

Conductivity and redox stability of new double perovskite oxide $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$)

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Conductivity and redox stability of new double perovskite oxide $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$)

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Abstract A series of new perovskite oxides $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) were synthesised by solid state reaction method. Synthesis of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) was achieved above 700 °C in 5 % H_2/Ar , albeit with the formation of impurity phases. Phase stability upon redox cycling was only observed for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$. Redox cycling of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) demonstrates a strong dependence on high temperature reduction to achieve high conductivities. After the initial reduction at 1200 °C in 5 % H_2/Ar , then re-oxidation in air at 700 °C and further reduction at 700 °C in 5 % H_2/Ar , the attained conductivities were between 0.1 and 58.4 % of the initial conductivity after reduction 1200 °C in 5 % H_2/Ar depending on the composition. In the investigated new oxides, sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ is most redox stable also retains reasonably high electrical conductivity, ~70 S/cm after reduction at 1200 °C and 2–3 S/cm after redox cycling at 700 °C, indicating it is a potential anode for SOFCs.

Introduction

Solid oxide fuel cells (SOFCs) are electrochemical devices to efficiently convert chemical energy into electricity. The conventional Ni-based cermet anode for SOFCs exhibits excellent catalytic activity and high conductivity which is good but suffers sintering/coarsening at high temperature. The significant volume change between NiO and Ni on reduction may lead to delamination between the anode and electrolyte interface. Carbon deposition on the Ni-based anode when hydrocarbon fuels are used in the SOFCs is a challenge [1, 2]. Therefore, it is desired to develop new anode, particularly redox stable anode for SOFCs [3–10]. Double perovskite $\text{Sr}_2(\text{TM})\text{MoO}_{6-\delta}$ (TM = Mn, Mg, Fe, Co, Ni, Cu, Zn) as potential anode materials for SOFCs has been the subject of a substantial body of research [3, 5, 8, 9, 11], and good fuel cell performance has been achieved for $\text{Sr}_2\text{MgMoO}_{6-\delta}$ [5], $\text{Sr}_2\text{MnMoO}_{6-\delta}$ [5], $\text{Sr}_2\text{CoMoO}_{6-\delta}$ [12] and $\text{Sr}_2\text{FeMoO}_{6-\delta}$ [13] anodes; of these compounds, only $\text{Sr}_2\text{MgMoO}_{6-\delta}$ (SMMO) has been proven to be redox stable [14]. Despite achieving redox stability, the chemical reactivity of SMMO with common electrolytes, such as LSGM and YSZ, limits its utility [6].

Xiao et al. improved both the formability and stability of $\text{Sr}_2\text{FeMoO}_{6-\delta}$ through an increase in the iron content of the sample, with $\text{Sr}_2\text{Fe}_{1.33}\text{Mo}_{0.66}\text{O}_{6-\delta}$ formed at 800 °C in H_2 , 300 °C below the synthesis temperature of $\text{Sr}_2\text{FeMoO}_{6-\delta}$ in 5 % H_2/Ar [13, 15]. The conductivity of $\text{Sr}_2\text{Fe}_{1.33}\text{Mo}_{0.66}\text{O}_{6-\delta}$ in 5 % H_2/Ar ranges between 15 and 30 S cm^{-1} from 700 °C to 300 °C, sufficient for an IT-SOFC anode material. Further development of this series by Liu et al. formed $\text{Sr}_2\text{Fe}_{1.5}\text{Mo}_{0.5}\text{O}_{6-\delta}$ in air at 1000 °C and demonstrated high conductivity in both oxidising and reducing atmospheres [11]. Good performance of $\text{Sr}_2\text{Fe}_{1.5}\text{Mo}_{0.5}\text{O}_{6-\delta}$ as a symmetrical electrode was achieved,

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attaining $\sim 500 \text{ mWcm}^{-2}$ at $800 \text{ }^\circ\text{C}$ in humidified H_2 with good stability over successive redox cycles.

It has been reported that potassium doping of $\text{Sr}_2\text{FeMoO}_{6-\delta}$ could improve the ionic conductivity of the parent material with minimal disruption to the compound structure, and good fuel cell performance has been achieved when $\text{Sr}_{1.6}\text{K}_{0.4}\text{FeMoO}_{6-\delta}$ was used as the anode for a SOFC [16]. Synthesis of the potassium-doped strontium molybdenum ferrite was noted to improve the formability of these compounds, with the formation of single-phase $\text{Sr}_{1.6}\text{K}_{0.4}\text{FeMoO}_{6-\delta}$ observed after reduction at $850 \text{ }^\circ\text{C}$ in H_2 , $250 \text{ }^\circ\text{C}$ lower than is required for the pure strontium analogue in $5 \text{ } \%$ H_2/Ar [13]. The conductivity of these compounds was comparable to that of the pure strontium iron molybdate, despite the lower synthesis temperature. Acceptable fuel cell performance, 766 mWcm^{-2} at $800 \text{ }^\circ\text{C}$ in H_2 , was also observed for a $\text{Sr}_{1.6}\text{K}_{0.4}\text{FeMoO}_{6-\delta}/\text{LSGM}/\text{Sr}_{0.9}\text{K}_{0.1}\text{FeO}_{3-\delta}$ cell [16], although the anodic composition was later determined to exhibit a mixture of SrMoO_3 and $\text{SrFe}_{0.6}\text{Mo}_{0.4}\text{O}_{2.7}$ phases. As the introduction of potassium had exhibited an improvement in the compound formability, it was posited that a further increase could be elicited through an increase in the iron content, as for the pure strontium analogue [11, 17]. To this end, the formability and redox stability of a series of new materials of the composition $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) was determined, and conclusions as to the suitability of these compounds for use as SOFC anode materials were drawn.

