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# Large Uniaxial Magnetic Anisotropy of Co–Pt Perpendicular Films Induced by Lattice Deformation

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A study of the thickness dependence of the uniaxial magnetic anisotropy  $K_u$  of Co–Pt perpendicular films deposited on Ru seed layers was extended over a wide range of Pt content from 10 to 30 at%, to further investigate the origin of the significant increase in  $K_u$  on reducing the film thickness, which was previously reported for  $Co_{80} Pt_{20}/Ru$  films.  $K_u$  increased significantly as the thickness decreased in all series of films with various Pt content. The rate of increase of  $K_u$  with reducing thickness became increasingly large as the Pt content decreased. It should be noted that the  $K_u$  of 2-nm-thick films was an almost constant  $2 \times 10^7$  erg/cm<sup>3</sup> over the range of Pt contents studied, although  $K_u$  was maximized at 25–30at% Pt content for films thicker than 5 nm. Experimental results suggest that the increase in  $K_u$  on reducing the thickness was mostly related to a reduction of the hcp Co–Pt lattice c/a ratio.

Index Terms—Co–Pt films, fcc volume fraction, hcp lattice, lattice deformation, Ru underlayer, surface anisotropy, uniaxial magnetic anisotropy.

### I. INTRODUCTION

LARGE uniaxial magnetic anisotropy  $K_u$  of  $1.5-2 \times 10^7$  erg/cm<sup>3</sup> was obtained in Co-20at%Pt perpendicular films deposited on Ru seed layers with no evidence of a Co<sub>3</sub>Pt-type ordered structure [1], [2].  $K_u$  increased significantly as the thickness decreased, and our results suggested that the increase in  $K_u$  was mostly related to a reduction of the c/a ratio of the hcp Co-Pt lattice due to epitaxial growth of Co-Pt on the Ru layers.

In this paper we extend the study of the thickness dependence of  $K_u$  to Co–Pt films with 10, 14, 20, 25, and 30at%Pt content, to further investigate the origin of the high  $K_u$ . The lattice mismatch between Co–Pt and Ru increased as the Pt content decreased; therefore, it was expected that reducing the Pt content would result in a further enhancement of  $K_u$ . We discuss the variation of  $K_u$  as a function of crystal structure in relation to the Pt content. The lattice deformation for films thinner than 10 nm was analyzed by synchrotron radiation grazing incidence X-ray diffraction (XRD), and included in the discussion. Moreover, the thickness dependence of  $K_u$  for Co–14at%Pt films deposited on Au seed layers was also examined for comparison; the lattice mismatch between Co–Pt and Au is much larger than that between Co–Pt and Ru.

### II. EXPERIMENTAL PROCEDURE

Co–Pt films were deposited on 4-in  $SiO_x/Si$  substrates by the cosputtering method with Co and Pt targets using a dc-magnetron sputtering system [2]. The film thickness was varied from 2 to 60 nm. Ru (20 nm) films were used as seed layers with a preseed layer of Pt(10 nm)/Ta(5 nm). No substrate heating was carried out during the deposition process. Capping layers of Pt (2 nm) were deposited on top of the Co–Pt layers. For comparison, Co–14at%Pt films with Au seed and capping layers were also



Fig. 1.  $K_u$  as a function of thickness  $\delta$  for Co–Pt films with various Pt content. The inset figure shows the Pt content dependence of  $K_u$  for the same films.

deposited.  $K_{\rm u}$  was obtained by subtracting the shape anisotropy  $2 \pi M_s^2$  from the value measured by torque magnetometry. The accuracy of the measured  $K_{\rm u}$  values was confirmed in our previous work [2].

# **III. RESULTS AND DISCUSSIONS**

# A. $K_{\rm u}$ as Functions of Pt Content and Film Thickness

The saturation magnetization  $M_s$  was almost constant in all series of films with various Pt content, except for films thinner than 5 nm, for which  $M_s$  increased as the thickness decreased. This increase in  $M_s$  was due to polarization of the Pt capping layer [2], suggesting that no significant diffusion occurred at the top/bottom surfaces of the Co–Pt layer.

