

Recent Research in Science and Technology 2012, 4(8): 130-132  
 ISSN: 2076-5061  
 Available Online: <http://recent-science.com/>



# Thermoluminescence Study of Y<sub>2</sub>O<sub>3</sub>: Tb

Nameeta Brahme, Anuradha Gupta, D.P.Bisen and U.Kurrey

SoS in Physics and Astrophysics, Pt. Ravishankar Shukla University, Raipur (C. G.), 492010, India

## Abstract

Nanophosphors were prepared by the combustion technique using yttria and terbia powder as precursors and urea as fuels. The particle is investigated by X-ray diffraction analysis. X-ray diffraction patterns confirm the formation of pure cubic phase of Y<sub>2</sub>O<sub>3</sub>. The resulting product was characterized by FTIR spectra to evaluate the vibrational feature of the sample. TL emission spectra show intense peak around 520 nm.

**Keywords:** Nanophosphors, Y<sub>2</sub>O<sub>3</sub>, combustion method. PACS: 78.60.Kn

## INTRODUCTION

Over past few years, the synthesis and characterization of nanomaterials have attracted much attention in display and lighting technology. Nanostructured materials possess properties, which are often uniquely different from those exhibited by their macroscopic counterparts due to perturbations of the electronic distribution induced by reduced dimensionality. The development of nanocrystalline phosphors has become pivotal due to the ever-shrinking size of today's technology.

In particular, this oxide is chosen as host because it is characterized by a low-phonon frequency, which makes the non-radiative relaxation of excited states inefficient. Moreover, it has excellent refractory properties with a melting point of 2450 °C and a high thermal conductivity [1].

As regards the doping ions, REs are considered to be the most promising elements as activator of nanophosphors since they have particular optical properties because of their special electronic configuration. The characteristic luminescence of RE ions is attributed to the 4f shell, which is not entirely filled [2].

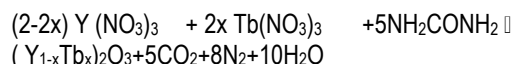
Particular attention has been drawn to trivalent Tb ions (4f<sup>8</sup>) as a doping element in hosts as yttrium or gadolinium oxide because of its narrow green bands originating from intra 4f-transitions. Soo et al., [3] investigated the local structure of Tb-doped Y<sub>2</sub>O<sub>3</sub> nanocrystals. Meng et al [4] reported the influence of size confinement on the energy transfer between Tb<sup>+3</sup> ions in Y<sub>2</sub>O<sub>3</sub> nanopowders.

Incorporation of rare earth in the nanocrystalline yttrium oxide matrix can be done by several techniques, such as the, polyol method [5], hydrothermal [6, 7], liquid phase reaction [8], aerosol pyrolysis [9], and sol-gel techniques [10], Pechini method [11]. In recent works, Combustion method has been successfully used in the synthesis of dielectrics networks doped with respect to prepare luminescent powder materials [1, 12-15].

## EXPERIMENTAL DETAILS

In this study Europium oxide (99.99%), Yttrium nitrate (99.99%), nitric acid, urea (90%) were used as starting raw material. To prepare Y<sub>2</sub>O<sub>3</sub>:Tb<sup>+3</sup>, RE(NO<sub>3</sub>)<sub>3</sub> stock solutions was prepared by dissolving Tb<sub>2</sub>O<sub>3</sub> in nitric acid. These two solution of Y(NO<sub>3</sub>)<sub>3</sub> and Tb(NO<sub>3</sub>)<sub>3</sub> were mixed according to the formula (Y<sub>0.95</sub>Tb<sub>0.05</sub>)<sub>2</sub>O<sub>3</sub> in a beaker and then a suitable amount of urea was added. A urea to metal nitrate was employed to prepare the precursor solution. Finally

this sample was transferred to crucible and fired in a furnace at 600°C. The Synthesis reaction is



The morphologies and sizes of the Tb doped Y<sub>2</sub>O<sub>3</sub> were determined by X-ray diffraction studies with Cu K $\alpha$  radiation ( $\lambda=1.5418 \text{ \AA}$ ). XRD data were collected over the range 20°-80° at room temperature. The X-ray diffraction patterns have been obtained from X-ray powder diffractometer. The particle size was determined using the sherrer's formula. The Shimadzu 1700 UV-VIS spectrometer is used to obtain the optical absorption spectra of Eu doped Y<sub>2</sub>O<sub>3</sub>. For recording TL, samples were exposed to UV radiations at different time. TL glow curves were recorded with the help of TLD reader (Model 1009I). TL spectra were recorded by using interference filters. FTIR spectra was also recorded to evaluate the vibrational feature of the sample.

