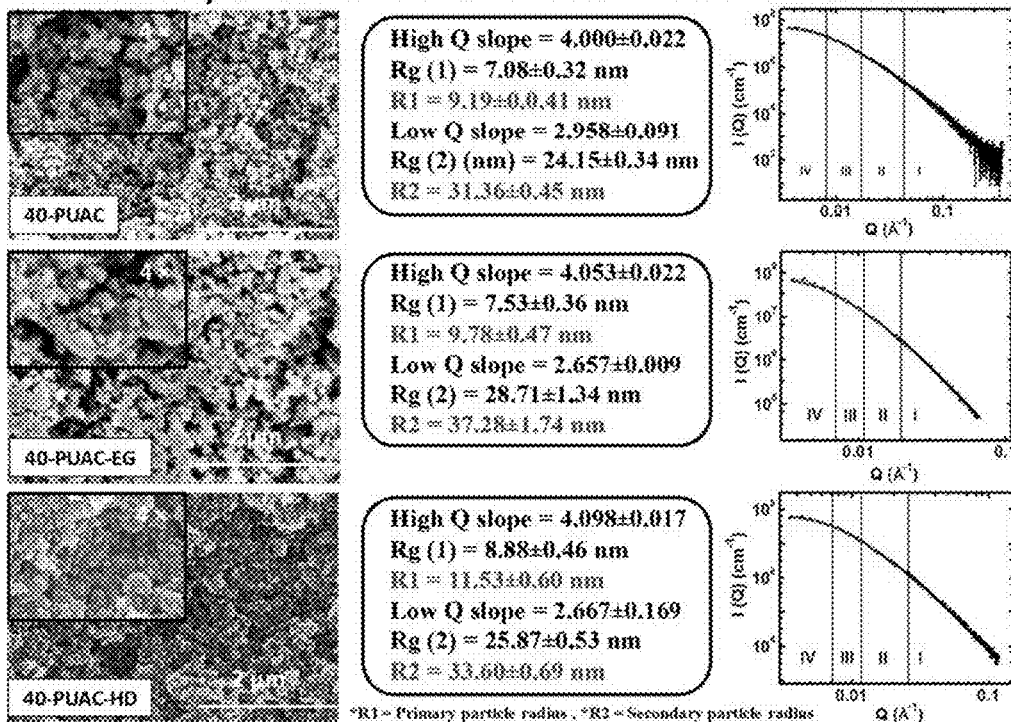
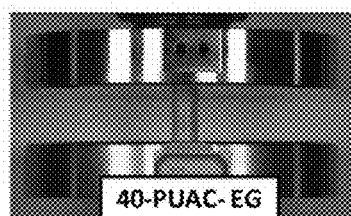
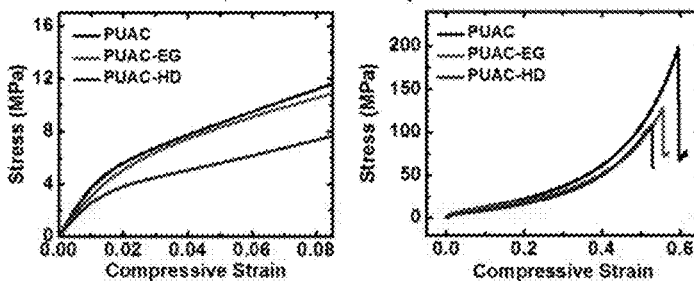


Figure 14

Mechanical Characterization of Polyurethane-acrylate Aerogels



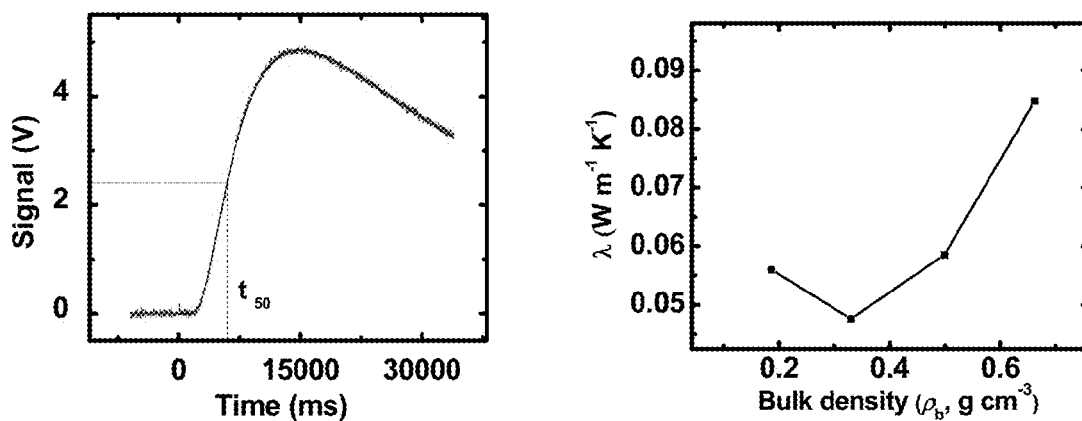
Quasi-static compression



samples	bulk density ($\rho_b, \text{g cm}^{-3}$)	Young's Modulus (E, MPa)	specific energy abs. ($\Gamma_{ab}, \text{J g}^{-1}$)
40-PUAC	0.662 \pm 0.004	370 \pm 33	45 \pm 3
40-PUAC-EG	0.616 \pm 0.009	316 \pm 47	38 \pm 7
40-PUAC-HD	0.627 \pm 0.012	285 \pm 23	47 \pm 1

Figure 15

Thermal Conductivity of PUAC Aerogels



$$\lambda = \rho_b \times c_p \times R$$

λ = thermal conductivity

ρ_b = bulk density

c_p = heat capacity (using MDSC)

R = thermal diffusivity (using Laser Flash Diffusivity)

	Thermal conductivity (W m ⁻¹ K ⁻¹)
20-PUAC	0.047
Glass wool	0.040

Figure 16

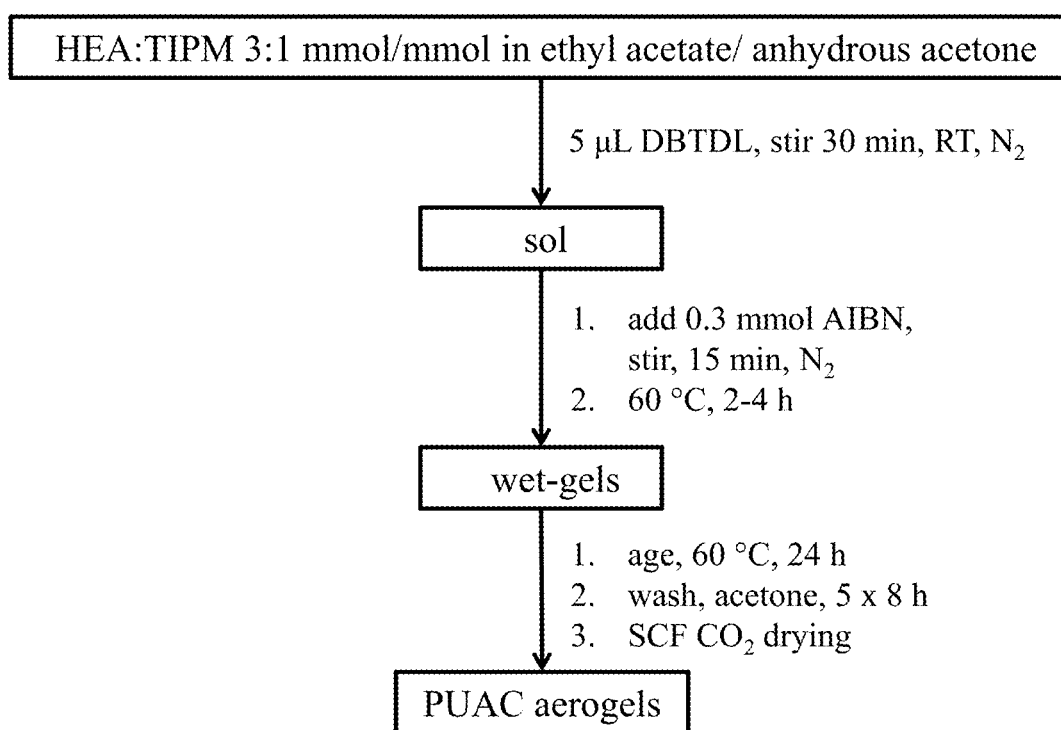


Figure 17

1

**FLEXIBLE TO RIGID NANOPOROUS
POLYURETHANE-ACRYLATE (PUAC) TYPE
MATERIALS FOR STRUCTURAL AND
THERMAL INSULATION APPLICATIONS**

CROSS-REFERENCE TO RELATED
APPLICATIONS

This application is a continuation of, and claims priority to, co-pending U.S. patent application Ser. No. 14/208,892, filed on Mar. 13, 2014, the entire disclosure of which is incorporated herein by reference.

