

1 **Surfactant micelle self-assembly directed highly symmetric ultrasmall**  
2 **inorganic cages**

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15 **Nanoscale objects with highly symmetrical cage-like polyhedral shapes, often with**  
16 **icosahedral symmetry, have recently been assembled using DNA<sup>1-3</sup>, RNA<sup>4</sup>, and**  
17 **proteins<sup>5,6</sup> for biomedical applications. These achievements relied on advances in the**  
18 **development of programmable self-assembling biomaterials<sup>7-10</sup>, as well as rapidly**  
19 **developing single-particle three-dimensional (3D) reconstruction techniques of cryo**  
20 **electron microscopy (cryo-EM) images that provide high-resolution structural**  
21 **characterization of biological complexes<sup>11-13</sup>. In contrast, such single-particle 3D**

22 reconstruction approaches have not been successfully applied to help identify  
23 unknown synthetic inorganic nanomaterials with highly symmetrical cage-like  
24 shapes. In this paper, employing a combination of cryo-EM and single-particle 3D  
25 reconstruction, we suggest the existence of isolated ultrasmall silica cages  
26 (“silicages”) with dodecahedral structure. We hypothesize that this highly  
27 symmetric self-assembled cage forms via arrangement of primary silica clusters in  
28 aqueous solutions on the surface of oppositely charged surfactant micelles. This  
29 discovery paves the way for such nanoscale cages from silica and other inorganic  
30 materials to be used as building blocks for a wide range of advanced functional  
31 materials applications.

32 For the search of dodecahedral silica cage structures (Figure 1), we looked at the  
33 early formation stages of surfactant micelle directed silica self-assembly<sup>14</sup>. Our synthesis  
34 system contained cetyltrimethylammonium bromide (CTAB) surfactant micelles and  
35 tetramethyl orthosilicate (TMOS) as a sol-gel silica precursor (see Methods section).  
36 Hydrophobic mesitylene (TMB) was added into the aqueous CTAB micelle solution,  
37 increasing micelle size and deformability<sup>15</sup>. TMOS was selected as the silica source due  
38 to its fast hydrolysis rate in water, and the initial reaction pH was adjusted to ~8.5.  
39 Following TMOS addition, its hydrolysis to silicic acid reduced the reaction pH to  
40 neutral<sup>16,17</sup>. The lowered pH accelerated silane condensation, forming primary silica  
41 clusters with diameter around 2 nm<sup>18</sup>. The negatively charged silica clusters were  
42 attracted to the positively charged CTAB micelle surface, assembling into micelle  
43 templated nanostructures<sup>14</sup>. This experimental design, where fast hydrolysis and

44 condensation of the silica precursor quickly terminated the reaction process, allowed  
45 preservation of early formation stages of micelle directed silica self-assembly<sup>17</sup>.

46 In order to improve particle dispersity on transmission electron microscopy  
47 (TEM) grids, low molar mass silane modified monofunctional polyethylene glycol (PEG)  
48 was added into the solution one day prior to TEM sample preparation, thereby covalently  
49 coating the accessible silica surface<sup>19</sup>, yielding PEGylated nanoparticles (Extended Data  
50 Figure 1) that could be further purified and isolated from the synthesis solution. Narrowly  
51 size distributed particles were observed under TEM with average diameter around 12 nm  
52 (Figure 2a and inset), consistent with silica structures wrapped around TMB swollen  
53 CTAB micelles<sup>15</sup>. The detailed particle structure was difficult to identify, however.  
54 Therefore, TEM samples were subsequently plasma etched on carbon grids for five  
55 seconds prior to imaging to remove excess organic chemicals (*e.g.* PEG-silane),  
56 otherwise contributing to background noise. To further improve the signal-to-noise ratio,  
57 a series of images were acquired of the same sample area and averaged. Stripes and  
58 windows in zoomed-in images of individual particles became more clearly recognizable,  
59 suggesting the presence of cage-like structures (Figure 2b and insets).

60 The study of thousands of such single particle TEM images revealed the  
61 prevalence of two cage projections with two- and, in particular, five-fold symmetry,  
62 respectively (Figure 2c), too few to allow for a successful 3D reconstruction. We  
63 therefore shifted our attention to cryo-EM characterization of the native reaction solution.  
64 The silica surface PEGylation step was omitted as the high PEG concentration  
65 substantially increased radiation sensitivity of the samples, resulting in difficulties  
66 obtaining clear cryo-EM images.

67 Cryo-EM provided direct visualization of particles in solution with arbitrary  
68 orientation, *i.e.* without disturbances due to sample drying on TEM substrates, including  
69 structure deflation. The background noise was significantly reduced as a result of the  
70 absence of a TEM substrate as well as chemicals dried onto the substrate during sample  
71 preparation (Figure 2c). Although particle aggregation was occasionally observed in  
72 cryo-EM (Figure 2d), individual silica nanoparticles with cage-like structures could  
73 always be identified (Figure 2c and d). No particle aggregation was observed in dry-state  
74 TEM of PEGylated particles, suggesting that particle aggregation observed by cryo-EM  
75 was a reversible process that could be overcome via insertion of PEG chains.

76 We manually identified ~19,000 single particle images from cryo-EM  
77 micrographs, clustered them and averaged the images in each cluster in order to improve  
78 the signal-to-noise ratio<sup>20</sup>. The averages showed different orientations of silica  
79 nanoparticles with cage-like structures, *i.e.* silicages (Extended Data Figure 2a). Averages  
80 were identified that were consistent with selected projections of a pentagonal  
81 dodecahedral cage (Figure 2c and Extended Data Figure 2b). The dodecahedral silicage  
82 (icosahedral point group,  $I^h$ , Figure 1) is the simplest of a set of Voronoi polyhedra  
83 suggested to form the smallest structural units of multiple forms of mesoporous silica<sup>21</sup>.  
84 Although such highly symmetric ultrasmall silica cages have never been isolated before,  
85 it seemed likely that this should be possible.

86 Guided by this structural insight, single-particle 3D reconstruction of silicages  
87 were performed using the “Hetero”<sup>22</sup> model-based maximum likelihood algorithm, in  
88 which a two-class reconstruction was computed to overcome challenges associated with  
89 structural heterogeneity (Methods section) and rotational icosahedral symmetry was

90 imposed on both classes (Extended Data Figure 2). One of the two-class reconstructions  
91 was a dodecahedral cage (Figure 3a and b). Low intensity signal was identified inside the  
92 reconstructed cage, consistent with the presence of TMB swollen CTAB micelles inside  
93 the silicage, whose electron density is lower than silica but higher than the surrounding  
94 ice. The other class (*i.e.*, non-cage) did not provide an interpretable structure, likely due  
95 to heterogeneity in the structure of the corresponding particles. Such two-class  
96 reconstructions were performed using different numbers of single particle images (2000,  
97 7000, and 10000) and yielded consistent results. Single-class reconstructions were also  
98 performed, using only the images in the class showing dodecahedral cages in two-class  
99 reconstructions, by the Hetero algorithm. Equivalent two-class and single-class  
100 reconstructions were also performed by the widely-used RELION 2.1<sup>24</sup> system.  
101 Dodecahedral cage structures were obtained in all these reconstructions (Extended Data  
102 Figure 3 and Supplementary Videos 1 and 2). The resolution of the reconstructions was  
103 approximately 2 nm<sup>25</sup> (Extended Data Figure 4). Silica in these cages is amorphous at the  
104 atomic level, which prevented atomic resolution in these reconstructions.

