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Author(s)	Ryoki, Akiyuki; Ida, Daichi; Terao, Ken
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**Scattering function of semi-rigid cyclic polymers analyzed in terms of wormlike rings: Cyclic amylose tris(phenylcarbamate) and cyclic amylose tris(*n*-butylcarbamate)**

**Akiyuki Ryoki,<sup>1</sup> Daichi Ida,<sup>2</sup> and Ken Terao<sup>1</sup>**

<sup>1</sup> Department of Macromolecular Science, Graduate School of Science, Osaka University, Osaka, Japan.

<sup>2</sup> Department of Polymer Chemistry, Graduate School of Engineering, Kyoto University, Kyoto, Japan.

Correspondence: Dr K Terao, Department of Macromolecular Science, Graduate School of Science, Osaka University, 1-1 Machikaneyama-cho, Toyonaka, Osaka 560-0043, Japan.

E-mail: [kterao@chem.sci.osaka-u.ac.jp](mailto:kterao@chem.sci.osaka-u.ac.jp)

Running Head: Scattering function of cyclic polysaccharides

## ABSTRACT

Recently reported data of the particle scattering function  $P(q)$  with the magnitude  $q$  of the scattering vector for rigid cyclic amylose tris(phenylcarbamate) (cATPC) and cyclic amylose tris(*n*-butylcarbamate) (cATBC) in different solvents were analyzed in terms of a novel simulation method based on the Kratky–Porod wormlike chain model. Although similar wormlike chain parameters were evaluated for both relatively flexible cyclic chains and the corresponding linear polymers, an appreciable decrease in the chain stiffness and slight extension of the local helical structure were found for cyclic chains with a higher chain stiffness. The difference in the wormlike chain parameters between the cyclic and linear chains cannot be realized in the previously reported molar mass dependence of the radius of gyration. This suggests that analyses of  $P(q)$  are decisively important to understand the conformational properties of rigid and/or semiflexible cyclic chains in solution if the molar mass range of the cyclic polymer samples is limited.

**Keywords:** scattering function; cyclic polysaccharide; Monte Carlo simulation; small-angle X-ray scattering; amylose derivative; chain stiffness; Kuhn segment number

The local conformation of ring or cyclic chains is substantially the same as that of their linear analogues if the chain length is sufficiently longer than the Kuhn segment length  $\lambda^{-1}$ , which is the chain stiffness parameter of the Kratky–Porod wormlike chain.<sup>1</sup> The discrepancy between cyclic and linear polymers may become significant upon shortening or stiffening the main chain because the difference in the curvature distribution becomes prominent owing to the topological constraint. Little is known, however, about the chain stiffness (or length) dependent local conformational change, except for the case of the super-helical structure of cyclic DNA,<sup>2</sup> although abundant studies have examined the physical properties of rather flexible cyclic polymers including polystyrene,<sup>3</sup> polydimethylsiloxane,<sup>4</sup> and polysaccharides.<sup>5</sup> This is likely because cyclic polymers are synthesized by end-linking or ring expansion methods. A variety of novel cyclic polymers have recently been reported not only for fundamental synthetic studies but also for building blocks of higher order structures consisting of amphiphilic block copolymers.<sup>6</sup> Considering that dense macrocyclic comb polymers that should have relatively stiff main chains form tube-like complexes,<sup>7</sup> the chain stiffness effect on the dimensional properties and intermolecular interactions should be an important topic.

Semiflexible and rigid cyclic polymers were recently obtained by means of ‘stiffening’ the main chain of rather flexible cyclic chains. Cyclic amylose tris(phenylcarbamate) (cATPC)<sup>8</sup> and cyclic amylose tris(*n*-butylcarbamate) (cATBC)<sup>9</sup> prepared from enzymatically synthesized cyclic amylose<sup>10</sup> behave as semiflexible and/or rigid cyclic macromolecules in solution. The molar mass dependence of the mean square radius of gyration  $\langle S^2 \rangle$  for cATPC and cATBC in various solvents is fairly explained by current theories for the cyclic wormlike chain<sup>11</sup> with the parameters,  $\lambda^{-1}$  and the helix pitch (or helix rise)  $h$  per unit chain length, determined for the corresponding linear chains.<sup>8a,9</sup> The experimental values for cATPC in some ketones and esters

