

CONF-8403148--2

MATERIALS SCIENCE WITH SYNCHROTRON RADIATION :
THE CASE FOR A SUPERPROBE*

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presentation to the

MAJOR MATERIALS FACILITIES COMMITTEE
NATIONAL RESEARCH COUNCIL

March 17, 1984

Washington, D.C.

MASTER

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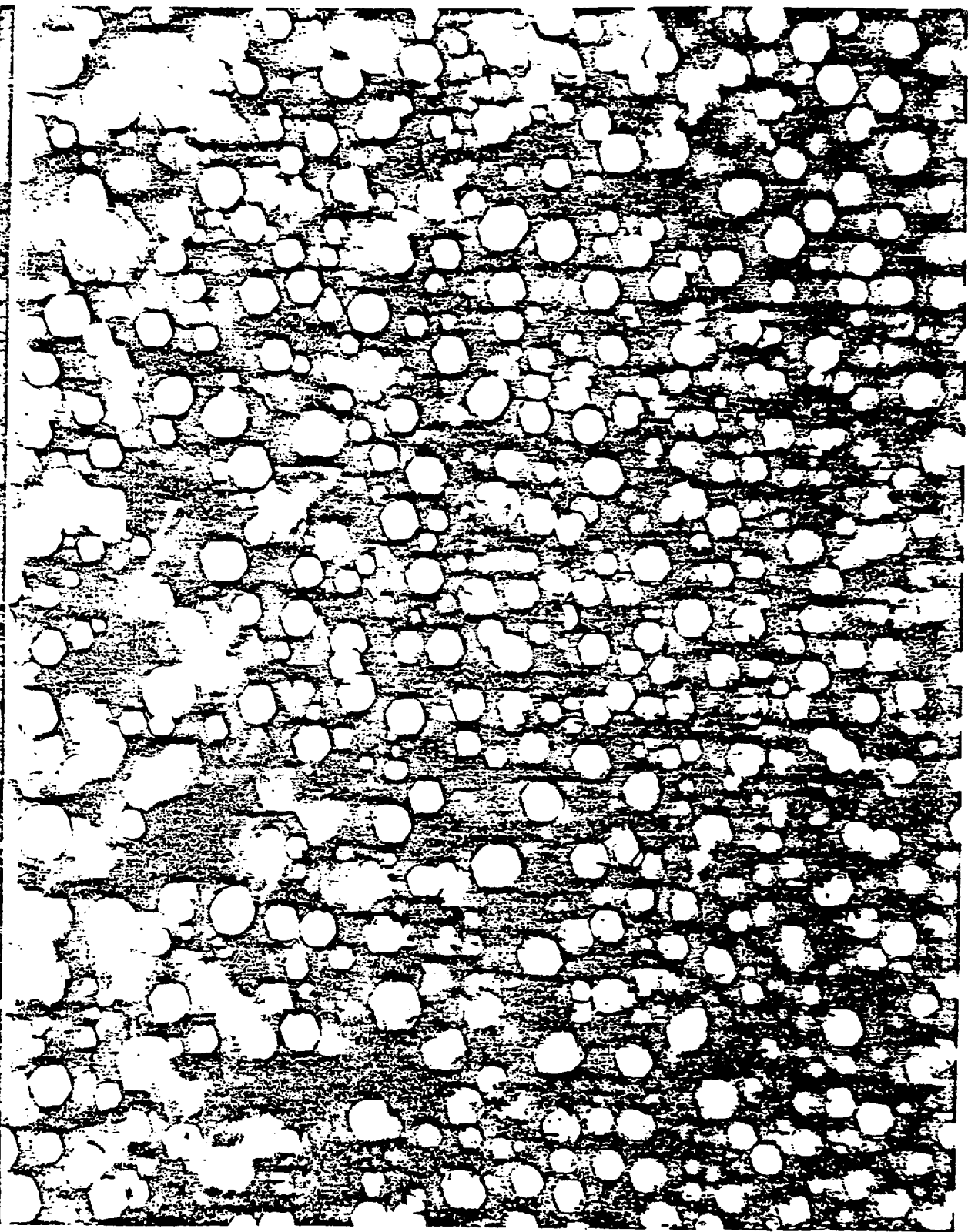
WE WILL TALK ABOUT THE FOLLOWING

- I. THE MECHANICAL STRENGTH, DUCTILITY, DIFFUSION, CORROSION, AND ELECTRICAL PROPERTIES OF MATERIALS ARE PROFOUNDLY MODIFIED BY DEFECTS (BOTH STRUCTURAL AND IMPURITY) ABOUT WHICH WE NEED MICRO-CHARACTERIZATION.
- II. AN X-RAY MICROPROBE IS THE NEXT LEAP FORWARD IN MICRO-CHARACTERIZATION.
- III. UNDULATORS ON A 6 GeV STORAGE RING WILL PROVIDE FOR AN X-RAY MICROPROBE OF EXTRAORDINARY SENSITIVITY.

H 78238

B

Fe-15Nx

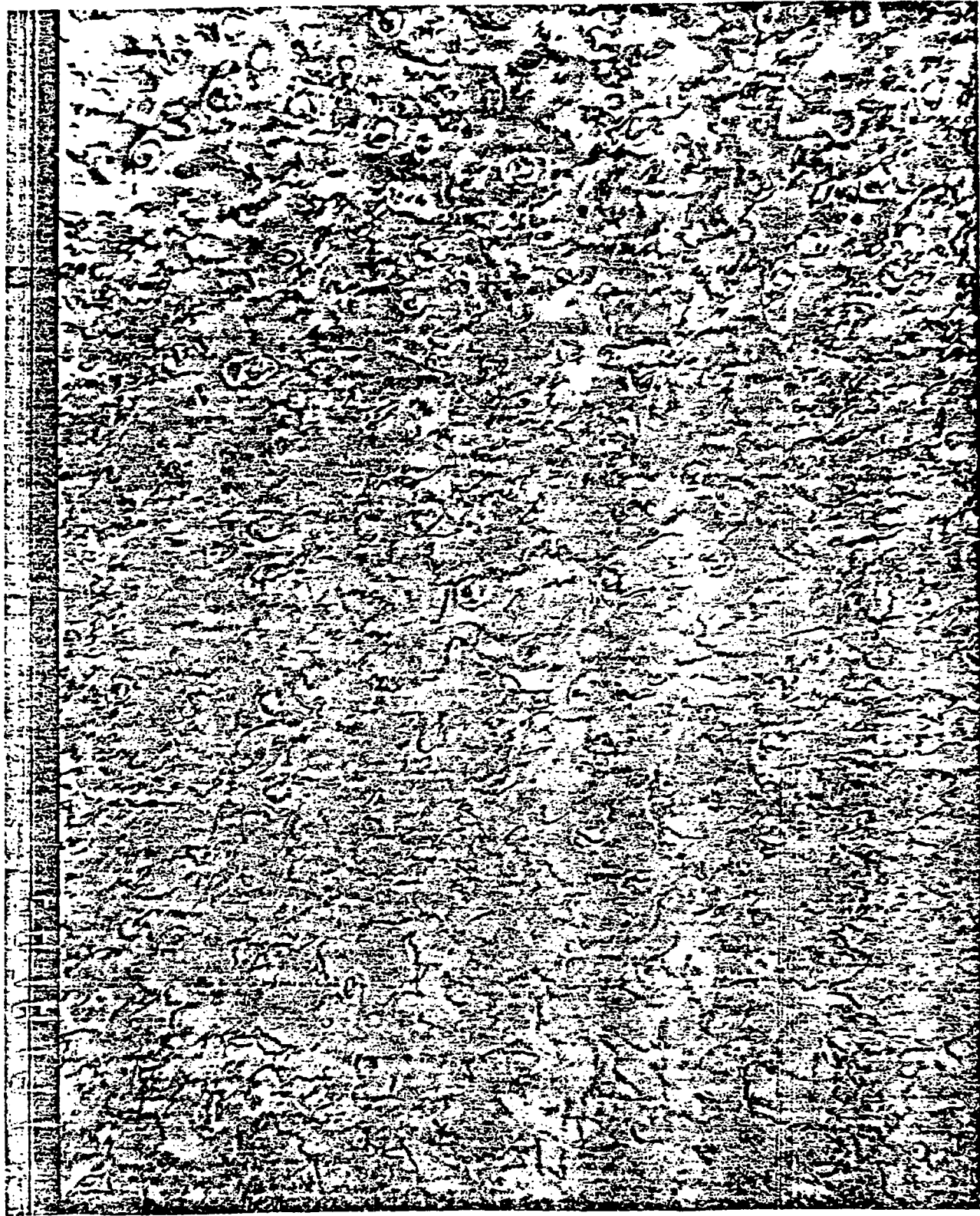


Yoder at 675°C (no He injection)

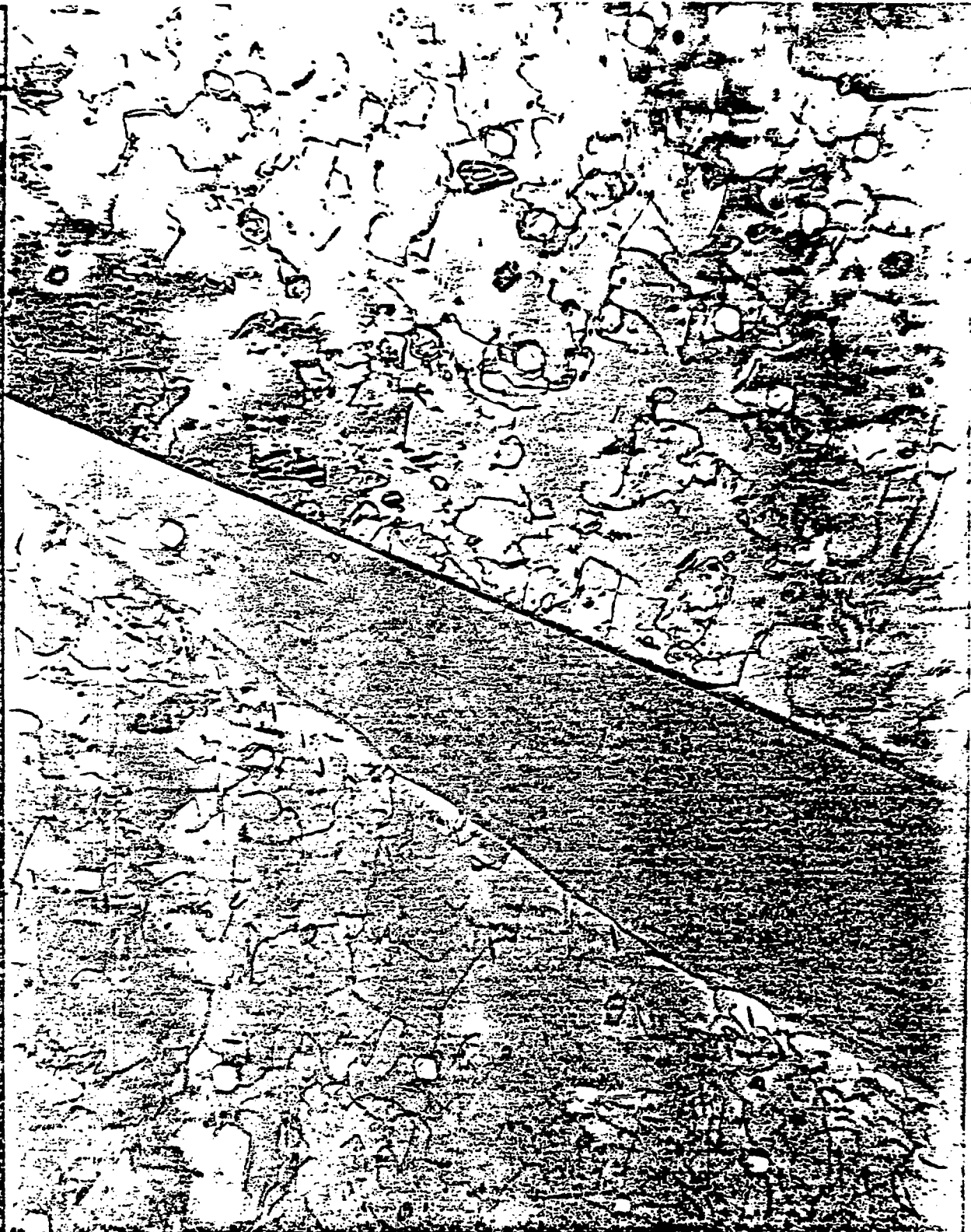
70 days at 675°C (no H₂ in jet)

#78197
B3 70

Fe-13Cr-15Ni-0.8Si

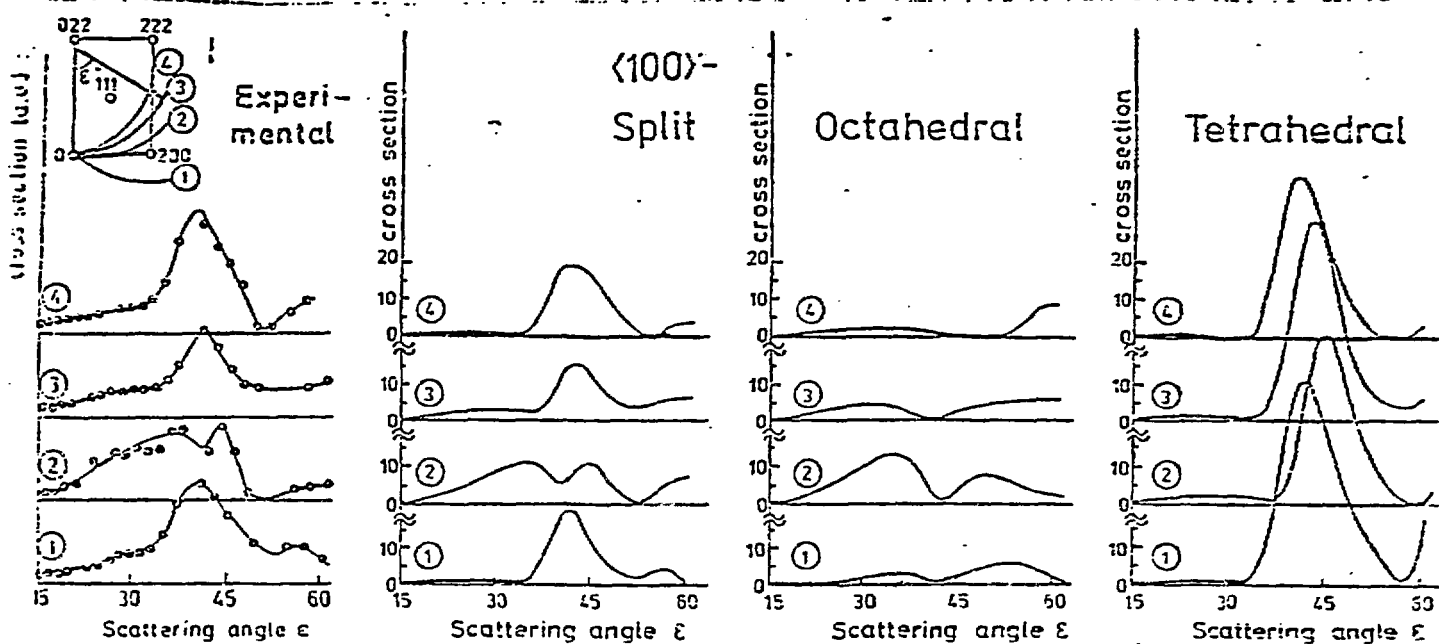


H78236 B7 Fe-13Cv-15Ni ± 0.2Ti



700x at 675°C (no He-injection)

