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RECENT ADVANCES IN HYDROGELS FOR CARTILAGE TISSUE ENGINEERING

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Abstract

Articular cartilage is a load-bearing tissue that lines the surface of bones in diarthrodial joints. Unfortunately, this avascular tissue has a limited capacity for intrinsic repair. Treatment options for articular cartilage defects include microfracture and arthroplasty; however, these strategies fail to generate tissue that adequately restores damaged cartilage. Limitations of current treatments for cartilage defects have prompted the field of cartilage tissue engineering, which seeks to integrate engineering and biological principles to promote the growth of new cartilage to replace damaged tissue. To date, a wide range of scaffolds and cell sources have emerged with a focus on recapitulating the microenvironments present during development or in adult tissue, in order to induce the formation of cartilaginous constructs with biochemical and mechanical properties of native tissue. Hydrogels have emerged as a promising scaffold due to the wide range of possible properties and the ability to entrap cells within the material. Towards improving cartilage repair, hydrogel design has advanced in recent years to improve their utility. Some of these advances include the development of improved network crosslinking (*e.g.* double-networks), new techniques to process hydrogels (*e.g.* 3D printing) and better incorporation of biological signals (*e.g.* controlled release). This review summarises these innovative approaches to engineer hydrogels towards cartilage repair, with an eye towards eventual clinical translation.

Keywords: Hydrogel, biomaterial, cartilage, tissue engineering.

Introduction

In simplistic terms, articular cartilage is a tissue consisting of a single cell type (chondrocyte) embedded within an extracellular matrix (ECM). However, the structure is more complex and includes three depth-dependent layers: the superficial zone, the middle zone and the deep zone, with changes in ECM content, structure and chondrocyte behaviour with depth. In the middle zone – the largest region – cartilage tissue is stiff, avascular, and features a low density of rounded chondrocytes surrounded by an ECM consisting of glycosaminoglycans (GAGs) and type II collagen (Vunjak-Novakovic and Freed, 1998; Wong and Carter, 2003). Cartilage damage due to trauma typically begins as a focal defect, which later progresses to a full-thickness defect once the lesion comes into contact with surrounding bone. Due to low cellularity, low vascularisation, minimal proliferative capacity of residing chondrocytes and low cell migration to areas of damage, articular cartilage is intrinsically unable to repair itself (Ahmed and Hincke, 2010).

Current strategies to repair focal and full-thickness cartilage defects have evolved from surgery aimed at inducing endogenous repair mechanisms (*e.g.* microfracture), towards osteochondral transplantation, and more recently to cell-based repair techniques, such as autologous chondrocyte implantation (ACI) (Brittberg, 2008; Goldring, 2006; Hunziker, 2002). ACI is a procedure in which patient chondrocytes are implanted into a debrided cartilage lesion (Peterson *et al.*, 2010). Both microfracture and ACI lack 3D scaffolds to define where cells reside and to instruct matrix formation; however, there have been advances in these therapies in recent years. Matrix-induced autologous chondrocyte implantation (MACI) is a cell-based therapy that supplements ACI by providing chondrocytes with a supportive scaffold material for matrix formation (Makris *et al.*, 2015). Autologous matrix-induced chondrogenesis (AMIC), on the other hand, supplements microfracture with an acellular scaffold in the lesion as an attempt to increase mechanical stability at the site, cell migration and cartilage production (Benthien and Behrens, 2011). Despite promising results, these clinical techniques are not without their limitations, including the surgical procedures and the quality of the formed cartilage.

Towards further expanding the use of biomaterials in cartilage repair, hydrogels have gained a great deal of interest in cartilage tissue engineering (Balakrishnan and Banerjee, 2011). Hydrogels are 3-dimensional (3D) polymer networks that are highly swollen and porous on the molecular scale, allowing the diffusion of various solutes and nutrients. Their fabrication can be cell amenable, allowing the encapsulation of different cell types (*e.g.* chondrocytes, stem cells). Lastly, an assortment of

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hydrogel parameters can also be tuned, including polymer chemistry, crosslinking density, degradation, mechanical properties and release kinetics of biochemical factors, towards improving their utility in tissue repair (Spiller *et al.*, 2011). Numerous studies have characterised the effects of hydrogel crosslinking – tuned through parameters like macromer concentration – on neocartilage formation (Chung *et al.*, 2006). These studies have shown that matrix distribution can be hindered when a hydrogel with a high degree of crosslinking is used. Also, differences in the composition of hydrogels have been investigated in the context of cartilage formation. For example, several groups report that type II collagen hydrogels induce a higher amount of chondrogenic outputs, in comparison to type I collagen hydrogels with similar properties (Lu *et al.*, 2010; Nehrer *et al.*, 1997).

To serve as constructs to replace articular cartilage, hydrogels have been used as either acellular scaffolds or as cell-laden biomaterials (Spiller *et al.*, 2011). For both strategies, hydrogel implants must integrate with surrounding tissue and support the production of cartilage. Acellular constructs are almost exclusively assayed *in vivo*, where the goal is structural and mechanical properties similar to native cartilage, while allowing efficient load transfer, or the recruitment of cells for cartilage repair (Kobayashi and Oka, 2004; Stammen *et al.*, 2001). Cell-laden hydrogels present residing cells with either developmental cues to trigger chondrogenesis, or microenvironmental cues that mimic native cartilage to maintain a chondrogenic phenotype and encourage matrix formation. Hydrogels are being developed to present these signals either as molecules bound to the polymer or through their controlled release. Also, these cultures may be performed *in vitro*, such as in bioreactors, or may be implanted directly for tissue growth.

With rapid advances being made towards hydrogels for cartilage tissue engineering, the goal of this review is to cover current and emerging developments in hydrogel design for cartilage repair. It is not meant to be comprehensive, but rather highlight representative advances in the field in recent years with a particular focus on hydrogel design. Specific areas include the use of unique crosslinking to produce hydrogels with strong mechanical properties, the improved processing of hydrogels into macroporous structures and the incorporation of biological signals to improve cell behaviour.

Improvements in hydrogel structure

One of the key design criteria for hydrogels towards cartilage regeneration is mechanical integrity. Traditional hydrogel designs, based on networks of a single polymer, generally result in hydrogel constructs with mechanical properties far inferior to those of native cartilage. Also, the increase in hydrogel modulus through increased crosslink density may compromise the viability of encapsulated cells through reduced diffusion. With the aim of increasing the mechanical properties of hydrogels to approach those of hyaline cartilage, the focus is shifting from conventional hydrogels that use a single polymer for hydrogel fabrication

(Fig. 1a), to more complex hydrogel systems with mixtures of multiple polymers, often including two or more independent networks. These systems not only typically achieve stronger mechanical properties than networks of single polymers, but may exhibit superior integration with surrounding tissue *in vivo*. In this section, we examine recent advances within the framework of different network types (Fig. 1).

Interpenetrating networks

Hydrogels based on interpenetrating networks (IPNs) are comprised of two or more separate crosslinked networks not covalently bound to each other, but rather partially intertwined such that chemical bonds have to be broken to separate the components that form the networks (Fig. 1b). Consequentially, the mechanical properties of IPN hydrogels tend to be stronger than those created with individual component networks, which makes them appealing for cartilage tissue engineering applications. In recent years, the design of IPNs for hydrogel formation has turned to the incorporation of two or more additional networks to either better mimic the physical properties of native tissue or to recapitulate the presentation of bioactive cues available to cells in the hydrogel constructs.

Recent examples of IPNs include the work done by Ingavle and colleagues, in which they explored the incorporation of methacrylated chondroitin sulphate, a major component of the cartilage ECM, as the second network after diffusion into an existing agarose-poly(ethylene glycol) diacrylate (PEGDA) network. Incorporating chondroitin sulphate into the network significantly increased the viability of encapsulated chondrocytes for at least six weeks and promoted greater biosynthesis of collagen and GAGs in the pericellular matrix (Ingavle *et al.*, 2013). Reinforcing networks such as those provided by methacrylated hyaluronic acid (HA) within a fibrin hydrogel have been established as well (Fig. 2a) (Snyder *et al.*, 2014). Snyder and colleagues found that the incorporation of a crosslinked HA network increased the compressive moduli of hydrogel constructs and modulated gene expression of encapsulated human mesenchymal stem cells (MSCs) by decreasing expression of the dedifferentiation marker type I collagen and increasing expression of the chondrogenic transcription factor Sox9 (Snyder *et al.*, 2014).