Fig. 1 shows the values of  $K_{\rm u}$  for films with various Pt content as a function of film thickness  $\delta$ .  $K_{\rm u}$  increased as  $\delta$  decreased, and reached about  $2 \times 10^7$  erg/cm<sup>3</sup> at  $\delta = 2$  nm in all series of films. The  $K_{\rm u}$  values of these films were a few times larger than those of Co–20at%Pt films previously reported [3]–[6]. The  $K_{\rm u}$ values of films thinner than 5 nm exceed the  $K_{\rm u}$  of Co<sub>3</sub> Pt ordered alloy films [7]. Moreover, the rate of increase of  $K_{\rm u}$ 

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Fig. 2. The fcc volume ratio,  $V_{\rm fcc}/V_{\rm total}$ , for 20 nm-thick films as a function of Pt content. The inset figure shows the thickness dependence of  $V_{\rm fcc}/V_{\rm total}$  for films with 14at%Pt content.

with reducing  $\delta$  became increasingly large as the Pt content decreased, as expected. The  $K_{\rm u}$  of films of  $\delta = 2$  nm was almost constant for films with Pt content from 10 to 30at%, as seen in the inset figure, although  $K_{\rm u}$  was maximized at 25–30at%Pt content for films thicker than 5 nm. Our investigation of the first- and second-order energy terms of the uniaxial anisotropy,  $K_{u1}$  and  $K_{u2}$ , for Co–20at%Pt films revealed [1], [2] that the ratio of  $K_{u2}$  to  $K_{u1}$  was 0.16 for films with  $\delta = 40$  nm, but reduced to less than 0.03 on decreasing  $\delta$  to 5 nm. The  $K_{u2}$ values of these films were particularly small, unlike in previous reports [5], [6]. A similar thickness dependence of  $K_{\rm u}$  was also observed in Co–14at%Pt films deposited on Au seed layers, as shown in the figure; however, the  $K_{\rm u}$  values were smaller than those of Co–14at%Pt films deposited on Ru seed layers.

## B. Structure

X-ray diffraction patterns revealed that the Co-Pt films with Ru seed layers had good c-axis orientation perpendicular to the film plane, with a c-axis distribution  $\Delta \theta_{50}$  of about 2.8°, independent of Pt composition. No evidence of a Co<sub>3</sub>Pt type ordered structure was found in the X-ray diffraction patterns. The diffraction of the fcc(111) plane was weakly observed by tilting the sample an angle  $\chi = 70.5^{\circ}$ .  $\chi$  is the angle between the normal axis to the film plane and the optical plane. This means that hcp Co-Pt lattice of grains contains a small number of fcc layers. We measured the diffraction intensities for the hcp(101) and fcc(111) planes, respectively, and calculated the fcc volume fraction,  $V_{\rm fcc}/V_{\rm total}$ , taking account of the preferred grain orientation, the multiplicity, form factor, Lorentz-polarization factor, and Debye–Waller factor [8]. Fig. 2 shows the values of  $V_{\rm fcc}/V_{\rm total}$  as a function of Pt content for films with  $\delta = 20$  nm.  $V_{\rm fcc}/V_{\rm total}$  could not be estimated for films thinner than about 10 nm because of the very low diffraction intensities.  $V_{\rm fcc}/V_{\rm total}$  was about 0.15 for films with a 10at%Pt content; however,  $V_{\rm fcc}/V_{\rm total}$  decreased significantly as the Pt content increased, becoming less than 0.02 at 30at%Pt content. It should be noted that the value of  $V_{\rm fcc}/V_{\rm total}$  was nearly constant as a function of  $\delta$ , as shown in the inset figure for Co–14at%Pt films. This was supported by cross section (dark field) transmission electron microscopy (TEM) images; thin fcc layers form nearly



Fig. 3. Values of c/a ratios and  $V_{\text{lattice}}$  of the hcp Co–Pt lattice as a function of thickness. Symbols: X = 20 (circle), 14 (square), and 10 (triangle).

independently of the film thickness. The use of Au instead of Ru seed layers resulted in  $V_{\text{fcc}}/V_{\text{total}}$  nearly doubling.

## C. Appearance of High $K_{\rm u}$ in Relation to Structure

The magnetic anisotropy of fcc Co–Pt should be negligibly small compared to that of hcp Co–Pt, and the Pt content dependence of  $V_{\rm fcc}/V_{\rm total}$  was qualitatively in good agreement with that for  $K_{\rm u}$ , shown Fig. 1. This result suggests that an increase in  $V_{\rm fcc}/V_{\rm total}$  plays a role in reducing  $K_{\rm u}$  in the low Pt content region. However, the change in  $K_{\rm u}$  was significantly large compared to that in  $V_{\rm fcc}/V_{\rm total}$ . Moreover, the value of  $V_{\rm fcc}/V_{\rm total}$ was nearly constant as a function of  $\delta$ . It is likely that the thickness dependence of  $K_{\rm u}$  was mainly caused by other factors.

