

Rheological evaluation of the fabrication parameters of cellulose acetate butyrate membrane on CO₂/N₂ separation performance

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Abstract

1
2 The rise in emission of greenhouse gases (GHGs) mainly carbon dioxide (CO₂) in recent years due to
3 rapid development of modern civilisation, has been listed as the primary contributor to global warming.
4 To address this global issue, membrane technology was applied and developed intensively because of
5 its superior performance in terms of efficiency and economic advantages. In this study, the cellulose
6 acetate butyrate (CAB) polymer was selected as the polymer matrix material since it exhibited excellent
7 film-forming properties. In addition, the wet-phase inversion technique was adopted to synthesise the
8 membrane based on different casting conditions. The optimum outcomes of the fabrication conditions
9 were then characterised with the scanning electron micrograph (SEM) to determine the best CAB
10 membrane for CO₂/N₂ separation. The results showed that CAB-70000 fabricated with 4 wt% of CAB
11 polymer concentration, casting thickness of 250 µm, solvent evaporation time of 5 minutes, and 30
12 minutes of solvent exchange for isopropyl alcohol and n-hexane, exhibited the best gas separation
13 performance. Further, CAB-70000 showed an average selectivity of 6.12 ± 0.09 and permeance up to
14 227.95 ± 0.39 GPU for CO₂ and 37.28 ± 0.54 GPU for N₂, respectively. In summary, this study is
15 expected to show a detailed outline of the future direction and perspective of the novel CAB polymeric
16 membrane that is suitable to be applied in the industry, and serves as an insight for researchers and
17 manufacturers working in the related field of gas separation.

18 **1.0 Introduction**

19 There is a trend of rapid increase in world population, which is expected to hit 10 billion by 2050 (Lalia
20 et al., 2013). In this regard, higher demand in energy will be required for the 21st century to meet the
21 urgent needs. It is predicted that the energy demand will increase by 57 per cent in 2030 (Conti et al.,
22 2016). As a major contributor to the world energy supply, fossil fuel solely contribute around 40 per
23 cent of the total carbon dioxide (CO₂) emission into the environment, which is mainly attributed to the
24 massive coal combustion activities (Carapellucci and Milazzo, 2003). Global warming has become a
25 genuine problem due to the excessive discharge of pollutants emitted from the combustion activities in
26 the primary industries (Yang et al., 2008).

27 In the past few decades through their efforts, researchers have contributed in combating this global
28 issue to limit and minimise the impact of greenhouse gases (GHGs). They have outlined three feasible
29 options. The first comprises of saving energy used intensively with methods that are more efficient. The
30 second option is to minimise the usage of carbon-based material source or replace it with renewable
31 energy, and the third is to improve the effectiveness of CO₂ sequestration with more advanced
32 technology development (Yang et al., 2008). For the past few years, membrane separation technology
33 has been utilised intensively for both water treatment and gas separation purpose (Yang et al., 2008,
34 Kappel et al., 2014, Barnes et al., 2014, Zhu et al., 2014). The membrane's chemical and physical
35 properties, and interaction between permeance and membrane are relatively crucial factors in
36 determining the diffusion characteristics of the gas separation field (Shekhawat, 2003). This is because
37 the separation selectivity and permeance are two critical parameters that indicate membrane separation
38 performance. In an ideal situation, high selectivity and permeance are preferred as both induce less
39 capital costs and operating expenses for the industries (Paradise and Goswami, 2007, Low et al., 2013).
40 Hence, the selection of material plays an influential role, in determining the specific gas separation
41 performance (Lalia et al., 2013, Zha et al., 2015, Feng et al., 2015).

42 The cellulose acetate butyrate (CAB) possesses few interesting characteristics that include, film-
43 forming properties, acetyl and butyryl functional groups, which can effectively improve and further
44 expand the capacity of cellulose chain giving high sorption characteristic, as well as high impact,

45 weather and chemical resistant (Feng et al., 2015, Basu et al., 2010, Kunthadong et al., 2015). The CAB
46 was first investigated and studied by Sourirajan back in 1958, then followed by Manjikian and others in
47 reverse osmosis (RO) separation (Wang et al., 1994). They reported that the CAB membrane owned
48 high solute separation with tolerable membrane flux result, and also provided ease of fabrication as
49 some pre-treatment was negligible (Ohya et al., 1980, Wang et al., 1994). However, limited studies have
50 been conducted on the effects of the acetyl group content on CAB membranes in the CO₂/N₂ gas
51 separation field. Further, no reports or systematic studies have been performed on the effects of
52 membrane production procedure and fabrication parameters. This includes membrane-casting thickness,
53 solvent exchange time for both isopropyl alcohol and n-hexane with different CAB molecular weights
54 as well as the polymer matrix material structure and performance of CAB membranes. Therefore, the
55 primary objective of this study is to investigate the effects of membrane production procedure and
56 fabrication parameters. Discussions on how the mentioned parameters can affect the membrane in terms
57 of morphology and gas separation performance are presented in this report. The separation performance
58 of the synthesised CAB membrane was selected to evaluate the specified parameters towards CO₂/N₂.

59 **2.0 Methodology**

60

61 **2.1 Materials**

62 The cellulose acetate butyrate (CAB, M_n ~12000, 65000, 70000) in powder form was purchased from
63 Sigma-Aldrich (Malaysia) for membrane preparation. Solutions required for membrane preparation i.e.,
64 chloroform, isopropyl alcohol, and n-hexane were purchased from Merck (Malaysia). Distilled water
65 was used for the phase-inversion steps, specifically for immersion precipitation for membrane formation.

66 **2.2 CAB Polymer dope preparation**

67 The CAB membrane was prepared using the wet-phase inversion method, followed by solvent exchange
68 to dry the membrane. A dope solution consisting of 4 wt% CAB (M_n =70000) powders and 96 wt%
69 chloroform was prepared following the condition of each parameter. The solution was stirred for 24
70 hours, and then sonicated for 20 minutes to eliminate the gas bubbles in the solution (Ahmad et al., 2014,
71 Feng et al., 2015). The solution was then poured into space within the casting bars with glass plate

72 underneath. An automatic film applicator (Elcometer 4340, E.U.) was then used for the casting of the
73 membrane. Referring to our previous work, 5 minutes of solvent evaporation time was allowed
74 following each parameter's condition before immersing the membrane in distilled water (27 °C) for a
75 duration of 24 hours (S.Minhas, 1992, Lee et al., 2017). The solvent exchange was performed on the as-
76 spun membrane first with 60 minutes immersion period in isopropyl alcohol and then another 60 minutes
77 immersion period in n-hexane. The resultant membrane was then dried at ambient temperature to
78 eliminate the remaining volatile liquid in between two glass plates filled with filter paper for 24 hours
79 before use (S.Minhas, 1992, Jawad et al., 2015a)

80 2.2.1 *Effect of casting thickness*

81
82 The membranes were prepared at different casting thicknesses following the fabrication method as
83 described in section 2.2 where the study range for this parameter was from 200 µm (CAB-200), 250 µm
84 (CAB-250) to 300 µm (CAB-300).

85 2.2.2 *Effect of exchange time with isopropyl alcohol*

86
87 Meanwhile, for the effect of solvent exchange time, the membranes were prepared following the
88 fabrication method as described in section 2.2. The solvent exchange duration studied was 15 minutes
89 (CAB-15Iso), 30 minutes (CAB-30Iso), and 60 minutes (CAB-60Iso) for isopropyl alcohol, followed
90 by 60 minutes of n-hexane.

91 2.2.3 *Effect of exchange time with n-hexane*

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93 In addition, the solution of the solvent exchange time with n-hexane was prepared following the
94 fabrication method as described in section 2.2. The resultant membranes were first solvent exchanged
95 with isopropyl alcohol for 30 minutes followed by solvent exchange times ranging from 15 minutes
96 (CAB-15H), 30 minutes (CAB-30H) to 60 minutes (CAB-60H) for n-hexane.

97 2.2.4 *Effect of CAB at different molecular weight (M_n)*

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99 The membranes were prepared with different CAB molecular weights (M_n) of 12000 (CAB-12000),
100 65000 (CAB-65000), and 70000 (CAB-70000) for the preparation of the dope solution. After that,
101 following the fabrication method as mentioned in section 2.2, the solvent exchange time for isopropyl
102 alcohol and n-hexane were set for 30 minutes each.

103 **2.3 Membrane permeability test**

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105 The procedure for gas permeation measurement was discussed in our previous published work (Lee et
106 al., 2017).

107 **2.4 Membrane characterization**

108 109 *2.4.1 Scanning Electron Microscopy (SEM)*

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111 The CAB membrane structures including surface and cross-sectional, were observed via SEM (Hitachi
112 TM3000, Tokyo, Japan). Each membrane sample was cut into small pieces, and then kept on a plastic
113 petri dish in the cryogenic freezer at a temperature of up to -80°C for 24 hours to give a consistent and
114 clean-cut by freezing. The samples were coated with a platinum layer to prevent high-energy beam
115 damage before the characterisation works. Further, each sample's average membrane thickness was
116 calculated based on the frequency count as measured by the Image-J software. Approximately, 100
117 measurements were taken to confirm the average membrane thickness.

118 *2.4.2 X-ray Photoelectron Spectroscopy (XPS)*

119
120 The CAB membranes fabricated at different molecular weights were characterised with the High
121 Resolution Multi-Technique X-Ray Spectrometer (Axis Ultra DLD XPS, Kratos, Shimadzu Corporation,
122 Japan). The analysis was carried out using a PHI 1600 spectrometer with hybrid lens mode, 150 W
123 (Anode: Mono), 1000 meV step, and 5 sweeps for each membrane at room temperature.

124 **3.0 Results and Discussion**

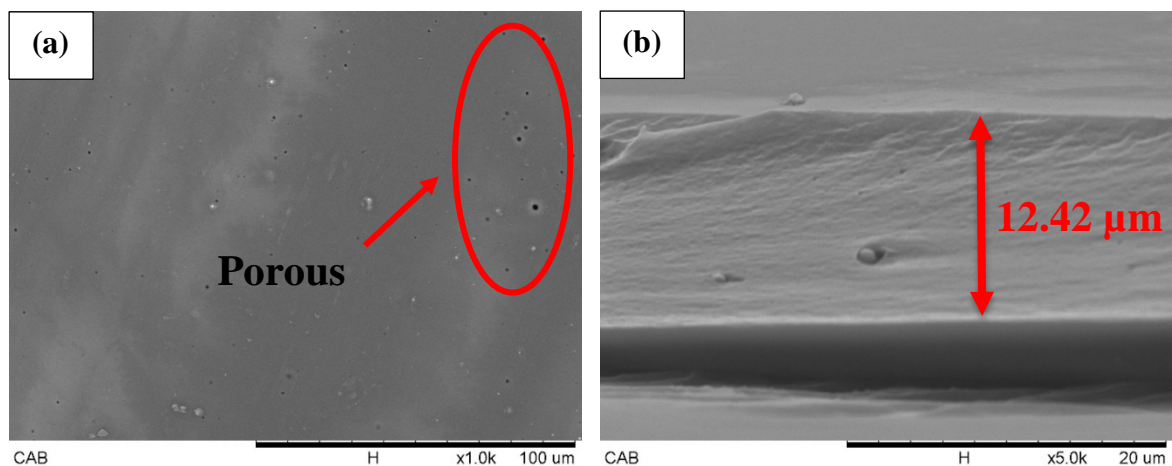
125 126 **3.1 Effect of casting thickness**

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128 The effect of casting thickness on the structure and performance of the CAB membrane was investigated,
129 as depicted in **Fig. 1**. As observed from **Fig. 1a**, the structure of CAB-200 (200 µm) was porous. As the
130 casting thickness of the membranes increased, a smooth surface was observed for CAB-250 (250 µm),
131 as demonstrated in **Fig. 1c**. Alternatively, a rough surface was formed for CAB-300 (250 µm), as seen
132 in **Fig. 1e**. The change in the structure was due to the different rates of demixing that occurred as the
133 phase precipitation proceeded when high casting thickness was applied, causing the deposition speed of
134 the membrane to reduce during the membrane formation phase. The slow deposition rate avoids rapid

135 exchange of non-solvent and solvent within the membrane. As a result, the surface structure of the CAB
136 membrane was built-up based on the sufficient phase precipitation period given (Ahmad et al., 2013,
137 Thomas et al., 2014).

