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Extending the environmental lifetime of unpackaged perovskite solar cells through interfacial design†

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Solution-processed oxo-functionalized graphene (oxo-G₁) is employed to substitute hydrophilic PEDOT:PSS as an anode interfacial layer for perovskite solar cells. The resulting devices exhibit a reasonably high power conversion efficiency (PCE) of 15.2% in the planar inverted architecture with oxo-G₁ as a hole transporting material (HTM), and most importantly, deploy the full open-circuit voltage (V_{oc}) of up to 1.1 V. Moreover, oxo-G₁ effectively slows down the ingress of water vapor into the device stack resulting in significantly enhanced environmental stability of unpackaged cells under illumination with 80% of the initial PCE being reached after 500 h. Without encapsulation, ~60% of the initial PCE is retained after ~1000 h of light soaking under 0.5 sun and ambient conditions maintaining the temperature beneath 30 °C. Moreover, the unsealed perovskite device retains 92% of its initial PCE after about 1900 h under ambient conditions and in the dark. Our results underpin that controlling water diffusion into perovskite cells through advanced interface engineering is a crucial step towards prolonged environmental stability.

Perovskite solar cells have attracted considerable research interest due to their remarkable PCEs exceeding the 20%

benchmark and their potential to be manufactured by low-cost solution-processing technologies.^{1–3} Device operational lifetime is the third corner of the magic triangle evaluating the performance potential of photovoltaic technologies, and this is of particular concern for the perovskite-based technology owing to its water soluble Pb-salt component.^{4,5} Limited shelf as well as operational lifetime have been reported in the early stages of perovskite research and appeared to be one of the major roadblocks towards further driving this technology.⁶ The precise degradation mechanism of perovskites in the presence of water is still under discussion. Previous studies have reported the loss of methyl-amine and the formation of yellowish PbI₂, while more recent studies rather suggest the partially reversible formation of (CH₃NH₃)₄PbI₆·H₂O hydrate complexes as an intermediate step.^{7,8} In addition, oxygen has been found to have only little effect on the degradation of perovskite devices.⁸

To address the stability issue of the perovskite solar cells, many efforts have recently been made. For example, several photo- or moisture-stable perovskite materials such as (C₆H₅(CH₂)₂NH₃)₂(CH₃NH₃)₂[Pb₃I₁₀], FA_{1-x}Cs_xPbI₃ and (CH₃NH₃-Pb(SCN)₂) have been developed.^{9–11} Recently, it has also been suggested that cross-linking perovskite grains with phosphonic acid ammonium derivatives may further decrease the moisture sensitivity of perovskite devices.¹²

A technically straightforward possibility to guarantee long-living perovskite devices is to provide a water and humidity dense package by using barriers and adhesives with a low water vapor transmission rate (WVTR).^{13,14} Hydrophobic carbon nanotubes/poly(methyl methacrylate) composites and Teflon have been used as effective barriers to slow down the ingress of moisture and the extended lifetime of perovskite cells are observed.¹³ In 2014, Han *et al.* employed a carbon layer as a back contact and a water-retaining layer for HTM-free perovskite solar cells. The resulting device exhibited a certified PCE of 12.8% and excellent stability under light soaking.¹⁵ In another study by Wei *et al.*, a free-standing flexible carbon film was employed as the cathode of HTM-free perovskite solar cells, obtaining a PCE of up to 13.53% and good device stability.¹⁶ Li

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et al. also demonstrated that HTM-free perovskite solar cells can withstand elevated temperature and constant outdoor illumination.¹⁷