In addition to IPNs of two networks, several groups have also investigated how tri-component IPNs increase mechanical properties and cartilage formation using both non-cell instructive (Dinescu *et al.*, 2015; Liao *et al.*, 2013) and cell-instructive materials (Guo *et al.*, 2012). In a non-cell instructive scaffold, comprised of gelatin, alginate and polyacrylamide, Dinescu and co-workers reported higher cell proliferation, lower cytotoxicity and greater chondrogenic gene expression (Sox9, type II collagen) of human adipose-derived stem cells than in one- or two-component hydrogels. The authors attributed these findings to the tri-component network's ability to better retain its internal structure and porosity in long-term culture (Dinescu *et al.*, 2015). Guo *et al.* (2012) further used this tri-component IPN concept to better approximate the diversity of biomaterials present in articular cartilage. They

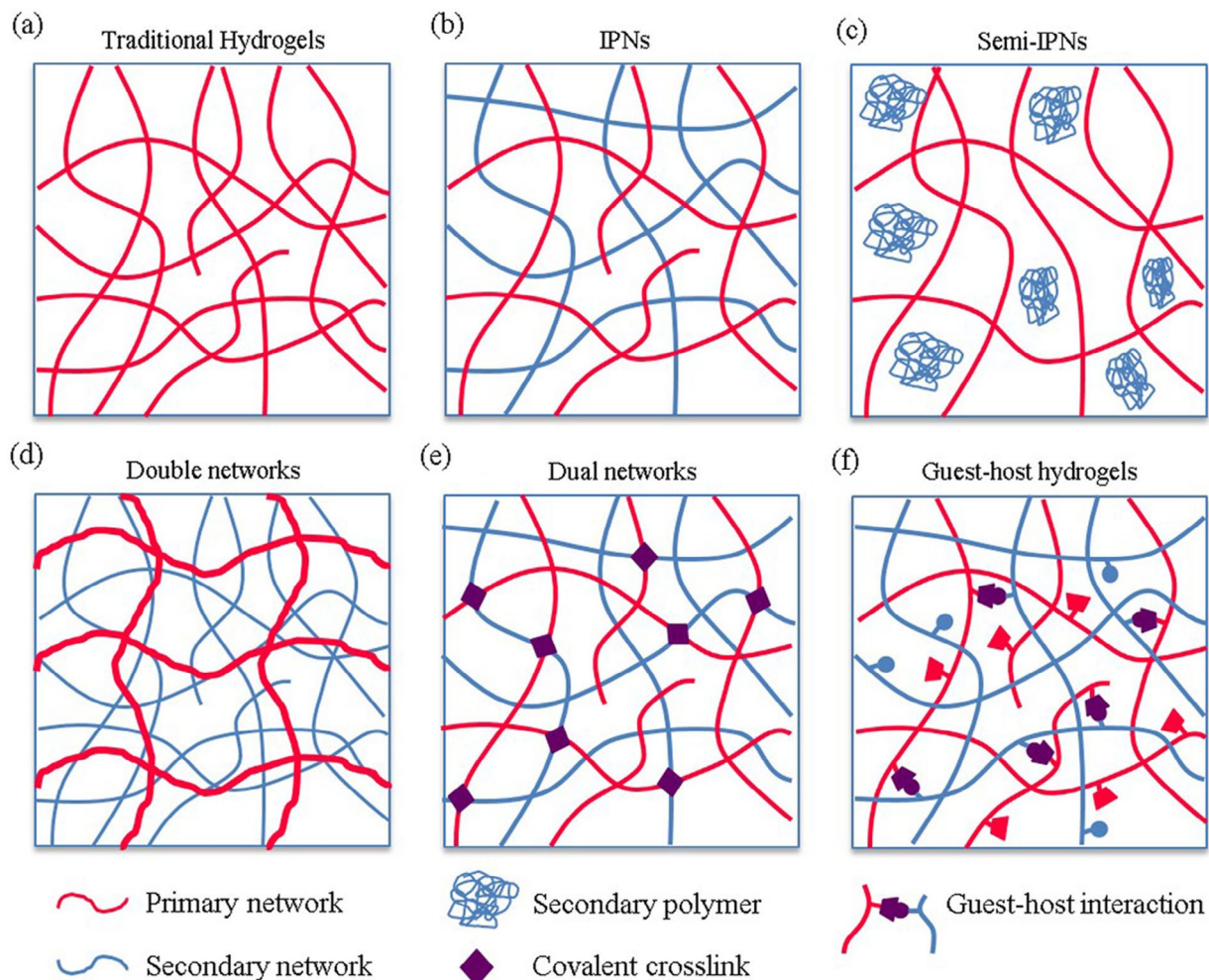


Fig. 1. Schematic representation of different designs utilised in hydrogels, from (a) traditional single polymer networks to those that include (b,c) multiple networks and (d-f) mixtures of polymers. Double networks may be linked together, but this is not a requirement. Generally, the network design controls properties such as mechanics and degradation.

demonstrated the feasibility of generating a tri-component IPN using only cell-instructive components (collagen combined with methacrylate-modified chondroitin sulphate and HA) (Guo *et al.*, 2012). By comparing it to a semi-IPN using the same components (where the chondroitin sulphate and HA were not methacrylated) and by varying the extent of methacrylation, they demonstrated tuneable mechanics with the tri-component IPNs, increased gene expression of chondrogenic markers (*i.e.* aggrecan, type II collagen and Sox9), downregulation of the dedifferentiation marker type I collagen and increased collagen and GAG synthesis by encapsulated rabbit chondrocytes that correlated with the bulk mechanics of the IPN constructs (Guo *et al.*, 2012).

Semi-interpenetrating networks

Unlike IPNs, semi-IPNs consist of a crosslinked network with either linear or branched polymers entrenched within the network (Fig. 1c). As such, the polymers and the network can theoretically be separated from one another without breaking chemical bonds. One of the most common macromolecules distributed in these networks is HA. In a recent study, high molecular weight HA was distributed within injectable networks of photopolymerised

methacrylated chitosan (MeGC) (Park *et al.*, 2013). An optimum formulation and gelation protocol was developed, and the presence of entangled (but unbound) HA (350 kDa) incorporated into the chitosan network resulted in stronger proteoglycan and GAG staining (as measured by alcian blue and safranin O) in the lacunae of chondrocytes encapsulated and cultured for up to 21 d (Park *et al.*, 2013).

Another approach towards utilising the properties of semi-IPNs is exploiting their ability to leach low-molecular weight macromolecules over time, as demonstrated by Skaalure *et al.* in a degradable poly(ethylene glycol) (PEG) hydrogel with low-molecular weight HA (29 kDa) (Skaalure *et al.*, 2014). By comparing PEG networks infiltrated by either low- or high-molecular weight HA (2 MDa), the authors found that leaching of low molecular weight HA over a 28 d period led to the greatest soluble GAG deposition during construct maturation, while collagen biosynthesis was comparable to that generated in constructs with high molecular weight HA (which remained in the network during this period) (Skaalure *et al.*, 2014). Conceptually similar approaches taken by Little *et al.* suggest that low-molecular weight HA and chondroitin sulphate in a fibrin-alginate hydrogel may