are, however, appreciably smaller than the theoretical values calculated with the parameters so determined.<sup>8b</sup> This suggests that both the local helical structure and the chain stiffness depend on the molar mass of the cyclic polymer chains. Although the particle scattering function  $P(q)$  was determined as a function of the magnitude  $q$  of the scattering vector for the two cyclic polymers in different solvents,<sup>8-9</sup> appropriate theories or simulation data are not yet available for semi-rigid cyclic chains. Recently, Ida et al.<sup>12</sup> calculated  $P(q)$  for wormlike rings with a variety of reduced chain lengths, that is, the ratio of the contour length  $L$  to  $\lambda^{-1}$  or the Kuhn segment number  $N_K$ , allowing us to compare them with experimental  $P(q)$  data. We thus reanalyzed the previously published  $P(q)$  data to discuss the difference between the molecular structures of linear and cyclic amylose derivatives in solution.

The previously reported  $P(q)$  data<sup>8-9</sup> for cATPC in 1,4-dioxane (DIOX), 2-ethoxyethanol (2EE), methyl acetate (MEA), ethyl acetate (EA), and 4-methyl-2-pentanone (MIBK) and cATBC in methanol (MeOH), 2-propanol (2PrOH), and tetrahydrofuran (THF) were determined at the BL40B2 beamline in SPring-8 and the BL-10C beamline in KEK-PF. They were obtained for five cATPC and nine cATBC samples ranging in weight-average molar mass  $M_w$  between  $1.25 \times 10^4 \text{ g mol}^{-1}$  and  $1.49 \times 10^5 \text{ g mol}^{-1}$  for cATPC and  $1.60 \times 10^4 \text{ g mol}^{-1}$  and  $1.11 \times 10^5 \text{ g mol}^{-1}$  for cATBC, corresponding to numbers of saccharide units between 24 and 290 for the former polymer and between 35 and 240 for the latter. The dispersity index  $\mathcal{D}$  defined as the ratio of  $M_w$  to the number-average molar mass was estimated to be 1.05 – 1.23.

Monte Carlo simulations were performed using the method reported by Ida et al.<sup>12a,13</sup> to calculate the particle scattering function  $P_0(q)$  of wormlike rings without chain thickness as a function of  $N_K$  ( $\equiv \lambda L$ ). A discrete wormlike chain model originally proposed by Frank-Kamenetskii et al.<sup>14</sup> was used with the bond number being 200. For each  $N_K$ ,  $10^5$  configurations

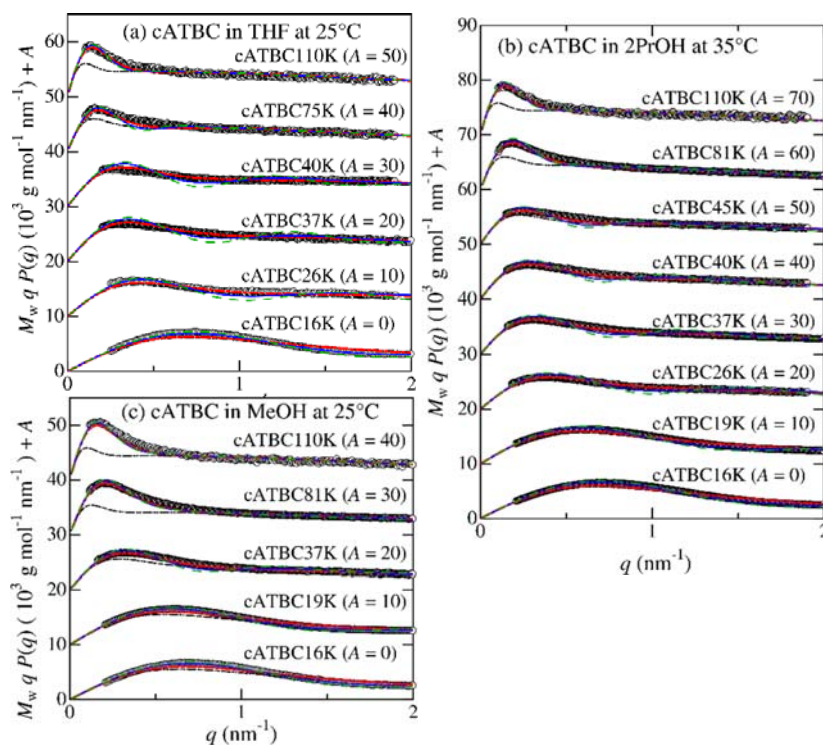
were generated to obtain an ensemble average with the appropriate monitoring steps. According to our previous study,<sup>8-9</sup> the chain thickness significantly affects  $P(q)$  at a relatively high  $q$  range. The relationship can be considered by means of the touched bead model<sup>15</sup> as follows.

