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Optical and Electronic Properties of Femtosecond Laser-Induced Sulfur-Hyperdoped Silicon N+/P Photodiodes

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Abstract

Impurity-mediated near-infrared (NIR) photoresponse in silicon is of great interest for photovoltaics and photodetectors. In this paper, we have fabricated a series of n⁺/p photodetectors with hyperdoped silicon prepared by ion-implantation and femtosecond pulsed laser. These devices showed a remarkable enhancement on absorption and photoresponse at NIR wavelengths. The device fabricated with implantation dose of 10¹⁴ ions/cm² has exhibited the best performance. The proposed method offers an approach to fabricate low-cost broadband silicon-based photodetectors.

Keywords: Ion-implantation, Hyperdoped silicon, NIR photoresponse

Background

Traditional silicon-based devices could not show desirable NIR photoresponse due to limitation of optical bandgap (1.12 eV) of silicon [1], and many attempts have been made to enhance the absorptance of silicon material, especially at NIR wavelengths [2–9]. The discovery of chalcogen-supersaturated silicon fabricated by laser irradiation in SF₆ atmosphere demonstrated an approach to enhance the sub-bandgap absorption [10, 11]. In this process, the material can be doped beyond the solubility limit [12]. Besides, light trapping effect caused by the unique pointed cone structure on silicon surface also increases the efficiency of light absorption [13]. In this paper, we have fabricated hyperdoped silicon prepared by ion-implantation and femtosecond pulsed laser. Hall measurement was carried out to measure the electrical properties of hyperdoped silicon. Photodetectors based on n⁺/p junction demonstrated high performances on both NIR absorption and photoresponse.

Methods

One-side polished p-type silicon [100] wafers (300 μm) with resistivity 8–12 Ω cm were ion-implanted with 1.2 keV ³²S⁺ into a depth of approximately 40 nm at room temperature. The implantation doses were 1 × 10¹⁴, 1 × 10¹⁵, and 1 × 10¹⁶ ions/cm². Pulsed laser melting (PLM) was carried out by 1 kHz train of 100 fs, 800 nm femtosecond laser pulses with a fluence of 0.5 J/cm². Then, laser spot of 200 μm diameter is focused on the silicon and patterned square areas up to 10 mm × 10 mm. Rapid thermal annealing (RTA) was implemented at 600 °C for 30 min in a N₂ atmosphere.

We determined the absorptance (*A*) of the samples by measuring reflectance (*R*) and transmittance (*T*) by using a UV-Vis-NIR spectrophotometer (UV3600, Shimadzu, Tokyo, Japan) equipped with an integrating sphere detector [3]. The absorptance was calculated by $A = 1 - R - T$. The concentration and mobility of carriers were measured by Hall Effect measurement system at room temperature (via van der Pauw technique) [14]. To investigate whether the impurity/intermediate band (IB) formed by sulfur impurities in silicon enhances the sub-bandgap photoresponse, we employed a Fourier-transform photocurrent spectroscopy method as described in Ref. [15, 16], where the chopped FTIR global light source is focused onto the sample, and the

