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Abstract

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Keywords

wear, friction, nano-tio2, containing, temperatures, nanolubricant, elevated, water-based, effect, ambient, steel, chrome, behaviour

Disciplines

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Effect of water-based nanolubricant containing nano-TiO₂ on friction and wear behaviour of chrome steel at ambient and elevated temperatures

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Abstract: A novel experimental method was developed through a ball-on-disk tribometer to study the friction and wear behaviour of a chrome steel at temperatures of 25, 200 and 500 °C. Water-based nanolubricants containing different concentrations of nano-TiO₂ from 0.4 to 8.0 wt% were used to investigate their effects on friction-reduction and anti-wear mechanisms, in comparison to the benchmarks under dry and water conditions. The results show that the water-based nanolubricants can significantly reduce coefficient of friction (COF) and improve wear resistance of the chrome steel at both ambient and elevated temperatures. In particular, the use of nanolubricant containing an optimal concentration (4.0 wt%) of nano-TiO₂ leads to the lowest COF and the smallest ball wear among all the lubrication conditions. The friction-reduction and anti-wear mechanisms are ascribed to the rolling & polishing effects, semisolid film and solid layer contributed by the nano-TiO₂ at temperatures of 25, 200 and 500 °C, respectively.

Keywords: Nano-TiO₂; Water-based nanolubricant; Coefficient of friction; Elevated temperature

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1. Introduction

Energy saving and environmental protection are being subjected to considerable concern in modern times [1]. Recently, great efforts have been made to minimise energy loss and environmental pollution in engineering applications [2-5]. Lubrication technology is one of the most useful approaches to solve these issues as the primary causes of energy loss in engineering fields are friction and wear [6]. It is welldocumented that the conventional lubricating system including base oils such as mineral oil [7], synthetic oil [8] and biological oil [9], oil-in-water emulsions [10, 11] and other oil-based lubricants [12-14] can decrease friction and wear to a certain extent. The application and discharge of oil, however, inevitably trigger environmental issues due to its nonbiodegradable nature and inherent toxicity [7]. In this regard, developing environment-friendly water-based lubricants with excellent tribological performance is increasingly important and urgently needed. Considering the insufficient film thickness of water due to its low viscosity and chemical reaction with metals [3], many researchers have focused on using nanomaterials as nanoadditives dispersed in water in order to enhance the lubricating property of water. These nanoadditives include metals [15, 16], metallic oxides [17-25], non-metallic oxides [26, 27], metal sulphides [28-30], composites [31, 32], fullerene [33, 34] and other carbon materials [35-43]. For examples, Zhang et al. [16] prepared water-based nanolubricant by adding surface-capped nano-Cu into distilled water for four-ball test. They found that the nano-Cu with an average diameter of 2 nm could be uniformly dispersed in distilled water without apparent agglomeration. The improved tribological properties of distilled water were ascribed to the formation of boundary lubrication film produced by tribo-chemical reaction between the nano-Cu and sliding steels. Radice and Mischler [17] prepared water-based nanolubricants by adding 10 vol% nano-Al₂O₃ into deionised water with an acetate buffer solution. The reciprocating wear testing results showed that the lubricant containing nano-Al₂O₃ decreased COF by 40-50%, compared to that without nanoparticles (NPs) due to a third body formed by NPs trapped within the contact area. Bao et al. [27] added surface modified nano-SiO₂ into water to prepare well-dispersed waterbased lubricants. The tribological properties evaluated by four-ball and pin-on-disk tests demonstrated that the lubricant containing 0.3 wt% nano-SiO₂ caused the best load-carrying and anti-wear performance due to the miniature ball bearing effect and the self-repairing ability of SiO₂ NPs between the friction pairs. Meng et al. [30] prepared water-dispersible MoS_2 NPs, and investigated tribological behaviour of the synthetic nano-MoS₂ dispersed in distilled water using a four-ball tester. The results indicated that nano-MoS₂ can effectively improve the friction-reduction, anti-wear and load-carrying abilities of distilled water, showing a promising potential as a nanoadditive in water-based lubricant. Zhang et al. [31] synthesised Cu/SiO₂ nanocomposite by use of a sol-gel method, and studied its tribological properties when added into distilled water via a four-ball tester. The results showed that the water-based lubricant containing nanoCu/SiO₂ was able to significantly reduce the friction and wear of steel friction pairs, compared to distilled water. This was ascribed to the formation of a protective film composed of Cu and SiO₂. Liang et al. [38] prepared aqueous graphene solution with the assistance of a non-ionic surfactant i.e. Triton X-100. The results obtained from ball-on-disk tribological tests showed that a small amount addition of graphene enabled a substantial reduction of COF by 81.3% and also the wear volume by two orders of magnitude in comparison with deionised water. In particular, research efforts have been directed towards using TiO₂ NPs as additives that can be dispersed uniformly in water, showing promising and inspiring lubrication performance [18, 23-25].