138 The cross-sectional micrographs of the fabricated CAB membrane at casting thickness of 200 μm
139 (CAB-200), 250 μm (CAB-250), and 300 μm (CAB-300) were revealed in **Figs. 1b, d, and f,**
140 respectively. From the micrographs, dense structures were depicted from all the cross-sectionals of the
141 CAB membranes. The dense structure formation was due to the densification of the membrane during
142 the immersion period, whereby the remaining solvent imbedded in the polymer matrix was replaced by
143 distilled water. As the volatility of the solvent was generally higher than distilled water the membrane
144 thickness changed from $12.42 \pm 0.05 \mu\text{m}$ to $11.32 \pm 0.06 \mu\text{m}$ and $12.89 \pm 0.10 \mu\text{m}$ for CAB-200, CAB-
145 250 and CAB-300, respectively. The reduction of membrane thickness from $12.42 \pm 0.05 \mu\text{m}$ (CAB-
146 200) to $11.32 \pm 0.06 \mu\text{m}$ (CAB-250) was due to thicker casting thickness applied during membrane
147 fabrication, which allows more solvent embedded in the polymer matrix to be replaced by non-solvent
148 (H_2O) during the immersion period, resulting in a denser and thinner membrane thickness for CAB-250
149 (Ahmad et al., 2013). In contrast, a thicker membrane was obtained when increasing the membrane
150 thickness further to 300 μm for CAB-300 ($12.89 \pm 0.10 \mu\text{m}$). This is correlated to the increase resistance
151 of inward diffusion of non-solvent, due to higher casting thickness applied, causing a delay transition
152 demixing in the film membrane (Tiraferrri et al., 2011).

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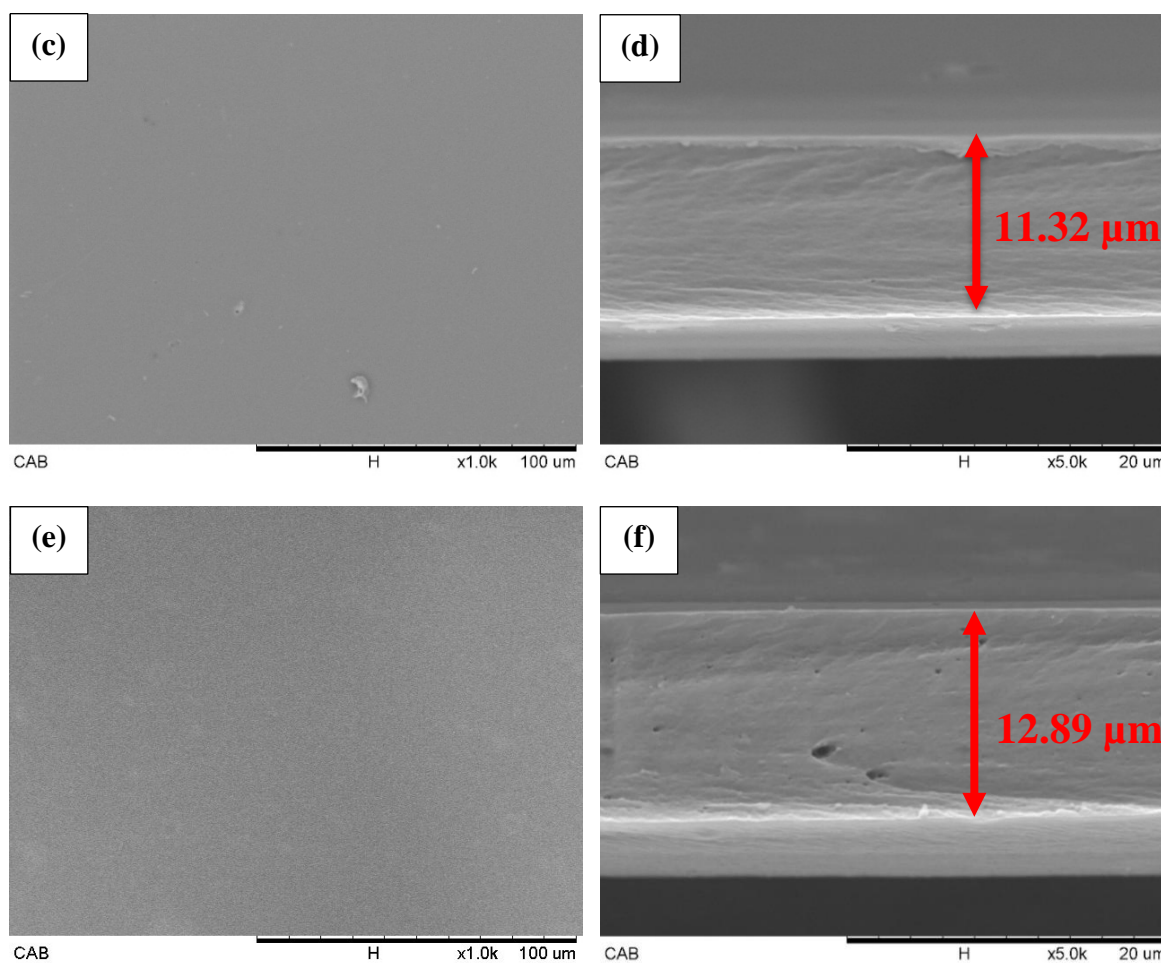


Fig 1. Top and cross-sectional SEM of CAB membrane at casting thickness (a-b) 200 μm (CAB-200), (c-d) 250 μm (CAB-250), and (e-f) 300 μm (CAB-300), with 4 wt% CAB polymer concentration and 5 minutes solvent evaporation time

154 The CO_2 permeance of CAB-200, CAB-250, and CAB-300 are illustrated in **Fig. 2**. Notably,
 155 CAB-250 demonstrated a higher permeance result of 398.46 ± 1.43 GPU, as compared to CAB-200
 156 (143.03 ± 0.62 GPU) and CAB-300 (12.93 ± 0.34 GPU). This was because of the reduction in its
 157 membrane thickness (11.32 μm , **Fig. 1d**) and its selective smooth surface structure, which allowed the
 158 solution diffusion mechanism to occur efficiently. Therefore, the CO_2 permeance of CAB-250 increased
 159 (Jawad et al., 2015a). Meanwhile, the CO_2 permeance of CAB-300 reduced to 12.93 ± 0.34 GPU,
 160 indicating that a higher casting thickness beyond 250 μm can exert extra resistance towards gas diffusion
 161 within the membrane, which in turn affects the efficiency of gas permeation due to the thick dense
 162 membrane synthesised (**Fig. 1f**).

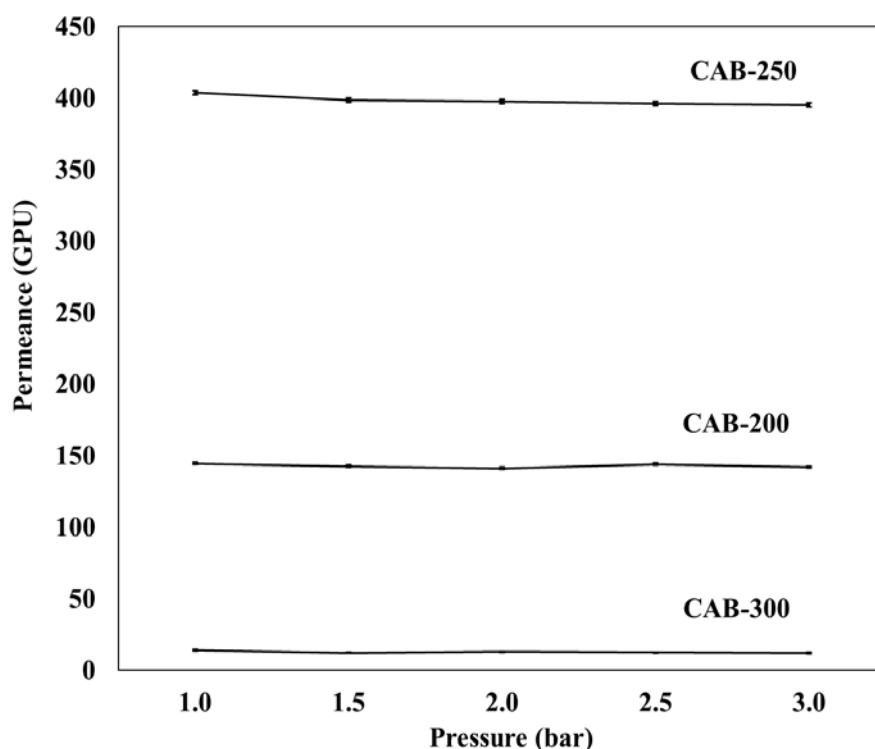


Fig 2. CO₂ permeance for membrane fabricated at 200 μm (CAB-200), 250 μm (CAB-250), and 300 μm (CAB-300), with 4 wt% CAB polymer concentration and 5 minutes solvent evaporation time

163
 164 On the other hand, the N₂ permeance for CAB-200, CAB-250 and CAB-300 were 112.83 ± 0.85 ,
 165 121.55 ± 1.30 , and 11.26 ± 0.31 GPU, respectively, as illustrated in **Fig. 3**. The CAB-250 exhibited
 166 higher N₂ permeance results. This was due to the initial casting thickness applied, resulting in a smooth
 167 membrane structure, which created less resistance towards the permeance of N₂ gas within the
 168 membrane (Freeman, 1999). The low N₂ permeance result yield for CAB-300 (11.26 ± 0.31 GPU) was
 169 due mainly to the thick dense membrane structure (12.89 ± 0.10 μm), which ultimately governed the
 170 solution diffusion rate of the membrane, as a thicker membrane usually induces more resistance to gas
 171 diffusion (Koros et al., 1988a).

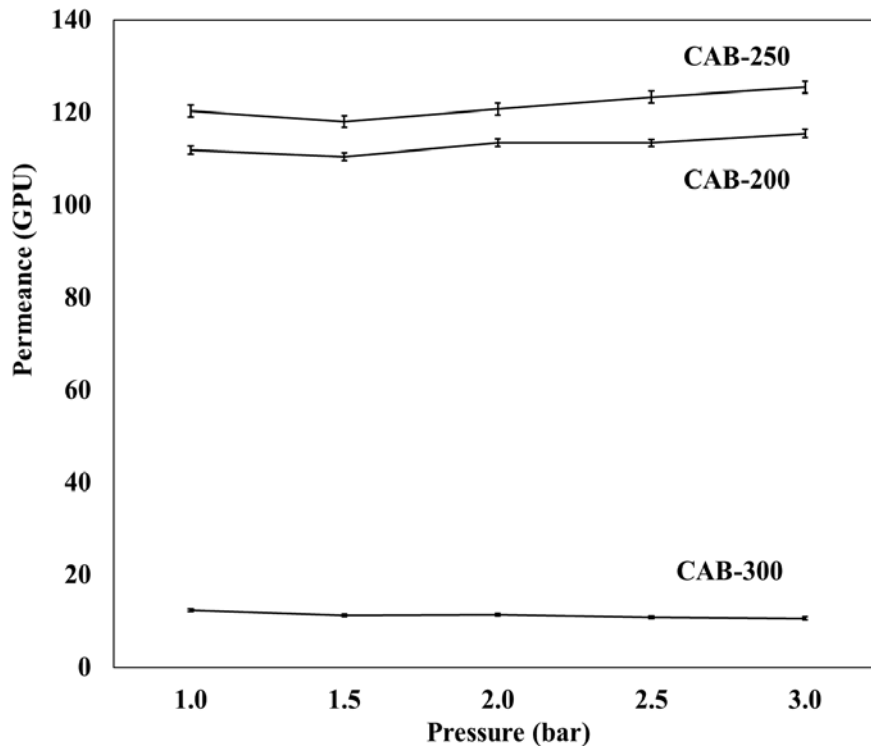


Fig 3. N₂ permeance for membrane fabricated at 200 μm (CAB-200), 250 μm (CAB-250), and 300 μm (CAB-300), with 4 wt% CAB polymer concentration and 5 minutes solvent evaporation time

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The ideal selectivity of CO₂/N₂ separation performance for CAB-200, CAB-250, and CAB-300 are shown in **Fig. 4**. As observed from the results when increasing the casting thickness from 200 μm to 250 μm the selectivity increased from 1.27 ± 0.01 GPU (CAB-200) to 3.28 ± 0.04 GPU (CAB-250). The acceptable result obtained for CAB-250 was due to the membrane structure formation, which eventually increased the CO₂ permeance against the N₂ permeance attained. However, the selectivity reduced to 1.15 ± 0.01 GPU when the higher casting thickness (300 μm) was implemented for CAB-300. Even though the thickness of the membrane was essential for effective gas separation, however, excessive membrane thickness restricted the gas diffusion within the membrane.

