

An Experimental Derivation of Partial Structure Functions of Amorphous Pd\_<80>Si\_<20> Alloy Using Combination of X-ray, Electron and Neutron Diffraction Experiments

著者	Fukunaga Toshiharu, Suzuki Kenji			
journal or	Science reports of the Research Institutes,			
publication title	Tohoku University. Ser. A, Physics, chemistry			
	and metallurgy			
volume	28			
number	2			
page range	208-217			
year	1980-03-29			
URL	http://hdl.handle.net/10097/28170			

An Experimental Derivation of Partial Structure Functions of Amorphous Pd<sub>80</sub>Si<sub>20</sub> Alloy Using Combination of X-ray, Electron and Neutron Diffraction Experiments\*

Toshiharu Fukunaga and Kenji Suzuki

The Research Institute for Iron, Steel and Other Metals

( Received January 31, 1980 )

# Synopsis

Three partial interference functions corresponding to Pd-Pd, Pd-Si and Si-Si correlations have been separated using the combination of three total interference functions observed by neutron, X-ray and electron diffraction experiments respectively. The number-concentration correlation functions were calculated from the partial Pd-Pd, Pd-Si and Si-Si correlation functions.

#### I. Introduction

In metal-metalloid amorphous alloys metalloid atoms play an important role to stabilize the amorphous structure. However, it is not easy to obtain experimentally the structural information concerning metalloid atoms in metal-metalloid amorphous alloys even by neutron diffraction as well as conventional X-ray diffraction, because the scattering factor and atomic fraction of metalloid atoms are usually small in these alloys.

Recently, we<sup>1,2)</sup> have observed well resolved RDF's of amorphous Pd-Si alloys which show distinct separation between two first-neighbor peaks associated with Pd-Si and Pd-Pd correlations respectively, using a time-of-flight pulsed neutron total scattering technique based on electron LINAC. Unfortunately, no reliable information of Si-Si correlation could be found out with that experiment.

In this paper, three partial structure functions related to Pd-Pd , Pd-Si and Si-Si correlations in amorphous  $Pd_{80}Si_{20}$  alloy are sepa-

<sup>\*</sup> The 1708th report of the Research Institute for Iron, Steel and Other Metals.

rated using combination of X-ray, electron and neutron diffraction experiments and then number-concentration correlation functions are derived.

### II. Experimental

Amorphous Pd-19.8at%Si alloy was prepared in a form of thin ribbon with about 2 mm in width and less than 30  $\mu$ m in thickness by splat quenching from the molten state using a single roll technique. Silicon content in the alloy was determined by conventional chemical analysis. Three total interference functions of the alloy were measured by neutron, X-ray and electron diffraction respectively. Neutron total interference function was obtained by pulsed neutron total scattering experiment<sup>3)</sup>. X-ray<sup>4)</sup> and electron<sup>5)</sup> total interference functions were measured by conventional methods. Details of these measurements have been described elsewhere  $^{3-5)}$ . Three total interference functions observed for amorphous Pd-19.8at%Si alloy are shown in Figure 1.

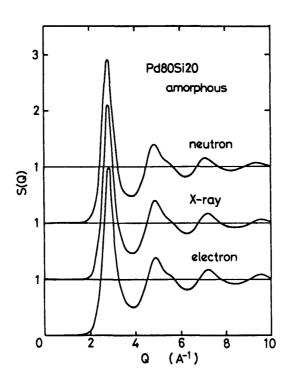


Fig. 1. Total interference functions of amorphous Pd-19.8 at%Si alloy observed by neutron, X-ray and electron diffraction.

### III. Partial pair Structure Functions

The measured total interference function S(Q) for an amorphous A-B binary alloy can be written as a weighted sum of partial pair interference functions  $S_{AA}(Q)$ ,  $S_{AB}(Q)$  and  $S_{BB}(Q)$  corresponding to A-A, A-B and B-B correlations in the alloy as follows

$$S(Q) = W_{AA}S_{AA}(Q) + 2W_{AB}S_{AB}(Q) + W_{BB}S_{BB}(Q) , \qquad (1)$$

where

$$W_{AA} = c_A^2 f_A^2 / (c_A^2 f_A + c_B^2 f_B^2)^2,$$

$$W_{AB} = c_A^2 c_B^2 f_A^2 f_B / (c_A^2 f_A + c_B^2 f_B^2)^2,$$

$$W_{BB} = c_B^2 f_B^2 / (c_A^2 f_A + c_B^2 f_B^2)^2,$$
(2)

and

 $c_A$  and  $c_B$  are the concentration fractions of A and B atoms in the alloy, and  $f_A$  and  $f_B$  are the atomic scattering factors of A and B atoms respectively.