The previous studies regarding tribological properties of water-based nanolubricants, however, were mainly conducted at the ambient temperature. The evaluation of tribological properties using water-based nanolubricants at elevated temperatures has rarely been performed or reported until now. In our previous studies [23, 24], the lubrication mechanisms of water-based nanolubricant containing TiO₂ NPs were systematically investigated via ball-on-disk tests at ambient temperature by varying surface conditions (surface roughness and oxidation) of applied disk and concentrations of nano-TiO₂ dispersed in water. The objective of this study is to reveal the effects of temperature and nano-TiO₂ concentration on friction and wear behaviour of a chrome steel (roll material) with water-based nanolubricant, using an innovative ball-on-disk configuration. This study will provide a reliable and effective way to evaluate the roll service life in hot steel rolling when using water-based nanolubricant.

2. Experimental details

2.1 Materials

An E52100 chrome steel ball and a mild steel (MS) disk were used as the friction pair in a ball-on-disk tribometer, in which the ball represented the roll material while the disk represented the strip steel. The chemical compositions of the two materials are listed in Table 1. All the balls being used had a diameter of 9.5 mm and a surface roughness of $0.02 \,\mu\text{m}$ in R_a. The disks were machined to a dimension of Ø40 mm × 8 mm with a surface roughness of $0.11 \,\mu\text{m}$ in R_a. The hardnesses of the ball and disk materials are approximately 780 and 90 HV, respectively. Surface morphologies and 3D profiles of the friction pair are shown in Fig. 1. It can be seen that the ball surface is quite smooth, while the disk has significant scratches on its surface. This rough surface was designed to differentiate the relevantly smooth one based on the previous study [24].

Materials	С	Si	Mn	Cr	Cu	Мо	Nb+V+Ti	P+S
Ball-E52100	1.0	0.25	0.35	1.5	0.30	0.10	-	≤0.03
Disk-MS	0.05	0.02	0.25	0.01	-	-	< 0.01	≤0.03

Table 1 Chemical compositions of the ball and disk materials (wt%).



Fig. 1 Surface morphologies and 3D profiles of (a, b) Cr steel ball, and (c, d) mild steel disk.

The water-based nanolubricants being used in the test consist of TiO_2 NPs, polyethyleneimine (PEI), glycerol and balanced water. The nano- TiO_2 is P25, a mixture of 75% of anatase and 25% of rutile with around 20 nm in diameter. It is worth noticing that the reason of choosing P25 as the nanoadditive in water is due to its much lower cost than those of pure anatase and rutile.

The synthesis procedure of nanolubricants can be found elsewhere, showing excellent dispersibility and stability [4, 24]. The chemical compositions of the nanolubricants being used in this study are shown in Table 2. For comparison, testing conditions without lubrication (i.e. dry condition) and with pure water lubrication were also used.

