# Instantaneous bandgap collapse in photoexcited monoclinic VO<sub>2</sub> due to photocarrier doping

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Using femtosecond time-resolved photoelectron spectroscopy we demonstrate that photoexcitation transforms monoclinic  $VO_2$  quasi-instantaneously into a metal. Thereby, we exclude an 80 femtosecond structural bottleneck for the photoinduced electronic phase transition of  $VO_2$ . First-principles many-body perturbation theory calculations reveal a high sensitivity of the  $VO_2$  bandgap to variations of the dynamically screened Coulomb interaction, supporting a fully electronically driven isostructral insulator-to-metal transition. We thus conclude that the ultrafast band structure renormalization is caused by photoexcitation of carriers from localized V 3d valence states, strongly changing the screening before significant hot-carrier relaxation or ionic motion has occurred.

Since its discovery in 1959 [1], studies of the VO<sub>2</sub> phase transition (PT) from a monoclinic (M<sub>1</sub>) insulator (Fig.1, top left) to a rutile (R) metal at  $T_{\rm C}=340K$  (Fig.1, top right) have revolved around the central question [2–5] of whether the crystallographic PT is the major cause for the electronic PT or if strong electron correlations are needed to explain the insulating low-T phase. While the M<sub>1</sub> structure is a necessary condition for the insulating state below  $T_{\rm C}$ , the existence of a monoclinic metal (mM) and its relevance to the thermally driven PT is under current investigation [6–10]. In particular, the role of carrier doping at temperatures close to  $T_{\rm C}$  by charge injection from the substrate or photoexcitation has been increasingly addressed [6, 8, 11–14].

One promising approach to disentangling the electronic and lattice contributions is to drive the PT non-thermally using ultrashort laser pulses in a pump-probe scheme. Time-resolved X-ray [15, 16] and electron diffraction [14, 17] showed that the lattice structure reaches the R phase quasi-thermally after picoseconds to nanoseconds. Transient optical spectroscopies have probed photoinduced changes of the dielectric function in the THz [18–20], near-IR [9, 10, 15, 21] and visible range [21]. The nonequilibrium state reached by photoexcitation (hereinafter transient phase) differs from the two equilibrium phases, but eventually evolves to the R phase [15–26]. The observation of a minimum rise time of 80fs in the optical response after strong excitation (50mJ/cm<sup>2</sup>), described as a structural bottleneck in VO<sub>2</sub> [22], challenged theory to describe the photoinduced crystallographic and electronic PT simultaneously [13, 23].

Time-resolved photoelectron spectroscopy (TR-PES) directly probes changes of the electronic structure. Pre-

vious PES studies of  $\rm VO_2$  used high photon energies generating photoelectrons with large kinetic energies to study the dynamics of the electronic structure; however, with low repetition rate (50 Hz [25]) and inadequate time resolution (> 150fs) the ultrafast dynamics of the electronic PT was inaccessible [26]. Thus, fundamental questions - namely, how fast and why the bandgap closes and whether this happens before or simultaneously with the crystallographic PT (Fig.1 top, center) - remained unanswered.

In this Letter, we use TR-PES to directly monitor the photoinduced changes of the density of states (DOS) around the Fermi energy  $E_{\rm F}$  which define the conduction properties of VO<sub>2</sub>. We show that the insulating gap collapses during the exciting laser pulse (< 60fs) with no sign of a structural bottleneck. The transient phase is an excited mM with carrier relaxation times on the order of 200fs. This interpretation of the experimental results is supported by first-principles calculations based on manybody perturbation theory. They reveal that the bandgap in the  $M_1$  phase is extremely sensitive to small changes in the occupation of the localized d bands that alter the dynamically screened Coulomb interaction. We thereby identify the origin of the metallization: photoexcitation induces holes by depletion of the V 3d orbital population [27, 28] strongly affecting the screening and collapsing the bandgap. Finally, the analysis demonstrates that, due to their strong localization, photoinduced holes are more effective than electrons in driving bandgap renormalization. In fact, hole doping can completely close the gap without the need of a structural change thus initiating a "hole-driven insulator-to-metal transition".

The 45 nm epitaxial  $VO_2$  film on a c-cut sapphire crys-

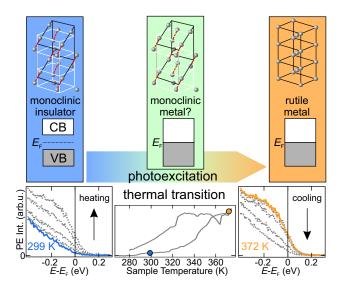


FIG. 1. (color online) Top: The photoinduced PT from the insulating  $M_1$  phase (left) to the metallic R phase (right) could proceed via a transient mM phase (green) or concurrently with the structural PT. Grey balls illustrate the V atom position. Bottom: Heating leads to the build-up of PE intensity in the bandgap (left) which is depleted upon cooling (right). Integrated PE intensity ( $-0.2 \text{eV} < E - E_F < 0.0 \text{eV}$ ) exhibits a hysteresis (center).

tal was grown at room temperature by pulsed laser ablation of a V target in an oxygen ambient [29]. For PES, it is kept under ultrahigh vacuum conditions and prepared by annealing cycles in an oxygen atmosphere. TR-PES is performed using a regeneratively amplified femtosecond laser working at a repetition rate of 40 kHz. A pump pulse (h $\nu_{\text{pump}} = 1.54\text{eV}$ ) launches the non-equilibrium dynamics and its fourth harmonic (h $\nu_{\text{probe}} = 6.19\text{eV}$ ) serves as probe pulse for photoemission. The incident pump fluence was 6.7(8) mJ/cm<sup>2</sup> (approx. 0.08 electrons per V atom) and probe fluence was kept below 8  $\mu$ J/cm<sup>2</sup> to avoid charging of insulating VO<sub>2</sub> and space-charge effects [30].

