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The Magnetic and Electric Properties of $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$
Crystalline Compounds and Amorphous Alloys*

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Synopsis

The manganese concentration x dependence of the average magnetic moment μ_0 , the Curie temperature T_c , the pressure effect on T_c and the electric resistance for the ferromagnetic amorphous alloys $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ were investigated. The Curie temperature decreases linearly with increasing x and μ_0 had a maximum around $x=0.15$. The magnetic susceptibility vs. temperature curves for all the prepared amorphous alloys obey the Curie-Weiss law above T_c . The pressure effect on T_c is that T_c has a value of 1.06 K/kbar at $x=0$ and is decreasing with increasing x and becomes zero at $x=0.4$. These results are analyzed on the basis of the pair interaction model and the local environment effect. There arise two kinds of minimum in the resistance vs. temperature curves. That the resistance minimum at high temperature has a strong correlation with T_c is assured by measuring the transverse effect of the resistance with amorphous alloys $x=0.4$. The resistance minimum for the amorphous alloys and crystalline compounds are found at low temperature under ferromagnetic state.

I. Introduction

There are three kinds of compounds Co_3B , Co_2B and CoB in the crystal structure of which is D011 for Co_3B , C16 for Co_2B and B27 for CoB in the Co-B system. The compounds Co_3B and Co_2B are ferromagnetic and magnetic moment of Co atom was determined as 747 K, $1.12\mu_B$ for Co_3B and 422 K, $0.896\mu_B$ for Co_2B ¹⁾. The variation of the average magnetic moment per

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magnetic atom in (Me, Me')-B compounds (Me, Me' =Cr, Mn, Fe, Co and Ni) was discussed by Lundquist et al.²⁾ with the band model. According to their discussion the electronic state of the cobalt atom in the Co₂B compound is very similar to that in the pure nickel. On the other hand, the magnetic moment per cobalt atom (μ_0) in Co_{1-x}Mn_x binary amorphous alloys decreases with increasing boron concentration and its concentration dependence of μ_0 shows that the values of μ_0 at x=0.25 and 0.33 are almost the same as that in the crystalline compounds Co₃B and Co₂B respectively and the magnetic moment of the cobalt will become zero near x=0.48 from $\mu_0=1.3$ at x=0.12. It is known that most of the ferromagnetic Co-metalloid amorphous alloys show the similar metalloid concentration dependence of μ_0 . The number of nearest neighbor atoms around a cobalt on Co_{1-x}Mn_x amorphous alloys is in the range of 10 to 13 for x \leq 0.4 and then the cobalt atom may be surrounded with those as in packing of fcc.

When the manganese atoms replace cobalt atoms partially in Co₂B crystalline compound the average magnetic moment per magnetic atom vs. manganese concentration curve has a maximum around 25 at %Mn and the Curie temperature curve as a function of manganese concentration also has a maximum at 10 at %Mn. It was reported by Kadomatsu et al.³⁾ that the compounds (Co_{1-x}Mn_x)₂ are ferromagnetic for x \leq 0.6 and for x>0.6 may be paramagnetic. They have interpreted the x dependence of μ_0 and T_c on the basis of the pair interaction model which was proposed by Kouvel⁴⁾ to explain the magnetic properties of Ni-Mn alloys. It was concluded that the exchange interaction of Co-Co and Co-Mn should be ferromagnetic and that of Mn-Mn should be antiferromagnetic. Obi et al.⁵⁾ performed the same analysis for (Co_{1-x}Mn_x)_{100-y}B_y ternary amorphous alloys and obtained the average manganese magnetic moment per magnetic atom curve as a function of B concentration. The value of the magnetic moment of MnB crystalline compound which is ferromagnetic appears well on the above curve. Mn₂B, however, may be paramagnetic¹⁾.