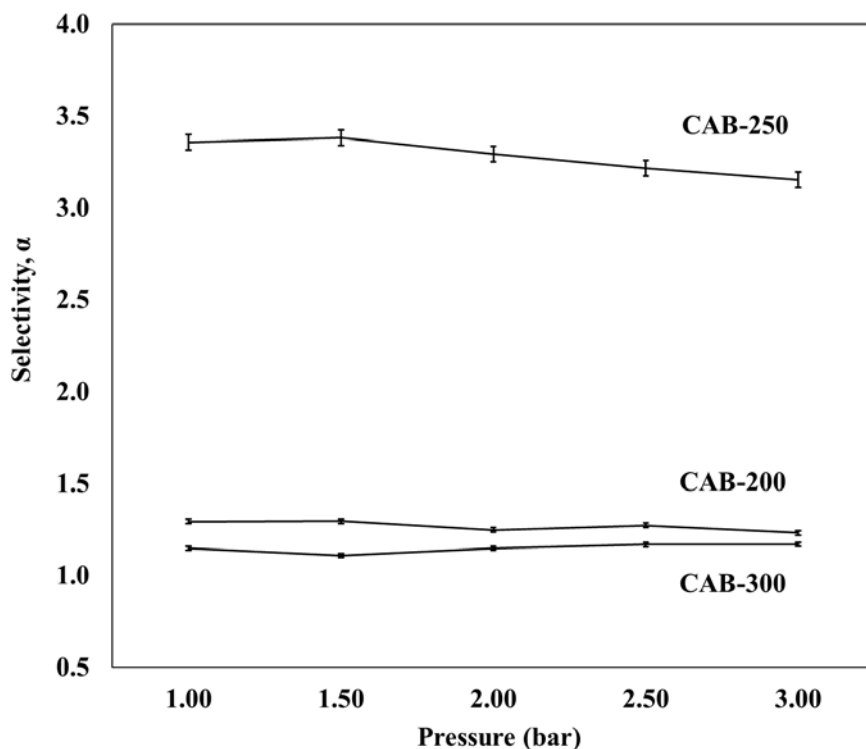


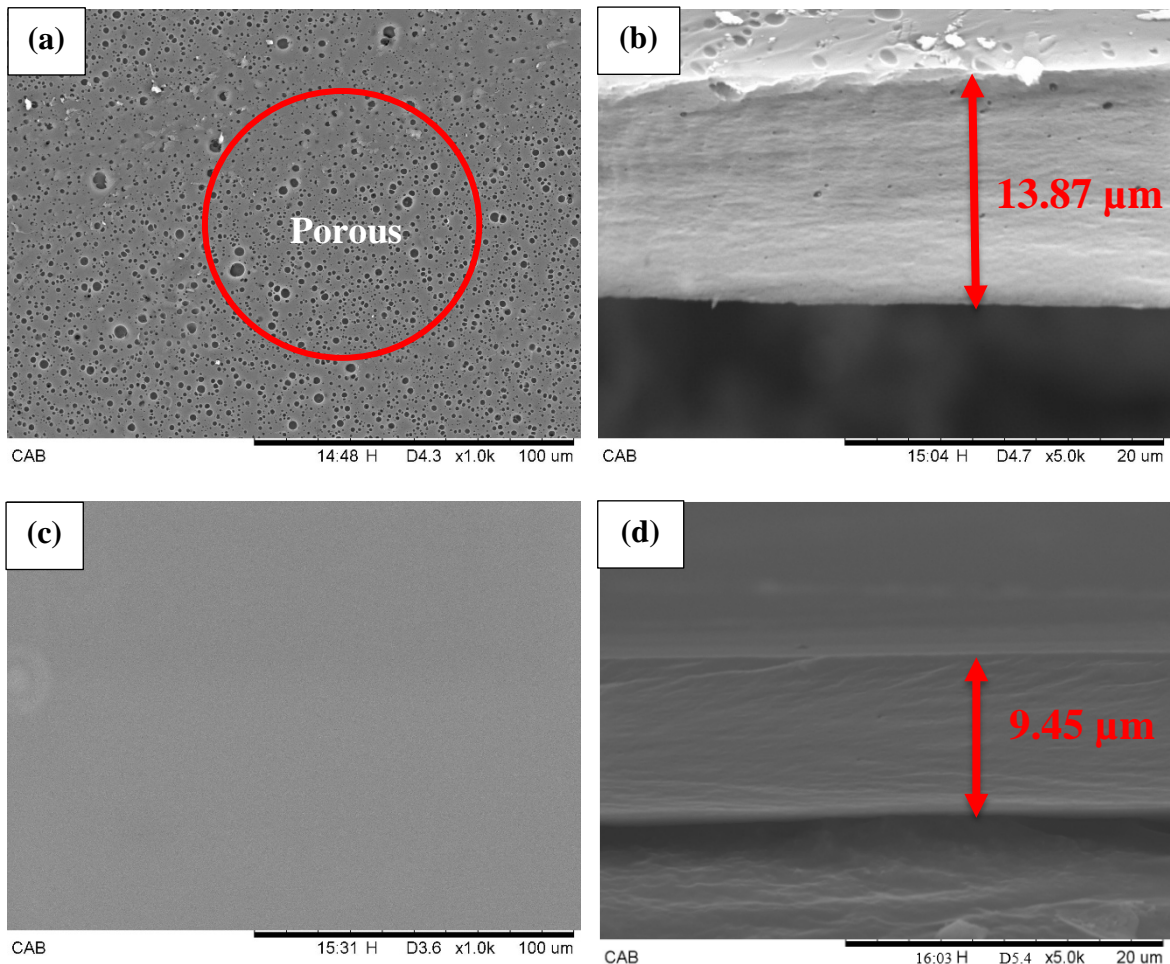
Fig 4. Ideal selectivity of CO₂/N₂ for membranes fabricated at different casting thickness 200 μm (CAB-200), 250 μm (CAB-250), and 300 μm (CAB-300), with 4 wt% CAB polymer concentration and 5 minutes solvent evaporation time

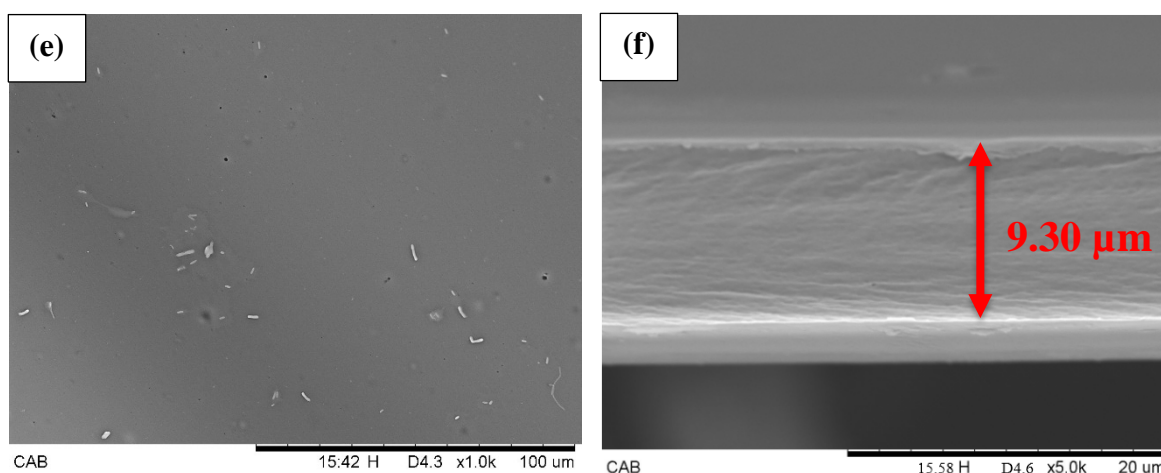
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3.2 Effect of solvent exchange with Isopropyl alcohol

The solvent exchange was performed after the precipitation immersion process of the CAB membrane with the purpose of drying or removing any remaining volatile liquid in the membrane. As displayed in Figs. 5a and b, the CAB-15Iso (15 minutes) exhibited a porous surface and irregular dense cross-sectional structure with a membrane thickness of $13.87 \pm 0.23 \mu\text{m}$. This porous structure surface was caused by the rapid solvent exchange between the water molecules available within the CAB structure and the first solvent (isopropyl alcohol) (Lui et al., 1988). During the first step of the solvent exchange process, an enormous amount of water molecules embedded in the membrane were generally replaced by isopropyl alcohol. As a result, due to the short 15 minutes solvent exchange immersion period allocated, vigorous pore formation appeared throughout the film membrane of CAB-15Iso (Lui et al., 1988). Thus, CAB-15Iso demonstrated thick membrane thickness because of the short solvent exchange time applied, resulting in more water molecules retaining inside the membrane.