## RESULTS AND DISCUSSION

### Morphological Analysis

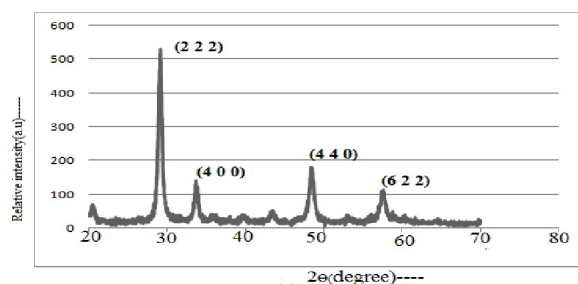


Fig1. XRD pattern of Y<sub>2</sub>O<sub>3</sub>:Tb

Fig.1 presents the XRD result for nanophosphor Y<sub>2</sub>O<sub>3</sub>: Tb. Four different peaks are obtained at 2 $\theta$  values of 29.12°, 33.78°, 48.46° and 57.56° and the peaks correspond to diffraction at (222), (400), (440) and (622) planes, respectively. The spectrum clearly shows the expected body centered cubic crystalline phase for the nanopowders in agreement with (ICSD #160890). Crystallite size estimated based on a simple Debye scherrer method yield value of around 15 nm.

**Optical Absorption Spectra**

The study of optical absorption is important to understand the behavior of nano-crystals. A fundamental property is the band gap-the energy separation between the filled valence band and the empty conduction band. Optical excitation of electrons across the band gap is strongly allowed, producing an abrupt increase in absorption at the wavelength corresponding to the band gap energy. This feature in the optical spectrum is known as the optical absorption edge.

Fig. 2 shows the absorption spectra of Y<sub>2</sub>O<sub>3</sub>: Tb phosphor in the range of 190 nm -700 nm. The absorption edge is found at 223 nm. The band gap was calculated corresponding to absorption edge (223 nm). The band gap was found to be 5.5eV for Y<sub>2</sub>O<sub>3</sub>: Tb phosphors.

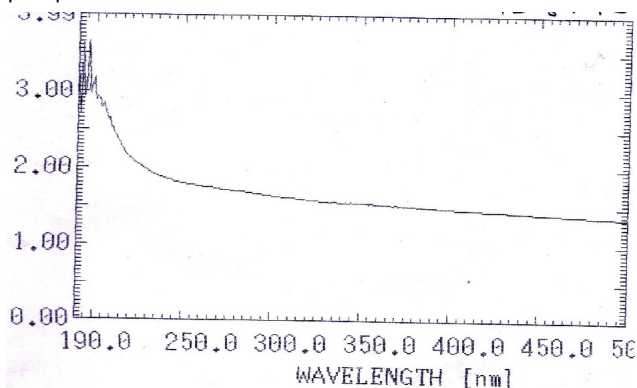


Fig 2. Absorption spectra of Y<sub>2</sub>O<sub>3</sub>: Tb phosphors

**FTIR Analysis**

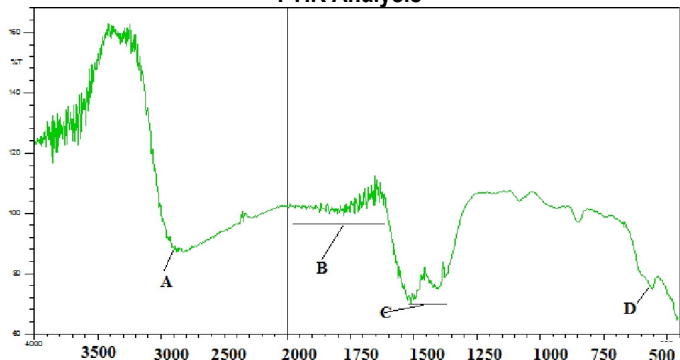


Fig 3. FTIR spectra of Y<sub>2</sub>O<sub>3</sub>: Tb

FTIR spectra of Y<sub>2</sub>O<sub>3</sub>: Tb that was synthesized at the heating temperature peak of 600°C is shown in Figure 3. A broad peak (A) at around 3000 cm<sup>-1</sup> is originated from O-H stretching in hydroxyl groups. Overtone or combination band [B] located in the range 2000–1650 cm<sup>-1</sup> [16]. Absorption at around [C]

1500– 1350 cm<sup>-1</sup> might originated from C-H bend and C-C stretching, around 562 cm<sup>-1</sup> from Y-O stretching [D] [17]. The absorption peaks resulted by bonding of metal ions usually appears at short wave numbers [18].

