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Low-Velocity Impact Characterization of Fiber-Reinforced Composites with Hygrothermal Effect

Reference

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ABSTRACT

In this article, low-velocity impact characteristics of UHN125C carbon fiber/epoxy composite, including unidirectional (0°), cross-directional (0°/90°), and quasi-isotropic layups, were experimentally measured. The effect of the fiber orientation angle and stacking sequences on impact force and induced strain were measured via an instrumented drop-weight apparatus with special concern for the moisture absorption effect. Dried specimens were immersed in distilled water for a certain period of time to absorb water for intermediate and saturated moisture content. It was observed that the impulse was reduced with the increase in moisture content; on the other hand, strain increased with moisture, as measured by DBU-120A strain-indicating software (MADSER Corp., El Paso, TX). Impact damage is widely recognized as one of the most detrimental damage forms in composite laminates because it dissipates the incident energy by a combination of matrix damage, fiber fracture, and fiber-matrix debonding. Therefore, it is extremely important to know the impact strength of a structure, especially for applications in industries such as aerospace, ship design, and some other commercial applications. The use of composite materials in engineering applications is increasing rapidly because they have higher strength-to-weight ratios than metals. The strength, stiffness, and, eventually, the life of composite materials are affected more than conventional materials by the presence of moisture and temperature. Thus, it is necessary to analyze the response of composites in a hydrothermal environment.

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Keywords

impulse, dynamic strain, layup, moisture content, fiber orientation

Introduction

Fiber-reinforced composite materials have been increasingly used in load-bearing structures because of a number of advantages they have, such as high specific strength and stiffness, good fatigue performance, and corrosion resistance, over more conventional materials [1]. Carbon fiber is a material that consists of extremely thin fibers that measure about 0.0002–0.0004 in. (0.005–0.010 mm) in diameter and are composed mostly of carbon atoms.

The carbon atoms are bonded together in microscopic crystals that are more or less aligned parallel to the long axis of the fiber. The crystal alignment makes the fiber incredibly strong for its size. Carbon fiber can be combined with epoxy and then wound or molded to form composite materials, such as carbon fiber–reinforced plastic (also referred to as carbon fiber), to create a high strength-to-weight ratio material.

The need to develop and use light-weight structural components in the design of aircraft, automotive, and various sporting goods has brought increased application of composite material. Reliable performance of the advanced, high-strength material in critical applications depends on ensuring that each part placed in service satisfies the conditions selected in design. It is therefore necessary to ensure the quality of materials used and the integrity of the product during various stages of manufacturing as well as the final product.

A serious obstacle to the widespread utilization of composite laminates is its sensitivity to impact and static loads in the direction of thickness [2,3]. Low-velocity impact tests are often simulated by simple static indentation-flexure tests, which neglect the influence of dynamic effects on the structural response [4,5]. In epoxy-based laminates, static and dynamic tests are observed to produce the same damage modes and comparable force-deflection behavior [6]. Low-velocity impact is one of the most subtle threats to composite materials in aeronautics, owing to the weak bonds between the plies; even small energies imparted by out-of-plane loads can result in hardly detectable damages, causing considerable strength losses in tension and, especially, in compression [7].The energy absorbed during impact is mainly dissipated by a combination of matrix damage, fiber fracture, and fiber-matrix debonding. This may lead to significant reductions in the load-carrying capability of the laminate. Generally, the earliest observable damage affecting a laminate subjected to low-velocity impact is delamination, which is mainly responsible for compression strength impairment. For this reason, much research work has been devoted to the mechanisms of delamination initiation and growth [8–11]. Since the residual material properties after impact are of primary concern in applying damage-tolerance concepts, many authors have also tried to correlate analytically or experimentally the residual tension and compression strength with impact energy and damage mechanisms [12,13]. The problem of calculating the maximum contact force has also been addressed extensively by other authors who used spring-mass models or the principle of conservation of energy [14,15]. Fewer data are available on the penetration process and related energy at low velocity although this phenomenon has received considerable attention in the domain of moderately high-velocity or ballistic conditions [16].

