

Communication

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Quantum Enhancement of Charge Density Wave in NbS₂ in the 2D Limit

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Abstract

At ambient pressure, bulk 2H-NbS₂ displays no charge density wave instability at

odds with the isostructural and isoelectronic compounds 2H-NbSe₂, 2H-TaS₂ and 2H-TaSe₂, and in disagreement with harmonic calculations. Contradictory experimental results have been reported in supported single layers, as 1H-NbS₂ on Au(111) does not display a charge density wave, while 1H-NbS₂ on 6H-SiC(0001) endures a 3 × 3 reconstruction. Here, by carrying out quantum anharmonic calculations from first-principles, we evaluate the temperature dependence of phonon spectra in NbS₂ bulk and single layer as a function of pressure/strain. For bulk 2H-NbS₂, we find excellent agreement with inelastic X-ray spectra and demonstrate the removal of charge ordering due to anharmonicity. In the 2D limit, we find an enhanced tendency toward charge density wave order. Freestanding 1H-NbS₂ undergoes a 3 × 3 reconstruction, in agreement with data on 6H-SiC(0001) supported samples. Moreover, as strains smaller than 0.5% in the lattice parameter are enough to completely remove the 3 × 3 superstructure, deposition of 1H-NbS₂ on flexible substrates or a small charge transfer via field-effect could lead to devices with dynamical switching on/off of charge order.

Keywords

Transition metal dichalcogenide, monolayer, charge density wave, anharmonicity, size-dependent properties, phonons

Transition metal dichalcogenides (TMDs) are layered materials with generic formula MX₂, where M is a transition metal (Nb, Ta, Ti, Mo, W, ...) and X a chalcogen (S, Se, Te). The layers, made of triangular lattices of transition metal atoms sandwiched by covalently bonded chalcogens, are held together by weak van der Waals forces, and TMDs can be readily exfoliated into thin flakes down to the single layer limit, with mechanical or chemical techniques.¹⁻⁴ In TMDs, the interplay between strong electron-electron and electron-phonon interactions gives rise to rich phase diagrams, with a wide variety of cooperating/competing collective electronic orderings as charge-density wave (CDW), Mott insulating, and superconductive phases.^{5,6} Of the several polytypes, we focus here on the most common one for

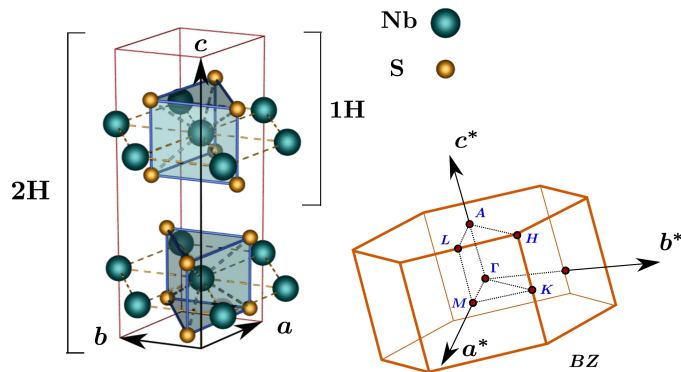


Figure 1: (Color online) Left-hand side: crystal structure of trigonal NbS₂ in the 1H monolayer and in the 2H (bulk) stacking layer configuration. Right-hand side: Corresponding hexagonal Brillouin zone (BZ) with the high-symmetry points (in the monolayer configuration only the points $\Gamma M K$ are relevant, and they are customarily indicated with a line over the letter).

NbS₂, the H polytype,^{7,8} where the transition metal is in trigonal prismatic coordination with the surrounding chalcogens. In Fig. 1 the 1H (monolayer) and 2H (bulk) crystal structures are shown.