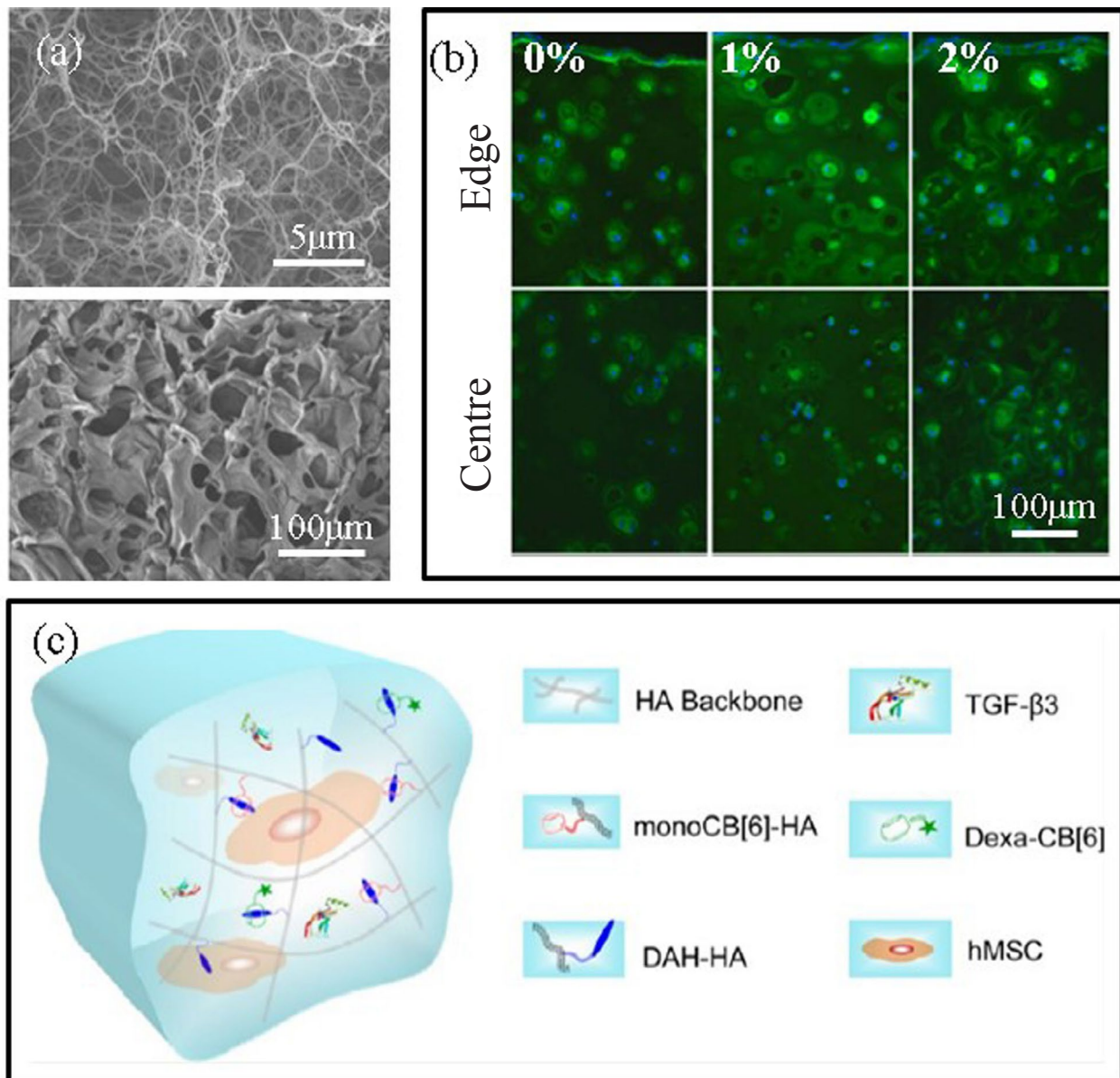


Fig. 2. (a) Scanning electron microscopy (SEM) of a fibrinogen hydrogel with (bottom) and without (top) interpenetrating methacrylated HA network (Snyder *et al.*, 2014). (b) MSCs encapsulated in gelatin methacrylamide hydrogels exhibit more aggrecan (green) with increasing concentrations of methacrylated HA, which acts as a dual network (Levett *et al.*, 2014). (c) Hydrogel molecular structures and crosslinking schemes can become quite complex, as seen by this schematic representation of supramolecular hydrogels prepared with CB[6]-HA, DAH-HA and drug conjugated DEXA-CB[6] (Jung *et al.*, 2014).

exert post-transcriptional effects on collagen expression to influence construct composition as cells deposit matrix over time (Little *et al.*, 2014).

Double networks

A double network consists of two networks with significantly different mechanical properties crosslinked together (Fig. 1d). Typically, the first network provides a rigid structure and the second network is ductile, resulting in greater toughness than the corresponding single networks alone would have achieved since the network can yield under mechanical load (Gong, 2010). These types of networks have gained interest in cartilage tissue engineering due to their superior mechanical properties over traditional hydrogels, including those that can approach the mechanics

of native hyaline cartilage (Arnold *et al.*, 2011). Double networks are also conceptually appealing since cartilage and other skeletal tissues inherently incorporate double networks into their ECM in order to achieve their robust mechanical properties (Arnold *et al.*, 2011).

Commonly utilised double networks for cartilage tissue engineering consist of a combination of two acrylamide polymers, poly(2-acrylamido-2-methylpropanesulfonic acid) (PAMPs), poly(acrylamide) (PAAm) and/or poly(N,N-dimethylacrylamide) (PDMAAm) (Yasuda *et al.*, 2009). Work with these double network hydrogels has recently extended to *in vivo* studies in both rabbit and sheep models of critical size defects, in which acellular plugs comprised of this double network implanted in defects resulted in enriched GAG and type II collagen content as

measured by histological staining after 4 weeks (Kitamura *et al.*, 2016). In all cases, the double-network plug was not infiltrated by cells and instead seemed to serve as a support material for tissue regeneration above and around it.

A direct comparison between double network hydrogels and traditional single network hydrogels of either only PAMPS or PDMAAm demonstrated that double network hydrogel constructs resulted in superior cartilage repair by histological scoring (Ogawa *et al.*, 2012). The tissue formed in the presence of double networks also exhibited similar gene expression profiles (Imabuchi *et al.*, 2011) and tissue surface roughness of native cartilage, even as observed by confocal laser scanning microscopy. Furthermore, combinatorial therapies using double network hydrogels and intra-articular injections (*e.g.* HA) have shown potential in tissue quality as measured by histological scoring and volume of cartilage generated by double network hydrogels implanted in cylindrical osteochondral defects in rabbit femoral trochlea at both 4 and 12 weeks, using acellular plugs of the PAMPS/PDMAAm double network (Fukui *et al.*, 2014).

However, the application of novel and cell-instructive materials for double network construct design in cartilage tissue engineering has not been widely explored until recently. Polymers used for double networks are often non-degradable, and while this stability renders them potentially useful materials for mechanical support in tissue defects, this stable and rigid network may limit cell infiltration and impede long-term matrix deposition and distribution. Despite these properties of double-network hydrogels, Levett and co-workers developed a double network consisting of a combination of gelatin-methacrylamide and hyaluronic acid-methacrylate. This system leveraged the relatively higher reactivity of methacrylate groups to create a double network, thus generating greater increases in compressive modulus and cartilage matrix component synthesis by encapsulated human chondrocytes (Levett *et al.*, 2014). New hydrogels making use of this type of network are continuously being developed, such as an injectable and cytocompatible double network hydrogel based on HA that is formed through the combination of dynamic (*i.e.* guest-host pairs) and stable (*i.e.* covalent) crosslinks (Rodell *et al.*, 2016).

Dual networks

Unlike double networks that use two materials with different mechanical properties, dual networks are defined as two materials crosslinked together into the same network and with similar crosslinking mechanisms (Fig. 1e). Although dual networks do not possess the toughness of double networks, each material in dual networks can imbue other useful properties to the hydrogel. For instance, one material can enable effective integration with the surrounding tissue, while the other can attract cells and encourage migration into the hydrogel. Moreira Teixeira *et al.* utilised a dextran-tyramine and heparin-tyramine dual network hydrogel to encapsulate bovine chondrocytes *in vitro* and reported improved cell viability and proliferation. Additionally, they observed increased deposition of chondroitin sulphate and collagen compared to cells in a single-component dextran-tyramine hydrogel (Moreira

Teixeira *et al.*, 2012). Jin *et al.* (2011) reported similar findings using the same dual network polymers *ex vivo* with bovine articular cartilage explants ($0.5 \times 0.5 \times 0.2$ cm) to determine tissue-adhesion of the hydrogel constructs (Jin *et al.*, 2011). In a more recent study, *in situ* forming ethylenediamino-functionalised HA (HA-EDA) and divinylsulfone-inulin (INU-DV) dual network hydrogels were used to encapsulate chondrocytes, with the first material selected due to its native presence in articular cartilage and the second for its biocompatibility and status as a widely-used FDA-approved polymer (Palumbo *et al.*, 2015).

Along the same lines of using HA as the biologically relevant component, Pirinen and co-workers developed a dual network hydrogel using high molecular weight HA (> 1,600 kDa) and low molecular weight PVA (27 kDa) functionalised with aldehydes and primary amines for crosslinking. This dual hydrogel system was amenable to tuneable swelling properties by varying the size of the smaller PVA component, and encapsulation of bovine knee chondrocytes showed favourable cell viability for at least 2 weeks in culture (Pirinen *et al.*, 2015). Similar hydrogel systems (*e.g.* with HA and gelatin) have also produced favourable results in construct formation and maturation. Levett and colleagues reported that the addition of HA as the second component resulted in retention of a rounded chondrocyte morphology, greater aggrecan deposition and compressive moduli and suppression of type I collagen accumulation over the course of a 56 d study (Fig. 2b) (Levett *et al.*, 2014).

Guest-host networks

Injectable hydrogels are an attractive approach for cartilage tissue engineering since they can be delivered *via* a direct injection or arthroscopically (Guvendiren *et al.*, 2012). Shear-thinning hydrogels are a particularly important method towards this as a hydrogel can be disrupted during the injection process and then self-heal upon injection. One shear-thinning HA system developed is based on the reversible bonds between guest (adamantane modified HA) and host (β -cyclodextrin modified HA) polymers, termed guest-host interactions (Fig. 1f) (Rodell *et al.*, 2013). These types of interactions rapidly form hydrogels by non-covalent interactions in the guest-host bonds. In addition to cyclodextrin-adamantane guest-host pairs, guest-host interactions between cucurbit[6]uril and diamino-hexane have also been investigated (Jung *et al.*, 2014). Both of these interactions are reversible and give rise to shear-thinning and self-healing hydrogels.