105         The Hetero reconstruction algorithm provided estimates of the projected  
106 orientation (*i.e.*, three Euler angles) for each experimental image, which were used to  
107 compute predicted projections. Nine predicted projections and corresponding  
108 experimental images were manually clustered, and averages were computed for each  
109 cluster (Figure 3c). The similarity of the projections of the 3D reconstruction and the  
110 averaged experimental images supports the dodecahedral cage structure. Furthermore, the  
111 theoretical probabilities of finding each of the nine projections (Figure 3c) were  
112 calculated based on the assumption that the orientations of silicages in cryo-EM are

113 random. The results were then compared to the probabilities observed by single particle  
114 3D reconstruction (Extended Data Figure 5). The high consistency between theoretical  
115 and experimental projection probabilities further supports the dodecahedral cage  
116 reconstruction.

117 While at this early point we can only speculate about the exact formation  
118 mechanism of the observed silicage structure, there are clues found in the details of the  
119 reconstruction. The vertices of the dodecahedral silicage had a diameter around 2.4 nm  
120 (Figure 3b), only slightly larger than the diameter of primary silica clusters, *i.e.*  $\sim 2$  nm<sup>18</sup>  
121 (Extended Data Figure 6). The interstitial spacing between two nearby vertices was  
122 estimated to be about 1.4 nm (*i.e.*, edge length, 3.8 nm, minus diameter of vertices, 2.4  
123 nm, see Figure 3b), much smaller than the diameter of such clusters. Bridges between  
124 vertices forming the edges of the dodecahedron were substantially thinner than the size of  
125 the primary clusters (Figure 3a and b). This suggests that negatively charged primary  
126 silica clusters formed in solution may start to come down onto the positively charged  
127 micelle surface attracted by Coulomb interactions. As more and more silica clusters  
128 assemble on the micelle surface, as a result of their repulsive interactions and possible  
129 interactions with other micelles, they may move to the vertices of a dodecahedron.  
130 Additional silane condensation onto the surface of growing clusters may eventually lead  
131 to bridge formation resulting in the final observed cage structure (Figure 3). The origin of  
132 icosahedral symmetry in viruses has been associated with the energy minimization of two  
133 opposing interactions, repulsive interactions associated with the bending rigidity and  
134 attractive hydrophobic interactions<sup>27</sup>. In a related way, in addition to electrostatic  
135 interactions, deformation of the micelle surface around the silica clusters may be another

136 important contributor to the free energy in our system. This is supported by experiments  
137 showing that the cage structures do not form in the absence of TMB (Extended Data  
138 Figure 7), which is expected to enhance micelle surface deformability.

139         Micelle self-assembly directed ultrasmall cage structures could also be fabricated  
140 from other inorganic materials with similar feature sizes and surface chemistry  
141 characteristics to silica. In preliminary experiments silica was replaced by two metals,  
142 gold and silver, and a transition metal oxide, vanadium oxide. Gold and silver structures  
143 were prepared by the reduction of metal precursors,  $\text{HAuCl}_4 \cdot 3\text{H}_2\text{O}$  and  $\text{AgNO}_3$ ,  
144 respectively, in the presence of the micelles (Extended Data Figure 8, also see Methods  
145 section). Tetrakis(hydroxymethyl)phosphonium chloride (THPC) was used as both the  
146 reductant and the capping agent to stabilize primary gold and silver nanoparticles and  
147 provide negative surface charges<sup>28</sup>. In contrast, primary vanadium oxide nanoparticles  
148 with native negatively charged particle surface were prepared via sol-gel chemistry<sup>29</sup>,  
149 similar to the synthesis of silicages (Methods section). Images of individual particles  
150 obtained by TEM revealed similar internal structure (Figure 4). These nanoparticles did  
151 not appear to be dense but instead showed cage-like structures (compare Figure 2 and 4),  
152 further corroborated by associated projection averages<sup>20</sup> revealing cages with rotational  
153 symmetry (bottom insets in Figure 4), similar to the prevalent projection in case of the  
154 silicage (*vide supra*). Micelle self-assembly directed cages like the dodecagonal structure  
155 described in this paper may therefore not be unique to amorphous silica, but may provide  
156 direct synthesis pathways to crystalline material cages (Extended Data Figure 9).

157         There are a number of important ramifications that derive from our silicage  
158 discovery. For example, such cages are considered as individual structural units from

159 which larger scale mesoporous silica is built up in a bottom-up manner<sup>15,21</sup>. However,  
160 since a dodecahedron cannot be used to generate a tessellation of 3D space, other silica  
161 cage structures would be required. This motivates better understanding of early formation  
162 pathways of surfactant directed silica self-assembly, including the search for micelle  
163 directed ultrasmall silicages with other structures, and from other materials (*vide supra*).  
164 Furthermore, the chemical and practical value of this polyhedral structure may prove  
165 extremely high. Considering the high versatility of silica surface chemistry, and the  
166 ability to distinguish cage inside and outside via micelle directed synthesis<sup>17</sup>, one can  
167 readily conceive cage derivatives of many kinds, which may exhibit unusual properties  
168 and be useful in applications ranging from catalysis to drug delivery. For example, based  
169 on recent successes in clinical translation of ultrasmall fluorescent silica nanoparticles  
170 with similar particle size and surface properties<sup>30</sup>, a whole range of novel diagnostic and  
171 therapeutic probes with drugs hidden in the inside of the cages can be envisaged.

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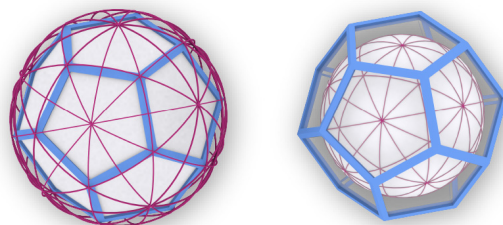
248 **Supplementary Information** is available in the online version of the paper.

249 **Acknowledgments** This project was support by the National Cancer Institute of the  
250 National Institutes of Health under Award Number U54CA199081. Y.G. and P.C.D.  
251 gratefully acknowledge financial support by the National Science Foundation under grant  
252 number 1217867, and Y.G. gratefully acknowledges financial support by a 2017 Google  
253 Ph.D. Fellowship in Machine Learning. T.A. acknowledges financial support by the  
254 Ghent University Special Research Fund (BOF14/PDO/007) and the European Union's  
255 Horizon 2020 research and innovation program (MSCA-IF-2015-702300). M.Z.T.  
256 gratefully acknowledges fellowship support by the Ministry of National Education of the  
257 Republic of Turkey. Work made use of shared facilities of the Cornell Center for  
258 Materials Research (CCMR) with funding from the NSF MRSEC program (DMR-  
259 1719875), and the Nanobiotechnology Center shared research facilities at Cornell. The  
260 authors gratefully thank V. Elser and Y. Jiang, Cornell Physics Department, for helpful  
261 discussions.

262 **Author Contributions** K.M., T.A. and U.W. designed the experimental work. Y.G. and  
263 P.C.D. performed the reconstructions. K.M. synthesized the silica-based materials. T.A.  
264 synthesized the metal and transition metal oxide based materials. K.M. and T.A.  
265 performed TEM and cryo-EM characterizations. K.M., T.A., M.T., and T.K. processed  
266 the images for reconstructions. K.M., T.A., and U.W. discussed the experimental work.  
267 U.W. wrote the manuscript with input of all coauthors. U.W. supervised the work.