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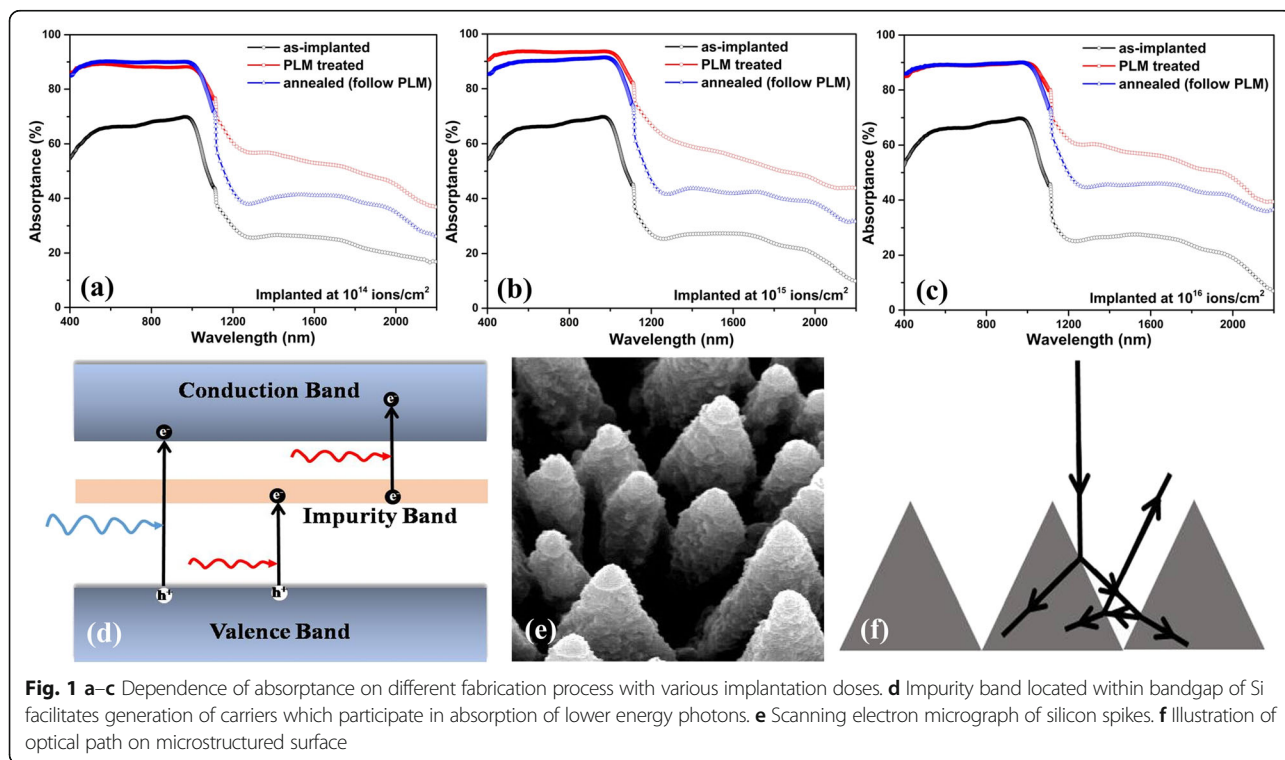
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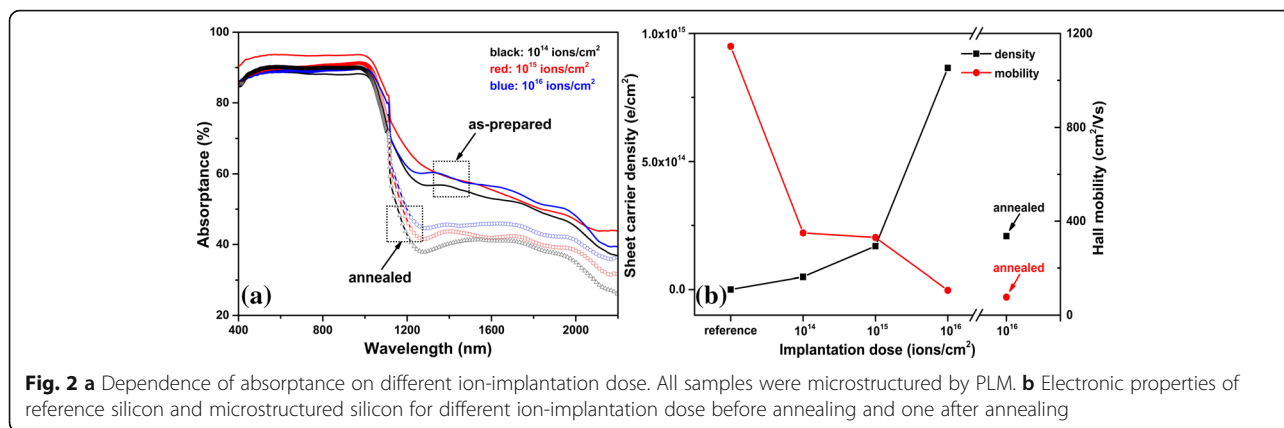
generated photocurrent is then demodulated by an external lock-in amplifier and finally fed back to the external port of the FTIR.

Results and Discussion

Figure 1 shows the absorbance of silicon samples implanted at different doses. The samples processed with PLM showed highest absorbance at visible and NIR wavelengths while as-implanted samples showed lowest absorbance. However, the annealing process reduces the absorption in NIR region of spectra. The high Vis-NIR absorbance of microstructured silicon is ascribed to the following reasons: hyperdoping-induced impurity band and microstructured surface-generated light trapping effect. As illustrated in Fig. 1d, an impurity band induced by dopants is formed in silicon, which is responsible for sub-bandgap absorption [17]. Consequently, the hyperdoped silicon shows high absorbance in NIR range. Meanwhile, laser melting reconstructs the silicon surface and produces an array of cones that leads to multiple reflection and absorption [13], as displayed in Fig. 1e, f. The processed annealing evidently reduces absorbance at NIR wavelengths range, which mainly caused by the two aspects: (1) annihilate the nanostructures on the silicon surface, decreasing the light trapping effect [18]; and (2) result in the bond rearrangement within silicon matrix, which optically inactivate sulfur impurities [11].

Because of the similar surface structure created by same laser parameters, the intensity of absorption in NIR range mainly depends on the dopant's impurity levels [19]. In the past, we have illustrated the possible S-related energy levels corresponding to the photo-response spectral features [20]. It showed the large enhancement observed at NIR region dependently resulted from the S-related energy level (~ 614 meV), which greatly enhanced the sub-bandgap absorbance. Prior to annealing process, absorption has no dramatic change with respect to the doping dose as shown in Fig. 2a. The microstructured silicon with 10^{16} and 10^{15} ion/cm² implantation dose show similar absorbance, and the sample implanted at 10^{14} ions/cm² shows unnoticeable decrease. We consider the lower absorbance for annealed samples in NIR range can be ascribed to the two aspects. M. A. Sheehy et al. [21] proposed the absorption decrease of below bandgap after annealing process is attributed to the diffusion out of the crystalline grains to the grain boundaries of the supersaturated dopants and defects. These defects include vacancies, dangling bonds, and floating bonds. Once the defects diffuse to the grain boundaries, they would no longer make a contribution to impurity bands in the Si, thus reducing the absorption of below bandgap radiation. Moreover, the literature [22] reported that no remarkable redistribution of S occurred until the annealing temperature reached at 650 °C. During this process, the