Fig.1 bottom (left/right) depicts photoelectron (PE) spectra of VO<sub>2</sub> in equilibrium at energies within the  $\approx 0.6$  eV bandgap of the insulating phase [31]. The blue curve (299K) exhibits the high-energy tail of the VO<sub>2</sub> valence band (VB). Heating the sample leads to a buildup of intensity (dotted curves, left) and a Fermi-Dirac (FD) like spectrum centered around  $E_{\rm F}$  at 372K (orange curve). This thermally induced PE intensity is suppressed upon cooling (dotted curves, right). The spectral weight below  $E_{\rm F}$  follows a hysteresis centered at 330K with a width of 25K (center panel) in line with optical experiments [30]. The difference between the high- and low-temperature PE spectra is plotted in Fig.2b (green). Due to the excellent agreement of the thermally induced change of PE intensity with the parameter-free FD distribution for 372K (black) [32], we conclude that, using

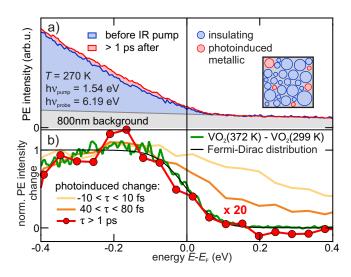


FIG. 2. (color online) (a) PE spectra before (blue) and >1ps after pump (red,  $6.7(8) \mathrm{mJ/cm^2}$ ). Inset: Illustration of partial switching of selected domains. b) Comparison of thermally induced (green) and persistent photoinduced change (yellow, orange, red). The latter spectra were binned in energy at a step size of  $\Delta E = 45 \mathrm{meV}$ .

 $h\nu_{\rm probe} = 6.19 {\rm eV},$  we are sensitive to the electronic PT in VO<sub>2</sub>.

In order to elucidate how the photoinduced electronic PT evolves, we perform time-resolved PES. We expect photoinduced changes on the order of 1-10% of the thermally induced change, as reported for optical experiments at comparable pump fluences [10, 15, 24]. This is because only parts of the probed volume are transformed into the transient phase (Fig.2a, inset) at excitation densities below the saturation regime ( $F_{\rm sat} \approx 4 \cdot F_{TH}, F_{TH}$ threshold fluence for the PT) [14, 22, 24] similar to the thermally driven PT [33, 34]. While optical experiments probe photoinduced changes due to carrier dynamics in the conduction band (CB) and VB of insulating VO<sub>2</sub> and close to  $E_{\rm F}$  of (potentially photoexcited) metallic  $VO_2$ , the energy selectivity of PES permits us to exclusively monitor photoinduced changes due to metallization by probing the dynamics in the gap of insulating  $VO_2$ . Fig.2a shows PE spectra before (blue) and > 1ps after the pump pulse. Indeed, the photoinduced change is very small. The difference is plotted in panel b (red markers) and compared to the thermally induced metallic spectral function (green).

As in optical experiments at comparable excitation fluences (see, e.g., Ref.24), the photoinduced signal is considerably smaller than the thermally induced one (here 5 %). Yet, the curves show remarkable agreement, implying that the pump pulse has metallized individual grains of the sample (Fig.2a, inset). Moreover, the signature of this transient metallic phase is practically identical with the thermally switched rutile VO<sub>2</sub>. This direct observation of metallicity (defined by the presence of a Fermi

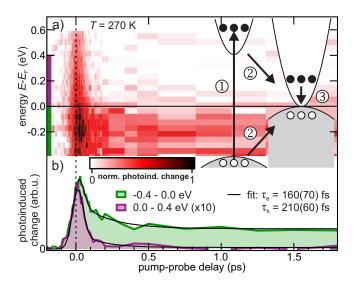


FIG. 3. (color online) a) PE intensity change versus pumpprobe delay close to  $E_{\rm F}$ . Photoelectrons are detected immediately in the gap, showing the quasi-instantaneous collapse (cf. cartoon, energy axis is to scale). Integration of the PE intensity above (purple) and below (green)  $E_{\rm F}$  yields the respective transients in b). The empirical fit (black) quantifies the averaged hot electron (hole) lifetimes [30].

edge) in photoexcited  $\mathrm{VO}_2$  on ultrafast timescales goes beyond optical probes of metal-like dielectric functions, because those can be influenced strongly by highly excited electron-hole plasma in the CB and VB or structural changes that alter the dielectric function.

