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Relation	

Electronic structure of Kondo lattice compounds YbNi_3X_9 ($X = \text{Al}, \text{Ga}$) studied by hard x-ray spectroscopy

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We have performed hard x-ray photoemission spectroscopy (HAXPES) for Yb-based Kondo lattice compounds; an antiferromagnetic heavy-fermion system YbNi_3Al_9 and a valence fluctuation system YbNi_3Ga_9 . The Yb $3d_{5/2}$ spectra of YbNi_3Ga_9 showed both Yb^{2+} and Yb^{3+} -derived structures indicating strong valence fluctuation, and the intensity of Yb^{2+} (Yb^{3+}) structures gradually increased (decreased) on cooling. The Yb $3d_{5/2}$ spectra of YbNi_3Al_9 mostly consisted of Yb^{3+} -derived structures and showed little temperature dependence. The Yb valences of YbNi_3Ga_9 and YbNi_3Al_9 at 22 K were evaluated to be 2.43 and 2.97, respectively. Based on the results of the Ni $2p$ and valence-band HAXPES spectra together with soft x-ray valence-band spectra, we described that the difference of physical properties of YbNi_3X_9 ($X = \text{Al}, \text{Ga}$) is derived from the differences of the $4f$ -hole level relative to the Fermi level (E_F) and Ni $3d$ density of states at E_F . The HAXPES results on the Yb valences were consistent with those obtained by x-ray absorption spectroscopy using the partial fluorescence yield mode and resonant x-ray emission spectroscopy at the Yb L_3 edge.

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I. INTRODUCTION

The $4f$ orbitals in rare-earth ions are almost localized inside the $5d$ and $6s$ valence shells. However, the tails of $4f$ orbitals extend out of the ions and may hybridize with conduction electrons. In strongly correlated $4f$ -electron systems, the hybridization between the $4f$ and conduction electrons (c - f hybridization) produces a wide variety of physical phenomena such as heavy-fermion behavior, magnetic transition, valence fluctuation, and unconventional superconductivity.¹ A large number of studies have been performed for Ce compounds because of simplicity of their $4f^1$ electron configuration. The studies for the Yb compounds with the $4f^1$ hole configuration (Yb^{3+}) are limited in number because the known Yb-compounds have often shown strong hybridization with fully occupied Yb $4f$ orbitals (Yb^{2+}). Recently, the first Yb-based superconductor β - YbAlB_4 with $T_C = 80$ mK was discovered.² This compound is considered to be located near a quantum critical point, showing strong valence fluctuation with the mean Yb valence of $\text{Yb}^{2.75+}$.³ It is suggested that the strong valence fluctuation is one possible driving force of the unconventional superconductivity in β - YbAlB_4 .

Recently, Ohara *et al.* has succeeded in synthesizing single crystals of new Yb-based Kondo lattice compounds YbNi_3X_9 ($X = \text{Al}, \text{Ga}$) whose physical properties strongly depend on X atoms.^{4,5} YbNi_3Al_9 is an antiferromagnetic heavy-fermion system with the Néel temperature of $T_N = 3.4$ K. The magnetic susceptibility exhibits the Curie-Weiss behavior above 80 K and an effective magnetic moment is estimated to

be $\mu_{\text{eff}} \sim 4.37\mu_B/\text{Yb}$, corresponding to the Yb valence close to $3+$. In contrast, the magnetic susceptibility of YbNi_3Ga_9 shows a typical valence fluctuation behavior with the Pauli paramagnetic ground states and it reaches a broad maximum around 200 K. The Kondo temperature T_K changes from a few K to several hundred K by substituting the X atom from Al to Ga.

YbNi_3X_9 possesses the trigonal ErNi_3Al_9 -type crystal structure with a space group of $R\bar{3}2$.^{6,7} The Ni_3 , X_3 and Yb_2X_3 layers with a triangular lattice stack along the c axis to be close packed. The Yb ion in the Yb_2X_3 layer forms a honeycomb lattice with a X_3 cluster at the center. The lattice volume varies less than 1% on the substitution of X atoms of YbNi_3X_9 . Furthermore, the Al and Ga ions are isovalent with their nominal valence electron configurations of $(3s^23p^1)$ and $(4s^24p^1)$, respectively. In spite of the same crystal structure and similar valence electrons, YbNi_3Al_9 and YbNi_3Ga_9 are located at a distance in the Doniach phase diagram.⁸ Therefore YbNi_3X_9 is suitable to investigate the difference in the electronic structures between an antiferromagnetic heavy-fermion system and a valence fluctuation system. As a similar system, we may refer to YbM_2Si_2 ($M = \text{Rh}, \text{Ir}$),⁹ and the other isostructural series of YbMCu_4 ($M = \text{Au}, \text{Ag}, \text{Cd}, \text{In}, \text{Mg}, \text{Tl}, \text{Zn}$)^{10,11} with a wide variety of physical phenomena.

In this study, we carried out hard x-ray photoemission spectroscopy (HAXPES) with $h\nu = 5.95$ keV in order to clarify the electronic structure of YbNi_3X_9 . The Yb $3d_{5/2}$ HAXPES spectra show that the Yb valence of YbNi_3Al_9 is almost trivalent, while that of YbNi_3Ga_9 strongly fluctuates in

agreement with the magnetic susceptibility measurement.^{4,5} We report the temperature-dependent Yb valences evaluated from the fitting analysis of the Yb $3d_{5/2}$ spectra. The Yb valence of YbNi_3Ga_9 at 300 K is $z \sim 2.59$ and monotonically decreases to $z \sim 2.43$ at 22 K, while that of YbNi_3Al_9 almost stays at $z \sim 2.97$ between 300 and 22 K. These HAXPES results are consistent with those of the x-ray absorption spectroscopy using the partial fluorescence yield mode (PFY-XAS) and resonant x-ray emission spectroscopy (RXES) at the Yb L_3 edge, except that the Yb valences obtained from the PFY-XAS and RXES experiments are closer to Yb^{3+} . We discuss the physical properties of YbNi_3X_9 from a viewpoint of the electronic structure based on the valence-band and Ni $2p$ HAXPES spectra and describe that the $4f$ -hole level and amount of the Ni $3d$ density of states (DOS) at the Fermi level (E_F) are keys to understand their different properties of YbNi_3X_9 .

II. EXPERIMENTS

Single crystals of YbNi_3X_9 were synthesized by the self-flux method as described in the literature.^{4,5} The x-ray powder diffraction indicated the samples were in a single phase without any impurity phases. The residual resistivity of less than $1 \mu\Omega\text{cm}$ shows high quality of the samples.⁴

The HAXPES experiments with $h\nu = 5.95$ keV were performed at undulator beamline BL15XU¹² of SPring-8. Synchrotron radiation was monochromatized with a Si (111) double-crystal monochromator and a post Si (333) channel-cut monochromator.¹³ The HAXPES spectra were taken by using a hemispherical analyzer (VG SCIENTA R4000) and the overall energy resolution was set to 240 meV for the Yb $3d_{5/2}$ and valence-band spectra and 150 meV for the Ni $2p_{3/2}$ spectra. Clean surfaces of the samples were obtained by fracturing *in situ* under the base pressure of 1×10^{-7} Pa at room temperature. The spectra were measured in the temperature range of 300–22 K. The binding energy of the spectra was calibrated by the Fermi edge of a gold film.