GOVERNMENT RIGHTS

This invention was made with Government support under Grant No. CHE-0809562 awarded by the National Science Foundation and Grant No. W911NF-10-1-0476 awarded by the Army Research Office. The U.S. government has certain rights in the invention.

TECHNICAL FIELD

The present invention relates, inter alia, to structural insulation materials, thermal insulation materials, and similar materials. More specifically, the present invention relates to a series of nanoporous polyurethane-acrylate type materials and similar materials for applications in structural insulation, thermal insulation, and the like.

BACKGROUND AND SUMMARY OF THE
INVENTION

In one embodiment, the present invention discloses a series of nanoporous polyurethane-acrylate (PUAC) type materials for applications in structural insulation, thermal insulation, and the like. In another embodiment, the present invention discloses a series of urethane-acrylate (UAC) Star monomers, which are monomeric precursors for the preparation of the PUAC materials. In yet another embodiment, the present invention discloses one or more processes for the conversion of the UACs to their corresponding PUACs. In yet another embodiment, the present invention discloses methods of use of the PUACs as aerogel materials for applications in structural insulation, thermal insulation, and the like.

Aerogels have been known as three-dimensional light-weight porous materials formed by aggregation of nanoparticles, wherein the thickness dimension may vary greatly (from nanometers to meters) such that in some cases they may be viewed as quasi-two-dimensional objects. High surface areas and open porosities have been reported as being two of the most attractive properties for applications in thermal insulation, drug delivery, catalysis, and other applications. Aerogels are believed to have been first reported in 1931 by Kistler, who is believed to have introduced supercritical fluid (SCF) drying as a means to retain the network morphology of wet-gels into the final dry objects. Most post-Kistler attention appears to have focused on silica aerogels. Organic aerogels seem to have started getting attention after the synthesis of resorcinol-formaldehyde (RF) aerogels.

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For a number of years, polymer- and RF-aerogels were almost synonymous terms. In the last few years, however, research on other polymer-based aerogels seems to have gained significant momentum. It is believed that the reasons include well-defined chemistry, molecular design flexibility, readily available raw materials, and the excellent mechanical, thermal, chemical, and other properties of these polymers.

In another embodiment, described herein are polymer aerogels obtained by using polyurethane-acrylate (PUAC) chemistry. PUAC polymers have been popular in the automotive and coating industries. It is believed that this is so because PUAC polymers incorporate properties of both polyurethanes and polyacrylates, and because they can be prepared easily by free-radical polymerization using UV light or heat. It has been observed that low density PUAC aerogels tend to be flexible, while higher density PUAC aerogels tend to possess greater stiffness and stronger mechanical properties.

It has been reported that flexible aerogels are interesting for a plethora of applications, illustratively, in planetary descent and landing (EDL) systems, thermal insulation, such as in sub-sea oil pipes (e.g., see <http://www.aerogel.com/markets/subsea.html>) and buildings (e.g., acoustic insulation), where flexibility and foldability are considered to be necessary properties. It has also been reported that the impressive strength-to-weight ratio of higher-density PUAC aerogels renders them attractive as energy absorbers for defense applications.

In another embodiment of the invention herein, thermally stable PUAC aerogels are described. In one aspect, the thermal stability of these PUAC aerogels can be as high as 300° C. In another aspect, the thermally stable PUAC aerogels are synthesized from inexpensive isocyanates and hydroxyl functionalized acrylates.

In another embodiment of the invention, described herein are polyurethane and polycarbonate aerogels, which are synthesized from urethane-acrylate or carbonate-acrylate star monomers. In one aspect, these star monomers are stable and easy to handle, as compared to highly-reactive isocyanates or phosgene. In another aspect, the raw materials used herein are inexpensive and readily available.

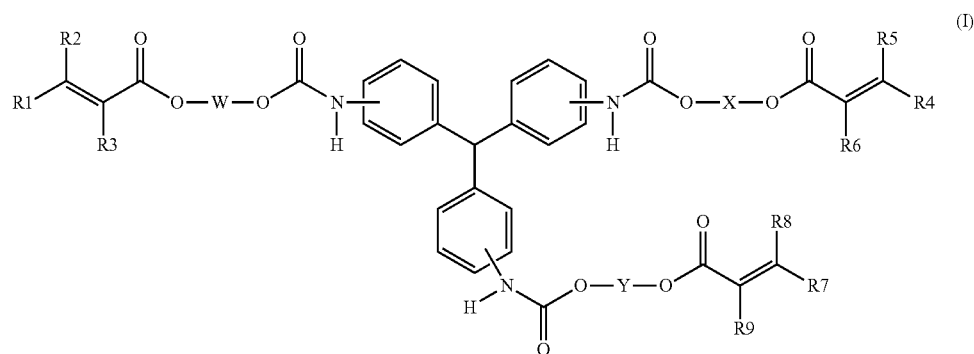
In another embodiment of the invention, described herein are inexpensive, multifunctional, light-weight, bendable nanoporous materials with high degree of flexibility. In one aspect, these materials may be prepared via easy, one-step synthesis methods, as will become apparent from the following discussion.

In another embodiment, as contemplated herein, the materials described in this invention may be useful as multifunctional thermal and acoustic insulating materials for buildings, sub-sea oil pipes, automobiles and aircraft, and as energy absorbers for anti-ballistic applications, such as, illustratively, armor and blast wave mitigation against improvised explosive devices (IEDs).

In another illustrative embodiment of the invention, the UAC Star monomers are compounds of the formula (I):

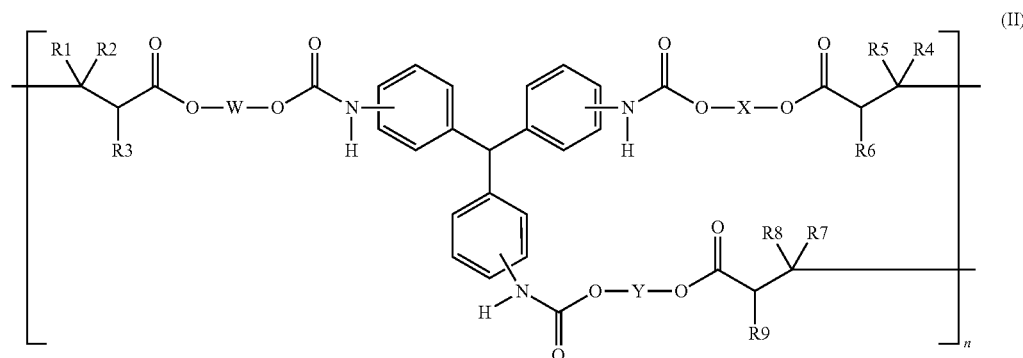
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wherein the nitrogen atoms of the urethane moieties are independently attached to their respective aryl rings at the 2, 3, or 4-positions of the aryl rings; wherein each of W, X, and Y independently represents a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group; and wherein each of R1-R9 independently represents an H, a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group. It is understood that other related variations on the illustrative UAC Star monomers of formula (I) are contemplated herein, as will become apparent in the detailed discussion below.

In another illustrative embodiment of the invention, the PUAC polymers are compounds of the formula (II):



wherein the nitrogen atoms of the urethane moieties are independently attached to their respective aryl rings at the 2, 3, or 4-positions of the aryl rings; wherein each of W, X, and Y independently represents a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group; and wherein each of R1-R9 independently represents an H, a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group; and wherein n is an integer ranging upwardly from 2 upwardly to 100, 200, 500 or even higher. It is understood that other related variations on the illustrative PUAC polymers of formula (II) are contemplated herein, as will become apparent in the detailed discussion below.

In another embodiment of the invention, processes are described herein for the conversion of the UAC Star monomers to the PUAC aerogel polymers of the invention, as will be further described in the following discussion.

BRIEF DESCRIPTION OF THE DRAWINGS

FIG. 1 shows the ¹³C NMR spectrum of the UAC Star monomer of Example 1, in CD₃COCD₃.

FIG. 2 shows the ¹H NMR spectrum of the UAC Star monomer of Example 1, in CD₃COCD₃.

FIG. 3 shows the high resolution mass spectrum of the UAC Star monomer of Example 1.

FIG. 4 shows the solid phase ¹³C NMR spectrum of a 9-PUAC aerogel sample prepared as described in Example 2.

FIG. 5 shows SEM images of 9-PUAC and 40-PUAC, respectively.

FIG. 6 shows a comparison of the mechanical behavior of 9-PUAC and 40-PUAC aerogels. 9-PUAC: (A) Flexing, (B)

after releasing the stress. 40-PUAC: (C) Before compression, (D) after compression using an Instron 4469 Universal Testing Machine.

FIG. 7 shows flexibility data for 9-PUAC and 40-PUAC aerogels.

FIG. 8 shows a comparative characterization by solid phase ¹³C NMR of PUAC, PUAC-EG, and PUAC-HD.

FIG. 9 shows a comparative characterization by FTIR and TGA of PUAC, PUAC-EG, and PUAC-HD.

FIG. 10 shows a comparison of the mechanical properties of PUAC, PUAC-EG, and PUAC-HD.