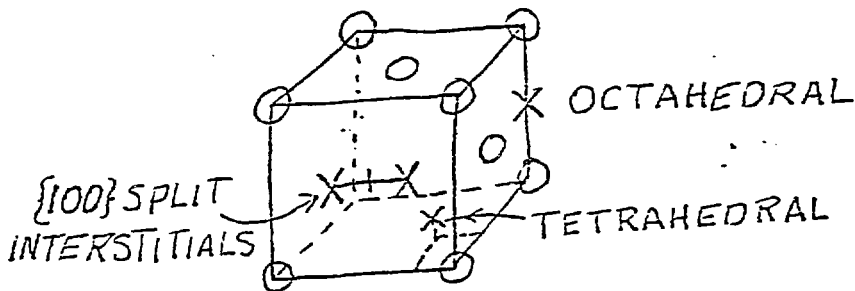
DIFFUSE X-RAY SCATTERING MEASUREMENTS SHOW INTERSTITIAL CONFIGURATION
IN ELECTRON IRRADIATED ALUMINUM AT 4°K

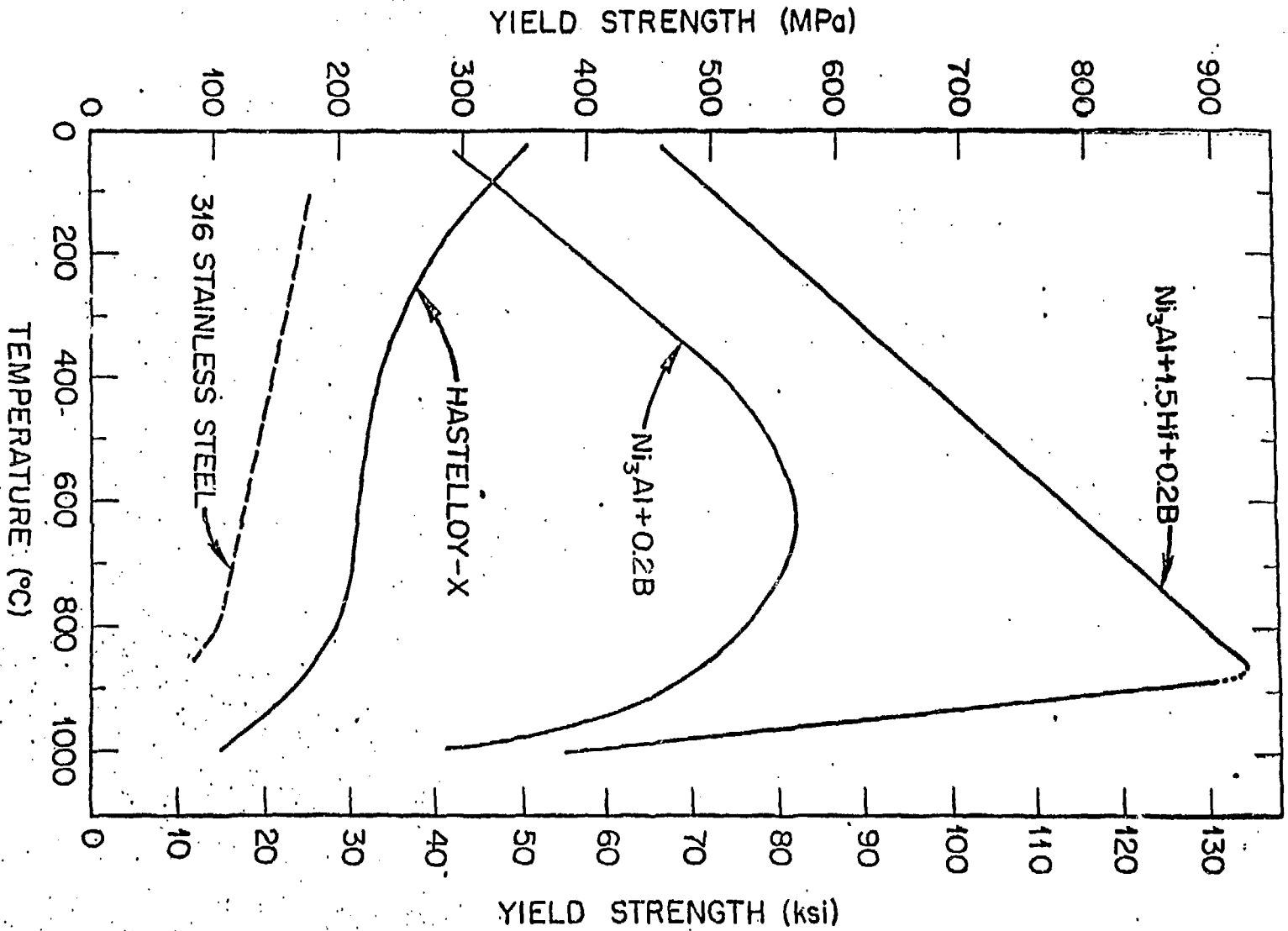


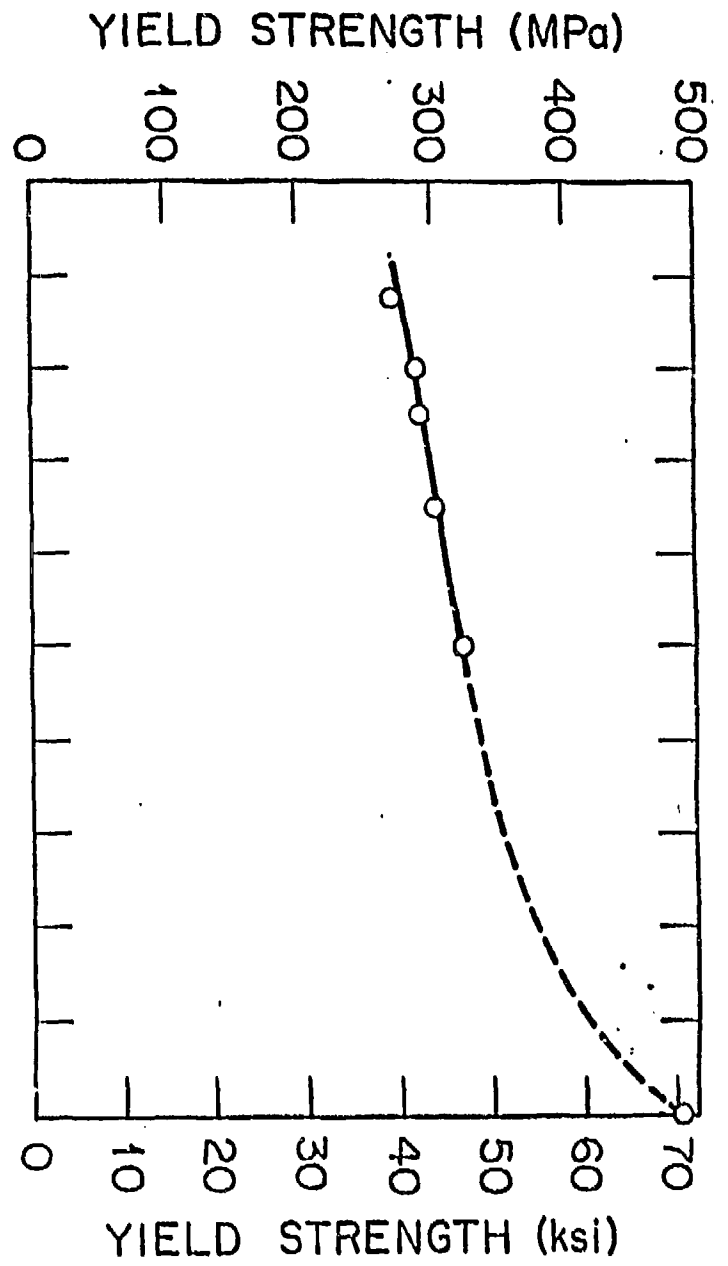
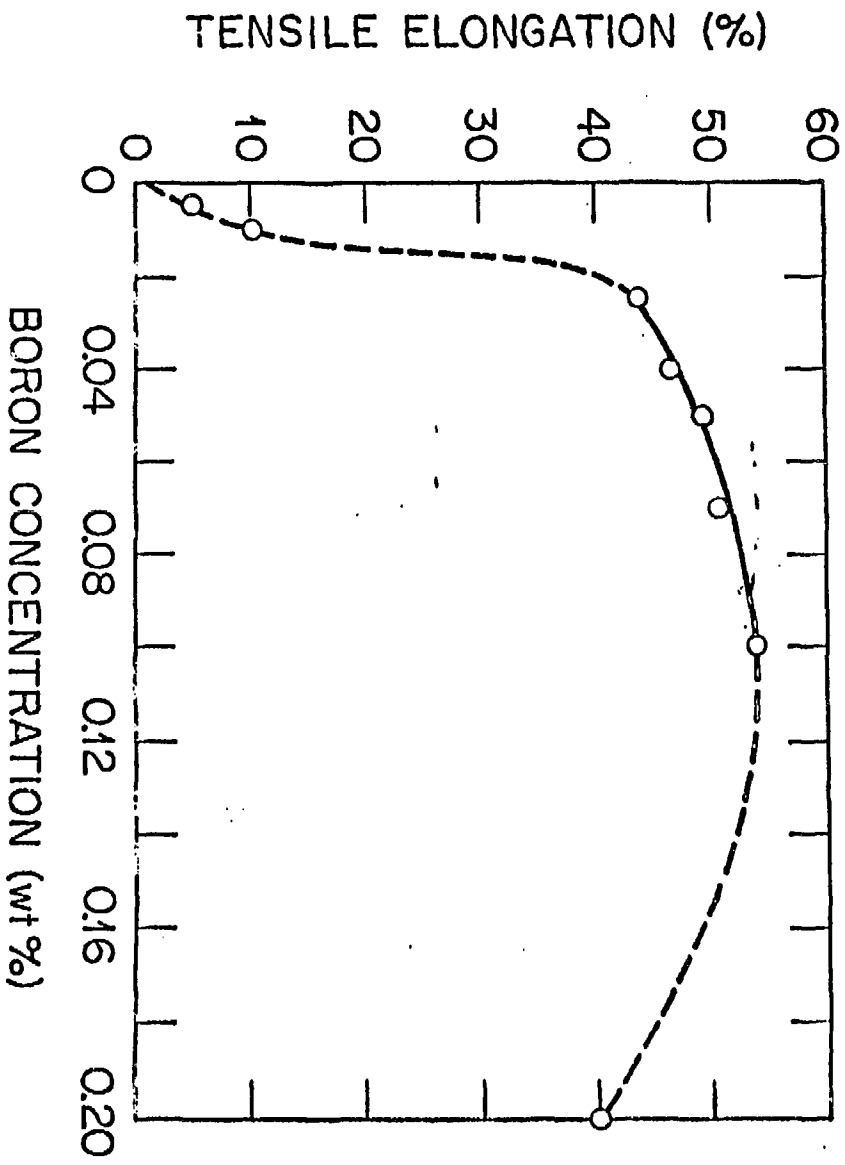
(P. Ehrhart, H. G. Haubold, and W. Schilling, Julich, Germany)