Perovskite solar cells can be processed in various architectures such as planar structures and mesoporous structures.^{18–21} As our focus is on low temperature and solution-processing compatible architectures, we selected the inverted planar architecture (p–i–n) for our studies. According to the literature, the most popular inverted planar architectures generally consist of a transparent electrode (*e.g.* fluorine- or indium-doped tin oxide (FTO or ITO)), a hole-transporting layer, a perovskite absorbing layer, an electron-transporting layer (typically phenyl-C61-butyric acid methyl ester (PCBM)) in combination with a buffer layer (*e.g.* ZnO) and a cathode (*e.g.* Al, Ag, and Au).^{20,22,23} Several conducting polymers (*e.g.*, poly(3,4-ethylenedioxythiophene):polystyrene sulphonate (PEDOT:PSS), poly(bis(4-phenyl)(2,4,6-trimethylphenyl)amine), and polythiophene) have been suggested as hole-transporting layers for perovskite devices.^{24,25} PEDOT:PSS is the most widely used hole-transporting layer because of its prominent properties, such as low cost, excellent film-forming properties, “state of the art” ink standing time, green solvent and its proven track record for industrial-scale roll-to-roll processing.²⁶ However, PEDOT:PSS has disadvantages of being acidic and hygroscopic. Due to the fairly high equilibrium saturation concentration of water, PEDOT:PSS is prone to a quick accumulation of water in the interface layer. None of these disadvantages of PEDOT:PSS appear to be inevitable, and only recently we could demonstrate water-free PEDOT:PSS formulations that can be processed as a hole extraction interface layer on top of perovskite cells in the regular architecture (n–i–p).²⁷ Coming back to the hysteresis free replacements for PEDOT:PSS in the inverted architecture, various inorganic charge transporting layers, such as NiO_x, CuSCN and graphene oxide have been recently investigated as alternative HTMs for perovskite solar cells.^{28–33} Replacing PEDOT:PSS with NiO_x- or Li⁺-doped Ni_xMg_{1–x}O, indeed, results in almost hysteresis free solar cells with improved air stability.^{28,34,35} Reduced graphene oxide has been exploited as a barrier layer as well and a drop in PCE from about 10% to 6% within 120 h in an ambient atmosphere was found.³² Graphene oxide and reduced graphene oxide stem from natural graphite and are produced under harsh oxidation conditions. These protocols result in a rupture of the carbon framework and it was reported that nanometer huge holes were introduced.³⁶ In contrast to graphene oxide synthesis, a milder method was developed that largely excludes the formation of in-plane defects.³⁷ In addition, functional groups were quantified and anionic organic sulfate ester groups were found in that graphene derivative (oxo-G₁) with about one organosulfate on 20–30 carbon atoms.³⁸ Moreover, those organosulfate groups determine the negative charge of the functionalized graphene sheets.^{38,39}

In this work, solution-processed covalently sulfated graphene oxide (oxo-G₁) is employed to substitute hydrophilic PEDOT:PSS as the anode interfacial layer for perovskite solar cells. The resulting devices exhibit reasonably high performance with a PCE of up to 15.2% in the planar inverted

architecture with oxo-G₁ as a HTM, and most importantly, deploy the full V_{oc} of up to 1.1 V. In addition, oxo-G₁ effectively slows down the ingress of water vapor into the device stack resulting in significantly enhanced environmental stability. Unpackaged cells retain 80% of their initial PCE after 500 h of white light illumination and approximately 60% of their initial PCE after ~1000 h of light soaking under ambient conditions maintaining the temperature beneath 30 °C. Moreover, unsealed perovskite devices can retain 92% of their initial PCE after about 1900 h under ambient conditions and in the dark (shelf lifetime). Our results underpin that controlling the water diffusion into perovskite cells through advanced interface engineering is a crucial step towards prolonged environmental stability.