Theoretical Analysis

IMPULSE

Using an oscilloscope, an impact force response with respect to time can be obtained, as shown in Fig. 1, and will later be expanded for measuring the impact response of first striking, excluding noise and rebound.

The first impact with the force plate, which is area under the curve P_{max}, can be simplified to a triangle in order to find the impact energy, as shown in Fig. 1. By the area of a triangle as shown in Eq 1, P_{max} is the maximum impact force.

$$I = \frac{1}{2}(P_{max})(\text{Time}) \tag{1}$$

IMPACT ENERGY

Impact energy represents the amount of work that can be performed by a single blow of a hammering mechanism. This is called potential energy, and the force is called a restoring force. As a general rule, Eq 2 represents the work done by a conservative force. For this particular experiment, the impact energy will be same for all tests.

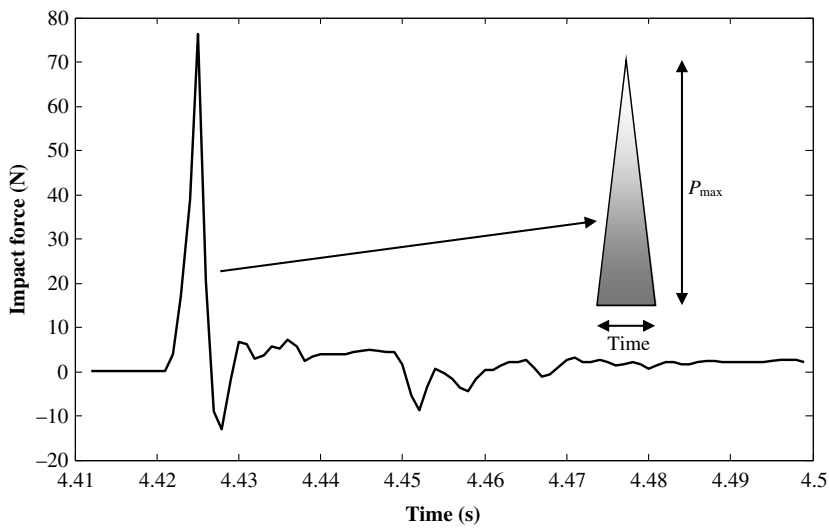
$$W = -\Delta PE \tag{2}$$

$$\Delta PE = mg\Delta h \tag{3}$$

where W is the work done, PE is mgh, ΔPE is the change in potential energy, m is the mass of the striker, g is the gravitational acceleration, and h is the height above impact surface.

FIG. 1

Impact force response for a particular specimen.



Specimen Preparation

PROCESSING OF LAMINATES

Three laminates of different stacking sequences were produced using unidirectional carbon/epoxy prepreg tape that was 0.113 mm thick. They were manufactured with high-strength carbon fiber that had a UHN125C grade, 39.32 GPa tensile modulus, 4.61 GPa tensile strength, 1.82×10^{-3} g/mm^3 fiber density, and 1.2×10^{-3} g/mm^3 resin density. Three different layup sequences (0° unidirectional, 0°/90° cross-directional, and quasi-isotropic layup) were experimentally examined, as given in Table 1.

Before putting layup in the autoclave for curing, a vacuum bagging process for carbon/epoxy laminates needed to made, as shown in Fig. 2. Once the vacuum was achieved, laminates were simultaneously cured in the autoclave at 125°C and 0.49033 MPa, according to the curing cycle as shown in Fig. 3. Initially, a square plate that measures 300 by 300 mm was fabricated and was later cut using a low-vibration wheel cutter in order to get the specimen of desired dimension. The composite laminates consisted of eight plies, resulting in nominal thickness of ~0.96 mm; the final specimen was a square plate measuring 127 by 127 mm.

HYGROTHERMAL EFFECT

In order to investigate the hot-wet environmental effect, the specimens were exposed to both temperature and moisture using a CW-20G refrigerating bath circulator (Jeio Tech Co., Ltd, Seoul, South Korea).

TABLE 1

Specimen layup sequence.

