10/12-93 220

CUNF-921131--5

SLAC-RB--6033 SLA Jant DE93 007738

HIGH POLARIZATION PROTOCATHODE R&D AT SLAC

Texashi MARUYAMA and Edward L. GARWIN. Stanford Linear Accelerator Center, Stanford University Stanford, California 94309, USA

Richard PREPOST and Georgie H ZAPALAC. Department of Physics, University of Wisconsin Madison, Wisconsin 53706, USA

# ABSTRACT

This paper describes recent progress on the development of high polarization photocathodes for polarized electron sources A strained inGEAs esthede has achieved a maximum electron-spin polarization of 71% and has demonstrated the strain enhancement of polarization for the first time Strained GEAs cathodes have yielded polarizations as high as 90% with much higher quantum efficiency

## 1. Introduction

Polarized electron sources based on negative-electron-affinity GaAs were pioneered at  $SLAC^{-1}$  The original PEGCY II source produced  $2x10^{11}$  e- per bunch at 120 Hz with an average spin polarization of 37%. The maximum polarization of this type of source were immited to 50% due to the valence-band degeneracy of the heavy-hole and light-hole bands of GaAs Much effort has been devoted to achieving higher polarization by primarily focusing on the following three types of pholocathodes 1) ternary chalcopyrites which naturally have the appropriate band structure, 2) GaAs-AfGaAs superintices and 3) strained materials These materials do not have a valence band degeneracy and selective excitation of a single transition is possible Eacher efforts at SLAC were devoted to studies of the (inst two types of pholocathodes<sup>2</sup>

When a thin layer of a crystal A is grown on a different crystal B with a slightly <u>smaller</u> lattice constant crystal A may grow pseudomorphically, introducing a biastal compressive strain in the plane of the interface. The strain alters the band structure of crystal A such that the strain dependent energy difference of the heavy-hole and light-hole bands relative to the conduction band is given by

 $\begin{array}{l} E_0^{CHH} \bullet E_0 \bullet \delta E_H \cdot \delta E_S \\ E_0^{CLH} \bullet E_0 \bullet \delta E_H \cdot \delta E_S \cdot (\delta E_S)^2 / 2 \Delta_0 \end{array} \end{array}$ 

Presented at the 10th International Symposium on High Energy Spin Physics Napova Japan November 9 14 1992 where  $E_0$  is the direct band gap of fully relaxed crystal A and  $D_0$  is the spin orbit splitting. The quantities  $dE_X$  and  $dE_S$  represent the hydrostatic shift of the center of gravity of the  $P_{3/2}$  multiplet and the linear splitting of the  $P_{3/2}$  multiplet respectively, and are given in terms of the biazzal strain e parallel to the interface by

 $\delta E_{H^{-}} 2 \in \{(C_{11} - C_{12})/C_{11}\} e$  and  $\delta E_{s^{-}} = b \mid (C_{11} + 2C_{12})/C_{11}\} e$ 

where the parameters a and b are the interband hydrostatic pressure and uniazial deformation potentials respectively, and the  $C_{11}$  are the elasticstiffness constants. Since the biatial strain e is compressive, the effect of the strain is to increase the band gap energy and remove the degeneracy of the heavy-hole and light-hole levels such that the heavy-hole band moves up in energy and the light-hole band moves down relative to the unstrained case This energy splitting makes it possible to preferentially excite the heavy-hole band by tuning the excitation photon energy An electron spin polarization of higher than 50% is then in principle possible for photoexcited electrons the heavy-hole band to conduction band transition would result in 160% electron spin polarization. On the other hand if an epitagial layer were grown on a crystal with a larger lattice constant a tensile strain would develop in the plane of the interface lifting the light-hole band higher in energy than the heavyhole band Selective excitation of the light hole band results in a polarization of -100% opposite in sign from the 50% polarization obtained when the heavy-hole and light-hole bands are degenerate.