FIG. 11 shows a comparison of properties of PUAC, PUAC-EG, and PUAC-HD prepared at 9% w/w concentration of Star monomer precursors.

FIG. 12 shows a comparison of particle properties of PUAC, PUAC-EG, and PUAC-HD prepared at 9% w/w concentration of Star monomer precursors.

FIG. 13 shows a comparison of properties of PUAC, PUAC-EG, and PUAC-HD prepared at 40% w/w concentration of Star monomer precursors.

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FIG. 14 shows a comparison of particle properties of PUAC, PUAC-EG, and PUAC-HD prepared at 40% w/w concentration of Star monomer precursors.

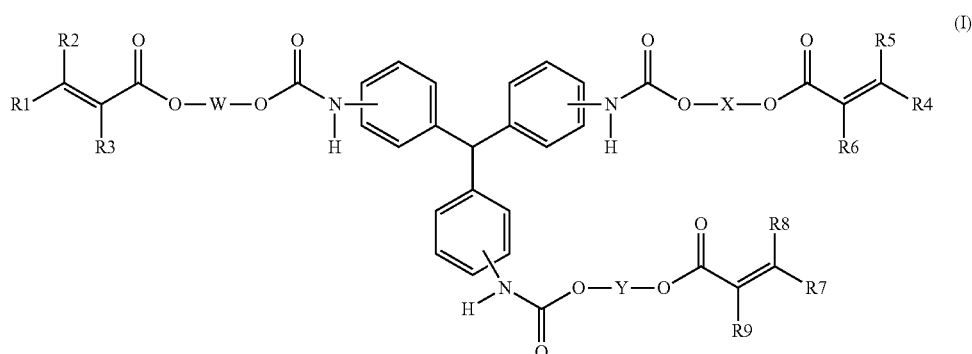
FIG. 15 shows a comparison of quasi-static compression properties of PUAC, PUAC-EG, and PUAC-HD.

FIG. 16 displays thermal conductivity properties of PUAC aerogels.

FIG. 17 describes an illustrative preparation of PUAC aerogels.

DETAILED DESCRIPTION

In one embodiment of the invention, described herein are UAC Star monomers of the formula (I):

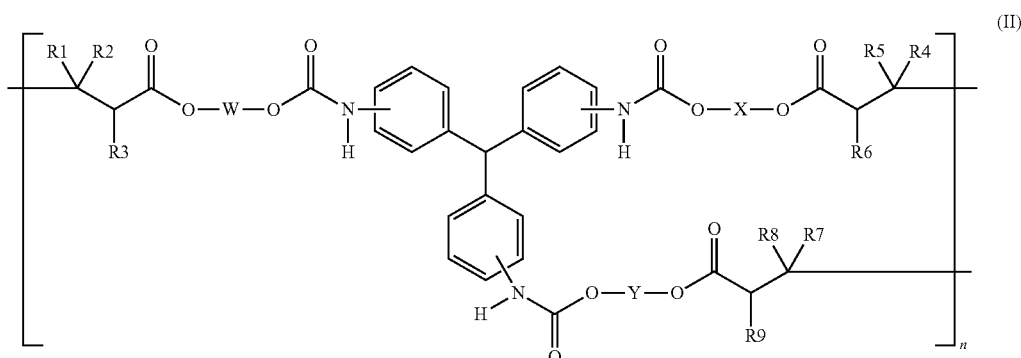


wherein the nitrogen atoms of the urethane moieties are independently attached to their respective aryl rings at the 2, 3, or 4-positions of the aryl rings; wherein each of W, X, and Y independently represents a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group; and wherein each of R1-R9 independently represents an H, a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group. It is understood that other related variations on the illustrative UAC Star monomers of formula (I) are contemplated herein, as will become apparent in the detailed discussion below. In one illustrative example, it is contemplated herein that one or more of the urethane moieties of the foregoing UAC Star monomers of formula (I) may alternatively be replaced with

carbonate groups, urea groups, and the like. The replacement of the urethane groups of the UAC Star monomers with carbonate groups or urea groups would provide carbonate-acrylate Star monomers or urea-acrylate Star monomers, respectively. In another illustrative example, it is contemplated herein that one or more of the aryl rings of the triarylmethane moiety of the UAC Star monomer of formula (I) may be substituted with any of a variety of substituent groups known in the art, illustratively, one or more C₁-C₆ straight chain alkyl group or C₁-C₆ branched alkyl group, one or more halogen atom, one or more alkoxy or alkylthio group, one or more ether or thioether group, one or more carboxy group, one or more ester group, one or more carboxamido group, one or more cyano group, one or more

nitro group, and the like; or that one or more of the aryl rings of the triarylmethane moiety of the UAC Star monomer may be replaced with alternative aromatic or heteroaromatic rings other than benzene rings, including single-ring or fused-ring aromatics or heteroaromatics, wherein the alternative aromatic or heteroaromatic rings may be substituted with one or more of the foregoing substituents.

In another illustrative embodiment of the invention, described herein are PUAC polymers of the formula (II):



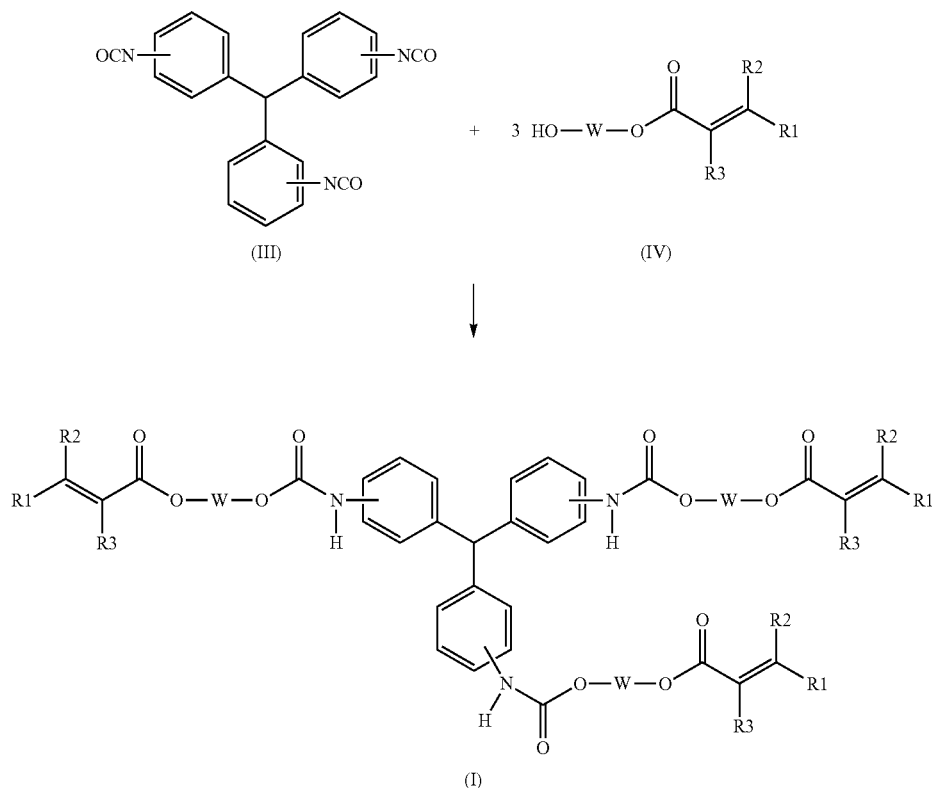
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wherein the nitrogen atoms of the urethane moieties are independently attached to their respective aryl rings at the 2, 3, or 4-positions of the aryl rings; wherein each of W, X, and Y independently represents a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group; and wherein each of R1-R9 independently represents an H, a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group; and wherein n is an integer ranging upwardly from 2.