195 Meanwhile, when the isopropyl alcohol solvent exchange time was increased to 30 minutes
196 (CAB-30Iso) and then subsequently to 60 minutes (CAB-60Iso), both revealed a smooth surface (**Figs.**
197 **5c** and **e**) with thin dense membrane thickness of $9.45 \pm 0.06 \mu\text{m}$ and $9.30 \pm 0.05 \mu\text{m}$, respectively, as
198 demonstrated in **Figs. 5d** and **f**. The formation of smooth surface and thin membrane was because of the
199 longer immersion period allocated. Therefore, this provided more relaxation time for the non-solvent
200 (H_2O) imbedded in the film membrane to exchange with the isopropyl alcohol (Radjabian et al., 2014).
201 This also allowed the formation of a thin dense membrane with homogeneous smooth surface structure
202 as revealed from CAB-30Iso and CAB-60Iso.
203





204 **Fig 5. Surface and cross-sectional SEM of CAB membrane dried with isopropyl alcohol first for**
 205 **a solvent exchange duration of (a-b) 15 minutes (CAB-15Iso), (c-d) 30 minutes (CAB-30Iso),**
 206 **and (e-f) 60 minutes (CAB-60Iso); then subsequently solvent exchanged with 60 minutes of**
 207 **n-hexane as the final solvent, at casting thickness of 250 μm and 5 minutes solvent**
 208 **evaporation time**

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210 As shown in **Fig. 6**, the CO₂ permeance rates increased from 65.53 ± 0.34 GPU (CAB-15Iso) to

211 262.29 ± 0.16 GPU (CAB-30Iso) and increased further to 398.82 ± 0.94 GPU (CAB-60Iso) by changing

212 the solvent exchange duration of isopropyl from 15 minutes to 30 minutes and subsequently to 60

213 minutes, respectively. This resulted in extensive water content reduction within the membrane structure

214 due to longer immersion period allocated. The steady exchange rate of water with isopropyl alcohol

215 within the CAB polymer matrix caused less CO₂ molecules to interact with the water, therefore allowing

216 more CO₂ gas to permeate through the membrane (Jawad et al., 2015b). In the meantime, the high CO₂

217 permeance rate for CAB-60Iso (60 minutes) contributed to the thin dense membrane structure, which

218 allowed the CO₂ feed gas to pass through the membrane with least resistance pathway as compared to

219 the thick dense membrane (Tirafferri et al., 2011). Thus, the CAB-60Iso (60 minutes) yielded the highest

220 CO₂ permeance rate amongst the other membranes (CAB-15Iso and CAB-30Iso).

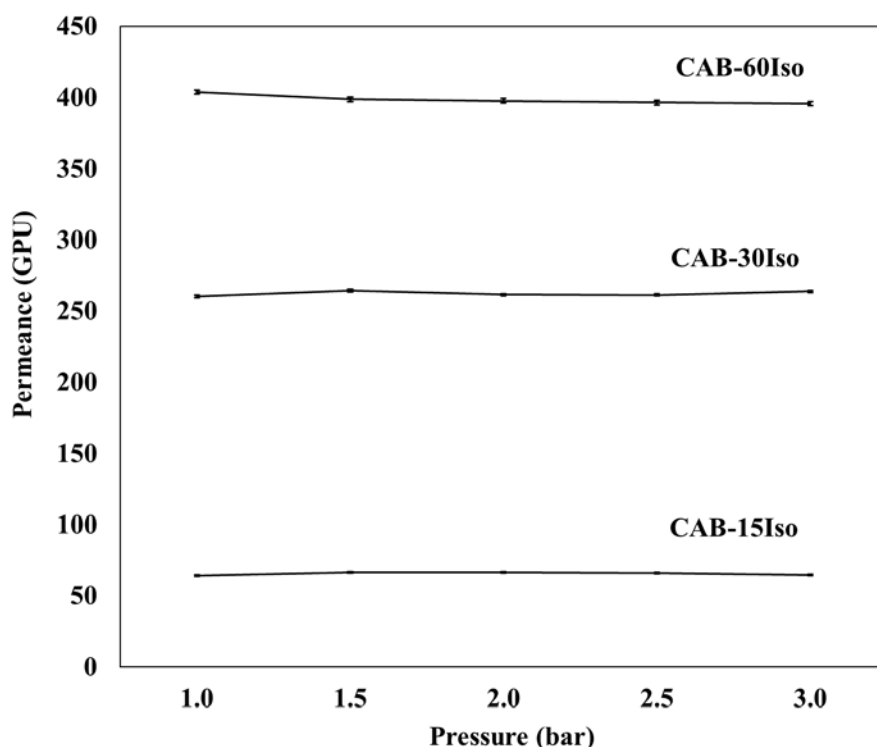


Fig 6. CO₂ permeance for membrane dried with 15 minutes (CAB-15Iso), 30 minutes (CAB-30Iso), and 60 minutes (CAB-60Iso) of isopropyl alcohol; then subsequently solvent exchanged with 60 minutes of n-hexane as the final solvent, at casting thickness of 250 μm and 5 minutes solvent evaporation time

222 The N₂ permeance rates for CAB-15Iso, CAB-30Iso, and CAB-60Iso are depicted in **Fig. 7**. The
 223 results obtained for N₂ permeance were 64.59 ± 0.41 , 70.49 ± 0.33 , and 121.76 ± 0.83 GPU for CAB-
 224 15Iso, CAB-30Iso, and CAB-60Iso, respectively. The possible explanation for this trend was due to the
 225 reduction in the membrane thickness from 13.87 μm to 9.3 μm (**Fig. 5**). In addition, as isopropyl alcohol
 226 was mainly made up from non-polar molecules, the remaining molecules within the CAB structure can
 227 easily attract light gas molecules (Katayama and Nitta, 1976). Thus, with longer solvent exchange
 228 duration, more isopropyl alcohol was retained within the polymer matrix, hence, attracting more N₂ gas
 229 molecules and resulting in the high N₂ permeance rate for CAB-60Iso (60 minutes). Eventually as the
 230 solvent exchange duration decreased, the N₂ permeance rate for CAB-15Iso and CAB-30Iso reduced as
 231 well.

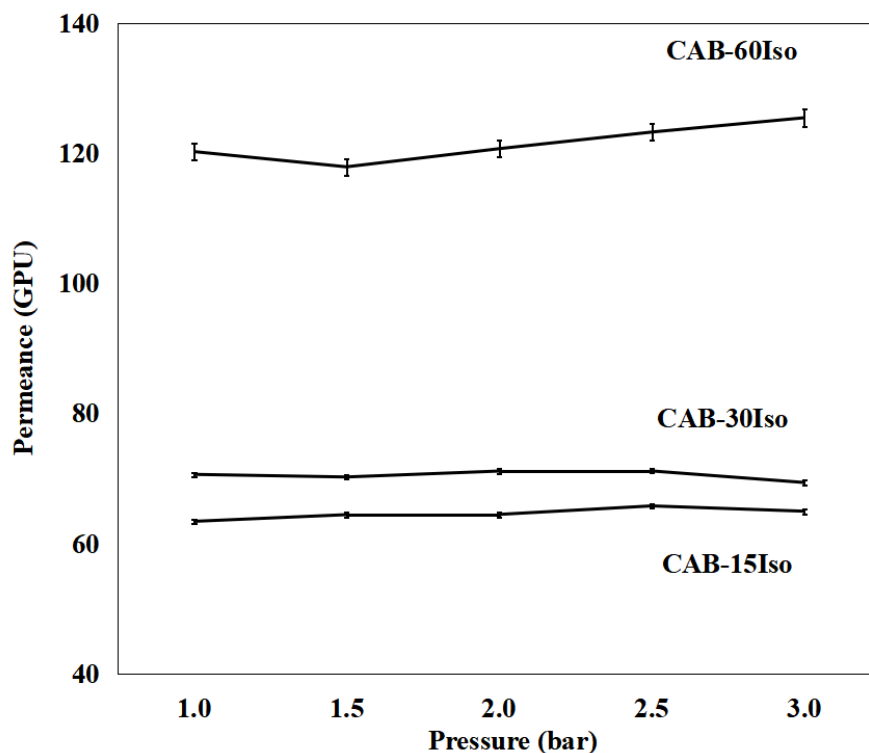


Fig 7. N₂ permeance for membrane dried with 15 minutes (CAB-15Iso), 30 minutes (CAB-30Iso), and 60 minutes (CAB-60Iso) of isopropyl alcohol; then subsequently solvent exchanged with 60 minutes of n-hexane as the final solvent, at casting thickness of 250 μ m and 5 minutes solvent evaporation time

232 As discussed previously, CAB-60Iso (60 minutes) showed a thin dense membrane formation with
 233 high CO₂ and N₂ permeance rates. However, based on **Fig. 8**, the CAB-30Iso (30 minutes) yielded the
 234 best selectivity performance. This was due to the smooth homogeneous surface and superior cross-
 235 sectional morphology, which selectively allowed a predetermined amount of CO₂ and N₂ to pass through
 236 the dense membrane. On the contrary, the CAB-15Iso (15 minutes) demonstrated low selectivity (**Fig.**
 237 **8**). This was due to the presence of a thick irregular surface morphology (**Figs. 5a and b**), which imposed
 238 an undesirable effect on membrane permeance performance due to extra resistance pathway generated
 239 (Rahimpour et al., 2008, Yang and Wang, 2006). Therefore, CAB-30Iso (30 minutes) was preferred as
 240 compared to CAB-15Iso (15 minutes) and CAB-60Iso (60 minutes) because of its excellent morphology
 241 and good selectivity performance.

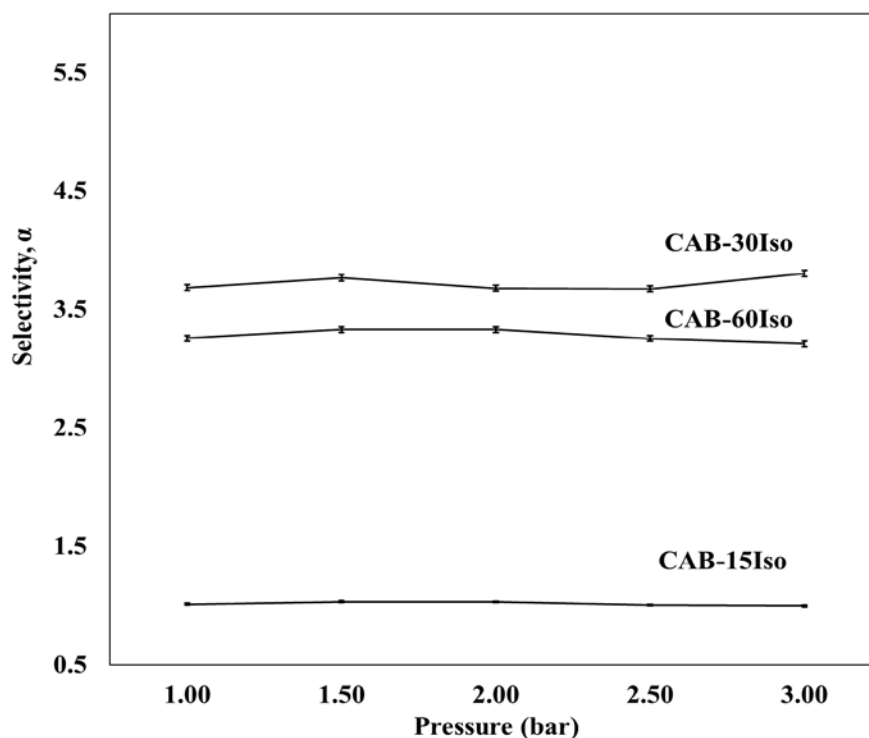


Fig 8. Ideal selectivity of CO₂/N₂ for membranes synthesised with solvent exchange duration of 15 minutes (CAB-15Iso), 30 minutes (CAB-30Iso), and 60 minutes (CAB-60Iso); then subsequently exchanged with 60 minutes of n-hexane as the final solvent, at casting thickness of 250 μ m and 5 minutes solvent evaporation time

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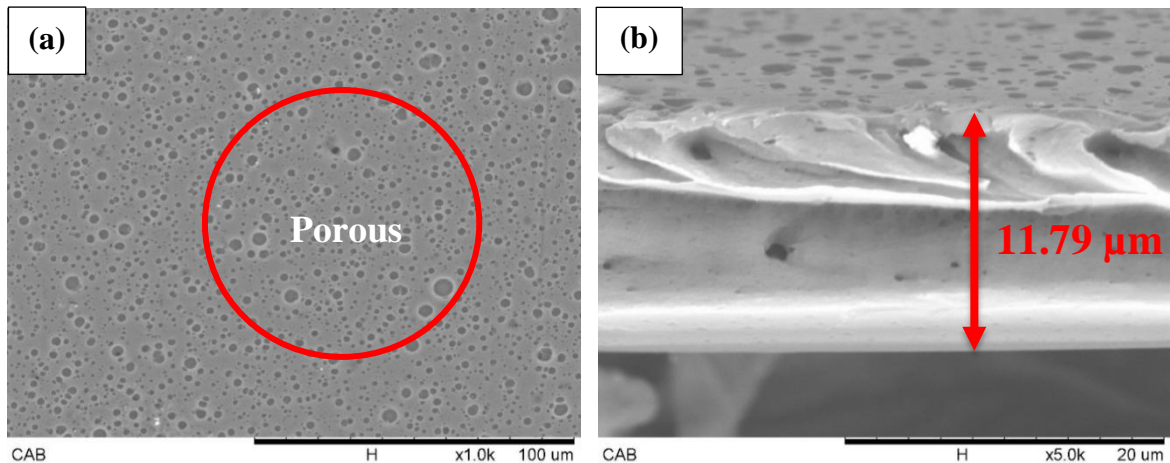
3.3 Effect of exchange time with n-hexane

As discussed in the previous section, the best solvent exchange time for isopropyl alcohol was 30 minutes (CAB-30Iso). Subsequently, the CAB membrane was subjected to further optimisation with the drying time of n-hexane. In this study, the CAB membranes were dried with solvent exchange times of 15 minutes (CAB-15H), 30 minutes (CAB-30H), and 60 minutes (CAB-60H) using n-hexane. As revealed from the SEM image in **Fig. 9**, the surface of CAB-15H (15 minutes) exhibited a porous structure, while CAB-30H (30 minutes) and CAB-60H (60 minutes) showed smooth surfaces. The main reason for the porous structure showed by CAB-15H was due to the rapid evaporation of the volatile solvent from the membrane structure itself and short duration of immersion period implemented (Chung and Kafchinski, 1997). Gradually by increasing the solvent exchange immersion period, the membrane had sufficient time for the solvent exchange to occur between isopropyl alcohol and n-hexane at a consistent and steady rate. Hence, suppressing a vigorous solvent exchange process within the polymer,

256 resulted in a smooth homogeneous surface as observed for CAB-30H (30 minutes) and CAB-60H (60
257 minutes) (Choi et al., 2006).