Lubrication type	Description					
1	Dry					
2	Water					
3	0.4 wt% TiO ₂ +0.004 wt% PEI + 10.0 vol% glycerol + balance water					
4	1.0 wt% TiO ₂ +0.01 wt% PEI + 10.0 vol% glycerol + balance water					
5	$2.0 \text{ wt\% TiO}_2+0.02 \text{ wt\% PEI} + 10.0 \text{ vol\% glycerol} + \text{balance water}$					
6	4.0 wt% TiO ₂ +0.04 wt% PEI + 10.0 vol% glycerol + balance water					
7	8.0 wt% TiO ₂ +0.08 wt% PEI + 10.0 vol% glycerol + balance water					

Table 2 Chemical compositions of employed lubricants.

2.2 Tribological tests

An Rtec MFT-5000 Muti-functional Tribometer was used to measure the coefficients of friction (COF) under the ball-on-disk testing protocols. The configuration of this tribometer for the tests at ambient temperature was the same to that reported in our previous study [23], in which the disk surface was covered by a layer of lubricant with a fixed volume of 2 ml prior to each tribological test under liquid lubrication. In this way, the initial conditions of the tribological tests can be well controlled. In contrast, the configuration was restructured with an innovative model to continuously supply the droplets of lubricants during sliding process, as shown in Fig. 2. It can be seen that the lubricants were contained in an infusion bag, and flowed along a plastic pipe that was connected to a copper pipe, and then dropped consecutively onto a rotating pre-heated disk. The copper pipe was bound with a ball holder to resist a high temperature close to the hot disk. The temperature of the disk can be well controlled by the induction heater ahead of each tribological test, and the flowing rate of the lubricants can be adjusted by a flow regulator equipped.



Fig. 2 Schematic of the ball-on-disk tribometer used for tribological tests at elevated temperatures.

The tribological testing conditions at ambient and elevated temperatures are shown in Table 3. A constant load of 50 N was applied to press the Cr steel ball against the rotating MS disk for a period of 5 minutes. The reason of choosing a relatively short sliding time was to decrease the stacking of nanolubricants caused by continuous droplets, which might impede movement of the ball. The linear speed and the radius of the wear track were 50 mm/s and 14 mm, respectively. It should be noted that a relatively low sliding speed of 50 mm/s was adopted for the minimisation of hydrodynamic effect. During testing, the time histories of COF were recorded. When a test was completed, both the ball and the disk were disassembled from the tribometer and then cleaned ultrasonically in an ethanol bath for 5 minutes to remove any loose debris and surplus lubricants. For each condition, the same test was repeated for three times.

 Table 3 Tribological testing conditions at ambient and elevated temperatures.

Normal load/N	Speed/mm·s ⁻¹	Time/min	Temperature/°C	Radius of wear track/mm
50	50	5	25, 200 and 500	14

2.3 Characterisation techniques

Wear scars of the balls produced after tribological tests were observed under a KEYENCE VK-X100K 3D Laser Scanning Microscope. Surface morphologies and 3D profiles of the worn zones were examined. Wear tracks of the disks were further examined using a JEOL model JSM-7001F Scanning Electron Microscope (SEM) equipped with an Energy Dispersive Spectrometer (EDS).

3. Results

3.1 Ball-on-disk tests at ambient temperature

Fig. 3 shows the COF curves and values measured at ambient temperature under different lubrication conditions. Fig. 3(a) shows that the COF curve caused by dry condition presents large fluctuation in the running-in period, and gradually drops to a stable level for the rest of the sliding. When using water lubrication, the COF is reduced to a lower and more stable level after a minor fluctuation in the first 10 seconds. The use of water-based nanolubricants further decreases the COF to a much lower level, even though there is also a slight fluctuation in the first 15 seconds. The COF values obtained at the stable stage of sliding are averaged from the three-time repeated tests, as shown in Fig. 3(b). It is found that the averaged COF value is significantly reduced from the dry and pure water lubrication of around 0.45 when the nanolubricants are applied. The increase of nano-TiO₂ concentration in water only slightly reduces the COF with the lowest value of 0.273 achieved when 4.0 wt% TiO₂ is used. A further increase of nano-TiO₂ concentration to 8.0 wt%, however, results in a slightly higher COF value. Therefore, a concentration of

4.0 wt% appears to be the optimal one in the water-based nanolubricants, which enables the reduction of COF by around 40% from that under the dry condition.