PFY-XAS and RXES measurements were performed at the Taiwan beamline BL12XU^{14–16} of SPring-8. A Johann-type spectrometer equipped with a spherically bent Si(620) crystal

(radius of ~ 1 m) was used to detect the Yb $L\alpha_1$ ($3d_{5/2} \rightarrow 2p_{3/2}$) and Raman emissions with a solid state detector (XFlash 1001 type 1201). The overall energy resolution was estimated to be about 1 eV around the emitted photon energy of 7.4 keV.

III. RESULTS

Figure 1(a) shows the temperature dependence of the Yb $3d_{5/2}$ HAXPES spectra of YbNi_3Ga_9 measured between 300 and 22 K. The Yb $3d_{5/2}$ spectral features of YbNi_3Ga_9 is similar to that of the other Yb compounds.^{3,17–20} The Yb^{2+} component is observed as a prominent peak at 1520 eV, while the Yb^{3+} components exist at 1524–1534 eV exhibiting multiplet structures arising from the Coulomb interaction between the $3d$ and $4f$ holes in the electron configuration of the $3d^9 4f^{13}$ final states. A broad structure around 1543 eV is attributed to a plasmon satellite accompanied by the $\text{Yb}^{3+} 3d_{5/2}$ photoemission process.^{3,17} The pronounced spectral intensity of the Yb^{2+} -derived structure exhibits a clear experimental evidence of the strong valence fluctuation in YbNi_3Ga_9 , as inferred from the temperature dependence of the magnetic susceptibility. With decreasing temperature, the intensity of the Yb^{2+} (Yb^{3+}) structure gradually increases (decreases), indicating that the Yb valence is closer toward Yb^{2+} as in the other valence fluctuating compounds.^{3,21,22}

Here, we comment on an additional structure around 1536 eV shown by a vertical bar in Fig. 1(a). This structure was not observed for YbInCu_4 (see Ref. 17), $\text{Yb}_{1-x}\text{Lu}_x\text{B}_{12}$ (see Ref. 18), and YbNi_3Al_9 [see Fig. 2(a)], and cannot be explained by the atomic multiplet calculations. Its intensity gradually increases with decreasing temperature along with the Yb^{2+} peak intensity, not with the Yb^{3+} -derived spectral intensity, indicating a correlation to the Yb^{2+} states. The structure at this energy region was reported in the Yb $3d$ spectrum of YbAl_3 in Ref. 20. It was attributed to the Yb^{2+} -derived plasmon satellite from the fact that the binding energy of the structure is higher by $\Delta E = 15.2$ eV than that of the Yb^{2+} peak, and it coincides well with the plasmon peak separation of $\Delta E = 15.2$ eV in the Yb^{3+} -derived structures. In the case of YbNi_3Ga_9 , the structure at 1536 eV is shifted by $\Delta E \sim 16$ eV from the

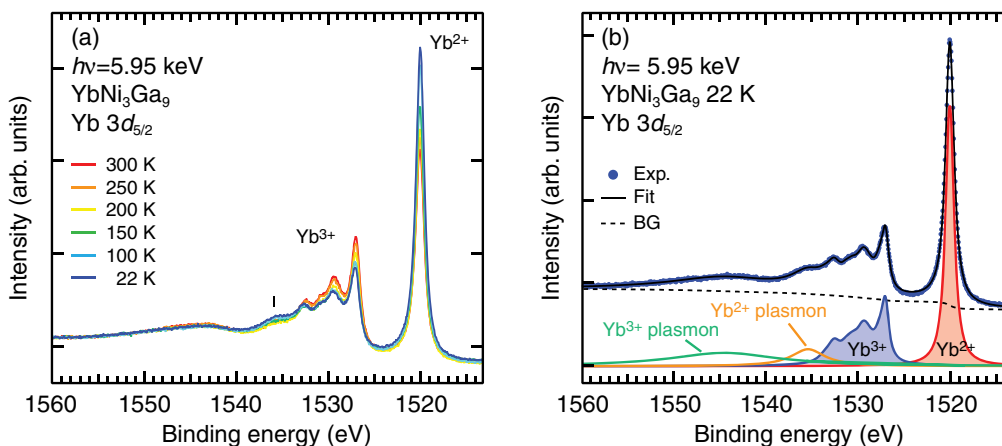


FIG. 1. (Color online) (a) Temperature dependence of the Yb $3d_{5/2}$ HAXPES spectra of YbNi_3Ga_9 measured from 300 to 22 K. A vertical bar shows the Yb^{2+} -derived plasmon structure. (b) Fit of the Yb $3d_{5/2}$ spectrum of YbNi_3Ga_9 measured at 22 K.

Yb^{2+} peak in agreement with the Yb^{3+} plasmon satellite. We believe, therefore, the additional structure is the Yb^{2+} plasmon satellite. The Yb^{2+} plasmon satellite is not clearly observed for the compounds with a small fraction of Yb^{2+} , such as YbInCu_4 (see Ref. 17), $\text{Yb}_{1-x}\text{Lu}_x\text{B}_{12}$ (see Ref. 18), and YbNi_3Al_9 .

It has been pointed out that the analysis of the Yb $3d$ spectrum can give Yb valence accurately in comparison with that of the Yb $4f$ spectrum in the valence-band region.^{17,18} We carried out the fitting analysis of the Yb $3d_{5/2}$ spectra and evaluated the Yb valence of YbNi_3Al_9 . The Yb valence is given by $z = 2 + I(\text{Yb}^{3+})/[I(\text{Yb}^{3+}) + I(\text{Yb}^{2+})]$, where $I(\text{Yb}^{3+})$ and $I(\text{Yb}^{2+})$ denote the integrated intensities of the Yb^{2+} and Yb^{3+} components.

Figure 1(b) shows the fit of the Yb $3d_{5/2}$ spectrum of YbNi_3Al_9 at 22 K as a typical example. We assume single line spectrum for the Yb^{2+} component and slightly modified line spectra derived from the atomic multiplet calculation for the Yb^{3+} component. The line spectra are convoluted with the Lorentzian function for lifetime broadening. In order to simulate the plasmon-derived spectral features, we assumed one Gaussian function. All components are convoluted with the Gaussian function taking into account the instrumental resolution of 240 meV. Finally, we add the background contribution derived from the secondary electrons according to the Shirley's method.²³ The obtained Yb valence of YbNi_3Al_9 is $z \sim 2.43$ at 22 K in the case of Fig. 1(b).

Figure 2(a) shows the temperature dependence of the Yb $3d_{5/2}$ HAXPES spectra of YbNi_3Al_9 between 300 and 22 K. The intensity of the Yb^{2+} peak at 1520 eV is considerably weak compared to that of the Yb^{3+} structures at 1524–1534 eV, indicating that the Yb ion in YbNi_3Al_9 is nearly trivalent. Almost no change with temperature is detected in the intensity of the Yb^{3+} structure. In contrast, the intensity of the Yb^{2+} peak gradually decreases and the spectral weight is shifted toward the lower binding-energy side from 300 to 150 K. The change in the Yb^{2+} spectral shape is saturated below 150 K. Note that the Yb^{2+} -derived peak of YbNi_3Al_9 has anomalous spectral shape and can not be represented by a single components.

