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Observation of large magnetoresistance of magnetic Heusler alloy $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ in high magnetic fields

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The magnetic and electrical properties on magnetic Heusler alloy $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ were studied in magnetic fields up to 18 T in 4.2–270 K temperature range. It was found that at the vicinity of 160 K the resistivity jump of 46% is accompanied by the magnetic phase transition. Furthermore, the large magnetoresistance effect of 50% by the magnetic field induced magnetic phase transition was observed. © 2006 American Institute of Physics. [DOI: 10.1063/1.2374868]

Recently, it has been found that ferromagnetic Heusler alloys $\text{Ni}_{50}\text{Mn}_{50-y}\text{X}_y$ ($X=\text{In}$, Sn , and Sb) with the cubic $L2_1$ -type ($L2_1$) structure show martensitic transformation below the Curie temperature T_C .¹ The result of neutron diffraction measurements for $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ shows that the martensite phase has an orthorhombic four-layered ($4O$) structure with space group of $Pmma$.² In addition, the magnetization (σ_O) in the $4O$ phase is smaller than that (σ_L) in the $L2_1$ phase.³⁻⁷ These results indicate that the $\text{Ni}_{50}\text{Mn}_{50-y}\text{X}_y$ alloy will exhibit field-induced magnetic and structural transitions such as those of the Ni_2MnGa system,⁸⁻¹⁰ which is called “ferromagnetic shape-memory alloys.” Especially, results on these Heusler alloys Ni-Mn-X attracted interest from the point of view of high performance magnetic materials controlled by magnetic fields.⁵⁻⁷

In the previous paper, we reported the magnetic field-induced reverse martensitic transformation from the $4O$ to the $L2_1$ structure, accompanied by the magnetic transition from the σ_O to the σ_L phase in $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$.¹¹ Furthermore, the result shows that high magnetic fields over 5 T are required to completely lead the field-induced reverse transformation in this compound. In this study, the magnetization and electrical resistivity measurements for Heusler alloy $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ were carried out in magnetic fields up to 18 T, in order to investigate the magnetoresistance effect.

Polycrystalline $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ compound has been prepared by induction melting under an argon atmosphere. The ingot was cut into a small pillar with a size of $1.08 \times 1.76 \times 2.50 \text{ mm}^3$. The pillar sample was confirmed to be a single phase with the $L2_1$ structure by x-ray powder diffraction measurements at room temperature. The magnetization σ was measured by an extraction-type magnetometer in magnetic fields B up to 18 T using a superconducting magnet. The electrical resistivity ρ was measured by a standard four-

probe technique in magnetic fields up to 17 T.

Figure 1 shows the temperature dependence of the magnetization (σ - T) at 1 mT (a) and 17 T (b). The magnetic phase transition is seen at the vicinity of 160 K for 1 mT and 120 K for 17 T with a large hysteresis over 50 K. From the data for 1 mT, the Curie temperature T_C is determined to be 325 K, and the martensitic transformation starting temperature M_s , the martensitic transformation finishing temperature M_f , the reverse transformation starting temperature A_s , and the reverse transformation finishing temperature A_f are determined to be 171, 125, 150, and 195 K, respectively. The σ - T behavior and T_C are consistent with previous results,^{1,2,11} although other characteristic temperatures are low by about 50 K. In this study, we selected the present sample having

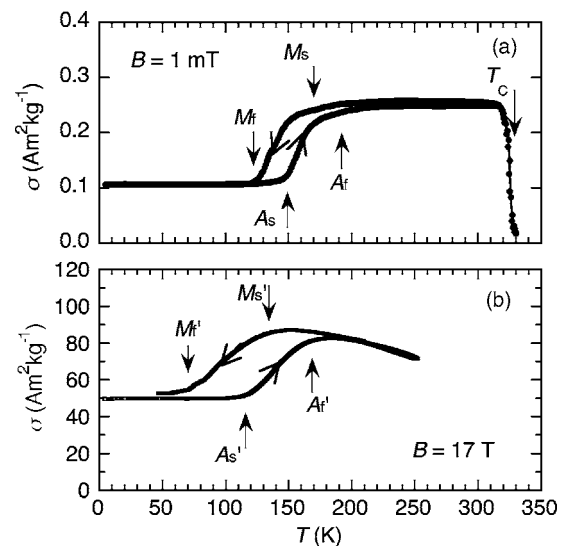


FIG. 1. Temperature dependence of the magnetization of $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ at 1 mT (a) and 17 T (b). The vertical arrows indicate the Curie temperature T_C and the characteristic temperatures of the martensitic transformation at 1 mT and 17 T. The measurements were carried out for the heating and cooling processes and the arrows indicate the thermal hysteresis.