Fig. 3 (a) COF curves as functions of sliding time, and (b) averaged COF values at the stable sliding stage for all the testing conditions at ambient temperature.

Fig. 4 shows the surface morphologies of the worn balls obtained at ambient temperature under different lubrication conditions. It is observed from Fig. 4(a) and (b) that both dry condition and pure water produce relatively large wear scars with significant scratches, in comparison to those generated by nanolubricants. The brown substances (arrow) on the ball surface are likely iron oxides, which should be the debris from the worn disk. In contrast, the wear scars are much smaller and smoother when lubricated by the water-based nanolubricants, suggesting enhanced anti-wear ability. The boundaries of the wear scars are much cleaner with less adhered impurities too.



Fig. 4 Surface morphologies of the worn balls obtained at ambient temperature under lubrication conditions of (a) dry, (b) water, (c) 0.4 wt% TiO₂, (d) 1.0 wt% TiO₂, (e) 4.0 wt% TiO₂, and (f) 8.0 wt% TiO₂.

Fig. 5 shows the 3D images and corresponding cross-sectional profiles of the ball wear areas obtained at ambient temperature under different lubrication conditions, whose surface morphologies can be found in Fig. 4. It is observed in Fig. 5(a) and (b) that dry condition brings forth a relatively deep wear scar with a wear area of 2128 μ m² along Z axis (depth), while pure water yields a shallower one with a smaller wear area of 1242 μ m². Apparently, the tests under both dry and water conditions lead to considerable ball wear. The use of nanolubricants results in significantly shallower wear scar with much smaller wear areas than those under dry and water lubrication. In particular, using the nanolubricant containing 4.0 wt% TiO₂ produces the smallest ball wear area of 372 μ m² (see Fig. 5(e)), which decreases the ball wear by 82.5% in comparison to the dry condition. Nevertheless, the use of a higher concentration of 8.0 wt% TiO₂ appears to have slightly aggravated the ball wear (see Fig. 5(f)).



Fig. 5 3D images and corresponding cross-sectional profiles of the ball wear areas obtained at ambient temperature under lubrication conditions of (a) dry, (b) water, (c) 0.4 wt\% TiO_2 , (d) 1.0 wt\% TiO_2 , (e) 4.0 wt\% TiO_2 , and (f) 8.0 wt\% TiO_2 .



Fig. 6 Wear areas of the balls obtained at ambient temperature under different lubrication conditions.

3.2 Ball-on-disk tests at 200 °C

Fig. 7 shows the COF curves and values measured at 200 °C under different lubrication conditions. It can be seen from Fig. 7(a) that general running-in periods last a bit longer, and overall fluctuations of COF curves are more significant, compared to those in Fig. 3(a). Similarly, dry condition presents the highest COF level, followed by that of water lubrication. The COF curve caused by the nanolubricant containing 0.4 wt% nano-TiO₂ fluctuates severely in the first 160 seconds, and then keeps stable until the end. With the increase of nano-TiO₂ concentration in the lubricant, the period with curve fluctuation is increasingly shortened. In order to differentiate the levels of COF curves in Fig. 7(a), averaged COF values extracted from stable stages are shown in Fig. 7(b). It is found that the variation trend of COF values is similar to that in Fig. 3(b), showing an optimal concentration (4.0 wt%) of nano-TiO₂. It is of great interest that the nanolubricants contribute to much lower COF values at 200 °C than those obtained at ambient temperature, when the nano-TiO₂ concentration is higher than 1.0 wt%. For the same optimal nano-TiO₂ concentration of 4.0 wt%, the COF can be reduced by 28.6% from ambient temperature to the elevated temperature of 200 °C. The COF entailed by dry condition at 200 °C is thus reduced by 58.9% to the largest extent when using the nanolubricant containing 4.0 wt% nano-TiO₂.