We also carried out the fitting analysis of the Yb $3d_{5/2}$ spectra to obtain the Yb valence of YbNi_3Al_9 as described

above except that no plasmon satellite was considered. The Yb^{2+} plasmon satellite is negligible because of the weak Yb^{2+} peak compared to the Yb^{3+} peak. The higher binding-energy side of the Yb^{3+} plasmon satellite (above 1540 eV) is incorporated as the background contribution. To analyze the spectral intensity of Yb^{2+} peak, we assume two components, labeled as P_1 and P_2 in Fig. 2(b). The intensity of the P_2 component decreases with decreasing temperature from 300 to 150 K. The Yb^{2+} peak is shifted toward lower binding-energy side, and P_2 intensity is almost unchanged below 150 K.

Although we cannot specify the origin of the two components for the Yb^{2+} $3d_{5/2}$ peak of YbNi_3Al_9 at present, we consider it to be intrinsic to YbNi_3Al_9 because its anomalous feature and temperature dependence are highly reproducible. In addition, similar spectral features are observed for the Cu substituted samples for the Ni ions of YbNi_3Al_9 . Judging from their energy positions, the P_1 component should correspond to the Yb^{2+} peak generally observed in Yb compounds.

For the P_2 component, we first examine the surface or subsurface effects as an origin. For Yb compounds, it is known that the Yb ion in the (sub)surface region has the smaller valence than that in the bulk,^{19,24} which may lead to a different Yb^{2+} $3d_{5/2}$ peak energy from that in the bulk regions. However, taking into account the escape depth of photoelectrons from Yb $3d$ core with the kinetic energy of $E_k \sim 4$ keV (~ 64 Å)²⁵ and the temperature dependence of the P_2 intensity, the origin of the P_2 component cannot be explained by the (sub)surface effect. Note that the Yb^{2+} peak for YbInCu_4 with the relatively thick (sub)surface region can be fitted with single line spectrum.¹⁷ Another possible explanation for the P_2 component is that there are actually two Yb-ion sites in YbNi_3Al_9 . Although the crystal structure of YbNi_3Al_9 was first confirmed as ErNi_3Al_9 -type structure with a single Yb site, according to the recent structure analysis by synchrotron radiation diffraction, the Al_3 clusters in the Yb_2Al_3 layer are partially replaced by the Yb ions yielding several Yb ion sites.²⁶ The temperature dependence of the P_2 intensity, however, indicates that the disorder within the Yb_2Al_3 layer decreases with decreasing temperature, which seems to be unlikely. To get further crystallographical information, the temperature-dependent x-ray diffraction experiment is in progress.

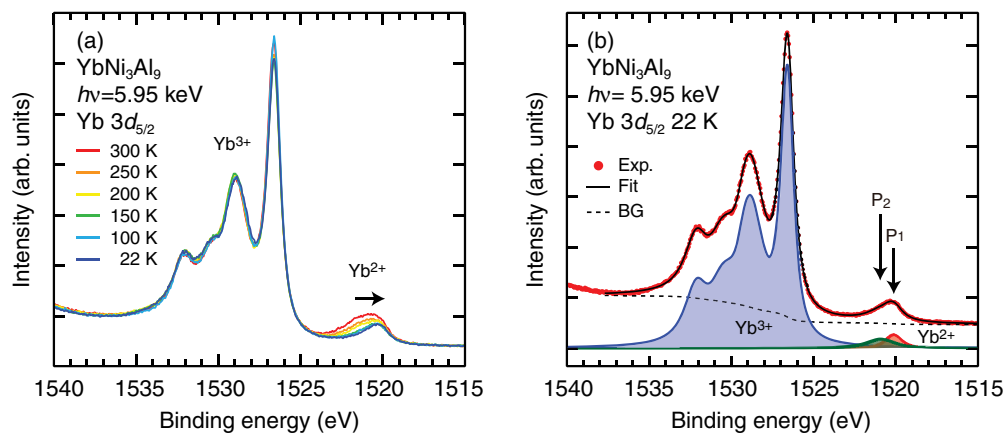


FIG. 2. (Color online) (a) Temperature dependence of the Yb $3d_{5/2}$ HAXPES spectra of YbNi_3Al_9 measured from 300 to 22 K. Spectral weight of the Yb^{2+} component is shifted toward the lower binding-energy side with decreasing temperature. (b) Fit of the Yb $3d_{5/2}$ spectrum of YbNi_3Al_9 measured at 22 K. We assume two components labeled as P_1 and P_2 for the Yb^{2+} peak.

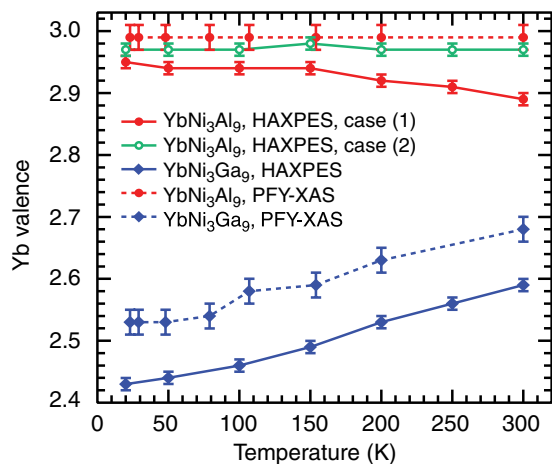


FIG. 3. (Color online) Temperature dependence of Yb valences of YbNi_3X_9 ($X = \text{Al}, \text{Ga}$) evaluated from the fitting analysis of $\text{Yb } 3d_{5/2}$ HAXPES and PFY-XAS spectra. Solid line connected closed and open circles represent the Yb valences from HAXPES of YbNi_3Al_9 evaluated for the case (1) and case (2), respectively, while solid line connected diamonds those of YbNi_3Ga_9 . Dashed line connected closed circles and diamonds represent the Yb valences of YbNi_3Al_9 and YbNi_3Ga_9 , respectively, from PFY-XAS.

Here, we evaluated the Yb valence of YbNi_3Al_9 for the two cases: (1) we assume both P_1 and P_2 and (2) only P_1 as the Yb^{2+} peak. In Fig. 2(b), the Yb valence at 22 K is evaluated as ~ 2.95 and 2.97 for the cases (1) and (2), respectively.

Figure 3 shows the evaluated Yb valences of YbNi_3X_9 as a function of temperature. Solid line connected closed and open circles represent the Yb valences of YbNi_3Al_9 evaluated for the cases (1) and (2), respectively, while solid line connected diamonds for those of YbNi_3Ga_9 . The Yb valence of YbNi_3Ga_9 at 300 K is $z \sim 2.59$ and monotonically decreases with temperature down to $z \sim 2.43$ at 22 K. The decrease of the Yb valence is often observed for the valence fluctuation compound such as YbAl_3 .²¹ According to the single impurity Anderson model (SIAM), the Yb valence approaches toward Yb^{2+} with decreasing temperature below T_K . The present experimental results indicate that T_K of YbNi_3Ga_9 is higher than room temperature, which is consistent with the temperature dependence of the magnetic susceptibility with a maximum around 200 K.^{4,5}

The Yb valence of YbNi_3Al_9 is close to $3+$ as expected from the Curie-Weiss behavior above 80 K of the magnetic susceptibility.^{4,5} For case (1), the evaluated Yb valence is $z \sim 2.89$ at 300 K and gets closer to $z \sim 3$ with decreasing temperature, which is very unusual among Yb compounds. The increase of the Yb valence at low temperature is caused by the decrease of the P_2 peak. For case (2), on the other hand, the Yb valence is $z \sim 2.97$ and shows little temperature dependence. This behavior can be explained within the framework of SIAM with low T_K less than 22 K.