The schematic structure of the photovoltaic devices and structural formula of the oxo-G₁ are shown in Fig. 1a. Oxo-G₁ is synthesized by a wet chemical synthesis method as outlined in the literature.³⁷ By mild oxidation, the carbon lattice is mostly preserved after oxidation and delamination in contrast to the material obtained by *e.g.*, the Hummer's method. Additionally, organic sulphate esters bound on both sides of the single carbon sheets are not hydrolysed because the reaction temperature is permanently kept below 10 °C.⁴⁰ Subsequent statistical Raman measurements of the oxo-G₁ are carried out. The defect density inside the carbon lattice is determined to be ~0.8% (Fig. S1†).^{41–44} Fig. 1b presents an AFM image of the oxo-G₁ film deposited on a SiO₂/Si substrate. Fig. 1c exhibits the J - V characteristics of a champion device based on oxo-G₁ compared to a reference device with a PEDOT:PSS interlayer. To achieve high PCE, it is necessary to realize a high V_{oc} and a high short-circuit current density (J_{sc}) mutually. As indicated in Fig. 1c, our best performing oxo-G₁-based perovskite cell shows a PCE of 15.2%, a $J_{sc} = 18.06 \text{ mA cm}^{-2}$, a $V_{oc} = 1.08 \text{ V}$ and a fill factor (FF) = 77.7%, while the reference device based on PEDOT:PSS exhibits a PCE of only 10.8%, with a $J_{sc} = 17.1 \text{ mA cm}^{-2}$, a $V_{oc} = 0.928 \text{ V}$, and a FF = 68.6%. The average performance of more than 30 oxo-G₁-based device peaks is between 13 and 14% (Fig. S2†). Cross calibration by external quantum efficiency (EQE) measurements confirmed the J_{sc} values recorded under the AM 1.5 G solar spectrum with deviations within about 5% (see Fig. 1d). Although J_{sc} is similar for the oxo-G₁ and reference devices, replacing the PEDOT:PSS with oxo-G₁ significantly improves V_{oc} and FF. The increased V_{oc} can be dominantly attributed to the higher work function of oxo-G₁ than that of PEDOT:PSS. The work function of oxo-G₁ and PEDOT:PSS determined with a Kelvin probe is found to be –5.2 and –5.0 eV, respectively (Table 1). The relatively high FF of 77.7% attained for the oxo-G₁-based devices can be partially attributed to the lower series resistance. The series resistance of 9.9 $\Omega \text{ cm}^2$ is determined for the oxo-G₁-based device from the J - V characteristics in the dark, whilst 15.5 $\Omega \text{ cm}^2$ are calculated for the PEDOT:PSS-based reference (Fig. 1e). Moreover, as can be seen from the atomic force microscopy images (Fig. 2), the perovskite layer on the PEDOT:PSS film exhibits a smaller grain size than that on oxo-G₁, which may lead to the weak field dependence of the PEDOT:PSS-based device at the transition from the 4th into the 3rd quadrant.

