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$\begin{array}{c} \mbox{MAGNETIC AND CRYSTALLOGRAPHIC ORDER IN} \\ \mbox{α-MANGANESE} \end{array}$

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We have made time-of-flight neutron diffraction measurements on α -manganese metal. Powder diffraction measurements were made at 14 temperatures between 15 and 305K, and single crystal measurements were made at 15 and 300K. We found that the crystal structure of α -Mn is tetragonal below its Néel point of 100K, with crystal symmetry $I\bar{4}2m$ and magnetic (Shubnikov) symmetry $P_I\bar{4}2_1c$. In agreement with the earlier results of Yamada et al., there are six independent magnetic atoms, and we found that their moments are weakly temperature dependent. The onset of magnetic order causes slight changes in the atomic positions and in the average atomic elastic constant.

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I. INTRODUCTION

 α -manganese has a surprisingly complex cubic crystal structure with 58 atoms per body-centered cell.¹ The complexity is surprising because most elements have simple structures. In the case of Mn the complexity is thought to arise from an instability of the 3d electron shell that gives rise to the formation of "self-intermetallic" compounds. In other words, elemental Mn is actually an intermetallic compound between Mn atoms in different electronic configurations. This possibility was first pointed out by Bradley and Thewlis.¹

 α -Mn becomes anti-ferromagnetic at 95 K. Given the crystal structure, the magnetic structure is necessarily complex. This structure was solved by Yamada et al.², who used single crystal neutron diffraction data and first principles magneto-structural analysis of heroic proportions.²³⁴ We wished to examine the magnetic ordering in more detail by obtaining structural data at a large number of temperatures. The multi-temperature requirement dictated the use of powder neutron diffraction. However, we found that information from single crystal data was essential to the complete solution of the problem.

II. EXPERIMENTAL METHOD

a-Mn was prepared by heating Johnson Matthey Grade I electrolytic Mn in a turbomolecular pumped system in the β-phase region until outgassing was complete and then cooling slowly through the α - β transition. The material was cooled from 900C to 755C over 5 hours, from 755C to 655C over 200 hours, and finally furnace cooled to room temperature. Powder was prepared by grinding the resulting material. Small single crystals were cut from an arc-melted ingot that was prepared from this material and annealed for a long time just below the α - β transition. Neutron powder diffraction data were taken on an α -Mn sample at 14 temperatures between 15 and 305K. We used the High Intensity Powder Diffractometer (HIPD) and the Manuel Lujan, Jr., Neutron Scattering Center (LANSCE) at the Los Alamos National Laboratory.⁵ We also obtained single crystal diffraction data at 15 and 300K on the Single Crystal Diffractometer (SCD) at LANSCE. Powder data were analyzed by Rietveld analysis using General Structure Analysis System (GSAS).⁶ This refinement package allows for the refinement of magnetic reflections arising from structures that can be described by Shubnikov groups. Powder data from six detector banks were co-refined in the analysis. Allowance was made for 0.90% (vol.) of MnO that was present in the powder and for some weak aluminum lines from the cryostat. The single crystal data were also refined using GSAS. In this case, magnetic multi-domain effects were treated as a twinning problem...

III. RESULTS AND DISCUSSION

Since there was no evidence for an incommensurate structure, our approach to the analysis of our data was guided by the use of Shubnikov groups.⁷⁸ These are magnetic space groups that are supergroups of the ordinary crystallographic space groups, with certain symmetry elements replaced by "anti"-elements. The anti-elements are the ordinary translations, mirror planes, etc., familiar from crystallography, except that the spatial symmetry operator is augmented by a time-reversal operator that reverses the microscopic current driving the magnetic moment of the atom to which it is applied. In this way 1651 magnetic space groups can be obtained from the original 230 ordinary space groups. Only commensurate magnetic structures are described by the Shubnikov groups.

The space group of α -Mn is I $\overline{4}3m$. The neutron diffraction data indicate a bodyanti-centered lattice, so that the Shubnikov group $P_1\overline{4}3m$, or one of its subgroups is indicated. The cubic and rhombohedral subgroups did not work, and we found that the best fit was obtained with the tetragonal Shubnikov group $P_1\overline{4}2'm' \equiv P_1\overline{4}2c$, which is a subgroup of the tetragonal space group I $\overline{4}2m$. This is equivalent to the magnetic model of Yamada et al.² The meaning of the symbol $P_1\overline{4}2'm'$ is the same as the ordinary space group symbol I $\overline{4}2m$, except that (1) the ordinary body-centering operator is replaced by an operator that flips the spin (reverses the current) as the body centering is applied, (2) the mirror planes are replaced by operators that reverse the current as a magnetic atom is reflected, and (3) the two-fold axes are replaced by operators that flip the spin (reverse the current) as the dyad operation is applied to a magnetic atom. (The distinction between spin-flipping and reversing the current is important only for improper symmetry operators such as mirror planes.)