Among metallic 2H bulk TMDs, NbS₂ occupies a special place since no CDW has been reported,^{9,10} contrary to its isoelectronic and isostructural 2H-TaSe₂, 2H-TaS₂ and 2H-NbSe₂. All these systems have very similar band structures and are conventional (i.e. phonon-mediated) superconductors with critical temperatures T_c that increases from a sub-Kelvin value in 2H-TaSe₂ and 2H-TaS₂ (around 0.2 K and 0.5 K, respectively) up to 5.7 K in 2H-NbS₂ and 7.2 K in 2H-NbSe₂.^{11–14} They also show quite a different CDW transition strength.^{15,16} 2H-TaSe₂, 2H-TaS₂ and 2H-NbSe₂ undergo a triple incommensurate CDW transition to a superlattice with hexagonal symmetry corresponding roughly to the same wave-vector $\mathbf{q}_{\text{CDW}} = \Gamma M(1 - \delta)2/3$ ($\delta \simeq 0.02$ is the *incommensurate factor*) of the Brillouin zone. However, the transition temperature T_{CDW} increases from 30 K for 2H-NbSe₂ to 80 K for 2H-TaS₂ and 120 K for 2H-TaSe₂ (2H-TaSe₂ actually shows a further commensurate first-order CDW transition at 92 K with δ dropping continuously to zero).¹⁷ Therefore, 2H-NbS₂ considerably stands out as it shows only an incipient instability near \mathbf{q}_{CDW} , but it remains stable even at the lowest temperatures. This circumstance is even more surprising if it is

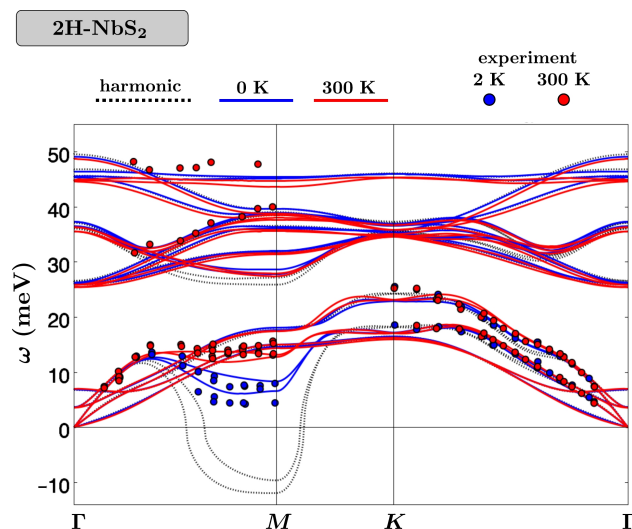


Figure 2: (Color online) 2H-NbS₂ harmonic (black dashed lines) and SSCHA anharmonic phonon dispersion at 300 K (red solid lines) and 0 K (blue solid lines), calculated using the experimental lattice parameters. The results are compared with the IXS measures of Ref. 7 performed at 300 K (red dots) and 2 K (blue dots). The SSCHA dispersion corrects the errors of the pure harmonic result near *M*: the instability of the two longitudinal acoustic and optical modes is removed and the softening on lowering temperature is well reproduced.

considered that 2H-NbSe₂ and 2H-NbS₂ display superconductivity at similar temperatures.

In TMDs, the behavior of the CDW ordering in the two-dimensional limit cannot be inferred from the knowledge of their bulk counterparts, since two competing mechanisms are expected to play a major role. On the one hand, reduced dimensionality strengthens Peierls instabilities (due to Fermi surface nesting) and electron-phonon interactions (due to reduced dielectric screening), thus favoring stronger CDW. On the other hand, stronger fluctuation effects from both finite temperatures and disorders should tend to destroy long-range CDW coherence in low-dimensional systems.¹⁸ In particular, the effect of dimensionality on the CDW ordering in the H polytype is a current active research area. In 1H-TaS₂, the CDW vanishes in the 2D limit,¹⁹ while in 1H-TaSe₂ it remains unchanged with respect to the bulk.²⁰ For 1H-NbSe₂ and 1H-NbS₂ the situation is more debated. In the 1H-NbSe₂ case, 3 × 3 CDW is observed, but some controversy is still present in literature, tentatively attributed either to the sample exposure to air or to the different substrates, concerning the thickness dependence of the T_{CDW} (lower/higher T_{CDW} of the monolayer with respect to the bulk has been reported