It is understood that other related variations on the illustrative PUAC polymers of formula (II) are contemplated herein, as will become apparent in the detailed discussion

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carrying out the process step include ketone solvents such as acetone, ester solvents such as ethyl acetate, and the like, or any other suitable solvents known to those skilled in the relevant art. In the case that a catalyst is used for the process step, suitable catalysts include various organometallic or organic catalysts known to those skilled in the relevant art, such as, illustratively, dibutyltin dilaurate (DBTDL). Following is an illustrative scheme of said process for preparing a UAC Star monomer of formula (I), wherein W=X=Y, R1=R4=R7, R2=R5=R8, and R3=R6=R9:



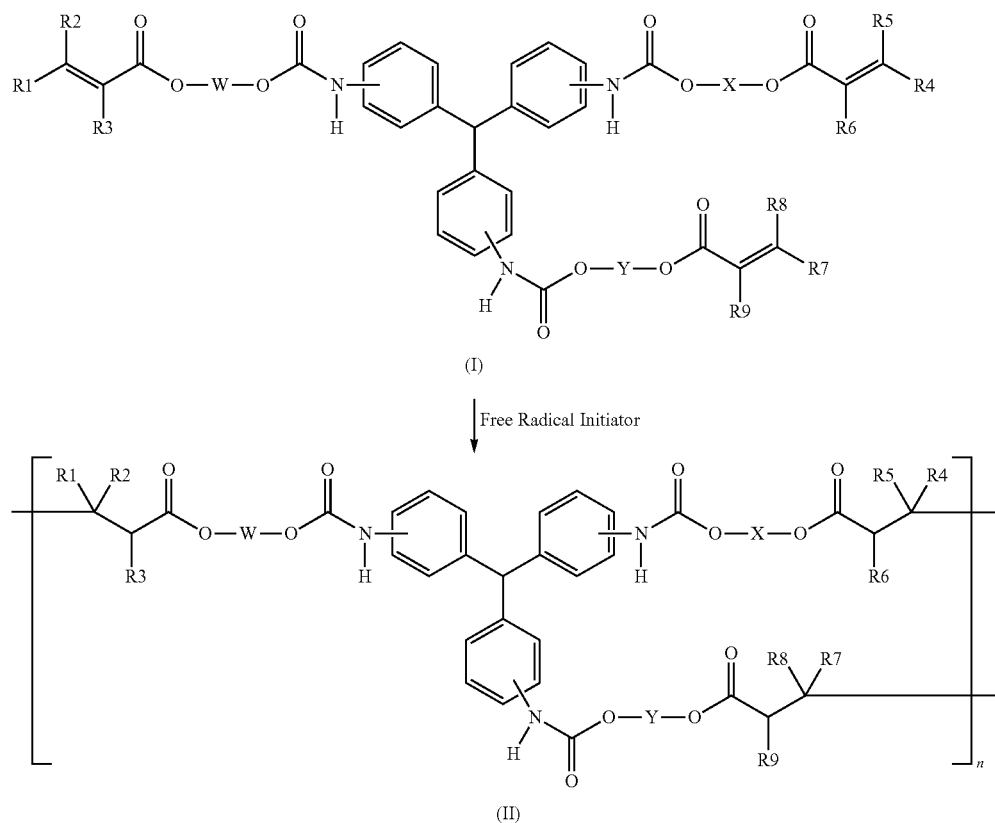
below. For example, one or more of the aryl rings of the triarylmethane moiety of the PUAC polymer of formula (II) may be substituted with any of a variety of substituent groups known in the art, similar to the substituent groups described in the previous paragraph in regard to the UAC Star monomers; or one or more of the aryl rings of the triarylmethane moiety of the PUAC polymer of formula (II) may be replaced with alternative aromatic or heteroaromatic rings other than benzene rings, including single-ring or fused-ring aromatics or heteroaromatics, wherein the alternative aromatic or heteroaromatic rings may be substituted with one or more of the foregoing substituents.

In another embodiment of the invention, a process for the preparation of the UAC Star monomers of formula (I) is described herein. In one illustrative example, the process comprises the step of reacting a tris(isocyanatoaryl)methane of formula (III) with one or more hydroxyalkyl acrylate of formula (IV), in a suitable solvent or mixture of solvents, following procedures known in the art, either with or without a catalyst, wherein the hydroxyl groups react with the isocyanato groups, to produce a UAC Star monomer of formula (I). Suitable solvents or mixtures of solvents for

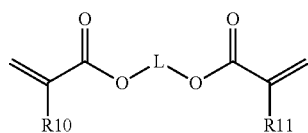
In another embodiment of the invention, a process for the preparation of the PUAC polymers of formula (II) is described herein. In one illustrative example, the process comprises the step of treating a solution of a UAC Star monomer of formula (I), in a suitable solvent or mixture of solvents, with a suitable polymerization catalyst, such as, illustratively, a free radical initiator. Suitable solvents or mixtures of solvents for carrying out the process step include ketone solvents such as acetone, ester solvents such as ethyl acetate, and the like, or any other suitable solvents known to those skilled in the relevant art. Suitable free radical initiators include any of a wide variety of agents known in the relevant art, such as, illustratively, the commonly used free radical initiator 2,2'-azobisisobutyronitrile (AIBN). In the following illustrative scheme of said process for the preparation of the PUAC polymers of formula (II), wherein W=X=Y, R1=R4=R7, R2=R5=R8, and R3=R6=R9, a solution of the UAC Star monomer (I) in a suitable solvent or mixture of solvents is treated with a catalytic amount of a free radical initiator, resulting in the formation of (II):

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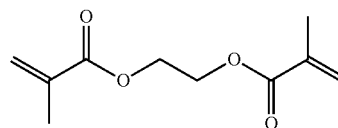


In another embodiment of the invention, a variation on the above process for generation of the PUAC polymers is described herein, which entails addition into the reaction mixture, prior to introduction of the polymerization catalyst (e.g., the free radical initiator), of one or more compounds termed herein as chain extenders. These chain extenders are monomeric compounds capable of copolymerizing with the UAC Star monomers. Illustrative of these chain extenders are diacrylate compounds of the following general formula (V):

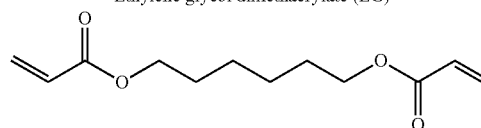


wherein L represents a linker group selected from C₁-C₁₂ straight chain or branched chain alkyl, alkoxyalkyl, alkoxy-carbonylalkyl, and alkoxy-carboxyalkyl; and wherein each of R₁₀ and R₁₁ independently represents an H, a C₁-C₆ straight chain alkyl group or a C₁-C₆ branched alkyl group. In one aspect of the invention, the inclusion of these chain extenders in the polymerization step results in PUACs possessing favorable flexibility, thermal insulation and other properties, as will become apparent in the discussion below. Representative of the chain extenders of the invention are compounds of the following formulae; it is to be understood that these formulae are shown only for the purpose of illustration, that they are not to be construed as limiting the

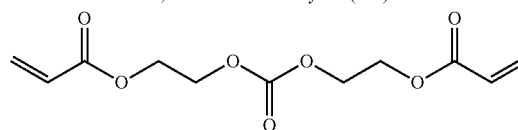
invention, and that other similar chain extenders well known to those skilled in the relevant art are contemplated herein:



Ethylene glycol dimethacrylate (EG)



1,6-Hexanediol diacrylate (HD)



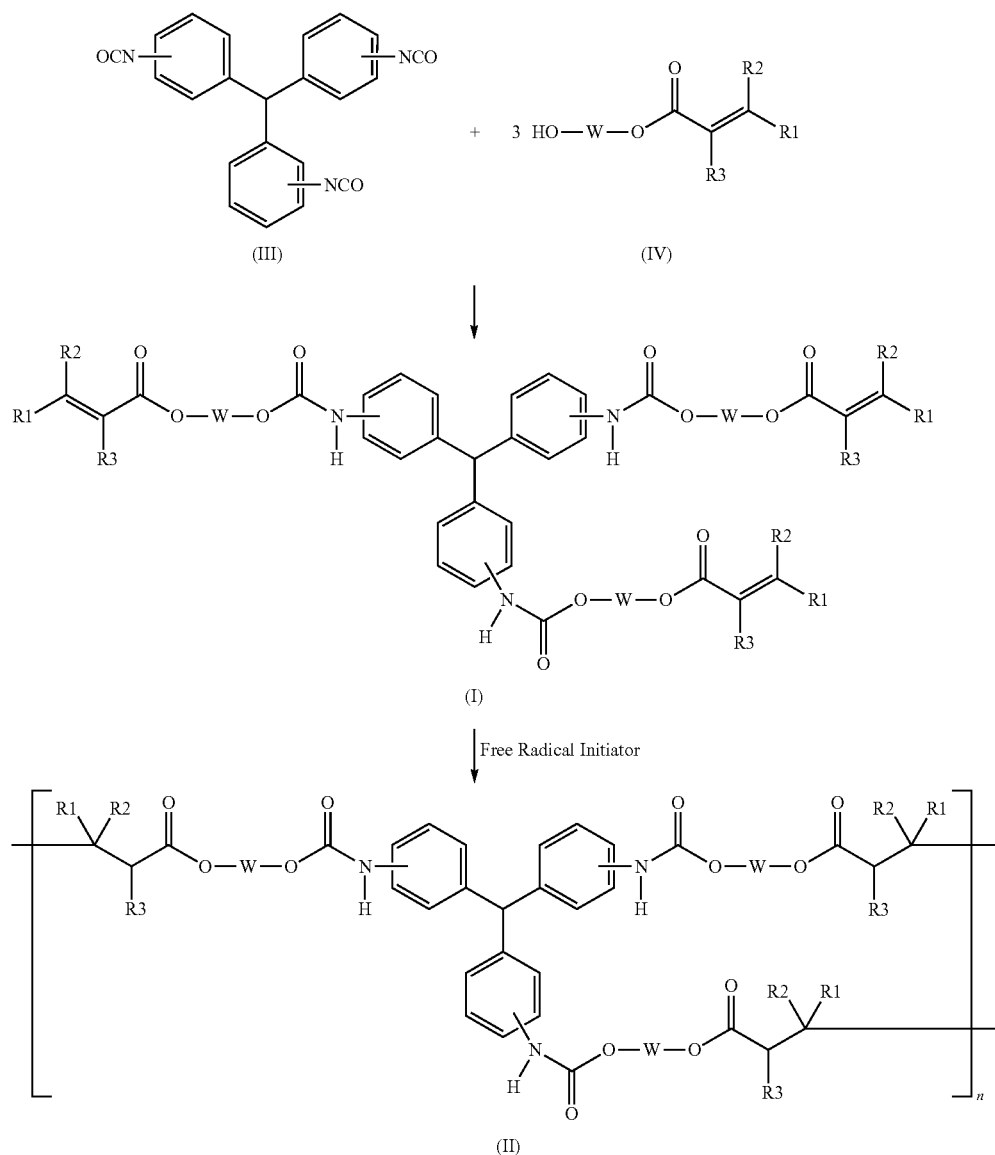
Bisacryloylethyl carbonate

In another embodiment of the invention, PUAC polymers are described herein that are obtained by the above polymerization process that includes the addition of one or more of the chain extenders to the UAC Star monomers, as described above. These PUAC polymers are similar in chemical structure to the chemical structure of the PUACs of formula (II), but include copolymerized chain extenders.