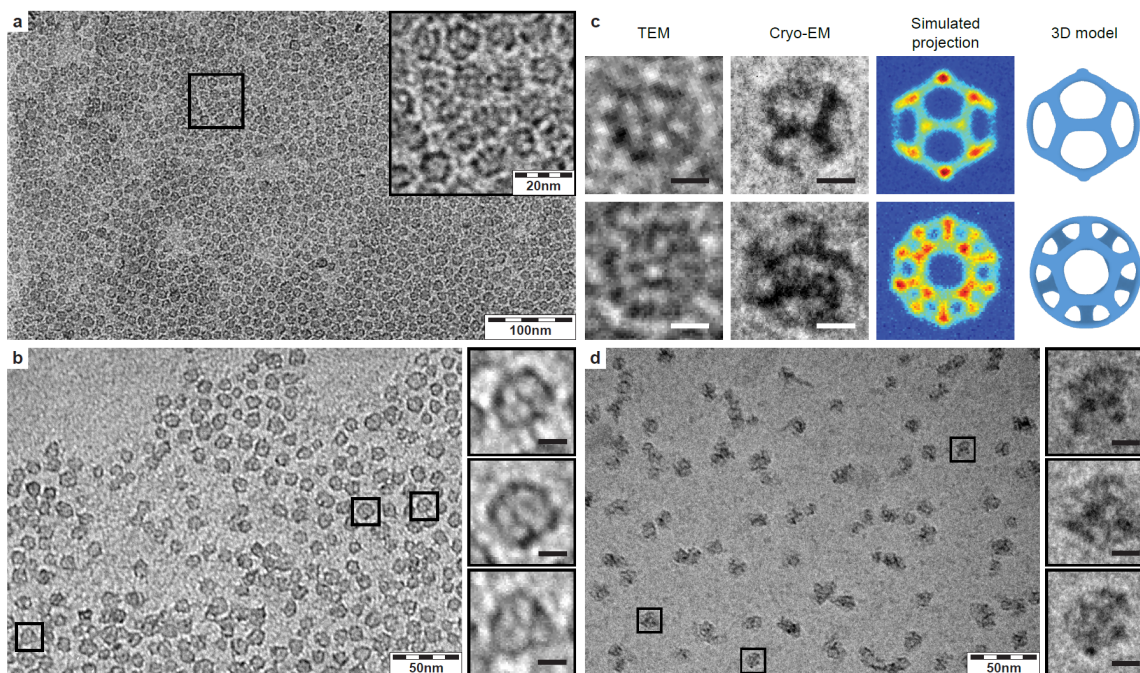
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270 Cornell University. Readers are welcome to comment on the online version of this article  
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272 addressed to U. W. ([ubw1@cornell.edu](mailto:ubw1@cornell.edu)).

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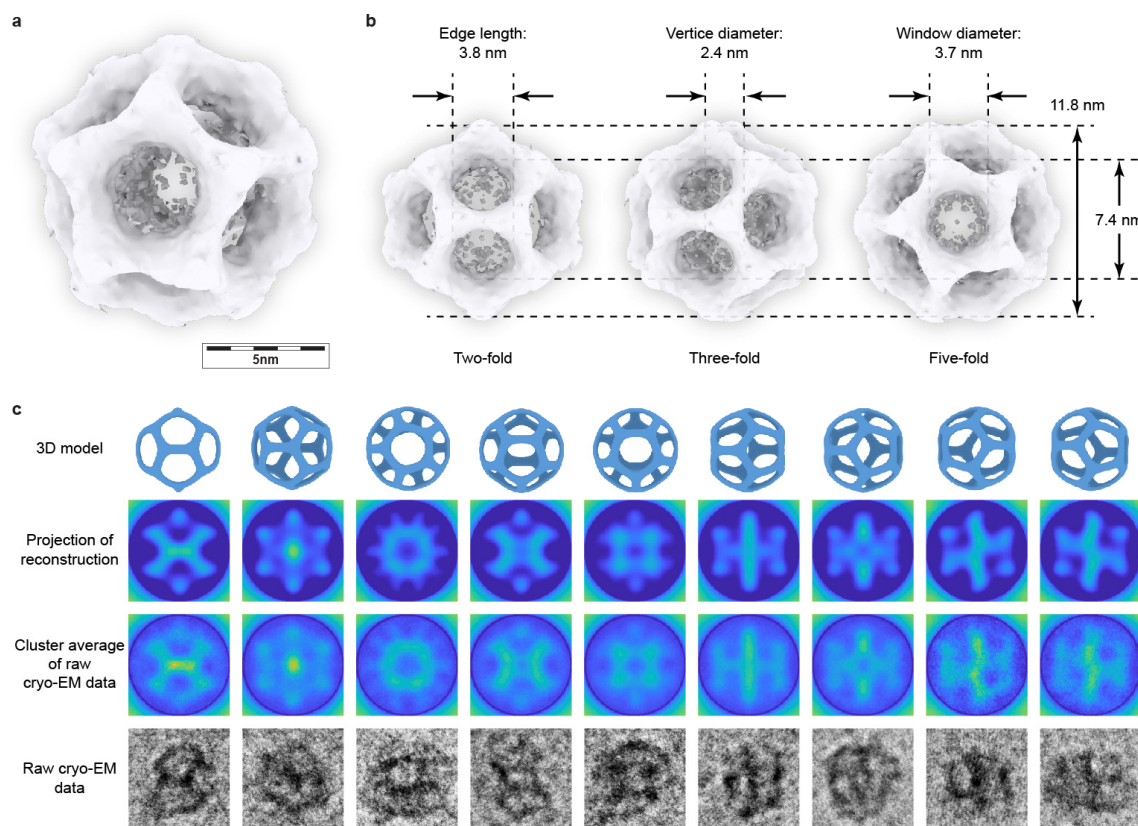
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**Figure 1 | Representation of dodecahedron.** Among the platonic solids, the dodecahedron best fills out its circumscribed sphere, *i.e.* a sphere that passes through all its vertices (left). The inscribed sphere passing through all facets (right) is shown for comparison.

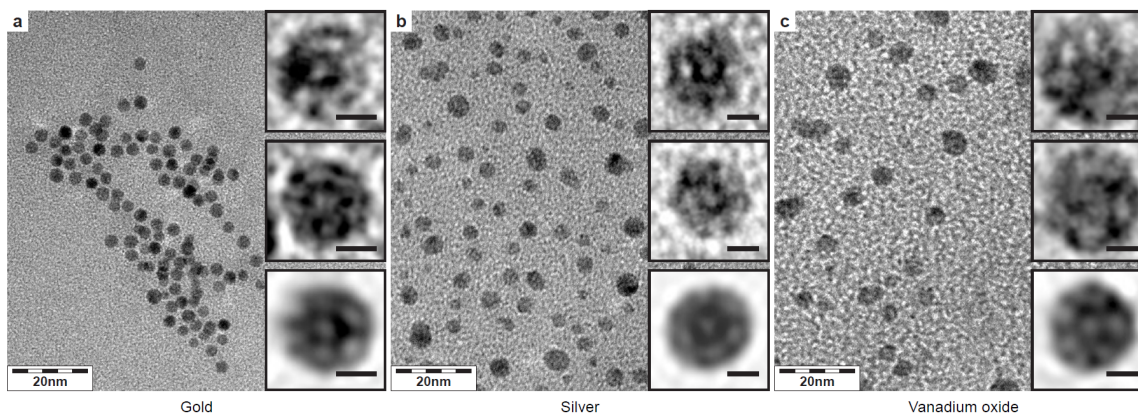


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**Figure 2 | TEM and cryo-EM characterizations of silicages.** **a**, TEM images at low magnification of PEG-coated silicages on carbon substrate. The inset in **(a)** shows a zoomed-in image. **b**, Averaged TEM image using eleven images acquired of the same sample area of PEG-coated silicages with insets showing representative individual structures at higher magnification. The sample was plasma etched for five seconds prior to TEM characterization to reduce background noise. **c**, Comparison between silicages observed in TEM and cryo-EM with projections of simulated dodecahedral cages and models. **d**, Cryo-EM images of silicages without PEG coating. Scale bars in the insets in **(b, c and d)** are 5 nm.