**Thermoluminescence Process**

Fig. 4 shows the TL glow curve of UV-irradiated Y<sub>2</sub>O<sub>3</sub>: Tb nanophosphor for different irradiation time. The TL glow curve show single peak at 297°C. It is seen that the TL intensity increase with

increasing

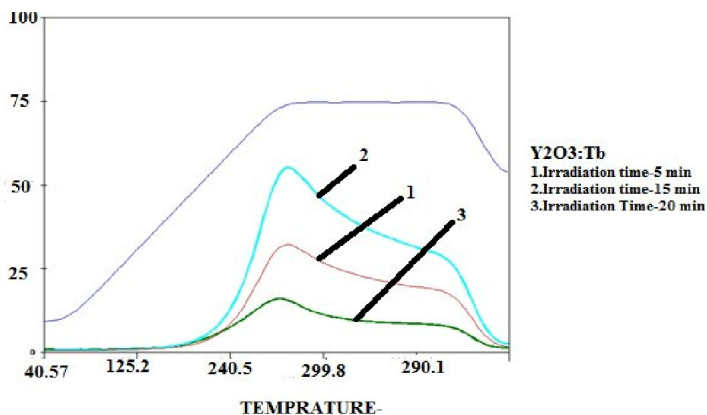


Fig 4. TL glow curve of UV-irradiated Y<sub>2</sub>O<sub>3</sub>: Tb

irradiation time. The terbium-doped yttria phosphor presents a TL glow curve with one peaks (T<sub>max</sub>) centered around 297°C. Tb<sup>3+</sup> ion acts as a recombination center affecting the TL response of the Y<sub>2</sub>O<sub>3</sub> material. As the particle become smaller, ions at the surface are not saturated in coordination. Electrons or holes may be excited easily and escape from the ions and they are trapped at surface states located in the forbidden gap. When the sample is heated the electrons are de-trapped and their subsequent recombination with holes gives rise to the light emission. . The TL intensity is increased for higher exposed time because the new trapping center is created by increasing exposure time of UV radiation. The trapping centers provide energy states available between valence band and conduction band responsible for radiative recombination. As the radiation dose is increased the color centers increase so initially the intensity increase with increasing UV dose. After the color center get saturated so that TL intensity saturated for higher UV dose.

**Calculation of Kinetic parameters**

The activation energy of Y<sub>2</sub>O<sub>3</sub>: Tb phosphor is calculated by Initial rise method.

**Initial Rise Method**

The method is based on the fact that as the glow curve initially begins to rise, the density of unoccupied recombination centers and the density of trapped electrons remain approximately constant, and hence the TL intensity is strictly proportional to exp(-E/kT).

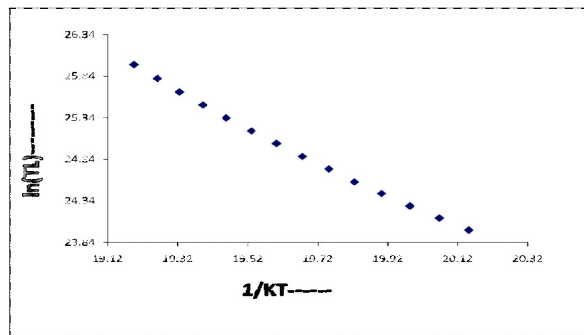


Fig 5. ln(TL) versus 1/KT graph to determine activation energy for maximum

## TL peak

Fig. 5 show graph of  $\ln(TL)$  Vs  $1/KT$  for maximum TL glow peak of fig 4. In applying initial rise method, a straight line is obtained. From the slope of line, activation energy  $E$  is evaluated for  $Y_2O_3:Tb$  nanophosphors for 15 min irradiation time. The activation energy of Eu doped  $Y_2O_3$  is found 1.64 eV.