258 As presented in **Fig. 9**, the membrane thickness for CAB-15H (15 minutes), CAB-30H (30
259 minutes), and CAB-60H (60 minutes) were 11.79 ± 0.18 , 9.50 ± 0.10 , and 9.45 ± 0.06 μm , respectively.
260 As seen from these results, the increased exchange time of n-hexane caused the CAB membrane to
261 become more compact due to membrane densification as time passed (Sabde et al., 1997). In addition,
262 the main reason for the reduction in the membrane thickness was due to the isopropyl alcohol imbedded
263 within the membrane slowly being replaced by n-hexane with time. The replacement of isopropyl
264 alcohol with n-hexane occurred when the molecular affinity of n-hexane was greater than isopropyl
265 alcohol (Hansen, 2007). Referring to the Hansen solubility chart, the solubility for isopropyl alcohol, n-
266 hexane, and water are 23.6, 14.9, and 47.9 $\text{MPa}^{1/2}$, respectively (Egan and Dufresne, 2008, Hansen,
267 2007). Therefore, the molecular affinity is in the order of CAB-water>CAB-isopropyl alcohol>CAB-n-
268 hexane. The order of the molecular affinity represents the attraction force between the polymer and the
269 solvent and non-solvent used (Kim and Oh, 2001).

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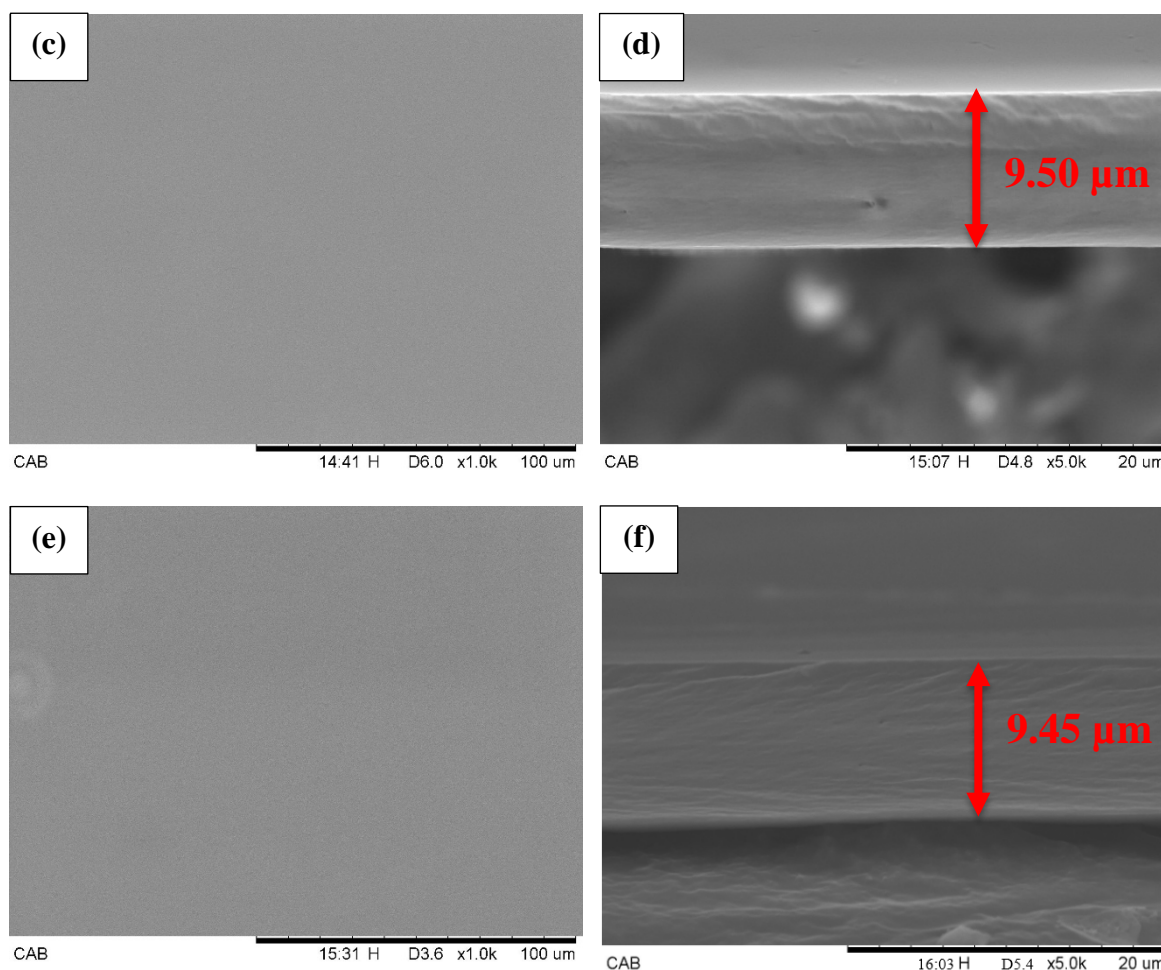


Fig 9. Surface and cross-sectional SEM of CAB membrane dried with 30 minutes of isopropyl alcohol first then followed by; (a-b) 15 minutes (CAB-15H), (c-d) 30 minutes (CAB-30H), and (e-f) 60 minutes (CAB-60H) of solvent exchange time using n-hexane, at casting thickness of 250 μm and 5 minutes solvent evaporation time

According to the CO_2 permeance results displayed in **Fig. 10**, there was clear indication that CAB-60H showed the highest CO_2 permeance rate followed by CAB-30H and subsequently, by CAB-15H. As seen in **Fig. 10**, the CO_2 permeance increased significantly from 21.55 ± 0.03 GPU to 227.95 ± 0.39 GPU when the solvent exchange time increased from 15 minutes (CAB-15H) to 30 minutes (CAB-30H). This was because when the exchange time was increased, sufficient time was provided for the exchange of the isopropyl alcohol content with n-hexane and therefore, generating a relatively thinner and compact cross-sectional membrane, which favoured CO_2 permeation through the membrane (Jawad et al., 2015b). In addition, the CO_2 permeance increased further when the solvent exchange duration was increased from 30 minutes to 60 minutes, as observed from CAB-30H (227.95

± 0.39 GPU) to CAB-60H (262.29 ± 0.16 GPU). The increase in the CO₂ permeance could be related to the increase in the number of the remaining polar n-hexane molecules within the membrane structure, resulting in a more active interaction with the CO₂ molecules and hence, higher CO₂ permeance yield (Jawad et al., 2015b).

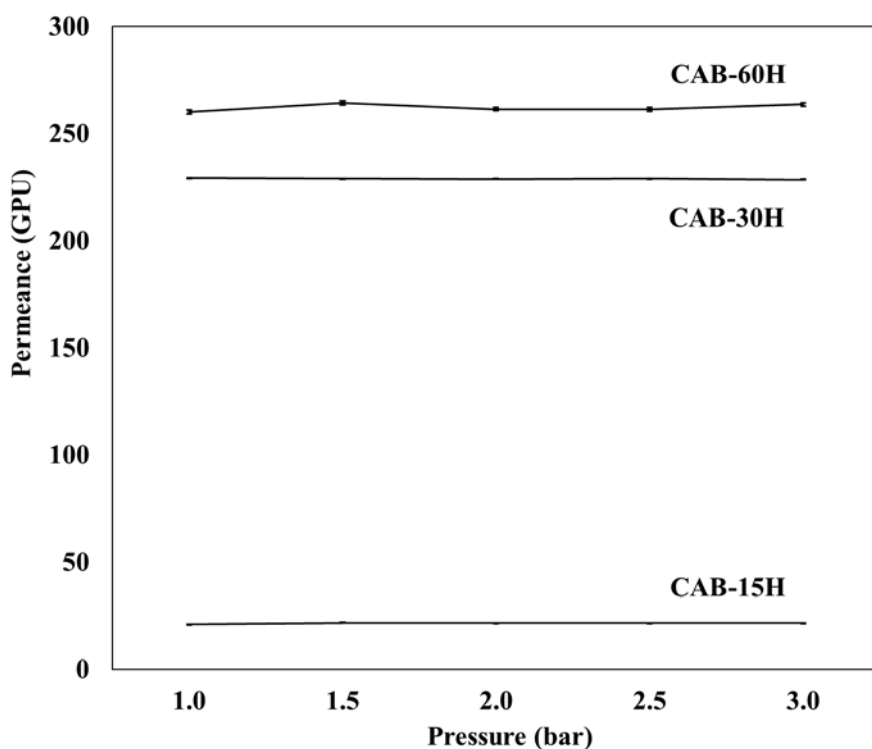


Fig 10. CO₂ permeance for membrane dried with 15 minutes (CAB-15H), 30 minutes (CAB-30H), and 60 minutes (CAB-60H) of n-hexane, at casting thickness of 250 μ m and 5 minutes solvent evaporation time

271 Meanwhile, **Fig. 11** illustrated a drastic increase of N₂ permeance from 10.03 ± 0.02 GPU to 37.28
 272 ± 0.54 GPU when the solvent exchange time of n-hexane was increased from 15 minutes (CAB-15H)
 273 to 30 minutes (CAB-30H). The reason for this increment was mainly due to the thin dense membrane
 274 structure of CAB-30H (9.50 ± 0.10 μ m), which allowed the feed of N₂ gas to pass through a least
 275 resistance pathway. However, the high N₂ permeance for CAB-60H (70.49 ± 0.33 GPU) was due to
 276 stress of surface tension caused by high capillary forces because of the evaporation of residual n-hexane
 277 within the membrane, which led to the collapse in the structure (Matsuyama et al., 2002).
 278