Specimen Number	Specimen Layup Sequence	Description
1	$[0_8]$	Unidirectional
2	$[0/90/0/90]_S$	Cross-ply
3	$[0/45/-45/90]_S$	Quasi-isotropic

FIG. 2

Vacuum bagging process for graphite/epoxy composites.

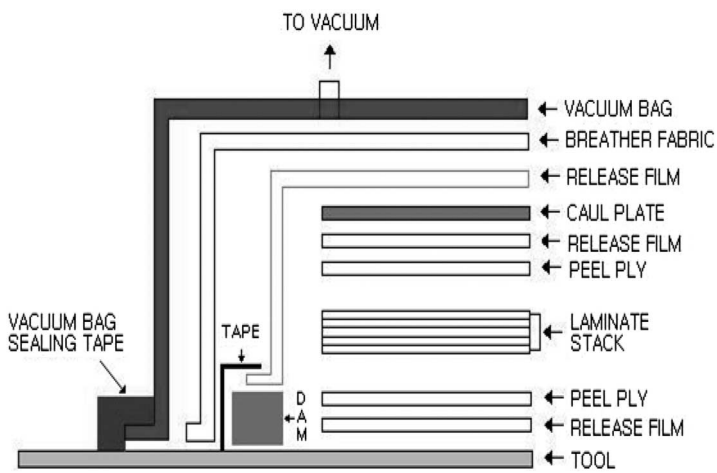
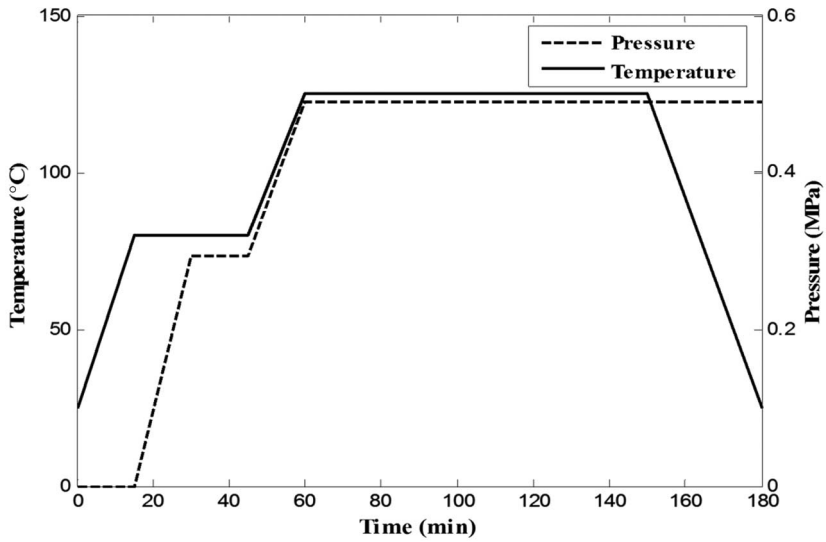


FIG. 3

Curing cycle in autoclave.



We followed these steps:

(1) Dry the specimens in the oven at 50°C to remove all moisture.
(2) Immerse the specimens in distilled water at 80°C for 8, 16, 24, and 35 days for different moisture contents, which were calculated using Eq 4.
(3) Dry the sample for a short period of time before impact testing.

$$M\,\% = \frac{W_m - W_d}{W_d} \cdot 100 \tag{4}$$

where, M is the moisture content, W_m is the weight of the wet sample, and W_d is the weight of the dry sample.

Strain Measurement and Experimentation

DYNAMIC STRAIN MEASUREMENT

Strain gauges are excellent for the measurement of dynamic strain processes. Mounting one strain gauge on each square plate, as shown in Fig. 4, allows the measurement of maximum strain in the specimen close to the impact load at 0°, 45°, and 90°. The gauge is mounted on a specimen under preload conditions, and the dynamic strain is measured using DBU-120A software. When the striker falls on the specimen, the striker penetrates it and the specimen splits. The measured strain is a long peak curve.