Experimental information

Materials synthesis

$\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) were produced by solid state synthesis technique. Stoichiometric amounts of SrCO_3 ($>99.9 \text{ } \%$, Sigma Aldrich), KHCO_3 ($99 \text{ } \%$, Alfa Aesar), Fe_2O_3 ($99.5 \text{ } \%$, Alfa Aesar) and MoO_3 ($99.5 \text{ } \%$, Alfa Aesar) were weighed and mixed in a planetary ball mill (Fritsch P6) for 2 h prior to firing at $900 \text{ }^\circ\text{C}$ for 5 h. A second firing at $1100 \text{ }^\circ\text{C}$ for 2 h was then performed. Pellets of all the samples ($\phi \approx 13 \text{ mm} \times 2 \text{ mm}$) were uniaxially pressed at 221 MPa and sintered in air at $1200 \text{ }^\circ\text{C}$ for 2 h. To study the redox stability, some of the as-prepared pellets were further fired in $5 \text{ } \%$ H_2/Ar for 10 h at 700 and $1200 \text{ }^\circ\text{C}$, respectively.

Materials characterisation

Phase purity and crystal parameters of the samples were examined by X-ray diffraction (XRD) analysis using a PANalytical X'Pert PRO MPD Multipurpose

diffractometer (Cu $\text{K}_{\alpha 1}$ radiation, $\lambda = 1.5405 \text{ \AA}$). GSAS [18] software was used to perform a least squares refinement of the lattice parameters of all suitable samples.

The densities of the pellets were determined from the measured mass and volume. Theoretical densities were calculated using experimental lattice parameters and the chemical formula of the sample. The relative densities were calculated from the actual and theoretical density values. The density of the pellets was around $90 \text{ } \%$ for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$).

Thermal analysis was conducted using a Stanton Redcroft STA 1500 Thermal Analyser on heating from room temperature to $800 \text{ }^\circ\text{C}$ and on cooling from $800 \text{ }^\circ\text{C}$ to room temperature in air, with a heating/cooling rate of $10 \text{ }^\circ\text{C min}^{-1}$ in $5 \text{ } \%$ H_2/Ar with a flow rate of $5 \text{ } \%$ H_2/Ar of 50 ml min^{-1} .

Conductivity measurements

Pellets for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) were coated on opposing sides using silver paste after firing at $1200 \text{ }^\circ\text{C}$ for 8 h in $5 \text{ } \%$ H_2/Ar . The conductivity of the samples was measured primarily in $5 \text{ } \%$ H_2/Ar between 300 and $700 \text{ }^\circ\text{C}$. Secondary measurements over the same temperature range were conducted in air following an equilibration step of 12 h at $700 \text{ }^\circ\text{C}$ in air. Final measurements over the same temperature range were conducted after an equilibration step of 12 h at $700 \text{ }^\circ\text{C}$ in $5 \text{ } \%$ H_2/Ar . Measurements were conducted using a pseudo four terminal DC method using a Solartron 1470E potentiostat/galvanostat controlled by CellTest software with an applied current of $1.0\text{--}0.1 \text{ A}$ [19].

Results and discussion

Synthesis of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) in air

XRD of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) after synthesis in air exhibited a single-phase double perovskite structure for composition $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ with space group $Fm\text{-}3m$ as shown in Fig. 1. However, an additional $\text{SrMoO}_{4-\delta}$ second phase (PDF: 01-085-0809) was observed for both $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ and $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$. The formation of a single-phase double perovskite structure for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ correlates with previous research into iron-rich strontium iron ferrites, which suggests that the formability limit in air for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ lies between $x = 0.4$ and $x = 0.5$ [15, 17, 20]. The Fe-rich composition will have better charge balance with the presence of Mo^{6+} ions in the perovskite oxide, facilitating the formation of single phase.

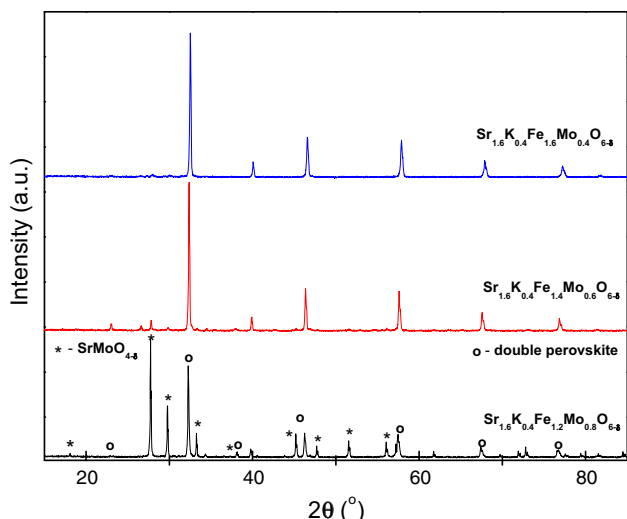


Fig. 1 XRD patterns of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4$ and 0.6) synthesised in air

At $x = 0.2$, if the charge for elements Fe and Mo in $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ is +3 and +6 respectively, the total positive charge is 12 then the oxygen sub-lattice should be stoichiometric, i.e. $\delta = 0$. However, the oxygen sub-lattice in perovskite oxides tends to be non-stoichiometric with the formation of oxygen vacancies, i.e. $\delta > 0$, after firing in air at high temperature [21, 22]. Under the circumstance, the positive charge will be greater than the negative charge, thus not balanced and difficult to form single-phase perovskite oxide. Therefore single-phase double perovskite oxide was not formed, while stable second-phase SrMoO_4 was formed (Fig. 1). With the increase of x to 0.4, the total positive charge is fewer than 12, which allows the formation of oxygen vacancies, and thus the second-phase SrMoO_4 was significantly reduced. Single-phase double perovskite oxide was formed with the Fe-rich sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ (Fig. 1).