Wei and colleagues described the assembly of a two-component guest-host hydrogel with adamantane-functionalised HA as the first component and monoacrylated β -cyclodextrin (assembled into clusters by photopolymerisation) as the second component (Wei *et al.*, 2016). The hydrogels formed from these components exhibited nanoclustered interactions that allowed for robust drying and re-swelling without changes in water content or shape. They also enabled the chondrogenesis of encapsulated human MSCs with greater collagen deposition compared to a covalently crosslinked methacrylated HA network, although the differences in

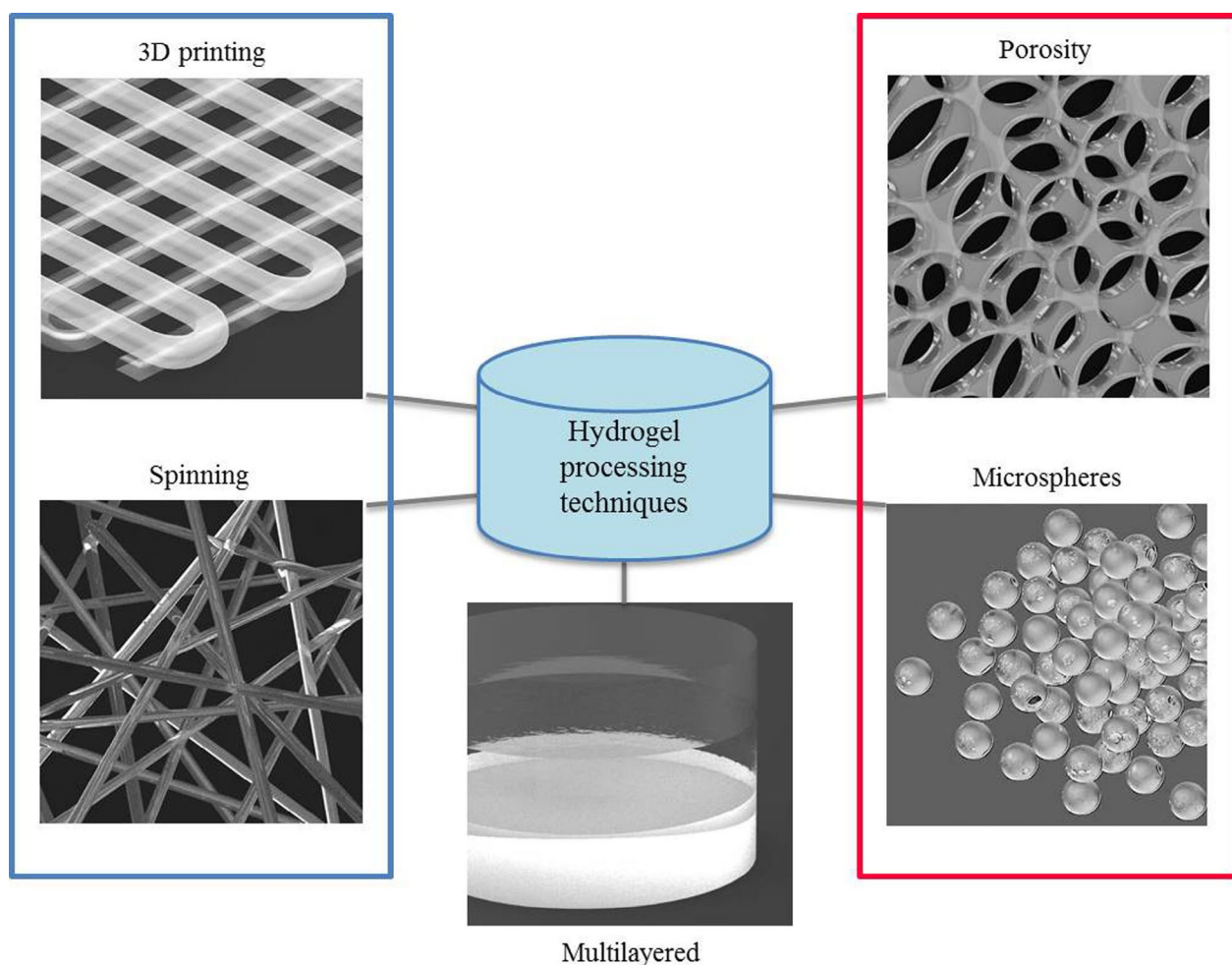


Fig. 3. Overview of different macroporous scaffold structures used for cartilage tissue engineering. To create hydrogel fibres, 3D printing and spinning techniques have been employed (blue box). In contrast, porous hydrogels and complementary microsphere hydrogels can also be fabricated (red box). To recapitulate native cartilage structures (e.g. different regions of cartilage, the osteochondral interface), multi-layer hydrogels incorporating several fabrication techniques can also be utilised (bottom).

biophysical properties and network structure limited a direct comparison (Wei *et al.*, 2016). Jung *et al.* (2014) demonstrated the use of guest-host interactions between curcubit[6]uril and diamino hexane to create hydrogels for the encapsulation of human MSCs as well as simultaneous release of dexamethasone for chondrogenesis. Cells in this injectable guest-host system, especially with the additional function of controlled dexamethasone release, exhibited increased cell proliferation, GAG synthesis, chondrogenic gene expression (*i.e.* type II collagen, COMP, aggrecan, Sox9) and neocartilage formation in an *in vivo* subcutaneous study (Fig. 2c).

Advances in the processing of hydrogel scaffolds

Parallel advances in manufacturing technologies and material design have paved the way for new possibilities in how hydrogels can be manipulated and formed into 3D macroporous scaffolds. Architectures that aid viability, retention and chondrogenic induction of cells have been achieved using several techniques including 3D printing, spinning and doping hydrogels with degradable porogens

or microspheres (Fig. 3). Due to their ability to generate unique microenvironments for cells and better recapitulate tissue structure on the macroscale, these techniques have become more prominent in tissue engineering in general and in cartilage engineering in particular. Recent studies making use of some of these more widely adopted methods are outlined in the subsections below.

Hydrogel fibres

Fibre fabrication techniques at length scales ranging from 150 nm to 1600 μm are typically generated by either spinning or 3D printing. The resultant fibrous networks offer a number of potential advantages, from more facile cell infiltration and greater hydrogel surface area (and thus improved diffusion of nutrients and other soluble factors) to increased mechanical properties. The techniques involve both the encapsulation of cells within macrofibres to the seeding of meshes formed from nanofibres. Numerous groups have engineered hydrogel fibres using material composites and copolymers that allow for subsequent photo and ionic crosslinking, resulting in a wide spectrum of material properties.

3D printing

3D printing, or additive manufacturing, typically involves the building of 3D structures layer-by-layer with the controlled deposition of materials. A chief concern with 3D printing of cell-laden hydrogels is the optimisation of the printing parameters and control over material properties to enable efficient printing while preserving cell viability and phenotype. Schuurman *et al.* adapted a methacrylated gelatin formulation by introducing HA to increase the viscosity of the precursor material and demonstrated print fidelity and chondrocyte viability with post-print photocrosslinking (Schuurman *et al.*, 2013). Conceptually similar fabrication techniques have been explored with other methods, including the use of ionically crosslinkable materials in lieu of photocrosslinking. For example, Fedorovich and colleagues used chondrocyte-laden alginate crosslinked with calcium chloride. In this case, constructs were printed with varying nozzle diameter, print speed and fibre orientation and then cultured or implanted subcutaneously to demonstrate cell viability and tissue formation (Fedorovich *et al.*, 2012). Building upon this approach, Markstedt and co-workers engineered additional shear-thinning properties into this ionically crosslinkable network by incorporating nanofibrillar cellulose into the alginate precursors (Markstedt *et al.*, 2015). Cui *et al.* (2012) examined the timing of crosslinking in this context as well, demonstrating direct 3D printing into osteochondral defects using a PEG dimethacrylate hydrogel in an explant model. The authors found greater chondrocyte viability and more uniform cell distribution with tandem (simultaneous) crosslinking, where crosslinking occurs at the same time as gel extrusion, rather than with traditional post-print crosslinking in which the construct can be crosslinked after printing (Cui *et al.*, 2012).

Thermally responsive hydrogels use temperature to modulate their gelation behaviour, allowing for the transition from liquid to hydrogel, exclusively as a function of temperature. This is particularly appealing in cartilage tissue engineering, since the change from ambient temperature to physiological conditions can induce rapid gel formation (Klouda and Mikos, 2008) or stabilise materials in 3D printing processes. A common composite choice for thermoresponsive “bioinks” consists of HA or chondroitin sulphate mixed with a thermoresponsive polymer. Numerous groups have tested different thermoresponsive materials in this regard, from poly(N-isopropylacrylamide) (pNIPAAm) (Kesti *et al.*, 2015) to triblock copolymers composed of PEG linked to N-(2-hydroxypropyl) methacrylamide (HPMA) (Boere *et al.*, 2015).