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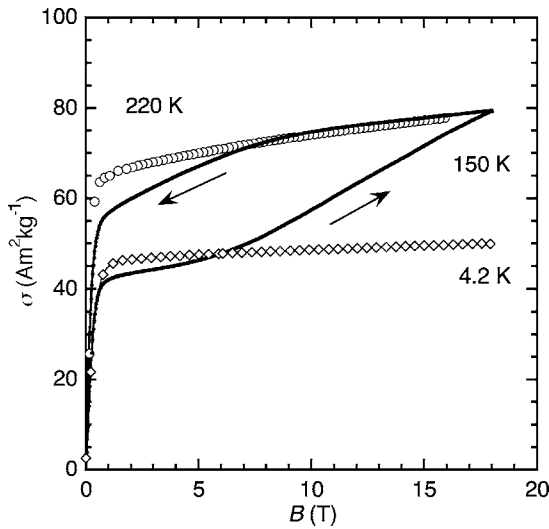


FIG. 2. High field magnetization curves of $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ at 4.2 K (open diamonds), 150 K (solid lines), and 220 K (open circles). The magnetization curves at 150 K were measured after zero-field heating from 4.2 K. The arrows indicate the magnetization process with increasing and decreasing magnetic fields B .

low M_s , M_f , A_s , and A_f , which is slightly different from the sample treatment reported in the previous study,¹¹ because it was difficult to measure the properties in high magnetic fields over 250 K. By applying 17 T, the characteristic temperatures decrease ($M'_s=135$ K, $M'_f=75$ K, $A'_s=110$ K, and $A'_f=166$ K) and the thermal hysteresis extends. The results obtained show that σ_O in the $4O$ phase is smaller than σ_L in the $L2_1$ phase in fields up to 17 T.

Figure 2 shows the magnetization (σ - B) curves at 4.2, 150, and 220 K in magnetic fields up to 18 T. The σ - B curve at 220 K ($L2_1$ phase) shows a ferromagnetic behavior and σ is $77.8 \text{ A m}^2 \text{ kg}^{-1}$ at 16 T. On the other hand, σ at 4.2 K ($4O$ phase) is $49.9 \text{ A m}^2 \text{ kg}^{-1}$ at 18 T, which is 40% smaller than that at 220 K. At $A_s < T < A_f$, a magnetic phase transition with large magnetic hysteresis is observed on the σ - B curves. The σ - B curve at 150 K is shown in this figure as a typical result, which was measured after zero-field heating from 4.2 K. This σ - B curve of 150 K indicates that the magnetic phase is lower σ_O phase in $B < 5$ T, and it transforms into higher σ_L phase in $5 \leq B \leq 18$ T. At 18 T, σ reaches up to $79.2 \text{ A m}^2 \text{ kg}^{-1}$, which is almost the same value at 220 K. That is, magnetic field induces the reverse martensitic transformation from the $4O$ to the $L2_1$ structures accompanied by the magnetic phase transition from lower σ_O to higher σ_L .

Figure 3 shows the temperature dependence of the electrical resistivity (ρ - T) at $B=0$ and 17 T. In these measurements, we observed the thermal hysteresis at the vicinity of 160 K for 1 mT and 120 K for 17 T, which is consistent with the magnetic phase transition, as shown in Fig. 1. The thermal variation of the ρ - T curve is very small below 100 K ($[\rho(100 \text{ K}) - \rho(4.2 \text{ K})] / \rho(4.2 \text{ K}) = 1\%$), but it abruptly changes by 46% ($=[\rho(100 \text{ K}) - \rho(160 \text{ K})] / \rho(100 \text{ K})$) at the vicinity of 160 K, and then it increases with increasing T .

Figure 4 shows the magnetoresistance (ρ - T) in magnetic fields up to 17 T at 4.2, 150, and 220 K. In $B=0$ T, ρ at 4.2 ($4O$ phase) and 220 K ($L2_1$ phase) are 320 and $180 \mu\Omega \text{ cm}$, respectively, and they decrease linearly with increasing B . On the other hand, we observed the large negative magnetoresistance effect in $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ at the vicinity of 160 K.