$$P(q) = 9 \left( \frac{2}{qd_b} \right)^6 \left( \sin \frac{qd_b}{2} - \frac{qd_b}{2} \cos \frac{qd_b}{2} \right)^2 P_0(q) \quad (1)$$

Since actual cATPC and cATBC samples have finite molar mass distributions, the  $z$ -average particle scattering function  $P_z(q)$  with a log-normal distribution was calculated numerically to compare the experimental data.

The calculated  $P_z(q)$  or  $P(q)$  values with the best fit parameters  $\lambda^{-1}$  and  $h$  for cATBC in three solvents are plotted in Figure 1. The calculated values of  $P_z(q)$  with appropriate  $\mathcal{D}$  [1.05 (blue) and 1.20 (red)] well explain the behavior of the experimental data, while those of  $P(q)$  for the monodisperse ring (green) slightly deviate downward in the middle- $q$  range from the experimental values for the lower- $M_w$  samples. Good agreement was observed between the simulation and experimental data for cATPC in six solvent systems (see Figure S1 in the supporting information). We note that the values of  $\lambda^{-1}$  may be determined without ambiguity only for the samples with high  $M_w$ , for which the theoretical values for the rigid ring calculated with the appropriate  $h$  value (black dot-dashed curves) underestimate  $P(q)$  at approximately  $q = 0.2 \text{ nm}^{-1}$ . We then adopted the  $\lambda^{-1}$  value determined for the samples with high  $M_w$  to calculate  $P_z(q)$  or  $P(q)$  for the samples with low  $M_w$ . We also note that  $P_z(q)$  and  $P(q)$  become insensitive

to changes in  $\lambda^{-1}$  with decreasing  $M_w$ . The other two parameters,  $h$  and  $d_b$ , were uniquely determined for more samples, except for the two lowest  $M_w$  samples, for both cATBC and cATPC. The latter parameter  $d_b$  was consistent with those for the corresponding linear chain within  $\pm 11\%$  if we estimate  $d_b$  for the linear polymer from the literature chain diameter  $d$  values<sup>16</sup> for the cylinder model by means of the known relationship  $d_b = 1.118 d$ .<sup>17</sup> The negligible difference between the  $P(q)$  data for the discrete wormlike chain and those for the continuous rigid cyclic chain in the high  $q$  region supports that the current simulation results are substantially the same as those for the continuous chain.