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3 with bilayer graphene²¹/silicon¹⁸ substrate, respectively). Supported single layers of 1H-
4 NbS₂ have become recently available, and while no traces of CDW have been observed down
5 to 30 K for monolayers grown on top of Au(111),²² a 3 × 3 CDW ordering has been observed
6 at ultra-low temperature (measurements performed below 5 K) for monolayers grown on top
7 of graphitized 6H-SiC(0001).²³
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13 In this letter we investigate, from first-principles, the vibrational properties of bulk 2H-
14 NbS₂ (at zero and finite pressure) and suspended 1H-NbS₂, taking into account quantum
15 anharmonic effects at non-perturbative level in the framework of the stochastic self-consistent
16 harmonic approximation (SSCHA).^{24–27} For bulk 2H-NbS₂, we show that quantum anhar-
17 monic effects remove the instability found at harmonic level, and give temperature dependent
18 phonon energies in quantitative agreement with experiment. Previous anharmonic calcula-
19 tions for 2H-NbS₂ anticipated the role of anharmonicity, but were limited to a low dimen-
20 sional subspace of the total high dimensional configurations space and did not account for the
21 temperature dependence.¹⁴ We also show that quantum anharmonic effects are noticeable
22 even at high pressure. Moreover, we demonstrate that the difference between 2H-NbS₂ and
23 2H-NbSe₂ is not simply ascribable to the different chalcogen mass. Finally, we analyze the
24 2D limit and show that freestanding single-layer 1H-NbS₂ undergoes a 3 × 3 CDW instability
25 in agreement with data on 6H-SiC(0001) supported samples. However, strains smaller than
26 0.5% are sufficient to completely remove the instability, suggesting a strong dependence of
27 the CDW on the environmental conditions (substrate, charge transfer...) and reconciling the
28 apparent contradiction with supported Au(111) samples.
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45 For bulk 2H-NbS₂, in Fig. 2 we compare the computed anharmonic phonon dispersions
46 with the results of the inelastic X-ray scattering (IXS) experiment of Ref. 7, at low and
47 ambient temperature. We also show the (temperature-independent) harmonic phonon dis-
48 persion. Calculations were performed with the $a_{\text{Exp}}^{2\text{H}} = b_{\text{Exp}}^{2\text{H}} = 3.33 \text{ \AA}$ and $c_{\text{Exp}}^{2\text{H}} = 11.95 \text{ \AA}$ bulk
49 experimental lattice parameters at zero pressure.⁷ The phonon dispersion is almost every-
50 where well reproduced with the harmonic calculation, except close to M , where it predicts
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3 that two longitudinal acoustic and optical modes become imaginary. Experimental phonon
