

High-Triplet-Energy Bipolar Host Materials Based on Phosphine Oxide Derivatives for Efficient Sky-Blue Thermally Activated Delayed Fluorescence Organic Light-Emitting Diodes with Reduced Roll-of

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High-Triplet-Energy Bipolar Host Materials Based on Phosphine Oxide Derivatives for Efficient Sky-Blue Thermally Activated Delayed Fluorescence Organic Light-Emitting Diodes with Reduced Roll-off

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1 We designed and synthesized two new ambipolar host 2 materials, namely CzPO and Cz3PO, which contain 3 electron-donating carbazole and electron-accepting 4 triphenylphosphine oxide moieties. Thermally activated 5 delayed fluorescence (TADF)-based OLEDs employing 6 CzPO and Cz3PO as host materials and the 9-(4-(4,6diphenyl-1,3,5-triazin-2-yl)phenyl)-1,3,6,8-tetramethyl-7 8 9H-carbazole (CzTRZ2) as the emitter resulted in improved maximum external quantum efficiencies, EQEmax, 9 10 of 13.1% and 13.2%, respectively, together with small 11 efficiency roll-offs, while the device based on bis[2-12 (diphenylphosphino)phenyl]ether oxide (DPEPO) showed a 13 much more pronounced efficiency roll-off. The reduced efficiency roll-off in the devices based on CzPO and Cz3PO 14 15 can be ascribed to their improved ambipolar charge transport capacity compared to that of **DPEPO**, which results in a 16 17 broader recombination zone in the emitting layer using these 18 ambipolar hosts. 19

20 Keywords: Thermally activated delayed fluorescence (TADF) | 21 Organic light-emitting diode (OLED) | Ambipolar host

22 After the invention of fluorescence-based organic 23 light-emitting diodes (OLEDs) in 1987¹, the flat panel 24 display market based on OLEDs is emerging to become 25 the dominant technology. Both organic-based 26 fluorescence and organometallic-based phosphorescence emitters are currently used in OLEDs^{2,3}. In 2012, a series 27 28 of highly efficient OLEDs based on thermally activated 29 delayed fluorescence (TADF) were demonstrated, in 30 which an efficient process of reverse intersystem crossing 31 (RISC) from a triplet excited-state to a singlet excitedstate was realized in purely organic molecules⁴. OLEDs 32 33 based on TADF emitters achieved comparable an internal 34 quantum efficiency (IQE) of 100%. Since then, a large 35 number of TADF emitters have been reported⁵. In 36 particular, TADF emitters have shown tremendous 37 potential to replace the state-of-the-art fluorophores in 38 high performance deep-blue OLEDs. During the survey of 39 blue-emitting devices, significant efficiency roll-off at 40 high current density was identified in these devices, and 41 the triplet energies (E_T) of the host materials must be 42 sufficiently high compared to the emitters in order to 43 confine the excitons on the emitters and thereby realize 44 efficient blue OLEDs^{5c}. Further, the balance of charge 45 transport carriers must also be considered to mitigate the 46 efficiency roll-off.

47 For blue TADF-based OLEDs, there are some general 48 requirements for the host material: (1) the $E_{\rm T}$ of host 49 materials must be higher than that of the emitter to prevent back energy transfer⁶; (2) to facilitate effective charge 50 51 injection, both the highest occupied molecular orbital 52 (HOMO) and lowest unoccupied molecular orbital 53 (LUMO) energy levels of the host materials should be well 54 matched with both those of the emitter and the other layers 55 of the device⁷; (3) a good host material should possess ambipolar character to best manage charge transport 56 through the emissive layer^{8,9}. In general, the 57 58 recombination region tends to occur closer to the emissive 59 layer/electron-transporting layer (EML/ETL) interface 60 when using hole-transporting (HT) hosts. Likewise, the 61 recombination region is likely to be located in the 62 EML/HTL interface when using an electron-transporting 63 (ET) host such as the frequently used DPEPO. It has been found that poor and unbalanced carrier mobility within the 64 EML is responsible for inferior efficiencies, low stability 65 66 and efficiency roll-off¹⁰. OLEDs with narrow charge recombination zones lead to devices presenting severe 67 efficiency roll-off due to the local accumulation of high-68 69 density triplet excitons, especially at high current densities¹¹⁻¹⁵. As a result, an ambipolar host is more 70 71 suitable for improving the device performance in blue 72 OLEDs. The simple incorporation of both HT and ET moieties within host materials should give rise to 73 ambipolar host materials¹⁶⁻¹⁹. **DPEPO** is the most popular 74 host material for blue-emitting OLEDs²⁰. However, the 75 76 use of such ET-type host materials is usually coupled with 77 reduced device stability and efficiency roll-off²¹⁻²⁴. Thus, 78 in this study, we designed two host materials, CzPO and

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1 **Cz3PO**, by linking carbazole donor units with an electronaccepting triphenylphosphine oxide group (Fig. 1). Both hosts show high E_{TS} and ambipolar charge transport properties. Sky-blue OLEDs based on these two bipolar 5 host materials not only showed smaller efficiency roll-off 6 characteristics but also operated under relatively low 7 driving voltages.