Fig.3 presents the ultrafast dynamics of the photoinduced electronic PT. The pump-induced change of PE intensity is depicted in false colors a) and is characterized by a fast component (fs timescales) and a longlived intensity below  $E_{\rm F}$ , which is spectrally equivalent to the photoinduced change in Fig.2b. More precisely, the pump-induced intensity below  $E_{\rm F}$  represents the spectral signature of the transient metallic phase. Fully established at 1ps, it is never modulated by coherent oscillations and remains unchanged for up to 400ps (not shown). This is noteworthy, as some time-resolved diffraction experiments on VO<sub>2</sub> demonstrate an evolution of the atomic lattice over nanoseconds [14–17]. The invariance of the TR-PE spectra on picosecond timescales shows that intermediate steps of the crystallographic PT have no effect on the FD distribution observed here, implying that the photoinduced electronic PT, i.e., the bandgap collapse, is completed before 1ps elapses.

Note that the photoinduced *change* of the TR-PES signal in the gap below  $E_{\rm F}$  results *only* from individual pump-induced metallized crystal domains, as photoexcited carrier dynamics in insulating VO<sub>2</sub> would occur at different energies (in CB and VB). Thus, the PE intensity above  $E_{\rm F}$  (purple bar in Fig.3a) corresponds to excited electrons in the transient (metallic) phase and PE intensity below  $E_{\rm F}$  (green bar) to dynamics in the

occupied electronic band structure. The latter cannot result from defect states in the gap, as these would already be occupied in equilibrium. The temporal evolution of integrated PE intensity in these energy windows is shown in Fig.3b. Both traces are well fit by single exponential decays (bl[35]ack curves) with constant offsets [36], convolved with the laser pulse envelope (duration: 61(5)fs [30]). The fits yield average decay constants  $\tau_e = 160(70)$ fs and  $\tau_h = 210(60)$ fs. Importantly, the PE intensity in the gap is observed quasi-instantaneously at photoexcitation. We do not observe a delayed rise of intensity below  $E_{\rm F}$  with a timescale of 80fs as expected for a structural bottleneck [22, 30]. On the contrary, the photoinduced PE intensity decreases on a timescale of 210fs as expected for hole relaxation towards  $E_{\rm F}$  from lower energies. Fig.2b depicts difference PE spectra at t = 0fs and 60fs. They clearly display transiently occupied states close to  $E_{\rm F}$ , superimposed by the lifetimeless intensity from two-photon absorption through virtual states. Therefore, the electronic PT occurs with the photo excitation of the *electronic* structure and precedes any significant ionic motion towards the R phase.

In order to identify the physical mechanism for ultrafast metallization of photoexcited VO2, we performed first-principles calculations of the quasiparticle DOS within a many-body Green's-function approach [30, 37]. We adopted the GW approximation for the self-energy  $\Sigma$  [38], because quasiparticle self-consistent GW, which naturally accounts for the localized character of the V 3d electrons, yields reliable quasiparticle band structures [39–41] compared to experiments [31]. Alternatively, cluster DMFT (dynamical mean-field theory) is also able to describe the electronic structure of monoclinic VO<sub>2</sub> [5, 42]. In the GW approximation  $\Sigma$  is given by the product of the one-particle Green's function G and the dynamically screened Coulomb interaction  $W(\omega) = \epsilon^{-1}(\omega)v$ . Here v is the bare Coulomb interaction and  $\epsilon^{-1}$  is the inverse dielectric function calculated in the random-phase approximation including electronhole and plasmon excitations [30].

The abruptness of the experimentally observed gap collapse justifies a Born-Oppenheimer approach with a "frozen lattice". In this spirit, we redistribute a portion of VB electrons equivalent to the experimental excitation density (0.075 electrons per V atom) to the unoccupied states. Note that our findings are robust with respect to excitation density [30]. We then calculate the screened interaction  $\Delta W$  that is changed by the presence of the additional carriers and the quasiparticle DOS with the self-energy  $\Delta \Sigma = G\Delta W$  [30, 43]. This redistribution of the electron and hole populations in the VB and CB is sufficient to lead to a collapse of the bandgap (Fig. 4a). In contrast to ordinary semiconductors where freecarrier doping leads to a moderate bandgap narrowing [43–47] and never results in a complete bandgap collapse purely electronically [47], this extreme sensitivity of VO<sub>2</sub>

to changes of the V 3d occupation is a distinctive and unique property of correlated materials [48].

The dynamical screening  $e^{-1}(\omega)$  in fact increases significantly in the low-energy region (< 1.5eV, Fig. 4b) due to creation of new VB-VB and CB-CB intraband electron-hole channels in the photoexcited system, while it remains almost unchanged in the high-energy region. This finding is robust with respect to variations of the charge redistributions involving the depopulation of the V 3d bands [30]. In order to unravel the microscopic mechanism at the origin of the bandgap collapse, we separately analyze the effect of changing the occupations of only VB or CB. We find that hole doping at the top of VB alone indeed induces the bandgap breakdown [30] as suggested before [49]. We rationalize these findings by the fact that VO<sub>2</sub> possesses an almost non-dispersive top VB corresponding to localized V 3d states [39, 50]. Population changes of these states strongly enhance lowenergy screening, leading to instantaneous metallization (bandgap closure). Note that pure electron doping also leads to a reduction of the bandgap but without metallization (no bandgap closure) [30].