Recently the pressure effect in the Curie point of (Co_{1-x}Mn_x)₂B compounds has been reported by Kodomatsu et al.³⁾. When the hydrostatic pressure is applied the Curie temperature decreases and the absolute value of its pressure deviation (dT_c/dP) increases with increasing Mn concentration. The thermal expansion of Fe-B amorphous alloys showed the anomaly below the Curie point and its behavior is invar-like⁶⁾. The thermal expansion of Co-B amorphous alloys, however, did not show the anomalous behavior near the Curie point even if the pressure effect on T_c for Fe-B and Co-B amorphous alloys yields in the same order. The resistivity vs. temperature curves for the ferromagnetic amorphous alloys Fe-B and Co-B have a min-

imum at low temperature. While the ferromagnetic amorphous alloys Re-nickel⁷⁾ (Re: rear earth metals) show a minimum in electric resistivity vs. temperature curve near the Curie temperature. It was reported that there appear 2 kinds of minimum in the resistivity vs. temperature curves for Fe-(Ni,Co)-Zr alloys⁸⁾. One of the two minima appears near Curie temperature and another at low temperature. It is interesting that there appears a minimum in the electric resistivity vs. temperature curve under the ferromagnetic state. It is useful for understanding the electronic state of Co atoms to study the resistance minimum in both crystalline and amorphous Co-B alloys having the same constitution.

In this paper, the Curie temperature, pressure effect on the Curie temperature and electric resistance on $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys are reported and discussed in the basis of the local environment effect and the pair interaction model comparing with those of the crystalline compounds with the same composition.

II. Experimental Procedure

The $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ ($0 \leq x \leq 0.4$) amorphous alloys were prepared a single roller quenching technique in the form 30 μm thick ribbon. The structure of all the prepared specimens was examined by X-ray diffraction and confirmed to be amorphous. The temperature dependence of magnetization and susceptibility were measured at temperatures between 4.2-600 K in a magnetic field of 6 kOe by a magneto balance. Hydrostatic pressure was applied to a specimen in a Teflon pressure cell filled with kerosine or silicone oil by using a piston cylinder type device⁹⁾. The pressure was calibrated by using the Hg-solid-liquid transition temperature. The Curie temperature was determined by an ac transformer method. Density was measured by a normal Archimedeian method. The error of density is smaller than about 0.1%. The crystalline $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ (all elements 3N pure) in the desired proportion under a purified argon gas atmosphere the products were checked to be in a single phase by X-ray diffraction. The specimens were cut carefully into a parallelepiped (1x2x4 in mm). Electric resistance was measured by a four probes method with dc current in the temperature range from 4.2 to 500 K.

III. Experimental Results

The magnetic susceptibility of $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys was measured in the temperature range from room temperature to 680 K. Its reciprocal magnetic susceptibility χ_g^{-1} is shown in Fig.1. χ_g^{-1} depending

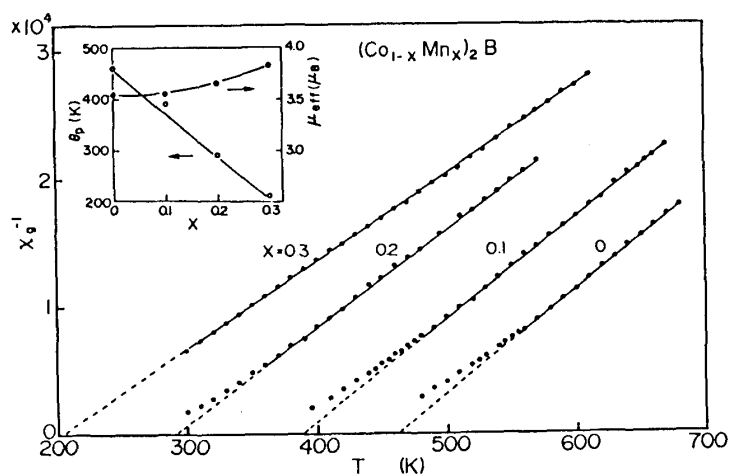


Fig.1 Temperature dependence for $(\text{Co}, \text{Mn})_2\text{B}$ amorphous alloys. The inset shows the paramagnetic Curie temperature (θ) and effective magnetic moment per transition atom (μ_{eff}).