In another embodiment of the invention, a one-pot process for the preparation of the PUAC polymers of formula

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(II) starting from a tris(isocyanatoaryl)methane of formula (III) and one or more hydroxyalkyl acrylate of formula (IV) is described herein. In one illustrative example, the process comprises the step of reacting a tris(isocyanatoaryl)methane of formula (III) with one or more hydroxyalkyl acrylate of formula (IV), in a suitable solvent or mixture of solvents as described above, following procedures known in the art, either with or without a catalyst, to produce a solution of the corresponding UAC Star monomer of formula (I). This is followed directly, without isolation of (I), by treatment with a suitable polymerization catalyst, such as, illustratively, a suitable free radical initiator as described above, to produce (II). Following is an illustrative scheme of said one-pot process for the preparation of PUAC (II), wherein $W=X=Y$, $R_1=R_4=R_7$, $R_2=R_5=R_8$, and $R_3=R_6=R_9$:



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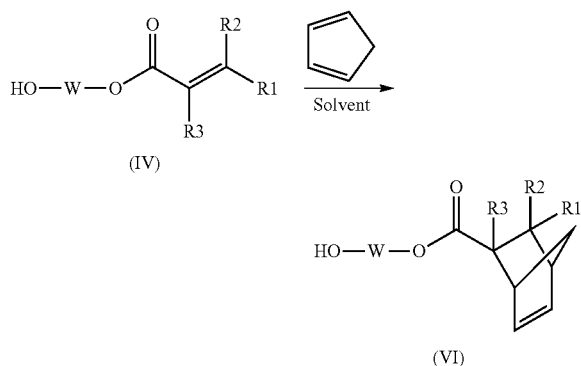
includes the addition into the reaction mixture, prior to introduction of the polymerization catalyst (e.g., the free radical initiator), of one or more of the chain extenders described above, thus resulting in PUACs that have copolymerized chain extender units incorporated therein.

In another embodiment of the invention, described herein are compounds that are derived by the electrocyclic (e.g., Diels-Alder) reaction in a suitable solvent of the alkene moiety of the hydroxyalkyl acrylate of formula (IV) with a diene, to produce the corresponding Diels-Alder adduct; wherein R₁, R₂, R₃, and W are as defined in the foregoing. Suitable solvents for this reaction are the common solvents used in the art to carry out Diels-Alder reactions, such as

In another embodiment of the invention, an alternative one-pot process for the preparation of PUAC polymers is described herein. This alternative one-pot process is similar in all respects to the one-pot process described above for the preparation of the PUAC polymers of formula (II), but

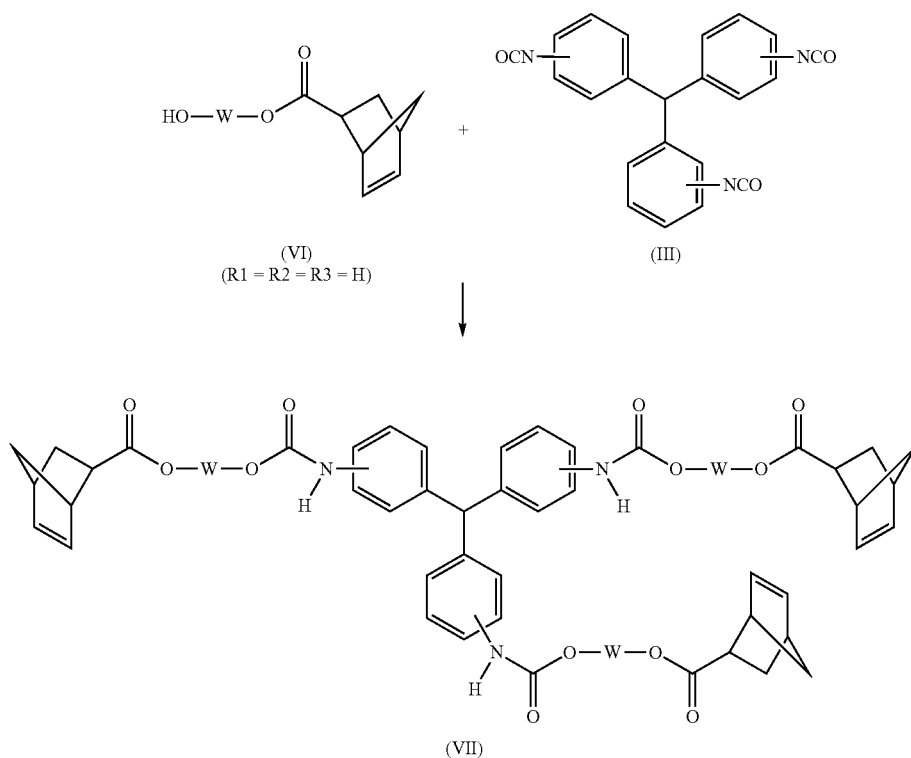
toluene, the xylenes, and similar solvents. Illustrative of these compounds are those represented by formula (VI), which are obtained by reaction of the hydroxyalkyl acrylate of formula (IV) with cyclopentadiene to produce the corresponding norbornene, as shown in the following scheme:

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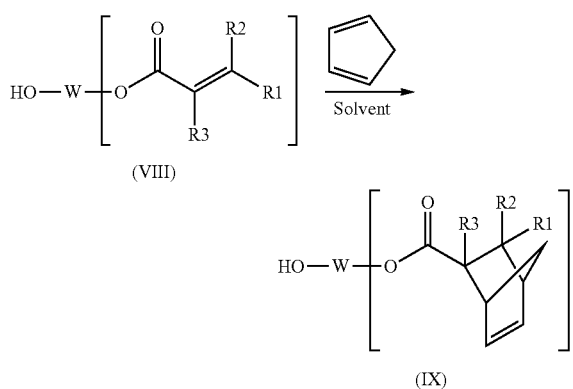
A specific example of the foregoing is the reaction of 2-hydroxyethyl acrylate (i.e., compound (IV) wherein $W=CH_2CH_2$; $R_1=R_2=R_3=H$) with cyclopentadiene to produce norbornene adduct (VI) wherein $W=CH_2CH_2$; $R_1=R_2=R_3=H$. It is to be understood that, as contemplated herein, other dienes may be used, including any of a plethora of non-cyclic and cyclic dienes known in the art.

In a further embodiment of the invention, Star monomers incorporating norbornene groups are described herein. These norbornene containing Star monomers are obtained by the reaction of the norbornene compounds (VI) with a tris(isocyanato) methane of formula (III) under conditions that are similar to those described above for the reaction of (III) with (IV). Illustrative of those Star monomers that incorporate norbornene groups are those represented by the formula (VII), which are obtained by the reaction of the norbornene compounds (VI), wherein $R_1=R_2=R_3=H$, with a tris(isocyanato) methane of formula (III), as shown in the following scheme:



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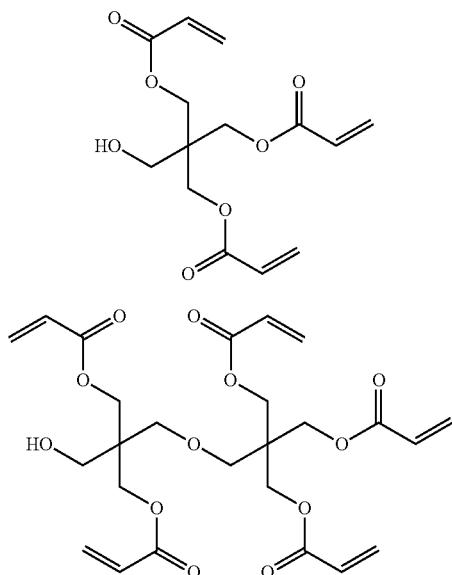
In another embodiment of the invention, described herein are compounds that are derived by the electrocyclic (e.g., Diels-Alder) reaction in a suitable solvent of the alkene moieties of hydroxyalkyl polyacrylate compounds of formula (VIII) with a diene, to produce the corresponding Diels-Alder adduct. Suitable solvents for this reaction are the common solvents used in the art to carry out Diels-Alder reactions, as described above. Illustrative of these compounds are those represented by formula (IX), which are obtained by reaction of the hydroxyalkyl polyacrylate compounds of formula (VIII) with cyclopentadiene to produce the corresponding adducts that contain polynorbornene moieties, as shown in the following scheme, wherein $R_1, R_2, R_3,$ and W are as defined in the foregoing, and m is an integer from 2 to 12:



It is to be understood that, as contemplated herein, various other hydroxyalkyl polyacrylate compounds may be used in

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the Diels-Alder reaction with the various dienes (e.g., cyclopentadiene). Illustrative of such other hydroxyalkyl polyacrylate compounds are the following examples represented by formulae (X) and (XI) below:

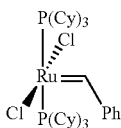


Specifically, in the case of the reaction of (X) with cyclopentadiene, a Diels-Alder adduct is obtained that contains three norbornene moieties; likewise, in the case of the reaction of (XI) with cyclopentadiene, a Diels-Alder adduct is obtained that contains five norbornene moieties.

In another embodiment of the invention, Star monomers containing multiple norbornene groups are described herein. These Star monomers may be obtained by the reaction of the hydroxyalkyl polynorbornene compounds represented by the formula (IX) with a tris(isocyanatoaryl)methane of formula (III), under conditions similar to those described earlier. Illustratively, the polynorbornene-containing hydroxyalkyl compounds obtained from the reactions of (X) and (XI), respectively, with cyclopentadiene are further reacted with tris(isocyanatoaryl)methane of formula (III) to produce the corresponding Star monomers.

In another embodiment of the invention, polynorbornene-polyurethane (PNUP) aerogel polymers are described herein. These PNUP aerogel polymers are obtained via a process wherein the norbornene-containing Star monomers represented by formula (VII), or the norbornene-containing Star monomers described in the previous paragraph, are polymerized under ring-opening metathesis polymerization conditions. Any of a variety of suitable conditions for ring-opening metathesis polymerization that are well-known in the art may be used. Illustrative of these conditions are as follows:

First and second generation Grubbs catalysts (GC-I and GC-II) may be used,

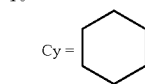
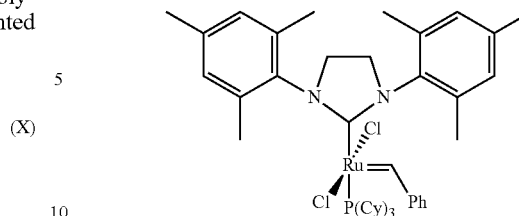


GC-I

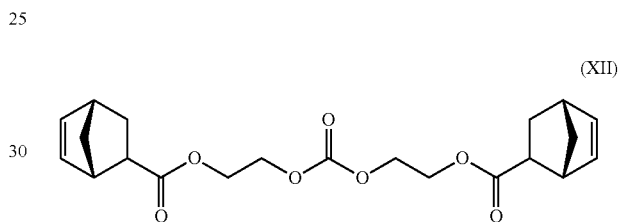
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-continued

GC-II



which offer the advantages of tolerance to moisture, oxygen, protic solvents and functional groups. GC-I has been used in toluene, GC-II has been used in toluene, acetone, tetrahydrofuran, THF and methanol. Further and alternatively, the ring-opening metathesis polymerization process may be carried out in the presence of chain extenders that are known to persons skilled in the relevant art. The following bisnorborneneacryloxyethyl carbonate compound (XII) is illustrative of such chain extenders.



In another embodiment, the aerogels of the invention herein are porous, low-density 3D assemblies of nanoparticles with large surface-to-volume ratios. In one aspect, the aerogels possess flexibility that makes them particularly attractive materials for thermal insulation. The flexible aerogels herein were characterized at the molecular level (solid phase ^{13}C NMR), nanoscopic level (SEM, SAXS), and macroscopic level (compression and 3-point bending). The lower density PUAC aerogels (0.14 g cm^{-3}) consist of large primary particles (88 nm in diameter) and are macroporous and flexible. The higher density PUAC aerogels (0.66 g cm^{-3}) consist of smaller particles (18 nm in diameter), and are rigid and mechanically strong.

While the novel technology herein has been illustrated and described in detail in the foregoing description, and in the following examples and figures, the same is to be considered as illustrative and not restrictive in character. It is understood that one of ordinary skill in the art could readily make a nigh-infinite number of insubstantial changes and modifications to the above-described embodiments and that it would be impractical to attempt to describe all such embodiment variations in the present specification. Accordingly, it is understood that all changes and modifications that come within the spirit of the novel technology are desired to be protected.

Examples

The following examples further illustrate specific embodiments of the invention. However, the following examples should not be interpreted in any way to limit the invention.

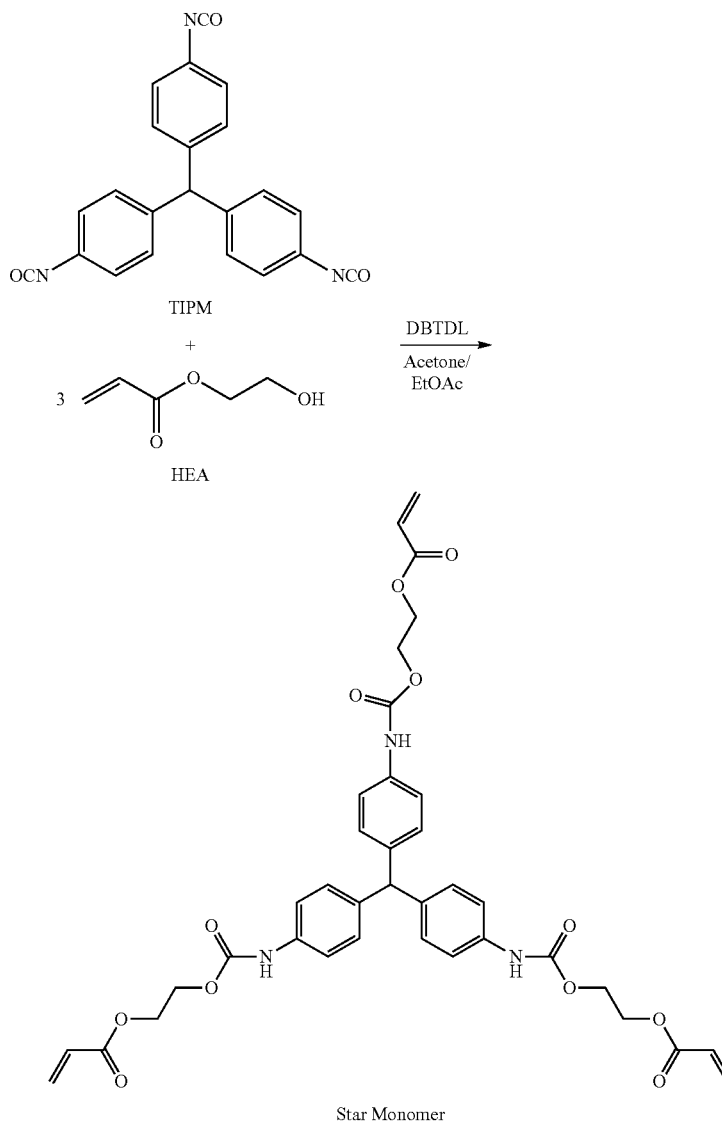
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MATERIALS. All reagents and solvents were used as received unless noted otherwise. 2-Hydroxyethyl acrylate (HEA), 2,2'-azobisisobutyronitrile (AIBN) and dibutyltin dilaurate (DBTDL) were purchased from Sigma-Aldrich. Anhydrous acetone was purchased from Fisher Scientific. Siphon-grade CO₂ was purchased from Ozark Gas Co. Tris(4-isocyanatophenyl)-methane (TIPM) (27% w/w solu-

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varied depending upon the desirable weight percent of solids in the solution. The UAC Star monomer can be isolated, recrystallized from CH₂Cl₂/hexane, and fully characterized. However, for routine preparation of aerogels, gelation was induced in one pot by adding a radical initiator, as described in the following example.

Scheme 1



tion in ethyl acetate) was obtained from Bayer Corporation USA (TIPM is referred to by Bayer Corporation as Desmodur RE).