Fig. 7 (a) COF curves as functions of sliding time, and (b) averaged COF values at the stable sliding stage for all the testing conditions at 200 °C.

Fig. 8 shows the surface morphologies of the worn balls obtained at 200 °C under different lubrication conditions. The wear scar obtained under dry condition is the biggest among all the worn balls. The wear scar caused by water has apparent scratches, and there are abundant black iron oxides adhered to the surroundings. It is acknowledged that these iron oxides are formed when the water contacts the steel surface at 200 °C. The adhered iron oxides can be found around the wear scar with nanolubricant containing 0.4 wt% TiO₂, above which the worn ball surfaces become smoother and cleaner.



Fig. 8 Surface morphologies of the worn balls obtained at 200 °C under lubrication conditions of (a) dry, (b) water, (c) 0.4 wt% TiO₂, (d) 1.0 wt% TiO₂, (e) 4.0 wt% TiO₂, and (f) 8.0 wt% TiO₂.

Fig. 9 shows the 3D images and corresponding cross-sectional profiles of the ball wear areas obtained at 200 °C under different lubrication conditions, which refers to the surface morphologies of the worn balls shown in Fig. 8. It is similar to the results in Fig. 5 that the dry and water condition produce the most severe ball wear in terms of the wear areas along Z axis (depth) at 200 °C, which are slightly larger than those obtained at ambient temperature. Another similar result is that the ball wear area continues to decrease until it reaches the smallest value (310 μ m²) with the increase of nano-TiO₂ concentration up to 4.0 wt%, followed by a bit larger ball wear area of 322 μ m² when using 8.0 wt% TiO₂. In this case, the ball wear caused dry condition can be maximally reduced by 88.1%. The variation trend of ball wear areas obtained at ambient temperature when the nano-TiO₂ concentration is above 0.4 wt%.



Fig. 9 3D images and corresponding cross-sectional profiles of the ball wear areas obtained at 200 °C under lubrication conditions of (a) dry, (b) water, (c) 0.4 wt% TiO₂, (d) 1.0 wt% TiO₂, (e) 4.0 wt% TiO₂, and (f) 8.0 wt% TiO₂.



Fig. 10 Wear areas of the balls obtained at 200 °C under different lubrication conditions.

3.3 Ball-on-disk tests at 500 °C

Fig. 11 shows the COF curves and values measured at 500 °C under different lubrication conditions. As can be seen from Fig. 11(a) that the overall running-in periods are much longer than those at ambient temperature and 200 °C (see Figs. 3(a) and 7(a)), and the COF curves caused by 4.0 and 8.0 wt% TiO₂ appear to be more fluctuant than those caused by other lubrication conditions. The variation of averaged COF values obtained from the stable stages is shown in Fig. 11(b). It is similar to the trends in Figs. 3(b) and 7(b) that the nanolubricant containing 4.0 wt% TiO₂ enables the maximal COF reduction by 39.5% based on that of dry condition. The difference is that the highest concentration of 8.0 wt% TiO₂ increases the COF value to the utmost among all the nanolubricants.



Fig. 11 (a) COF curves as functions of sliding time, and (b) averaged COF values at the stable sliding stage for all the testing conditions at 500 °C.

Fig. 12 shows the surface morphologies of the worn balls obtained at 500 °C under different lubrication conditions. It is observed that the wear scars produced by water and water-based nanolubricants are wider than that produced under dry condition. The sticking phenomenon of the ball only occurs when lubricated by water. The apparent scratches appear after water and 0.4 wt% TiO_2 are used in the sliding process. In contrast, the nanolubricants containing an even higher concentration of nano- TiO_2 lead to smoother wear scars. In particular, the wear scar caused by 4.0 wt% TiO_2 is the smallest (see Fig.12 (e)), while the one produced by 8.0 wt% TiO_2 becomes the biggest instead (see Fig.12 (f)).