on the temperature varies linearly as following the Curie-Weiss law. The paramagnetic Curie temperature θ_p and effective magnetic moment per transition metal atom μ_{eff} depend on the manganese concentration x as shown in the inset of the figure. The values of θ_p and μ_{eff} at $x=0$ obtained to be 460 K, 3.6 and the variation for θ_p decreases linearly and for μ_{eff} increases monotonously with increasing x . The permeability for $x=0$ and 0.2 depending on the temperature under pressure is shown in Fig. 2. The Curie temperature T_c was defined with intersection point of two extrapolated lines of the permeability vs. temperature curve as shown in the figure. This determination is very effective since the magnetization

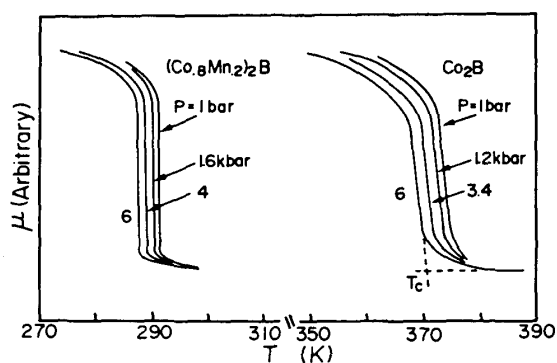


Fig.2 Temperature dependence of the initial susceptibility under hydrostatic pressure for the amorphous Co_2B and $(\text{Co}_{0.8}\text{Mn}_{0.2})_2\text{B}$ alloys.

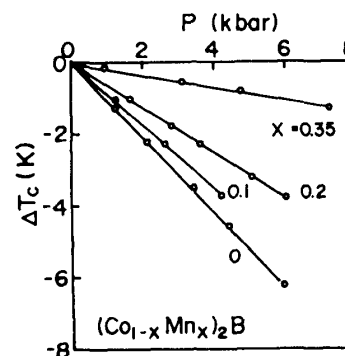


Fig.3 Curie temperature shift by compression for $(\text{Co}, \text{Mn})_2\text{B}$ amorphous alloys.

vs. temperature curve is not shown the sharp variation in the vicinity of the Curie temperature. The T_c of all the present amorphous alloys shifts to the lower temperature side by compression. This Curie temperature shift ΔT_c as a function of pressure was plotted in Fig.3 as the manganese concentration x is a parameter. The slope of the ΔT_c vs.

pressure (P) curve (dT_c/dP) increases with increasing x and this x dependent dT_c/dP curve is shown in Fig.4 together with that of the crystalline compounds by Kadomatsu et al.³⁾. The sign of the grade of the

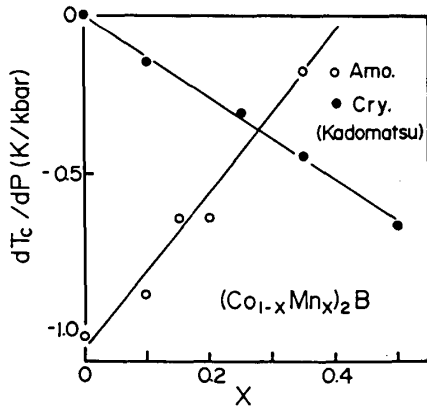


Fig.4 Mn concentration x dependence of the Curie temperature shift vs. pressure curve for $(\text{Co}, \text{Mn})_2\text{B}$ amorphous alloys and crystalline compounds after Kadomatsu et al. (ref.2).