1 POINT DEFECT

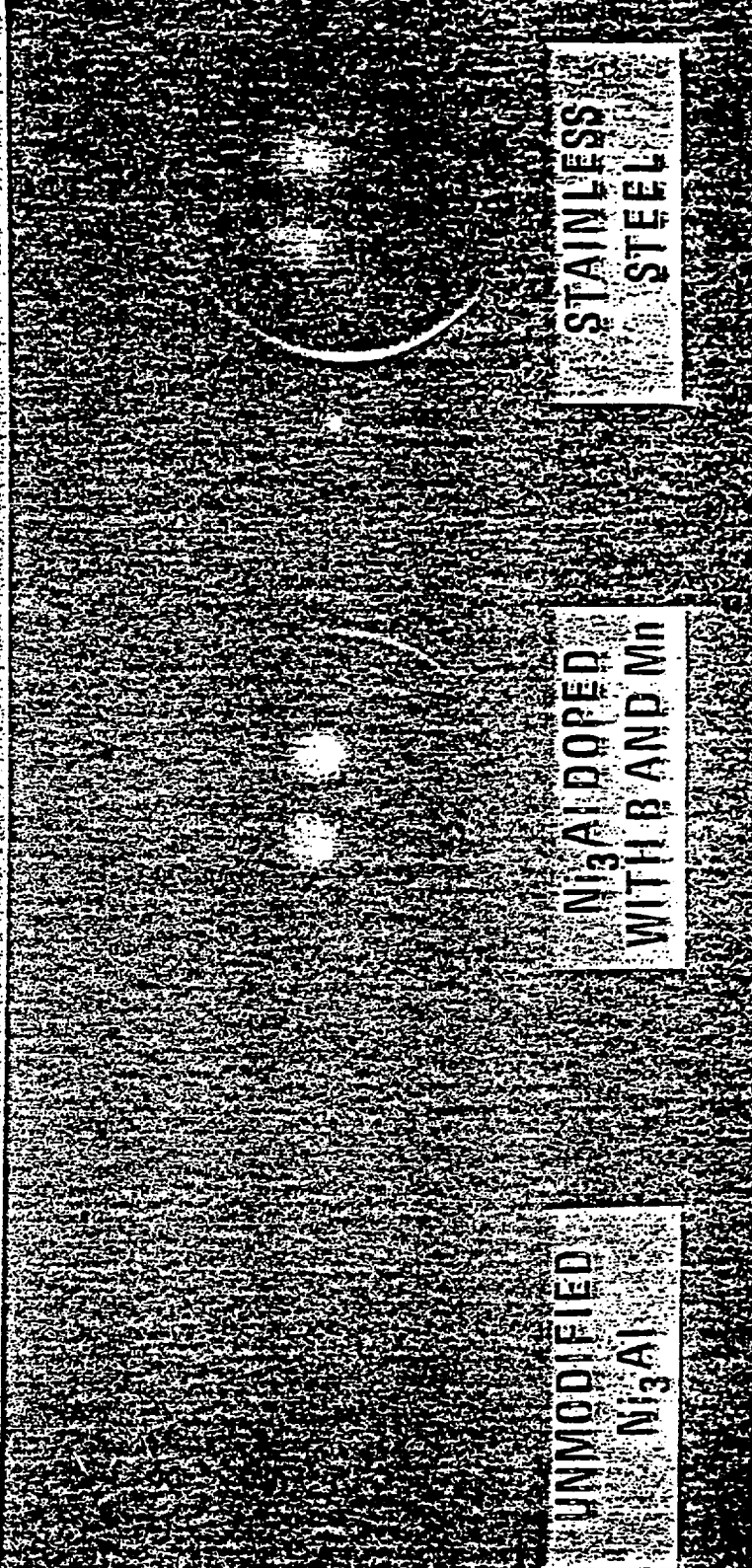
10,000 ATOMS







INTERMETALLIC ALLOYS BASED ON NICKEL ALUMINIDES
CAN BE MADE MORE DUCTILE THAN STAINLESS STEELS



UNMODIFIED
Ni₃Al

Ni₃Al DOPED
WITH B AND Mn

STAINLESS
STEEL

• THE ALUMINIDES ARE MUCH STRONGER THAN STAINLESS
STEELS AND SUPERALLOYS AT ELEVATED TEMPERATURES

PURE TiB_2 CERAMIC EXHIBITS EXTENSIVE MICROCRACKING DUE TO THERMAL EXPANSION ANISOTROPY

GRAIN BOUNDARY FRACTURE



TRANSGRANULAR FRACTURE

MICROCRACK TIP



CRACK TIP

30 μ m

POLARIZED LIGHT

HOT PRESSING CONDITIONS

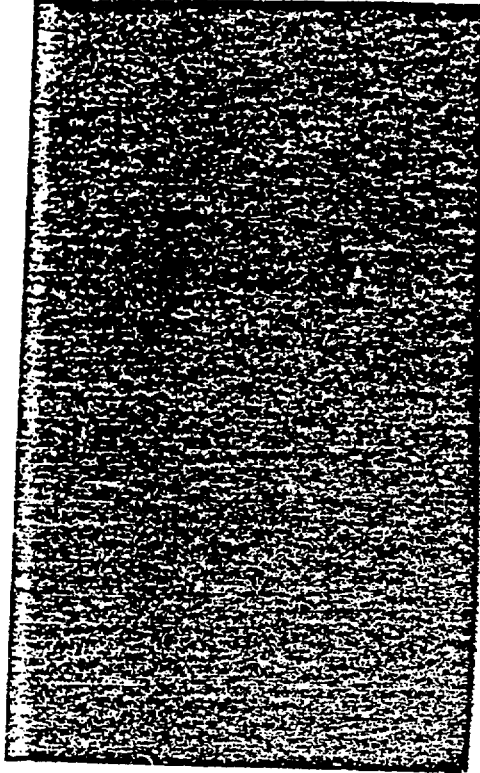
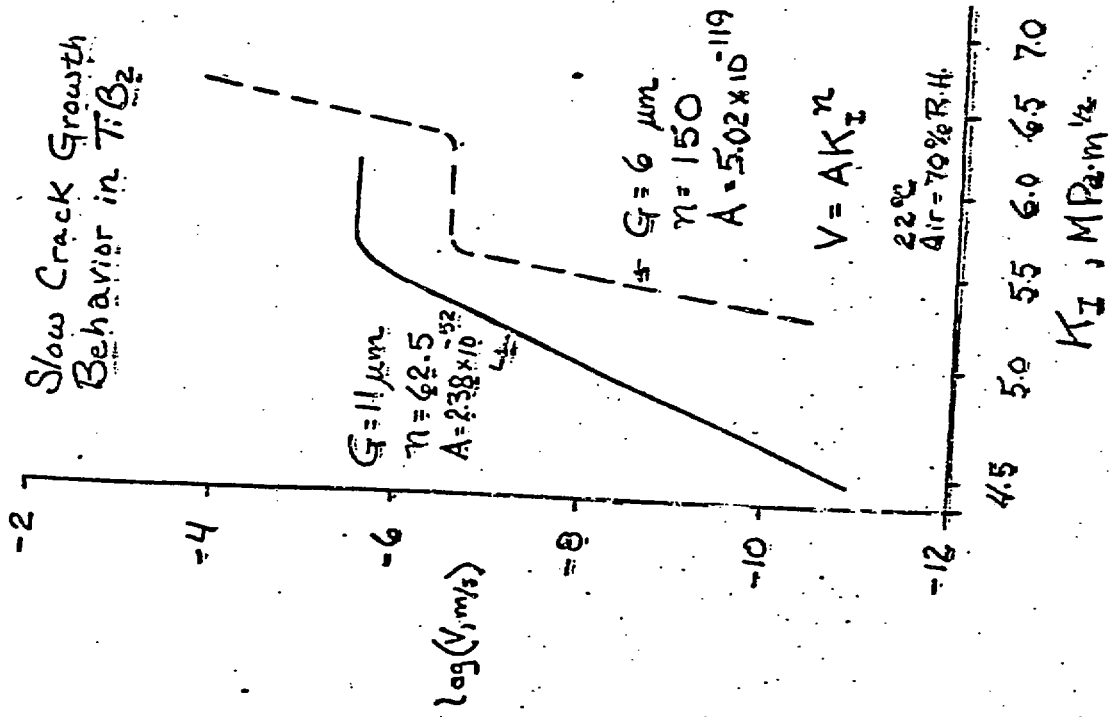
Temperature: 2050°C

Pressure: 25 MPa

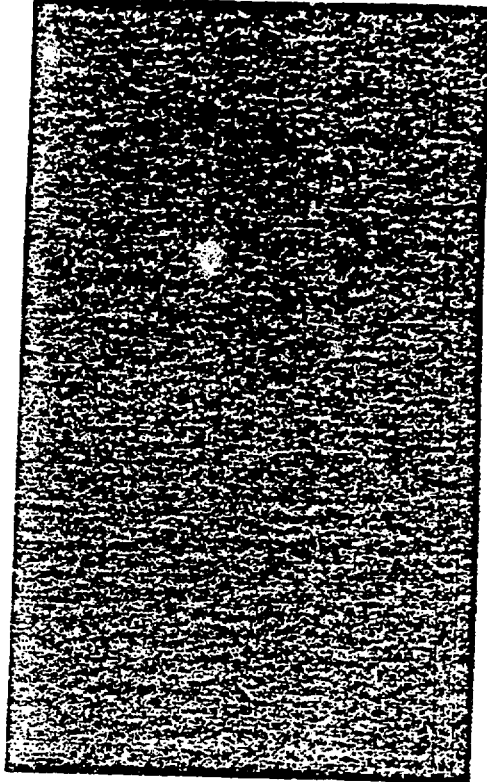
Time: 4 h

Atm: Vac

Slow Crack Growth Behavior in TiB_2

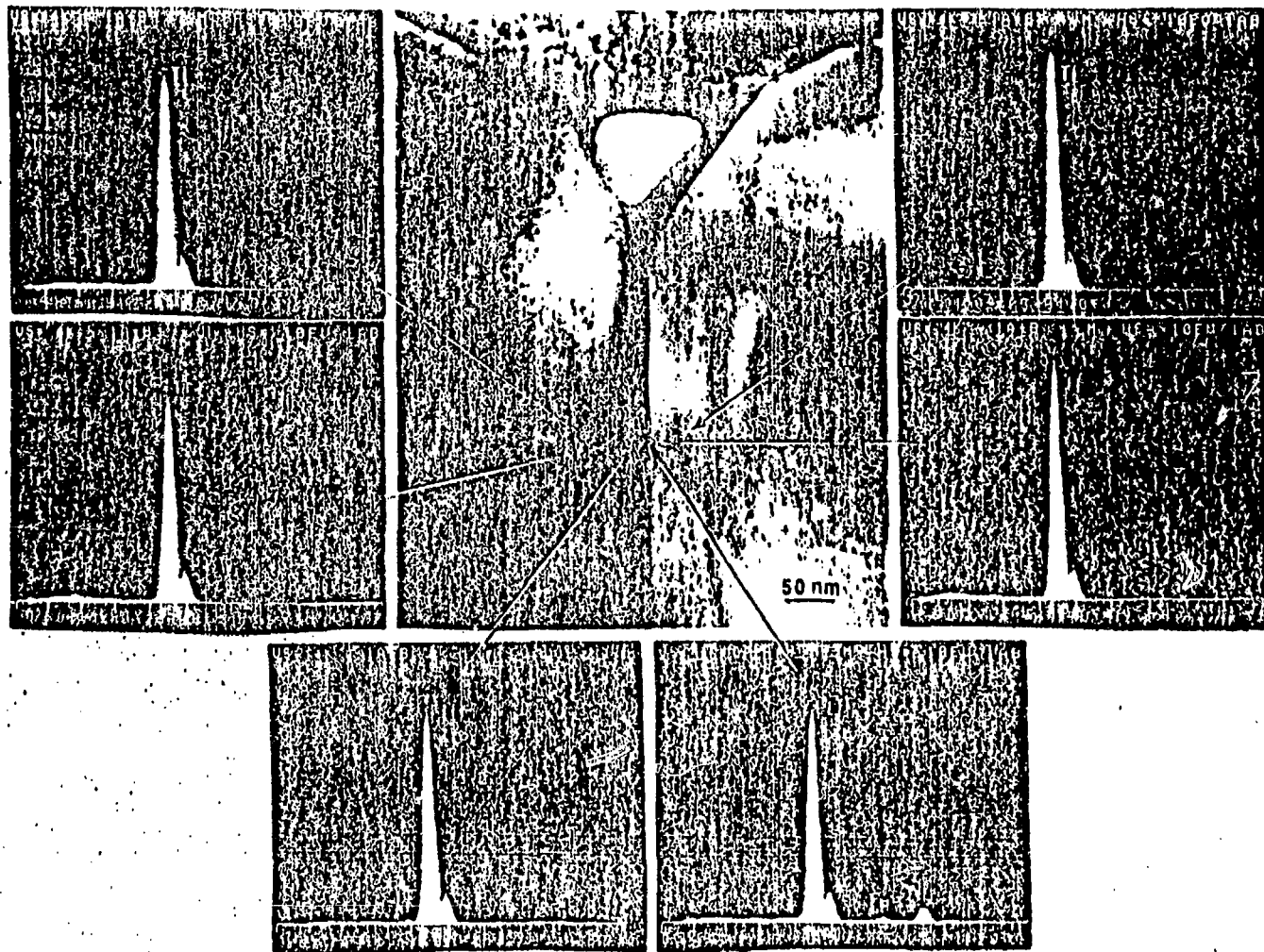


NT 90AD Gave 11 μm $\sigma_f = 450 MPa$ $\times 1000$
 N1 - 1.3 wt % E = 578 GPa
 O2 = 0.6 wt % $\nu = 0.115$



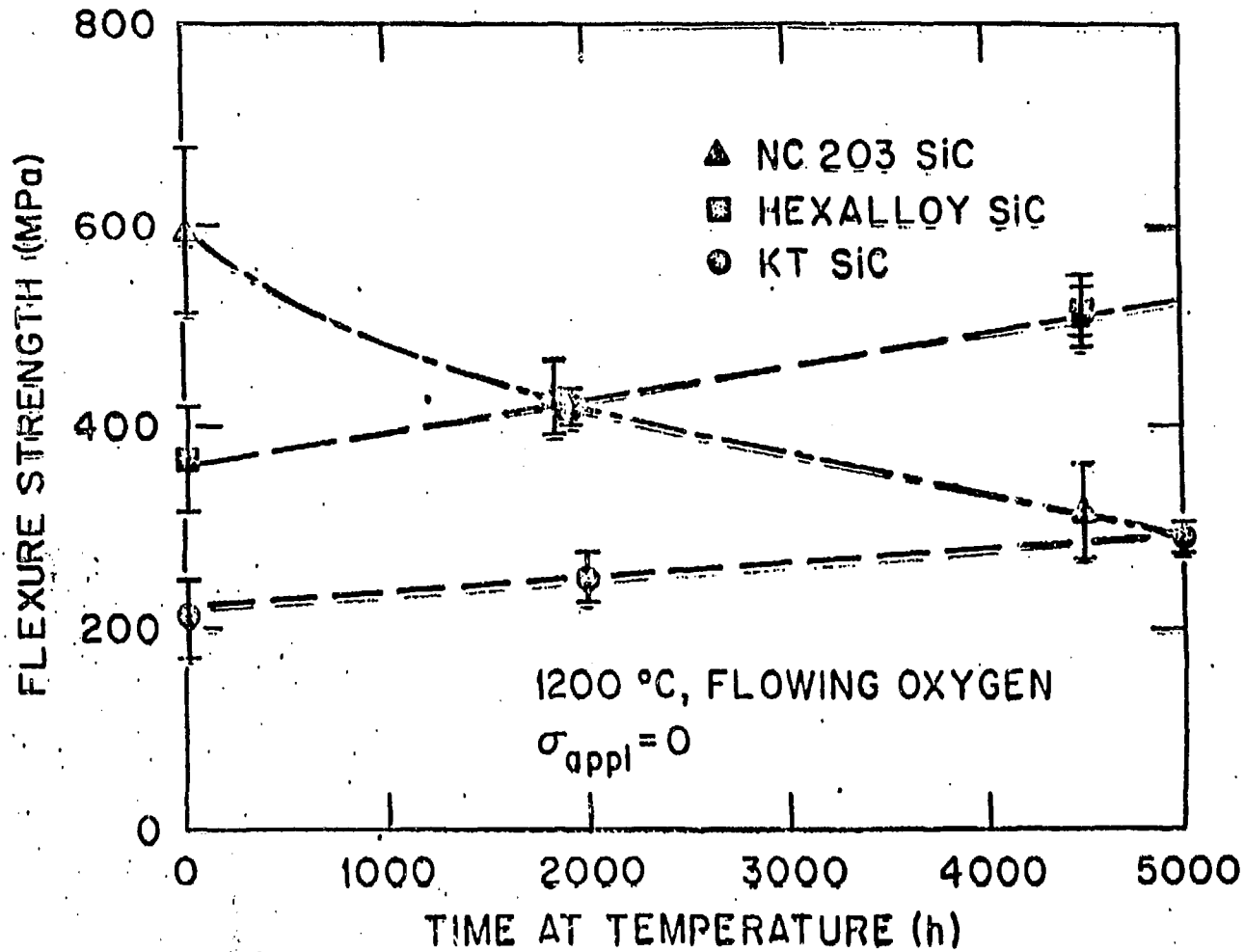
NT 90AB Gave 6 μm $\sigma_f = 720 MPa$ $\times 1000$
 N1 - 1.2 wt % E = 564 GPa
 O2 - 1.1 wt % $\nu = 0.115$

X-RAY ENERGY DISPERSIVE SPECTROSCOPY (EDS) OF A NUMBER
OF BOUNDARIES NEAR TRIPLE POINTS REVEALS THE PRESENCE
OF Ni AND Fe