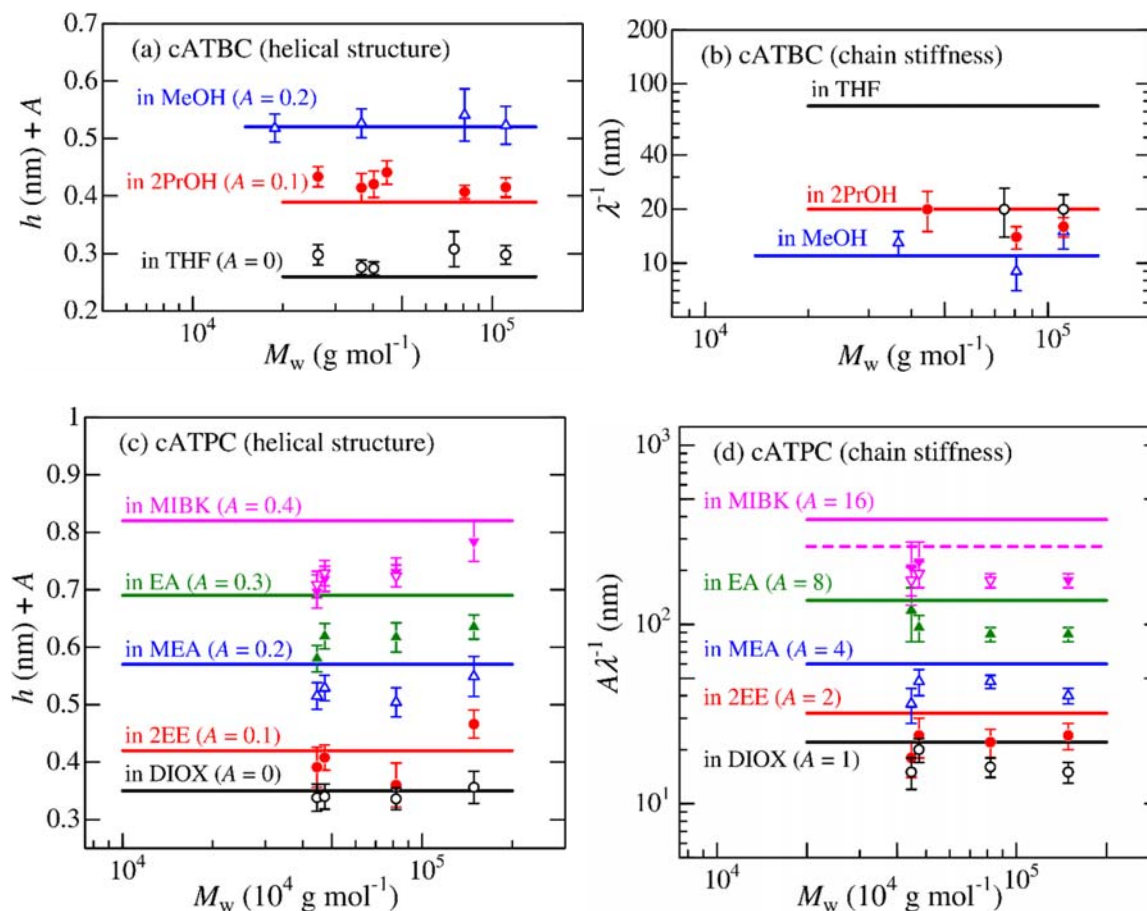
**Figure 1** Comparison between the experimental  $P(q)$  data<sup>9</sup> (unfilled circles) and the theoretical values for the indicated cATBC samples in tetrahydrofuran (THF) at 25 °C (a), in 2-propanol (2PrOH) at 35 °C (b), and in methanol (MeOH) at 25 °C (c). Dashed, solid, and dot-dashed curves are the simulation values for the wormlike ring model with the dispersity index  $\mathcal{D} = 1.00$ ,

1.05, and 1.20, respectively. The double-dot-dashed black curve represents the theoretical values for the rigid ring ( $D = 1.2$ ). The ordinate values are shifted by  $A$  for clarity.