291  
 292 **Figure 3 | Single particle reconstruction of dodecahedral silicage.** **a** and **b**,  
 293 Dodecahedral silicage reconstruction result (**a**) and its three most unique projections  
 294 along the two-, three- and five-fold symmetry axes (**b**). The average dimensions of  
 295 silicages were estimated based on the reconstructed dodecahedral silicage (**b**). **c**,  
 296 Representative comparison of nine unique projections from the reconstruction and cryo-  
 297 EM cluster averages with projections of a 3D dodecahedral cage model (top row in **c**).  
 298 Corresponding single cryo-EM images are displayed at the bottom in (**c**) highlighting  
 299 the difference between raw data and reconstruction. Scale bar in (**c**) is 10 nm. Visualizations  
 300 in panels (**a**) and (**b**) are by UCSF Chimera<sup>23</sup>.  
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302  
 303 **Figure 4 | Cage-like structures with different inorganic compositions.** **a**, **b**, and **c**,  
 304 Similar cage-like nanoparticles were obtained when silica was replaced by other

305 materials, including gold (**a**), silver (**b**), and vanadium oxide (**c**). The insets display  
306 zoomed-in images of individual particles (top two rows) and averaged images<sup>20</sup> (bottom  
307 row). The scale bars in all insets are 2 nm.  
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310 METHODS

311 **Chemicals and materials.** All chemicals were used as received.  
312 Cetyltrimethylammonium bromide (CTAB), ammonia (2 M in ethanol), mesitylene (1,3,5  
313 trimethylbenzene, TMB), tetramethyl orthosilicate (TMOS), gold chloride trihydrate  
314 ( $\text{HAuCl}_4 \cdot 3\text{H}_2\text{O}$ ), silver nitrate ( $\text{AgNO}_3$ ), tetrakis(hydroxymethyl)phosphonium chloride  
315 (THPC), dimethyl sulfoxide (DMSO), acetic acid, and ethanol were purchased from  
316 Sigma-Aldrich. Vanadium oxytriisopropoxide was purchased from Alfa Aesar.  
317 Anhydrous potassium carbonate ( $\text{K}_2\text{CO}_3$ ) was purchased from Mallinckrodt. Anhydrous  
318 ethanol was purchased from Koptec. Silane modified monofunctional polyethylene glycol  
319 (PEG-silane) with molar mass around 500 g/mol (6-9 ethylene glycol units) was  
320 purchased from Gelest. Carbon film coated copper grids for TEM and C-Flat holey  
321 carbon grids for cryo-EM were purchased from Electron Microscopy Sciences.

322 **Synthesis, TEM, and cryo-EM characterization of silicages.** Silicages were  
323 synthesized in aqueous solution through surfactant directed silica condensation. 125 mg  
324 of CTAB was first dissolved in 10 ml of ammonium hydroxide solution (0.002 M). 100  
325  $\mu\text{l}$  of TMB was then added to expand CTAB micelle size, and the water: CTAB: TMB  
326 molar ratio was about 1620: 1: 2. The solution was stirred at 600 rpm at 30 °C overnight,  
327 followed by the addition of 100  $\mu\text{l}$  of TMOS. The reaction was then left at 30 °C  
328 overnight under stirring at 600 rpm.

329 To prepare cryo-EM samples, 5  $\mu\text{l}$  of the native reaction solution was applied to glow  
330 discharged CF-4/2-2C Protochips C-Flat holey carbon grids, blotted using filter paper and  
331 plunged into a liquid mixture of 37% ethane and 63% propane at -194 °C using an EMS

332 plunge freezer. Cryo-EM images were acquired on a FEI Tecnai F20-ST TEM operated  
333 at an acceleration voltage of 200 kV using a Gatan Orius CCD camera. All cryo-EM  
334 images used for reconstruction were acquired at the same magnification, with a pixel size  
335 of 0.16 nm, and nominal defocus was kept between 1  $\mu\text{m}$  and 2  $\mu\text{m}$ .

336 To prepare dry-state TEM samples, 100  $\mu\text{l}$  of PEG-silane was added into the reaction  
337 solution. The reaction solution was left at 30  $^{\circ}\text{C}$  overnight under stirring at 600 rpm to  
338 surface modify silicages covalently with PEGs to improve their dispersity on TEM grids.  
339 Afterwards, 30  $\mu\text{l}$  of the reaction solution was dropped onto a copper grid coated with a  
340 continuous carbon film, and blotted using filter paper. TEM images were acquired using  
341 a FEI Tecnai T12 Spirit microscope operated at an acceleration voltage of 120 kV. In  
342 order to improve the signal to noise ratio in recorded images, TEM sample grids were  
343 plasma etched for 5 seconds before TEM characterization, and a series of images was  
344 acquired of the same sample area, which was then averaged.

345 In order to quench individual primary silica clusters formed at the very early stages of  
346 cage formation, 100  $\mu\text{l}$  of PEG-silane was added into the reaction solution about three  
347 minutes after the addition of TMOS. The rest of the procedures, including particle  
348 synthesis, dry-state TEM sample preparation, and TEM characterization, were the same  
349 as described above.

350 **Particle purification.** To remove CTAB and TMB from the cages, after adding PEG-  
351 silane and stirring at room temperature for a day (Methods section), the solution was  
352 heat-treated at 80  $^{\circ}\text{C}$  overnight to further enhance the covalent attachment of PEG-silane  
353 to the silica surface of the silicages. The PEGylated nanocages were first dialyzed



354 (molecular weight cut off, MWCO, 10 kDa) in a mixture of acetic acid, ethanol, and  
355 water (volume ratio 7:500:500) for three days, and were then dialyzed in DI water for  
356 another three days<sup>17</sup>. In both cases the dialysis solutions were changed once per day. The  
357 dry-state TEM sample preparation and TEM characterization methods were the same as  
358 described in the Methods section.

359 **Synthesis of particles without TMB.** The synthesis and TEM characterization methods  
360 used for particles without TMB were identical to those with TMB as described in the  
361 Methods section, except that the TMB addition step was omitted.

362 **Silicage surface area and yield of production.** The specific surface area of the silicages  
363 was assessed by a combination of nitrogen sorption measurements and theoretical  
364 estimations. After PEGylated silicage synthesis and purification, particles were first up-  
365 concentrated using a spin filter (Vivaspin 20, MWCO 10 kDa) and dried at 60 °C.  
366 Particles were then calcined at 550°C for 6 hours in air. The production yield was then  
367 estimated via dividing the remaining weight after calcination, *i.e.* weight of inorganic  
368 silica, by the theoretical weight of silica, *i.e.* calculated based on the amount of silica  
369 source added into the synthesis. Nitrogen adsorption and desorption isotherms were  
370 acquired using a Micromeritics ASAP 2020 (Extended Data Figure 1c) yielding a specific  
371 surface area of 570 m<sup>2</sup>/g using the Brunauer–Emmett–Teller (BET) method. For  
372 comparison, using the dodecahedral cage model with the dimensions from the  
373 reconstruction shown in Figure 3, a theoretical surface area of silicages was estimated to  
374 be around 790 m<sup>2</sup>/g. Overestimation of the experimental value is consistent with expected  
375 losses of surface area during sample calcination.