The relevant elementary processes are sketched in the inset of Fig.3a. Absorption of pump photons lifts localized electrons from the top VB into the CB of insulating VO<sub>2</sub>. This photocarrier doping causes an instantaneous breakdown of the gap [51] and excited electrons and holes subsequently relax at a slower rate towards equilibrium at  $E_{\rm F}$ . The experimentally determined hot carrier relaxation times of order 200fs agree nicely with observations from other experiments. A similar time constant characterizes the incoherent time-dependent response of the conductivity in THz measurements of Pashkin et al. [20], which may well originate from excited carriers in the CB of VO<sub>2</sub>. Also, pump-probe experiments of the transient phase revealed that the optical response of photoexcited VO<sub>2</sub> starts to resemble that of the thermally metallized sample after 200fs [24]. Photoexcitation of  $VO_2$  creates an excited metal whose optical and electronic properties become similar to those of the thermally driven material only after the hot carriers have equilibrated. It is possible that the subsequent evolution of photoexcited VO<sub>2</sub> towards the R phase occurs quasi-thermally, as the hot carriers thermalize with the lattice and heat it above  $T_{\rm C}$ . Unlike the thermal PT, where lattice distortion and Coulomb interaction cooperatively drive the formation of the insulating gap, photoexcitation of the electronic system instantaneously modifies the electronic correlations causing the gap collapse, which is subsequently stabilized by the structural evolution.

In conclusion, the present experimental and theoretical work provides a comprehensive description of the elementary steps of the photoinduced electronic PT in  $VO_2$ . The bandgap of the insulating phase collapses instantaneously upon photoexcitation due to carrier doping, revealing an ultrasensitivity of  $VO_2$  to variations of

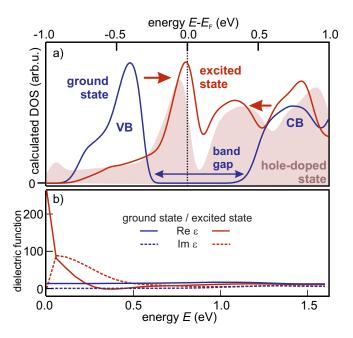


FIG. 4. (color online) a) Calculated DOS (integrated over the entire Brillouin zone) of the  $M_1$  ground (blue) and excited states (red) broadened by the experimental resolution (90 meV). b) Real and imaginary parts of the dielectric function for a representative small momentum transfer  $\mathbf{q} = (1/6, 0, 0)$ .

the screening by holes at the top of the V valence bands. The bandgap collapse is followed by hot carrier relaxation in the transient metallic phase on a 200fs timescale and quasi-thermal evolution of the system towards the high-T phase. The abrupt vanishing of the bandgap proves the absence of a structural bottleneck in the photoinduced electronic PT. Moreover, the electronic PT precedes the timescales observed for the crystallographic PT [16, 17] in excellent agreement with recent observations of a photoexcited mM-like VO<sub>2</sub> [14]. These new insights into the character of the isostructural insulator-to-metal transition in VO<sub>2</sub> provide not only a novel understanding of the physical mechanisms of the phenomenon, but also establish the basis for new experimental and theoretical studies of photoinduced dynamics in the transient metallic phase of  $VO_2$  as well as other correlated materials.

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## Supplementary materials

(Dated: August 28, 2018)

# I. EXPERIMENTAL DETAILS: SAMPLE GROWTH

The  $VO_2$  film was deposited in an Epion PLD-3000 system using a Lambda Physik (Coherent COMPex 205) excimer laser (248 nm (KrF), 4 J/cm<sup>2</sup> per pulse, 25 Hz repetition rate, and nominal 25 ns pulse duration). Prior to deposition, the chamber was pumped down to 9x10<sup>-6</sup> Torr. A pure vanadium metal target was ablated at room temperature in an ultra-high purity oxygen ambient  $(1.1 \times 10^{-2} \text{ Torr}, 2 \text{ sccm flow rate})$ . The laser beam was rastered across the rotating target while the substrate holder also rotated. The average deposition rate was 0.3 Å/s and the film thickness (nominal 45 nm) was verified using a Dektak profilometer. The deposited film was annealed inside a tube furnace in 250 mTorr of O<sub>2</sub> at 723 K for 10 minutes. After annealing, the film was allowed to cool before exposing them to ambient conditions.

White-light transmission hysteresis measurements were taken to characterize the transmitted light as a function of temperature, controlled with a Peltier heater. Light from the near-blackbody tungsten lamp was focused onto the sample using a 5x~(NA~0.20) microscope objective; the light transmitted through the film was collected with a 5x~(NA~0.12) microscope objective and focused onto an InGaAs detector. The resulting optical hysteresis curve shows a switching contrast of 0.32, critical temperature of 326 K, and hysteresis width of 6 K, consistent with Beer's law for a stoichiometric and switching  $VO_2$  film and this spectral source.

# II. EXPERIMENTAL DETAILS: PHOTOELECTRON SPECTROSCOPY

The experiments were performed at a repetition rate of the laser of 40 kHz, as the photoexcited VO<sub>2</sub> did not fully recover upon arrival of the next pulse pair at higher repetition rates. In contrast to optical experiments under ambient conditions, the reduced heat transport in UHV gave, even at the low repetition rate, an upper limit to the incident pump fluence of 7 mJ/cm<sup>2</sup>. The  $VO_2$  sample was annealed in an oxygen atmosphere at  $10^{-4}$  mbar (T = 600 K, 30 min) until the photoemission (PE) spectra exposed the thermal phase transition (see Fig.1 in the main text). The emitted photoelectrons were detected using a hemispherical electron energy analyzer that was held at a fixed bias voltage of -0.5 eV with respect to the sample holder. All data shown are angle-integrated  $(\pm 7^{\circ})$ . Photoelectron spectra are plotted as a function of energy with respect to the equilibrium Fermi level of the sample, which is in electrical contact with the sample holder.