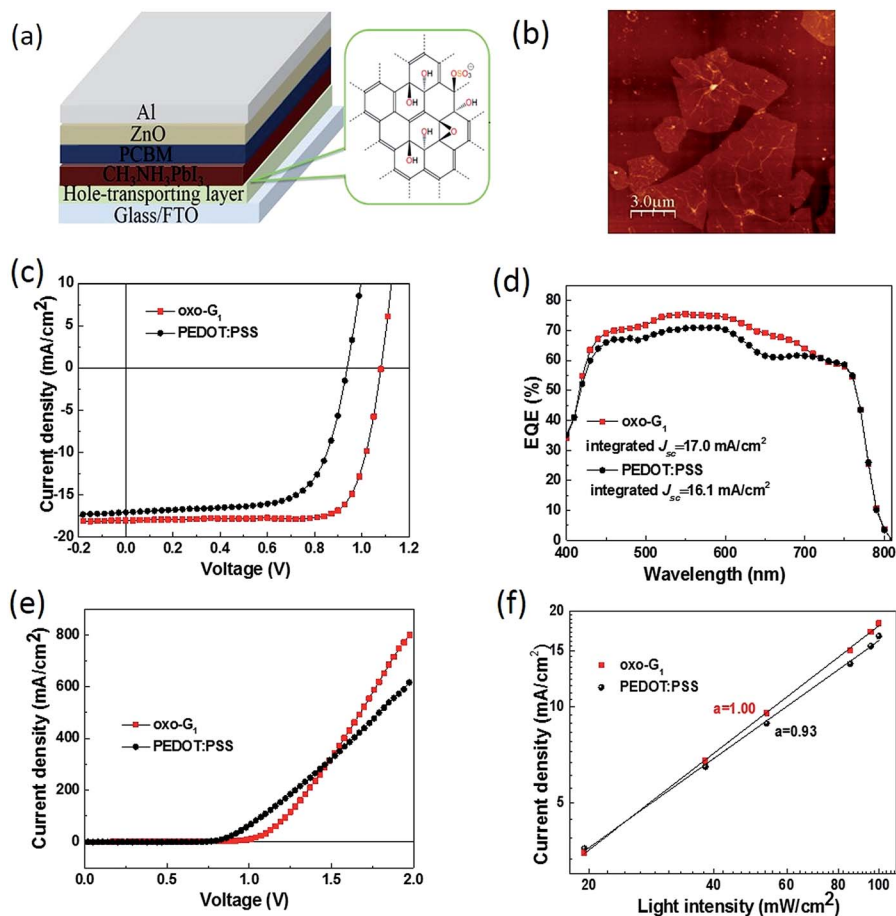


Fig. 1 (a) Schematic device structure of the perovskite solar cells with different electron-blocking layers, (b) a representative atomic force microscope (AFM) image of oxo- G_1 on a SiO_2/Si substrate, (c) J - V and (d) EQE characteristics of oxo- G_1 /CH₃NH₃PbI₃/PCBM/ZnO/Al solar cells (red squares) and PEDOT:PSS/CH₃NH₃PbI₃/PCBM/ZnO/Al solar cells (black circles), (e) comparison of the dark J - V curves of PEDOT:PSS and oxo- G_1 devices, and (f) short-circuit current as a function of the incident light intensity for different photovoltaic devices: oxo- G_1 /CH₃NH₃PbI₃/PCBM/ZnO/Al solar cells (red squares) and PEDOT:PSS/CH₃NH₃PbI₃/PCBM/ZnO/Al solar cells (black circles).

Table 1 Work function of PEDOT:PSS and oxo- G_1 measured with a Kelvin probe in air

	Work function (eV)	Contact angle
PEDOT:PSS	-5.0	18°
Oxo- G_1	-5.2	48°

Furthermore, light-intensity-dependent photocurrent measurements are performed to investigate the recombination mechanism of the devices (Fig. 1f). The relationship between J_{sc} and the incident light intensity (I) typically follows a power law ($J_{sc} \propto I^\alpha$). The light intensity dependence of J_{sc} is plotted on a log-log scale in Fig. 1f for both types of devices and fitted to a power law. With an exponent $\alpha \approx 1$, the 2nd order recombination is absent in oxo- G_1 -based devices under J_{sc} conditions.⁴⁵ Interestingly, PEDOT:PSS-based devices show a small but distinct contribution from the 2nd order recombination as evidenced by their lower $\alpha = 0.93$.

The operational lifetime of the fabricated perovskite devices based on oxo- G_1 and PEDOT:PSS interlayers is investigated with

a home-built setup. Illumination is provided by LED lamps in order to prevent the negative influence of ultraviolet light on the light-soaking stability. In order to systematically evaluate the long-term stability of the fabricated devices, we detect the photovoltaic performance of unpackaged oxo- G_1 and PEDOT:PSS devices under three different conditions: (a) storage in the dark and an ambient environment (Fig. 3a), (b) light soaking under 0.5 suns in a N_2 atmosphere (Fig. 3b) and (c) light soaking under 0.5 suns in an environmental atmosphere (Fig. 3c). Oxo- G_1 -based devices show superior stability under all three conditions over their counterpart based on PEDOT:PSS. Oxo- G_1 -based solar cells are unusually stable under ambient conditions in the dark. Solar cells are stored for more than 1900 h under ambient conditions (a temperature of approximately 15–25 °C and a relative humidity (RH) between 30 and 50%) and retain 92% of their initial PCE (Fig. 3a). In contrast, PEDOT:PSS-based solar cells lose about 50% of their initial PCE already within about 400 h. A similar trend is found for unpackaged cells operated in a N_2 atmosphere under 0.5 suns (Fig. 3b). Oxo- G_1 solar cells exhibit higher photo-stability than that of the PEDOT:PSS-based solar cells, although the difference is much