376 **Synthesis and TEM characterization of metal cage-like structures.** The gold and  
377 silver cage-like structures were prepared by the reduction of metal precursors,  
378  $\text{HAuCl}_4 \cdot 3\text{H}_2\text{O}$  and  $\text{AgNO}_3$ , respectively, in the presence of micelles with the same water:  
379 CTAB: TMB ratio as for the silicage work. In a typical batch, 50 mg of CTAB was  
380 dissolved in 4 ml of water at 30 °C, then 40  $\mu\text{l}$  of TMB and 200  $\mu\text{l}$  of ethanol were added  
381 to the mixture. After stirring the reaction at 30 °C overnight at 600 rpm, 16  $\mu\text{l}$  of either  
382  $\text{HAuCl}_4 \cdot 3\text{H}_2\text{O}$  (25 mM) or  $\text{AgNO}_3$  (25 mM) was added, followed after 5 minutes by 8  $\mu\text{l}$   
383 of THPC (68 mM). After another 5 minutes, 6  $\mu\text{l}$  of potassium carbonate (0.2 M) was  
384 finally added.

385 Dry-state TEM samples for gold and silver cage-like structures were prepared after one  
386 day and 6 hours of reaction, respectively, due to different reaction rates as described in  
387 the Methods section. In both cases, the samples were prepared by drying 8  $\mu\text{l}$  of the  
388 native reaction mixture diluted three times in ethanol on a TEM grid in air overnight. In  
389 order to remove the thick CTAB layer before imaging, the grid was immersed in ethanol  
390 for 2 minutes and then dried in air. TEM images of metal cage-like structures were  
391 acquired using a FEI Tecnai T12 Spirit microscope operated at an acceleration voltage of  
392 120 kV.

393 **Synthesis and TEM characterization of vanadium oxide cage-like structures.** The  
394 vanadium oxide cage-like structures were prepared based on sol-gel chemistry very  
395 similar to the silicages, using vanadium oxytriisopropoxide as the precursor. In a typical  
396 batch, 50 mg of CTAB was dissolved in 4 ml of water at 30 °C, then 40  $\mu\text{l}$  of TMB was  
397 added to the mixture. After stirring the reaction at 30 °C overnight at 600 rpm, 50  $\mu\text{l}$  of  
398 vanadium oxytriisopropoxide diluted in 100  $\mu\text{l}$  of DMSO was added to the reaction.

399 Dry-state TEM samples for vanadium oxide cage-like structures were prepared after one  
400 day of reaction by drying on a TEM grid 8  $\mu\text{l}$  of the native reaction mixture diluted 10  
401 times in water. At such dilution, the amount of CTAB was low enough so that the TEM  
402 samples did not require any plasma cleaning or soaking in ethanol prior to imaging. The  
403 TEM images of vanadium oxide cages were acquired using a FEI Tecnai T12 Spirit  
404 microscope operated at an acceleration voltage of 120 kV.

405 **Particle reconstruction.** The “Hetero” model-based maximum likelihood algorithm<sup>22</sup>  
406 was used which can simultaneously estimate: (1) a reconstruction for each type of  
407 particle shown in the images, (2) the type of particle shown in each image, and (3) the  
408 projection orientation for each image. Such joint estimation is a central feature of the  
409 algorithm and is a natural approach to process data from complicated mixtures. The  
410 estimates in (2) and (3), which are based on 3D structure, are independent of the  
411 clustering of 2D images, which is based on pixel values (*e.g.*, Extended Data Figure 2). In  
412 addition to the Hetero algorithm, the widely used RELION 2.1<sup>24</sup> system was applied to  
413 compute equivalent two-class and single-class reconstructions. The images were  
414 corrected for the CTF by phase flipping.

415 **Additional details and optical characterization of the metal and vanadium oxide**  
416 **based syntheses of cage-like structures.** In contrast to the sol-gel reaction leading to the  
417 silicage, the gold and silver cage-like structures syntheses relied on reduction reactions.  
418 To this end, tetrakis(hydroxymethyl)phosphonium chloride (THPC) was used as it reacts  
419 in water at basic pH to form trimethoxyphosphine, which can play both the role of  
420 reductant and capping agent for the metal nanoparticles. THPC has been widely used for

421 the synthesis of ultra-small ( $< 3$  nm) and negatively charged phosphine-stabilized gold  
422 nanoparticles<sup>28</sup>. These nanoparticles are often used as seeds for the subsequent growth of  
423 continuous gold shells on the surface of aminated silica nanoparticles thanks to their high  
424 affinity and binding efficiency with amine groups<sup>31,32</sup>. Alcohol was added to the reaction  
425 mixture in order to mimic the conditions of the silicage synthesis where methanol is  
426 formed upon hydrolysis of TMOS. Early stage preliminary experiments showed that  
427 resulting structures were less size dispersed when using ethanol in slightly higher  
428 concentration than the released methanol in the silicage synthesis. Gold and silver cage-  
429 like structure syntheses were performed at a much lower concentration ( $[\text{Au}]$  or  $[\text{Ag}] =$   
430  $93.7 \mu\text{M}$ ) as compared to the silicages ( $[\text{Si}] = 65.9 \text{ mM}$ ). Attempts at synthesizing gold  
431 and silver cages at higher concentrations resulted in much larger nanoparticles with no  
432 apparent internal structure.

433 **Gold based synthesis.** The addition of gold precursor to the reaction initially resulted in  
434 the formation of a pale yellow precipitate which turned into a clear, *i.e.* non-turbid, darker  
435 orange solution within a couple of minutes under stirring at  $30 \text{ }^\circ\text{C}$  (see Extended Data  
436 Figure 8 for a survey of the absorption characteristics at each step of the synthesis). Since  
437 neither the precipitate nor the darker orange coloration was observed in the absence of  
438 CTAB, we attribute these observations to some interaction between the gold chloride  
439 anions and the ammonium groups of the CTAB. After the addition of THPC, the solution  
440 turned colorless within a couple of minutes, indicating that gold(III) had been reduced to  
441 gold(I). The subsequent transformation of THPC into trimethoxyphosphine by increasing  
442 the pH with the addition of potassium carbonate happened within the first hour of  
443 reaction (see also description for the case of silver below). However, the reduction from

444 gold(I) to gold(0) was found to be rather slow with the first hint of coloration appearing  
445 after 8 hours of reaction. After one day of reaction, the solution ended up exhibiting a  
446 brown coloration. This brown coloration was the signature of gold nanoparticles which  
447 are too small or not dense enough to show a strong surface plasmon resonance, as  
448 evidenced by the absorption profile in Extended Data Figure 8 which only shows a faint  
449 feature around 510 nm.

450 **Silver based synthesis.** The addition of silver precursor to the reaction did not initially  
451 translate into any visible effects, neither in the presence of CTAB/TMB nor after adding  
452 THPC. Nevertheless, after adding potassium carbonate to the silver based synthesis, the  
453 solution started to turn pale yellow within the first hour of reaction and resulted in an  
454 intense yellow coloration after 6 hours, at which point the TEM samples were prepared.  
455 This yellow coloration is classic for such small silver nanoparticles showing a surface  
456 plasmon resonance centered around 420 nm as shown in Extended Data Figure 8.