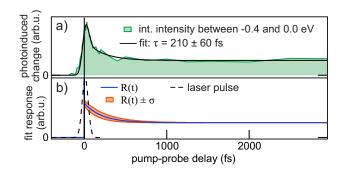


FIG. S1. Population dynamics and fits

In contrast to traditional ARPES experiments using high photon energies, our experiments enable the clear distinction of sample charging and space charge effects that result from the cloud of photoemitted electrons: Using low photon energies, we always detect the complete PE spectrum starting at the low energy cut-off of secondary electrons with zero kinetic energy up to the electrons photoemitted from the Fermi energy  $E_{\rm F}$ . While space charge effects generally lead to a broadening of the whole spectrum (fast electrons are accelerated, slow electrons become slower), charging of the sample leads to a shift of the entire spectrum to higher or lower energies due to negative (positive) charging. Both processes depend on the photon density of the probing light pulse; the latter also depends on temperature (as it influences the electonic conductivity). To avoid such effects of the insulating VO<sub>2</sub>, experiments were performed at sufficiently low probe photon flux  $(8 \mu J/cm^2)$  and sufficiently high temperatures (still  $\langle T_{\rm C}\rangle$  in order to maintain ample conductivity.

### III. FITTING OF TIME-RESOLVED DATA

The population dynamics (e.g. green curve in Fig. S1a) are empirically fitted with the function

$$R_1(t) = \Theta(t) \cdot (a + b \cdot e^{-t/\tau}) + c \cdot \delta(t) \tag{1}$$

where the first term is a single exponential decay with a constant offset starting at time zero (blue curve in Fig. S1b) and the second term accounts for photoelectrons emitted after absorption of two photons via (virtual) states without lifetime. Convolution with the pump and probe laser pulses' envelopes (dashed curve) yields the black fit in Fig. S1a. Their cross correlation (two Gaussians) is determined at high energies, and has a width of 89 fs, leading to a mean pulse duration of pump and probe pulses of 63 fs, which gives a conservative upper limit for the error of the decay time determination.

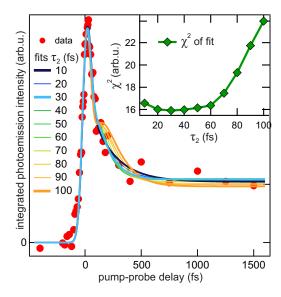


FIG. S2. Fitting of the rise time

Fig. S1 illustrates the deviation that results from a 60 fs error with the other fit parameters fixed (red).

In order to test, whether the data is consistent with the 80 fs structural bottleneck, we also fitted the empirical function

$$R_2(t) = \theta(t) \left( A_1 e^{-t/\tau_1} - A_2 e^{-t/\tau_2} + C \right) + D \cdot \delta(t)$$
 (2)

to the time-dependent change of PE intensity below  $E_{\rm F}$ . The results are shown in Fig. S2. The first term empirically describes the population dynamics below  $E_{\rm F}$ while the second one accounts for the two-photon PE spike at t = 0. Again, this function was convolved with the laser pulses' envelope as discussed above. Assuming that the DOS is zero before the pump pulse arrives, the population response must be zero at t=0 and then rise  $(\tau_2)$  as the structural bottleneck is passed. The decay  $(\tau_1)$  accounts for possible hot hole relaxation towards  $E_{\rm F}$ . Thus,  $A_1 - A_2 + C = 0$  is a boundary condition for this fit. Fig. S2 shows a series of different fits with fixed rise times  $\tau_2$  and the corresponding  $\chi^2$  in the inset. Obviously, the fits yield satisfactory results only for rise times below 60 fs, i.e. below the structural bottleneck time of 80 fs that was observed in optical experiments<sup>1</sup>.

#### IV. THEORETICAL FRAMEWORK

In the present work we use the GW self-energy approximation on top of a density-functional theory (DFT) calculation of VO<sub>2</sub> to account for correlation effects. This approach is known to provide an accurate description of both insulating and metallic phase of VO<sub>2</sub><sup>2</sup>. In the GW approximation (GWA)<sup>3</sup> the self-energy  $\Sigma$  is given by the

convolution in frequency space:

$$\Sigma(\mathbf{r}_1, \mathbf{r}_2, \omega) = \frac{i}{2\pi} \int d\omega' e^{i\eta\omega'} G(\mathbf{r}_1, \mathbf{r}_2, \omega + \omega') W(\mathbf{r}_1, \mathbf{r}_2, \omega')$$
(3)

between the time-ordered one-particle Green's function G:

$$G(\mathbf{r}_1, \mathbf{r}_2, \omega) = \sum_{i} \frac{\phi_i(\mathbf{r}_1)\phi_i^*(\mathbf{r}_2)}{\omega - \epsilon_i + i\eta \operatorname{sgn}(\epsilon_i - \mu)}, \quad (4)$$

(where  $\mu$  is the Fermi energy and  $\eta \to 0^+$  ) and the dynamically screened Coulomb interaction W:

$$W(\mathbf{r}_1, \mathbf{r}_2, \omega) = \int d\mathbf{r}_3 \, \epsilon^{-1}(\mathbf{r}_1, \mathbf{r}_3, \omega) v(\mathbf{r}_3 - \mathbf{r}_2). \tag{5}$$