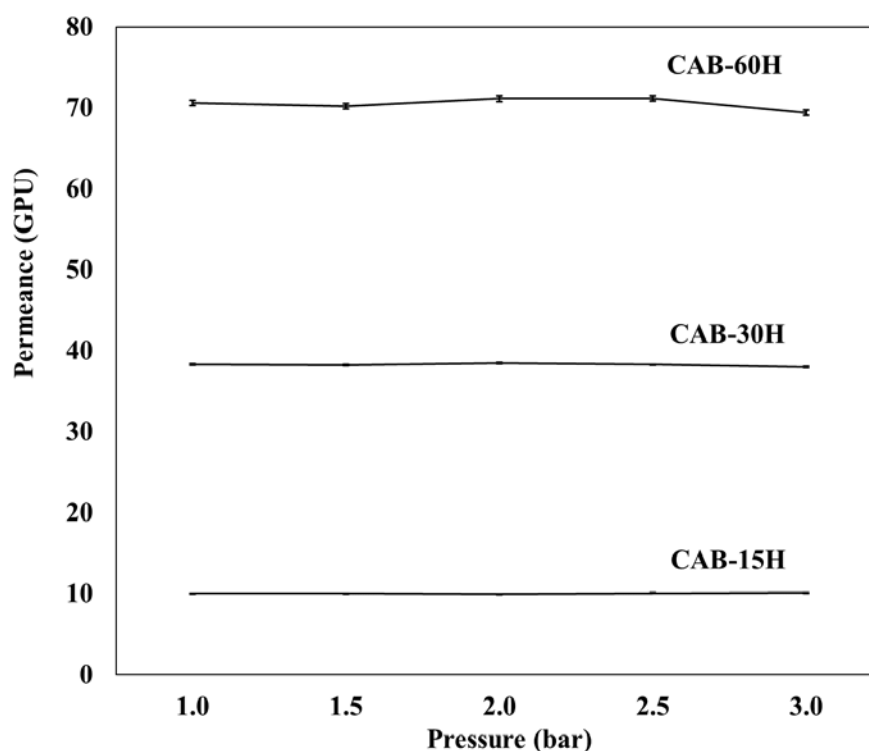
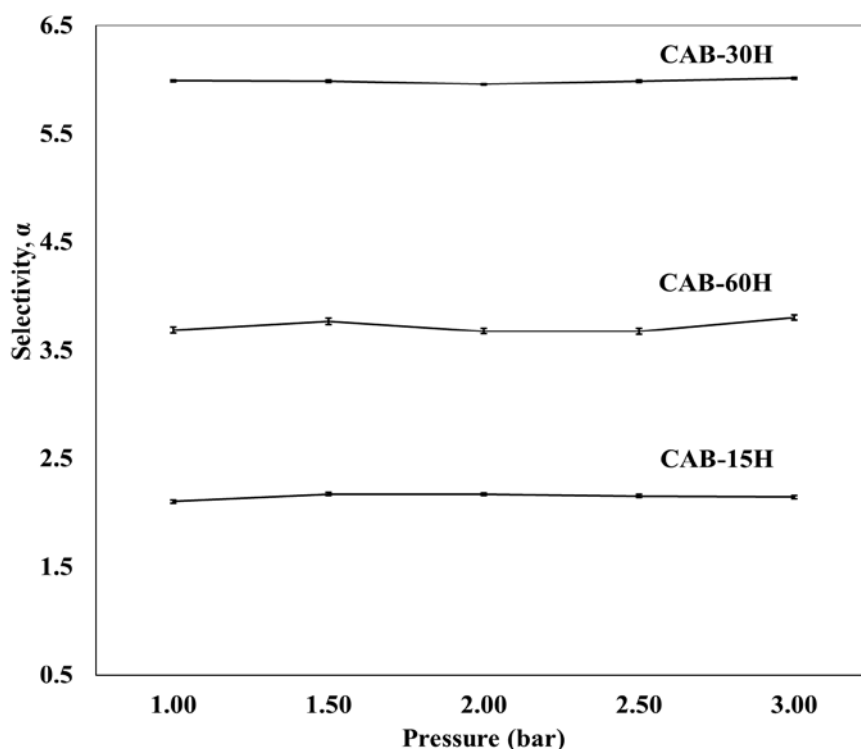


Fig 11. N₂ permeance for membrane dried with 15 minutes (CAB-15H), 30 minutes (CAB-30H), and 60 minutes (CAB-60H) of n-hexane, at casting thickness of 250 μm and 5 minutes solvent evaporation time

279 As seen in **Fig. 12**, the CAB-30H membrane showed the highest gas selectivity, which was
 280 achieved at 6.12 ± 0.09 . This result further proved that to have a high gas separation performance a
 281 smooth surface with regular thin dense membrane morphology was preferable (**Figs. 9c and d**) (Huang
 282 and Feng, 1995, Jansen et al., 2005, Matsuyama et al., 2002, Lui et al., 1988). On the other hand, CAB-
 283 15H showed a lower separation performance of 2.15 ± 0.17 . This was due to the collapse in the
 284 membrane structure caused by the short solvent immersion time, thereby generating an uneven porous
 285 surface and thick dense membrane structure, as presented in **Figs. 9a and b**. However, CAB-60H
 286 exhibited a smooth surface and thinner dense membrane morphology ($9.45 \pm 0.06 \mu\text{m}$), as depicted in
 287 **Figs. 9e and f**. In addition, the low selectivity performance for CAB-60H (3.72 ± 0.03) was as a result
 288 of the excessive exchange time with n-hexane, which deformed the functionality of the membrane and
 289 therefore, generated moderate selectivity performance (Budd et al., 2005).

290

291



292
 293 **Fig 12. Ideal selectivity of CO₂/N₂ for CAB membrane dried with 30 minutes of isopropyl alcohol**
 294 **first then followed by; 15 minutes (CAB-15H), 30 minutes (CAB-30H), and 60 minutes**
 295 **(CAB-60H) of solvent exchange with n-hexane, at casting thickness of 250 μm and 5**
 296 **minutes solvent evaporation time**

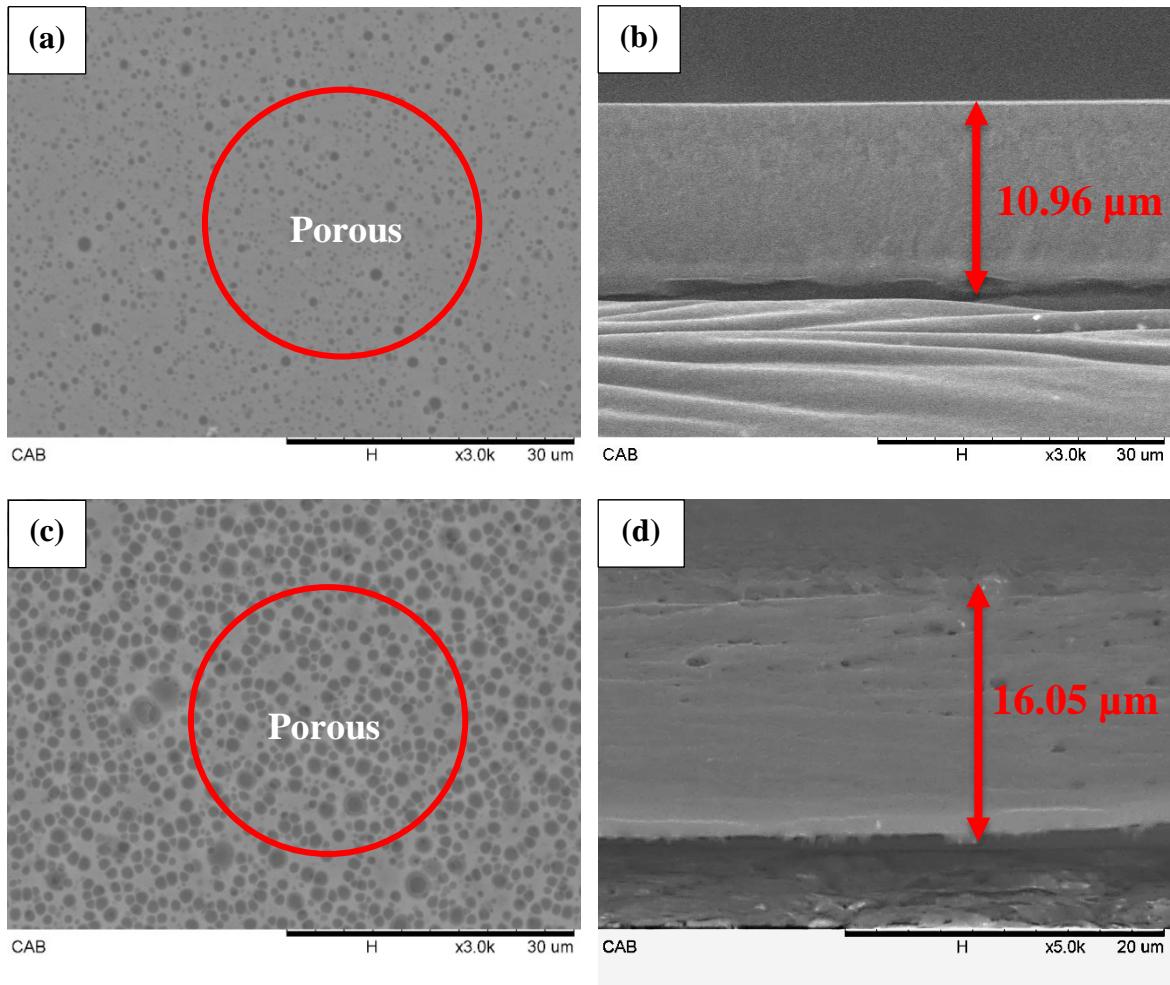
297 3.4 The Effect of CAB polymer at different molecular weight (M_n)

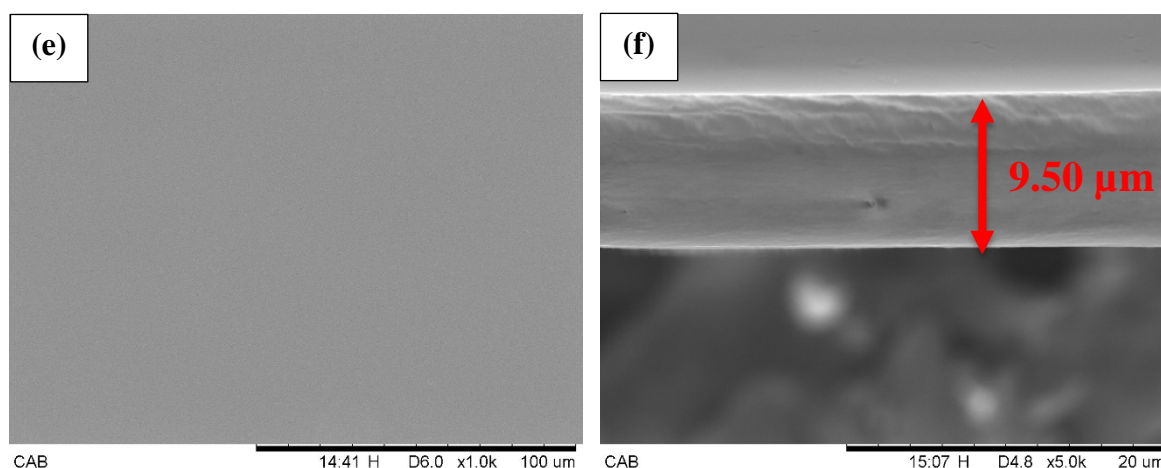
298
 299 According to Coltelli et al. (2008), the acetyl group has been deduced to have prominent effect on the
 300 membrane gas separation performance, as excessive acetyl composition in the membrane could promote
 301 plasticisation within the membrane (Coltelli et al., 2008, Ismail and Lorna, 2002). Thus, different CAB
 302 molecular weights with different acetyl, butyryl, and hydroxyl groups were investigated, as
 303 demonstrated in **Fig. 13**.

304 As depicted in **Figs. 13a** and **c**, a porous structure was observed for both CAB-12000 ($M_n=12000$)
 305 and CAB-65000 ($M_n=65000$), while CAB-70000 ($M_n=70000$) showed a smooth surface (**Fig. 13e**). The
 306 reason the membrane surface changed from porous to smooth was due to the high molecular weights of
 307 CAB, which caused the increase in the number of entanglements between the macromolecular chains in
 308 the solution (Jansen et al., 2006). Therefore, the high molecular weights of CAB favoured the gelation
 309 of the polymer rich phase after the phase-inversion occurred and hence, suppressed the formation of the
 310 porous structure during the early stages (Jansen et al., 2005).