Fig. 12 Surface morphologies of the worn balls obtained at 500 °C under lubrication conditions of (a) dry, (b) water, (c) 0.4 wt% TiO₂, (d) 1.0 wt% TiO₂, (e) 4.0 wt% TiO₂, and (f) 8.0 wt% TiO₂.

Fig. 13 shows the 3D images and corresponding cross-sectional profiles of the ball wear areas obtained at 500 °C under different lubrication conditions, such as dry, 0.4, 4.0 and 8.0 wt% TiO₂. It can be seen that the balls are significantly worn out under both dry condition and 0.4 wt% TiO₂ in terms of the wear areas along Z axis (depth) at 500 °C, which are much larger than those obtained at ambient temperature and 200 °C (see Figs. 5 and 9). The ball wear can be greatly alleviated by using the nanolubricant containing 4.0 wt% TiO₂, instead of being aggravated when the nano-TiO₂ concentration increases to 8.0 wt%. The wear areas of the balls yielded at 500 °C under applied lubrication conditions are quantitatively shown in Fig. 14. The results indicate that the ball wear produced under dry condition can be minimised by 85.5% when using the nanolubricant containing 4.0 wt% TiO₂.



Fig. 13 3D images and corresponding cross-sectional profiles of the ball wear areas obtained at 500 °C under lubrication conditions of (a) dry, (b) 0.4 wt% TiO₂, (c) 4.0 wt% TiO₂, and (d) 8.0 wt% TiO₂.

Fig. 14 Wear areas of the balls obtained at 500 °C under different lubrication conditions.

4. Discussion

4.1 Rolling and polishing effects

As shown in Fig. 15(a)-(d), the SEM images of worn disk surfaces using the nanolubricants containing 0.4, 1.0, 4.0 and 8.0 wt% TiO₂, respectively. The EDS spectra acquired from Area A in Fig. 15(c) and Area B in Fig. 15(d) are shown in Fig. 15(e) and (f), respectively, demonstrating that TiO₂ NPs are distributed in the wear tracks. As the TiO_2 NPs have spherical shapes, they could serve as rollers between the ball and the disk to reduce friction, at the same time polish the surfaces [44]. This is likely why the COF and ball wear are significantly reduced in comparison to the respective values obtained from dry and pure water conditions. Also, smoother surfaces of worn balls are produced, as can be seen in Figs. 3-6. The NPs that effectively rub surfaces should be originated from two sources, those trapped in surface valleys and existed outside contact region [23]. For the nanolubricant containing the lowest nano-TiO₂ concentration of 0.4wt%, few TiO₂ NPs act inside the contact region, thus limit their contribution to the reduction of COF and ball wear. With progressively increasing the nano-TiO₂ concentration up to 4.0 wt%, an increasing number of TiO₂ NPs are trapped in the disk valleys due to the presence of initial scratches over the disk surface (see Fig. 1(c) and (d)). These trapped TiO_2 NPs and supplementary ones can be furnished from the valleys and the outer region, respectively, onto the rubbing surfaces. This relieves NP starvation and thus improves the friction-reduction and anti-wear abilities, leading to a declining trend of the COF and ball wear [45-47]. When TiO₂ NPs are excessive, such as the case of 8.0 wt%, the TiO₂ NPs are sparsely populated in the wear track. This is because the NPs can accumulate around the contact region to form a barrier, which reduces the extraneous supply of NPs from the outer region [48]. Moreover, the NPs tend to agglomerate and hence increase their size. Both the block of NPs into the contact region and agglomerated NPs would aggravate the friction and wear [49], and therefore result in a higher COF and more severe ball wear, compared to those of 4.0 wt% TiO₂.

Fig. 15 SEM images of worn disk surfaces lubricated by (a) 0.4 wt% TiO₂, (b) 1.0 wt% TiO₂, (c) 4.0 wt% TiO₂, and (d) 8.0 wt% TiO₂, and EDS spectra of (e) 4.0 wt% TiO₂ and (f) 8.0 wt% TiO₂.