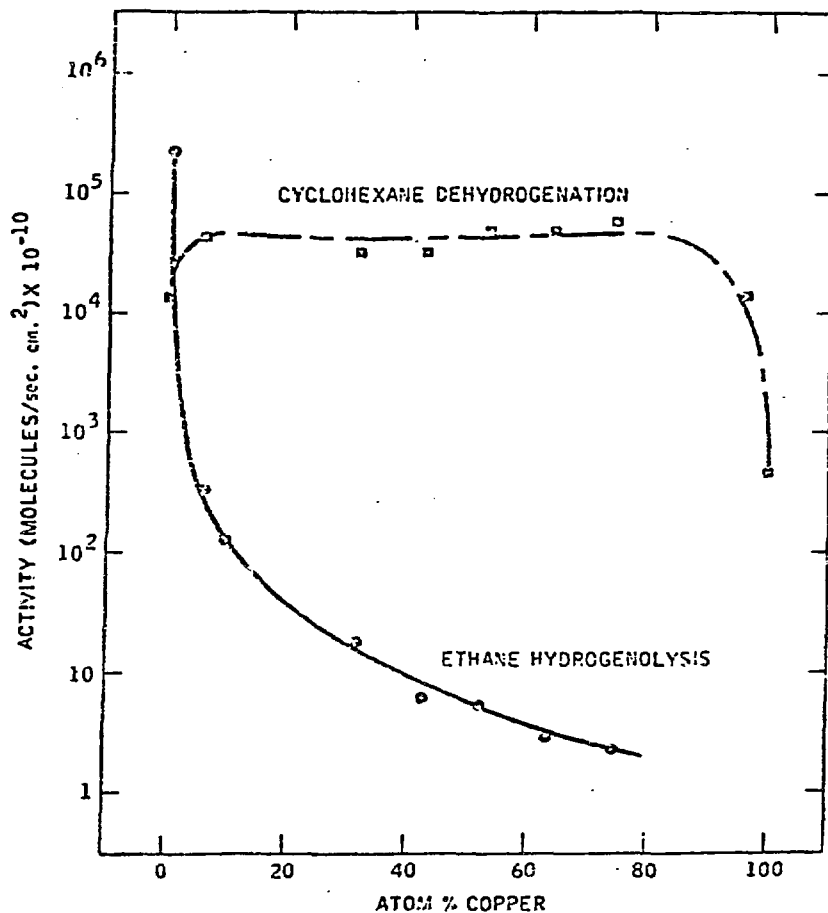
Quantitative analyses indicate that there is ~ 8 at % Ni
and ~ 1.5 at % Fe in the analyzed regions at boundaries

STRENGTH RETAINED OF SiC AFTER OXIDATION AT 1200°C IN FLOWING OXYGEN DEPENDS UPON INITIAL COMPOSITION

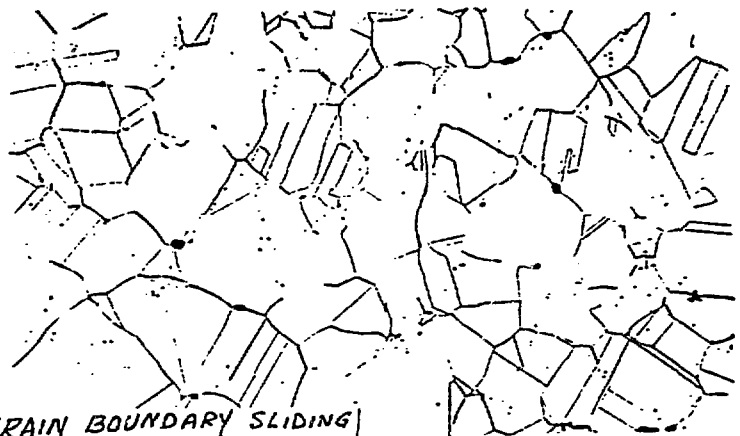


"COMPLEXITY OF CATALYTIC PHENOMENA AND OUR LIMITATIONS IN OBTAINING STRUCTURAL AND ELECTRONIC INFORMATION AT A SUFFICIENTLY MICROSCOPIC LEVEL HAVE IMPEDED OUR SCIENTIFIC UNDERSTANDING."