457 **Vanadium oxide based synthesis.** The vanadium oxide cage-like nanoparticles were  
458 prepared under the same conditions as the silicage, however the pH was not adjusted with  
459 ammonia due to the fast hydrolysis and condensation rate of the vanadium oxide  
460 precursor. In contrast to the metal cage-like nanoparticles synthesis, no alcohol was  
461 added here since the hydrolysis of the vanadium precursor, vanadium oxytriisopropoxide,  
462 produces alcohol similar to the silicage synthesis. The addition of this precursor to the  
463 TMB micelles resulted in the immediate formation of a red precipitate. Under stirring, the  
464 precipitate dispersed homogeneously in solution, which remained turbid, and turned  
465 orange after one day of reaction at 30 °C.

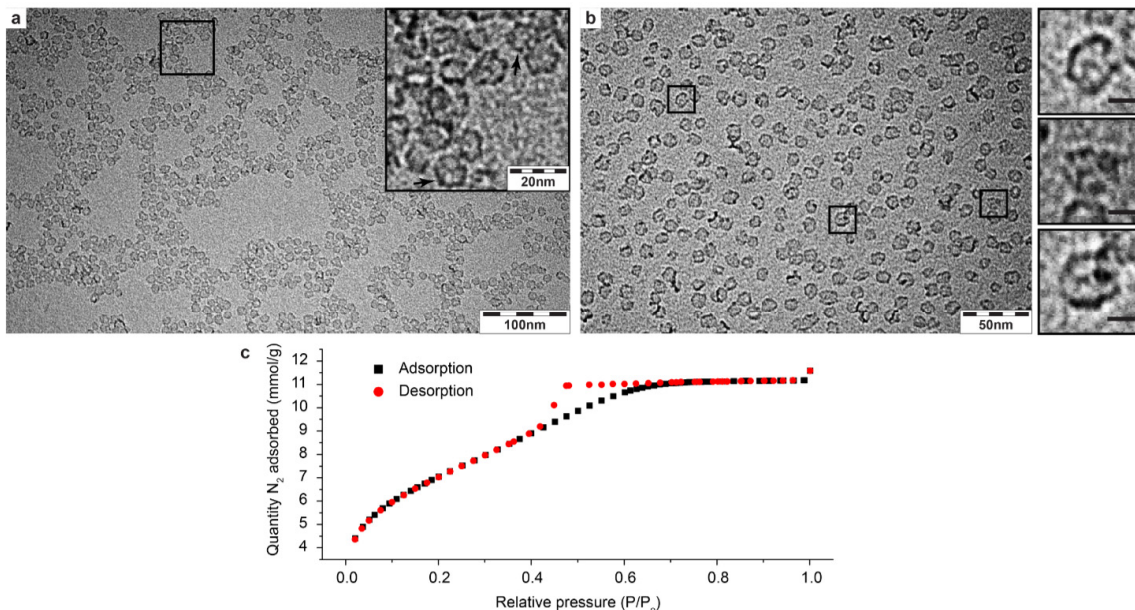
466 **Data availability.** The figures, which have associated raw data, include: Figures 2, 3, and  
467 4, as well as Extended Data Figures 1, 2, 3, 4, 5c, 6, 7, 8, and 9. There are no restrictions  
468 on data availability. The datasets generated and analyzed during the current study are  
469 available from the corresponding author on reasonable requests.

470 **Code availability.** The custom code and algorithm used for the 3D reconstruction and  
471 related analysis during the current study are available from the corresponding author on  
472 reasonable requests.

#### 473 **Additional references**

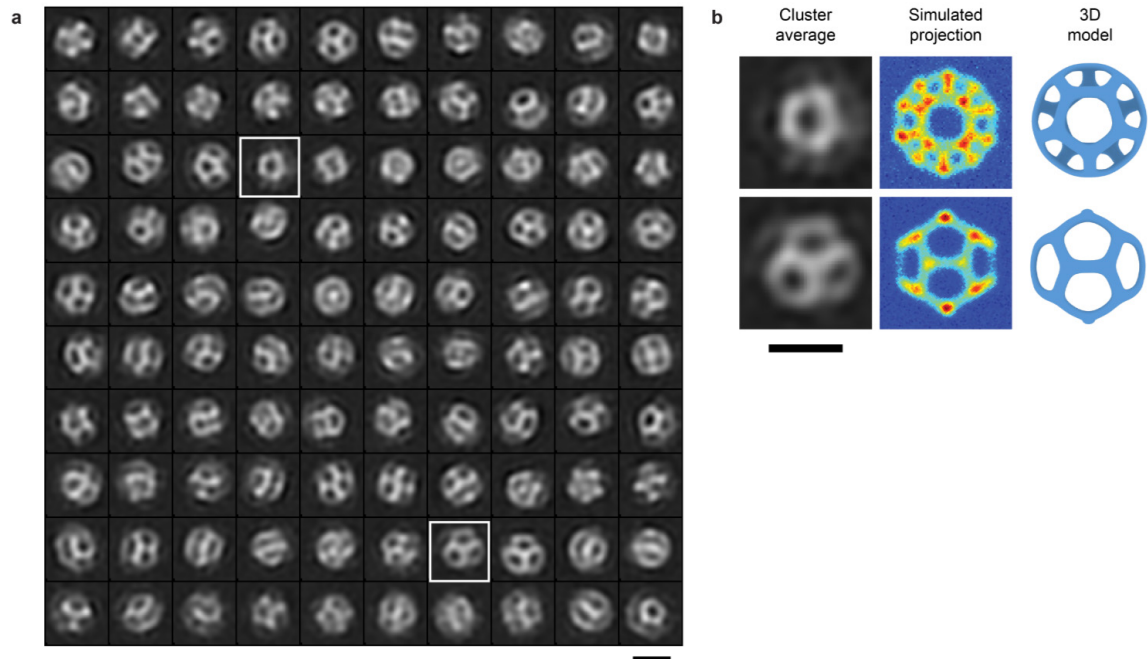
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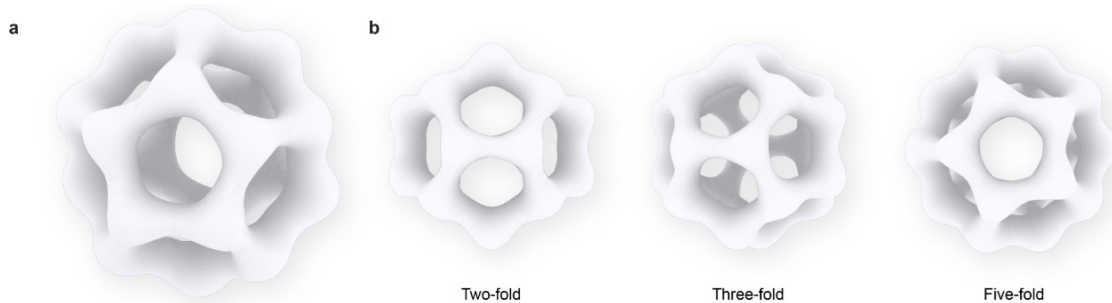


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**Extended Data Figure 1 | PEGylated silicages after cleaning and nitrogen sorption measurements on calcined cages.** **a** and **b**, Representative dry-state TEM images at different magnifications of PEGylated silicages after the removal of surfactant and TMB (Methods section). Particles in **(a)** (inset, black arrows) as well as in the insets in **(b)** exhibit cage-like structure, suggesting structure preservation after the removal of CTAB and TMB. **c**, Nitrogen adsorption and desorption isotherms of calcined silicages. After the removal of surfactant and TMB, particles were calcined at 550 °C for 6 hours in air prior to nitrogen sorption measurements. A particle synthesis yield of 67% was estimated from the weight of the calcined powder. The surface area of calcined silicages as assessed by the Brunauer–Emmett–Teller (BET) method was 570 m<sup>2</sup>/g, consistent with theoretical estimations (Methods section). Scale bars in the insets in **(b)** are 5 nm.