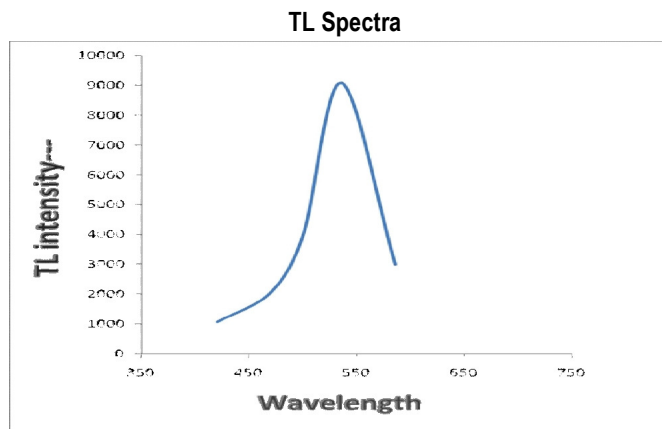


Fig 6. TL spectrum of  $Y_2O_3:Tb$

Figure 6 shows the TL spectrum of  $Y_2O_3:Tb$  ranges from 300 to 700 nm. The emission band was observed at 520 nm in green region of the spectrum.

## CONCLUSION

We have investigated the Thermoluminescence phenomena in the  $Y_2O_3:Tb$  (5 mol %) phosphor prepared with combustion synthesis. The TL properties of Tb doped  $Y_2O_3$  nanophosphor has been investigated for UV irradiation and it is seen that at a particular irradiation time 15 min high intensity of glow peak was found. From optical absorption edge, the band gap energy was found to be 5.5 eV for  $Y_2O_3:Eu$ . TL spectrum of  $Y_2O_3:Tb$  shows the emission band at 520 nm in green region of the spectrum.

## REFERENCES

[1]. Anh TK, Benal loul P, Barthou C, Giang LTK, Vu N, Minh

LQ(2007) *J Nanomater.*

- [2]. Blasse G, Grabmaier BC (1994) Springer, Berlin
- [3]. Y. L. Soo, S. W. Huang, Z. H. Ming, and Y. H. Kao, G.C. Smith, E. Goldburt, R. Hodel, B. Kulkarni, J. V. D. Veliadis, and R. N. Bhargava *Journal of Applied Physics*, Volume 83, Issue 10, pp. 5404-5409 (1998).
- [4]. Meng Q, Chen B, Xu W, Yang Y, Zhao X, Di W, Lu S, Wang X, Sun J, Cheng L, Yu T, Peng Y (2007) *J Appl Phys* 102:093505-1.
- [5]. Flores-Gonzalez MA, Ledoux G, Roux S, Lebbou K, Perriat P, Tillement O (2005). *J Solid State Chem* 178: 989-997, 2004.
- [6]. Wang L, Shi L, Liao N, Jia H, Du P, Xi Z, Wang L, Jin D (2010) *Mater Chem Phys* 119:490-494.
- [7]. Liu Z, Yu L, Wang Q, Tao Y, Yang H (2011) *J Lumin* 131:12-16.
- [8]. Park JH, Back NG, Kwak MG, Jun BE, Choi BC, Moon BK, Jeong JH, Yi SS, Kim JB (2007) *Mater Sci Eng C* 27:998-1001.
- [9]. Sotiriou GA, Schneider M, Pratsinis SE (2011) *J Phys Chem C* 115:1084-1089.
- [10]. Wu YC, Garapon C, Bazzi R, Pillonnet A, Tillement O, Mugnier J (2007) *J Appl Phys* A697-704.
- [11]. Saengkerdsub S, Im HJ, Willis C, Dai S (2004) *J Mater Chem* 14:1207-1211.
- [12]. Song H, Wang J (2006) *J Lumin* 18:220-226.
- [13]. Muenchausen RE, Jacobsohn LG, Bennett BL, McKigney EA, Smith JF, Valdez JA, Cooke DW (2007) *J Lumin* 126:838-842.
- [14]. Guinier, A. X-ray Diffraction, Freeman, San Francisco (1963).
- [15]. Nameeta Brahme, Anuradha Gupta, D.P. Bisen, R.S. Kher, S. J. Dhoble, *Physics Procedia* 29(2012) 97 - 103,
- [16]. R. M. Silverstein, G. L. Bassler, and T. C. Morrill, John Wiley & Sons, New York, NY, USA, 1991.
- [17]. G. Yao, L. Su, X. Xu, and J. Xu, *Journal of Alloys and Compounds*, vol. 462, no. 1-2, pp. 381-385, 2008.
- [18]. K. Nakamoto, *Infrared and Raman Spectra of Inorganic and Coordination Compounds*, John Wiley & Sons, New York, NY, USA, 4th edition, 1986.