EXPERIMENTAL SETUP

The actual experimental setup and block diagram are shown in Figs. 5 and 6. Low-energy impact tests were conducted on samples that measure 127 by 127 by 0.96 mm in a Dynatup/GRC minitower drop-impact machine (Instron, Norwood, MA) with a striker mass of 2.02 kg. The specimen was placed on a rectangular plate that had a hole in the center and was fixed from the corners. The impact machine is equipped with a

FIG. 4

Strain gauges set up on a square plate.

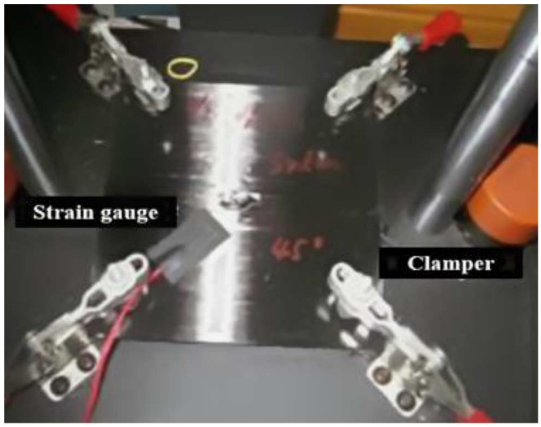
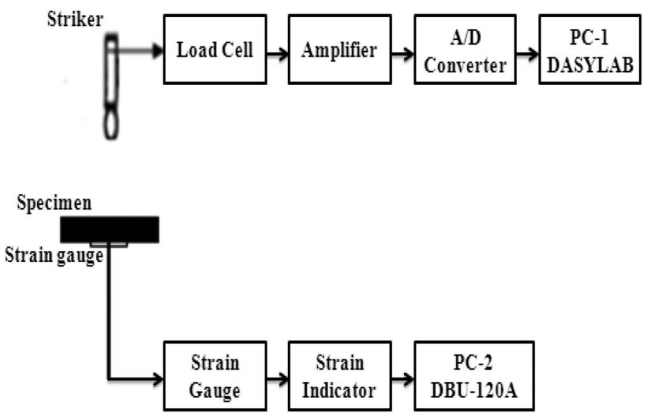


FIG. 5

Block diagram of complete experimental setup.



hemispherical nose of 12.7 mm diameter with rebound effect. The height of the striker was fixed (445 mm) with a falling velocity of 3.0 m/s. A piezometer load cell was placed on the upper part of hemispherical nose with the amplifier; the signals were recorded on a computer using DASYLab software (Measuring Computing Corp., Norton, MA) via A-D converter, as shown in Fig. 6.

Results and Discussion

Twenty-seven specimens were tested, and a total of nine specimens for each layup are mentioned in Table 1. Out of the nine specimens, three specimens were taken for each of three conditions (dry, intermediate, and saturated) and their mean value was considered. Moisture absorption takes place through a diffusion process governed by Fick's law, in which water molecules are transported from areas with higher moisture concentration to areas with lower moisture concentration [17,18]. Fig. 7 shows the weight increase as a function of exposure time. Like any other polymers, epoxies can absorb moisture when

FIG. 6

Complete experimental setup.