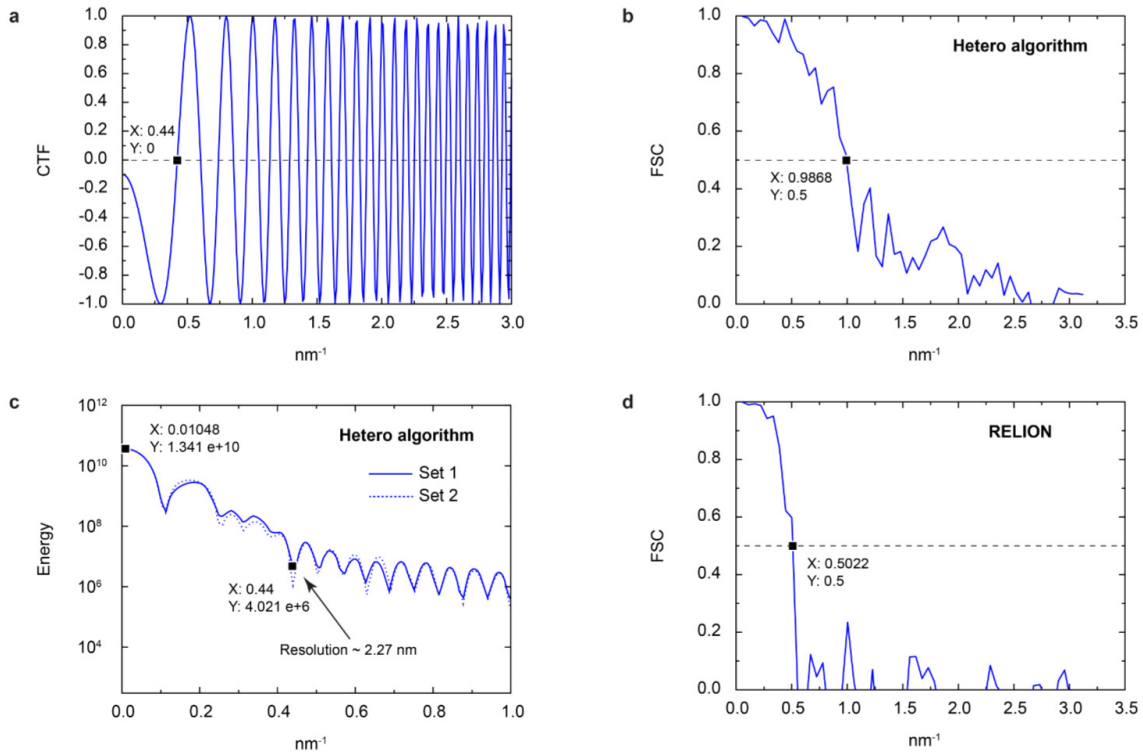


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 494 **Extended Data Figure 2 | Cluster averages of 2D images of silicages.** **a**, ~19,000  
 495 single particle cryo-EM images were sorted into 100 clusters<sup>20</sup>. **b**, Some of the  
 496 projections (examples highlighted in **a**) exhibited features similar to projections of  
 497 dodecahedral cage structure obtained by simulation. Also shown are projection models.  
 498 The scale bars are 10 nm.  
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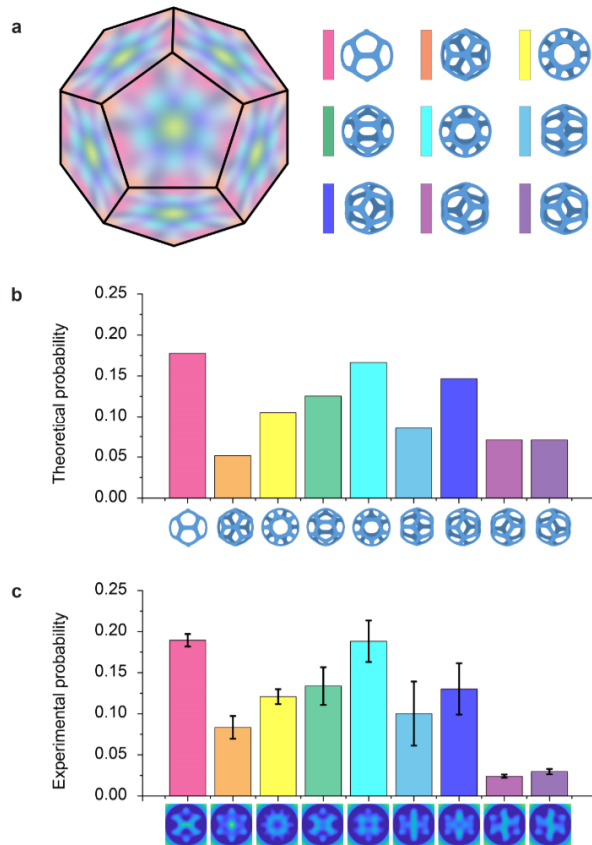
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 502 **Extended Data Figure 3 | Reconstruction of silicage using RELION 2.1 system.**  
 503 Dodecahedral silicage reconstruction result<sup>24</sup> (**a**) and its three most unique projections  
 504 along the two-, three- and five-fold symmetry axes (**b**). The reconstruction was obtained  
 505 from a single-class calculation run by RELION 2.1 using the same set of single particle  
 506 images as was used in the class of the dodecahedral cage shown in Figure 3a.  
 507 Visualization is by UCSF Chimera<sup>23</sup>.  
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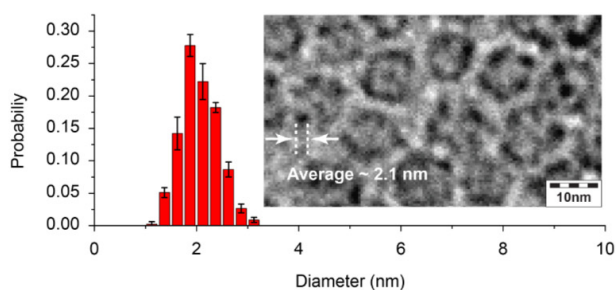
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**Extended Data Figure 4 | A typical CTF and determination of reconstruction resolution.** CTFFIND4.1.8<sup>26</sup> was used to estimate defocus for individual micrographs or set of micrographs with results consistent with the nominal defocus values of 1 to 2 microns. **a**, Contrast transfer function (CTF) for defocus 1.98 microns. Since the first zero-crossing of CTF occurs at 0.44 nm<sup>-1</sup>, the CTF has little effect on reconstructions unless the resolution is greater than 1/0.44 = 2.27 nm. **b**, Fourier Shell Correlation (FSC)<sup>25</sup> computed by standard package<sup>20</sup> for two Hetero reconstructions that are independent starting at the level of separate sets of images each containing 2000 images (*i.e.*, “gold standard” FSC). The resolution implied by the FSC curve (at 0.5 threshold) is 1/0.99 = 1.01 nm. **c**, Energy function for the same pair of reconstructions as in (b). Energy is the spherical average of the squared magnitude of the reciprocal-space electron scattering intensity, where the denominator of FSC is the square root of a product of two Energy functions, one for each reconstruction. The observations that Energy has dropped by more than 10<sup>-3</sup> times its peak value and the character of the curve has become oscillatory and more slowly decreasing, both by 0.44 nm<sup>-1</sup>, indicates that the resolution implied by the FSC curve (at 0.5 threshold) is exaggerated<sup>22</sup> and that a more conservative resolution is 1/0.44 = 2.27 nm. **d**, FSC computed by a standard package<sup>20</sup> for two RELION 2.1 reconstructions computed from the same images as the reconstructions in (b), from which the resolution (at 0.5 threshold) is estimated to be around 1/0.50 = 2.00 nm.