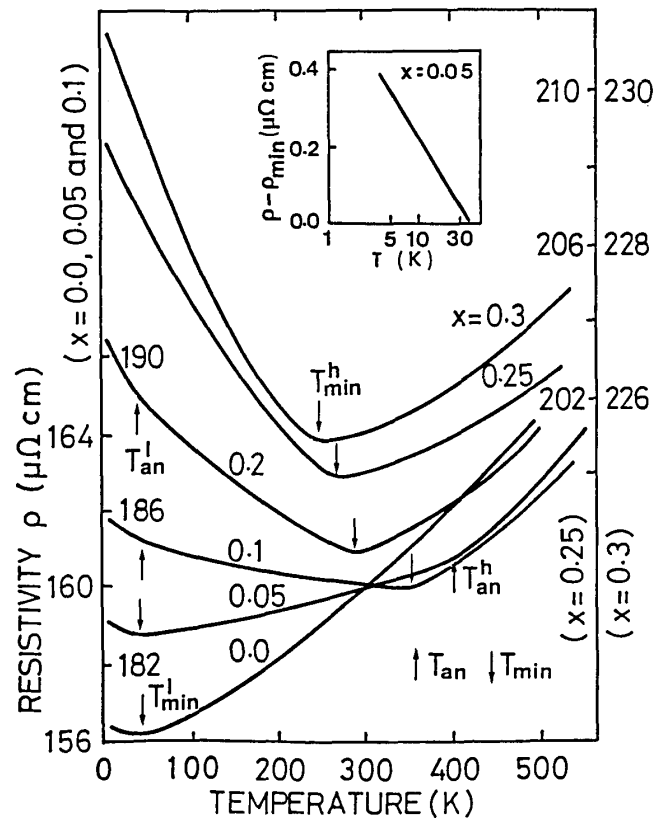


Fig.5 Temperature dependence of the electric resistivity ρ for $(\text{Co}, \text{Mn})_2\text{B}$ amorphous alloys; T_{\min}^l and T_{an}^l stand for the temperature of the resistivity minimum and anomaly at low temperatures respectively, and T_{\min}^h and T_{an}^h stand for those at high temperatures respectively.

curve for $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ crystalline specimens is opposite to that for the amorphous alloys. That is, the value of dT_c/dP for the amorphous alloys is 1.06 K/kbar at $x=0$ and decreases linearly and becomes zero K/bar at 0.4 when x increases. However, that for the compounds is zero K/kbar at $x=0$ and increases linearly and 0.5.

Fig.5 shows the resistivity vs. temperature curves for the $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys. The curves for $x=0$ and 0.05 show a minimum at low temperatures and those for $x=0.1$ and 0.2 show an anomaly at low temperatures. On the other hand, another minimum appears also near the Curie temperature for $x \geq 0.1$ in addition to the low temperature resistivity-minimum. The anomaly, however, appears near the Curie point for $x=0.05$. The resistivity for the crystalline compounds also was measured in

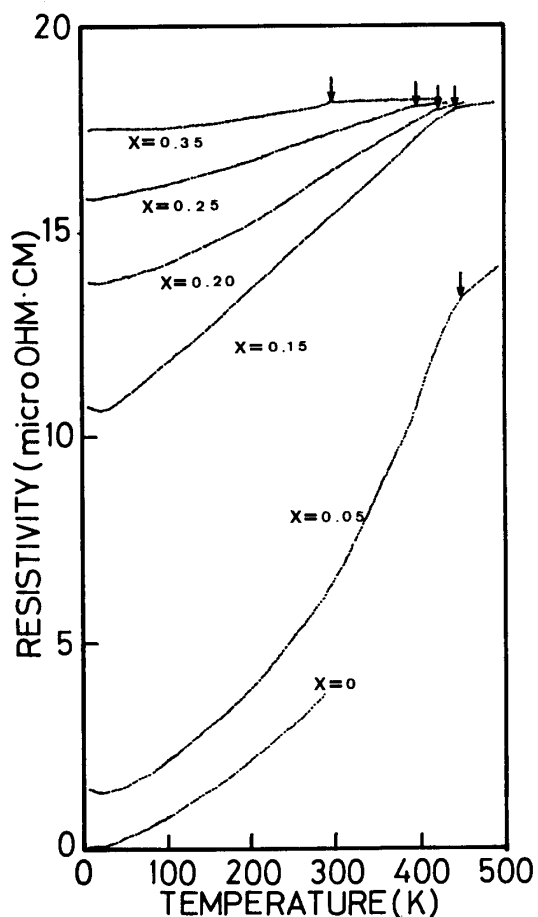


Fig.6 Temperature dependence of the resistivity for $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ crystalline compounds. Arrows show the Curie temperature.