Example 1

Illustrative preparation of urethane-acrylate (UAC) Star monomers. A urethane-acrylate Star monomer was synthesized via reaction of TIPM (1 mmol) with HEA (3 mmol) using DBTDL (5 μ L) as a catalyst in anhydrous acetone (see Scheme 1). The reaction mixture was stirred at room temperature for 30 min. The quantity of anhydrous acetone was

55 The UAC Star monomer was characterized by ¹³C NMR, the spectrum of which is shown in FIG. 1. Peak f (154 ppm) corresponds to the carbonyl group of the urethane linkage. 60 Peaks j and k correspond to the acrylate moieties, which were used for free radical polymerization. The UAC Star monomer was also characterized by ¹H NMR; the spectrum is shown in FIG. 2. The UAC Star monomer was also characterized by high resolution mass spectroscopy; the data is shown in FIG. 3. 65

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Example 2

Illustrative preparation of PUAC aerogels. A UAC Star monomer was prepared as described in the previous example. Gelation was induced in one pot by adding AIBN (0.3 mmol) into the reaction mixture containing the UAC Star monomer, followed by transferring into molds and heating at 60° C. for 2-4 h (see FIG. 17). Wet-gels were aged in their molds for 24 h at 60° C., then transferred in acetone, washed 5 times, 8 h per wash, and finally were dried using SCF CO₂ to produce the PUAC aerogels. Polypropylene vials (4 mL, Wheaton polypropylene Omnivials, Part No. 225402, 1 cm in diameter), or polypropylene centrifuge tubes (50 mL, Fisher Scientific, Cat. No. 06-443-18, 2.8 cm in diameter) were used as molds. The PUAC aerogels are referred to herein as 'xx-PUAC', where 'xx' denotes percent weight of monomers in the solution.

Solid phase ¹³C NMR was used to confirm formation of the PUAC polymer and to determine whether all the double bonds of the Star monomer had reacted by free radical polymerization. FIG. 4 shows the solid phase ¹³C NMR of a 9-PUAC prepared as described above. The relative ratio of aromatic carbons (110-145 ppm) to aliphatic carbons (20-75 ppm) was found to be close to the theoretical value (1.31:1.00 vs 1.38:1.00, respectively), thus confirming complete reaction.

Example 3

Comparison of PUAC aerogels from the lowest Star monomer concentration (9-PUAC) with PUAC aerogels from the highest Star monomer concentration (40-PUAC). The PUACs were prepared as described in the previous example. It was found that 9-PUACs were flexible, while 40-PUACs were rigid. The SEM images of 9-PUAC and 40-PUAC are displayed in FIG. 5, which show the following data: SEM of 9-PUAC: $\rho_b=0.135\pm 0.004$ g cm⁻³, porosity=90% v/v; SEM of 40-PUAC: $\rho_b=0.662\pm 0.004$ g cm⁻³, porosity=50% v/v. At lower monomer concentrations, the polymer chains are expected to grow longer resulting late phase separation and formation of larger primary particles (88 nm in diameter by SAXS for 9-PUAC). At higher monomer concentrations, early phase separation leads to formation of smaller primary particles (18 nm in diameter by SAXS for 40-PUAC). Aggregation of primary particles in case of 40-PUAC yields clusters of secondary and tertiary particles (by SAXS). Therefore, 9-PUAC aerogels have lower inter-particle connectivity than 40-PUAC.

Example 4

Comparison of the Mechanical Behavior of 9-PUAC and 40-PUAC aerogels. A comparison of the mechanical behavior of 9-PUAC and 40-PUAC aerogels is shown in FIG. 6: 9-PUAC: (A) Flexing, (B) after releasing the stress. 40-PUAC: (C) Before compression, (D) after compression using an Instron 4469 Universal Testing Machine. It was found that 9-PUAC is flexible and when stress is released, it practically recovers its original shape.

Example 5

Flexibility of PUAC Aerogels. The flexibility of 9-PUAC was formally characterized with 3-point bending using a Dynamic Mechanical Analyzer (FIG. 7). (A) Stress-strain curves of 9-PUAC aerogels via a 3-point bending test (3 samples—only the early part of the curve is shown). (B)

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Stress-strain curves of 40-PUAC aerogels via compression testing (3 samples—only the early part of the curve is shown). With a higher inter-particle connectivity, 40-PUAC aerogels are rigid and mechanically strong (see FIG. 6D). The ultimate compressive strength, the compressive Young's modulus and the specific energy absorption were found to be equal to 175 MPa, 370 MPa and 45 J g⁻¹, respectively (see FIG. 7B).

Example 6

Illustrative preparation of PUAC aerogels with inclusion of ethyleneglycol dimethacrylate (EGDMA) or hexamethylene diacrylate (HMDA) as chain extenders. A UAC Star monomer was prepared exactly as described in Example 1 above. EGDMA or HMDA (0.75 mmol) was added to the UAC Star monomer solution. Gelation with AIBN, and subsequent aging of the wet-gels in molds, and drying using SCF CO₂ were done exactly as described in the procedure of Example 2 above. This provided the desired ethyleneglycol diacrylate extended PUAC (PUAC-EG) or the hexamethylene diacrylate extended PUAC (PUAC-HD).

Example 7

Comparative characterization by solid phase ¹³C NMR of PUAC, PUAC-EG, and PUAC-HD. FIG. 8 shows the comparative ¹³C NMR spectra of PUAC, PUAC-EG, and PUAC-HD, showing that the chain extenders have been incorporated in the material in the prescribed molar ratio.

Example 8

Comparative characterization by FTIR and thermogravimetric analysis (TGA) of PUAC, PUAC-EG, and PUAC-HD. FIG. 9 shows a comparison of the FTIR spectra and TGA traces of PUAC, PUAC-EG, and PUAC-HD, showing the chain extenders do not alter substantially the functional group footprint or the thermal behavior of the basic PUAC materials.

Example 9

Comparative mechanical characterization of PUAC, PUAC-EG, and PUAC-HD. FIG. 10 shows a comparison of the mechanical properties of PUAC, PUAC-EG, and PUAC-HD, showing that chain extenders change substantially the flexural moduli of the material, but not its compressive behavior.

Example 10

Comparison of properties of PUAC, PUAC-EG, and PUAC-HD prepared at 9% w/w concentration of Star monomer precursors. FIG. 11 displays various comparative properties related to the porous structure of the PUAC, PUAC-EG, and PUAC-HD prepared at 9% w/w concentration of Star monomer precursors. For comparative bulk densities, progressively longer chain extenders give larger particles and significantly reduced surface areas.

Example 11

Comparison of particle properties of PUAC, PUAC-EG, and PUAC-HD prepared at 9% w/w concentration of Star monomer precursors. FIG. 12 displays comparative particle sizes determined with small angle x-ray scattering and SEM

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of the PUAC, PUAC-EG, and PUAC-HD prepared at 9% w/w concentration of Star monomer precursors, showing a generally good agreement between primary particle sizes with those calculated from gas sorption data (refer to FIG. 11), indicating absence of closed porosity.

Example 12

Comparison of properties of PUAC, PUAC-EG, and PUAC-HD prepared at 40% w/w concentration of Star monomer precursors. FIG. 13 displays various comparative properties related to the porous structure of the PUAC, PUAC-EG, and PUAC-HD prepared at 40% w/w concentration of Star monomer precursors. For comparative bulk densities, particle sizes are not affected by the presence of chain extenders.

Example 13

Comparison of particle properties of PUAC, PUAC-EG, and PUAC-HD prepared at 40% w/w concentration of Star monomer precursors. FIG. 14 displays comparative particle sizes determined with small angle x-ray scattering and SEM of the PUAC, PUAC-EG, and PUAC-HD prepared at 40% w/w concentration of Star monomer precursors, showing a generally good agreement between primary particle sizes with those calculated from gas sorption data (refer to FIG. 13), indicating absence of closed porosity.

Example 14

Comparison of quasi-static compression properties of PUAC, PUAC-EG, and PUAC-HD. FIG. 15 displays comparative quasi-static compression data of PUAC, PUAC-EG, and PUAC-HD. For comparative densities, the Young's

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modulus decreases and flexibility increases for longer chain extenders, but the overall energy absorption capability remains about the same.

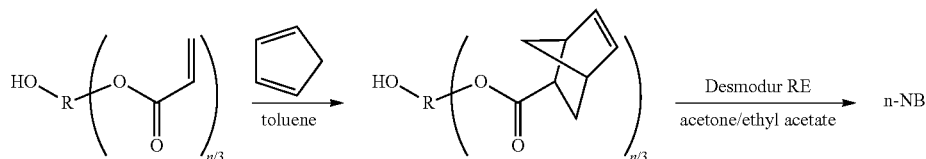
Example 15

Thermal conductivity properties of PUAC aerogels. FIG. 16 displays thermal conductivity data of PUAC aerogels. On the left, it shows raw data of heat transfer across an aerogel disk using a laser flash method. On the right, it shows the variation of thermal conductivity, X , with bulk density. 20-PUAC samples are about as good thermal insulators as glass wool (0.047 versus $0.040 \text{ W m}^{-1} \text{ K}^{-1}$).