4 energies show a sensible temperature dependence in this region of the *BZ* and are, obvi-
5 ously, always real. The SSCHA cures the pathology of the harmonic result: the anharmonic
6 phonon dispersions do not show any instability and give a very good agreement with the
7 experiment at both temperatures.
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13 Since SSCHA calculations give dispersions in good agreement with experiments, we can
14 perform a wider analysis. In the upper panel of Fig. 3 we show the SSCHA phonon dispersion
15 for different temperatures along the full high-symmetry path of the *BZ*. As temperature
16 decreases, anharmonicity causes the softening of two acoustic and optical longitudinal modes
17 close to both *M* and *L*, but there is no instability. Thus, quantum fluctuations strongly
18 affected by the anharmonic potential stabilize 2H-NbS₂. In the other two panels we show
19 the effect of hydrostatic pressure on the phonon dispersion. Since there are no available
20 experimental lattice parameters at high pressures, we estimated them by assuming that the
21 ratio between experimental and standard DFT theoretical lattice parameters (i.e. the lattice
22 parameters that minimize the DFT energy but do not take into account any lattice quantum
23 dynamic effects), $a_{\text{Exp}}^{2\text{H}}(P)/a_{\text{Th-DFT}}^{2\text{H}}(P)$ and $c_{\text{Exp}}^{2\text{H}}(P)/c_{\text{Th-DFT}}^{2\text{H}}(P)$, are independent of the applied
24 pressure *P*. Thus we computed those ratios at zero pressure and, for a given pressure *P*, the
25 calculations were performed using as lattice parameters $a = (a_{\text{Exp}}^{2\text{H}}/a_{\text{Th-DFT}}^{2\text{H}}) \times a_{\text{Th-DFT}}^{2\text{H}}(P)$ and
26 $c = (c_{\text{Exp}}^{2\text{H}}/c_{\text{Th-DFT}}^{2\text{H}}) \times c_{\text{Th-DFT}}^{2\text{H}}(P)$. Increasing pressure the anharmonicity of the lowest energy
27 modes around *M* and *L* decreases, but remains relevant even up to 14 GPa. A similar
28 conclusion was drawn for 2H-NbSe₂, where large anharmonic effects and strong temperature
29 dependence of these phonon modes were observed as high as 16 GPa, in a region of its phase
30 diagram where no CDW transition is observed.²⁸
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49 These results confirm the importance of quantum anharmonicity in 2H-NbS₂ to describe
50 experimental data and the absence of a CDW instability. It is tempting, at this point,
51 to use the same technique to shed light on the different CDW behavior exhibited by the
52 very similar compound 2H-NbSe₂. Indeed, as we showed in a previous work,²⁸ the SSCHA
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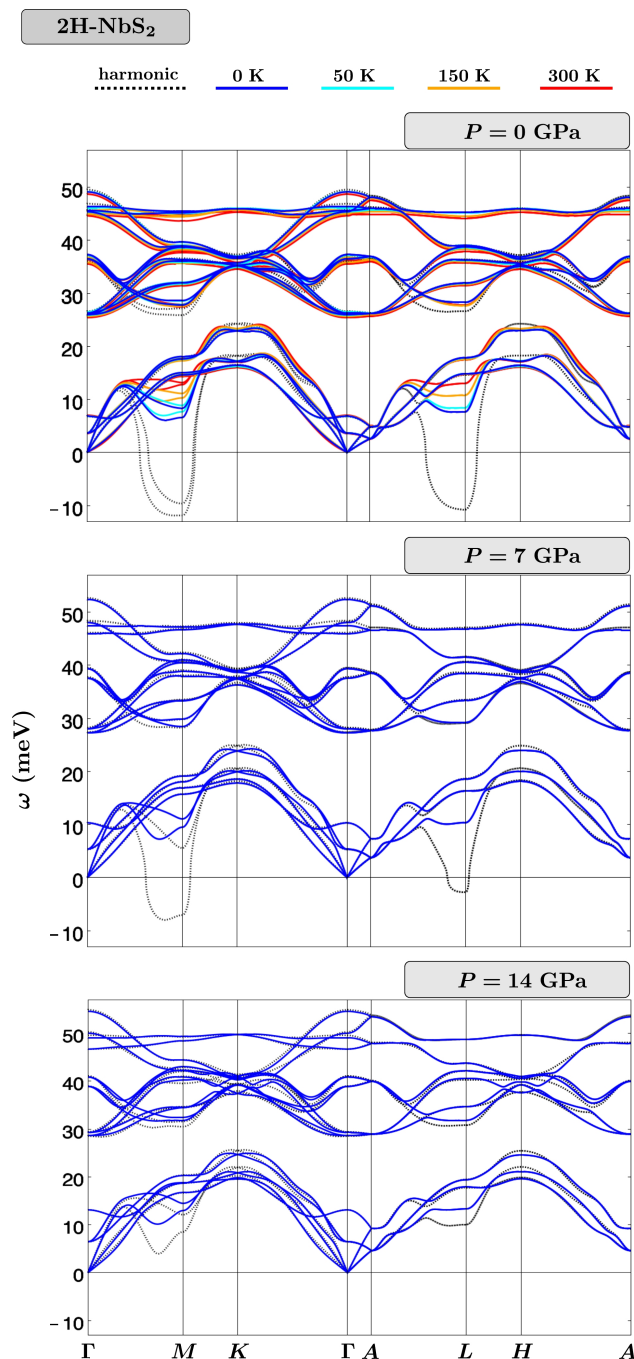
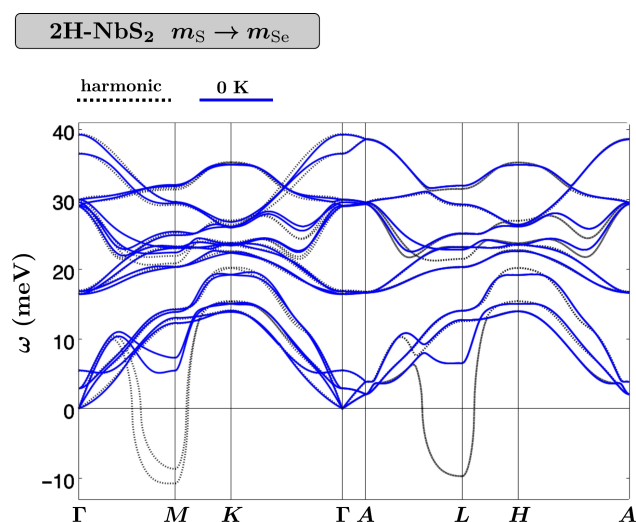