John H. Sinfelt: Structure of metal catalysts



CREEP RUPTURE OF Ni+20%Cr



GRAIN BOUNDARY SLIDING
PRODUCES VOIDS

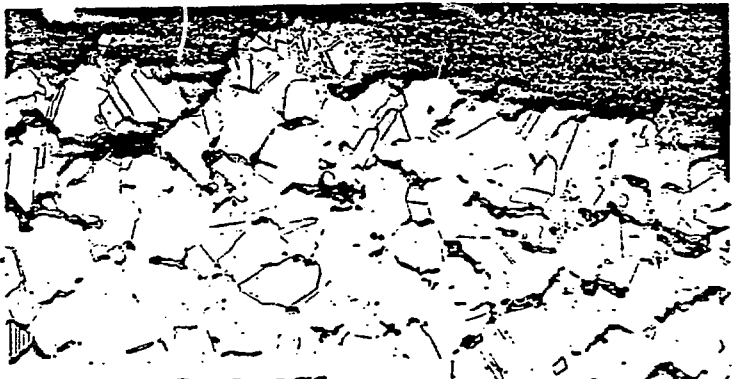
200X



GRAIN BOUNDARY
CRACKS FORM

100 μ m

200X $\text{Ni} 20\text{Cr}$, 12.7%
SMP, 163KS

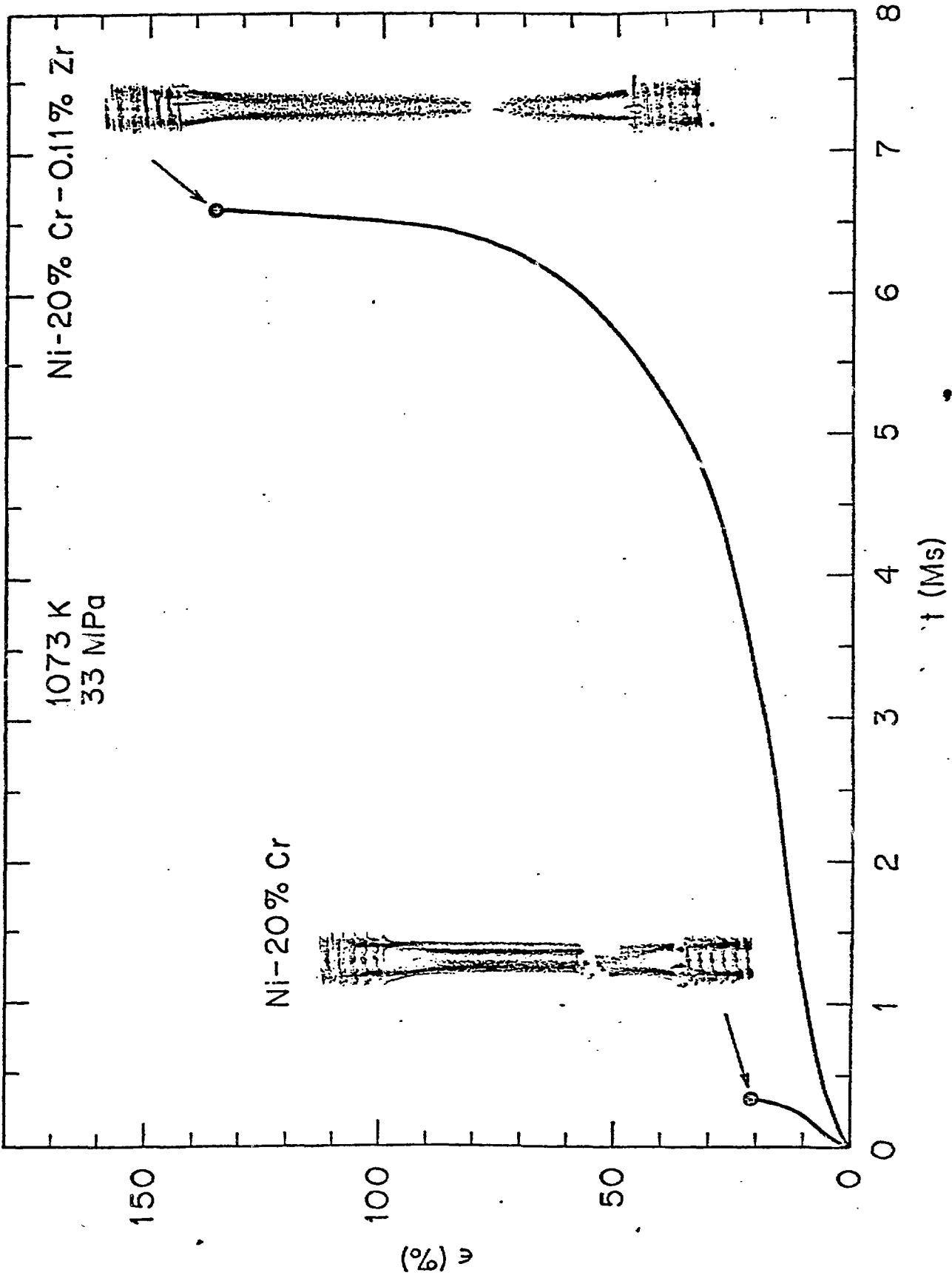


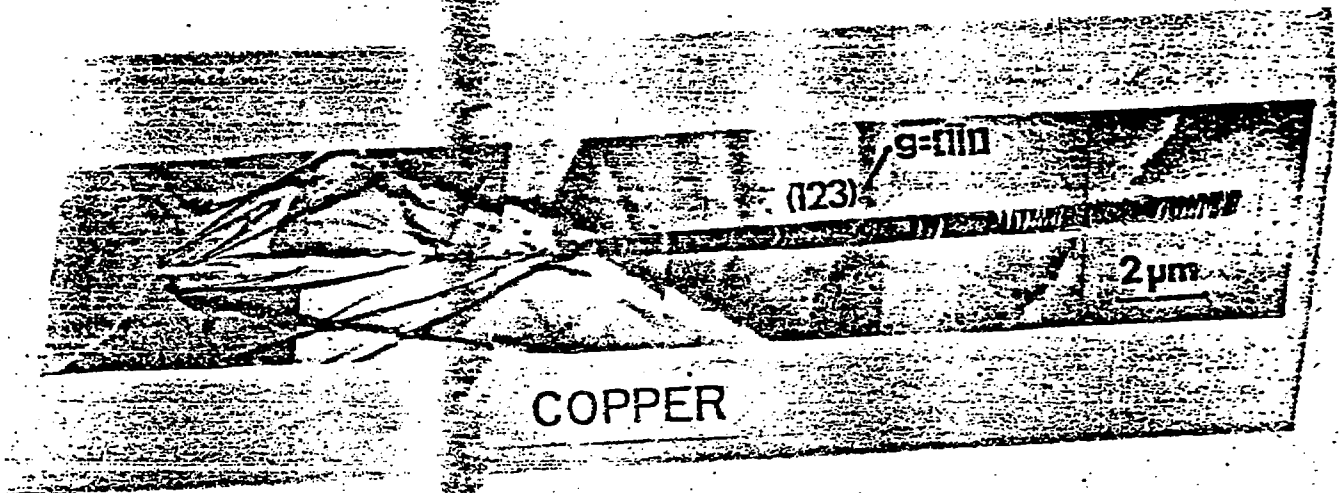
SPECIMEN RUPTURES
BY GRAIN BOUNDARY FAILURE

Ni-20% Cr - 0.11% Zr

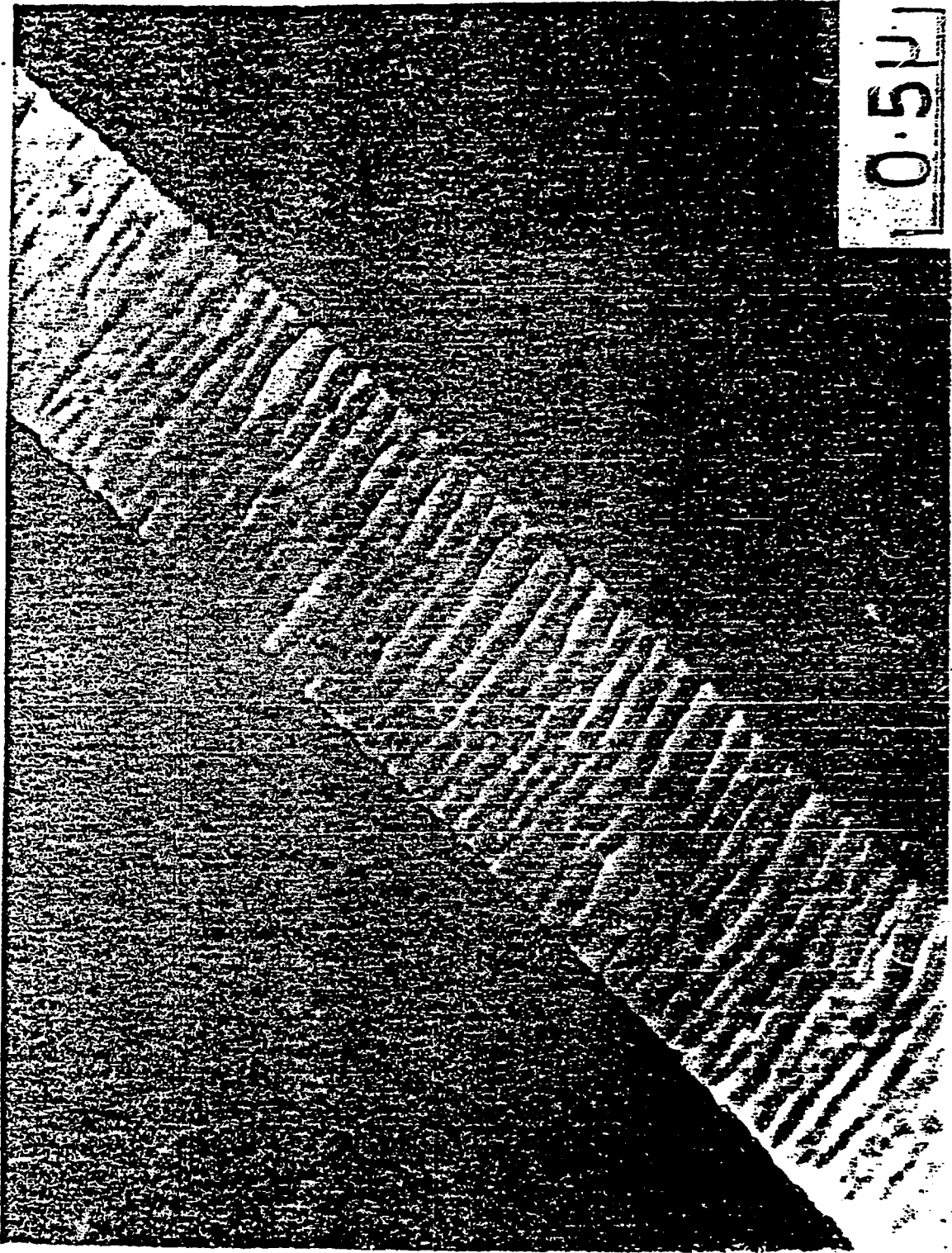
1073 K
33 MPa

Ni-20% Cr

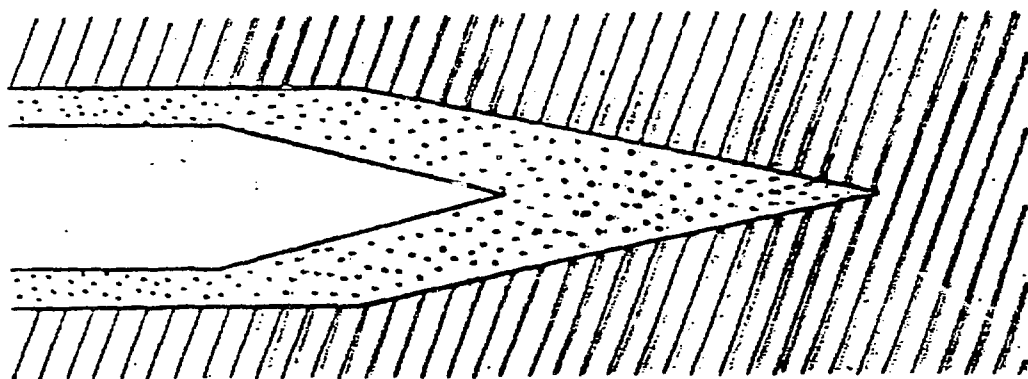




COPPER

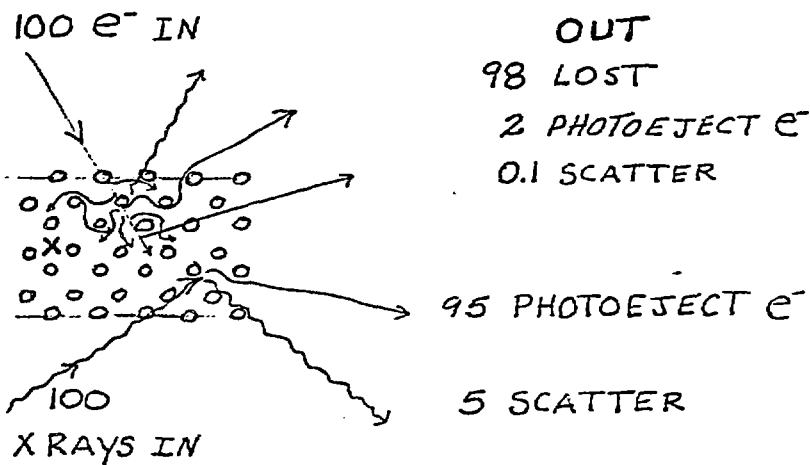


Transmission electron micrograph of a craze.



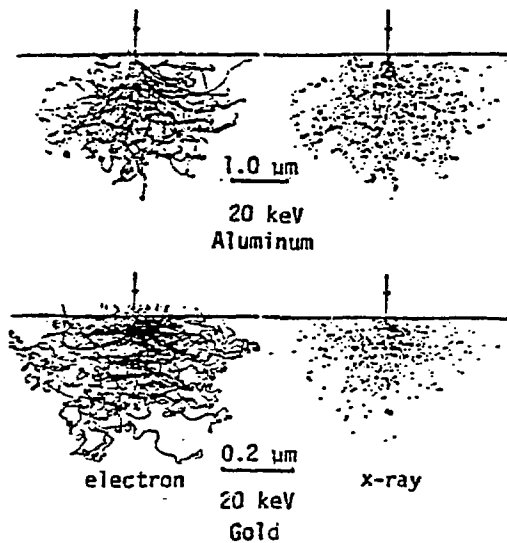
Variation of the microstructure along the length of a craze.
(a) transmission electron-micrograph of a fractured craze;
(b) proposed model for craze.

X RAYS ARE MOST DESIRED PROBE FOR MATERIALS



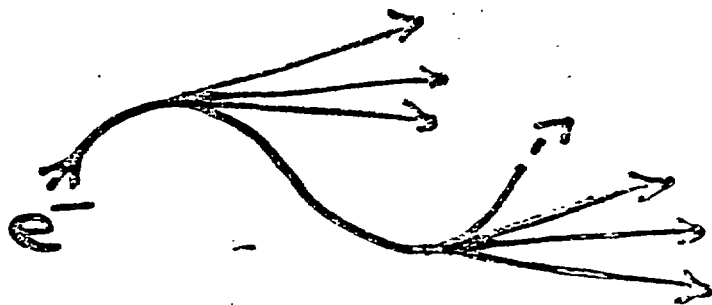
ORNL

INTERACTION VOLUME OF 20 KEV ELECTRONS RANGES
FROM 2 μm IN ALUMINUM TO 0.4 μm IN GOLD FOR
THICK SAMPLES

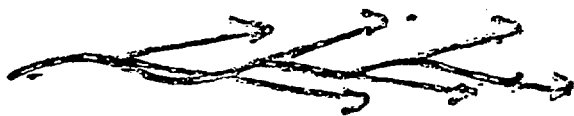


FROM CURGENVER AND DUNCUMB
TIRL REPORT 303, ESSEX, ENGLAND, 1971

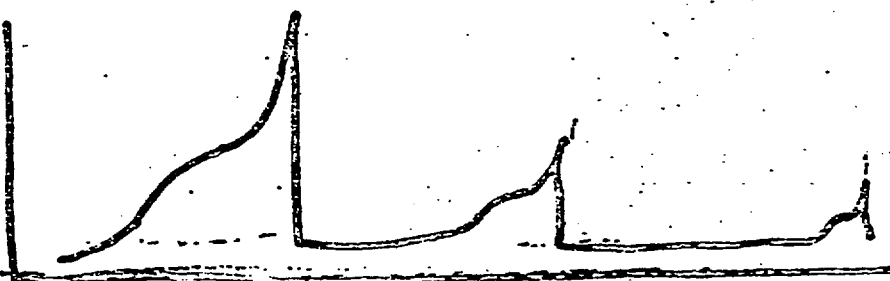
WIGGLERS SUPERIMPOSE THE RADIATION
FROM SEVERAL ARCS.



UNDULATORS PROVIDE VERY BRIGHT
QUASI-MONOCROMATIC RAD.



INTENSITY



$\leftarrow h\nu \rightarrow$

-SOMETIMES UNUSABLE FLUX IS WORSE
THAN NO FLUX-

HERMAN



Herman

2-4

© 1934 Universal Press Syndicate

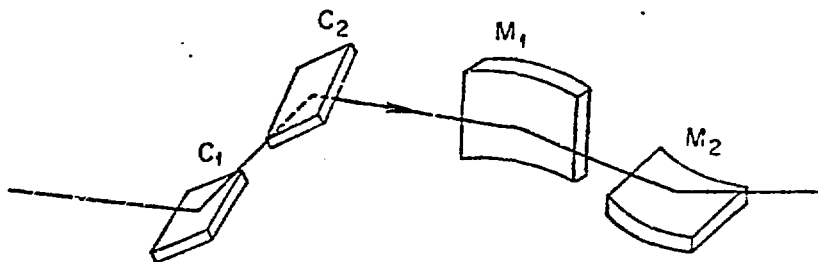
“Up and left.”

DEMAGNIFY THE SOURCE SIZE WHILE MAINTAINING THE FLUX

- SOURCE SIZE $2\sigma_x = 0.8 \mu\text{M}$, $2\sigma_y = 0.2 \mu\text{M}$
- UNDULATOR DIVERGENCE $2\sigma_x' = 0.01 \text{ MRAD}$, $2\sigma_y' = 0.01 \text{ MRAD}$

I. FRESNEL ZONE PLATES: TOO TRANSPARENT FOR HARD X RAYS,
CHOICE BELOW 1 KEV

	DEMAGNIFICATION	ACCEPTANCE AT 12 KEV		PROBE SIZE $\mu\text{M} \times \mu\text{M}$
		CYLINDRICAL MRAD	ELLIPTICAL MRAD	
II. MIRRORS	100:1	<u>0.013</u>	0.034	8×2
	1000:1	0.0013	0.0034	0.8×0.2
III. MULTILAYERS	100:1	<u>0.026</u>	0.136	8×2
	1000:1	0.0026	<u>0.0136</u>	0.8×0.2
IV. CRYSTALS	100:1	<u>0.058</u>	0.68	8×2
	1000:1	0.0058	<u>0.068</u>	0.8×0.2
V. OPTICS				



NONDISPERSIVE MULTILAYERS OR CRYSTALS FOLLOWED
BY CROSSED ELLIPTICAL OR CYLINDRICAL MIRRORS

GAIN IN X-RAY MICROPROBE PERFORMANCE WITH 6 GeV RING

	14 keV X-rays 2.5 GeV, 500 mA Wiggler	14 keV X-rays 6 GeV, 200 mA Undulator	100 keV e ⁻ Electron microprobe field emission
BRIGHTNESS P or e ⁻ /s μm ² mrad	10 ¹⁶	2 × 10 ¹⁹	3 × 10 ¹⁹
INTENSITY P or e ⁻ /Area s	$\frac{5 \times 10^{12}}{10 \mu\text{m}^2 \ 280 \text{ eV s}}$	$\frac{2 \times 10^{15}}{10 \mu\text{m}^2 \ 10 \text{ eV s}}$	
	$\frac{5 \times 10^{11}}{\mu\text{m}^2 \ 280 \text{ eV s}}$	$\frac{1 \times 10^{14}}{\mu\text{m}^2 \ 10 \text{ eV s}}$	$\frac{6 \times 10^{13}}{\mu\text{m}^2 \ \text{s}}$
	$\frac{1 \times 10^9}{500 \text{ \AA}^2 \ 280 \text{ eV s}}$	$\frac{3 \times 10^{11}}{500 \text{ \AA}^2 \ 10 \text{ eV s}}$	
			$\frac{6 \times 10^9}{30 \text{ \AA}^2 \ \text{s}}$
			$\frac{10^7}{4 \text{ \AA}^2 \ \text{s}}$

200 probes, MDL = 50 ppm/100 sec
 1100 SEM, MDL = 500 ppm/100 sec
 4000 SEM, microscopy only

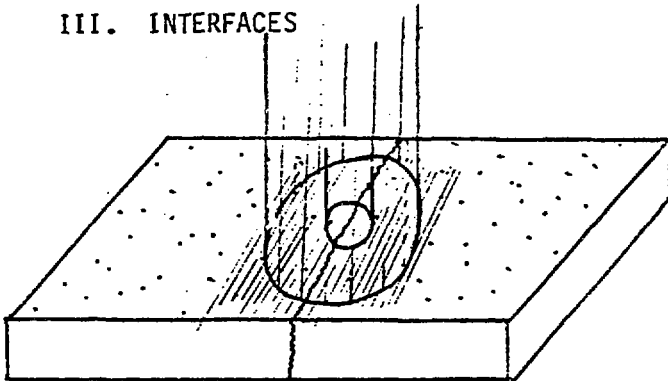
AN X-RAY PROBE CAN ACHIEVE SMALL DETECTABLE LIMITS

I. MINIMUM DETECTABLE LIMIT = $\frac{3.29 (\text{Mass Fraction})(\text{Background Counts})^{1/2}}{\text{Signal Counts}}$

II. $\frac{\text{Signal Counts}}{\text{Sec ppm}} = (\text{Thick-Target Yield}) \left(\frac{\text{Incident Photons}}{\text{Sec } \mu\text{m}^2} \right)$
 $\times (\text{Solid Angle} = 10^{-3}) \times 10^{-6} \text{ Mass Fraction}$

$$\frac{N_s}{\text{Sec ppm}} = 0.1 \times \frac{10^{14}}{\text{Sec } 1 \mu\text{m}^2} \times 10^{-9} = 10^4 \frac{\text{Counts}}{\text{Sec ppm}} \text{ in } 1 \mu\text{m}^2 \text{ Spot}$$

III. INTERFACES



$$1 \text{ Atom Plane} = 2 \text{ \AA}$$

$$\frac{1 \mu\text{m}}{2 \text{ \AA}} = 5000 \text{ Atomic Planes}$$

$$\frac{1 \text{ Monolayer}}{5000 \text{ Planes}} = 200 \text{ Parts Per Million}$$

WITH ALL THE IMPURITY IN THE INTERFACE

$$\text{MDL} = 3.29 \times 1 \text{ ppm} \times \left(\frac{1}{10}\right)^{1/2} \times \frac{1}{(10,000)^{1/2}} = 10.5 \text{ ppb/sec}$$

or

$$\text{MDL} = 5 \times 10^{-5} \text{ of a monolayer/sec}$$

IV. IN DIRTY ALLOYS AND CERAMICS WITH 0.1 wt % IMPURITY

A. Matrix Can Contain 1000 ppm of Impurity

IMPURITY DETECTION AT INTERFACES

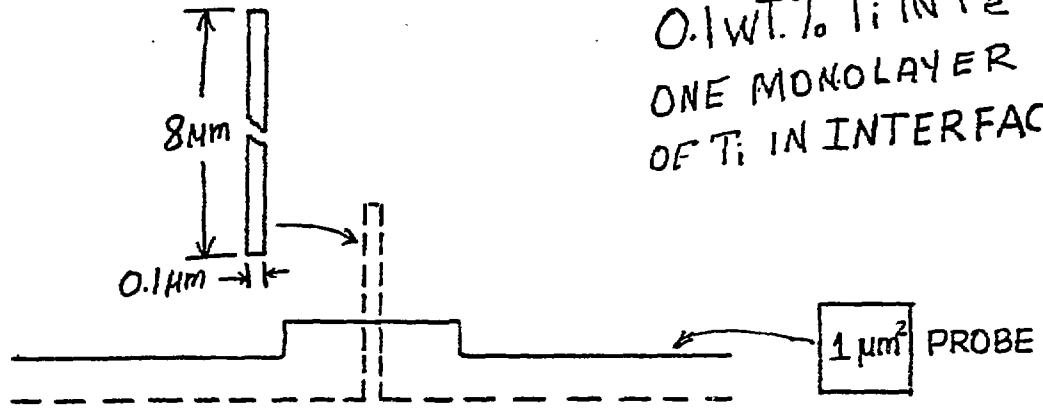
0.1 WT.% Ti IN Fe
ONE MONOLAYER
OF Ti IN INTERFACE

Ti K_{α}
X-RAYS
SEC

10^7

10^6

10^5



500 \AA^2 PROBE

2

1

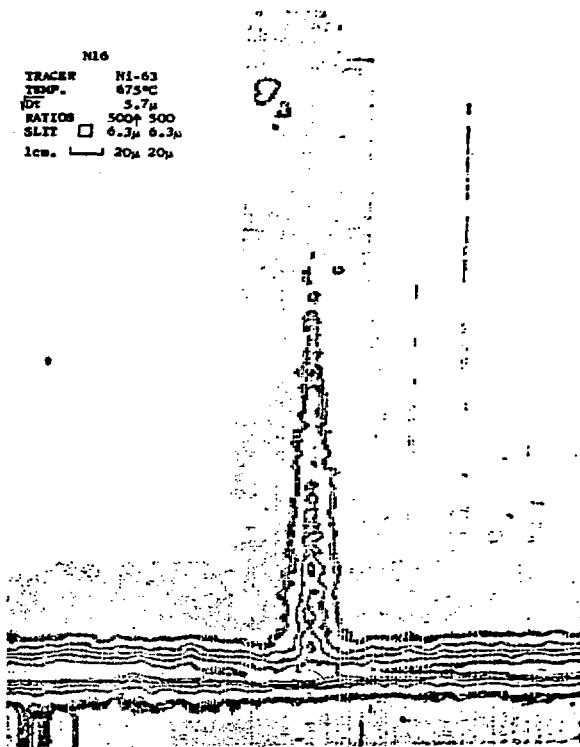
0

1

2

DISTANCE (μm)