STA of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) in 5 % H_2/Ar

Thermogravimetric analysis in 5 % H_2/Ar of the $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) samples synthesised in air, is shown in Fig. 2a. The initial weight loss at a temperature below 150 °C for samples $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ and $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$ is probably due to the desorption of adsorbed water and gases. The further weight loss for all samples at a temperature of ~500 °C is due to the reduction of both iron and molybdenum ions accompanying with the loss of lattice oxygen [23]. In general, it is believed that reduction of iron ions is easier than that of molybdenum ions. From this point of view, sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ is expected to have the

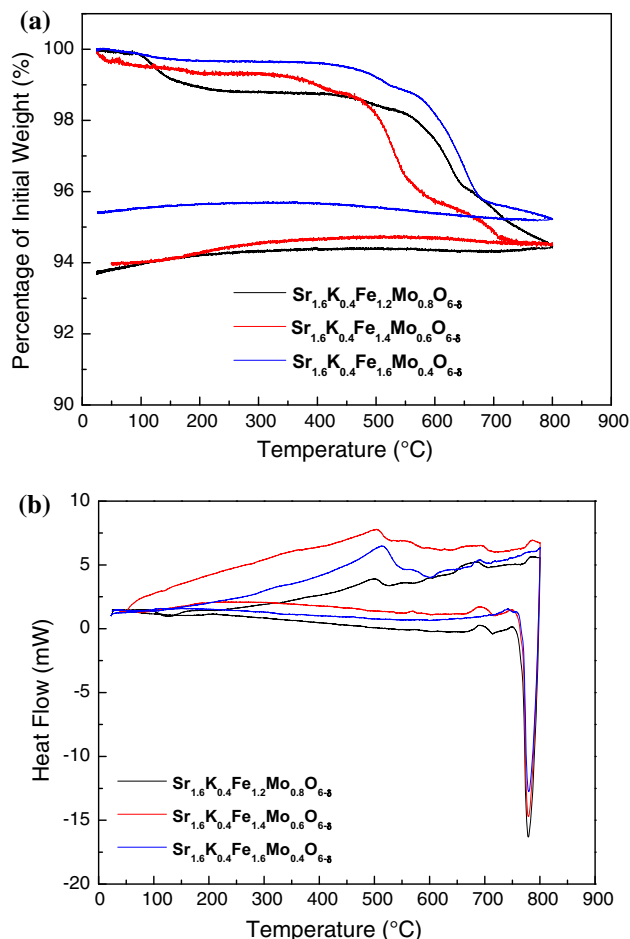
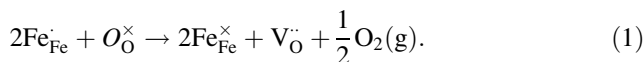


Fig. 2 Thermogravimetric analysis (a) and differential scanning calorimetry (b) of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4$ and 0.6) in 5 % H_2/Ar

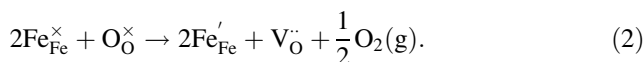
lowest reduction temperature. However, in the $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ series, the lower reduction temperature was observed for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$. This indicates that, in this study, the reduction of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ oxides is more complicated than expected and it is hard to decide which ions, iron, or molybdenum, will be reduced first.

In perovskite oxides obtained in air, the charges of iron and molybdenum ions are normally in the $\text{Fe}^{3+}/\text{Fe}^{4+}$ and Mo^{6+} respectively. In a reducing atmosphere, the state of iron is changed to $\text{Fe}^{3+}/\text{Fe}^{2+}$ while for molybdenum it is normally reduced to Mo^{5+} .

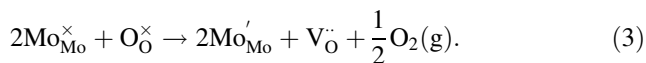
The reduction of Fe^{4+} to Fe^{3+} can be described as



The reduction of Fe^{3+} to Fe^{2+} can be described as



The reduction of Mo^{6+} to Mo^{5+} can be described as



In the defect equations above, Kröger–Vink notations are used.

It can be noticed that the reduction of both iron and molybdenum ions accompanies with the loss of lattice oxygen and formation of oxygen vacancies.

Differential scanning calorimetry, as shown in Fig. 2b, of all samples exhibits an exothermic peak at a temperature around 500 °C upon heating is probably due to the reduction of perovskite oxides because endothermic peaks were not observed on cooling therefore it is irreversible indicating not related to phase changes. For samples with second-phase SrMoO_4 , the reduction of SrMoO_4 starts at 750 °C in 4 % H_2/Ar which happens at a much higher temperature [24]. There were some exothermic effects around 700 °C on heating which could be related to the phase transformation associated to the reduction of perovskite oxides (Fig. 2b). Samples with $x = 0.2$ and 0.4 exhibit exothermic effects at ~ 700 °C on cooling (Fig. 2b) which might be associated to the second-phase SrMoO_4 which was presented in both samples. The sudden dip at DSC curve on cooling from 800 °C was caused by the STA system itself [25].