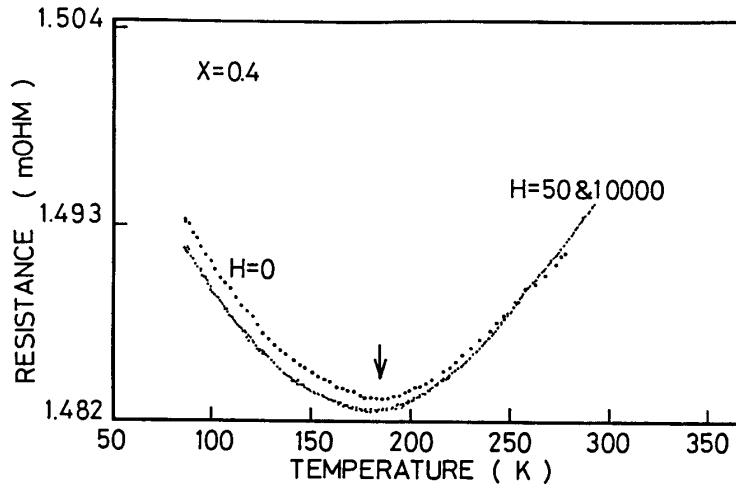


Fig.7 Temperature dependence of the resistance at magnetic field $H=0, 50$ and 10000 Oe . A arrow shows the resistance minimum.

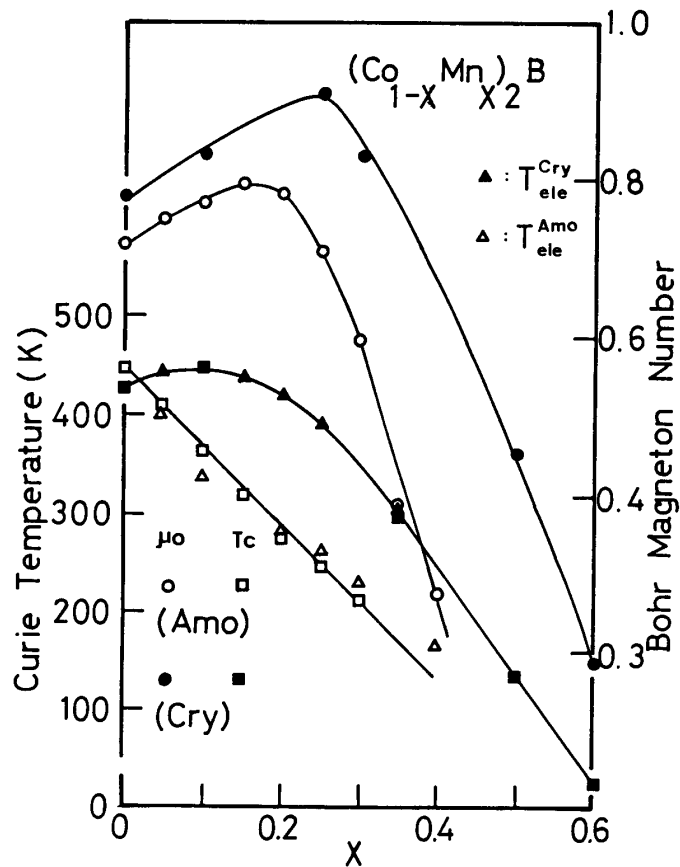


Fig.8 Manganese concentration x dependence of the average magnetic moment per magnetic atom and the Curie temperature for $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys and crystalline compounds. \triangle and \blacktriangle : T_c determined from the electric resistivity measurement.

the temperature range from 4.2 K to 450 K. The obtained result is shown in Fig.6. The curve for $x \geq 0.05$ has a minimum at low temperature and show a similar anomaly at the Curie point as those for the ordinary ferromagnetic metals have. The curve for $x=0$ does not have a minimum at low temperature. Fig.7 shows the experimental result in the case of applying the magnetic field in a perpendicular to the surface of the amorphous ribbon specimen. The temperature of the minimum at $H=0$ is 180 K and this value was plotted in Fig.8 as Curie temperature. This minimum temperature is not depend on the magnetic field. The values of the resistance for $H=50$ Oe and 10000 Oe is not different in the temperature range of 77 to 273 K. The resistance value at $H=5$ Oe is smaller than that at $H=0$ till 240 K.