Since three total interference functions S(Q)'s of amorphous Pd-19.8at%Si alloy observed by neutron, X-ray and electron diffraction experiments have different weighting factors  $W_{\rm PdPd}$ ,  $W_{\rm PdSi}$  and  $W_{\rm SiSi}$  one another, three partial pair interference functions  $S_{\rm PdPd}(Q)$ ,  $S_{\rm PdSi}(Q)$  and  $S_{\rm SiSi}(Q)$  can be solved from the simultaneous linear equation

$$\begin{pmatrix}
s^{N}(Q) \\
s^{X}(Q) \\
s^{E}(Q)
\end{pmatrix} = \begin{pmatrix}
w_{PdPd}^{N} & 2w_{PdSi}^{N} & w_{SiSi}^{N} \\
w_{PdPd}^{X} & 2w_{PdSi}^{X} & w_{SiSi}^{X} \\
w_{PdPd}^{E} & 2w_{PdSi}^{E} & w_{SiSi}^{E}
\end{pmatrix} \begin{pmatrix}
s_{PdPd}(Q) \\
s_{PdSi}(Q) \\
s_{SiSi}(Q)
\end{pmatrix}, (3)$$

where the superscripts N, X and E mean neutron, X-ray and electron diffraction respectively. The weighting factors  $W_{ij}$  (i,j=Pd or Si) in neutron, X-ray and electron total interference functions for amorphous Pd-19.8at%Si alloy are listed in Table 1.

Table 1. Weighting factors of amorphous Pd-19.8at%Si alloy in neutron, X-ray and electron diffraction experiments.

	W <sub>PdPd</sub>	<sup>2W</sup> PdSi	W <sub>SiSi</sub>	
neutron	0.730	0.249	0.021	
X-ray	0.895	0.101	0.003	
electron	0.835	0.158	0.007	

It is difficult to determine three partial pair interference functions by solving directly equation (3), because values of the weighting factor of Si-Si correlation for X-ray and electron diffraction are very small compared with those of Pd-Pd and Pd-Si correlations.

In this work, therefore, three partial pair interference functions were derived with two steps iterative procedure. First, trial  $S_{\mbox{PdPd}}(Q)$  and  $S_{\mbox{PdSi}}(Q)$  are calculated from combination of X-ray and electron total interference functions

$$S^{X}(Q) = W_{PdPd}^{X} S_{PdPd}^{X}(Q) + 2W_{PdSi}^{X} S_{PdSi}^{X}(Q) + W_{SiSi}^{X} S_{SiSi}^{X}(Q)$$

$$S^{E}(Q) = W_{PdPd}^{E} S_{PdPd}^{E}(Q) + 2W_{PdSi}^{E} S_{PdSi}^{E}(Q) + W_{SiSi}^{E} S_{SiSi}^{E}(Q),$$
(4)

where an initial requirement of  $w_{SiSi}^X = w_{SiSi}^E = 0$  is used. Second,  $s_{SiSi}(Q)$  is obtained using trial partial interference functions  $s_{PdPd}(Q)$  and  $s_{PdSi}(Q)$  and neutron total interference function as follows

$$s_{\text{SiSi}}(Q) = \frac{s^{N}(Q) - w_{\text{PdPd}}^{N} s_{\text{PdPd}}(Q) - w_{\text{PdSi}}^{N} s_{\text{PdSi}}(Q)}{w_{\text{SiSi}}^{N}}$$
(5)

The  $S_{SiSi}(Q)$  obtained is substituted into equation (4) and then  $S_{PdPd}(Q)$  and  $S_{PdSi}(Q)$  are again calculated. Such an iterative calculation is continued until three partial pair interference functions  $S_{PdPd}(Q)$ ,  $S_{PdSi}(Q)$  and  $S_{SiSi}(Q)$  are converged to reasonable solutions within experimental errors. Finally, the physical reliability of obtained partial pair interference functions was examined by the following criteria used by Enderby et al. and Edwards et al.