Structure of reduced $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$)

Reduction of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$ in 5 % H_2/Ar at 700 °C leads to a significant reduction in the proportion of the secondary SrMoO_4 phase (PDF: 01-085-0809) in the

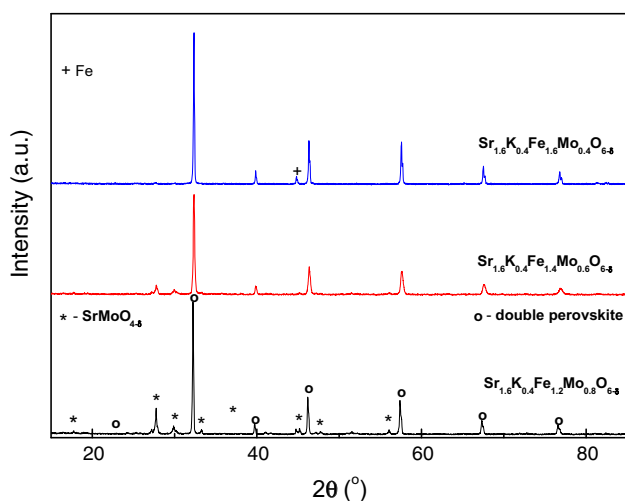


Fig. 3 XRD patterns of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4$ and 0.6) after reduction in 5 % H_2/Ar at 700 °C

XRD pattern, shown in Fig. 3. SrMoO_4 is a stable oxide. It has been reported that reduction of SrMoO_4 in 4 % H_2/Ar began at 750 °C and completed after holding at 800 °C for about 3 h [24]. The reducing temperature, 700 °C in this study was not enough to convert SrMoO_4 into SrMoO_3 . Therefore SrMoO_3 was not observed. During the firing process, some SrMoO_4 may react with the double oxide forming a new oxide with more Mo accommodated in the lattice. This indicates that a reducing atmosphere is in favour of the formation of single-phase double perovskite which was also observed in previous reports [5, 17]. The pattern of sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ was unchanged whilst an extra peak at $\sim 45^\circ$ was observed for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ which belongs to the strongest (110) peak of $\alpha\text{-Fe}$ (PDF: 6-696) after the reduction at 700 °C [23, 26]. The splitting of some peaks at low d-spacing for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ after reduction in 5 % H_2/Ar at 700 °C (Fig. 3) is probably due to the reduced symmetry of the perovskite phase [27].

XRD patterns of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) samples after reduction in 5 % H_2/Ar at 1200 °C for 10 h are shown in Fig. 4. All samples exhibited a double perovskite structure (SG: $Fm\text{-}3m$), albeit with presence of a small proportion of a secondary Fe phase (PDF: 6-696). The structure of these materials differs from that observed by Hou et al. [16] for $\text{Sr}_{1.6}\text{K}_{0.4}\text{FeMoO}_{6-\delta}$, which was refined as a mixture of two perovskite structures (SG: $Pm\text{-}3m$) with similar lattice parameters. GSAS analysis, shown in Table 1, demonstrates a linear reduction in the lattice parameter with reducing molybdenum content. The GSAS plots of samples after reduction in 5 % H_2/Ar at 1200 °C are shown in Fig. 5. The size of the molybdenum cation ($\text{Mo}^{6+} = 0.59$ Å, $\text{Mo}^{5+} = 0.61$ Å) at CN = 6 is

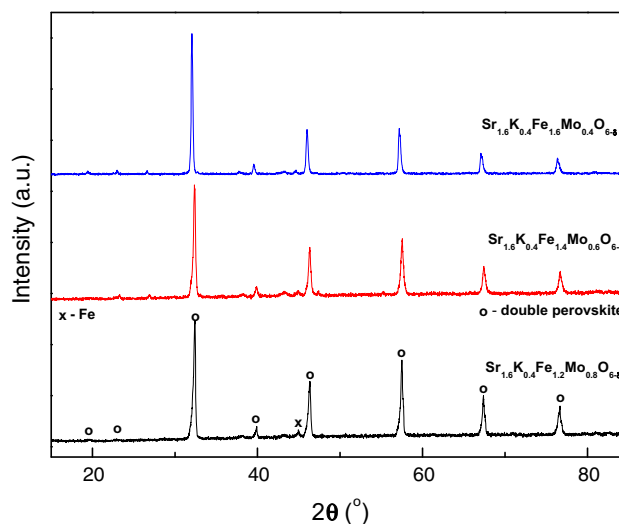


Fig. 4 XRD patterns of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4$ and 0.6) after reduction in 5 % H_2/Ar at 1200 °C

Table 1 Rietveld refinement and lattice parameters from GSAS refinement of Sr_{1.6}K_{0.4}Fe_{1+x}Mo_{1-x}O_{6-δ} (x = 0.2, 0.4, 0.6) after reduction at 1200 °C in 5 % H₂/Ar

		Sr _{1.6} K _{0.4} Fe _{1.2} Mo _{0.8} O _{6-δ}	Sr _{1.6} K _{0.4} Fe _{1.4} Mo _{0.6} O _{6-δ}	Sr _{1.6} K _{0.4} Fe _{1.6} Mo _{0.4} O _{6-δ}
	χ^2	2.941	1.868	1.758
	Rp (%)	11.45	8.82	7.23
	wRp (%)	8.60	6.70	5.57
	Space group	<i>Fm-3m</i>	<i>Fm-3m</i>	<i>Fm-3m</i>
	a (Å)	7.898(2)	7.881(2)	7.869(1)
	V (Å ³)	492.7(4)	489.6(4)	487.4(3)
	Fe (%)	3.3	2.4	2.1
	Space group	<i>Im-3m</i>	<i>Im-3m</i>	<i>Im-3m</i>
	a (Å)	2.871(1)	2.869(1)	2.864(1)
Sr/K	x	0.5	0.5	0.25
	y	0.5	0.5	0.25
	z	0.5	0.5	0.25
	U_{iso}	0.007(1)	0.014(1)	0.001(1)
Fe	x	0	0	0
	y	0	0	0
	z	0	0	0
	U_{iso}	0.005(3)	0.015(3)	0.005(2)
Fe/Mo	x	0.5	0.5	0.5
	y	0.5	0.5	0.5
	z	0.5	0.5	0.5
	U_{iso}	0.009(2)	0.061(4)	0.016(1)
O	x	0.237(2)	0.244(2)	0.252(1)
	y	0	0	0
	z	0.5	0.5	0
	U_{iso}	0.017(3)	0.032(2)	0.027(2)