To further examine the results obtained from the HAXPES measurements, we discuss PFY-XAS and RXES results at the Yb L_3 edge, which are bulk-sensitive spectroscopic techniques to study the valence of Yb compounds. Figure 4(a) shows the PFY-XAS spectra of YbNi_3Ga_9 taken between 300 and 23 K. The spectra were normalized to the integrated intensity. The

PFY-XAS results again indicate that YbNi_3Ga_9 is the strong valence fluctuation compound. The peak at 8938 eV is derived from the Yb^{2+} component and the peak at 8945 eV is the Yb^{3+} component. The energy of the Yb^{3+} peak is higher than that of the Yb^{2+} peak because the $\text{Yb}^{3+} 2p_{3/2}$ states with one $4f$ hole has lower binding-energy than the $\text{Yb}^{2+} 2p_{3/2}$ with no $4f$ hole. With decreasing temperature, the intensity of the Yb^{2+} (Yb^{3+}) component gradually increases (decreases). Figure 4(b) shows the RXES spectra of YbNi_3Ga_9 taken at the same temperatures as the PFY-XAS spectra. The incident photon energy is set at $h\nu = 8938$ eV, which corresponds to the Yb^{2+} peak in the PFY-XAS spectra. The horizontal axis represents an energy transfer defined as the difference between the incident and emitted photon energies. The RXES spectra mostly consist of the Yb^{2+} peak at the transfer energy of $\Delta h\nu = 1522$ eV with a tiny broad structure around 1526–1534 eV originating from the Yb^{3+} component. It is noted that the Yb^{2+} component is emphasized in the RXES spectra excited at the Yb^{2+} peak in the PFY-XAS spectra. The temperature dependence of these components in the RXES spectra is consistent with that in the PFY-XAS spectra. Actually, the temperature dependence of the intensity ratio of Yb^{3+} to Yb^{2+} (not shown here) shows the same trend as that of the Yb valence derived from the PFY-XAS spectra.

Figures 4(c) and 4(d) exhibit the PFY-XAS and RXES spectra of YbNi_3Al_9 measured with the same experimental condition for YbNi_3Ga_9 . As expected from the Yb $3d$ HAXPES spectra in Fig. 2(a), one can observe dominant Yb^{3+} component at 8945 eV in the PFY-XAS spectra and at 1528 eV in the RXES spectra. It should be noticed in the RXES spectra that the Yb^{2+} states are clearly observed as a shoulder structure at the lower transfer energy side of the Yb^{3+} peak ($\Delta h\nu = 1522$ eV). Taking into account that the PFY-XAS and RXES are more bulk sensitive compared to HAXPES, the existence of the Yb^{2+} component in RXES spectra indicate a small fraction of the divalent Yb ions in YbNi_3Al_9 . We find small peaks at 8934 eV in the PFY-XAS spectra and at 1518 eV in the RXES spectra. The similar feature is also observed in the Yb L_3 PFY-XAS and RXES spectra of YbCu_2Si_2 in Ref. 20, where these features are assigned to the $2p \rightarrow 4f$ quadrupole transition.

The Yb valences of YbNi_3X_9 are estimated from the fits to the PFY-XAS spectra. After the subtracting the arctan-like background from the spectra, we have fitted the spectra using the Voigt functions representing the Yb^{2+} and Yb^{3+} , $2p$ - $4f$ quadrupole transition and background components shown in Fig. 5. The Yb valences are derived from the intensity ratio of the Yb^{2+} and Yb^{3+} components. The dashed lines in Fig. 3 show fitting results. The Yb valence of YbNi_3Ga_9 monotonically decreases from $z \sim 2.68$ at 300 K to $z \sim 2.53$ at 23 K, while that of YbNi_3Al_9 stays at $z \sim 2.99$ at all temperatures. The Yb valences of YbNi_3Al_9 derived from the PFY-XAS spectra are almost consistent with those of the fits of the Yb $3d_{5/2}$ HAXPES spectra for case (1), suggesting that the analysis was appropriate. As regards YbNi_3Ga_9 , the mean Yb valences deduced from the PFY-XAS spectra is about 0.15 larger than those from the Yb $3d$ HAXPES spectra. This tendency has been widely observed for the Yb compounds. If one compares the fits in Figs. 1(b), 2(b) and 5, the analysis of the Yb $3d_{5/2}$ HAXPES spectra is more straightforward

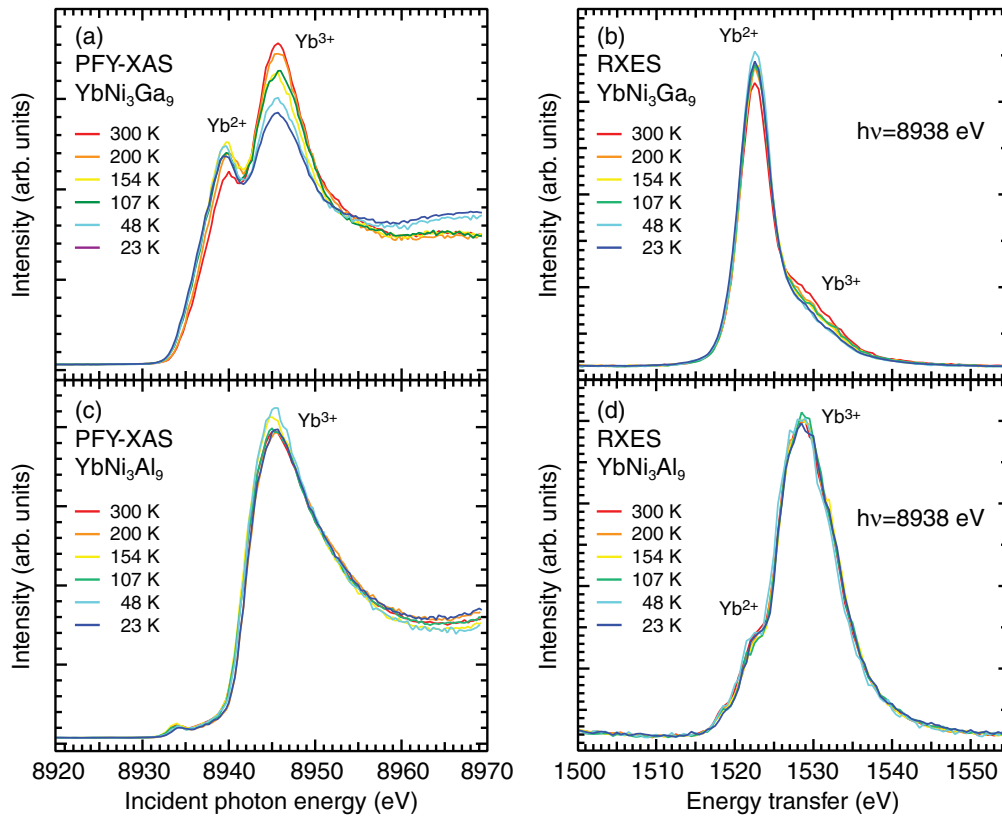


FIG. 4. (Color online) Temperature dependencies of Yb L_3 -edge PFY-XAS and RXES spectra measured at 8938 eV of YbNi_3X_9 between 300 and 23 K. (a) PFY-XAS and (b) RXES spectra of YbNi_3Ga_9 , and (c) PFY-XAS and (d) RXES spectra of YbNi_3Al_9 .