Figure 1. Molecular structures, HOMO, LUMO, S₁ and T₁
energy levels of CzPO and Cz3PO characterized by DFT
calculations at the PBE0/6-31G(d) level of theory, and the
electron density distributions of the frontier molecular
orbitals and spin density distributions of T₁ energy levels.

14 The energy levels of CzPO and Cz3PO were first 15 investigated computationally using the Gaussian 16 program package²⁵. Ground-state geometries in the gas phase were 16 17 optimized at the PBE0/6-31G(d) level, and the lowest singlet 18 and triplet excited states were calculated using time-19 dependent density functional theory (TD-DFT) based on the 20 optimized ground-state geometries. The electron density 21 distribution of the HOMOs and LUMOs of CzPO and 22 Cz3PO are depicted in Fig. 1 along with their energies and 23 the energies of the singlet and triplet excited states and the 24 spin density distribution of the T₁ states. The HOMOs of both 25 host materials are delocalized over the electron-donating 26 carbazole units and bridging phenylene while the LUMOs are 27 mainly localized over the electron-accepting phosphine oxide 28 moiety.



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Figure 2. UV-Vis absorption spectra of CzPO, Cz3PO and DPEPO in toluene (10^{-5} M) and (b) PL emission spectra of CzPO, Cz3PO and DPEPO in toluene (10^{-5} M) and neat films.

The UV–Vis absorption and photoluminescence (PL) spectra of **CzPO**, **Cz3PO** and **DPEPO** are shown in Fig. 2 respectively. The photophysical data are summarized in Table S1. In toluene, all host materials show strong absorption bands below 300 nm, which are assigned as π – π * transitions of the phosphine oxide moiety. The absorption band beyond 300 nm is absent in **DPEPO**, and the lowest41 energy absorption bands ranging from 310 to 350 nm are 42 attributed to the π - π * transition of the 9-phenylcarbazole unit 43 in CzPO and Cz3PO. In toluene, CzPO shows emission 44 maxima (λ_{PL}) at 344 and 359 nm, while **Cz3PO** shows λ_{PL} at 45 361 nm. DPEPO shows the most hypsochromically shifted 46 emission at 290 nm. The bathochromically shifted emission 47 in Cz3PO is due to the presence of three donor groups within 48 the molecule. In neat films, only CzPO and Cz3PO display 49 red-shifted emission at 377 and 406 nm, respectively, 50 whereas **DPEPO** show an additional emission band at 437 51 nm that may be ascribed to excimer emission.²⁸ In the present 52 study, CzTRZ2 was used as the TADF emitter in the emitting 53 layer as the absorption spectrum of CzTRZ2 overlaps 54 strongly with the neat film emission spectra of CzPO and 55 Cz3PO, thereby assuring efficient Förster resonance energy 56 transfer (FRET) from the excited hosts to CzTRZ2.



Figure 3. (a) Phosphorescence spectra of CzPO, Cz3PO
and DPEPO in toluene matrices at 77 K. (b)
Phosphorescence spectra of donor and acceptor moieties in
toluene matrices at 77 K.

62 The phosphorescence spectra of CzPO, Cz3PO and 63 **DPEPO** are depicted in Fig. 3a and these are cross-compared 64 with the phosphorescence spectra of N-phenylcarbazole and triphenylphosphine oxide (Fig. 3b), which act as reference 65 66 electron-donor and electron-acceptor moieties. From the 67 phosphorescence spectra, the ET values were determined to 68 be 3.06 and 3.05 eV for CzPO and Cz3PO, respectively. The 69 $E_{\rm T}$ of CzPO and Cz3PO are higher than that of CzTRZ2 70 (2.85 eV)^{5d}, ensuring a suppression of back energy transfer. Both CzPO and Cz3PO show similar vibronic-structured 71 72 emission profiles and $E_{\rm T}$ to that of N-phenylcarbazole. 73 Therefore, the lowest T₁ excited state of each of CzPO and 74 Cz3PO is localized on the donor moieties, consistent with the 75 computed spin density simulation.



Figure 4. (a) Steady-state PL spectra and (b) time-resolved PL decay profiles of 6 wt.%-CzTRZ2:host doped thin films (host = CzPO, Cz3PO, and DPEPO) measured at 300 K under N_2 . (excitation wavelength: 280 nm).

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1 The PL spectra of 6 wt %-CzTRZ2 doped films in CzPO, 2 Cz3PO and DPEPO are nearly identical, indicating efficient FRET (Fig. 4a). The CzPO-, Cz3PO- and DPEPO-doped 3 4 films of CzTRZ2 show broad and structureless sky-blue 5 emission with high Φ_{PL} of 81%, 75%, and 92%, respectively. 6 The transient decay profiles of the doped films clearly show 7 biexponential decay respectively, at 300 K (Figs. S1 and 4b). 8 Such behavior is characteristic of the TADF emission previously reported for this compound.^{5d} Furthermore, both 9 hosts materials possess a high decomposition temperature, T_{d} , 10 of 365 and 454 °C, for CzPO and Cz3PO, respectively, 11 which is indicative of high thermal and morphological 12 stabilities (Fig. S2). The T_d values are substantially higher 13 than that of **DPEPO** $(T_d = 322 \text{ °C})^{26}$. 14



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16 **Figure 5.** (a) The EL spectra measured at 8 V, and (b) 17 external quantum efficiency (η_{ext})-current density curves of 18 the TADF-based OLEDs.