generally smaller than that of iron in the *Fm-3m* structure (Fe_{LS}²⁺ = 0.61 Å, Fe_{HS}²⁺ = 0.78 Å, Fe_{LS}³⁺ = 0.55 Å, Fe_{HS}³⁺ = 0.645 Å) [28] which would intimate that a reduction in the lattice should occur with increasing molybdenum content. However, the presence of oxygen vacancies in the oxides may lead to enlarged lattice parameters. As the valency of these cations is known to alter with compositional modifications [29], it may be possible that the reduction in the molybdenum and iron ions could result in a modification of the valency of the B-site cations in Sr_{1.6}K_{0.4}Fe_{1+x}Mo_{1-x}O_{6-δ} (x = 0.2, 0.4, 0.6) with formation of oxygen vacancies, resulting in the observed increase in the lattice parameter. Further investigation using Mössbauer spectroscopy could determine the feasibility of this supposition. The proportion of the secondary iron phase increased with increasing iron content. The exsolved Fe on the surface may improve the catalytic activity of the anode which was observed in previous reports [30–32].

Conductivity of Sr_{1.6}K_{0.4}Fe_{1+x}Mo_{1-x}O_{6-δ} (x = 0.2, 0.4, 0.6)

Figure 6 shows the d.c. conductivity of all three samples in different atmospheres and redox history. The conductivity

of samples in air measured from samples obtained from firing in air at 1200 °C for 2 h (red) is in the range of 10⁻²–10⁻¹ S/cm indicating that the conductivity in an oxidising atmosphere is very low thus the materials are not suitable to be used as cathode materials for SOFCs.

To measure the conductivity of the sample after reduction at a high temperature, the samples were reduced in 5 %H₂/Ar at 1200 °C for 10 h first, then the conductivity was measured in the same atmosphere. Samples Sr_{1.6}K_{0.4}Fe_{1.2}Mo_{0.8}O_{6-δ} and Sr_{1.6}K_{0.4}Fe_{1.4}Mo_{0.6}O_{6-δ} exhibited high electronic conductivity in 5 %H₂/Ar, >40 Scm⁻¹, over the entire temperature range (black). The conductivity of sample Sr_{1.6}K_{0.4}Fe_{1.6}Mo_{0.4}O_{6-δ} was significantly lower after the same treatment, <1 S cm⁻¹ over the same temperature range. In this Fe-rich sample, at a very strong reducing atmosphere, majority of iron ions is reduced to Fe²⁺ (3d⁶) or Fe³⁺ (3d⁵) ions while the high electronic conductivity of iron ions relies on the Fe⁴⁺ which has a 3d⁴ outer orbital. Moving of electrons in the low spin of Fe²⁺ ions or high spin of Fe³⁺ ion is quite difficult resulting in reduced conductivity [8]. The reduction of Mo⁶⁺ (4 d⁰) to Mo⁵⁺(4 d¹) will introduce an electron in the 4d orbital which can be an electron charge carrier thus making the oxide conductive [8, 9, 23].