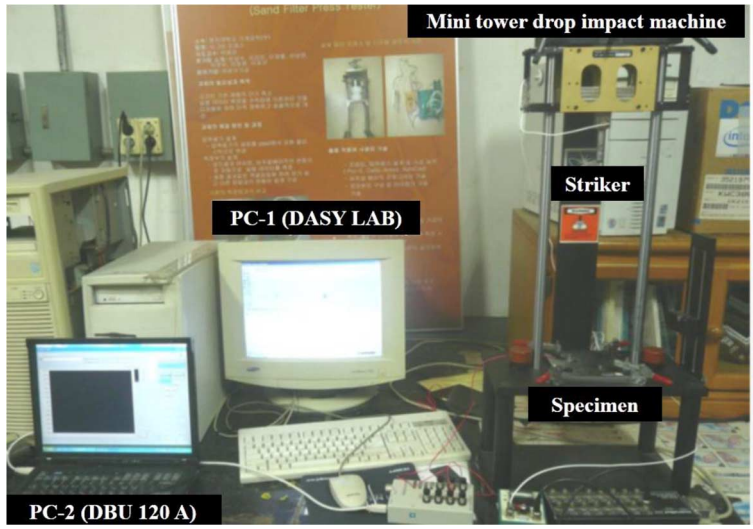
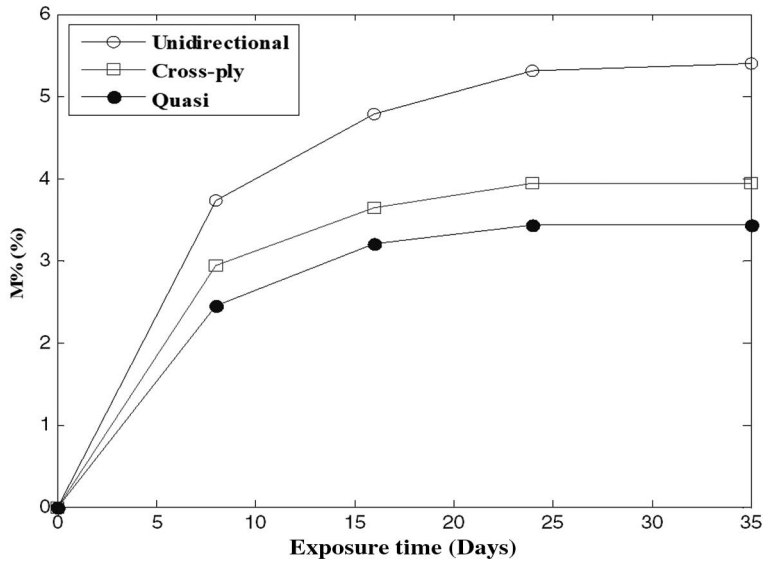


FIG. 7

Weight increase in terms of moisture content.



exposed to humid environments. From the results, it was found that the moisture absorption was not uniform: the unidirectional composites lead to absorption of 5.4 %, cross-ply composites with 3.9 %, and quasi-isotropic with 3.4 %. Similar results were obtained for moisture contents in previous research with different structures of the same composite [19]. From Fig. 7, it can be observed that the specimen's intermediate and saturated moisture content are at the exposure time of 8 and 35 days, respectively.

In the first experimental phase, the dynamic strain was measured, and the results were compared. Moreover, the strain variation that was due to the hygrothermal effect was

TABLE 2
Experimental results for all specimen types with moisture content.

Moisture	Specimen Type	Mass [g]	Moisture % [%]	Impulse [N-sec]	Strain [micro m/m]
Dry layup	0°	24.56	–	3.78	2,172.7
	0°/90°	24.81	–	4.51	1,842.6
	Quasi	24.68	–	5.68	1,640.9
Intermediate moisture	0°	24.86	1.61	3.12	2,415.7
	0°/90°	25.14	1.32	4.00	2,011.5
	Quasi	25.08	1.24	4.92	1,806.1
Saturated moisture	0°	24.74	1.76	2.57	2,587.6
	0°/90°	25.02	1.46	3.48	2,218.5
	Quasi	24.89	1.34	4.29	1,978.0

additionally investigated, as shown in Table 2 and Figs. 8–11. Three specimens were taken for experimentation, their individual responses are shown in Figs. 8 and 10, and the mean value for all parameters, including mass, moisture content, impulse, and strain is shown in Table 2 and plotted in Figs. 9 and 11. As far as the configuration of laminate is concerned, the maximum strain was observed in the unidirectional layup, and a minimum strain was observed in the quasi-isotropic layup. Investigation of the hygrothermal effect shows that the presence of moisture and heat damages the specimen because they increase the strain.

Using impact load, impulse was calculated, as shown in Figs. 10 and 11. The results are in strong agreement with the strain as the unidirectional specimen was the least rigid,

FIG. 8 Strain measurement for dry specimens.