than that of the PFY-XAS spectra. In the former, the Yb^{2+} and Yb^{3+} components are separately observed, while in the latter their components are overlapped on the respective edge jumps. On the other hand, PFY-XAS is photon-in and photon-out experiments and is a more bulk sensitive compared to HAXPES using electrons as probe. Even at excitation energies as high as $h\nu \sim 6$ keV, the HAXPES spectra might be

somewhat suffered from the surface and subsurface regions, where the Yb valence tends to be $z \sim 2$. This is considered to be one of the reasons why the Yb $3d$ HAXPES provides smaller Yb valence.

Next, we examine the valence-band HAXPES spectra of YbNi_3X_9 . In Fig. 6(a), the spectrum of YbNi_3Ga_9 taken at 22 K exhibits the Yb^{2+} and Yb^{3+} -derived structures, indicating the valence fluctuating property. The prominent $\text{Yb}^{2+} 4f_{7/2}$ and $4f_{5/2}$ peaks exist near E_F and 1.5 eV, respectively, and the $\text{Yb}^{3+} 4f$ multiplet structures due to the $4f^{12}$ final states are observed at 5–12 eV. On the other hand, the spectrum of YbNi_3Al_9 is mostly composed of the Yb^{3+} multiplet structures, consistent with the Yb $3d_{5/2}$ spectra in Fig. 2(a). We also note the tiny peaks due to the $\text{Yb}^{2+} 4f_{7/2}$ and $4f_{5/2}$ states exist as shown by vertical bars. The $\text{Yb}^{3+} 4f$ multiplet structure of YbNi_3Al_9 is shifted toward the lower binding-energy side compared to that of YbNi_3Ga_9 as a whole. The small peaks at 2.0 eV for YbNi_3Ga_9 and at 2.3 eV for YbNi_3Al_9 is ascribed to the Ni $3d$ states. As shown in Fig. 6(b), these peaks are remarkably observed in the vacuum ultraviolet photoemission (VUV PES) spectra taken at undulator beamline BL-1 of Hiroshima Synchrotron Radiation Center (HSRC) with $h\nu = 182$ eV, where the photoionization cross section of the Ni $3d$ states becomes large.²⁷ Note that the Ni $3d$ peak is shifted toward the lower binding-energy side by about 300 meV on going from $X = \text{Al}$ to Ga as indicated by a solid line.

Figure 7 shows the Ni $2p_{3/2}$ core HAXPES spectra of YbNi_3X_9 measured at 300 K. The inset exhibits the whole Ni $2p$ spectra in the wide binding-energy range. The Ni

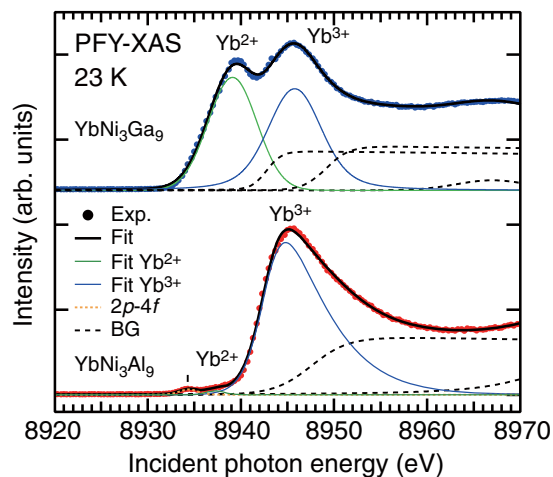


FIG. 5. (Color online) An example of the fits for PFY-XAS spectra of YbNi_3X_9 at 23 K. Thin solid lines represent the fits for the Yb^{2+} and Yb^{3+} components using the Voigt functions after the subtracting the arctan-like background from the spectra.

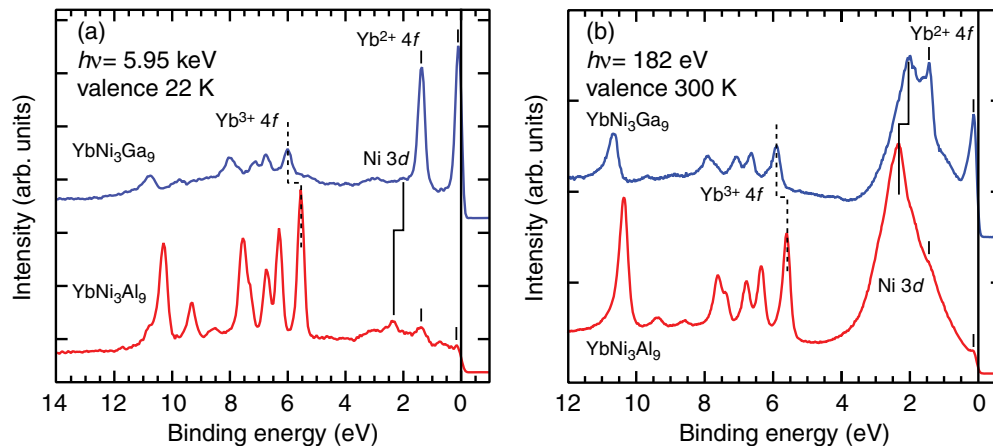


FIG. 6. (Color online) The valence-band (a) HAXPES and (b) VUV PES spectra measured at 22 and 300 K, respectively. The vertical bars show the $\text{Yb}^{2+}4f_{7/2}$ and $4f_{5/2}$ structures. The lowest peak of the Yb^{3+} multiplet structures is shown by a dashed line. The $\text{Yb}^{3+}4f$ multiplet structures of YbNi_3Al_9 is shifted toward the lower binding-energy side compared to that of YbNi_3Ga_9 as a whole. The small peaks at 2.0 and 2.3 eV in the HAXPES spectra for YbNi_3Ga_9 and YbNi_3Al_9 (solid line), respectively, are ascribed to the Ni $3d$ states, which are remarkably observed as prominent peaks in the VUV PES spectra due to the large photoionization cross section of the Ni $3d$ states.

$2p_{3/2}$ and $2p_{1/2}$ peaks are located around 853 and 871 eV, respectively. One can also see broad structures around 863, 880, and 887 eV, and a shoulder structure overlapping with the lower binding-energy side of the $2p_{1/2}$ peak around 870 eV.