- (1) Solutions are found in the range defined by the dashed line in Figure 2.
- (2) A smooth curve is drawn such that the sum rule is satisfied.

$$\int (S_{\text{pdPd}}(Q) - 1) Q^{2} dQ = \int (S_{\text{pdSi}}(Q) - 1) Q^{2} dQ$$

$$= \int (S_{\text{SiSi}}(Q) - 1) Q^{2} dQ = -2\pi^{2} \rho .$$
(6)

Figure 3 shows final pair interference functions  $S_{pdPd}(Q)$ ,  $S_{pdSi}(Q)$  and  $S_{SiSi}(Q)$  for amorphous Pd-19.8at%Si alloy. Error bands in  $S_{pdPd}(Q)$ ,  $S_{pdSi}(Q)$  and  $S_{SiSi}(Q)$  result from the statistical

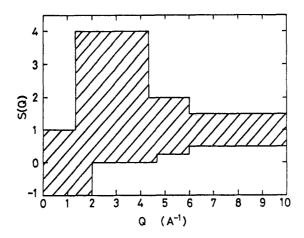


Fig. 2. A region allowed empirically for interference function of amorphous Pd-19.8at%Si alloy.

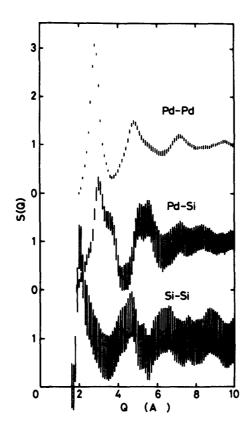


Fig. 3. Partial pair interference functions for Pd-Pd,
 Pd-Si and Si-Si correlations. Error bands originate
 from 2 % uncertainties in experimental total
 interference functions.

uncertainty in experimental total interference functions which is less than 2 % in each experiment.

The partial pair distribution functions  $g_{ij}(r)$  (i,j=Pd or Si) are defined as Fourier transforms of  $S_{ij}(Q)$  (i,j=Pd or Si)

$$g_{ij}(r) = 1 + \int_{0}^{\infty} \frac{1}{2\pi^{2} \rho r} Q(S_{ij}(Q) - 1) \sin Qr \ dQ$$
 (7)

Figure 4 shows  $g_{pdPd}(r)$ ,  $g_{pdSi}(r)$  and  $g_{SiSi}(r)$  of amorphous Pd-19.8at%Si alloy, where the integration of equation (7) is truncated at  $Q_{max}=10$  A<sup>-1</sup> in case of Pd-Pd correlation and at  $Q_{max}=7$  A<sup>-1</sup> in case of Pd-Si and Si-Si correlations because of increasing uncertainty included in  $S_{ij}(Q)$  at higher Q region.

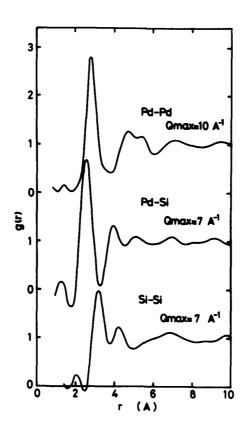


Fig. 4. Partial pair distribution functions for Pd-Pd, Pd-Si and Si-Si correlations.

# IV. Number-Concentration Structure Functions

Bhatia and Thornton $^{8)}$  have shown that total interference function S(Q) is divided into another set of the three partial interference

functions which correspond to number-number  $S_{NN}^{}(Q)$ , number-concentration  $S_{NC}^{}(Q)$  and concentration-concentration  $S_{CC}^{}(Q)$  correlations

$$S(Q) = \frac{1}{\langle f^2 \rangle} \{\langle f \rangle^2 S_{NN}(Q) + 2\Delta f \langle f \rangle S_{NC}(Q) + \langle \Delta f \rangle^2 S_{CC}(Q) \}, \quad (8)$$

where  $<f>=c_A^f_A+c_B^f_B$  and  $\Delta f=|f_A-f_B^f|$ .