Figure 3: (Color online) 2H-NbS₂ harmonic phonon dispersion (black dashed lines) and SSCHA anharmonic phonon dispersion at several temperatures (colored solid lines). Results for different pressures are shown. From the top to the bottom panel: 0 GPa, 7 GPa, 14 GPa. The zero pressure results are obtained using the experimental lattice parameters. The high pressure results are obtained assuming that the ratio between experimental and DFT theoretical lattice parameters are independent of the applied pressure (more details in the main text). Anharmonicity removes the instability, obtained at harmonic level, of the longitudinal acoustic and optical modes near *M* and *L* at 0 GPa and 7 GPa. Anharmonicity reduces as the pressure increases but it has a noticeable effect even at 14 GPa.



36 Figure 4: 2H-NbS₂ harmonic phonon dispersion (black dashed lines) and SSCHA anharmonic
37 phonon dispersion at 0 K (blue solid lines), at zero pressure, computed replacing the mass of
38 S with the mass of Se (more details in the main text). Anharmonicity removes the instability
39 also in this case.
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3 correctly displays the occurrence of CDW in 2H-NbSe₂ at ambient pressure. One evident
4 difference between 2H-NbS₂ and 2H-NbSe₂ is, of course, the mass of the chalcogen atom. We
5 then performed a SSCHA calculation at 0 K for 2H-NbS₂ with “artificial” S atoms having
6 unaltered electronic configuration but the mass of Se. In other words, we performed a SSCHA
7 calculation where the average displacements of the atoms from the equilibrium position is
8 ruled by the Se mass, but for each fixed position of the atoms the electronic structure is
9 computed with the normal S atoms. The results are shown in Fig. 4. Also in this case, when
10 quantum anharmonic effects are included the system does not show any CDW instability.
11 Thus the different behavior of 2H-NbS₂ and 2H-NbSe₂ cannot be ascribed to a mass effect
12 but has a more complex origin related to the different electron screening on the ions.
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23 The validity of the results obtained with the SSCHA method on bulk 2H-NbS₂ gives us
24 confidence that a similar calculation on the 1H-NbS₂ monolayer may shine light about the
25 effects that dimensionality and environmental conditions (substrate, doping) can have on
26 the CDW ordering in metallic TMDs. The suspended 1H-NbS₂ monolayer was simulated
27 leaving 12.55 Å of vacuum space between a 1H layer and its periodic replica. At conventional
28 static DFT level, we found that the theoretical zero pressure in-plane lattice parameter of
29 the monolayer and the bulk are essentially the same, $a_{\text{Th-DFT}}^{2\text{H}} \simeq a_{\text{Th-DFT}}^{1\text{H}} \simeq 3.34\text{Å}$. Therefore,
30 for the suspended monolayer we use as in-plane lattice parameter the bulk experimental
31 one, $a_{\text{Exp}}^{2\text{H}} = 3.33\text{Å}$. This value is also compatible with the recent experimental measures
32 $3.29 \pm 0.03\text{Å}$ and 3.34Å reported for the lattice parameter of monolayer grown on substrate
33 in Ref. 23 and Ref. 22, respectively.
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45 In the upper panel of Fig. 5, we show the harmonic and SSCHA anharmonic phonon