The bulk and surface plasmons are often observed in photoemission spectra. The energy of the surface plasmon is generally lower than that of the corresponding bulk plasmon.²⁸ The broad structure at 887 eV is ascribed to the bulk plasmon associating with the Ni $2p_{1/2}$ photoemission and are separated from the $2p_{1/2}$ main peak by $\Delta E \sim 16$ eV. The Ni $2p_{3/2}$ bulk plasmon is observed as the shoulder at 870 eV. The bulk plasmons are also observed as the broad structures at $\Delta E \sim 16$ eV higher binding-energy from the main peaks in the Ga $2p$ (not shown here) and Yb $3d$ [see Fig. 1(a)] spectra

of YbNi_3Ga_9 . In the inset of Fig. 7, there exist two peaks at 880 and 863 eV in the Ni $2p$ spectra (vertical bars) that seem to be surface plasmons. However, it is unlikely because there are no corresponding peaks associating with the Ga $2p$ spectra. Thus we consider that the structures at 880 and 863 eV are derived from the satellite structures with the $2p^53d^9$ final states due to the electron correlation effect.²⁹

In Fig. 7, dots and solid lines, respectively, represent the experimental spectra and the fits using the Lorentzian function convoluted with the Gaussian function for the instrumental resolution. The binding energies of the Ni $2p_{3/2}$ states of YbNi_3Al_9 and YbNi_3Ga_9 are 853.80 and 853.51 eV, respectively. There exists an energy shift toward the lower binding-energy side by about 300 meV on going from $X = \text{Al}$ to Ga. The core-level chemical shift reflects the change of the chemical bonding and charge distribution of the conduction electron around the specific atoms. Taking into account the fact that the similar energy shift is also observed for the Ni $3d$ states in the valence-band spectra in Fig. 6, the Ni $2p_{3/2}$ energy shift suggests that E_F of the Ni-derived conduction-band (CB) density of states (DOS) is shifted toward the lower energy side for YbNi_3Ga_9 . It suggests a charge transfer from the CB states to the Yb $4f$ states leading to a valence fluctuation in YbNi_3Ga_9 .

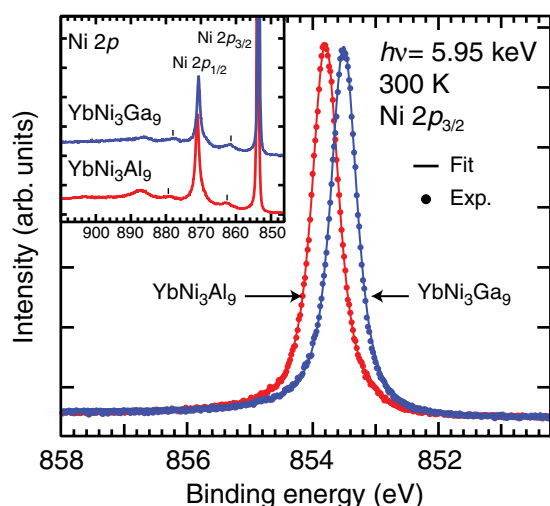


FIG. 7. (Color online) Ni $2p_{3/2}$ HAXPES spectra of YbNi_3Ga_9 and YbNi_3Al_9 measured at 300 K. Dots and solid lines are, respectively, experimental spectra fits using single line spectrum convolutes with Lorentzian and Gaussian functions. The inset shows the whole region of the Ni $2p$ spectra measured at 300 K. The broad structures shown by vertical bars are satellite structures.

IV. DISCUSSION

We find the difference of the binding energies of the $\text{Yb}^{3+}4f$ multiplet structures between YbNi_3Ga_9 and YbNi_3Al_9 as shown by a dashed line in Fig. 6. The binding energy of the $\text{Yb}^{3+}4f$ multiplet structures of YbNi_3Al_9 are shifted by 300 meV toward the E_F side compared to that of YbNi_3Ga_9 . We consider that the energy shift is originated from the difference of bare $4f$ -hole level, $-\varepsilon_f$, whose negative sign represents that $4f$ -hole level exists above E_F . An energy separation of the Yb^{2+} and $\text{Yb}^{3+}4f$ HAXPES spectra are approximately given by $-\varepsilon_f + U$, where U stands for the Coulomb interaction between the $4f$ holes on the same site.

If we assume that U is almost unchanged between YbNi_3X_9 , the deeper binding-energy of the Yb^{3+} multiplet structures for YbNi_3Al_9 suggests that the absolute value of $-\varepsilon_f$ in YbNi_3Al_9 is larger than that in YbNi_3Ga_9 . This means that the $4f$ -hole level exists far above E_F stabilizing the Yb^{3+} configuration (one hole state) in YbNi_3Al_9 . On the other hand, the $4f$ -hole level in YbNi_3Ga_9 is expected to be located closer to E_F leading to the valence fluctuation.³⁰ The energy shift of the Yb^{3+} multiplet structures was also observed in $\text{Yb}_2\text{Co}_3\text{X}_9$ ($X = \text{Ga}, \text{Al}$) by VUV PES with $h\nu = 40.8$ eV.³¹ $\text{Yb}_2\text{Co}_3\text{Al}_9$ is an antiferromagnetic heavy-fermion compound with $T_N = 1.16$ K and $\text{Yb}_2\text{Co}_3\text{Ga}_9$ is a valence fluctuation compound with $T_K \sim 210$ K as is the case of YbNi_3X_9 . In addition, $\text{Yb}_2\text{Co}_3\text{X}_9$ has similar crystal structures with that of YbNi_3X_9 .³²

As the X atoms are the nearest neighbor of the Yb ions, it is likely that the sp -like CB electrons derived from X atoms significantly contribute to the c - f hybridization, though the band-structure calculation is yet to be performed. Furthermore, the energy shift of the Ni $2p$ and $3d$ states indicates that the Ni-derived CB states also contribute to the c - f hybridization of YbNi_3X_9 . In particular, their Ni $3d$ DOS's should have tails toward the lower binding-energy side and some contribution to the DOS at E_F since there exist satellites in the Ni $2p$ spectra. From the energy positions of the Ni $3d$ peak in Fig. 6, the larger Ni $3d$ DOS at E_F is expected for YbNi_3Ga_9 compared to YbNi_3Al_9 , which qualitatively explains T_K of several hundreds K for YbNi_3Ga_9 and T_K of a few K for YbNi_3Al_9 . In addition, based on the discussion of the energy shift of the $\text{Yb}^{3+}4f$ multiplet structures, the Yb $4f$ -hole level of YbNi_3Ga_9 is closer to E_F than that of YbNi_3Al_9 . The large Ni $3d$ DOS and the Yb $4f$ -hole level near E_F of YbNi_3Ga_9 causes the strong hybridization between the Ni $3d$ and Yb $4f$ states and charge transfer from the Ni $3d$ to Yb $4f$ states, leading to the enhanced valence fluctuation. In the case of YbNi_3Al_9 , the Yb ion stays to be trivalent because the Ni $3d$ DOS at E_F is small and the Yb $4f$ -hole level exists away from E_F . Thus the Ni $3d$ DOS at E_F and the Yb $4f$ -hole level play a key role to determine the antiferromagnetic and valence fluctuating properties of YbNi_3X_9 . In order to confirm the role of the Ni $3d$ states, studies of solid solutions of YbNi_3X_9 and other transition-metal substituted systems for the Ni ions, which directly control the d band, are planned.

The Yb $3d$ HAXPES and PFY-XAS experiments reveal that the Yb valence of YbNi_3Ga_9 is close to $z \sim 2.5$, namely, equal population of the Yb^{2+} and Yb^{3+} states. Based on the experimental results, YbNi_3Ga_9 is expected to be one of systems easy to control the magnetic ground state. In fact, YbNi_3Ga_9 undergoes a transition into a ferromagnetic ordered state at ~ 9 GPa.³³ At the critical pressure, the Yb valence is considered to be trivalent.