There number-concentration partial interference functions  $S_{NN}\left(Q\right)$ ,  $S_{NC}\left(Q\right)$  and  $S_{CC}\left(Q\right)$  are written as a linear combination of partial pair interference functions as follows

$$S_{NN}(Q) = c_{A}^{2} S_{AA}(Q) + c_{B}^{2} S_{BB}(Q) + 2c_{A} c_{B} S_{AB}(Q) ,$$

$$S_{CC}(Q) = c_{A}^{2} c_{B}^{2} \{ S_{AA}(Q) + S_{BB}(Q) - 2S_{AB}(Q) \} + c_{A} c_{B} ,$$

$$S_{NC}(Q) = c_{A} c_{B} \{ c_{A} S_{AA}(Q) - c_{B} S_{BB}(Q) + (c_{B} - c_{A}) S_{AB}(Q) \} .$$

$$(9)$$

The same relationships are maintained between  $\{g_{NN}(r), g_{NC}(r), g_{CC}(r)\}$  and  $\{g_{AA}(r), g_{AB}(r), g_{BB}(r)\}$ . Figures 5 and 6 show the results of number-concentration structure functions obtained for amorphous Pd-19.8at%Si alloy.

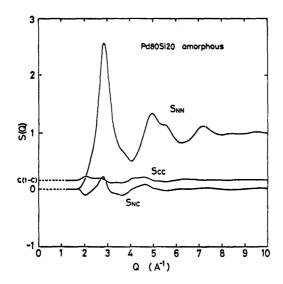


Fig. 5. Number-concentration interference functions for amorphous Pd-19.8at%Si alloy.

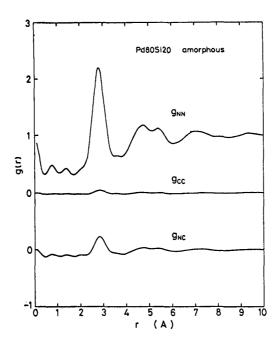


Fig. 6. Number-concentration pair distribution functions for amorphous Pd-19.8at%Si alloy.

## V. Results and Discussion

The overall behavior of  $S_{\mathrm{PdPd}}(Q)$  in Figure 3 is quite analogous to that of each experimental total S(Q). This confirms that the total S(Q) of amorphous Pd-19.8at%Si alloy is predominantly dominated by Pd-Pd correlation even in neutron diffraction.

The  $S_{\mathrm{PdSi}}(Q)$  has a shoulder at high Q side of the first peak. Such a shoulder has been found in  $S_{\mathrm{FeP}}(Q)$  of Fe-P and  $S_{\mathrm{NiP}}(Q)$  of Ni-P amorphous alloys obtained by Waseda et al. 10 using anomalous X-ray scattering technique. However, the first peak of  $S_{\mathrm{CoP}}(Q)$  of amorphous Co-P alloy reported by Sadoc and Dixmier 10 does not show a well-defined shoulder but an asymmetric profile extended to high Q region. These characteristic behaviors of partial metal-metalloid pair interference functions are definitely originated from the well-defined local coordination of metal atoms surrounding a metalloid atom.

The first peak of  $S_{\text{SiSi}}(Q)$  appears at low Q region compared with those of  $S_{\text{pdPd}}(Q)$  and  $S_{\text{pdSi}}(Q)$ . This result is in good agreement with the behavior of  $S_{\text{pp}}(Q)$  in amorphous Ni-P and Fe-P alloys observed by Waseda et al. and in amorphous Co-P alloy by Sadoc and Dixmier 11).