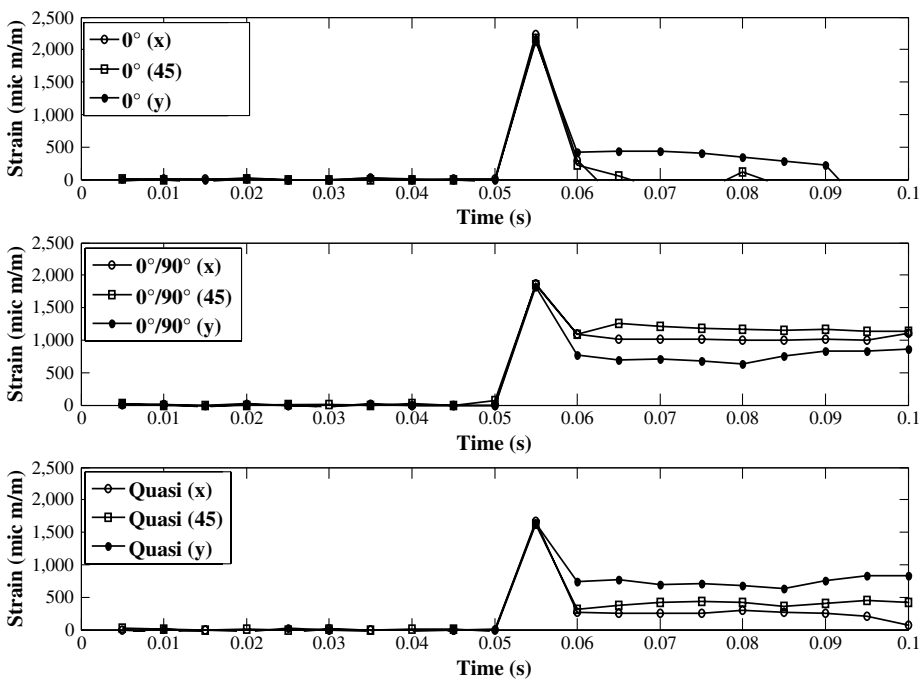


FIG. 9

Variation in strain for three layups with moisture content.

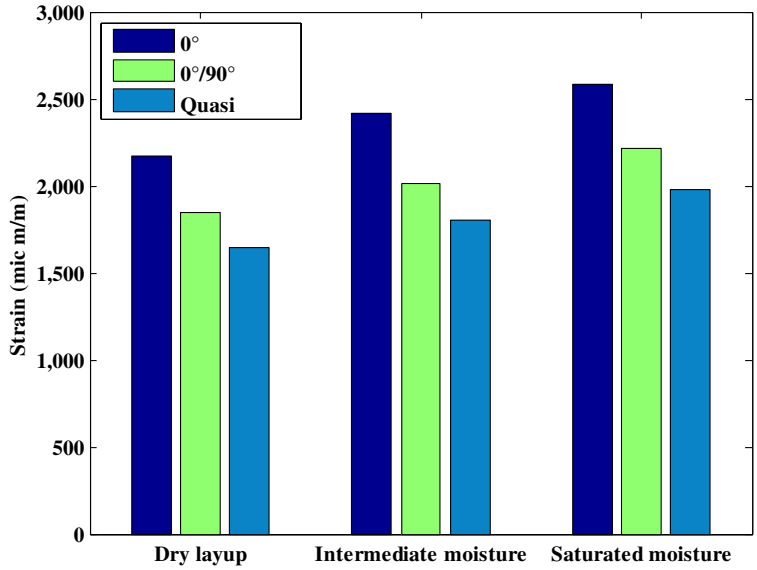
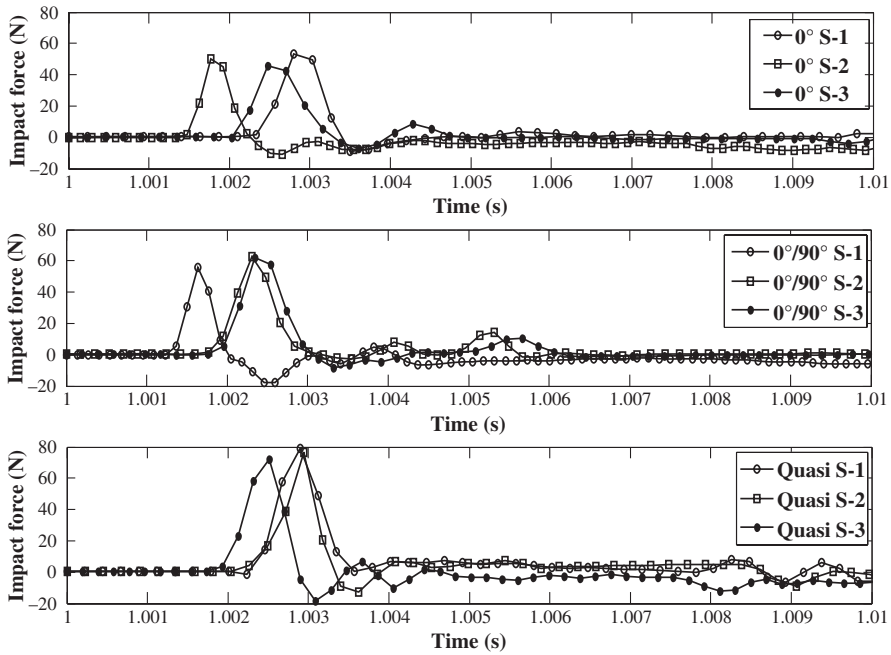