In exploring possible effects of thermoresponsive material structures at the nanoscale, Muller *et al.* showed that nanostructuring a thermoresponsive hydrogel allowed for high chondrocyte viability in a printed multilayer mesh and other more physiologically-relevant constructs including an ear (Fig. 4e) and a sheep meniscus (Fig. 4f,g) (Muller *et al.*, 2015). In this case, the authors incorporated both unmodified and acrylated Pluronic, crosslinked and then subsequently washed out the unmodified fraction. In each case mentioned, methacrylated HA or chondroitin sulphate was either used as a crosslinker or

as an independent reinforcing network to compensate for the generally poor mechanics of the thermoresponsive component.

Spinning hydrogel fibres

A number of techniques that use spinning technology have been explored towards fabricating hydrogel fibres. A wet-spinning technique was introduced by Han and colleagues in which an 8-arm methacrylated PEG precursor was injected through a syringe pump into a tris(2-aminoethyl) amine (TAEA) bath (Han *et al.*, 2014). Subsequent crosslinking yielded microribbons that could be further crosslinked together to form hydrogels with complex geometries at both the micro and macro scales, and these hydrogels supported adipose-derived stem cell viability as well as proliferation and spreading under various conditions with the potential of recapitulating the laminar matrix organisation of cartilage tissue (Han *et al.*, 2014).

Electrospinning has also been used to generate fibrous constructs from methacrylated HA that supported human MSC interactions and chondrogenesis, without outcomes dependent on the hydrogel fibre properties such as mechanics and adhesion (Kim *et al.*, 2013). Multi-polymer fibrous scaffolds were fabricated using HA hydrogels and implanted in combination with microfracture for tissue repair in a mini-pig model. The multi-polymer design allowed for stable fibres to maintain the fibre structure and degradable fibres to release chondroinductive factors. Here, the transforming growth factor beta 3 (TGF- β 3) spun into degradable HA fibres was verified to be active upon release and *in vivo* it resulted in improved histological scores and type II collagen content in the repaired defects (Kim *et al.*, 2015). Fibre spinning and printing have also been combined in cell-laden constructs, where, for instance, alternating rounds of electrospinning of polycaprolactone (PCL) fibres and printing of fibrin-collagen hydrogels were performed to yield scaffolds with greater mechanical properties and improved histological staining of GAGs and type II collagen in an 8 week subcutaneous implant study in mice (Xu *et al.*, 2013).

Porous hydrogels

While hydrogel fibres and scaffolds comprised of fibrous networks permit the incorporation of pores of varying shapes and sizes, several other methods for generating porous hydrogel constructs within the context of cartilage tissue engineering also exist. These may be advantageous in some cases, since they do not need specialised printing or spinning equipment and may produce scaffolds more rapidly. More recently, the design of porous hydrogels has turned to dynamic control of macroporosity as well as dynamic control of bulk material properties.

Porogen approaches

Han *et al.* developed stimuli-responsive porogens from alginate, gelatin and HA that respond to chelation, temperature and enzymatic activity. The alginate porogen was also used to deliver chondrocytes into the hydrogel and treatment with changes in temperature, EDTA or hyaluronidase activity released chondrocytes from the alginate components and subsequently increased hydrogel

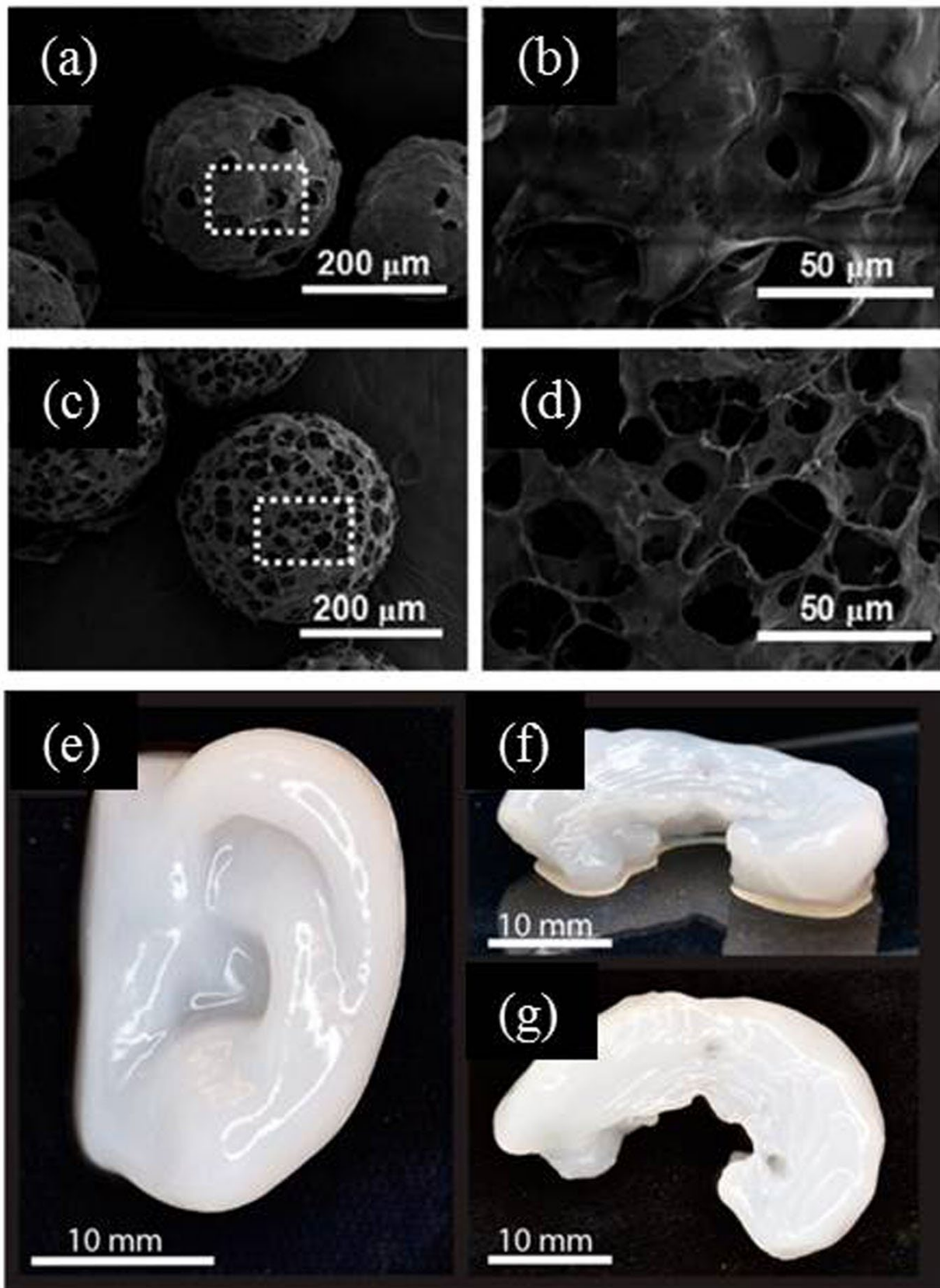


Fig. 4. SEM of PBLG microsphere hydrogels fabricated at a gelatin concentration of (a,b) 1.9 % and (c,d) 3.25 % (Fang *et al.*, 2015). 3D printing was used to print (e) a human ear and (f,g) a sheep meniscus, as seen from different angles with Ink8020 after crosslinking (Muller *et al.*, 2015).

macroporosity (Han *et al.*, 2013). While this controlled degree of macroporosity affected bulk mechanical properties, approaches have also been developed where mechanics can be dynamically increased with photocrosslinking, such as from 2.6 to 12.5 kPa, without changing porosity (Marklein *et al.*, 2012). In this case, across the range of mechanics tested, chondrogenic gene expression by human MSCs in porous constructs was slightly upregulated even in the absence of chondrogenic media (Marklein *et al.*, 2012). In a different approach, Ahrem *et al.* (2014) treated whole bacterial nanocellulose hydrogels with 3D laser perforation to produce a porous construct that promoted chondrocyte ingrowth and proliferation. Control of hydrogel porosity and functionalisation of pore interiors with relevant molecules such as type II collagen have also been explored in tandem with the use of stimuli-responsive materials. For example, Almeida and colleagues recently showed that “shape-memory” properties can be attained by covalent crosslinking of alginate scaffolds by making use of carbodiimide chemistry, which along with type II collagen incorporation led to higher sulphated GAG (sGAG) and collagen production in comparison to scaffolds functionalised with type I collagen (Almeida *et al.*, 2016).