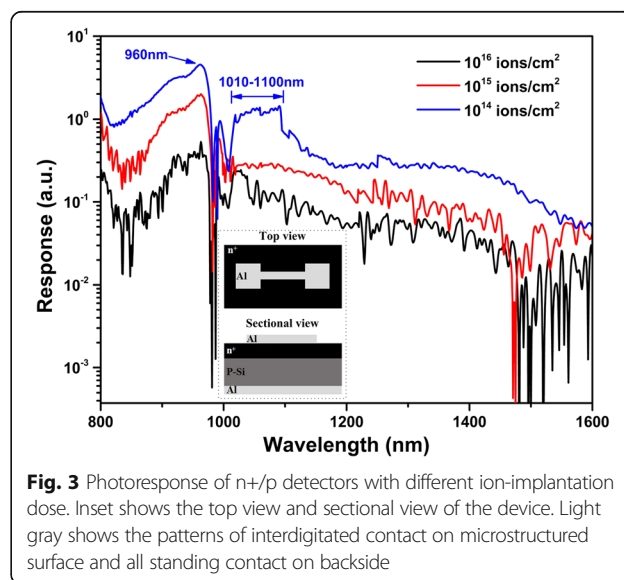
S appears to complex with defect clusters, which means the S atoms will combine with each other at the Si wafer surface. This phenomenon leads to a reduction of the active doping concentration.

The carrier density and mobility of microstructured silicon with different ion-implantation doses are shown in Fig. 2b. It is evident that sheet density increases with ion-implantation dose, and mobility decreases with increasing ion-implantation dose. According to Shockley-Read-Hall (SRH) recombination effect, in an indirect bandgap semiconductor such as Si and Ge, the carrier lifetime decreases with the increase of dopant concentration [23, 24]. The decrease of mobility leads to an increase of recombination probability, so the decrease of mobility results in a decrease on electron lifetime and the decrease on mobility with increasing doping dose is consistent with SRH recombination effect. After annealing, the sheet carrier density decreases dramatically due to thermal diffusion effect as we discussed previously.

Figure 3 shows the photoresponse with different doping dose, and the inset shows the diagram of n+/p photo-detector. The photoresponse at NIR range indicates the appearance of impurity-mediated band. The prominent peak at approximately 960 nm corresponds to the generation of electron-hole pairs in silicon substrate, which are separated by the built-in potential of n+/p junction and collected at the top and bottom Al contacts. This phenomenon is well known as the heterojunction theory in Si devices [25].

The observed photoresponse in NIR is ascribed to the sulfur impurity levels in hyperdoped silicon. Such impurity levels facilitate the below bandgap absorption as mentioned above. The absorbed NIR light is converted into electron-hole pairs, resulting in the enhancement of photoresponse in NIR range (1100 ~ 1600 nm) [20]. The device with implantation dose of 10^{14} ions/cm² shows the highest photoresponse in the wavelength range of 1010–1100 nm. The broad peak has been investigated owing to deep sulfur levels in femtosecond laser-processed silicon [20, 26]. In addition, we found that

the device with 10^{14} ions/cm² has showed higher photoresponse than those with 10^{15} and 10^{16} ions/cm². And the Hall measurement indicated that the sample implanted at 10^{14} ions/cm² had a bulk concentration of 10^{19} ions/cm³. As demonstrated by SRH recombination effect, carrier lifetime depends on dopant concentration in silicon. E. Mazur has concluded that the sample with 10^{19} ions/cm³ dopant concentration was expected to show longer carrier lifetime than 10^{20} and 10^{21} ions/cm³ [23]. Our Hall measurement results, sample implanted at 10^{14} ions/cm² shows the highest mobility, are in agreement with the conclusion. Based on this theory, although a sample with higher doping dose shows greater absorbance, there is still a balance between optical absorption and carrier mobility. As presented in Fig. 3, the device with 10^{14} ions/cm² is most probable to show the highest photoresponse, which is consistent with the conclusion reported in Ref. [23].



Conclusions

We have measured the response of photodetectors based on microstructured silicon with different ion-implantation dose. The incorporation of impurities leads to a remarkable enhancement on absorbance and photoresponse at NIR wavelengths. And device implanted at 10^{14} ions/cm² exhibits the highest photoresponse. PLM combined with ion-implantation demonstrates a considerable technique for the fabrication of NIR detectors. This technique may offer a feasible approach to fabricate low-cost broadband silicon-based photodetectors.

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Authors' Contributions

TZ, BL, and YX carried out the experiments and analyzed the data. XY and ZL participated in the work to measure the responsivity (a.u.). WA analyzed the theory and polished the English. SL and ZC gave equipment support. All authors read and approved the final manuscript.

Competing Interests

The authors declare that they have no competing interests.

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