In addition to the group $P_1\bar{4}2$ 'm', we also tried fits with $P_1\bar{4}2m$, $P_1\bar{4}'2m'$ and $P_1\bar{4}'2'm$. (Of these four, only the first does not exclude a magnetic moment on site I by symmetry.) The models with $P_1\bar{4}2m$ and $P_1\bar{4}'2m'$ can be excluded by the single crystal data, which clearly show the presence of (300), (500), etc. magnetic reflections, in agreement with Yamada et al ². This is indicative of the presence of the two-fold anti-axis 2°. These reflections are not directly observable in the powder data because of overlapping allowed reflections, so that powder data cannot distinguish between 2 and 2'. The quality of the powder fits for $P_1\bar{4}2'm'$ and $P_1\bar{4}'2m'$ is very similar, so that single crystal data was essential for making this distinction. However, the use of powder data is the

only practical way of making observations at a large number of temperatures in a reasonable amount of time. The combination of both types of experiment was necessary in this case.

Since the magnetic structure of α -Mn is tetragonal, a tetragonal nuclear (or chemical) structure is implied. This structure was found by Rietveld refinement. The space group is $1\overline{4}2m$, and there are six unique atoms in the tetragonal unit cell. Above the Néel point, the structure is cubic with four unique atoms in the cell. There are 58 atoms per unit cell in each case.

Fig. 1 shows the lattice constants of α -Mn plotted versus temperature. We emphasize that no tetragonal splitting of the diffraction lines was observed directly; rather the metric tetragonality was found by testing the powder data with Rietveld refinement. Simultaneous refinements of the strain broadening were well-behaved, and sensible atomic positions were found, so we have some confidence in this result. The average of the tetragonal lattice constants agrees with previous diffraction measurements for which the low-temperature structure was taken to be cubic.^{9 10}

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The magnetic moments are plotted versus temperature in Fig. 2 These were obtained from the powder refinements using the form factors for the 3d⁵4s² state of neutral Mn determined by Freeman and Watson. ¹¹ Thus we did not allow for a variation of the form factor among the atoms as did Yamada et al ². The moments we found are somewhat different from theirs, but the significant qualitative feature of groups of large and small moments is retained. The large moments on atoms I and II are somewhat temperature dependent, and the smaller moments on the remaining atoms are not.

The atomic positions also show the effects of magnetic ordering. Below the Néel temperature, the original four crystallographic sites of the cubic structure (Table 1) split into the six sites the tetragonal structure (Table 2). The atomic positions obtained from the powder refinements are plotted versus temperature in Fig. 3. The resulting magnetic structure is shown in Fig. 4 as a a-axis projection of the structure. Each sphere represents a Mn atom: the diameter of the sphere represents the magnitude of the magnetic moment, the arrow represents the projection of the magnetic moment in the tetragonal structure. (A different view of the magnetic structure is given in the compilation by Wijn.¹²) The Cartesian components of the magnetic moments at 15K are also shown in Table 2.

Magnetic ordering in α -Mn causes a stiffening of the lattice. Fig. 5 shows the average mean-square thermal displacement, <u2>, plotted versus temperature. The <u2> is obtained from the Debye-Waller factors from the Rietveld analysis of the powder data; it

is averaged over atomic sites and direction.^{13 14} We have made separate fits above and below the Néel temperature to the $<u^2>$ versus temperature data with a simple Debye model. The increase in Θ_{DW} observed at the Néel point (from 438 to 536K) is quite large.

IV. SUMMARY

We have made a study of the temperature dependence of the magnetic structure of α -Mn using single crystal and powder neutron diffraction. In agreement with the results of Yamada et al., the magnetic structure is tetragonal, with Shubnikov group $P_1 \overline{4}2$ 'm' = $P_1 \overline{4}2c$. The magnetic moments fall into two groups, with one group having much smaller moments than the other, and the larger moments are weakly temperature dependent. Magnetic ordering causes some changes in atomic position and elastic behavior.