AUTORADIOGRAPHIC IMAGE OF Ni-63 TRACER
DIFFUSION INTO COPPER GRAIN BOUNDARY



(a)

After T. J. Renduff

Fig. 2 Top

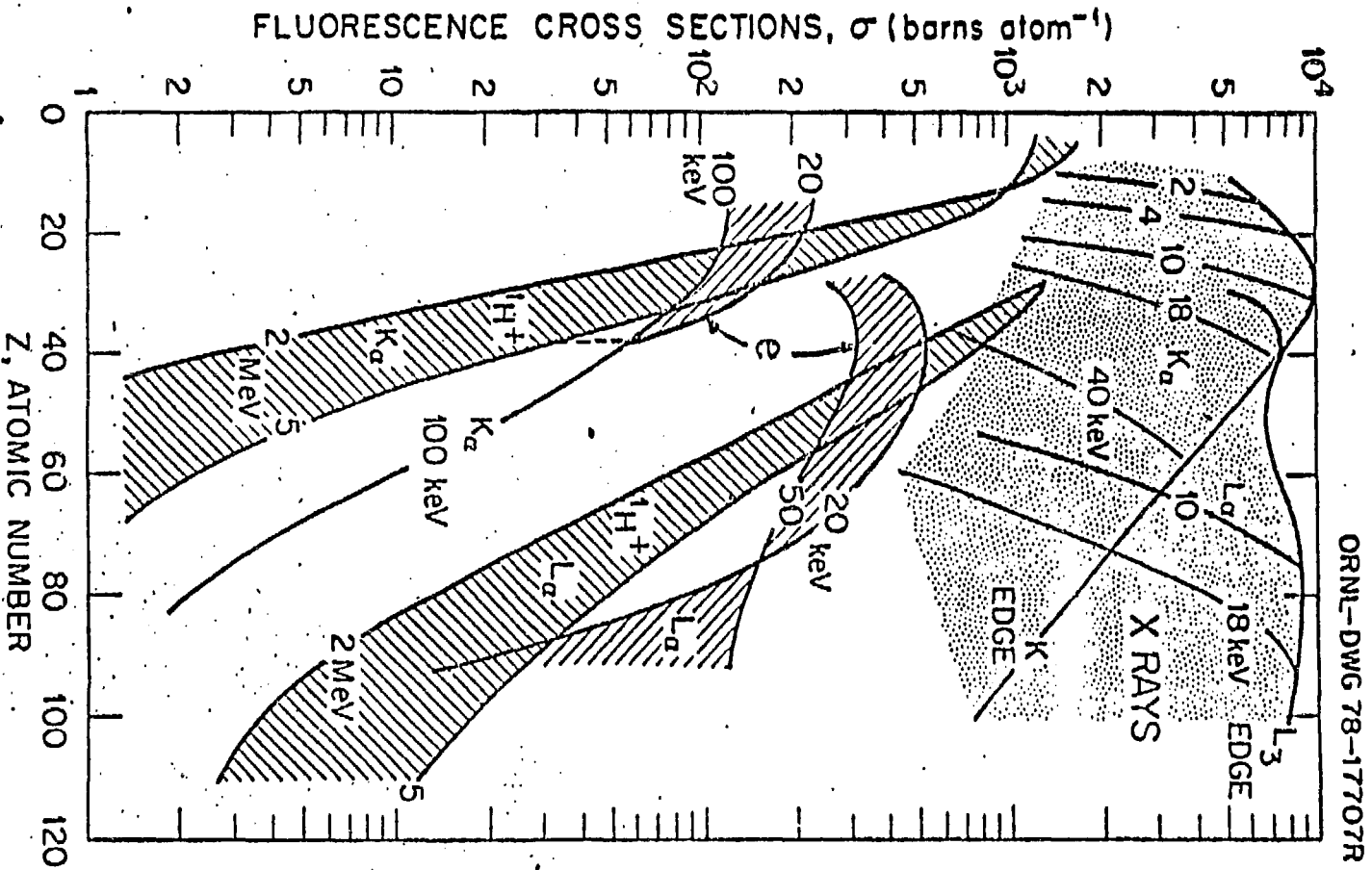
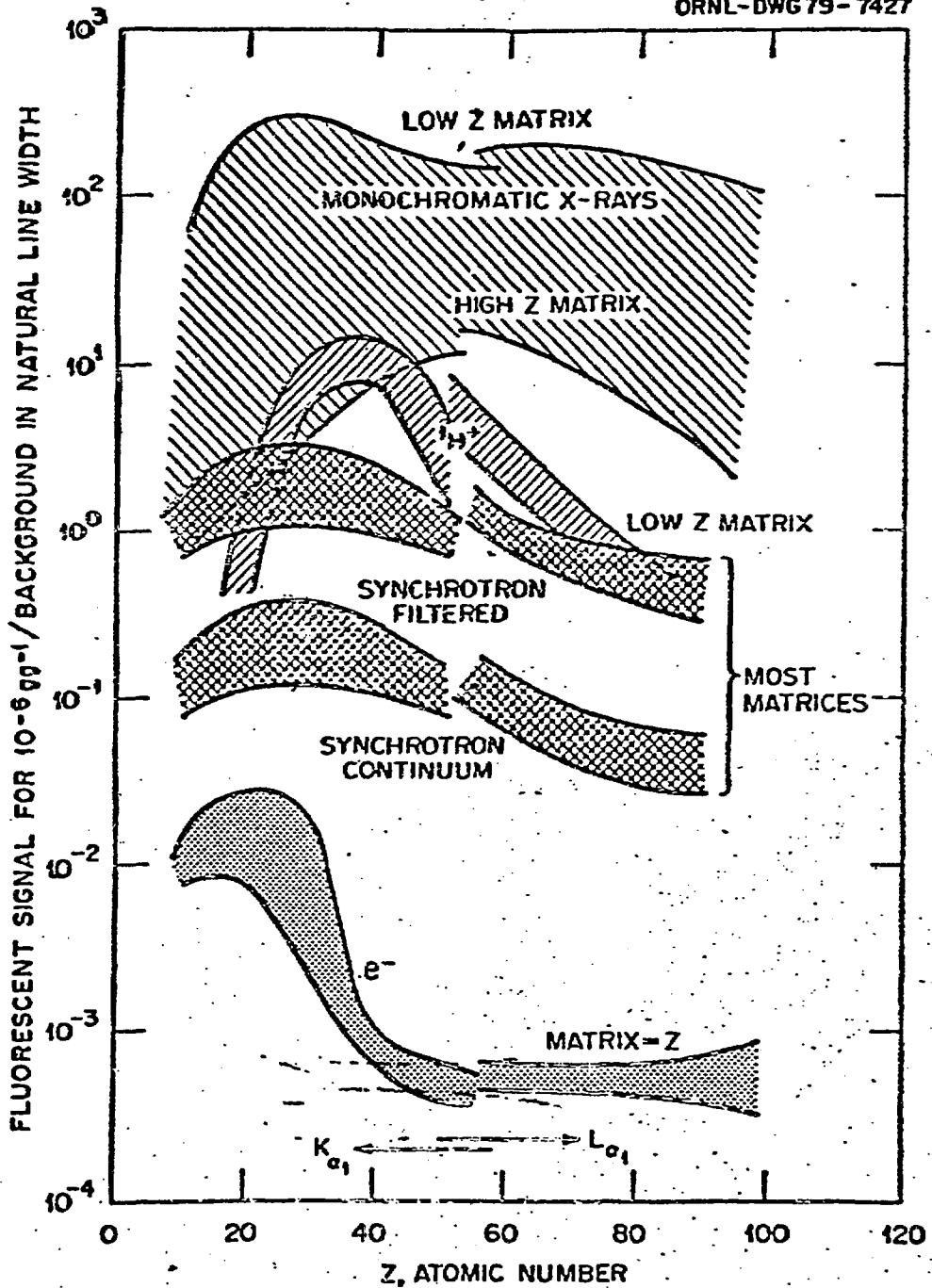
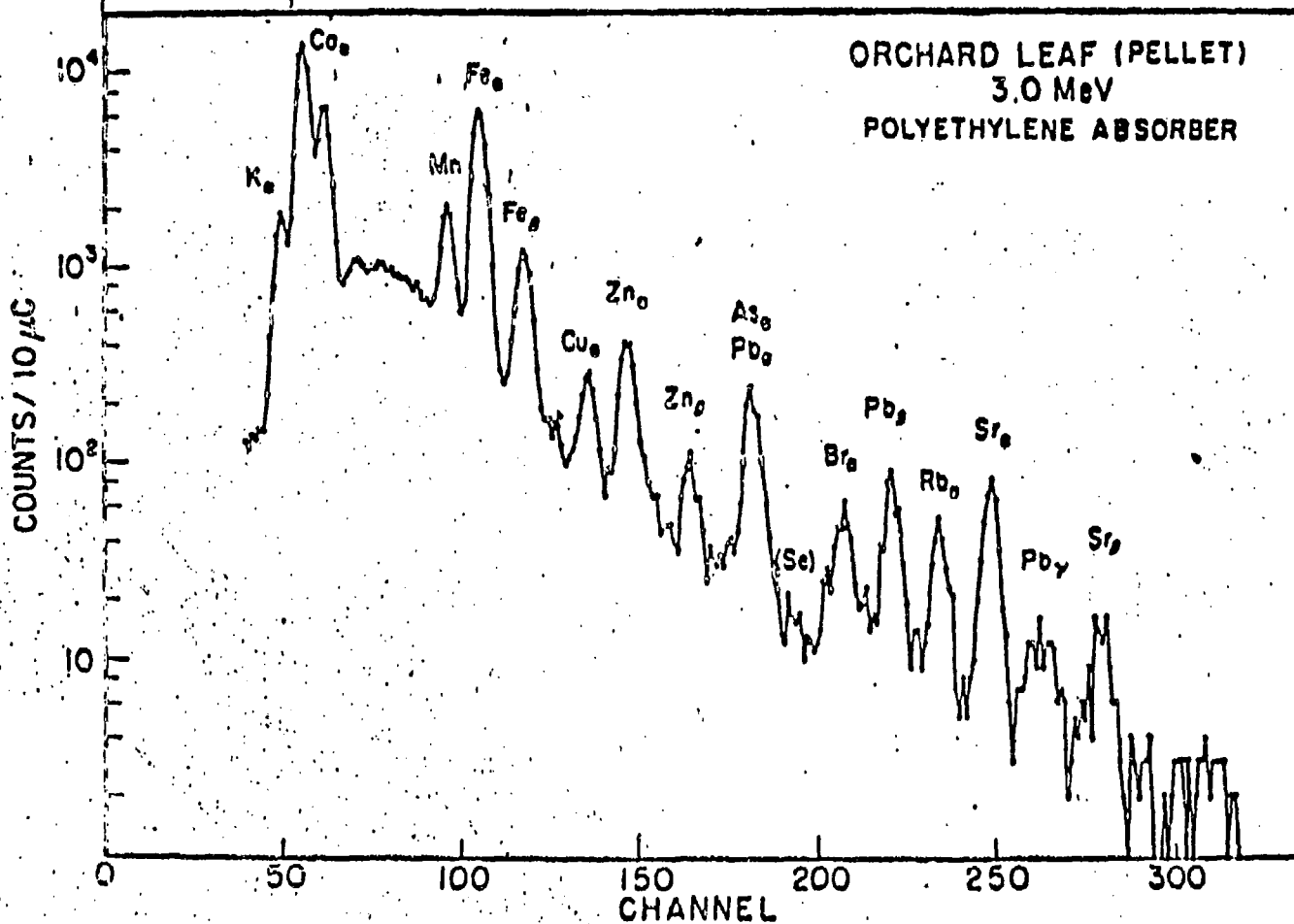