The determined  $\lambda^{-1}$  and  $h$  values are plotted against  $M_w$  in Figure 2. The former parameter  $\lambda^{-1}$  for cATBC in THF is much smaller than that for the linear ATBC. This difference is clearly recognized from the fact that the calculated values of  $P_z(q)$  and  $P(q)$  for cATBC in THF with the parameters determined for linear ATBC ( $h = 0.28$  nm and  $\lambda^{-1} = 75$  nm) are appreciably smaller than the experimental values, as shown in Figure S2 in the supporting information. On the other hand, the theoretical values of  $\langle S^2 \rangle$  for wormlike rings calculated from the Shimada–Yamakawa equation<sup>11</sup> with the parameters determined currently for cATBC ( $h = 0.33$  nm and  $\lambda^{-1} = 20$  nm) and those calculated using the previously estimated values ( $h = 0.28$  nm and  $\lambda^{-1} = 75$  nm)<sup>9</sup> assuming  $\lambda^{-1}$  for linear ATBC<sup>16c</sup> are indistinguishable in the investigated  $M_w$  range, as shown in Figure S3 in the supporting information. This indicates that the wormlike chain parameters for this system cannot be determined only from the reported  $\langle S^2 \rangle$  data. When we consider that the wormlike chain parameters of infinitely long cyclic chains should be the same as those for the corresponding linear chains, as mentioned in the Introduction, it is expected that a molar mass dependent chain stiffness could be observed in the higher molar mass range. In the case of  $h$ , its values for most cases of cyclic chains are fairly identical to those of the corresponding linear chains, except for cATPC in MIBK, cATPC in EA, cATBC in THF, and cATBC in 2PrOH. The former two polymer-solvent systems are rather specific because hydrogen bonding solvent molecules extend and stiffen the linear ATPC, while in the latter cases, the rigid helical main chain of ATBC in THF (or 2PrOH) is extended by the cyclization similarly to the case with bended helical springs. This local structural change also reduces the chain stiffness of the main



chain of cATBC. It is, however, noted that this local structural change is insensitive to the solution infrared absorption, which reflects the intramolecular hydrogen bonds.<sup>9</sup>

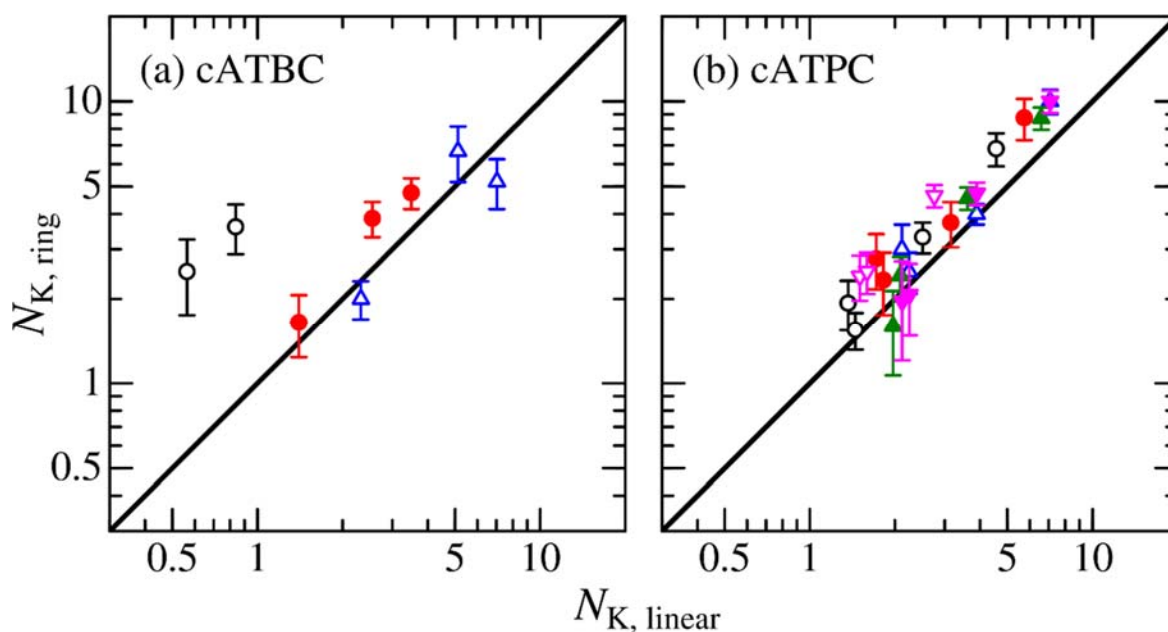


**Figure 2** Weight-average molar mass  $M_w$  dependence of the wormlike chain parameters for cATBC and cATPC in solution. In (a) and (b), the unfilled circles, filled circles, and triangles indicate the parameters for cATBC in THF at 25 °C, in 2PrOH at 33 °C, and in MeOH at 25 °C, respectively. The solid lines are  $h$  or  $\lambda^{-1}$  for the corresponding linear chain. In (c) and (d), the unfilled circles, filled circles, unfilled triangles, filled triangles, unfilled inverted triangles, and filled inverted triangles represent the parameters for cATPC in DIOX at 25 °C, in 2EE at 25 °C, in MEA at 25 °C, in EA at 33 °C, and in MIBK at 25 °C and 58 °C, respectively. The solid and

dashed magenta lines in (d) are the  $\lambda^{-1}$  values for linear ATPC in MIBK at 25 °C and 58 °C, respectively. The ordinate values are shifted by  $A$  for clarity.