As shown in Figure 4, the characteristic splitting of the second

peak remains in  $g_{pdPd}(r)$ . The first-neighbor position for Pd-Pd correlation is 2.85 A which is a little larger than the value(2.80 A) observed by high resolution neutron total pair distribution function of amorphous Pd-Si alloy  $^{1,2)}$ . The first peak of  $g_{\mathrm{PdSi}}(r)$  is located at 2.60 A. This distance is fairly larger than the value(2.41 A) in high resolution neutron total pair distribution function 1,2). These discrepancies are usually caused due to the truncation effect of Fourier transformation. Misawa and Suzuki<sup>12)</sup> have demonstrated that the position of narrow first peak in pair distribution function converges gradually with increasing truncation value  $\mathbf{Q}_{\text{max}}$  in interference function. Therefore,  $\mathbf{Q}_{\text{max}}$  values adopted in this work are too small to obtain exact values of the first peak positions particularly for narrow Pd-Si correlation. The first peak position in  $g_{SiSi}(r)$  is about 3.2 A and the second peak is located closely to the first peak. The oscillation of subsequent peaks in  $g_{SiSi}(r)$  is broader than those in  $g_{pdPd}(r)$  and  $g_{pdSi}(r)$ . It can be said that Si atoms never be at the nearest neighbor position in amorphous Pd-Si alloy.

As shown in Figures 5 and 6, both of  $S_{NN}(Q)$  and  $g_{NN}(r)$  vary in a manner similar with  $S_{pdpd}(Q)$  and  $g_{pdpd}(r)$ . This can be understood from equation (9) because of predominant content of Pd atoms with large scattering factor in amorphous Pd-19.8at%Si alloy. When the concentration fluctuation in A-B binary alloy is uncorrelated, i.e.  $S_{AA}(Q) = S_{AB}(Q) = S_{BB}(Q)$ , the relations of  $S_{CC}(Q) = c_{A}c_{B}$  and  $g_{CC}(r) = 0$  are realized. On the other hand,  $S_{CC}(Q) < c_{A}c_{B}$  and  $g_{CC}(r) < 0$  if unlike-atom A-B correlation predominantly exists, and  $S_{CC}(Q) > c_{A}c_{B}$  and  $g_{CC}(r) > 0$  if like-atom A-A and B-B correlations are preferred. Although the accuracy of partial structure functions obtained in this work is much limited, the spatial fluctuation and segregation of constituent atoms are concluded to be very small in amorphous Pd-19.8at%Si alloy.

### Acknowledgements

This work is partly supported by the Mitsubishi Foundation. One of the authors (T.F.) would like to thank the Japan Society for the Promotin of Science for the Fellowship to work at Tohoku university.

# References

(1) K. Suzuki, T. Fukunaga, M. Misawa and T. Masumoto, Mater. Sci. Eng., 23 (1976), 215.

- (2) T. Fukunaga, M. Misawa, K. Fukamichi, T. Masumoto and K. Suzuki, Proc. Third Int. Conf. Rapidly Quenched Metals III, 2 (1978), 325.
- (3) K. Suzuki, M. Misawa, K. Kai and N. Watanabe, Nucl. Inst. and Meth., <u>147</u> (1977), 519.
- (4) T. Fukunaga, Doctor Thesis, Tohoku University 1979.
- (5) S. Fujime, Japan J. Appl. Phys., 5 (1966), 764.
- (6) J.E. Enderby, D.M. North and P.A. Egelstaff, Phil. Mag., <u>14</u> (1966), 961.
- (7) F.G. Edwards, J.E. Enderby, R.A. Howe and D.I. Page, J. Phys. C: Solid Stat. Phys., <u>8</u> (1975), 3483.
- (8) A.B. Bhatia and D.E. Thornton, Phys. Rev., B2 (1970), 3004.
- (9) Y. Waseda, H. Okazaki and T. Masumoto, Sci. Rep. RITU, <u>A26</u> (1977), 202.
- (10) Y. Waseda and S. Tamaki, Z. Physik, B23 (1976), 315.
- (11) J.F. Sadoc and J. Dixmier, Mater. Sci. Eng., 23 (1976), 187.
- (12) K. Suzuki and M. Misawa, Proc. of Studies on Metallic Melts in Metallurgical Reactions held at Iizaka, Nov. 30 -Dec. 2, 1978 (edited by M. Shimoji) p.111.

A part of this work has been reported in the Topical Conference on Atomic Scale Structure of Amorphous Solids held at IBM Research Center, Yorktown Heights, April 3-5, 1978. [Bull. Amer. Phys. Soc., 23 (1978), 467.]