311 Based on **Figs. 13b, d, and f**, the thickness of CAB-12000, CAB-65000, and CAB-70000 were
312 10.96 ± 0.10 , 16.05 ± 0.17 , and 9.50 ± 0.10 μm , respectively. The increment in the CAB molecular
313 weights further influenced the membrane thickness through the rheological properties of the casting
314 solution (Jansen et al., 2005). This was due to the high molecular weights of the CAB polymer being
315 utilised for membrane fabrication, which gave the rapid gelation (Jansen et al., 2005). After the rapid
316 gelation, the porous structure was greatly suppressed and further evaporation of solvent and non-solvent
317 from the polymer matrix resulted in gradual shrinkage of the structure (Jansen et al., 2005). Therefore,
318 the thickness of CAB-70000 (9.50 ± 0.10 μm) was thinner than CAB-12000 (10.96 ± 0.10 μm) and
319 CAB-65000 (16.05 ± 0.17 μm).

320





321 **Fig 13. Surface and cross-sectional SEM of CAB membranes prepared with polymer**
 322 **concentration of 4 wt% and molecular weights (M_n) of (a-b) 12000 (CAB-12000), (c-d)**
 323 **65000 (CAB-65000), and (e-f) 70000 (CAB-70000), at casting thickness of 250 μm and 5**
 324 **minutes solvent evaporation time**

325
 326 The performance of CO_2 permeance achieved for the different molecular weights (M_n) of CAB-
 327 12000, CAB-65000, and CAB-70000 were 101.42 ± 0.97 , 74.37 ± 1.25 , and 227.95 ± 0.39 GPU,
 328 respectively, as shown in **Fig. 14**. The decrease in the CO_2 permeance rates observed from CAB-12000
 329 (28-31 wt%) to CAB-65000 (16-19 wt%) was due to the thick dense membrane morphology as presented
 330 in **Fig. 13d** ($16.05 \pm 0.17 \mu\text{m}$), which can hindrance the CO_2 permeance (Jawad et al., 2015a).
 331 Meanwhile, CAB-12000, which exhibited greater membrane thickness of $10.96 \pm 0.10 \mu\text{m}$, contradicted
 332 the results with higher CO_2 permeance, as illustrated in **Fig. 14**. The possible explanation for the increase
 333 in CO_2 permeance was caused by the acetyl groups rigidity and steric effects (Wan et al., 2003).
 334 Therefore, this allowed the higher intrinsic solubility of CO_2 due to the greater number of acetyl–acetyl
 335 interactions that existed (Koros et al., 1988b, Scholes et al., 2012). In addition, increasing the CAB
 336 molecular weight from 65000 to 70000 had increased the permeance rate drastically from 74.37 ± 1.25
 337 GPU to 227.95 ± 0.39 GPU. Even though, CAB-70000 (12-15 wt%) has the lowest acetyl-acetyl
 338 interactions due to low acetyl group composition compared to other CAB polymers. The significant
 339 increase in the CO_2 permeance was due to the thin dense membrane exhibited for CAB-70000, as thin
 340 dense membrane usually impose less flux resistance for the membrane (Pandey and Chauhan, 2001).
 341 Therefore, the permeance of CO_2 was highest among all as the membrane thickness was the thinnest.

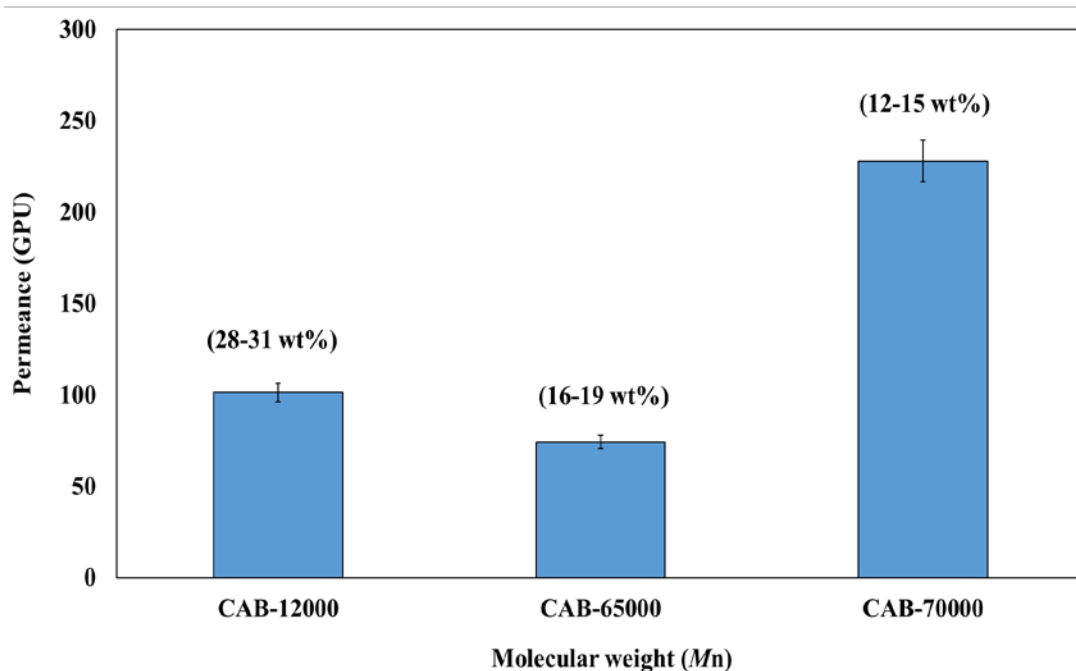


Fig 14. CO₂ permeance results for CAB membranes fabricated at different molecular weights comprising CAB-12000, CAB-65000, and CAB-70000 acetyl content of 28-31 wt%, 16-19 wt%, and 12-15wt%, respectively

342 As portrayed in **Fig. 15**, the N₂ permeance rate achieved for CAB-12000, CAB-65000, and CAB-
 343 70000 were 95.26 ± 1.06 , 48.94 ± 0.89 , and 37.28 ± 0.54 GPU, respectively. The reduction in N₂
 344 permeance was due to the high presence of the hydroxyl group (1.2-2.2 wt%) content within the CAB-
 345 70000 polymer. The reaction between the hydroxyl and carbonyl groups of the CAB polymer caused
 346 the formation of hydrogen bonds, which could delay the de-mixing between the coagulant and the non-
 347 solvent. This resulted in the smooth homogeneous formation of the membrane surface, which could
 348 influence the N₂ permeance rate (Childress and Elimelech, 1996). Thus, it may be deduced that with the
 349 increment of the hydroxyl group within the membrane composition, the formation of a homogeneous
 350 surface morphology was favoured. Further, the hydroxyl group can increase the preferential restrictions
 351 on membrane pore formation, whereby the permeance and diffusion coefficient can be suppressed,
 352 hence, enhancing the selectivity performance of the membrane (Yave et al., 2009).
 353

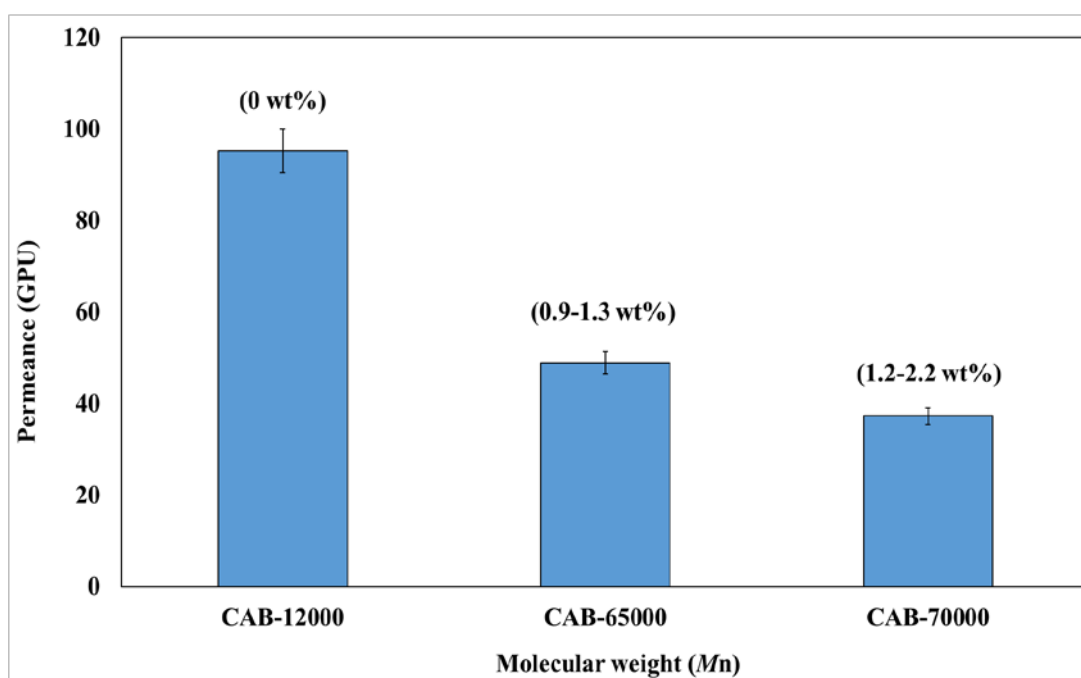


Fig 15. N₂ permeance results for CAB membranes synthesised at different molecular weights comprising CAB-12000, CAB-65000, and CAB-70000 hydroxyl content of 0 wt%, 0.9-1.3 wt%, and 1.2-2.2 wt%, respectively

354 **Fig. 16** reveals the selectivity results for the different CAB molecular weights of 12000 (CAB-
 355 12000), 65000 (CAB-65000), and 70000 (CAB-70000), respectively. From the selectivity results
 356 depicted in **Fig. 16**, CAB-70000 achieved the average highest selectivity of 6.12 ± 0.09 , followed by
 357 CAB-65000 with a moderate selectivity of 1.52 ± 0.04 , and CAB-12000 with the lowest selectivity of
 358 1.06 ± 0.01 . The high selectivity performance of CAB-70000 was due to the high presence of the butyryl
 359 group content (35-39 wt%), which promoted the CO₂ diffusion better due to the increase of the non-
 360 polar butyryl chain within the structure of the membrane, thus, making the membrane more hydrophobic
 361 in nature (Wan et al., 2004, Ong et al., 2012).

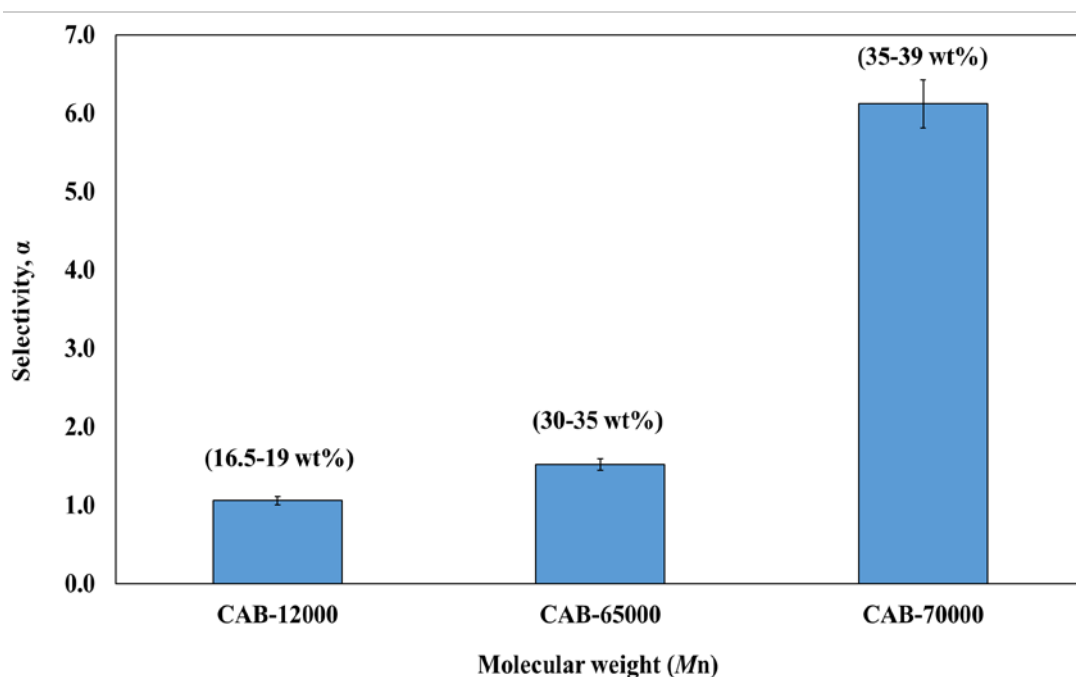


Fig 16. CO₂/N₂ selectivity results for CAB membranes at different molecular weights comprising CAB-12000, CAB-65000, and CAB-70000 butyryl content of 16.5-19 wt%, 30-35 wt%, and 35-39 wt%, respectively

362

363 3.4.1 X-ray Photoelectron Spectroscopy (XPS) Analysis

364 The XPS characterisation was adopted in this study to analyse the quantitative element composition of
 365 the CAB membrane fabricated. The quantitative element composition of the membrane surface can be
 366 determined from the spectrum obtained. Consequently, CAB-12000, CAB-65000, and CAB-70000 were
 367 analysed through XPS analysis. The surface chemical quantitative compositions are depicted in **Table**
 368 **1** and **Fig. 17**, respectively.