Generally, Yb^{3+} state is favored at high pressure indicating the decrease of T_K and the strength of c - f hybridization, and the Yb ground state changes from mixed valent to localized magnetic states.³⁴ In Yb metal, the pressure induces the shift of both Yb^{3+} and Yb^{2+} energy level toward E_F and the charge transfer occurs from Yb^{2+} to CB, resulting in the enhancement of Yb^{3+} .³⁵ Our results show that the substitution of Ga to Al site in YbNi_3Al_9 may correspond to apply the negative pressure and the charge transfer occurs from CB to Yb^{3+} ,

which is a reversed process of the case of Yb metal. Namely, the chemical pressure induces the charge transfer from the CB to $4f$ hole and the localized Yb state of YbNi_3Al_9 changes to the mixed valent state of YbNi_3Ga_9 as described above. It is, however, noted that the difference of the lattice parameters from $X = \text{Al}$ to Ga are about -0.4% for a axis and 0.8% for c axis.⁵ This indicates the difference between the chemical and hydrostatic pressure effects; in YbNi_3X_9 the ground state is more sensitive to the change in the electronic structure by the chemical substitution.

Finally, we comment on the present experimental results of YbNi_3X_9 in comparison with those of the valence transition compound YbInCu_4 .³⁶⁻³⁸ The sharp valence transition of YbInCu_4 was initially observed by Yb $L_{\alpha 1}$ resonant inelastic x-ray scattering experiments,³⁹ and then by the Yb $3d$ HAXPES experiments.¹⁷ The Yb valence derived from the HAXPES changes from $z \sim 2.90$ to $z \sim 2.74$ at $T_V = 42$ K on cooling.¹⁷ At T_V , T_K also changes from $T_{K+} \sim 25$ K to $T_{K-} \sim 400$ K.⁴⁰ Recently, we found that the Cu $2p_{3/2}$ and In $3d_{5/2}$ peaks in the HAXPES spectra of YbInCu_4 were shifted toward the lower binding-energy side below T_V by ~ 40 and ~ 30 meV, respectively.⁴¹ The energy shifts suggest that the valence transition is caused from the charge transfer from CB to Yb $4f$ states and the higher T_{K-} is interpreted by the higher CB-DOS at E_F . In addition, the $\text{Yb}^{3+}4f$ multiplet structures in the valence-band soft x-ray photoemission spectra of YbInCu_4 taken at $h\nu = 800$ eV are also shifted toward the higher binding-energy side below T_V ,¹⁹ again similar to the energy shift of the corresponding multiplet structures from YbNi_3Al_9 to YbNi_3Ga_9 (see Fig. 6). The energy shifts of the core-level and Yb^{3+} multiplet structures are, thus, possibly controlled by T_K , which may be changed through the valence transition or by the substitution of the constituent elements and pressure.

V. CONCLUSIONS

We have investigated the electronic structure of the Kondo lattice compounds YbNi_3X_9 by means of the Yb $3d_{5/2}$, valence-band and Ni $2p$ HAXPES with $h\nu = 5.95$ keV, the PFY-XAS and RXES at the Yb L_3 edge, and the valence-band PES with $h\nu = 182$ eV. Both Yb^{2+} and Yb^{3+} components were clearly observed in the Yb $3d_{5/2}$ spectra of YbNi_3Ga_9 , indicating the strong valence fluctuation, consistent with the magnetic susceptibility measurements.^{4,5} With decreasing temperature, the intensity of the Yb^{2+} (Yb^{3+}) structure gradually increases (decreases) and the evaluated Yb valence of YbNi_3Ga_9 changes from 2.59 at 300 K to 2.43 at 22 K. The Yb $3d_{5/2}$ spectra of YbNi_3Al_9 , on the other hand, indicate that the Yb^{3+} component is dominant and the Yb^{2+} component is considerably weak. The spectra show almost no temperature dependence and the evaluated Yb valence at 22 K is 2.97, consistent with the Curie-Weiss-like magnetic susceptibility above 80 K. The results on the Yb valence and its temperature dependence were consistent with the PFY-XAS and RXES experiments, though the derived valence is relatively higher than those obtained from the Yb $3d$ HAXPES experiments. In the valence-band spectra, we found that the binding energy of the $\text{Yb}^{3+}4f$ multiplet structures of YbNi_3Al_9 is ~ 300 meV higher than that of YbNi_3Ga_9 . This suggests that

the energy difference between $4f$ -hole level and E_F is small for YbNi_3Ga_9 and large for YbNi_3Al_9 . We also observed the energy shifts of the Ni $2p$ and Ni $3d$ states on going from $X = \text{Al}$ to Ga by 300 meV toward the lower binding-energy side. These energy shifts can be explained by the E_F shift of the Ni-derived CB-DOS, which leads to the enhanced Ni $3d$ DOS at E_F in YbNi_3Ga_9 . Taking into account the $4f$ -hole level close to E_F , the charge transfer from the Ni $3d$ to Yb $4f$ states easily takes place, which makes YbNi_3Ga_9 valence fluctuating system.

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¹C. M. Varma, *Rev. Mod. Phys.* **48**, 219 (1976).

²S. Nakatsuji, K. Kuga, Y. Machida, T. Tayama, T. Sakakibara, Y. Karaki, H. Ishimoto, E. Pearson, G. G. Lonzarich, H. Lee, L. Balicas, and Z. Fisk, *Nat. Phys.* **4**, 603 (2008).

³M. Okawa, M. Matsunami, K. Ishizaka, R. Eguchim M. Taguchim, A. Chainani, Y. Takata, M. Yabashi, K. Tamasaku, Y. Nishino, T. Ishikawa, K. Kuga, N. Horie, S. Nakatsuji, and S. Shin, *Phys. Rev. Lett.* **104**, 247201 (2010).

⁴S. Ohara, T. Yamashita, Y. Mori, and I. Sakamoto, *J. Phys.: Conf. Ser.* **273**, 012048 (2011).

⁵T. Yamashita, R. Miyazaki, Y. Aoki, and S. Ohara, *J. Phys. Soc. Jpn.* **81**, 034705 (2012).

⁶R. E. Gladyshevskii, K. Genzual, H. D. Flack, and E. Parthé, *Acta Cryst. B* **49**, 468 (1993).

⁷Y. Lutsyshyn, Y. Tokaychuk, V. Davydov, and R. Gladyshevskii, *Chem. Met. Alloys* **1**, 303 (2008).

⁸S. Doniach, *Physica B & C* **91**, 231 (1977).

⁹S. Danzenbächer, Yu. Kucherenko, D. V. Vyalikh, M. Holder, C. Laubschat, A. N. Yaresko, C. Krellner, Z. Hossain, C. Geibel, X. J. Zhou, W. L. Yang, N. Mannella, Z. Hussain, Z.-X. Shen, M. Shi, L. Patthey, and S. L. Molodtsov, *Phys. Rev. B* **75**, 045109 (2007).

¹⁰J. L. Sarrao, C. D. Immer, Z. Fisk, C. H. Booth, E. Figueroa, J. M. Lawrence, R. Modler, A. L. Cornelius, M. F. Hundley, G. H. Kwei, J. D. Thompson, and F. Bridges, *Phys. Rev. B* **59**, 6855 (1999).