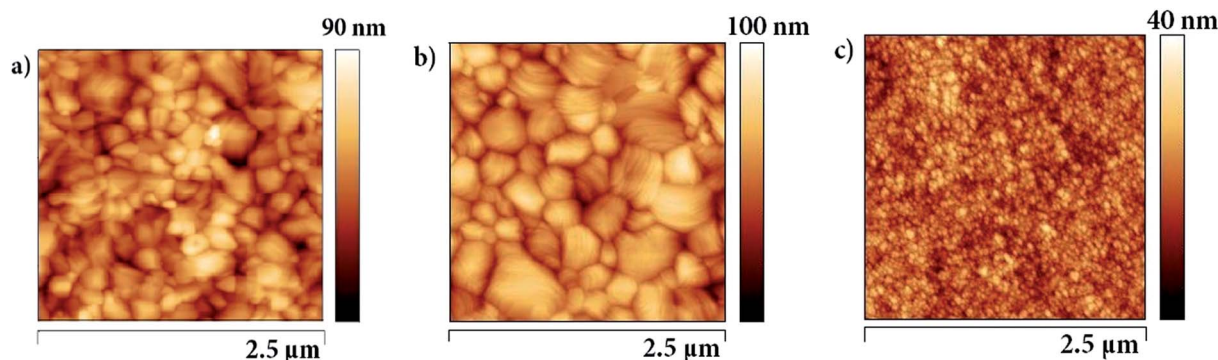


Fig. 2 AFM images of (a) perovskite grown on the PEDOT:PSS and (b) perovskite grown on the oxo-G₁; (c) ZnO on the oxo-G₁/perovskite/PCBM composite film.

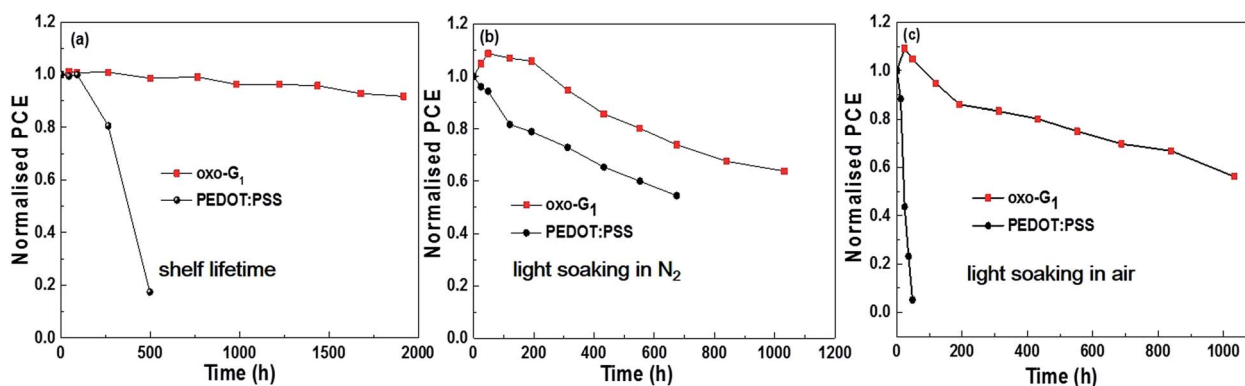


Fig. 3 Decline in PCE as a function of time for unsealed devices with oxo-G₁ compared to that of unsealed devices with PEDOT:PSS under three different conditions: (a) storage in the dark and an ambient environment, (b) light soaking under 0.5 suns in a N₂ atmosphere, and (c) light soaking under 0.5 suns in an environmental atmosphere (all *J*-*V* characteristics were measured under AM 1.5 G illumination) (all the devices were held at open circuit).

smaller. After about 670 h the oxo-G₁-based solar cells still retain 74% of their initial PCE, while PEDOT:PSS-based devices retain only 54% of their initial PCE after approximately 670 h. The most significant discrepancy in the lifetime is, however, observed under ambient environmental conditions (a temperature of approximately 20–30 °C and a RH between 30 and 50%) and illumination of approximately 0.5 sun equivalents provided by a white LED (Fig. 3c). Unsealed oxo-G₁ devices retain ~60% of their initial PCE after 1000 h. In contrast, PEDOT:PSS devices lose more than 95% of their initial PCE within only 50 h. This means the reference devices fail completely after 50 h and the oxo-G₁ devices did not yet show degradation. We summarize these measurements with the observation that unpackaged oxo-G₁-based perovskite solar cells show attractive environmental stability when compared to inverted architecture cells incorporating a PEDOT:PSS anode interfacial layer; it implies that oxo-G₁ may act as a barrier inhibiting the ingress of moisture into the perovskite layer.