When a latitice-mismatched layer is grown on a substrate the misfit between the layers is accomodated by the elastic strain in the epitaxial layer However, as the epitaxial layer enceeds a critical thickness misfit dislocations are generated which begin to relieve the strain Although the idea of using the strained photocathodes for increasing polarization has been known since the late 1970s it was generally considered impractical for the polarized electron source application because the critical thickness is typically 100Å <sup>3</sup> However recent experimental studies on latticemismatched layers have revealed that most of the lattice strain is preserved to thicknesses much beyond the critical thickness and significant strain relief can be observed only above a second critical thickness of about 1000Å <sup>4</sup>

# 2 Strained InGaAs grown on Galss

The electron spin polarization was measured for two samples with a InGEAS layer epitazially grown on a GaAs substrate with InGEAS thicknesses of 0.1  $\mu$ m and 1.14  $\mu$ m. The thin sample is expected to be highly strained while the thick sample is celated Figures 1.127 and (b) show the measured electron spin polarization as a function of electron photon wavelength for the 0.1  $\mu$ m thick and the 1.34  $\mu$ m thick samples respectively. In the wavelength region longes than 880  $\mu$ m the

maint

Work supported in part by Department of Energy Contracts DE AC03-765F00515 (SLAC) and DE AC02-76ER00861 (UW)

photoemission from GeAs diminishes sharply since the excitation photon energy is smaller than the GaAs band gap, and the major contribution to the photoemission can come only from the inGaAs layer. The spin polarizations of the two samples show a significant difference. The polarization of the D 10  $\mu$ m-thick sample is observed to increase sharply at about 925 nm reaching 71% at about 980 nm. The sharp enhancement at about 925 nm corresponds to the expected gap energy between the lighthole band and the conduction band for the sample. On the other hand the polarization of the 1.14  $\mu$ m-thick sample remains at 46% and does not show any enhancement.

# ). Strained GaAs grown on InGaAs

It is well known that the electron-spin polarization increases when the photoemission quantum efficiency decreases in particular polarizations higher than 50% have been observed for positive-electronaffinity surfaces it is important to demonstrate that the polarization enhancement observed for the thin inGaAs sample is indeed due to the crystal strain and not due to a positive-electron-affinity surface. As mentioned in the introduction a tensile strain in the plane of the interface should produce a spin polarization opposite to a compressive strain. To test this prediction a third sample was grown with a 0 1  $\mu$ mthick GaAs isystrate)<sup>6</sup>. As shown in Fig. 1(c) the polarization decreased from +50% changed its sign and reached 66%. This behavior is totally consistent with the expectation from the strain induced effect.



Figure 1 Electron-spin polarization as a function of excitation photon wavelength for a) 0.1  $\mu$ m InGaAs b) 1.14  $\mu$ m InGaAs and c) 0.1  $\mu$ m GaAs

4 Stained GaAs grown on GaAsp<sup>7</sup>

A systematic study of strain efforts on electron polarization has been made using samples with GAAs epitatistiv grown on a GaAsi  $_{\rm T}P_{\rm T}$  buffer layer Samples with varying spitagial layer thickness and varying

buffer layer phosphorus concentration were used to cover a range of strains, and the electron spin polarization and quantum efficiency were measured for each sample as a function of excitation photon energy.

Table I summarizes the parameters for the five samples studied the phosphorus fraction (z) the GaAs epitaxial layer thickness the introemismatch, the measured strain as a per cent of full strain and the critical thickness

TABLE J Strained CaAs samples

Sample	1	2	3	4	5
I	0210	0 243	0 279	0 244	0 238
GEAS thickness (µm)	011	015	011	0 20	0 30
Lattice mismatch (3)	076	086	191	0 85	0 86
Measured strain (%)	76	85	87	<b>8</b> 1	61
Critical thickness (Å)	133	110	92	110	113