We next evaluated the EL performance in OLEDs using 19 20 CzTRZ2 as the TADF emitter and CzPO- and Cz3PO and 21 **DPEPO** as host matrices in the EML. The devices were 22 fabricated with the following architecture : ITO (100 23 nm)/HATCN (10 nm)/TAPC (40 nm)/mCP (10 nm)/6 wt.%-24 CzTRZ2:host (20 nm)/PPT (10 nm)/TPBi (50 nm)/LiF (0.8 25 nm)/Al (100 nm), in which 2,3,6,7,10,11-hexacyano-26 1.4.5.8.9.12-hexaazatriphenylene (HATCN) is the hole-27 injection layer, 1,1-bis(4-ditolylaminophenyl) cyclohexane 28 (TAPC) and 1,3-bis(N-carbazolyl)benzene (mCP) are the 29 hole-transporting layer and electron-blocking layer, 30 respectively. The EML layer consists of 6 wt.% of CzTRZ2 31 doped Cz3PO or DPEPO. in CzPO, 2.8 -Bis(diphenylphosphoryl)dibenzothiophene (PPT) is the hole-32 33 blocking layer and 1,3,5-tris(N-phenylbenzimidazol-2-34 yl)benzene (TPBi), LiF and Al serve as the electron-35 transporting layer, the electron-injection layer and cathode, respectively. The ET of electron-blocking and hole-blocking 36 37 layers, i. e., mCP (2.9 eV) and PPT (2.9 eV), respectively, are 38 higher than that of CzTRZ2, which ensures confinement of 39 the triplet excitons within the EML²⁷. The EL spectra and 40 EQE characteristics of the corresponding OLEDs are shown in Fig. 5. Devices based on CzPO and Cz3PO as host 41 42 materials resulted in high EQE_{max} of 13.1 and 13.2% 43 respectively. By contrast, the device based on DPEPO 44 showed a higher EQE_{max} of 16.7%. Despite the higher EQE_{max}, 45 the **DPEPO**-based devices showed a significantly serious 46 efficiency roll-off with an EQE of only 1% at 100 mA cm⁻². 47 Meanwhile, the CzPO- and Cz3PO-based devices exhibited 48 reduced efficiency roll-off with EQE_{100} of 4.2% and 4.2%, 49 respectively, at a display-relevant current density of 100 mA 50 cm⁻². All the device data are shown in Table S2.



Figure 6. Current density-voltage (*J*-*V*) characteristics of
(a) hole-only devices (HODs) and (b) electron-only devices
(EODs) based on the CzPO, Cz3PO and DPEPO hosts.

55 To investigate the reason for the reduced roll-off 56 characteristics, hole-only devices (HODs) and electron-only 57 devices (EODs) with different host materials were fabricated 58 to evaluate the carrier transporting properties of the 59 corresponding OLEDs. These device configurations were: 60 ITO/HAT-CN (10 nm)/TAPC (20 nm)/ host (60 nm)/TAPC 61 (20 nm)/Al (100 nm) for the HODs and ITO/TPBi (20 62 nm)/host (60 nm)/TPBi (20 nm)/LiF (0.8 nm)/Al (100 nm) for the EODs. Figure 6 showed the J-V curves of the 63 64 fabricated HODs and EODs. For the HODs (Fig. 6a), the and Cz3PO-based devices showed hole 65 Cz3PO transporting/injection properties due to the presence of more 66 67 electron-donating carbazole units, while no significant 68 electron current was observed in DPEPO-based device. In 69 contrast. for EODs, the order of the electron 70 transporting/injection properties of devices was **DPEPO** > 71 CzPO > Cz3PO (Fig. 6b). From both the HODs and EODs 72 characteristics, CzPO and Cz3PO were confirmed to be 73 ambipolar in nature, in which CzPO- and Cz3PO-based 74 devices exhibited balanced hole and electron 75 transporting/injection properties. Thus, the well-balanced 76 charge flux and broad distribution of the charge-77 recombination zones within the EML resulted in low 78 efficiency roll-off characteristics for CzPO- and Cz3PO-79 based OLEDs.

80 In summary, reduced efficiency roll-offs were 81 demonstrated in sky blue TADF-based OLEDs using CzPO 82 and Cz3PO as ambipolar hosts compared to the device with 83 the reference DPEPO host. Sky blue TADF-OLEDs based 84 employing CzPO and Cz3PO host materials showed 85 maximum EQEs of 13.1 and 13.2%, respectively, and 86 reduced roll-offs with EQEs of 4.2% and 4.2%, respectively, 87 at current densities of 100 mA cm⁻². 88

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Graphical Abstract		
	Textual Information	
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Graphical Information		
Please insert your Graphical Abstract: The size is limited within 100 mm width and 30 mm height, or 48 mm square>(required)		
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