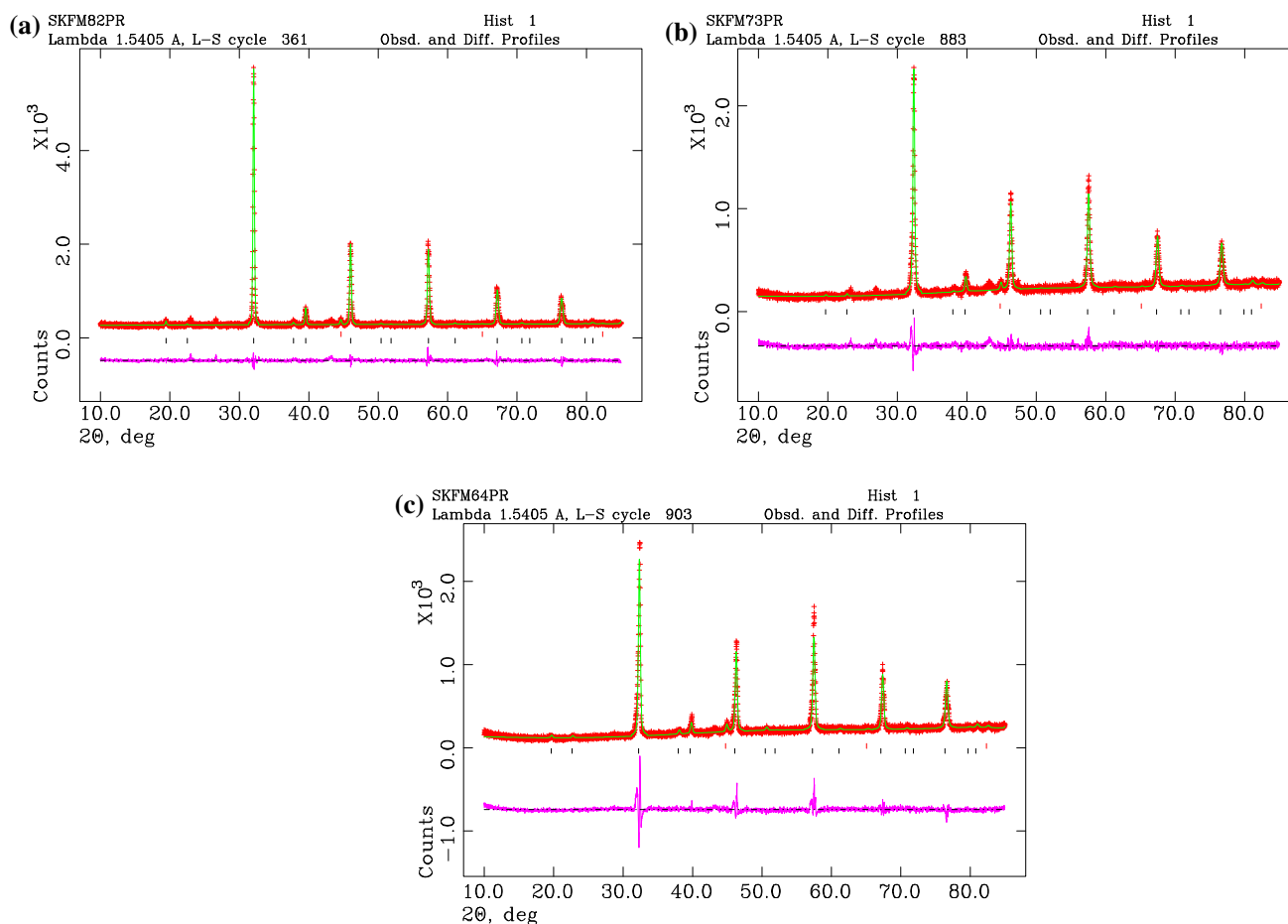


Fig. 5 GSAS plots of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ $x = 0.2$ (a), $x = 0.4$ (b) and $x = 0.6$ (c) after reduction in 5 % H_2/Ar at 1200 °C

In order to exam the effects of redox process on the dc conductivity, the pellets pre-reduced at 1200 °C in 5 % H_2/Ar for 10 h were re-oxidised in air at 700 °C for 12 h. The as-oxidised pellet was then held at 700 °C in 5 % H_2/Ar for 10 h to reach an equilibrium before conducting the conductivity measurement in 5 % H_2/Ar . As shown in Fig. 6, the conductivities of samples $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$ and $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ after the re-oxidation and reduction processes are in the range of 10^{-2} – 10^{-1} S/cm which is insufficient for these materials to be used as anode for SOFCs. Only sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ exhibits a conductivity of 2–3 S/cm which is just enough for planar design [6]. The compounds did not attain similar conductivities as those observed after reduction at 1200 °C in 5 % H_2/Ar , with the percentage retained increasing with iron content, 0.1 % for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$, 7.5 % for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ and 58.4 % for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ after the re-oxidation and reduction processes. As the ratio of both $\text{Fe}^{3+}/\text{Fe}^{2+}$ and $\text{Mo}^{6+}/\text{Mo}^{5+}$ is known to be

highly dependent on the reducing atmosphere and temperature [33, 34], it is expected that the reduction in the conductivity is a result of a lower degree of mixed valency due to the lower reduction temperature. In terms of total conductivity, sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ is the best among the investigated three samples. The possible reason is that, for Fe-rich sample with $x = 0.6$, in a reducing atmosphere, mainly it is iron that is reduced to $\text{Fe}^{3+}/\text{Fe}^{2+}$ ions with low conductivity as described above. In the Mo-rich sample with $x = 0.2$, the lattice is very strong and thus either iron or molybdenum can be reduced at mild condition (700 °C), leading to low conductivity as well. This could be the reason the conductivity of sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$ exhibits similar conductivity in both air and mild reducing atmosphere (Fig. 6a). In sample with $x = 0.4$, the perovskite lattice is less strong than sample with $x = 0.2$, thus Mo^{6+} is partially reduced to Mo^{5+} , resulting in high conductivity. SrMoO_4 can be reduced to SrMoO_3 (from Mo^{6+} to Mo^{4+}) at 750 °C in 4 % H_2/Ar [24]. In this study, we used