Here v is the (static) bare Coulomb interaction and  $\epsilon^{-1}$  is the inverse dielectric function that describes the screening of v through electron-hole and collective plasmon excitations. In the GWA  $\epsilon^{-1}$  is calculated in the random-phase approximation (RPA) from the independent-particle polarizability  $\chi_0$ :

$$\epsilon^{-1}(\omega) = 1 + v \frac{\chi_0(\omega)}{1 - v\chi_0(\omega)}.$$
 (6)

(Here integrations/inversions over spatial coordinates are implicitly understood.) Finally, the (time-ordered) independent-particle polarizability  $\chi_0$  is:

$$\chi_0(\mathbf{r}_1, \mathbf{r}_2, \omega) = \sum_{i,j} (f_i - f_j) \frac{\phi_i(\mathbf{r}_1) \phi_j^*(\mathbf{r}_1) \phi_i^*(\mathbf{r}_2) \phi_j(\mathbf{r}_2)}{\omega - (\varepsilon_i - \varepsilon_j) + i\eta \operatorname{sgn}(\varepsilon_i - \varepsilon_j)},$$
(7)

where  $f_i$  are the occupation numbers.

In the standard approach, G and W are constructed using the Kohn-Sham eigenvalues and wavefunctions obtained in a DFT calculation with local density approximation (LDA). However, such calculations fail to reproduce the insulating nature of the monoclinic VO<sub>2</sub> (M<sub>1</sub> phase)<sup>2</sup>. Then we need to rely on more sophisticated orbital-dependent potentials to capture the correct electronic structure of both insulating and metallic phase. Thus in this work, the quasiparticle (QP) wavefunctions  $\phi_i$  entering Eqs. (4)-(7) are obtained from a QP selfconsistent COHSEX calculation (see Refs. 2 and 4). COHSEX is a static approximation to the GWA selfenergy (Eq. (3)) that is given by the sum of a Coulombhole (COH) and a screened-exchange (SEX) terms<sup>3</sup>. The QP energies  $\varepsilon_i$  appearing in Eq. (4) are then calculated self-consistently by solving the QP equation

$$\left(-\frac{\nabla^2}{2} + V_{ext}(\mathbf{r}_1) + V_H(\mathbf{r}_1)\right) \phi_i(\mathbf{r}_1) + \int d\mathbf{r}_2 \Sigma(\mathbf{r}_1, \mathbf{r}_2, \varepsilon_i) \phi_i(\mathbf{r}_2) = \varepsilon_i \phi_i(\mathbf{r}_1),$$
(8)

following a  $GW_0$  scheme (i.e.  $\phi_i$  and W are kept fixed at the COHSEX level). In Eq. (8)  $V_H$  is the Hartree potential and  $V_{ext}$  is the electron-ion interaction. At self-consistency, the QP energies  $\varepsilon_i$  determine the QP density

of states (DOS), integrated over the whole Brillouin zone, that is plotted in the following figures (also the one-shot  $G_0W_0$  results on top of COHSEX do not differ qualitatively from the  $GW_0$  DOS).

### V. NUMERICAL DETAILS

Here, we have used the Abinit code to perform the calculations<sup>5</sup>. We have adopted the experimental crystal lattice of the monoclinic  $M_1$  structure<sup>6</sup> (atomic positions are kept frozen) and used Troullier-Martins pseudopotentials  $^{7}$  (with V 3s and 3p explicitly treated as valence electrons). A plane-wave basis set with cutoff of 180 Hartree has been employed. COHSEX results are obtained from Ref. 2. Convergence has been achieved using 200 bands for the calculation of the screening, and 150 bands in the calculation of self-energy corrections. We have used 5007 plane waves to expand the wavefunctions entering the screening and 14999 plane waves for the wavefunctions entering the self-energy. We have calculated the frequency convolution in Eq. (3) using an accurate contour-deformation technique with 60 frequencies along the real axis up to 1.5 Hartree and 10 frequencies along the imaginary axis. In the main part of the present article, the DOS are calculated with a  $6\times6\times6$   $\Gamma$ -centered grid of k points. The results discussed in the following of this Supplementary Material are instead based on less expensive calculations performed with a  $4\times4\times4$  grid of **k** points that provides the same qualitative results as the more converged calculations with the  $6 \times 6 \times 6$  k-grid.

In the time-resolved photoemission (TR-PE) experiment, the laser pump excites 0.08 electrons per V atom (i.e. 0.32 electrons per monoclinic unit cell), partially depleting V 3d valence states and partially occupying V 3d conduction states above the gap (the O 2p states instead cannot be excited for energy-conservation reasons: the top of the O 2p band has a binding energy that is larger than the energy of the laser pump). In order to simulate pump-probe TR-PE experiments, formally one should solve the non-equilibrium Baym-Kadanoff (BK) equations<sup>8</sup>. However, for the goals of the present work we investigate the effect of the photoinduced instantaneous modification of the occupations  $f_i$  on the measured QP DOS through the change of the screened Coulomb interaction  $\Delta W$ , which is one of the ingredients in the solution of the BK equations and it provides a physical understanding of the microscopic process leading to the bandgap collapse discussed in the main text. In fact, from Eq. (7) we see that variations of the  $f_i$  in the photo excited states directly affect  $\chi_0$  and hence the screening  $\epsilon^{-1}$  through Eq. (6). The QP DOS are recalculated using the GWA with a  $\Delta W$  modified according to different occupation distributions  $f_i$ , i.e. with  $\Sigma = G\Delta W^9$ . The differences between the resulting DOS are entirely due to different screened Coulomb interactions  $\Delta W$  originating from different occupation distributions  $f_i$ .