Microsphere hydrogels

As a complement or alternative to porous bulk hydrogels, scaffolds comprised of microspheres from a range of materials have also been developed. Such scaffolds have been studied most often as delivery vehicles or depots for small molecules or proteins, but they have also been used as potential cell carriers (*i.e.* cell microencapsulations). In one case, a xanthan gum derivative was selected as the encapsulation material because of its established safety record in pharmaceuticals, food and cosmetics, as well as its observed protective effects on joint cartilage – administered *via* an intra-articular injection (Hamcerencu *et al.*, 2007; Katzbauer, 1998). Murine chondrocytes were viable in these microgels for up to 21 d (Mendes *et al.*, 2012). Another interesting scaffold, comprised of the synthetic peptide poly(g-benzyl-L-glutamate) (PBLG), achieved tuneable porosity in microspheres by varying the amount of gelatin porogen followed by its removal, producing spherical hydrogel scaffolds that could be seeded with chondrocytes throughout their interior (Fig. 4a-d). *In vitro*, they retained chondrocyte roundness after seeding and supported ECM deposition in culture for 3 to 7 d and, when injected subcutaneously *in vivo*, they formed tissue with improved histological scoring and type II collagen content compared to cell-only injections (Fang *et al.*, 2015).

Multilayered scaffolds

The final scale of architectural complexity in cartilage engineering is one that attempts to recapitulate the zonal architecture of cartilage in the joint. Regardless of the particular technique used to deposit or arrange the material, numerous groups have worked to establish two-to-three-layered hydrogel constructs with differential characteristics to mimic the zones likely damaged by osteochondral defects. Feasibility of a 3D printed bilayer hydrogel construct has been demonstrated, both by 3D extrusion of material (Cui *et al.*, 2012) and by projection

stereolithography (Sun *et al.*, 2015). As for tri-layer hydrogels, Nguyen and co-workers described a PEG-based hydrogel with the superficial layer containing chondroitin sulphate and a matrix metalloproteinase (MMP)-sensitive peptide, the middle layer containing PEG and chondroitin sulphate and the deep layer composed of PEG and HA. In such a construct the investigators showed that with a single stem cell population evenly distributed in all layers, type II collagen deposition decreased gradually from superficial to deep layers along with an increase in type X collagen and proteoglycans, resulting in a gradient of compressive modulus across the construct (Nguyen *et al.*, 2011).

However, there are a number of different approaches to zonal design of hydrogel constructs, with alternatives including the use of different cell types (*e.g.* chondrocytes, osteogenic progenitors) (Fedorovich *et al.*, 2012) or culture conditions in each layer rather than substantially modifying the material characteristics of the layer itself. It remains to be seen how advances in the design of hydrogels with this degree of control over architecture will compare with other diverse strategies for hydrogel engineering towards cartilage repair.

Controlled presentation and delivery of biochemical factors

Cartilage formation is regulated by the dynamic spatial and temporal presentation of an assortment of biochemical factors that either interact with receptors at the cell surface or are internalised by cells, resulting in the induction of chondrogenesis. Thus, hydrogels are being designed where biomolecules are either directly tethered to the hydrogel or are encapsulated within the hydrogel for presentation to entrapped or surrounding cells. This control over biomolecules' spatial introduction and timing can aid in the production of heterogeneous constructs to mimic developing or adult tissues and to enhance the dynamic processes of differentiation.

The spatial and temporal presentation of biochemical factors is extremely important, since uncontrolled exposure may not only reduce effects on chondrogenesis and cartilage formation, but can also induce adverse effects. For example, supra-physiological levels of TGF- β delivered *via* injections to murine knee joints resulted in a decrease in proteoglycan synthesis, synovial fibrosis and endochondral ossification (Bakker *et al.*, 2001; van Beuningen *et al.*, 2000). To regulate the dose and timing of biochemical factors that interact with cell surface receptors, several techniques such as protease-degradable tethers and presenting factors in their latent form (Place *et al.*, 2012; Re'em *et al.*, 2012) have been recently proposed. To achieve controlled release of cartilage-inducing factors that are internalised by cells, novel design techniques towards the improved design of carrier vehicles [*e.g.* microparticles (MPs) or nanoparticles (NPs)] loaded with biochemical factors have been proposed. In this section we will highlight several recent advances in biochemical factor presentation using these mechanisms and their *in vivo* and *in vitro* cartilage regeneration potential using cell-laden and acellular hydrogels (Fig. 5).

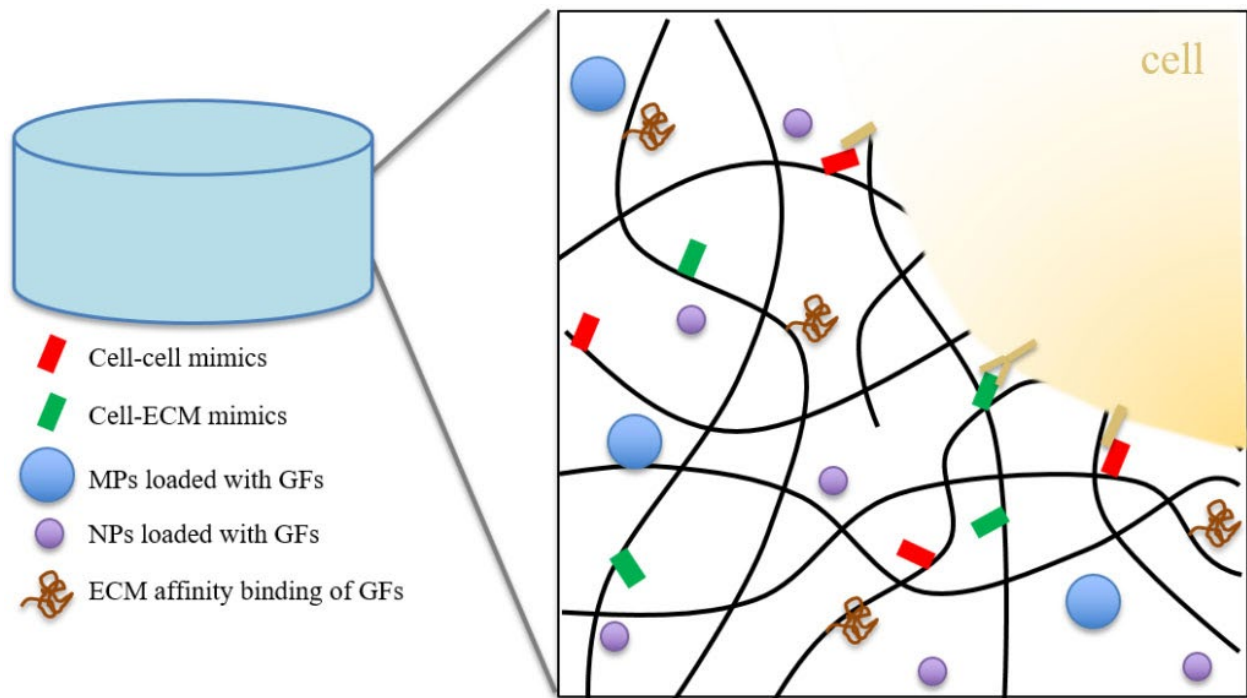


Fig. 5. Overview of controlled presentation of biochemical factors. These include cell-cell and cell-matrix interactions, as well as growth factors and other molecules that can be either tethered to the hydrogels *by* affinity or heparin binding, or encapsulated in MPs and NPs for controlled release.

Covalent conjugation and affinity binding

In order to present cells with biochemical cues that interact with integrins or receptors at the cell surface, these signals are typically incorporated into the hydrogel itself, either by covalent tethering or affinity binding by electrostatic interactions as is observed within the ECM. Chemical coupling of peptides to hydrogels is one of the simplest approaches that can be taken to modify hydrogels with biochemical signals.

During limb development, the first instance of cartilage formation occurs within a highly condensed cell-rich aggregate that is largely devoid of ECM (Bobick *et al.*, 2009). Cell-cell interactions in this context are mediated by the adhesion molecule N-cadherin (DeLise and Tuan, 2002a; DeLise and Tuan, 2002b; Oberlender and Tuan, 1994) and these interactions regulate signalling events that are critical to the initiation of chondrogenesis. Of note, these cell-cell adhesion signals are not present throughout development, but rather arise and peak during this condensation phase, after which adhesion-based signalling from the formed ECM dominates (Singh and Schwarzbauer, 2012). To this end, Bian *et al.* recently developed an HA hydrogel system that incorporates an N-cadherin mimetic peptide (*i.e.* HAV) (Fig. 6a) and found that the peptide conjugation promoted both early chondrogenesis of human MSCs, as well as cartilage-specific matrix production (Fig. 6b) (Bian *et al.*, 2013). This outcome was even observed *in vivo*, and it was shown recently that the presence of the N-cadherin signal from the hydrogel also influenced β -catenin signalling in human MSCs (Fig. 6c,d) (Vega *et al.*, 2016).

Due to the importance of the temporal presentation of biochemical signals, there has been increased interest in

regulating the presentation of these biochemical cues by enzymatic regulation. Towards this, Salinas and Anseth used thiol-acrylate photopolymerisation to tether RGD peptides with an MMP-13 cleavable linker in order to induce MSCs towards a chondrogenic phenotype. The authors found that MSCs encapsulated in these hydrogels produced active MMP-13 between 9 and 14 d in culture, resulting in an over ten-fold increase of GAG deposition in comparison to MSCs exposed to non-cleavable RGD peptides (Salinas and Anseth, 2008).