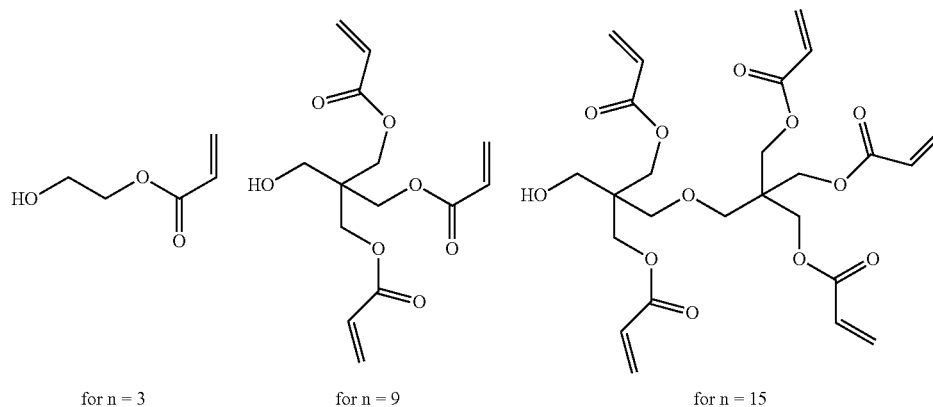
Example 16

Illustrative preparation of Star monomers containing norbornene groups. To a solution of 2-hydroxyethyl acrylate in toluene is added a solution of cyclopentadiene in toluene, the mixture is stirred under conditions known in the art to be suitable for Diels-Alder reaction, and the reaction between the alkene and diene is monitored until formation of the Diels-Alder adduct is complete. The resultant norbornene-containing hydroxyl compound is isolated using standard organic chemistry procedures. Subsequently, this norbornene-containing hydroxyl compound is subjected to reaction with TIPM in the presence of DBTDL as a catalyst in anhydrous acetone, following the procedure described in Example 1 above, thus producing the corresponding tris (norbornene)-containing Star monomer.

Additional examples of the preparation of norbornene-containing Star monomers following the procedure described herein are displayed in the following scheme, wherein n is an integer that is equal to 3, 9, or 15. The polynorbornene-containing Star monomer 15-NB shown below is obtained in the case where $n=15$:

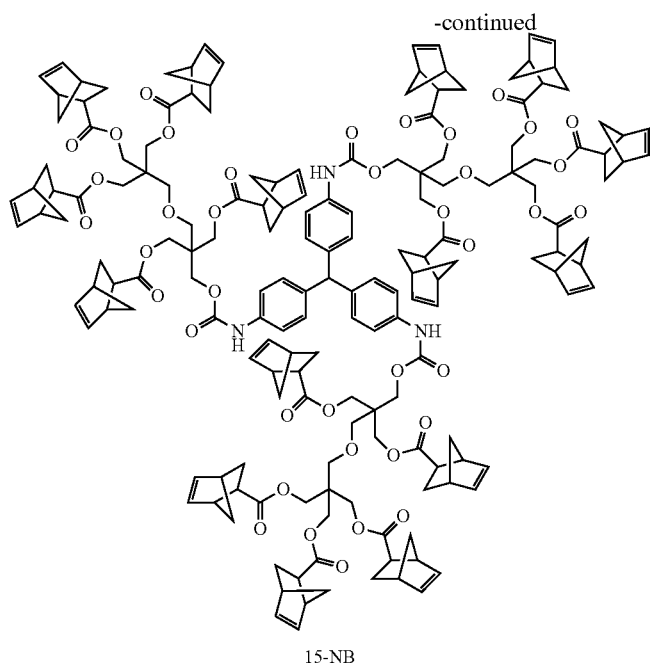


whereas:



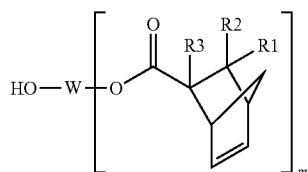
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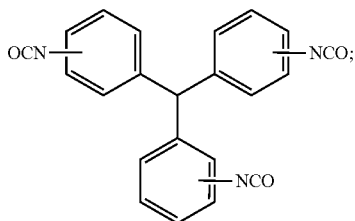


What is claimed is:

1. A norbornene-containing star monomer formed by the reaction of a norbornene-containing hydroxyl compound of the formula (IX)



with a tris(isocyanatophenyl)methane of formula (III)



wherein W is a C₁-C₆ straight chain or branched alkyl or alkoxyalkyl group;

wherein R₁, R₂, and R₃ are each independently hydrogen or C₁-C₆ straight chain or branched alkyl group; and, wherein m is an integer from 1 to 5.

2. A process for preparing the norbornene-containing star monomer of claim 1, the process comprising the step of mixing a tris(isocyanatophenyl)methane of formula (III) with three molar equivalents of the norbornene-containing hydroxyl compound of formula (IX) in an organic solvent.

3. The process of claim 2, further comprising the addition of a catalyst.

4. The process of claim 3, wherein the catalyst is an organotin compound.

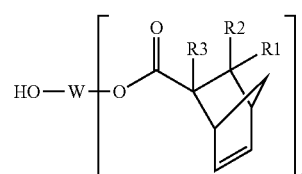
5. The process of claim 4, wherein the organotin compound is dibutyltin dilaurate.

6. The process of claim 2 wherein the organic solvent is selected from the group consisting of a ketone solvent, an ester solvent, and a combination thereof.

7. The process of claim 6, wherein the organic solvent is acetone, ethyl acetate, or a combination thereof.

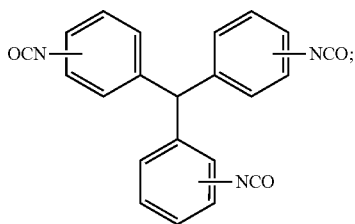
8. The norbornene-containing star monomer of claim 1, wherein, in the norbornene-containing hydroxyl compound of formula (IX), R₁=R₂=R₃=H, W=CH₂CH₂, and m=1; and, wherein, in the tris(isocyanatophenyl)methane of formula (III), the nitrogen atoms of the isocyanate groups are attached to their respective aryl rings at the 4-positions of the aryl rings.

9. A polynorbornene-polyurethane polymer formed by polymerization of a norbornene-containing star monomer obtained by the reaction of a norbornene-containing hydroxyl compound of the formula (IX)



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with a tris(isocyanatophenyl)methane of formula (III)

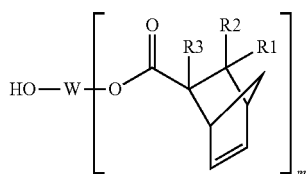


wherein W is a C₁-C₆ straight chain or branched alkyl or alkoxyalkyl group;

wherein R₁, R₂, and R₃ are each independently hydrogen or C₁-C₆ straight chain or branched alkyl group; and, wherein m is an integer from 1 to 5.

10. The polynorbornene-polyurethane polymer of claim 9, wherein, in the compound of formula (IX), R₁=R₂=R₃=H, W=CH₂CH₂, and m=1; and, wherein, in the tris(isocyanatophenyl)methane of formula (III), the nitrogen atoms of the isocyanate groups are attached to their respective aryl rings at the 4-positions of the aryl rings.

11. A polynorbornene-polyurethane polymer formed by copolymerization of a norbornene-containing star monomer with a dinorbornene-containing chain extender, wherein the norbornene-containing star monomer is obtained by the reaction of a norbornene-containing hydroxyl compound of the formula (IX)



(III)

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(IX)

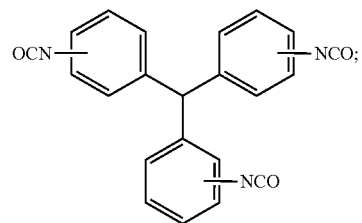
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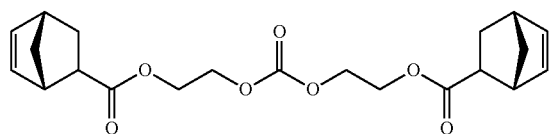
with a tris(isocyanatophenyl)methane of formula (III)



(III)

wherein W is a C₁-C₆ straight chain or branched alkyl or alkoxyalkyl group; wherein R₁, R₂, and R₃ are each independently hydrogen or C₁-C₆ straight chain or branched alkyl group; and, wherein m is an integer from 1 to 5.

12. The polynorbornene-polyurethane polymer of claim 11, wherein the dinorbornene-containing chain extender is a compound of the formula (XII):



(XII)

13. The polynorbornene-polyurethane polymer of claim 9 or claim 11, wherein the polymerization is ring-opening metathesis polymerization.

14. The polynorbornene-polyurethane polymer of claim 11, wherein, in the compound of formula (IX), R₁=R₂=R₃=H, W=CH₂CH₂, and m=1; and, wherein, in the tris(isocyanatophenyl)methane of formula (III), the nitrogen atoms of the isocyanate groups are attached to their respective aryl rings at the 4-positions of the aryl rings.

15. The polynorbornene-polyurethane polymer of claim 9 or claim 11, wherein said polymerization is carried out under ring-opening metathesis polymerization conditions.

16. The polynorbornene-polyurethane polymer of claim 9 or claim 11, wherein the polymer is an aerogel.

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