4.2 Effect of nanoparticles at elevated temperatures

Fig. 16 shows the SEM images and EDS mappings of the worn disk surfaces using the nanolubricants containing 0.4, 1.0, 4.0 and 8.0 wt% TiO₂ at the elevated temperature of 200 °C. It is found from Fig. 16(a) that TiO₂ NPs are scarcely distributed in the wear track, indicating that NPs may have played similar lubrication effect to that at ambient temperature. In contrast, small pieces of TiO₂ films that have loose microstructures are found to be deposited in the wear track when the nano- TiO_2 concentration increases to 1.0 wt% (see Fig. 16(b)). Such a lubricating film is observed over the entire wear track when the nano-TiO₂ concentration increases to 4.0 wt%, as shown in Fig. 16(c). The lubricating film appears like a type of "semisolid film" that consists of TiO₂ NPs and glycerol, since the water is expected to be fully evaporated at 200 °C, while glycerol has a much higher boiling temperature of 290 °C [50]. As glycerol is a viscous liquid, this semisolid film is supposed to have similar lubrication effect to the protective film formed by the oil containing NPs, which greatly facilitates the decrease of COF and ball wear [51, 52]. A further increase of nano-TiO₂ concentration to 8.0 wt% would enhance the semisolid film, i.e. the film becomes thicker and denser, as shown in Fig. 16(d). However, a thicker and denser film may impede the movement of the ball, and thus lead to a higher COF than that of 4.0 wt% TiO₂. Nevertheless, the ball wear is not deteriorated significantly, as the semisolid film can prevent the ball from direct contact with the disk. The semisolid film formed at 200 °C is expected to improve the tribological performance, also alleviate the sticking phenomenon occurred on the ball surface, as shown in Fig. 8.

It should be noted that the iron oxides are inevitably generated, especially when the water-based lubricants contact the hot disks at 200 °C. The EDS mappings of the element oxygen shown in Fig. 16 are also indicative of iron oxides other than TiO_2 NPs. It has been reported that oxidation of the steel surfaces leads to low friction and mild wear, irrespective of the initial lubrication condition [53, 54]. In the cases of pure water and water-based nanolubricants, the contribution of generated iron oxides to the friction and wear is deemed nearly identical, since water is the main composition in the lubricant. For water-based nanolubricants with high concentrations of nano-TiO₂, the roles of TiO₂ NPs are thus dominant.

Fig. 16 SEM images and EDS mappings of the worn disk surfaces lubricated by (a) 0.4 wt% TiO₂, (b) 1.0 wt% TiO₂, (c) 4.0 wt% TiO₂, and (d) 8.0 wt% TiO₂ at 200 °C.

Fig. 17 shows the SEM images and EDS mappings of the worn disk surfaces with the nanolubricants containing 0.4, 4.0 and 8.0 wt% TiO₂ at 500 °C. It is seen from Fig. 17(a) that there is a cluster of TiO₂ NPs and iron oxides scattered in the wear track, which greatly aggravates the friction and wear, and thus produces apparent scratches on the ball surface (see Fig. 12(c)) with severe ball wear (see Fig. 13(b)). When the nano-TiO₂ concentration increases to 4.0 wt%, the wear track is completely covered by a solid layer that is only composed of loose TiO₂ NPs (see Fig. 17(b)). This is because both water and glycerol are vaporised at 500 °C, the TiO₂ NPs thus form the solid layer under the pressure of contact between the ball surfaces from direct contact, and therefore improves the lubrication performance [2, 55]. In contrast, the nanolubricant containing 8.0 wt% TiO₂ appears to have formed a very compact solid layer under the elevated temperature of 500 °C, as shown in Fig. 17(c). As the compact layer is harder than the loose one,

the ball lubricated by 8.0 wt% TiO₂ thus suffers more severe wear than that lubricated by 4.0 wt% TiO₂. Besides, the TiO₂ NPs agglomerated around the contact region could cause extra friction on the rubbing surfaces [23], leading to a higher COF and a wider wear scar of the ball, compared to those of 4.0 wt% TiO₂. With the temperature increases from 200 to 500 °C, therefore, the increased COF and ball wear are dependent on a transfer from metal-on-semisolid film contact to metal-on-solid layer contact. It is similar to the case happened at 200 °C that the TiO₂ NPs dominate the contribution to improve the tribological properties of water-based nanolubricants.