Besides peptides, growth factors, such as those from the TGF- β superfamily, have also been tethered to hydrogels. Although TGF- β plays a key role in promoting chondrogenesis, its therapeutic utility is limited by its inherent protein instability, requiring high amounts of protein that can cause adverse side effects with inefficient cartilage formation. Choi and co-workers compared hydrogels containing TGF- β 1 covalently tethered to visible blue light-inducible chitosan (MeGC) hydrogels prior to photopolymerisation with type II collagen *versus* MeGC hydrogels with positively-charged TGF- β 1 ionically conjugated to type II collagen. The authors found that the cumulative release of TGF- β 1 was significantly higher in the covalently bound system, resulting in more sGAG production by adipose-derived stem cells *in vitro*, as well as improved integration with surrounding tissue of hydrogels implanted in a rat cartilage defect model (Choi *et al.*, 2015).

In vivo, control over the bioavailability of biomolecules is dependent on numerous factors. For example, growth factors including TGF- β are present as a complex including a latency associated peptide (LAP), which inactivates TGF- β by masking the receptor-binding domains (Shi *et al.*, 2011). As such, members of the TGF- β superfamily

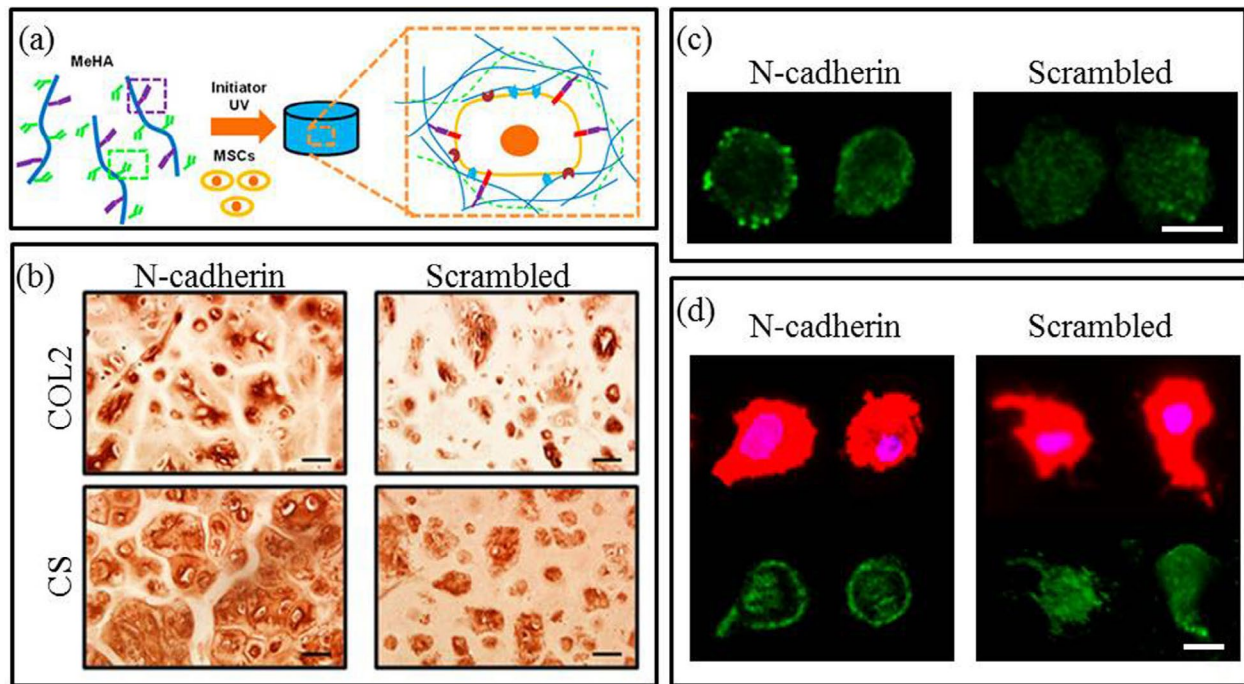


Fig. 6. Human MSCs were (a) photoencapsulated in hydrogels containing either N-cadherin mimics or scrambled sequence controls. (b) After 4 weeks of *in vitro* culture, N-cadherin mimics enhanced chondroitin sulphate (CS) and type II collagen (COL2) production by human MSCs, as seen by immunohistochemical staining (Bian *et al.*, 2013). (c,d) Single cell analysis of MSCs in these hydrogel environments showed an increase in N-cadherin mediated β -catenin signalling after 3 d in culture. (c) Cross-sectional images of MSCs stained for β -catenin (green) show that N-cadherin mimics recruit β -catenin to the cell membrane. Additionally, N-cadherin mimics induced an increase in nuclear β -catenin, as confirmed by (d) representative maximum (top) and average (bottom) projections of single MSCs stained for actin (red), nucleus (blue) and β -catenin (green) (Vega *et al.*, 2016). Scale bars: **b** = 50 μ m; **c,d** = 5 μ m.

remain inactive and protected from degradation until LAP is removed by one of several mechanisms including proteolytic cleavage of LAP. Place *et al.* sought to recapitulate this natural mechanism by coupling TGF- β 1 in its latent form to PEGDA prior to crosslinking with thiolated HA hydrogels by a Michael addition reaction (Place *et al.*, 2012). The authors found that encapsulated chondrocytes in these hydrogels were viable for up to 34 d and produced type II collagen without the presence of soluble TGF- β 1, as confirmed by immunocytochemistry (Place *et al.*, 2012).

In addition to covalent conjugation, affinity binding of macromolecules to growth factors is an alternative method towards the controlled release of growth factors. To this end, dendrimers, a particular type of branched macromolecules, can be functionalised with peptides that selectively bind to different molecules (Simon-Gracia *et al.*, 2013). Seelbach and colleagues recently investigated the use of dendrimers with affinity binding peptides towards the non-burst release of bone morphogenetic protein 2 (BMP-2) and TGF- β 1. Briefly, dendrimers containing four binding peptides presenting sequences targeting BMP-2 or TGF- β 1 protein binding were covalently conjugated to HA, mixed with poly(N-isopropylacrylamide)-grafted HA (Hyal-pN) brush copolymers, and then loaded with BMP-2 or TGF- β 1. The authors found a dependence of the release of these growth factors based on the length of the dendrimer arms (Seelbach *et al.*, 2015).

GAGs are highly negatively-charged polysaccharides that can modulate macromolecular binding, particularly with positively-charged molecules (Lindahl and Hook, 1978). By regulating the degree of GAG sulphation, the extent of this charge can be tuned to control the presentation of positively-charged growth factors (*e.g.* TGF- β 1). To this extent, Lim and Temenoff showed that MSCs cultured in TGF- β 1 containing media and encapsulated in desulphated chondroitin hydrogels featured significantly upregulated gene expression of type II collagen and aggrecan when compared to a PEG-based control hydrogel (Lim and Temenoff, 2013).

Particles for controlled release

Carrier vehicles such as MPs and NPs as a method to deliver encapsulated factors have been extensively studied and optimised to achieve continuous delivery of their payload by controlling the chemical (*e.g.* charge, degradability) and physical (*e.g.* carrier size, shell thickness) properties of the carriers. In contrast to simple diffusion of soluble cues into hydrogel constructs, loaded MPs encapsulated in hydrogels protect biochemical factors from degradation and help prevent rapid diffusion and clearance from hydrogels, allowing for greater control over the release kinetics of biochemical cues into the hydrogel. Although this technique has been used and investigated for decades, the utility of MPs (Reyes *et al.*, 2013a; Reyes *et al.*, 2013b) and NPs (Ertan *et al.*, 2013; Lim *et al.*, 2011; Siu *et al.*,

Table 1. Summary of hydrogel fabrication techniques for cartilage tissue engineering.