The average magnetic moment per transition metal atom at 0 K (μ_0) and Curie temperature of $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys are shown as a function of x in Fig.8 together with those of the compounds with the same composition. The closed marks in Fig.8 are for the crystalline specimens where the circles and squares were quoted from the data by Kadomatsu et al.³⁾ and the triangles were determined by the electric resistance measurement. The open marks mean for the amorphous alloys and the circles are on the magnetic moment and the squares are on the Curie point. The curves on the moment for both the crystalline and noncrystalline $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ have a maximum at $x=0.25$ and at $x=0.2$ respectively and after that decrease sharply to zero around $x=0.5$ for the amorphous alloys. The curve on the Curie point for the crystalline compounds has a maximum at $x=0.1$. The curve, however, for the amorphous alloys decreases monotonously with increasing x .

We found the discrepancy and agreement between the experimental results for $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys and crystalline compounds. In the next section, these results are discussed with the pair interaction model and local environment effect.

IV. Discussins and Conclusions

The density was measured to obtain the information of a mean distance between metal-metal atoms in $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys. The slope of the observed density D vs. manganese concentration x curve (dD/dx) was determined to be -1.6 g/cm^3 . The mean distance d_m between the nearest neighbor atoms in Co_2B amorphous alloys is estimated to be about 2.53 Å from the interpolation of the experimental values 2.5 Å for $x=0.185$ ¹⁰⁾ and 2.57 Å for $x=0.4$ ¹¹⁾ determined by X-ray diffraction. Assuming that the mean distance is proportional to a length of a cube

having the average volume for a cobalt atom which is defined from deviding the whole volume by the number of cobalt atoms, the mean distance can be calculated from the density by taking the proportional constant $K=1.067$ for Co_2B amorphous alloys which was obtained from deviding the mean distance ($d_m=2.53$ A) with the length of a side of the average cube(2.387). Therefore the d_m for $x=0.3$ is estimated to be 2.58 A. This assumption is proper if the packing on neighbor atoms is hold symmetrically, for example the K value is 1.122 in the case of fcc and 1.091 in the case of bcc packing.

The reciprocal magnetic susceptibility shown in Fig.1 follows the Curie-Weiss law so that it may be allowed to use the localized moment model. AS the Curie temperature is a function of manganese concentration x , let it denote $T_c(x)$. According to the pair interaction model, $T_c(x)$ is expressed as follow³⁾;

$$T_c(x) = T_{\text{CoCo}}(1-x)^2 + 2T_{\text{CoMn}}x(1-x) + T_{\text{MnMn}}x^2, \quad (1)$$

where T_{CoCo} , T_{CoMn} and T_{MnMn} stand for the interaction energy in the temperature unit with respect to Co-Co, Co-Mn and Mn-Mn pair respectively. The original pair interaction model was proposed by Kouvel⁴⁾. In many cases, however, this model was modified the simpler one assuming that the cobalt and manganese atoms have a constant magnetic moment independent of manganese concentration x and that a cobalt atom or a manganese atom as a center atom loses the moment or is coupled antiparallel with the nearest neighbor atom moments when a atom is surrounded by the more than a critical number no of manganese atoms. The values of T_{CoCo} , T_{CoMn} and T_{MnMn} can be determined as reproducing the experimental T_c vs. x curve in Fig. 8. The result is $T_{\text{CoCo}}=448$ K, $T_{\text{CoMn}}=48$ K and $T_{\text{MnMn}}=-352$ K. On the other hand, these values for the crystalline specimens reported by Kadomatsu et al.³⁾ are $T_{\text{CoCo}}=426$ K, $T_{\text{CoMn}}=596$ K and $T_{\text{MnMn}}=-934$ K. The average magnetic moment μ_0 vs. x curve in Fig.9 is specified as follow: when the Co atoms are replaced with a small amount of Mn the average moment increases since the manganese magnetic moment (1.35 μ_0) is larger than the cobalt one (0.715 μ_0). A cobalt or a Mn atom will be surrounded with more Mn atoms as Mn concentration increases. According to the local environment effect, center atoms (Co or Mn atoms) lose the magnetic moment or couple antiferromagnetically with the neighbor atoms when the number of manganese atom is greater than the critical no. and then the average magnetic moment has a maximum value and is decreasing with increasing x . Since the cobalt moment is zero in the range of $x \geq 0.3$ it is expected that the mictomagnetism takes place by the antiferromagnetic interaction of Mn-Mn pairs. Obi et al.⁵⁾ have found the