The surface anisotropy should be taken into account when discussing the value of  $K_{\rm u}$  in the thin film region. Assuming that the increase in  $K_{\rm u}$  was caused by the surface anisotropy, we tried to estimate the surface anisotropies at the top and the bottom surfaces,  $K_{\rm s1}$  and  $K_{\rm s2}$ . However, the  $K_{\rm u} \sim 1/\delta$  curve was not linear in all series of films, similar a series of Co–20at%Pt films [2]. Moreover, the calculated values of  $K_{\rm s1} + K_{\rm s2}$  in these films ranged from 5 to 8 erg/cm<sup>2</sup>, which are much larger than the surface anisotropy of Co-based alloy films [9]. These results indicate that the thickness dependence of  $K_{\rm u}$  cannot be explained only by the surface anisotropy.

The lattice mismatch between Co–Pt and Ru increased as the Pt content decreased; therefore, it was expected that reducing the Pt content would result in an enhancement of lattice deformation. We measured the hcp-CoPt lattice spacings of the (002), (004), and (112) planes of these films, and used them to calculate the lattice constants a and c. The diffraction of the hcp (112) plane was measured by tilting the sample at an angle  $\chi = 58-59^{\circ}$ . Fig. 3 shows the values of the c/a ratio and the lattice volume,  $V_{\text{lattice}}$ , (=  $(3^{1/2}/2)ca^2$ ) as a function of  $\delta$  for films with 10, 14, and 20at%Pt content. In the fig, the results for Co–14at%Pt films in the very thin film region (from 3 to 20 nm), analyzed by synchrotron radiation grazing incidence XRD method, are also shown. The c/a value decreased as  $\delta$  decreased for all series of media. This c/a reduction was mainly caused by



Fig. 4. Relationship between the c/a ratio of the hcp Co–Pt lattice and  $K_u$ . Symbols: X = 20 (circle), 14 (square), and 10 (triangle).

an increase in the *a* value (the lattice spacing in the lateral direction), probably due to epitaxial growth of Co–Pt on the Ru layers. The c/a reduction in Co–14at%Pt films reached 1% on decreasing  $\delta$  from 60 to 3 nm. We expected that increasing the lattice mismatch by reducing the Pt content would enhance the lattice deformation. However, the c/a values for the 10at%Pt films were large relative to the others, which may be related to the formation of a small number of fcc layers in the hcp-lattice. Moreover, it should be noted that  $V_{\text{lattice}}$  increased significantly, especially in the  $\delta$  region less than 20 nm. The increase in  $V_{\text{lattice}}$  for Co–14at%Pt films with Au seed layers showed larger c/a and smaller  $V_{\text{lattice}}$  compared to films with Ru seed layers.

Fig. 4 shows the relationship between the c/a ratios of the hcp Co–Pt and  $K_u$ . The increase in  $K_u$  on reducing  $\delta$  was coincident with a reduction of the c/a ratio. This relationship between the c/a ratio and  $K_u$  was qualitatively in good agreement with theoretical predictions [10]–[13]. The rate of increase of  $K_u$  with reducing c/a value was slightly larger in series of films with low Pt content; however, no significant difference was found.

Experimental results suggested that the increase in  $K_u$  on reducing the thickness was mostly related to a reduction of the hcp Co–Pt lattice c/a ratio. The results for Co–14at%Pt films with Au seed layers are located in the extrapolated area of the  $K_u - c/a$  plot for the films with Ru seed layers. We deposited Co–25at%Pt binary alloy films (10-nm thickness) on various seed layer materials, and found [14] that  $K_u$  and the c/a ratio showed a greater dependence on the seed layer materials used, and  $K_u$  increased significantly as the c/a ratio decreased. This result supports our conclusion. Moreover, it is likely that a huge increase in the lattice volume enhances the increase in  $K_u$  on reducing the thickness especially below 10 nm; however, more intensive effort is required to clarify the relationship between the lattice deformation and  $K_u$ .

### IV. CONCLUSION

The rate of increase of  $K_u$  of Co–Pt films with reducing thickness becomes increasingly large as the Pt content decreases as we expected, and  $K_{\rm u}$  of 2-nm-thick films was almost a constant  $2 \times 10^7$  erg/cm<sup>3</sup>, independent of Pt content in the region from 10 to 30at%Pt. Our results suggest that the increase in  $K_{\rm u}$  on reducing the thickness was mostly related to a reduction of the hcp Co–Pt lattice c/a ratio.

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