

Research Article

Flame-Retardant and Thermal Degradation Mechanism of Caged Phosphate Charring Agent with Melamine Pyrophosphate for Polypropylene

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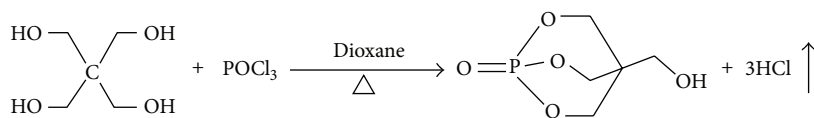
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An efficient caged phosphate charring agent named PEPA was synthesized and combined with melamine pyrophosphate (MPP) to flame-retard polypropylene (PP). The effects of MPP/PEPA on the flame retardancy and thermal degradation of PP were investigated by limiting oxygen index (LOI), vertical burning test (UL-94), cone calorimetric test (CCT), and thermogravimetric analysis (TGA). It was found that PEPA showed an outstanding synergistic effect with MPP in flame retardant PP. When the content of PEPA was 13.3 wt% and MPP was 6.7 wt%, the LOI value of the flame retardant PP was 33.0% and the UL-94 test was classed as a V-0 rating. Meanwhile, the peak heat release rate (PHRR), average heat release rate (AV-HRR), and average mass loss rate (AV-MLR) of the mixture were significantly reduced. The flame-retardant and thermal degradation mechanism of MPP/PEPA was investigated by TGA, Fourier transform infrared spectroscopy (FTIR), TG-FTIR, and scanning electron microscopy-energy dispersive X-ray spectrometry (SEM-EDXS). It revealed that MPP/PEPA could generate the triazine oligomer and phosphorus-containing compound radicals which changed the thermal degradation behavior of PP. Meanwhile, a compact and thermostable intumescent char was formed and covered on the matrix surface to prevent PP from degrading and burning.

1. Introduction

In recent years, intumescent flame retardants (IFRs) have aroused more and more attention in flame retardant polypropylene (PP), due to their being environmental-friendly, low smoke and antidripping [1–3]. Typically, an IFR system is composed of three components: an acid source, a charring agent, and a blowing agent [4]. The most reported IFR system is ammonium polyphosphate/pentaerythritol/melamine (APP/PER/MEL), which was systematically studied by Bourbigot's [5, 6] and Camino's groups [7]. When being burned, the IFR forms an intumescent multicellular char, acting as a physical barrier which slows down the heat and oxygen transfer and thus prevents the substrate from degrading and burning. However, the relatively low flame retardant efficiency, poor thermostability, and high moisture of the IFR system severely impede its actual commercial application [8].

To overcome these shortcomings, it is imperative to develop a new IFR system. One solution is to apply new acid source such as melamine pyrophosphate (MPP) [9, 10], which is widely used in flame retardant nylon because of its good thermostability and water-resistance property [11–13]. Another one is to synthesize novel charring agents which possess excellent char-forming capability and good thermostability to substitute PER, such as caged bicyclic phosphates [14–16] and triazine derivatives [17]. Recently, caged bicyclic phosphate 2,6,7-trioxal-1-phosphabicyclo[2.2.2]octane-4-methanol 1-oxide, named pentaerythritol phosphate (PEPA) was considered to be one of the most promising halogen-free flame retardants because of its good thermostability and outstanding char-forming capability [10, 11, 18]. Jiao and coworkers [9, 10] employed MPP and PEPA to flame-retard PP. It was found that MPP/PEPA could effectively improve the flame retardancy and thermostability of PP. When the content of PEPA and



SCHEME 1: Synthetic route of PEPA.

MPP was 15 wt% and 15 wt%, respectively, the LOI value of the flame retardant PP was 33.2% and the UL-94 test was classed as a V-0 rating. However, the flame-retardant and thermal degradation mechanism of MPP/PEPA for PP had not been investigated and revealed.