46 dispersions of suspended 1H-NbS₂ at several temperatures, calculated with the lattice pa-
47 rameter $a_{\text{Exp}}^{2\text{H}}$. As in the bulk case, the system is unstable at harmonic level, but it is stabilized
48 by quantum fluctuations strongly sensitive to the anharmonic potential down to 0 K. How-
49 ever, comparing Figs. 2 and 5, we observe that even if the used in-plane lattice parameter is
50 the same in both cases, at 0 K the softest theoretical phonon frequency is approximately 20%
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3 harder in the bulk than in the single layer case, demonstrating that there is a substantial
4 enhancement of the tendency toward CDW in the 2D limit. In the monolayer, the theoretical
5 phonon softening is localized in $\mathbf{q}_{\text{CDW}} = 0.72\overline{\Gamma M}$, which is quite close to the $\mathbf{q}_{\text{CDW}} \simeq 2/3\overline{\Gamma M}$
6 of the CDW instability experimentally found in 1H-NbS₂ on 6H-SiC(0001)²³ (and in 1H-
7 NbSe₂^{18,21}). Notice that, since the computed wave-vector of the instability may be affected
8 by the finite grids used in the calculations, we do not discard that it may be slightly shifted
9 in the infinite grid limit.
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17 Pressure tends normally to remove CDW ordering. Therefore, considering the proximity
18 of the instability, it cannot be discarded that a tensile dilatation due to the substrate may
19 induce the CDW transition observed for 1H-NbS₂ on graphitized 6H-SiC(0001). However,
20 for the same reason, we cannot exclude the more interesting prospect that the observed
21 CDW be an intrinsic property of this system. Indeed, even small variations of the lattice
22 parameter, compatible with the experimental uncertainty, could have a relevant impact on
23 the results of the calculations, and a more accurate theoretical analysis of the monolayer
24 structure is therefore necessary. As the energy of the soft-mode along $\overline{\Gamma M}$ is of the order of
25 $\simeq 58$ K, for a proper analysis of the CDW in the monolayer it is important to fully take into
26 account quantum effects. Including quantum anharmonic contributions to strain through
27 the technique introduced in Ref. 26, we find that with the used lattice parameter $a_{\text{Exp}}^{2\text{H}}$ the
28 structure is slightly compressed, with an in-plane pressure $P = 0.66$ GPa. Upon relaxation
29 we obtain the theoretical lattice parameter $a_{\text{Th}}^{1\text{H}} = 3.35$ Å, approximately 0.5% larger than
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45 The harmonic and quantum anharmonic phonons at 0 K calculated with the lattice pa-
46 rameter $a_{\text{Th}}^{1\text{H}}$ are shown in the bottom panel of Fig. 5. While at harmonic level the phonon
47 dispersion is not substantially different from the one computed with $a_{\text{Exp}}^{2\text{H}}$, when quantum
48 anharmonic effects are included the phonon dispersion at 0 K now shows an instability
49 at $\mathbf{q}_{\text{CDW}} = 0.72\overline{\Gamma M}$, thus in agreement with the CDW observed for 1H-NbS₂ on top of
50 6H-SiC(0001). The obtained instability is very weak (i.e. the obtained imaginary phonon
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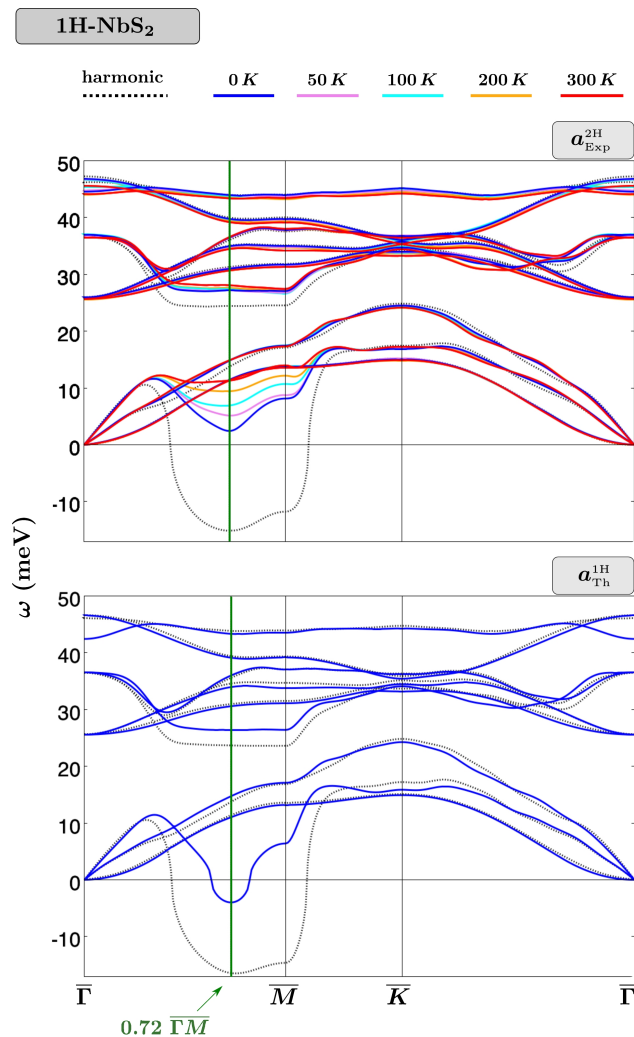