The Kuhn segment numbers  $N_{K,\text{ring}}$  for cATBC and cATPC are plotted against that for the corresponding linear chain  $N_{K,\text{linear}}$  in Figure 3. Note that  $N_{K,\text{linear}}$  was calculated from the wormlike chain parameters reported for the corresponding linear chain with the same  $M_w$ . This number is well-known as a universal parameter that mainly reflects the chain conformation of unperturbed linear polymer chains in solution. Interestingly, in the range of  $N_{K,\text{linear}} > 1.5$ ,  $N_{K,\text{ring}}$  is fairly close to  $N_{K,\text{linear}}$ , in contrast to the case of cATBC in THF in the range of  $N_{K,\text{linear}} < 1$ , for which  $\lambda^{-1}$  is clearly smaller than that of the linear ATBC [see Figure 2(b)]. The threshold value 1.0 – 1.5 of  $N_K$  is close to the value at which the (angle-independent) ring closure probability for the (linear) wormlike chain significantly increases.<sup>18</sup> It is, however, noted that  $N_{K,\text{ring}}$  for ATPC in EA and MIBK is incidentally close to  $N_{K,\text{linear}}$ , although  $h$  and  $\lambda^{-1}$  for cATPC are smaller than those for linear ATPC [see Figure 2(c) and (d)]. As mentioned above and in previous papers,<sup>16b,19</sup> the origin of the chain stiffness of these systems is different from that of other systems. We may thus conclude that some strain effects for cyclic chains arising from the relaxation of the internal-rotation-angle distribution in their main chain become appreciable for shorter (stiffer) cyclic chains with small  $N_K$ . In contrast,  $\lambda^{-1}$  for the cyclic chain is substantially the same as that for the corresponding linear chain when  $N_K$  is larger, and the local helical structure of the cyclic chain is almost identical to that of the corresponding linear chain. Although such a threshold value of  $N_K$  may depend on the polymer-solvent system and/or the origin of the chain stiffness, for example, differences in  $h$  and  $\lambda^{-1}$  between cyclic and linear

chains were reported for ATPC in EA and MIBK [see Figure 2(c) and (d)] in the range of  $N_K$  from 0.42 to 6.6,<sup>8b</sup> such strain effects seem to make cyclic chains more flexible than linear chains as far as amylose carbamates are concerned. Unfortunately, however, it is still not clear whether rigid cyclic chains should have a smaller  $\lambda^{-1}$  compared to the corresponding linear chains. It is desirable to test other rigid cyclic chain systems to clarify this issue.



**Figure 3** Double logarithmic plots of  $N_{K,ring}$  vs  $N_{K,linear}$  for (a) cATBC and (b) cATPC in various solvents. The symbols are the same as those used in Figure 2. The solid lines indicate  $N_{K,ring} = N_{K,linear}$ .

We reanalyzed our recently published scattering function data for cATBC in three and cATPC in five solvent systems in terms of the cyclic wormlike chain to determine the contour length per residue  $h$  and the Kuhn segment length  $\lambda^{-1}$  (stiffness parameter). While there was no

significant molar mass dependence of the wormlike chain parameters, a clearly small  $\lambda^{-1}$  was estimated for cATBC in THF, even though the difference cannot be distinguishable from the radius of gyration, showing that the analysis of the particle scattering function of the cyclic chain is important to determining the chain shape of the rigid cyclic chains in solution.

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## Graphical Abstract

Scattering function of semi-rigid cyclic polymers analyzed in terms of wormlike rings: Cyclic amylose tris(phenylcarbamate) and cyclic amylose tris(*n*-butylcarbamate)

Akiyuki Ryoki, Daichi Ida, and Ken Terao