In this work, an efficient caged bicyclic phosphate PEPA was synthesized and combined with MPP to flame-retard PP. The effects of MPP/PEPA on the flame retardancy and thermal degradation of PP were investigated by limiting oxygen index (LOI), vertical burning test (UL-94), cone calorimetric test (CCT), and thermogravimetric analysis (TGA). The flame-retardant and thermal degradation mechanism of MPP/PEPA was investigated and revealed by TGA, Fourier transform infrared spectroscopy (FTIR), TG-FTIR, and scanning electron microscopy-energy dispersive X-ray spectrometry (SEM-EDXS).

2. Experiment

2.1. Materials. Polypropylene (PP, T30S, a granulated product with a melt flow index of 3.0 g/10 min, 230°C, 2.16 kg) was provided by Maoming Petrochemical Co., Ltd., China. Melamine pyrophosphate (MPP) was obtained from Jiangmen Topchem Technology Co., Ltd., China. Pentaerythritol (PER) was purchased from Tianjin Kermel Chemical Reagent Co., Ltd., China. Dioxane and phosphorus oxychloride (POCl_3) were supplied by Changzheng Chemical Reagent Co., Ltd., China. POCl_3 was distilled under atmospheric pressure before use. Ethanol was purchased from Chinasun Specialty Products Co., Ltd., China. Antioxidant (B215) was provided by Ciba Specialty Chemicals Inc., Switzerland. All the commercial materials except POCl_3 were used directly without further purification.

2.2. Synthesis of 2,6,7-Trioxa-1-phosphabicyclo[2.2.2]octane-4-methanol 1-Oxide Pentaerythritol Phosphate (PEPA). PEPA was synthesized in our laboratory according to Balabanovich's work [19]. Exactly 68.1 g (0.5 mol) pentaerythritol, 22.5 mL of POCl_3 , and 200 mL dioxane and aqueous NaOH solution were added in a 500 mL round bottom flask. The mixture was stirred and heated up to 85°C. Another 22.5 mL of POCl_3 was dropped slowly into the flask. Afterwards, the temperature was heated up to 95°C, and the reaction was kept under reflux until there was no HCl released. The reaction mixture was then cooled to the room temperature and filtered. The white solid was washed with dioxane and ethanol for several times. After being dried at 100°C under vacuum to a constant weight, the white powdery PEPA was obtained (yield: 91.2%). The synthetic route of PEPA was shown in Scheme 1.

2.3. Preparation of Flame Retardant PP. PP, MPP, and PEPA were dried in a vacuum oven at 100°C for 6 h before use. Then PP, MPP, and PEPA were melt-mixed on a two-roll mill (XK-160, Changzhou Shuangfeng Machinery Factory, China) at 170°C for 15 min. The prepared mixtures were hot-pressed at 180°C into sheets of suitable thickness and size for the corresponding tests.

2.4. Measurements

2.4.1. Fourier Transform Infrared Spectrometry (FTIR). The sample was mixed with KBr powder and then pressed into a tablet. The FTIR spectrum of the sample was recorded on a Tensor 27 spectrometer (Bruker Optics Inc., Germany) by averaging 16 scans at a resolution of 4 cm^{-1} . The measurement was carried out in the optical range of 4000–400 cm^{-1} .

2.4.2. ^1H Nuclear Magnetic Resonance Spectrometry (^1H NMR). The ^1H NMR spectrum was obtained by using a Bruker AV300 NMR Spectrometer (Bruker, Germany) with $(\text{CD}_3)_2\text{SO}$ as a solvent.

2.4.3. Elemental Analysis (EA) and Inductively Coupled Plasma-Atomic Emission Spectrometry (ICP-AES). The element content of the sample was measured by an elemental analyzer (Vario EL cube, Elementar Analysensysteme GmbH, Germany) and an inductively coupled plasma-atomic emission spectrometer (ICAP 6500 Duo, Thermo Fisher Scientific Co., USA).

2.4.4. Thermogravimetric Analysis (TGA). The TGA was carried out with a TG209 thermal analyzer (Netzsch Instruments Co., Germany) from 30°C to 700°C at a linear heating rate of 20°C/min under an air flow of 30 mL/min. Each sample was measured in an alumina crucible with a weight about 10 mg.