369 Observing the results tabulated in **Table 1**, both the atomic and mass concentration of the oxygen
 370 (O) atom decreased with the increase in the CAB membrane molecular weights. The decreasing trend
 371 of atomic concentration from 34.02>30.88>27.30% and mass concentration from 40.72>37.31>33.35%
 372 of the O atom was due to the decrease of the acetyl group derived from each CAB polymer
 373 (Suttiwijitpukdee et al., 2011). As indicated clearly in **Fig. 14**, the acetyl group affected the permeance
 374 of CO₂ within the membrane. Hence, this further proved that increasing the acetyl group or O atom
 375 presence within the membrane subsequently, decreased the permeance of CO₂. The increase in the O
 376 element was mainly funded by the breaking of the carbonyl (C=O) group and prompted the formation

377 of a new carboxyl group (-COOH) (Liu et al., 2014). The increase in carboxyl group made the membrane
378 more hydrophilic, resulting in decline of the CO₂ permeance flux (Xia and Ni, 2015, Xu et al., 2014).

379 **Table 1. Element composition of the CAB membrane synthesized at different molecular weight**

Peak	CAB-12000		CAB-65000		CAB-70000	
	Atomic Conc %	Mass Conc %	Atomic Conc %	Mass Conc %	Atomic Conc %	Mass Conc %
O 1s	34.02	40.72	30.88	37.31	27.30	33.35
C 1s	65.98	59.28	69.12	62.69	72.70	66.65

380
381 On the other hand, when observing the carbon (C) element present within CAB-12000, CAB-
382 65000, and CAB-70000, the C atoms increased with increase in the polymer molecular weights. The
383 atomic concentration increased from 65.98>69.12>72.70 and the mass concentration increased from
384 59.28>62.69>66.65 for CAB-12000, CAB-65000, and CAB-70000, respectively. The increase in the C
385 element within the membrane was because of the increase in the butyryl group within the CAB polymer.
386 As indicated in **Fig. 16**, the butyryl group played a crucial role in manipulating the selectivity
387 performance of the membrane, because it can increase the CO₂ diffusion due to the increase of the non-
388 polar butyryl chain within the structure of the membrane (Wan et al., 2004). As a result the membrane
389 became more hydrophobic in nature, and hence, promoted better CO₂ permeance flux (Ong et al., 2012).

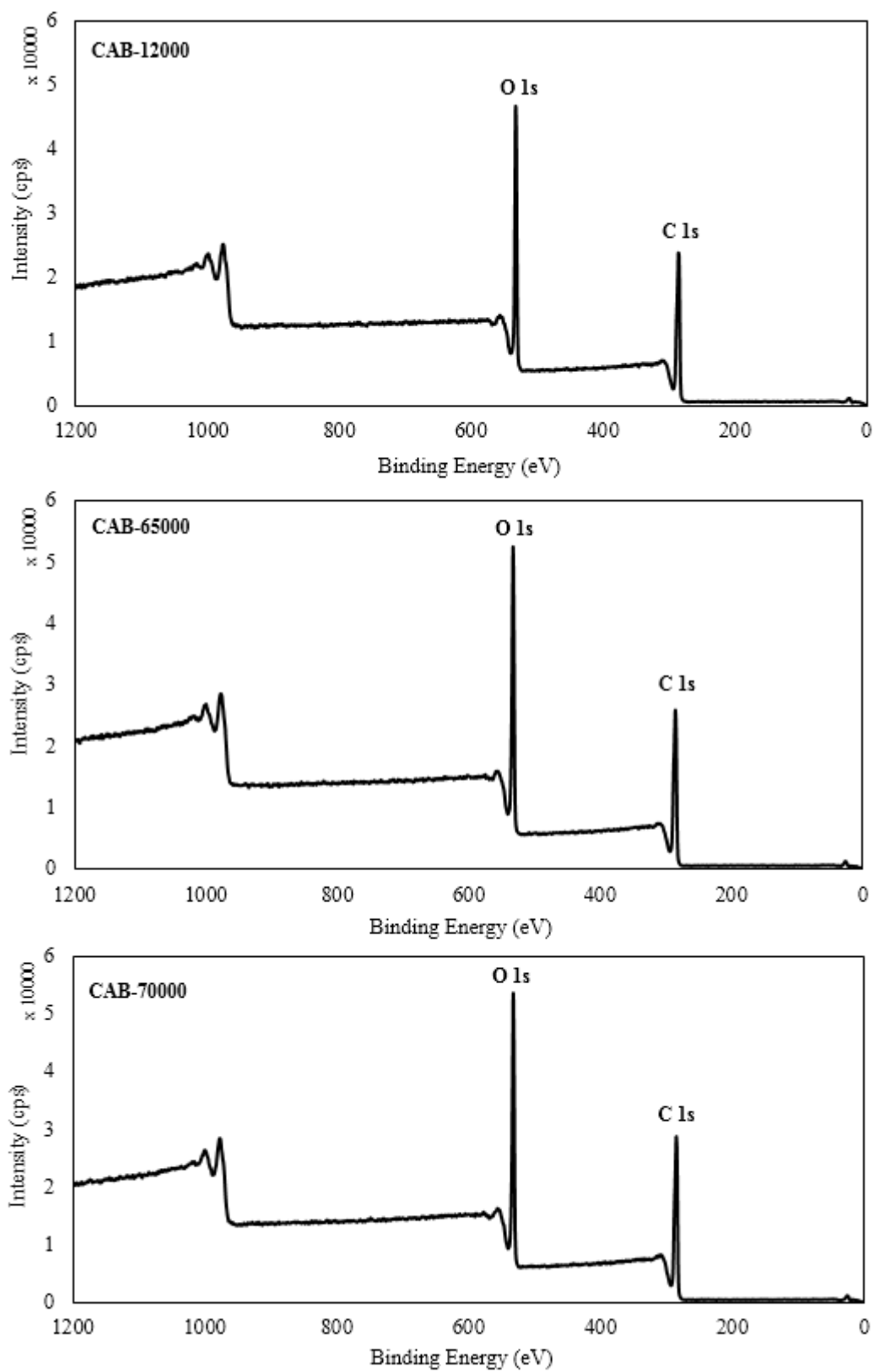


Fig 17. Element composition of XPS spectrum of CAB-12000, CAB-65000, and CAB-70000

390 The CO₂/N₂ separation performance of this current study were summarised and compared with
 391 other research works, as shown in **Table 2**. In this study, the best membrane performance achieved for
 392 both CO₂ permeance and selectivity was 227.95 ± 0.39 GPU and 6.12 ± 0.09, respectively for CAB-
 393 70000. This was a result of the dynamic contents of acetyl, butyryl, and hydroxyl composition present
 394 in CAB polymer. The acetyl and butyryl contributed significantly towards the permeance of CO₂/N₂ by
 395 enhancing the solubility of CO₂ within the membrane structure. As compared to other research works,
 396 the permeance results achieved in the present work showed higher CO₂/N₂ permeance with acceptable
 397 selectivity result. The selectivity performance achieved for this study shows the typical trade-off
 398 relationship of polymer membrane due to the high permeance results and low selectivity of the CAB
 399 polymer. Nevertheless, the low selectivity of the CAB polymer can eventually be overcome by
 400 incorporating the polymer matrix with inorganic filler to produce the hybrid system of mixed matrix
 401 membranes (MMMs) (Aroon et al., 2013, Chung et al., 2007, Ismail et al., 2009, Goh et al., 2011).

402 **Table 2. List of CO₂/N₂ permeation results achieved from current study with previous work**

Polymer	P (CO ₂)	P (N ₂)		Conditions	References
CAB	164.84 ± 0.73 ^a	26.36 ± 0.05 ^a	6.06 ± 0.23	1-3 x10 ⁵ Pa, casting thickness of 250µm	Present work
SPEEK	5.01	1.94	5.58	1-1.5x10 ⁵ Pa, 25 °C, casting thickness of 60-80µm	(Xin et al., 2015)
BPPO	76.78 ^b	N/A	30	0.7x10 ⁵ Pa, casting thickness of 50-90 µm	(Cong et al., 2007)
PES	10.98 ^a	0.80 ^a	13.73	3-4x10 ⁵ Pa, casting thickness of 150µm	(Ismail et al., 2011)
6FDA-durene	30.3 ^b	2.87 ^b	10.56	35°C, 10atm, casting thickness of 40µm	(Liu et al., 2001)

403 SPEEK- sulfonated polyetheretherketone, BPPO- brominated polyphenylene oxide, PES- polyethersulfone,
 404 6FDA- 2,2-bis(3,4-dicarboxyphenyl) hexafluoropropane dianhydride

405 ^a GPU. ^b Barrer. N/A- not available.

406

407 **4.0 Conclusion**

408 The optimisation of membrane morphology conducted with respect to the different parameters was
 409 found to be successful for the preparation of the highly selective CAB gas separation membrane. The

410 membrane formation and morphology were closely related to the rheological behaviour of the casting
411 solution. The results have shown that membrane casting thickness, solvent exchange duration for both
412 isopropyl alcohol and n-hexane, and the molecular weights of the CAB polymer had a significant role
413 in manipulating the CO₂/N₂ gas separation performance as well as the morphology of the membranes.
414 Under optimised conditions, the best membrane was found to be the CAB-70000, which was fabricated
415 with 4 wt% polymer concentration, 250 µm casting thickness, 5 minutes solvent evaporation time, 30
416 minutes solvent exchange with isopropyl alcohol followed by another 30 minutes of solvent exchange
417 with n-hexane. Moreover, the CAB-70000 had the best gas separation performance with an average
418 selectivity of 6.12 ± 0.09 and permeance up to 227.95 ± 0.39 GPU for CO₂ and 37.28 ± 0.54 GPU for
419 N₂, respectively. The superior CO₂/N₂ separation performance of the membrane was mainly contributed
420 by the quality formation of the smooth surface, with thin dense and defect-free membrane structure.
421 Further, it has been suggested that to improve the performance of the CAB membrane, inorganic
422 nanoparticle fillers such as carbon nanotubes (CNTs) be incorporated to produce mixed matrix
423 membrane (MMM).

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425
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430
431 **Keywords:** Membrane gas separation; Cellulose acetate butyrate; casting thickness; solvent exchange
432 time; isopropyl alcohol; n-hexane; molecular weight; functional groups; Scanning Electron Micrograph;
433 X- ray Photoelectron Spectroscopy

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