ACKNOWLEDGMENTS

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Tables

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Table 1. Atomic co-ordinates for α -Mn at 305K.

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| | 1 | | | IV |
|---|---|------------|------------|------------|
| x | 0 | 0.31836(5) | 0.35776(3) | 0.09081(4) |
| у | u | u | u | 16 |
| z | u | ** | 0.03543(4) | 0.28276(5) |

Table 2. Atomic co-ordinates and magnetic moments (μ_B) for α -Mn at 15K.

| | l | 11 | -1 | 111-2 | IV-1 | IV-2 |
|----|----------|-----------|-----------|-----------|-----------|-----------|
| x | 0 | 0.3192(2) | 0.3621(1) | 0.3533(2) | 0.0921(2) | 0.0895(2) |
| у | 56 | u | 4 | 0.0333(1) | 11 | 0.2850(1) |
| Z | " | 0.3173(3) | 0.0408(2) | 0.3559(2) | 0.2790(3) | 0.0894(2) |
| mx | 0 | 0.14(12) | 0.43(8) | -0.25(10) | 0.27(8) | -0.08(4) |
| m, | " | " | 46 | -0.25(10) | u | -0.45(8) |
| mz | 2.83(13) | 1.82(06) | 0.43(8) | -0.32(4) | -0.45(8) | 0.48 (5) |

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Figure Captions

Fig. 1. Lattice constants versus temperature for α -Mn. The error bars are smaller than the plotted points.

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Fig. 2. Magnetic moments versus temperature for α -Mn.

Fig. 3. Positional parameters for α -Mn. The error bars are smaller than the plotted points.

Fig. 4. Magnetic structure of α -Mn, a-axis projection. The diameter of each sphere represents the magnitude of the magnetic moment. The inscribed arrows represent the projection of the moment in the basal plane of the tetragonal unit cell.

Fig. 5. Average mean square thermal displacement, $<u^2>$, versus temperature for α -Mn. The lines through the points are fits to a simple Debye-Waller formula. The fitted Debye-Waller temperatures are different above and below the Néel temperature.

References

⁴ T. Yamada and S. Tazawa, J. Phys. Soc. Japan, <u>28</u>, 609 (1970).

⁵A. C. Lawson, J. A. Goldstone, J. G. Huber, A. L. Giorgi, J. W. Conant, A. Severing, B. Cort and R. A. Robinson, J. Appl. Phys <u>69</u> 5112 (1991)

⁶A. C. Larson and R. B. Von Dreele, Los Alamos National Laboratory Report LAUR 86-748 (1986).

¹ A. J. Bradley and J. Thewlis, Proc. Roy. Soc. <u>A115</u> 456 (1927).

²T. Yamada, N. Kunitomi, Y. Nakai, D. E. Cox and G. Shirane, J. Phys. Soc. Japan, <u>28</u>, 615 (1970).

³ T. Yamada, J. Phys. Soc. Japan, <u>28</u>, 596 (1970).

⁷A. V. Shubnikov and N. V. Belov, Colored Symmetry John Wiley and Sons, New York (1964) 198-210.

⁸V. A. Koptsik, *Shubnikov Groups* [in Russian], Izd., Moskovsk. Gos. Univ., Moscow (1966).

⁹ J. A. C Marples, Physics Lett. <u>24A</u>, 207 (1967).

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¹⁰C. P. Gazzara, R. M. Middleton, R. J. Weiss and E. O. Hall, Acta Cryst. <u>22</u> 859 (1967).

¹¹ A. J. Freeman and R. E. Watson, Acta Cryst. <u>14</u> 231 (1961).

¹² H. P. J. Wijn, Magnetic Properties of Metals: d-Elements and Compounds, (New York, Springer-Verlag, 1991) p. 4-5.

¹³A. C. Lawson, A. Williams, J. A. Goldstone, D. T. Eash, R. J. Martinez, J. I. Archuleta, D. J. Martinez, B. Cort, and M. F. Stevens, J. Less-Common Metals <u>167</u> 353 (1991).

¹⁴A. C. Lawson, G. H. Kwei, J. A. Goldstone, B. Cort, R. I. Sheldon, E. Foltyn, J. Vaninetti,
D. T. Eash, R. J. Martinez and J. I. Archuleta, .Advances in X-Ray Analysis, Vol. 36, John
V. Gilfrich et al, eds. (New York, Plenum Press, 1993) p. 577.



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