It seems obvious to correlate the attractive stability of oxo-G₁-based devices with their enhanced hydrophobicity and potentially better moisture barrier properties as compared to those of PEDOT:PSS. As presented in Fig. 4a and b, oxo-G₁ has a larger contact angle of 48° than that of PEDOT:PSS (18°), confirming

that oxo-G₁ is indeed more hydrophobic than PEDOT:PSS. To further understand the role of sulfated graphene oxide in blocking moisture ingress into the perovskite layer, FTO glasses coated with oxo-G₁ or PEDOT:PSS are sealed to the open mouth of an aluminum cup filled with CaCl₂ (Fig. S3†). A thin film of ultraviolet curable epoxy adhesive (Katiobond LP 655, Delo) is deposited onto the PEDOT:PSS or oxo-G₁-coated FTO glass. Then, these cups are subjected to damp heat conditions in a climate chamber. Note that this method requires an adhesive with a water vapor diffusion constant being significantly below the one of the material of interest. We previously determined the water vapor diffusion constant *D* for the adhesive Delo Katiobond LP 655 with approximately $1.1 \times 10^{-12} \text{ cm}^2 \text{ s}^{-1}$ at 60 °C/90% RH and found that in excellent agreement with the WVTR value from the provider ($\text{WVTR} = 6.1 \text{ g m}^{-2} \text{ d}^{-1}$ [http://www.delo.de]).⁴⁶ Feron *et al.* recently measured the diffusion coefficient of water in PEDOT:PSS and calculated a value of $D = \sim 5.0 \times 10^{-10} \text{ m}^2 \text{ s}^{-1}$, *i.e.*, approximately two orders of magnitude larger than that of *D* for our adhesive.⁴⁷ It is therefore reasonable to expect a major contribution to water vapor ingress from the hole-extraction layer. Indeed, Fig. S4† confirms an at least 10-fold higher WVTR in PEDOT:PSS-based cups as compared to that of oxo-G₁-based cups.