Figure 2 shows the measured electron spin polarization as a function of excitation photon wavelength for all the samples. The polarization of samples 1, 2 and 3 shows a systematic shift of the locatic, of the polarization enhancement toward shorter wavelength as the phosphorus fraction is increased. This shift is consistent with the expected change in the energy gap  $E_0^{\rm CLH}$  due to increased strain. The maximum polarization at the photon energy corresponding to the gap energy  $E_0^{\rm CHH}$  increases from 32% to 90% as the phosphorus fraction is increased most likely due to a more selective excitation of the heavy-hole band, since the heavy-hole light-hole splitting increases from 50 meV to 67 meV for these samples.

Although the buffer layers for samples 4 and 5 have the same photphorus fraction as sample 2 the polarization of these samples shows a systematically different behavior. The polarization enhancement shifts toward longer wavelength as the GaAs thickness is increased from 0.15  $\mu$ m (sample 2) to 0.20  $\mu$ m (sample 4) and to 0.30  $\mu$ m (sample 5) For sample 5 photoemission is observed beyond the expected band gap energy  $E_0^{C,KH}$ . This is a strong indication that the lattice is partially related and that the heavy and light-hole band energies are merging towards the values expected for related GaAs. However, even though there is considerable relatation of the strain the maximum polarization of both samples reaches more than 50% Since the epitatial layer of sample 5 is about 30 times thicker than the equilibrium trained thickness this high polarization demonstrates a significant persistence of lattice strain

Figure 3 shows the measured quantum efficiency as a function of excitation photon wavelength for all the samples As expected, the quantum efficiency increasing with increasing GRAS epitarial layer thickness However the gain in quantum efficiency in the high polarization region is offset by a commensurate decrease in polarization due to the increased relaxation of the sample strain For the present



samples, the highest quantum efficiency that corresponds to at least \$0% polarization is 0.13%, measured for sample 4

Figure 2 Electron-spin polarization as a function of wavelength for the strained GaAs samples Figure 3 Quantum efficiency as a function of wavelength

#### 5. Conclusions

The effect of strain on electron-spin polarization has been systematically studied with InGaAs and GaAs photocathodes Polarization in excess of 70% was observed for the 0.1 mm-thick strained inGaAs establishing the first observation of strain enhancement of electron-spin polarization for photoemitted electrons Polarization as high as 90% was observed for strained GaAs and the highest quantum efficiency corresponding to at least 80% polarization was 0.13%

## 6. References

- 1 C Y Prescott et al Phys Lett 77B 347 (1978) 84B 524 (1979)
- 2 C K Sinclair, Proc 6th Int Symp on High Energy Spin Phys Marseille 1984 Journal de Physique Colloque C2 suppl 2 p 669 (1985), and Proc 8th Int Symp on High Energy Spin Phys Munnespolis 1988 AIP Conf Proc 187 p 1412 (1989)
- 3 D T Pierce and R J Celotta in Optical Orientelion edited by F Meter and B P Zakharchenya (North-Bolland Amsterdam 1986) p 259
- 4 P J Orders and B F Usher Appl Phys Lett 50 980 (1987)
- 5 T Maruyama B L Garwin R Prepost G H 2spaine ] 5 Smith and ] D Walker, Phys Rev Lett 66 2376 (1991)
- 6 To prevent photoemission from the InGaAs layer a 200Å-thick AGaAs was added between InGaAs and GaAs layers
- 7 T Maruyama E L Garwin R Preposi and G H Zapalac Phys. Rev B45 4261 (1992)

# DISCLAIMER

This report was prepared as an account of work sponsored by an agency of the Uosted States Government Neither the Uosted States Government nor any agency thereof, nor any of their employees, makes any warranty, express or implied, or assumes any legal hability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product process, or service by trade name, trademark, manufacturer, or otherwise does not necessarily constituit or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof. The views and opicions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof.



# DATE FILMED 3/10/93