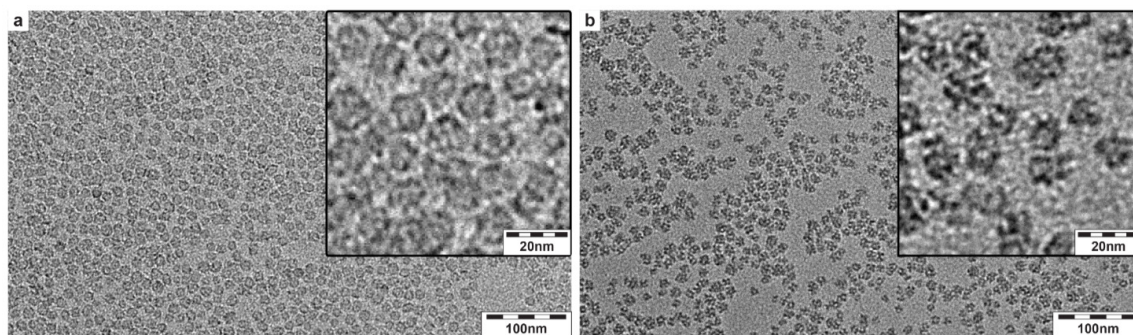
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**Extended Data Figure 5 | Probability analysis of silicage projections.** **a**, Orientation dependence of silicage projections. The orientations, at which the nine different silicage projections (right panel) can be seen, are calculated, and manually mapped on a surface of a dodecahedron (left panel). The orientations, corresponding to different projections, are assigned to different colors. **b**, Probability analysis for different silicage projections. The probability of imaging a particular projection in EM is estimated by dividing that subset of the surface area of a sphere which contains the orientations that correspond to the specific projection, by the total surface area of the sphere (**a**). **c**, Experimental probability of different silicage projections. The probability of each projection is calculated by dividing the number of the single particle images assigned to the specific silicage projection via 3D reconstruction by the overall number of silicage single particle images. The error bars in (**c**) are standard deviations calculated from three projection distributions, which were obtained from three independent reconstruction runs using different sets of single particle images, respectively.



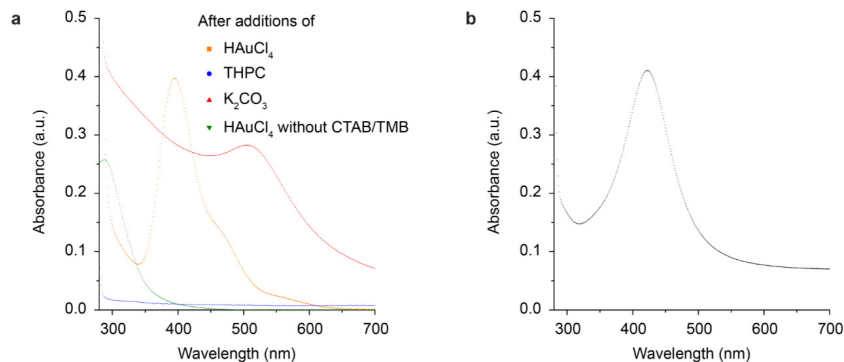
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**Extended Data Figure 6 | Size analysis of silica clusters at an early stage of cage formation.** Particle size distribution for primary silica clusters at an early stage of cage formation, obtained by manually analyzing 450 silica clusters using a set of TEM images. The measured silica clusters were randomly split into three groups, each containing 150 particles. A cluster size distribution was then obtained for each of the three groups, respectively, and the results were averaged. The error bars are standard deviations calculated from the three cluster size distributions. A representative TEM image is included in the inset. In order to quench the very early stages of cage formation, PEG-silane was added into the synthesis mixture about three minutes after the addition of TMOS thereby PEGylating early silica structures. TEM sample preparation and characterization were as described before (Methods section). Primary silica clusters with diameters around 2 nm were identified, consistent with the proposed cage formation mechanism.



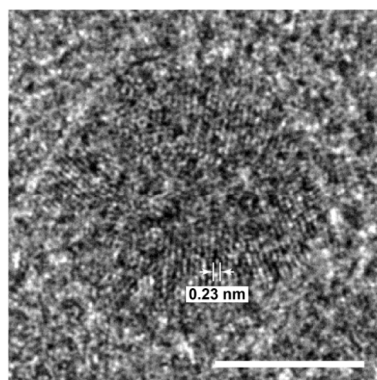
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**Extended Data Figure 7 | Role of TMB in cage formation.** TEM images at different magnifications of silica nanoparticles that were synthesized with (a) and without (b) TMB. Nanoparticles synthesized without TMB (b) exhibited stronger contrast at the particle center as compared to the nanocages (a), suggesting that these particles did not exhibit hollow cage-like structures but instead were conventional mesoporous silica nanoparticles with relatively small particle sizes (b).



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**Extended Data Figure 8 | Optical characterization of gold and silver based synthesis solutions.** **a**, Survey of the gold based synthesis showing the absorption profile of solutions after the successive additions of HAuCl<sub>4</sub> (orange), THPC (blue), one day after the addition of K<sub>2</sub>CO<sub>3</sub> (red), and compared to the same concentration of HAuCl<sub>4</sub> added to the equivalent water/ethanol solution but without any CTAB or TMB (green). **b**, Absorption profile of a solution obtained from the silver synthesis 6 hours after the addition of K<sub>2</sub>CO<sub>3</sub>.



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**Extended Data Figure 9 | High resolution TEM image of single cage-like gold nanoparticle.** The gold particle exhibited lattice fringes with a spacing of 2.3 Å, consistent with the lattice spacing between (111) planes of gold (JCPDS no. 04-0784). The scale bar is 5 nm.

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588 **Surfactant micelle self-assembly directed highly symmetric ultrasmall inorganic**  
589 **cages**

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## Supplementary Information

601 The Supplementary Information includes two supplementary videos of the 3D  
602 reconstructions obtained using Hetero algorithm and RELION 2.1 system, respectively.  
603 Both of the videos are in mp4 format, and their captions are as the following:

604

605 **Supplementary Video 1 | Reconstruction of silicage using Hetero algorithm.** The  
606 reconstruction was obtained from a single-class calculation by the Hetero algorithm using  
607 the same set of single particle images as was used in the class of the dodecahedral cage  
608 shown in Figure 3a.

609

610 **Supplementary Video 2 | Reconstruction of silicage using RELION 2.1 system.** The  
611 reconstruction was obtained from a single-class calculation by the RELION 2.1 system  
612 using the same set of single particle images as was used in the class of the dodecahedral  
613 cage shown in Figure 3a.

614