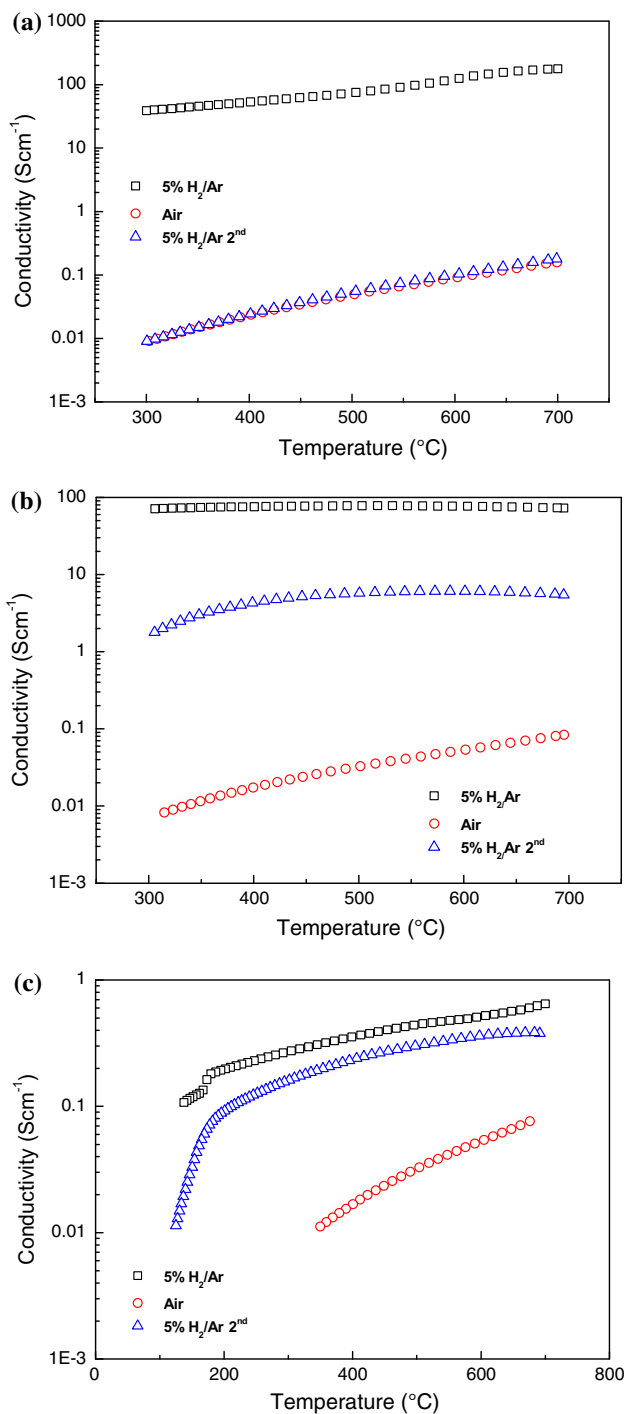


Fig. 6 Conductivity of Sr_{1.6}K_{0.4}Fe_{1+x}Mo_{1-x}O_{6-δ}, x = 0.2 (a), x = 0.4 (b) and x = 0.6 (c) in air (red), in 5 % H₂/Ar after reduction in 5 % H₂/Ar at 1200 °C (black), in 5 % H₂/Ar after further re-oxidation of the 1200 °C pre-reduced sample in air at 700 °C for 10 h then equilibrium in 5 %H₂/Ar at 700 °C for 10 h (blue) (Color figure online)

5 %H₂/Ar as the reducing reagent, and thus it is possible to partially reduce Mo⁶⁺ to Mo⁵⁺ at slightly lower temperature, say, 700 °C with the presence of a large amount of weak Fe–O bonds in the lattice.

STA on re-oxidation of Sr_{1.6}K_{0.4}Fe_{1+x}Mo_{1-x}O_{6-δ} (x = 0.2, 0.4, 0.6) in air

Re-oxidation of Sr_{1.6}K_{0.4}Fe_{1+x}Mo_{1-x}O_{6-δ} (x = 0.2, 0.4, 0.6) after reduction in 5 % H₂/Ar at 1200 °C caused an increase in weight proportional to the molybdenum content, as exhibited in Fig. 7a. The Mo-rich sample with x = 0.2 gained the greatest weight whilst the Fe-rich sample with x = 0.6 gained the fewest. This indicates iron, instead of molybdenum, has to be reduced to low valances in order to balance the positive and negative charge to form the single-phase perovskite oxide for sample Sr_{1.6}K_{0.4}Fe_{1.2}Mo_{0.8}O_{6-δ}. Re-oxidation was observed to begin between 400 and 600 °C, lower than the current operating temperature of SOFCs. Re-oxidation of the pre-reduced oxides is negligible at a temperature below 350 °C, which indicates the materials can be used as electrode materials for fuel cells or other electrochemical devices at low temperatures [35]. Re-oxidation of these materials occurs at a similar temperature to that of Sr₂FeMoO_{6-δ} [14],

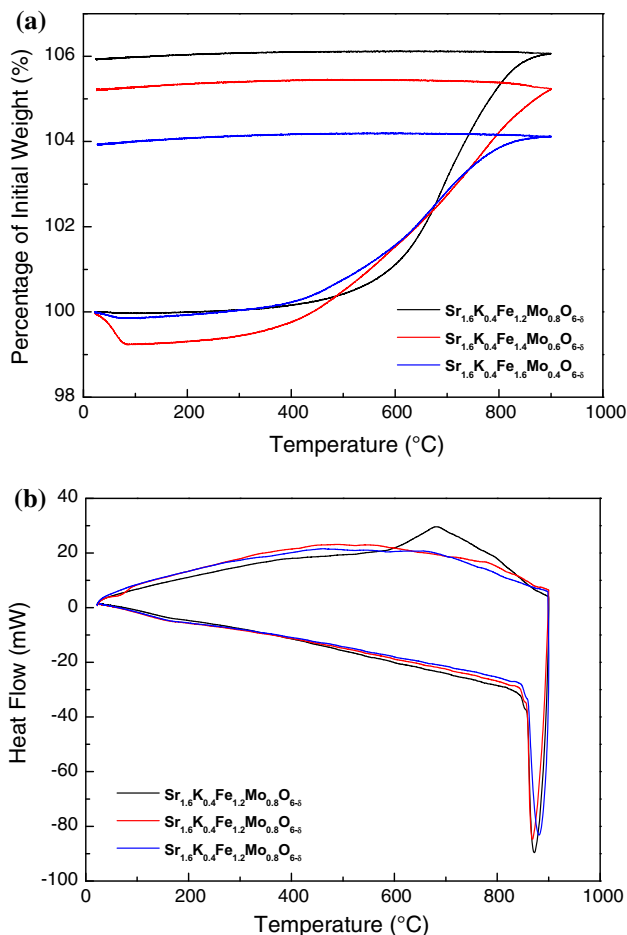


Fig. 7 Thermogravimetric analysis (a) and differential scanning calorimetry (b) of Sr_{1.6}K_{0.4}Fe_{1+x}Mo_{1-x}O_{6-δ} (x = 0.2, 0.4 and 0.6) in air after reduction in 5 % H₂/Ar at 1200 °C