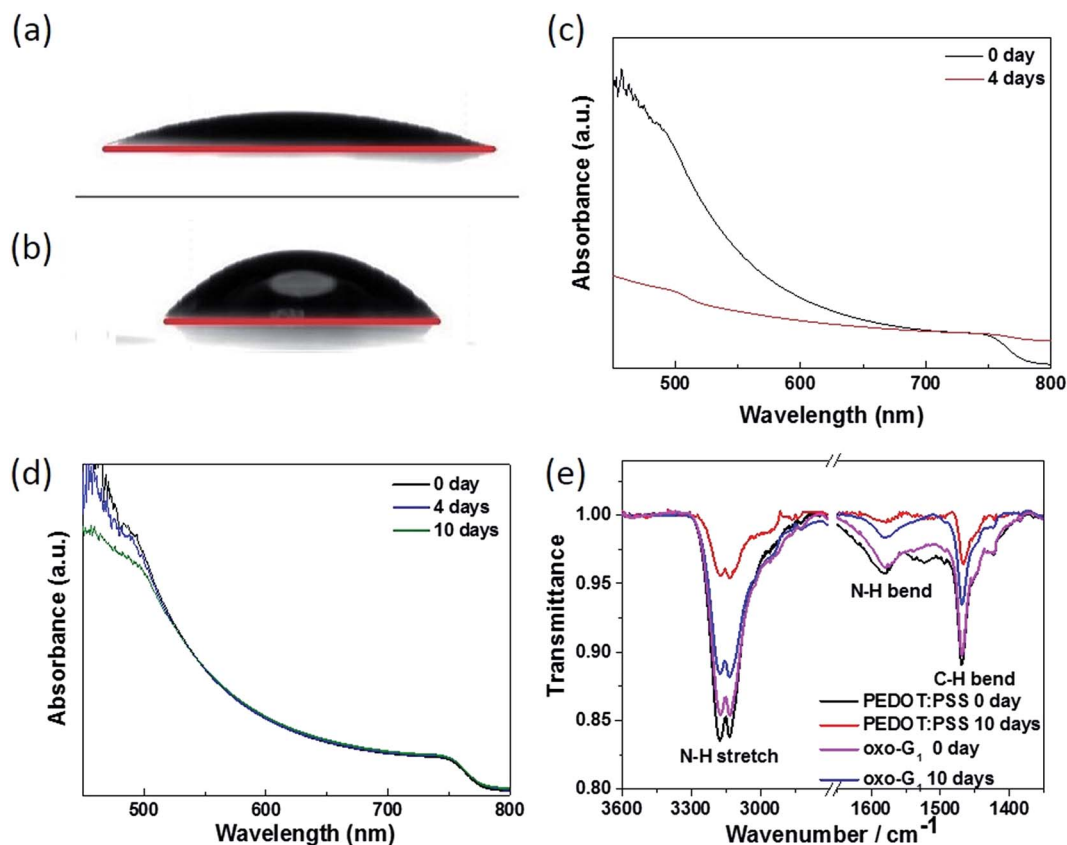


Fig. 4 Images of the contact angle of a water drop: (a) 18° on PEDOT:PSS and (b) 48° on oxo-G₁; UV-vis absorption of (c) unsealed PEDOT:PSS devices and (d) unsealed oxo-G₁ devices (light soaking under 0.5 suns in an environmental atmosphere); (e) ATR spectra of fresh and aged solar cells (light soaking under 0.5 suns in an environmental atmosphere).

Having elucidated that water vapor can indeed enter perovskite solar cells through the PEDOT:PSS layer, we next analyzed the degradation behavior of the respective devices under an ambient environment and light soaking by UV-vis and attenuated total reflectance-Fourier transform infrared (ATR-FTIR) spectroscopy. Fig. 4 reports a significant change in absorption for the PEDOT:PSS-based devices, while no or only negligible changes are observed for oxo-G₁-based devices after 10 days under 0.5 suns and at a humidity of 30–50%. We notice that all the features in Fig. 4c are in excellent correlation with the recent report by Kelly *et al.*⁸ who explained the three features of this spectrum by the formation of a mixture of perovskite, PbI₂ and PbI_x-based hydrate complexes. As reported by Venkatamaran *et al.*, ion migration is induced and accelerated by visible light, which explains why perovskite devices under light soaking in an environmental atmosphere degraded faster than the corresponding devices stored in the dark and an ambient environment.⁴⁸

ATR-FTIR spectroscopy is utilized to identify further degradation products on a molecular level. As depicted in Fig. 4e, the peak at 3200 cm⁻¹ is attributed to the N–H stretch vibration of the CH₃NH₃⁺ group.⁴⁹ After being exposed to an ambient environment and light soaking for ten days, the peak for the N–H stretch of the perovskite in the PEDOT:PSS devices dramatically declines, evidencing a serious loss of methyl amine in

PEDOT:PSS-based devices. As reported by Walsh *et al.*, water can protonate excess iodide to form HI and deprotonate the methylammonium cation to produce methylamine.⁵⁰ Both of these compounds are highly volatile. The evaporation of methylamine and HI will accelerate the rate of formation of the hydrated phase. In contrast, the signal of the N–H stretch vibration of perovskite in oxo-G₁-devices decreases only slightly (Fig. 4e), thus implying significantly reduced hydrolysis for oxo-G₁-based devices.

Conclusions

In summary, we demonstrate surprisingly long operating lifetimes for unpackaged perovskite solar cells under continuous illumination with a white-light LED with a light intensity of ~0.5 suns and under ambient conditions. The environmental light-soaking stability of unpackaged inverted architecture perovskite solar cells is significantly enhanced by replacing PEDOT:PSS with oxo-G₁. Oxo-G₁ not only acts as an efficient hole-extraction layer, but more importantly, contributes to stabilizing perovskite solar cells by suppressing moisture ingress due to its hydrophobic nature compared to that of PEDOT:PSS. A champion efficiency of up to 15.2% has been obtained using a graphene derivative layer oxo-G₁ and unpackaged oxo-G₁ solar cells retain ~60% of their initial

performance after 1000 h of light soaking. The findings demonstrated in this work highlight the importance of designing hydrophobic interfacial layers with low WVTR as a crucial development towards more stable perovskite solar cells.

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