#### VI. EXTENDED ANALYSIS OF THE THEORETICAL CALCULATIONS

We have calculated the QP DOS with some representative occupation distributions to analyze in detail the effect of the photoexcitation on the screening (neglecting electron-hole interactions). The results are shown in Fig. S3. In particular, Fig. S3(a, center) shows the QP DOS for the monoclinic  $M_1$  phase in the ground state (i.e. before the photoexcitation) calculated in the GWA, reproducing previous results<sup>2</sup>. It displays a gap of  $\sim 0.6$  eV between the top-valence and bottom-conduction V 3d states, in agreement with earlier PE spectra<sup>10</sup>. In the ground state all the valence states below the Fermi level, comprising V 3d states and O 2p at lower energy, are fully occupied ( $f_i = 2$ ) and those above the gap are fully empty ( $f_i = 0$ ), see Fig. S3(a, left).

Fig. S3(b)-(f)(center) show the QP DOS obtained for different excited occupation distributions [see Fig. S3(b)-(f)(left)], in all of which 0.075 electrons per V atom are excited from valence to conduction states. In Fig. S3(b)-(c) the occupation distributions have the same valence and conduction quasi-Fermi levels, but different effective electronic temperatures. In Fig. S3(d)-(f) the distributions are non-thermal.

In all cases with excited occupation distributions, we find important changes in the QP DOS with respect to the ground state: there is always at least a conspicuous bandgap narrowing. In Fig. SS3(b)-(d) we even observe a complete bandgap collapse. This finding is an evidence of the very high sensitivity of the  $\rm VO_2$  band structure with respect to changes in the V  $\rm 3d$  occupations.

In order to understand the microscopic mechanism that leads to the bandgap collapse, in Fig. S3(a)-(f)(right) we have also plotted the real and imaginary parts of the dielectric function  $\epsilon(\mathbf{q},\omega)$ , calculated in RPA in the low-energy range ( $\omega < 1.0 \text{ eV}$ ) for a small momentum transfer  $\mathbf{q} = (1/6, 0, 0)$  (for other  $\mathbf{q}$  similar considerations apply). In the ground state, see Fig. S3(a), both Re  $\epsilon$  and Im  $\epsilon$  are flat. In fact, in RPA there cannot be electron-hole transitions at energies below the fundamental direct bandgap ( $\sim 0.8 \text{ eV}$ ). The change in the occupations, see Fig. S3(b)-(f), develops new peaks in  $\operatorname{Im} \epsilon$  for  $\omega < 1.0 \text{ eV}$  that are linked to the opening of new electron-hole excitation channels at energies smaller than the ground-state bandgap (Re  $\epsilon$  is analogously modified as it is connected to  $\operatorname{Im} \epsilon$  by Kramers-Kronig relations). This can be immediately realised from Eq. (7): the independent-particle polarizability  $\chi_0$  have new poles for transitions between pairs of states for which  $f_i \neq f_j$ .

The new peaks in  $\operatorname{Im} \epsilon$  are mainly due to transitions within the valence band (VB) and within the conduction band (CB) ("intraband transitions"). In fact, when the new occupation distribution remains flat, see Fig. S3(f)(left), smaller structures are present in  $\operatorname{Im} \epsilon$ , see Fig. S3(f)(right), which are only due to transitions from partially filled CB to completely empty CB. In turn this leads to a smaller bandgap narrowing. This holds

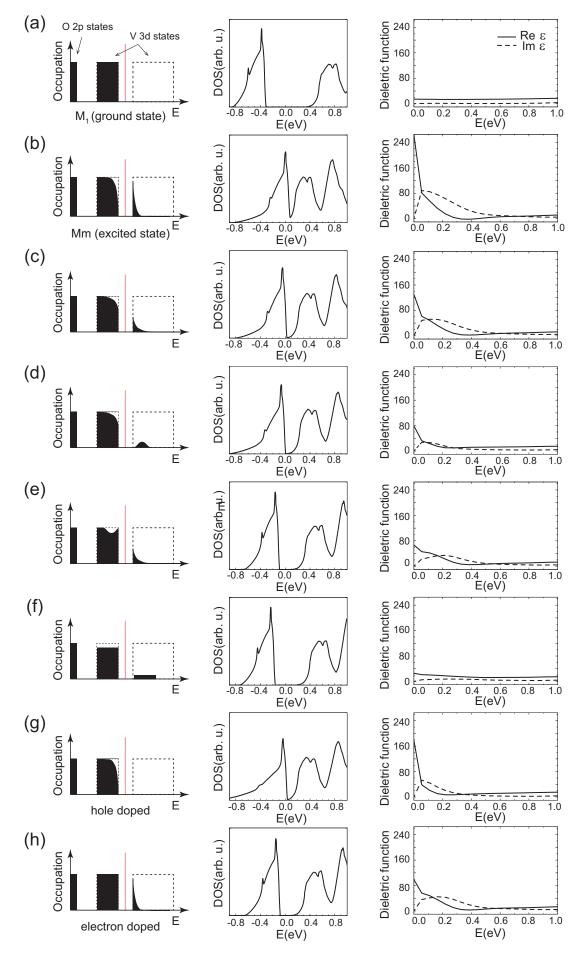


FIG. S3. (Left) Occupation distribution used to calculate the screened Coulomb interaction. (Center) Quasiparticle density of states obtained from the given occupation distribution. (Right) Real and imaginary parts of the dielectric function calculated in RPA at  $\mathbf{q} = (1/6, 0, 0)$ . The ground state is represented in panel (a), photoinduced excited states in panels (b)-(f), hole and