anomalous behavior of the magnetization at low temperatures. It suggests that there could exist the inhomogeneous magnetic state below the anomalous temperature. The critical number of n_0 is an important parameter when the environmental effect is discussed. The experimental result on the magnetic moment in the amorphous alloys can be explained well with the critical number $n_0=4$ and nearest neighbor atom number $N=10$.¹²⁾ The n_0 for Co-Mn alloys is estimated to be 4¹³⁾, but one for the crystalline $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ compound is given by Kadomatsu et al.³⁾ as 6. The effect that the value of the magnetic moment depends on the nearest neighbor distance has been neglected in the local environmental effect. The nearest neighbor distances for Co-Mn alloys and for $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys are almost same values (2.5 Å) and that for the compounds is about 2.75 Å. Therefore it seems that the distance effect can not be neglected. In fact, the maximum on the moment for the compounds in Fig. 8 shifts to the higher concentration of x comparing with that for the amorphous alloys in order that the effect reducing the average magnetic moment in the crystalline compounds is weaker than in the amorphous alloys. As $T_c(x)$ is differentiated with respect to pressure P , the following equation is obtained as;

$$\begin{aligned} \frac{dT_c}{dP} = & x^2 \left(\frac{dT_{\text{CoCo}}}{dP} + \frac{dT_{\text{MnMn}}}{dP} - 2 \frac{dT_{\text{CoMn}}}{dP} \right) \\ & + 2x \left(\frac{dT_{\text{CoMn}}}{dP} - \frac{dT_{\text{CoCo}}}{dP} \right) + \frac{dT_{\text{CoCo}}}{dP} \end{aligned} \quad (2)$$

The value of the differentiation of T_{ij} ($i, j = \text{Co or Mn}$) with respect to P are determined with the experimental values in Fig. 4. That is; $dT_{\text{CoMn}}/dP = -1.06$ K/kbar, $dT_{\text{CoMn}}/dP = 0.26$ K/kbar and $dT_{\text{MnMn}}/dP = 1.16$ K/kbar. The values for the crystalline $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ obtained by Kadomatsu et al.³⁾ are $dT_{\text{CoCo}}/dP = 0$, $dT_{\text{CoMn}}/dP = -0.65$ and $dT_{\text{MnMn}}/dP = -1.3$ K/kbar. Supposing that the pressure effect in the magnetic moment is neglected the interaction between Co-Co atoms or between Mn-Mn atoms for the amorphous alloys is weakened ferromagnetically or antiferromagnetically with increasing pressure. The ferromagnetic interaction of Co-Mn distance. According to the interaction curve for the manganese atoms shown in Fig. 9 which was proposed by Yamada¹⁴⁾ the antiferromagnetic interaction at 2.77 Å will be stronger when the Mn-Mn distance decrease. Therefore dT_{MnMn}/dP for the compounds has a negative value. All the figures show the neighbor atoms positions and the numerical number stand for the number of the atoms. On the other hand, the mean distance for the amorphous alloys is from 2.53 to 2.58 so that dT_{MnMn}/dP has a positive value.

The electric resistivity vs. temperature curves for Fe-B, Co-B

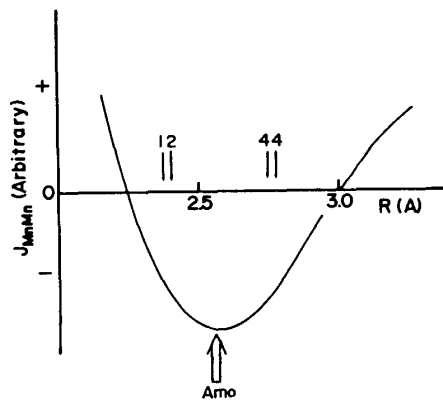


Fig.9 Interaction curve on $J_{MnMn} (\propto T_{MnMn}^{-1})$ as a function of the distance R (in Å unit) between Mn-Mn atoms in α -Mn by Yamada (ref.14). Small allows show the respective location of eleven neighbor Mn atoms in $(Co, Mn)_2B$ crystalline compounds. A bold arrow shows the average distance between Mn-Mn atoms in $(Co, Mn)_2B$ amorphous alloys.