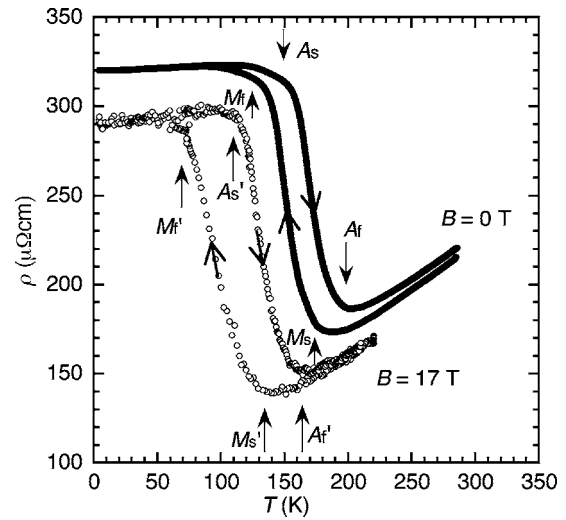


FIG. 3. Temperature dependence of the electrical resistivity of $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ at 0 T (solid lines) and 17 T (open circles). The vertical arrows indicate the characteristic temperatures of the martensitic transformation at 0 and 17 T. The measurements were carried out for the heating and cooling processes and the arrows indicate the thermal hysteresis.

The data at 150 K are shown in this figure as a typical result. On the initial (first) ρ - B process, ρ is $307 \mu\Omega \text{ cm}$ at $B=0$ T and decreases gradually in fields up to 5 T. Then, ρ decreases rapidly with increasing B , and it reaches down to $154 \mu\Omega \text{ cm}$ at 17 T. The change of the negative magnetoresistance is about 50% in magnetic fields up to 17 T.

The ρ - B curve for first decreasing B process is not traced on the initial one, but it increases linearly with decreasing B to 8.3 T. Subsequently, ρ increases rapidly below 8.5 T, and it reaches up to $259 \mu\Omega \text{ cm}$. This first ρ - B loop is consistent with the σ - B hysteresis loop, as shown in Fig. 2. That is, the large magnetoresistance is due to the reverse martensitic transformation induced by the field-induced magnetic phase transition. Just after this measurement, we remeasured the ρ - B curves (second) in $B < 12$ T at the same temperature,

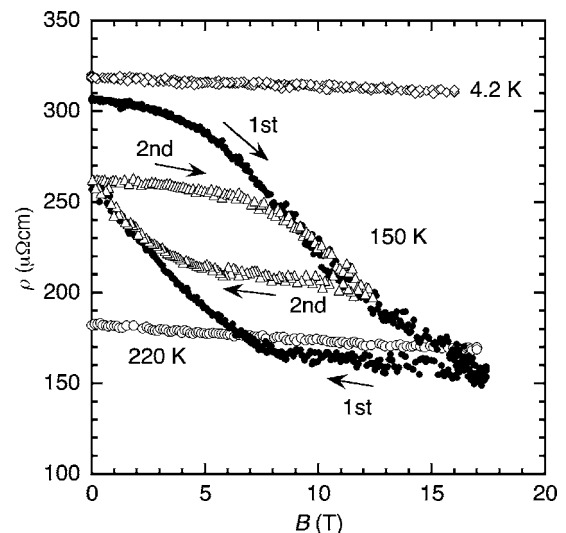


FIG. 4. Magnetoresistance of $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ at 4.2 K (open diamonds), 150 K (solid circles and open triangles), and 220 K (open circles). The solid circle and open triangle show the data for the initial (first) and second (second) scans, respectively. The data at 150 K were measured after zero-field heating from 4.2 K. The arrows indicate the magnetization process with increasing and decreasing magnetic fields B .

which shows the reversible ρ - B process for applying B (open triangle in Fig. 4). In addition, ρ of the second measurement is traced on that of the first measurement in $B > 8.3$ T for the field increasing process and $B < 2.3$ T for the field decreasing process.

In this study, we found that the electrical property as well as the magnetic one of $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ is very unique. Especially, ρ changes drastically at the vicinity of 150 K and is almost constant below 100 K. The properties of $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ are probably due to the drastic change of the magnetic and electronic structures, accompanied with the martensitic transformation. By the neutron diffraction study for $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$, Brown *et al.* suggested that the suppression of σ is due to some antiparallel alignment of moment in the $4O$ phase.² Our result of thermoelectric power measurement indicates that the density of state at the vicinity of the Fermi level modifies drastically, accompanied by the martensitic transformation in $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$.¹² Therefore, in order to understand the basic properties of $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$, it is required to clarify the magnetic and electronic structures.

In summary, the magnetic and electrical resistivity measurements for $\text{Ni}_{50}\text{Mn}_{36}\text{Sn}_{14}$ were carried out in magnetic fields up to 18 T. We found that the alloy exhibits a large-negative magnetoresistance effect of 50%, accompanied by the magnetic field-induced reverse transformation at the vicinity of the martensitic transformation temperature.

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