to a certain extent also for Fig. S3(e), where holes are created in the middle and not at the top of the VB. On the contrary, whenever intraband transitions are made possible, the screening is largely modified, leading to the bandgap collapse [independently of the electronic temperature, compare Figs. SS3(b)-(c)]. This finding is confirmed also when the excited electronic charge is larger, for example 0.1 and 0.125 electrons per V atom, see Fig. S4.

The overall picture that emerges from this analysis is clear: the change of the occupations allows intraband transitions to modify the low-energy screening of the Coulomb interaction. In turn, the modified  $\Delta W$  leads to a dramatic change of the electronic structure, revealing a high sensitivity of the electronic properties of VO<sub>2</sub> to external changes.

In order to analyze the effect of intraband transitions within the VB and the CB separately, we have followed two paths: (i) select the occupation distribution of Fig. S3(b) and perform calculations where we remove intraband transitions within the CB or the VB from  $\chi_0$  in Eq. (7); (ii) perform calculations with hole doping or electron doping with 0.075 electrons per V atom removed from the top VB or added to the bottom CB. These two approaches differ slightly by some interband transitions. The results of approach (i) are shown in Fig. S5(a)-(c). When all the intraband transitions are suppressed, see Fig. S5(c), the peak in  $\epsilon$  in the low-energy regions disappears. In this case, the bandgap remains open and is almost the same as in the ground state [see Fig. S3(a)]. When intraband transitions are allowed within the CB

only, see Fig. S5(b), the bandgap is reduced by 0.46 eV but not closed. Finally, when intraband transitions are allowed within the VB only, see Fig. S5(a), the bandgap collapses, nearly retrieving the result of the total calculation, see Fig. S3(b).

Therefore, this analysis clearly shows that intraband transitions within VB play a crucial role in changing the screening and closing the bandgap. This is due to the fact that the top VB states are largely non-dispersive owing to their effective electronically one-dimensional character which is linked to the bonding states of the V dimers along the c-axis<sup>11</sup>. This peculiar property gives rise to an intense peak in the low-energy region of Im  $\epsilon$  when intraband channels are activated by the electron excitation.

While the intraband transitions are artificially suppressed in approach (i), the realistic means to realize those conditions in experiments are through electron or hole doping, which are simulated in approach (ii). The results of approach (ii) for an electron/hole density of 0.075 carriers per V atom are shown in Fig. S3(g)-(h). In accordance with approach (i), we find that while for the pure electron doping the bandgap is strongly reduced but not completely closed, the change in the dielectric function due to hole doping is sufficient to close the bandgap. The same result holds for the higher electron/hole densities discussed in Fig. S4, namely hole doping closes the bandgap but electron doping leads to a positive gap of about 0.1 eV. These findings thus suggest that, in addition to the photoinduced phase transition reported in this work, alternative experimental approaches employing pure hole doping also provoke metallization of  $VO_2$ .

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The study of the effects due to variations of  $f_i$  on the Hartree potential  $\Delta V_H$ , for the change of the electron density, and on the Green's function  $\Delta G$ , for the change of the pole structure in Eq. (4), is left for a future work. We do not expect the conclusions of the present analysis will change once those contributions are included.

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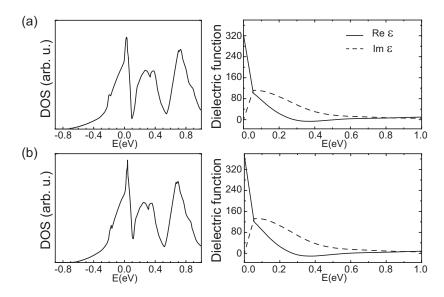


FIG. S4. Same as Fig. S3(b), here with a larger number of excited electrons: (a) 0.1 and (b) 0.125 per V atom.

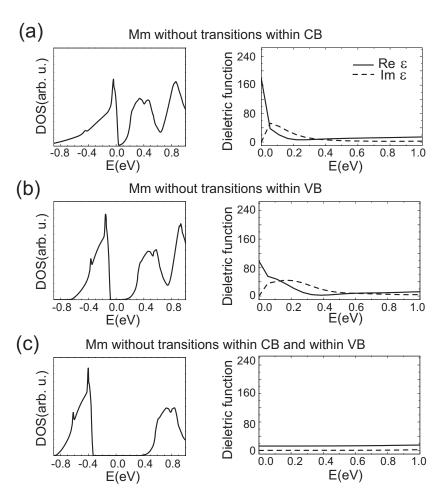


FIG. S5. Starting from the occupation distribution of Fig. S3(b), intraband transitions have been suppressed (a) within the CB, (b) within the VB, and (c) in both the VB and the CB.