Fig. 23 Top

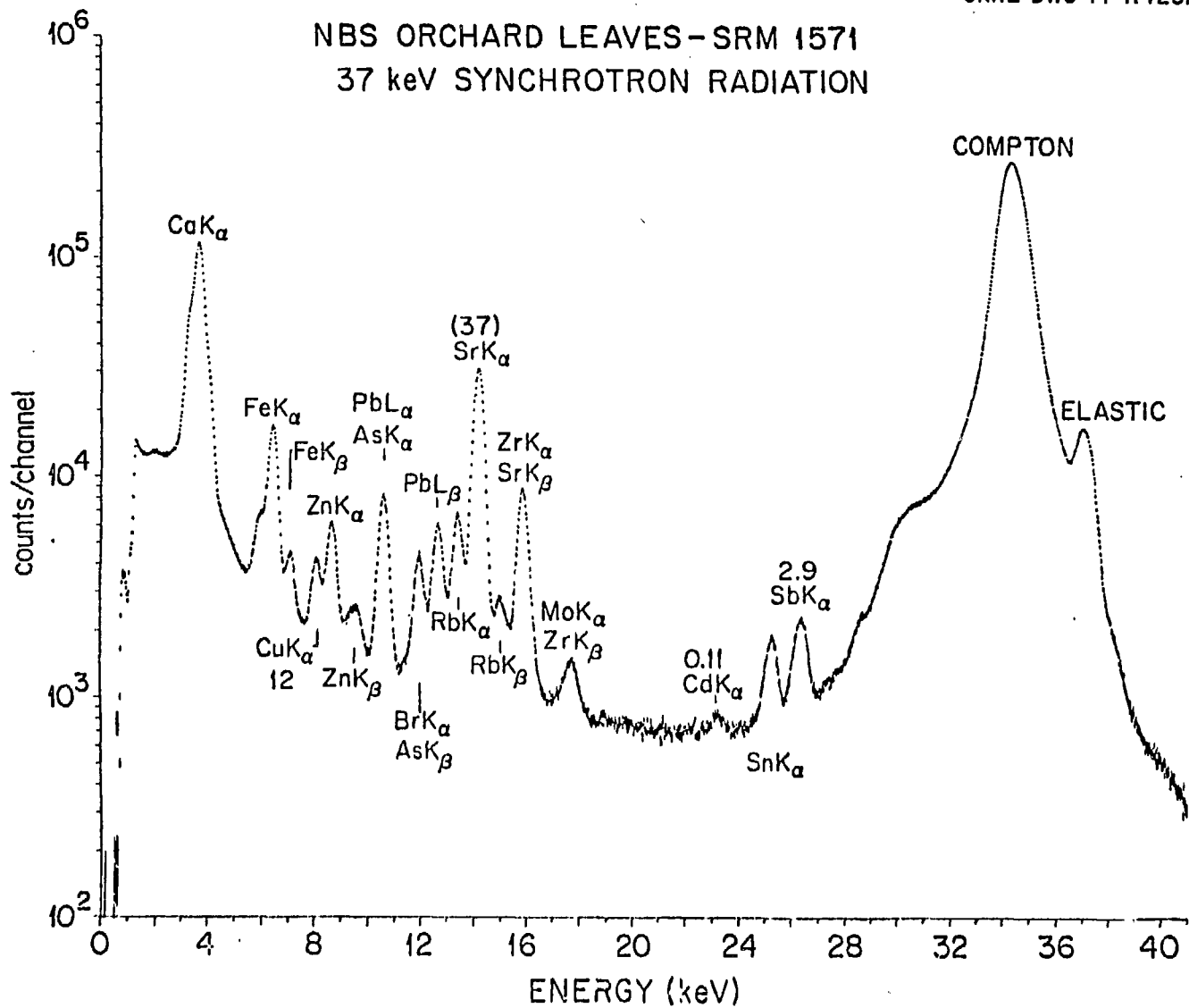
ORNL-DWG79-7427



ORNL-DWG 79-7429



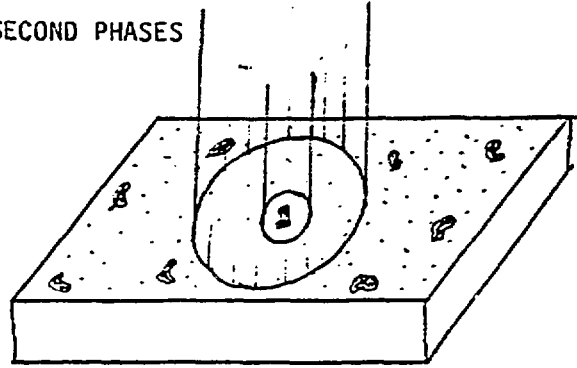
NBS ORCHARD LEAVES - SRM 1571
 37 keV SYNCHROTRON RADIATION



SURFACE IMPURITIES AND SECOND PHASES

I. FLUORESCENT DETECTION

$$N_s \text{ (Signal Counts)} = \frac{I_0 \sigma \text{ (Fluorescent Cross Section)}}{4\pi R^2}$$



$$N_s = \frac{10^{14} \text{ p}}{\text{Sec } \mu\text{m}^2} \times 10^{-3} \times 5 \times 10^{-21} \frac{\text{cm}^2}{\text{Atom}} \times \frac{10^8 \mu\text{m}^2}{\text{cm}^2} = \frac{10^{-2} \text{ p}}{\text{Sec Atom}}$$

$$\frac{\text{Atoms}}{\mu\text{m}^2 \text{ Monolayer}} = \frac{10,000 \text{ \AA} \times 10,000 \text{ \AA}}{2.35 \text{ \AA} \times 2.35 \text{ \AA}} = 1.8 \times 10^7 \text{ Atoms}$$

$$N_s = \frac{10^{-2} \text{ p}}{\text{Sec Atom}} \times 1.8 \times 10^7 \text{ Atoms} = \frac{1.8 \times 10^5 \text{ p}}{\text{Sec Monolayer}}$$

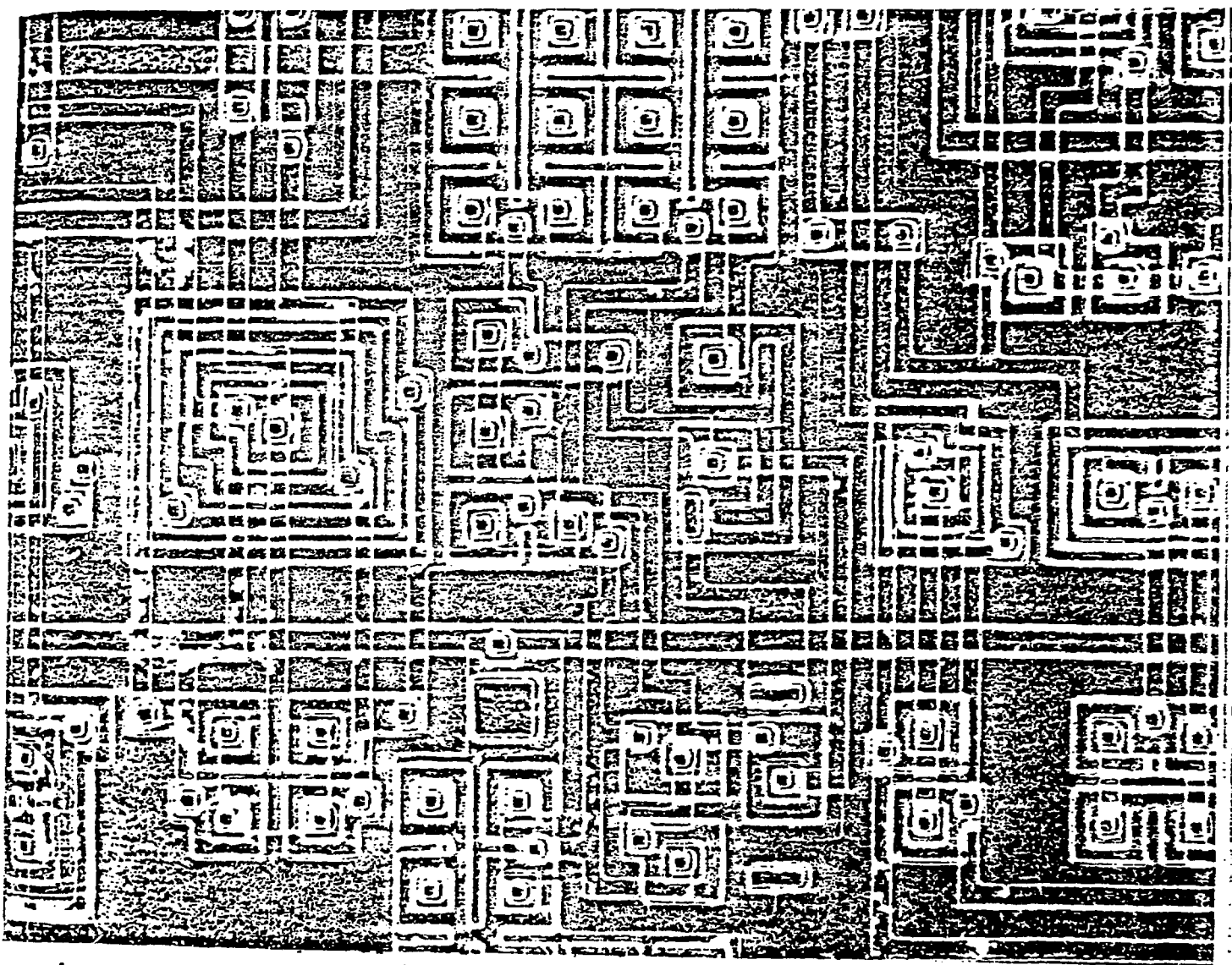
II. MINIMUM DETECTABLE LIMIT WITH μm^2 PROBE, $\tau = 1$ Sec

2.7×10^{-4} Monolayer
 5×10^3 Atoms
 40 \AA -diam Particle

III. MINIMUM DETECTABLE LIMIT WITH 500 \AA^2 PROBE, $\tau = 1$ Sec

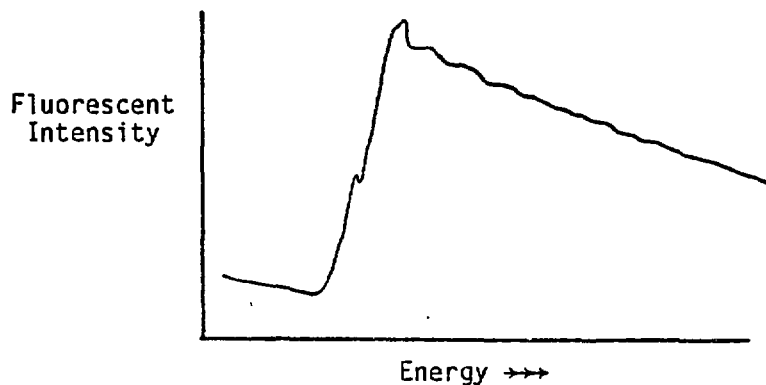
5.4×10^{-3} Monolayer
 250 Atoms
 15 \AA -diam Particle

ELECTRONIC MICROCIRCUIT DEVICE RESEARCH WILL BENEFIT ENORMOUSLY FROM THE HIGH SENSITIVITY OF AN X-RAY MICROPROBE



CHEMICAL INFORMATION: EXAFS AND DIFFRACTION

I. EXAFS BY MEASURING FLUORESCENCE



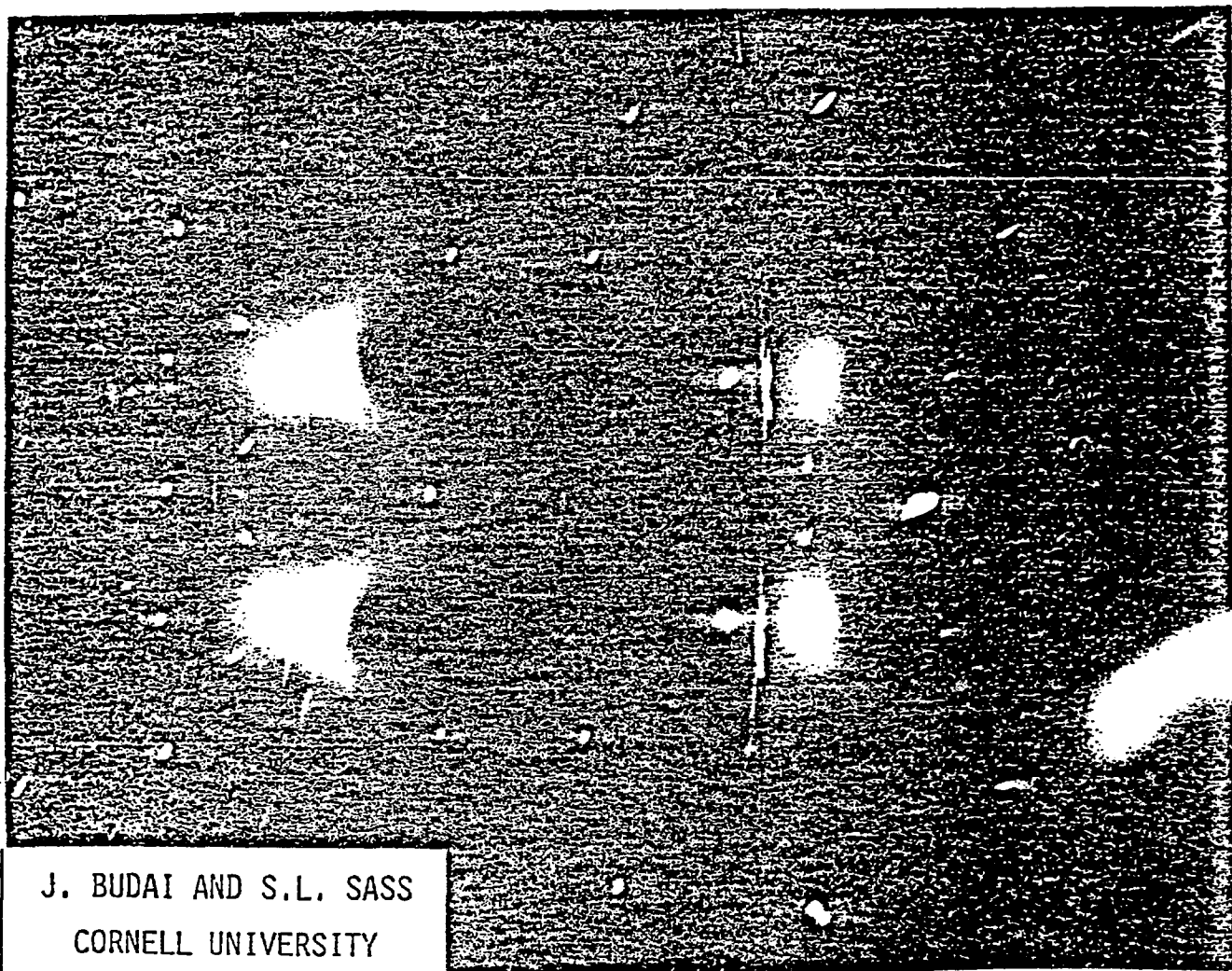
$$\left. \begin{array}{l} \text{Monolayer} \\ 600 \text{ \AA Particle} \\ 84 \text{ ppm} \end{array} \right\} 1.8 \times 10^4 \frac{p}{\text{sec}} \text{ for } \frac{10^{13} p}{\text{sec eV } \mu\text{m}^2} \text{ Incident}$$

II. DIFFRACTION: MINIMUM DETECTABLE LIMIT

CRYSTALLINE	1 μm^2 Probe $\tau = 1 \text{ Sec}$	10 ⁻² of a Monolayer 1.6 $\times 10^3$ Atoms in a Particle 28 \AA -diam Particle
-------------	---	--

AMORPHOUS	1 μm^2 Probe $\tau = 1 \text{ Sec}$	6 Monolayers
	500 \AA^2 Probe $\tau = 1 \text{ Sec}$	1 $\times 10^6$ Atoms 235 \AA -diam Particle

DIFFRACTION PATTERN FROM
Au $\Sigma = 13$ ($\theta = 22.6^\circ$) [001] TWIST BOUNDARY