Improvements in hydrogel structure		Advances in the processing of hydrogel scaffolds	
IPNs	(Dinescu <i>et al.</i> , 2015; Guo <i>et al.</i> , 2012; Ingavle <i>et al.</i> , 2013; Snyder <i>et al.</i> , 2014)	3D printing	(Fedorovich <i>et al.</i> , 2012; Kesti <i>et al.</i> , 2015; Markstedt <i>et al.</i> , 2015; Muller <i>et al.</i> , 2015; Schuurman <i>et al.</i> , 2013)
Semi-IPNs	(Little <i>et al.</i> , 2014; Park <i>et al.</i> , 2013; Skaalure <i>et al.</i> , 2014)	Spinning	(Han <i>et al.</i> , 2014; Xu <i>et al.</i> , 2013)
Double networks	(Arnold <i>et al.</i> , 2011; Kitamura <i>et al.</i> , 2016)	Porogens	(Ahrem <i>et al.</i> , 2014; Han <i>et al.</i> , 2013; Marklein <i>et al.</i> , 2012)
Dual networks	(Jin <i>et al.</i> , 2011; Levett <i>et al.</i> , 2014; Moreira Teixeira <i>et al.</i> , 2012; Palumbo <i>et al.</i> , 2015; Pirinen <i>et al.</i> , 2015)	Microspheres	(Fang <i>et al.</i> , 2015; Mendes <i>et al.</i> , 2012)
Guest-host networks	(Jung <i>et al.</i> , 2014; Wei <i>et al.</i> , 2016)	Multilayered	(Cui <i>et al.</i> , 2012; Fedorovich <i>et al.</i> , 2012; Nguyen <i>et al.</i> , 2011; Sun <i>et al.</i> , 2015)
Controlled presentation and delivery of biochemical factors			
Mimetic peptides	(Bian <i>et al.</i> , 2013; Salinas and Anseth, 2008; Vega <i>et al.</i> , 2016)		
MPs/NPs	(Ahearne and Kelly, 2013; Siu <i>et al.</i> , 2012; Spiller <i>et al.</i> , 2012)		
ECM affinity binding	(Choi <i>et al.</i> , 2015; Seelbach <i>et al.</i> , 2015)		

2012; Whitmire *et al.*, 2012) has seen recent advances in their use as delivery vehicles for cartilage repair.

To achieve a more sustained delivery of growth factors, Spiller and co-workers recently designed a hybrid scaffold where insulin-like growth factor 1 (IGF-1) was loaded into a poly(vinyl alcohol) (PVA) and poly(lactic-co-glycolic acid) (PLGA) hydrogel system. The authors used a novel double emulsion technique, such that PLGA MPs ($11.3 \pm 6.4 \mu\text{m}$) containing IGF-1 directly formed and evenly dispersed throughout the PVA hydrogel. IGF-1 release was linear and sustained for at least 45 d, and *in vivo* studies showed that the hydrogel resulted in the formation of thick cartilage layers and exhibited good integration between the formed cartilage and the surrounding neocartilage (Spiller *et al.*, 2012).

In addition to IGF-1, there are other biochemical cues that enhance cartilage formation and can benefit from a suitable delivery vehicle for their sustained release. Nell-like molecule 1 (Nell-1) has previously been shown to induce differentiation and growth towards bone and cartilage tissue *in vivo* (Siu *et al.*, 2011). In order to showcase the importance of sustained release of Nell-1, Siu *et al.* compared the release of Nell-1 from alginate hydrogels either directly encapsulated in the bulk hydrogel, or loaded into chitosan NPs (100-300 nm in size). The authors found that Nell-1 containing NPs induced a significantly more gradual release of the protein in comparison to hydrogels where Nell-1 was directly encapsulated. Additionally, Nell-1 NP containing hydrogels induced an increase in GAG synthesis and proteoglycan accumulation in a critical size osteochondral defect 12 weeks post-implantation (Siu *et al.*, 2012). This study showed not only the importance of Nell-1 in cartilage formation, but also the need for NPs for controlled release.

Although design parameters used to control the release kinetics of biochemical factors loaded in MPs and NPs have been thoroughly investigated, not much work has been done to study the influence of the encapsulating hydrogel on the release profiles of biochemical cues loaded

in MPs and NPs. To this end, Ahearne and co-workers studied how different hydrogel macromers (*i.e.* fibrin, agarose and gellan gum) affected the efficiency of TGF- β 3 loaded MPs to induce chondrogenesis of encapsulated MSCs. Piglet MSCs and TGF- β 3 loaded gelatin MPs (50-70 μm) were encapsulated in fibrin, agarose or gellan gum hydrogels, and the role of the MPs in inducing cartilage formation from the MSCs over the course of 21 d *in vitro* was evaluated. The authors observed a significant increase in sGAG accumulation in the agarose and gellan gum hydrogels in comparison to the fibrin hydrogels. Additionally, histological staining with alcian blue showed a stronger and more homogeneous distribution of sGAG in the gellan gum hydrogels, in contrast to a more pericellular presentation of sGAG in the agarose hydrogels (Ahearne and Kelly, 2013). This study shows that not just MPs and NPs, but the hydrogel macromers as well, are important determinants of the release profiles of biochemical factors loaded in these vehicle carriers.

Summary and future directions

Although hyaline cartilage is perceived as a simple tissue, developing biomaterials that can achieve mechanical properties of native cartilage and complete integration with surrounding tissues remains a challenge. Cartilage tissue engineering has seen a rapid advance in fabrication techniques, resulting in hydrogel constructs that are improving the quality of produced cartilage (summarised in Table 1). To increase the mechanical properties of hydrogels, traditional single network hydrogels are being supplemented with either additional networks or mixtures of polymers. These techniques have also been used to fabricate hydrogels which can induce integration with surrounding tissue while promoting chondrogenesis *in vivo*. Novel processing techniques have also been employed towards fabricating hydrogel fibres and porous hydrogels for improved cartilage formation, as well as

the fabrication of multilayered hydrogels that mimic the zonal architecture of native cartilage. In addition to these advances, the field has also seen an increased interest in presenting biochemical cues in a controllable and temporal fashion.

Looking ahead, combinatorial screening approaches to robustly identify synergies between the various hydrogel parameters presented in this review would enable the development of cell-laden and acellular hydrogels that will eventually adequately recapitulate native cartilage. The technology to achieve this kind of screening has been explored for the discovery of hydrogels with a variety of properties, such as cell adhesion, proliferation and foreign body responses (Le *et al.*, 2016; Vegas *et al.*, 2016; Zant and Grijpma, 2016a; Zant and Grijpma, 2016b). So far this has been achieved by changing the extent of material modification, the combinations of different materials at different ratios or the amount of a peptide of interest in the precursor. Although none of these combinatorial hydrogel platforms have been directly applied to probe chondrogenesis or cartilage matrix synthesis, they are extremely promising for this application. Additionally, a better understanding of cartilage both during development and in its mature state will be critical to the success of advancing these exciting techniques towards developing hydrogel systems that robustly recapitulate native cartilage.

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Discussion with Reviewers

Dirk Grijpma: Would other polymer systems, such as porous rubber-like materials be suited for cartilage tissue engineering? If not, why would the use of hydrogels be preferred?

Authors: Porous rubber-like materials do present their share of interesting properties and potential advantages to this application, including mechanical properties and resistance to load. However, they are limited in their use for the encapsulation of cells; additionally, injection of such macroporous constructs is difficult, limiting non-invasive implantation techniques. Hydrogels can be used for the encapsulation of cells with uniform distribution and can be injected into irregular defects. Hydrogels also permit nutrient and soluble factor diffusion and in developing tough systems are able to attain mechanical properties comparable to rubber-like materials.

Dirk Grijpma: What would the ideal degradation behaviour of cartilage tissue engineering scaffolds be? And how could that be achieved?

Authors: In principle, the degradation of scaffolds should match or correlate with the formation of matrix by encapsulated or infiltrating cells in the material, which is arguably determined by the differentiation state or phenotype of said cell population. Thus, it is difficult to state a specific timeframe to permit and not inhibit matrix elaboration. Degradation can be tuned through numerous means, including the introduction of hydrolytically degradable crosslinkers or through crosslinkers that degrade in the presence of cell-produced proteases.

Matteo D'Este: Can the authors envisage double network hydrogels based on degradable and cell-instructive (bio) polymers? Could these biodegradable double networks be featured by high toughness but at the same time being capable of cell encapsulation or attraction from surrounding tissues and be used for cartilage regeneration rather than replacement?

Authors: These are great considerations towards the design of hydrogel-based repair for cartilage. Indeed,

double networks permit tougher hydrogels, and this, in tandem with cell-instructive biomimetic polymers, could promote cartilage regeneration. Some recent works from Jos Malda's and Travis Klein's laboratories did just this, and this has been included in this review. Other work on tough double networks using cell-instructive biopolymers includes a study from Rodell *et al.* (2016) with hyaluronic acid as the base material, where the material can be both injected and used for cell encapsulation, but to date this has not yet been applied specifically to cartilage tissue engineering.

Dimitrios Zeugolis: Is there a clear hydrogel formulation/cargo that "leads the race"?

Authors: While we would be interested in identifying a hydrogel formulation that is the best, it is quite difficult

to make a specific claim due to the wide variety of experimental parameters explored within the various reports, such as the time points selected or the outcomes measured. While a few studies have compared different hydrogels, none to date are broad or controlled enough to enable us to confidently argue for a specific formulation. Also, only select hydrogels have progressed to clinically relevant animal models and success may depend on the injury model chosen. Furthermore, the best hydrogel will depend on the specific clinical scenario (*e.g.* patient age, size of defect for repair) and treatment parameters, such as whether cells are delivered or recruited to the tissue site.

Editor's note: The Scientific Editor responsible for this paper was Mauro Alini.