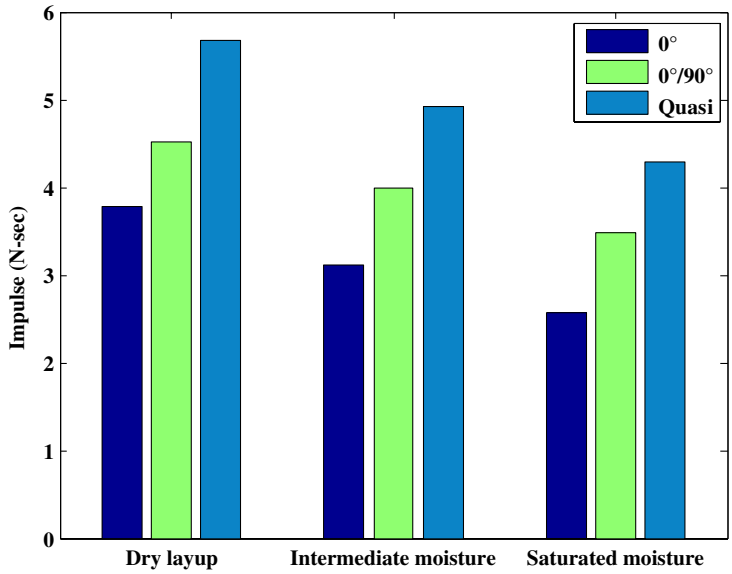
FIG. 10 Impulse force measurement in dry specimens.



so it produced maximum strain, and it was expected that its impact response would be the lowest. Moisture and temperature minimize the impact force and therefore a lower impulse was obtained for intermediate and saturated moisture content. The hygrothermal effect gradually reduces the impulse and increases the strain.

FIG. 11

Variation in impulse for three layups with moisture.



Conclusion

From this research on carbon fiber–reinforced composites, the following conclusion can be derived.

(1) Water absorptivity in unidirectional laminates was found to be higher than in cross-ply and quasi-isotropic laminates.
(2) During low-velocity impact tests, higher impulses were found in dry laminates in all stacking sequences; with the increment of moisture content, a reduction in impulse was observed.
(3) As far as the stacking sequence is concerned, the impulses in quasi-isotropic laminates were found to be much higher than in other stacking sequences in all proposed conditions.
(4) With the increment in moisture content, higher strain was found in all layups.
(5) Strain found in quasi-isotropic laminates was lower than unidirectional and cross-ply laminates in all proposed conditions.

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