and Fe-P amorphous alloys have a minimum at low temperature.¹⁵⁾ There also appears such a minimum in the case of $(Co_{1-x}Mn_x)_2B$ ($x < 0.2$) amorphous alloys.¹⁶⁾ When the x is more than 0.2 the resistivity vs. temperature curve has an anomal instead of the minimum. The origin of the minimum has been considered to be the Kondo effect or the two level model. The resistivity for the $(Co_{1-x}Mn_x)_2B$ crystalline compounds in Fig. 6 also show the minimum near 20 K. This minimum may be attributed to the Kondo effect which is expected to appear even in the ferromagnetic materials if there are a few of manganese atoms being in extreme weak internal field. In fact, it is shown from the analysis with the pair interaction model that when a manganese atom is surrounded by Co atoms and Mn atoms it is possible that the gross interaction for the center Mn atom becomes to be zero by cancellation with the Co-Mn ferromagnetic and Mn-Mn anti-ferromagnetic interaction. The resistivity vs. temperature curve for the crytalline specimens shows a kink at the Curie temperature. The curves for the amorphous alloys ($x \geq 0.2$), however, show the minimum near the Curie points. The close correlation between the minimum and the Curie temperature is concluded from the fact that the minimum temperature are on the Curie temperature vs. Mn concentration curve as shown in Fig.8.

The temperature dependent electric resistance curves for $x=0.4$ on $(Co, Mn)_2B$ amorphous alloys at the applied magnetic field $H=50$ and 10000 Oe in Fig. 7 are almost coincident each other so that the magnetization should be saturated at 50 Oe. Below the Curie temperature, the resistance decreases with applying magnetic field such a phenomena found in the pure nickel and is known as the transverse effect. The difference between the resistance at 0 and 50 Oe is almost constant till the temperature of the minimum and become small above 180K. It means that there exist Curie ponit in the vicinity of the resistance minimum and then the strong correlation between the Curie temperature and minimum. Obi et al.⁵⁾ pointed out that the minimum is not intrinsic since the resistance vs.

temperature curve showed the hysteresis near the Curie temperature. In the present case, however, the curve showed a reversible variation. This discrepancy is in order that their amorphous alloys specimens have higher Curie temperature near which there is a crystallization point and then it is possible to crystallize partially in the specimens once they are heated up over the Curie temperature.

The concept of the pair interaction model and local environment effect is very useful for analyzing our experimental results. The Co atom in $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys loses the magnetic moment at smaller Mn concentration than that for the compounds. The strength of the Co-Mn interaction becomes weaker with increasing the number of Co atoms having no magnetic moment. This thing causes the phenomenon that the T_c vs. x curve for the amorphous alloys shows the monotonous decrease with increasing Mn concentration, nevertheless that for the compounds have a minimum. Furthermore, it is predicted that there may arise the mictomagnetism for $(\text{Co}_{1-x}\text{Mn}_x)_2\text{B}$ amorphous alloys ($x \geq 0.4$) by Mn-Mn pair antiferromagnetic interaction. The other hand, it is pointed out from the electric resistivity measurement that the existence of the Mn atom which is in a local paramagnetic state under the ferromagnetic state is possible for both the amorphous alloys and crystalline compounds. The value of the effective magnetic moment for CoB amorphous alloys obtained from the reciprocal susceptibility is about four times larger than the average magnetic moment. It means that the variation of the magnetic moment in the pair interaction model and local environment effect cannot be neglected. It is important how to understand that the Co or Mn magnetic moment is annihilated by the manganese atoms neighbor. The investigation with FMR and the pressure dependence of the magnetization will be performed in the future in order to make the electronic state of the magnetic atoms clear.

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