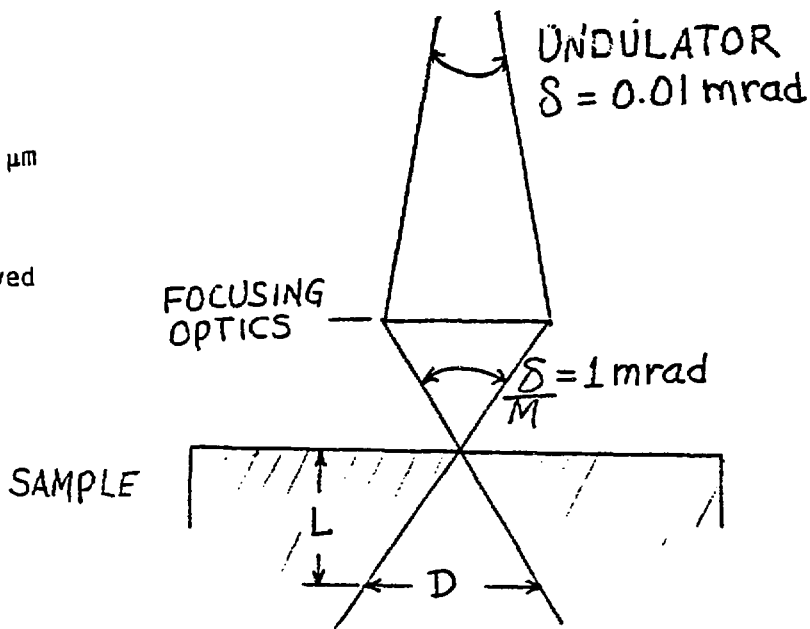
J. BUDAI AND S.L. SASS
CORNELL UNIVERSITY

BEAM PENETRATION AND DIFFRACTION LIMITATIONS FOR X-RAY MICROPROBE

I. BEAM PENETRATION:

$$L = \frac{D}{\delta} = \frac{1 \mu\text{m}}{1 \text{ mrad}} = 10^3 \mu\text{m}$$

With $L < 10^3 \mu\text{m}$, $1 \mu\text{m}$
Probe Size Is Preserved



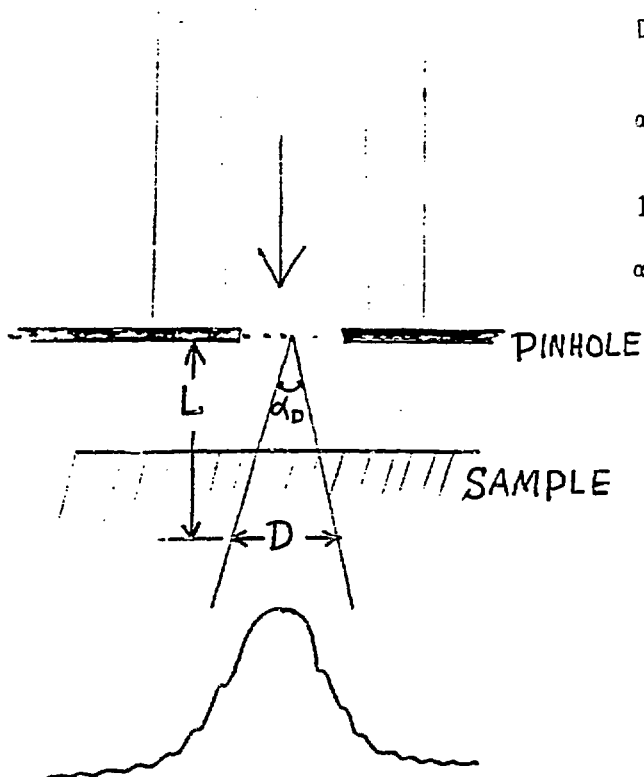
II. PROBE SIZE: Mag = 1/100, $2 \mu\text{m} \times 8 \mu\text{m}$

NEED PINHOLE COLLIMATION FOR $< \mu\text{m}$
DIFFRACTION LIMIT

$$\alpha_D \text{ mrad} = \frac{0.244 \lambda \text{ \AA}}{D_{\mu\text{m}}}$$

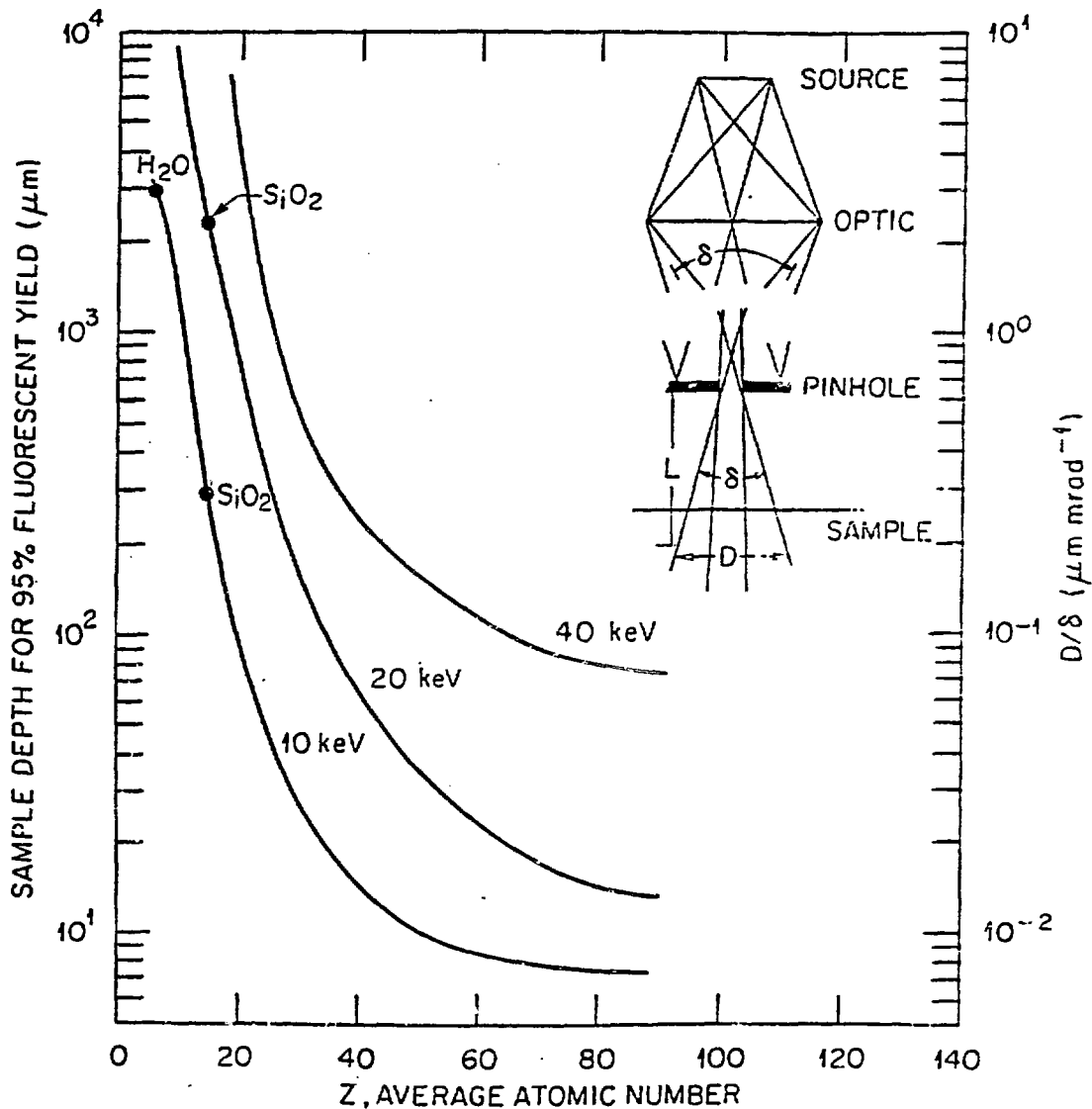
12.4 keV X-Ray and $D = 500 \text{ \AA}$,

$$\alpha_0 = 4.9 \text{ mrad}, \text{ and } L = \frac{D}{\alpha_0} = 10 \mu\text{m}$$



PENETRATION DEPTH OF X RAYS INTO MATERIALS FOR 95% OF THE FLUORESCENT YIELD

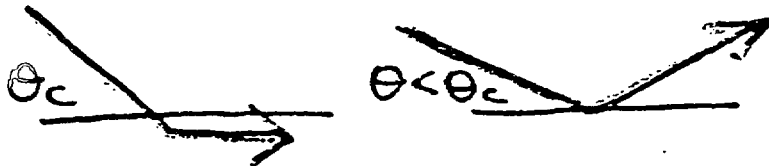
ORNL-DWG 83-16645R



SURFACE PENETRATION OF X-RAYS

CAN BE VARIED BY USING A

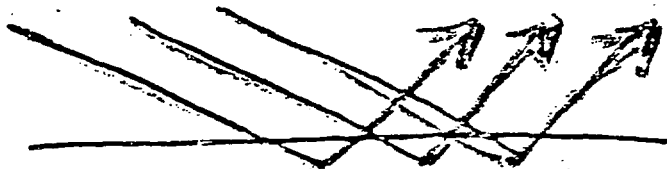
GLANCING ANGLE GEOMETRY.



DEPTH OF PENETRATION DEPENDS ON

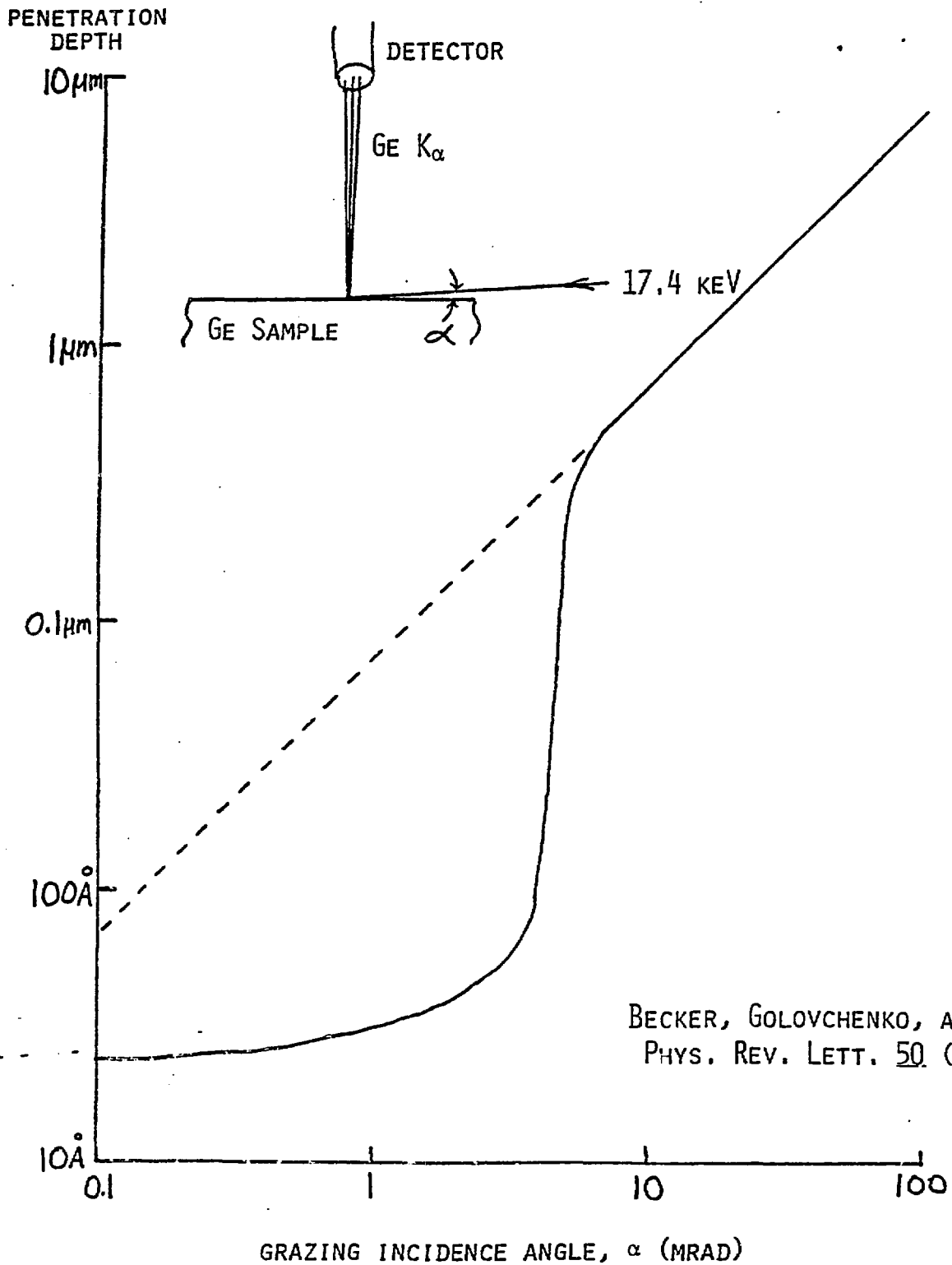
ABSORPTION AND DENSITY OF SAMPLE, θ !

(TYPICAL 50-1000 Å SKIN DEPTH)



$\sim 0-10 \text{ MRAD} = \theta_c$ HENCE SMALL
OPENING ANGLE OF SYNCH. VITAL.

DEPTH PROFILING WITH DIFFRACTION AND FLUORESCENCE



- I. GEOCHEMISTRY HAS SUPPLIED THE MAJOR PUSH FOR INSTRUMENTING AN X-RAY MICROPROBE AT THE NSLS.
 - MEASUREMENTS OF TRACE ELEMENT CONCENTRATIONS AS A FUNCTION OF DISTANCE FROM SOURCE POINTS (SURFACE INTERFACES AND INCLUSIONS) COMBINED WITH CHEMICAL DATA WILL BRING NEW INFORMATION CHARACTERIZING THE TIMES, TEMPERATURES, PRESSURES, AND CHEMICAL ENVIRONMENT FOR RECONSTRUCTION OF THE GEOLOGICAL AGE AND CONDITIONS UNDER WHICH TERRESTRIAL AND EXTRA-TERRESTRIAL MATTER FORMED.

- II. ENVIRONMENTAL SCIENCES AND BIOLOGY HAS INITIATED THE INSTRUMENTATION OF AN X-RAY MICROPROBE AT SSRL AND A PROPOSAL FOR ONE ON PEP.

X-RAY MICROPROBES BASED ON UNDULATORS IN A 6 GeV STORAGE RING WILL PRODUCE NEW INFORMATION: NOT JUST MORE OF THE SAME.

- MATERIALS DEFECTS AND IMPURITIES: MECHANICAL, CHEMICAL, AND ELECTRICAL PROPERTIES.
- CHEMISTRY, PHYSICS, MEDICINE, GEOCHEMISTRY, ENVIRONMENTAL: SYNERGISTIC EFFECT.
- $1 \mu\text{M} \rightarrow 500 \text{ \AA}$ RESOLUTION IN THICK SAMPLES.
- PPB AND 10^{-5} MONOLAYER SENSITIVITY.
- OPERATES IN AIR, WATER: IN VIVO.
- 10^{-6} THE ENERGY DEPOSITED BY CHARGED PARTICLES FOR SAME MDL.