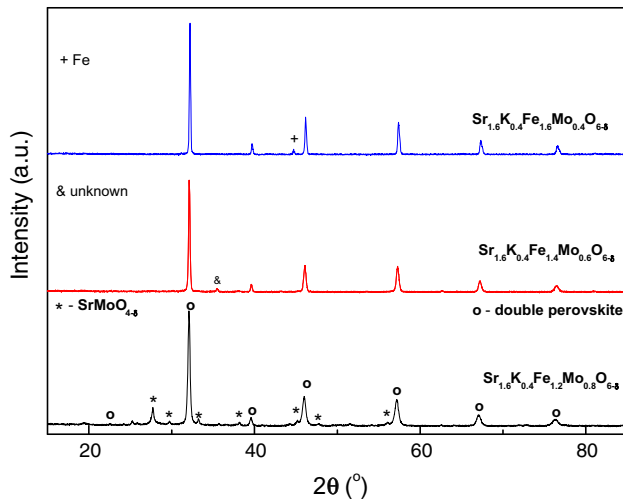


Fig. 8 XRD patterns of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4$ and 0.6) after re-oxidation of $1200\text{ }^\circ\text{C}$ pre-reduced samples in air at $700\text{ }^\circ\text{C}$ for 10 h

suggesting a minimal modification of the material redox stability after partial replacement of strontium by potassium at the A-site of double perovskite $\text{Sr}_2\text{FeMoO}_{6-\delta}$. A significant deviation was observed between 600 and $800\text{ }^\circ\text{C}$ by differential scanning calorimetry, shown in Fig. 7b, for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$, which can be attributed to the formation of the secondary SrMoO_4 phase observed in the XRD pattern after redox cycling, Fig. 8.

XRD of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) after redox cycling

After the samples were reduced in $5\% \text{H}_2/\text{Ar}$ at $1200\text{ }^\circ\text{C}$ for 10 h, they were further oxidised in air at $700\text{ }^\circ\text{C}$ for 10 h and then cooled down to room temperature in air. XRD patterns of re-oxidised $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) are shown in Fig. 8. Trace amount of Fe (PDF: 6-696) phase was observed for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$ after redox cycling, with an increase from

Table 2 Rietveld refinement and lattice parameters from GSAS refinement of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) after re-oxidation and re-reduction at $700\text{ }^\circ\text{C}$ in $5\% \text{H}_2/\text{Ar}$ of the compounds previously reduced at $1200\text{ }^\circ\text{C}$ in $5\% \text{H}_2/\text{Ar}$

	$\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$	$\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$	$\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.6}\text{Mo}_{0.4}\text{O}_{6-\delta}$
χ^2	4.541	1.714	1.547
Rp (%)	10.44	6.41	5.95
wRp (%)	7.59	4.92	4.72
Space group	<i>Fm-3m</i>	<i>Fm-3m</i>	<i>Fm-3m</i>
a (Å)	7.862(2)	7.872(1)	7.861(1)
V (Å ³)	486.0(3)	487.9(5)	485.8(2)
Secondary phase	SrMoO_4	–	Fe
Space group	<i>I4₁/m</i>	–	<i>Im-3m</i>
Second phase (%)	13	–	3.2
a (Å)	5.394(1)	–	2.864(5)
b (Å)	5.394(1)	–	2.864(5)
c (Å)	12.013(3)	–	2.864(5)
Sr/K			
	x	0.5	0.5
	y	0.5	0.5
	z	0.5	0.5
	U_{iso}	0.026(1)	0.003(1)
Fe			
	x	0	0
	y	0	0
	z	0	0
	U_{iso}	0.010(3)	0.019(3)
Fe/Mo			
	x	0.5	0.5
	y	0.5	0.5
	z	0.5	0.5
	U_{iso}	0.031(5)	0.041(3)
O			
	x	0.231(2)	0.238(1)
	y	0	0
	z	0.5	0
	U_{iso}	0.061(4)	0.032(4)

2.1 % (Table 1) to 3.2 % (Table 2) in the phase fraction, whilst a SrMoO_4 phase (PDF: 01-085-0809) was observed for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.2}\text{Mo}_{0.8}\text{O}_{6-\delta}$. Sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ was dominated by double perovskite structure (SG: $Fm\bar{3}m$) although a weak peak at $\sim 35^\circ$ cannot be indexed by known compounds (Fig. 8) indicating it is most redox stable. GSAS analysis, shown in Table 2, demonstrated a reduction in the lattice parameters after redox cycling due to the oxidation of Fe and Mo ions, although the lattice parameters of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) samples exhibit no observable trend, reflecting the complexity of two multi-valent elements at the B-sites.

Introduction of potassium into $\text{Sr}_2\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) appears to have a negligible effect on the formability of these compounds, with the variation of the iron content exhibiting greater influence on material formability. Sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ that is most redox stable also retains reasonably high electrical conductivity which is a potential anode for SOFCs.

Conclusion

Potassium substitution into $\text{Sr}_2\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) with the intention of increasing the formability and ionic conductivity was successful only for $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ composition. Synthesis of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) was achieved above 700°C in 5 % H_2/Ar , albeit with the formation of some impurity phases. Phase stability upon redox cycling was observed for sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$.

Redox cycling of $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1+x}\text{Mo}_{1-x}\text{O}_{6-\delta}$ ($x = 0.2, 0.4, 0.6$) demonstrates a strong dependence on high temperature reduction to achieve high conductivities, with re-reduction at lower temperatures attaining between 0.1 and 58.4 % of the initial conductivity observed after high-temperature reduction. The reliance of these compounds on high-temperature reduction is expected to limit their utility as SOFC anode materials, as the vulnerability to oxidation can have disastrous consequence for fuel cell durability. However, the re-oxidation process is negligible at a temperature below 350°C indicates they can be used as electrode materials for low temperature electrochemical devices including low temperature fuel cells. In the investigated new oxides, sample $\text{Sr}_{1.6}\text{K}_{0.4}\text{Fe}_{1.4}\text{Mo}_{0.6}\text{O}_{6-\delta}$ that is most redox stable also retains reasonably high electrical conductivity, $\sim 70\text{ S/cm}$ after reduction at 1200°C and $2\text{--}3\text{ S/cm}$ after redox cycling at 700°C , indicating it is a potential anode for SOFCs.

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