¹¹J. M. Lawrence, P. S. Riseborough, C. H. Booth, J. L. Sarrao, J. D. Thompson, and R. Osborn, *Phys. Rev. B* **63**, 054427 (2001).

¹²S. Ueda, Y. Katsuya, M. Tanaka, H. Yoshikawa, Y. Yamashita, S. Ishimaru, Y. Matsushita, and K. Kobayashi, *AIP Conf. Proc.* **1234**, 403 (2010).

¹³K. Kobayashi, M. Yabashi, Y. Takata, T. Tokushima, S. Shin, K. Tamasaku, D. Miwa, T. Ishikawa, H. Nohira, T. Hattori, Y. Sugita, O. Nakatsuka, A. Sakai, and S. Zaima, *Appl. Phys. Lett.* **83**, 1005 (2003).

¹⁴I. Jarrige, H. Ishii, Y. Q. Cai, J.-P. Rueff, C. Bonnelle, T. Matsumura, and S. R. Shieh, *Phys. Rev. B* **72**, 075122 (2005).

¹⁵H. Yamaoka, I. Jarrige, N. Tsujii, N. Hiraoka, H. Ishii, and K.-D. Tsuei, *Phys. Rev. B* **80**, 035120 (2009).

¹⁶H. Yamaoka, H. Sugiyama, Y. Kubozono, A. Kotani, R. Nouchi, A. M. Vlaicu, H. Oohashi, T. Tochio, Y. Ito, and H. Yoshikawa, *Phys. Rev. B* **80**, 205403 (2009).

¹⁷H. Sato, K. Shimada, M. Arita, K. Hiraoka, K. Kojima, Y. Takeda, K. Yoshikawa, M. Sawadam M. Nakatake, H. Namatame, M. Taniguchi, Y. Takata, E. Ikenaga, S. Shin, K. Kobayashi, K. Tamasaku, Y. Nishino, D. Miwa, M. Yabashim, and T. Ishikawa, *Phys. Rev. Lett.* **93**, 246404 (2004).

¹⁸J. Yamaguchi, A. Sekiyama, S. Imada, H. Fujiwara, M. Yano, T. Miyamachi, G. Funabashi, M. Obara, A. Higashiya, K. Tamasaku, M. Yabashi, T. Ishikawa, F. Iga, T. Takabatake, and S. Suga, *Phys. Rev. B* **79**, 125121 (2009).

¹⁹H. Sato, K. Yoshikawa, K. Hiraoka, M. Arita, K. Fujimoto, K. Kojima, T. Muro, Y. Saitoh, A. Sekiyama, S. Suga, and M. Taniguchi, *Phys. Rev. B* **69**, 165101 (2004).

²⁰L. Moreschini, C. Dallera, J. J. Joyce, J. L. Sarrao, E. D. Bauer, V. Fritsch, S. Bobev, E. Carpena, S. Huotari, G. Vankó, G. Manaco, P. Lacovig, G. Panaccione, A. Fondacaro, G. Paolicelli, P. Torelli, and M. Grioni, *Phys. Rev. B* **75**, 035113 (2007).

²¹S. Suga, A. Sekiyama, S. Imada, A. Shigemoto, A. Yamasaki, M. Tsunekawa, C. Dallera, L. Braicovich, T.-L. Lee, O. Sakai, T. Ebihara, and Y. Ōnuki, *J. Phys. Soc. Jpn.* **74**, 2880 (2005).

²²K. Yamamoto, H. Yamaoka, N. Tsujii, A. M. Vlaicu, H. Oohashi, S. Sakakura, T. Tochio, Y. Ito, A. Chainani, and S. Shin, *J. Phys. Soc. Jpn.* **76**, 124705 (2007).

²³D. A. Shirley, *Phys. Rev. B* **5**, 4709 (1972).

²⁴F. Reinert, R. Claessen, G. Nicolay, D. Ehm, S. Hüfner, W. P. Ellis, G.-H. Gweon, J. W. Allen, B. Kindler, and W. Assmus, *Phys. Rev. B* **58**, 12808 (1998).

²⁵S. Tanuma, C. J. Powell, and D. R. Penn, *Surf. Interface Anal.* **21**, 165 (1994).

²⁶C. Moriyoshi (private communication).

²⁷J. J. Yeh and I. Lindau, *At. Data Nucl. Data Tables* **32**, 1 (1985).

²⁸J. A. Leiro and M. H. Heinonen, *Surf. Sci.* **346**, 73 (1996).

²⁹F. U. Hillebrecht, J. C. Fuggle, P. A. Bennett, Z. Zołnierak, and Ch. Freiburg, *Phys. Rev. B* **27**, 2179 (1982).

- ³⁰We carried out theoretical calculation based on SIAM with noncrossing approximation (NCA) and the NCA results reproduced the energy shift of Yb^{3+} multiplet structures toward the deeper binding-energy side with increasing the absolute value of ε_f .
- ³¹T. Okane, S.-I. Fujimori, A. Ino, A. Fujimori, S. K. Dhar, C. Mitra, P. Manfrinetti, A. Palenzona, and O. Sakai, *Phys. Rev. B* **65**, 125102 (2002).
- ³²R. E. Gladyshevskii, K. Cenzual, and E. Parthé, *J. Alloys Compd.* **182**, 165 (1992).
- ³³K. Matsubayashi, T. Hirayama, T. Yamashita, S. Ohara, and Y. Uwatoko (in preparation).
- ³⁴A. V. Goltsev and M. M. Abd-Elmeguid, *J. Phys.: Condens. Matter* **17**, 5913 (2005).
- ³⁵E. R. Ylvisaker, J. Kunes, A. K. McMahan, and W. E. Pickett, *Phys. Rev. Lett.* **102**, 246401 (2009).
- ³⁶I. Felner and I. Nowik, *Phys. Rev. B* **33**, 617 (1986).
- ³⁷I. Felner, I. Nowik, D. Vaknin, Ulrike Potzel, J. Moser, G. M. Kalvius, G. Wortmann, G. Schmiester, G. Hilscher, E. Gratz, C. Schmitzer, N. Pillmayr, K. G. Prasad, H. de Waard, and H. Pinto, *Phys. Rev. B* **35**, 6956 (1987).
- ³⁸K. Kojima, *J. Magn. Magn. Mater.* **81**, 267 (1989).
- ³⁹C. Dallera, M. Grioni, A. Shukla, G. Vankó, J. L. Sarrao, J. P. Rueff, and D. L. Cox, *Phys. Rev. Lett.* **88**, 196403 (2002).
- ⁴⁰J. M. Lawrence, S. M. Shapiro, J. L. Sarrao, and Z. Fisk, *Phys. Rev. B* **55**, 14467 (1997).
- ⁴¹Y. Utsumi, H. Sato, H. Kurihara, H. Maso, K. Hiraoka, K. Kojima, K. Tobimatsu, T. Ohkuchi, S.-I. Fujimori, Y. Takeda, Y. Saitoh, K. Mimura, S. Ueda, Y. Yamashita, H. Yoshikawa, K. Kobayashi, T. Oguchi, K. Shimada, H. Namatame, and M. Taniguchi, *Phys. Rev. B* **84**, 115143 (2011).