Figure 5: (Color online) Suspended 1H-NbS₂ harmonic phonon dispersion (black dashed lines) and SSCHA anharmonic phonon dispersion at several temperatures (colored solid lines), at zero pressure. Upper panel: results obtained with the experimental in-plane bulk lattice parameter $a_{\text{Exp}}^{2\text{H}}$. The softening of the acoustic mode, localized at $\mathbf{q}_{\text{CDW}} = 0.72\bar{\Gamma}\bar{M}$, is more pronounced than in the 2H bulk case. However, the frequencies remain real even at 0 K. Lower panel: results obtained with the theoretical lattice parameter $a_{\text{Th}}^{1\text{H}}$, obtained by fully relaxing the structure taking into account quantum anharmonic effects. At 0 K the frequency at $\mathbf{q}_{\text{CDW}} = 0.72\bar{\Gamma}\bar{M}$ becomes imaginary.

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3 frequency is very small). Therefore, this result is also compatible with the hypothesis that
4 charge doping from the substrate could be at the origin of the CDW suppression for 1H-
5 NbS₂ on top of Au(111), similarly to what it was proposed for the case of 1H-TaS₂ on top
6 of Au(111).²⁹ Our results show that if quantum anharmonic effects are included, then even
7 a small compression/dilatation of approximately 0.5% removes/induces the charge density
8 wave instability on 1H-NbS₂. The extreme sensitivity of the CDW on environmental con-
9 ditions therefore suggests that deposition of 1H-NbS₂ on flexible substrates,³⁰⁻³² or a small
10 charge transfer via field effect, could lead to devices with dynamical on/off switching of the
11 3×3 order.
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21 In conclusion, we have shown that quantum anharmonicity is the key interaction for the
22 stabilization of the crystal lattice in bulk 2H-NbS₂, as it removes the instability found at
23 the harmonic level. The calculated temperature dependence of the phonon spectra are in
24 excellent agreement with inelastic X-ray scattering data. Anharmonicity remains important
25 even at large pressures. Given the good agreement between theory and experiment in bulk
26 2H-NbS₂, we have studied the behavior of the CDW in the 2D limit by considering single
27 layer 1H-NbS₂. We found that suspended 1H-NbS₂ undergoes a quantum phase transition
28 to a CDW state with approximately 3×3 charge ordering in the 2D limit, in agreement
29 with experimental results on supported samples on 6H-SiC(0001). However, the CDW is
30 extremely sensitive to environmental conditions, as it is very weak and compressive strains
31 smaller than 0.5% are enough to suppress it. This explains the absence of CDW observed in
32 1H-NbS₂ on top of Au(111). This also suggest that devices with dynamical on/off switching
33 of the 3×3 charge order can be obtained with deposition of 1H-NbS₂ on flexible substrates,
34 or through a small charge transfer via field effect.
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19 (10) Recent diffuse X-ray scattering experiments report faint traces of CDW in the 2H
20 polytype of NbS_2 commensurate with a $\sqrt{13} \times \sqrt{13}$ supercell, therefore identical to the
21 supercell modulation associated with the CDW of 1T-TaSe₂ and 1T-TaS₂.⁸ However,
22 this is attributed to the presence of isolated rare and dilute octahedral 1T layers, caused
23 either by dilute amount of Nb in the van der Waals interlayer space of 3R-like stacking
24 faults or rotational disorder in the stacking of 1H layers.
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3 and a $40 \times 40 \times 1$ Monkhorst-Pack grid³⁵ of the Brillouin zone (*BZ*), respectively. These
4 meshes were adjusted accordingly in the supercell calculations. We used ultrasoft pseu-
5 dopotentials,³⁶ a plane-wave cutoff energy of 35 Ry for the kinetic energy and 400 Ry
6 for the charge density, and a Methfessel-Paxton smearing of 0.005 Ry.³⁷ The SSCHA
7 calculations were performed on a $4 \times 4 \times 1$ supercell for the bulk and on a $6 \times 6 \times 1$ su-
8 percell for the monolayer. The short-range part of the anharmonic dynamical matrices
9 were computed with SSCHA in the static approximation for the phonon self-energy,
10 retaining only the so called “bubble term”.^{25,38} The long-range part of the anharmonic
11 dynamical matrices were obtained first by Fourier interpolating the difference between
12 the SSCHA and the harmonic dynamical matrices to finer grid, and then adding the
13 harmonic contribution calculated on this grid ($6 \times 6 \times 4$ for the bulk and $16 \times 16 \times 1$ for
14 the monolayer). The phonon dispersion along high-symmetry paths were subsequently
15 obtained by Fourier interpolation.
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Graphical TOC Entry