2.4.5. Thermogravimetry-Fourier Transform Infrared Spectrometry (TG-FTIR). The TG-FTIR instrument consists of a thermogravimeter (TG209, Netzsch Instruments Co., Germany), a Fourier transform infrared spectrometer (Tensor 27, Bruker Optics Inc., Germany), and a transfer tube with an inner diameter of 1 mm connected the TG and the infrared cell. The investigation was carried out from 30°C to 750°C at a linear heating rate of 20°C/min under a nitrogen flow of 30 mL/min. In order to reduce the possibility of pyrolysis gas condensing along the transfer tube, the temperature of the infrared cell and transfer tube was set to 230°C.

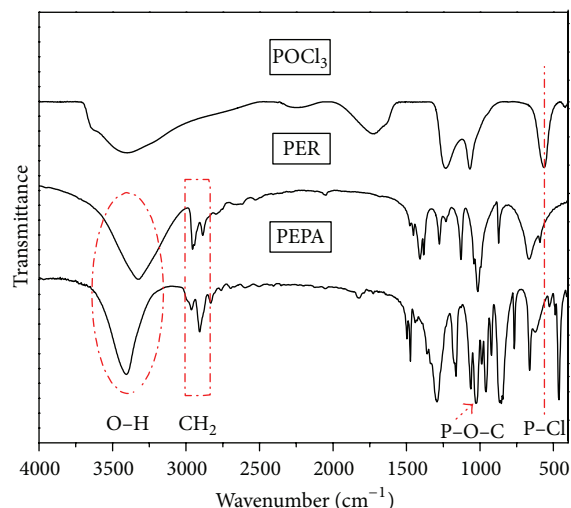


FIGURE 1: FTIR spectra of POCl_3 , PER, and PEPA.

2.4.6. Limiting Oxygen Index (LOI). The limiting oxygen index test was measured by using an oxygen index meter (HC-2, Jiangning Analysis Instrument Co., China) according to ASTM D2863-2008. The dimensions of the specimens were 80 mm × 10 mm × 4.0 mm.

2.4.7. Vertical Burning (UL-94) Test. The vertical burning (UL-94) test was conducted on a vertical burn instrument (CFZ-3, Jiangning Analysis Instrument Co., China) according to ASTM 3801-2010. The dimensions of samples were 127 mm × 12.7 mm × 3.2 mm.

2.4.8. Cone Calorimetric Test (CCT). The cone calorimetric test was carried out by a cone calorimeter (Fire Testing Technology Co., UK) according to ISO5660. Each specimen, with the dimensions of 100 mm × 100 mm × 4 mm, was wrapped in aluminum foil and exposed horizontally to an external heat flux of 35 kW/m².

2.4.9. Scanning Electron Microscopy-Energy Dispersive X-Ray Spectrometry (SEM-EDXS). The morphology of the residue was observed by a scanning electron microscope (EV0-18, Carl Zeiss Jena Co., Germany) with an accelerating voltage of 20.0 kV. The surface of the residue was sputter-coated with a conductive gold layer before observation. EDXS result of the char was measured by an energy dispersive X-ray spectrometer.

3. Results and Discussion

3.1. Characterization of PEPA. Figure 1 shows the FTIR spectra of POCl_3 , PER, and PEPA. POCl_3 was mainly characterized by the absorption of 1200–1300 cm⁻¹ ($\nu_{\text{P=O}}$) and 567 cm⁻¹ ($\nu_{\text{P-Cl}}$). PER was mainly characterized by the peaks of 3390 cm⁻¹ ($\nu_{\text{O-H}}$) and 2960 cm⁻¹ ($\nu_{\text{C-H}}$). When POCl_3 reacted with PER, the $\nu_{\text{P-Cl}}$ peak disappeared, and a new peak appeared at 1019 cm⁻¹ ($\nu_{\text{P-O-C}}$), indicating that