Fig. 17 SEM images and EDS mappings of the worn disk surfaces lubricated by (a) 0.4 wt% TiO₂, (b) 4.0 wt% TiO₂, and (c) 8.0 wt% TiO₂ at 500 °C.

4.3 Friction and wear mechanisms

Fig. 18 schematically illustrates the friction and wear mechanisms involved in the ball-on-disk tests at ambient and elevated temperatures using the water-based nanolubricants. When the Cr steel ball slides against the relatively rough mild steel disk at ambient temperature (see Fig. 18(a)), the spherical TiO₂ NPs suspended in the lubricant play the roles of ball bearings in the contact region, so their rolling and polishing effects enable the reduction of COF between the rubbing surfaces, and thus decrease the ball wear. Because the disk surface has machined striations, TiO_2 NPs can be trapped in the valleys working together with the NPs supplemented from the outer region. With the nano- TiO_2 concentration increases from 0.4 to 4.0 wt%, there is an increasing number of effective TiO_2 NPs behaving in the contact region, indicating progressively improved lubrication performance. However, a further increase of nano-TiO₂ concentration up to 8.0 wt% may accelerate the agglomeration of TiO_2 NPs. This thus promotes the formation of a barrier that blocks the continuous supply of TiO₂ NPs from the outer region, leading to the increased COF and ball wear. The lubrication mechanisms at 200 °C are somehow different from those at the ambient temperature, which are associated with the formation of a semisolid film that consists of TiO₂ NPs and glycerol, as shown in Fig. 18(b). The semisolid film prevents the ball from direct contact with the disk, thereby decreasing the friction and ball wear. A maximal nano-TiO₂ concentration of 8.0 wt%, however, results in a thickened semisolid film, which may impede the movement of the ball and thus increase the COF. When the disk is further heated up to 500 °C, the semisolid film is transformed to a solid layer which covers the entire disk surface, as shown in Fig. 18(c). As the solid layer is only composed of TiO_2 NPs which is harder than the semisolid film, this would result in higher COF values and more severe ball wear, compared to those caused by the semisolid film formed at 200 °C.

Fig. 18 Schematic illustration of friction and wear mechanisms using the water-based nanolubricants at (a) ambient temperature, (b) 200 °C and (c) 500 °C.

5. Conclusions

In this study, ball-on-disk tribological tests were conducted at ambient and elevated temperatures of 200 and 500 °C using the water-based nanolubricants containing different nano-TiO₂ concentrations from 0.4 to 8.0 wt%. The friction and wear behaviours of the chrome steel ball were investigated in terms of the COF and wear area of the ball. The conclusions are drawn as follows.

- (1) The use of water-based nanolubricants containing different nano-TiO₂ concentrations greatly decreases the COF and ball wear at both ambient and elevated temperatures, in comparison to the benchmark values obtained from the dry and water conditions. The optimal concentration of 4.0 wt% TiO₂ exhibits the best lubrication performance among all the lubrication conditions.
- (2) At ambient temperature, the COF and ball wear obtained under dry condition can be decreased by 40% and 82.5%, respectively, which is ascribed to the rolling and polishing effects contributed by TiO₂ NPs.
- (3) At 200 °C, the COF and ball wear obtained under dry condition are decreased by 58.9% and 88.1%, respectively, due to the formation of a semisolid film that consists of TiO₂ NPs and glycerol.
- (4) At 500 °C, the COF and ball wear obtained under dry condition are reduced by 39.5% and 85.5%, respectively. The lubrication mechanism is dependent on the formation of a solid layer that is only composed of TiO_2 NPs.

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