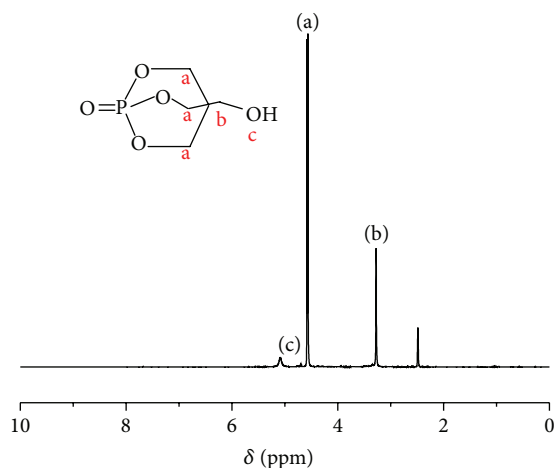


FIGURE 2: ¹H-NMR spectrum of PEPA.

TABLE 1: The experimental and theoretical elemental analysis results of PEPA.

| Elements | C | H | O | P |
|----------------------------|-------|------|-------|-------|
| Theoretical content (wt%) | 33.33 | 5.00 | 44.44 | 17.23 |
| Experimental content (wt%) | 33.51 | 3.90 | 45.23 | 17.36 |

POCl_3 was completely reacted with PER *via* nucleophilic substitution reaction. In addition, the appearance of the characteristic peak of caged bicyclic phosphate (870 cm⁻¹) further confirmed that PEPA was successfully synthesized [20].

The ¹H-NMR spectrum of PEPA was shown in Figure 2. The peak at about 4.5 ppm was assigned to the $-\text{CH}_2-$ protons (a) of the caged bicyclic ring. The peak at 3.2 ppm was attributed to the $-\text{CH}_2-$ protons (b) adjacent to the caged ring. The peak at about 5.1 ppm was assigned to the $-\text{OH}$ (c).

To further confirm the chemical structure of PEPA, the contents of C, H, and O were measured by an elemental analyzer (EA), and P was tested by an inductively coupled plasma-atomic emission spectrometer (ICP-AES). Table 1 presents the EA and ICP-AES data of PEPA. The content of C, H, O, and P was 33.51 wt%, 3.90 wt%, 45.23 wt%, and 17.36 wt%, which were approximately in accordance with their theoretical content, respectively. Based on the above analysis, it was confirmed that PEPA had been synthesized successfully.

3.2. LOI and UL-94 Analysis. Table 2 lists the effect of MPP and PEPA content on the LOI and UL-94 test results of PP. It could be seen that the PP/MPP mixture showed low flame retardancy when MPP was used alone. The LOI value of PP-1 was only 23.0%, and the UL-94 test classed no rating. This was mainly due to the lack of charring agent. MPP could not form an intumescent char layer on the surface of PP to prevent the substrate from burning. When employing some PEPA to replace part of MPP, the flame retardancy of the PP mixture was remarkably enhanced. With the increase of PEPA, the LOI value of the mixture increased first and then decreased.

TABLE 2: Effect of MPP and PEPA content on the LOI and UL-94 test results of PP.

| Samples | Composition (wt%) | | | LOI (%) | UL-94 |
|---------|-------------------|------|------|---------|-------|
| | PP | MPP | PEPA | | |
| PP | 100.0 | 0 | 0 | 17.5 | NR |
| PP-1 | 80.0 | 20.0 | 0 | 23.0 | NR |
| PP-2 | 80.0 | 0 | 20.0 | 24.0 | NR |
| PP-3 | 80.0 | 15.0 | 5.0 | 26.5 | NR |
| PP-4 | 80.0 | 13.3 | 6.7 | 28.0 | V-1 |
| PP-5 | 80.0 | 10.0 | 10.0 | 30.0 | V-1 |
| PP-6 | 80.0 | 6.7 | 13.3 | 33.0 | V-0 |
| PP-7 | 80.0 | 5.0 | 15.0 | 31.0 | V-1 |

TABLE 3: Characteristic parameters of the cone calorimetric test for PP and flame retardant PP.

| Samples | PHRR (kW/m ²) | AV-HRR (kW/m ²) | THR (MJ/m ²) | AV-MLR (g/s·m ²) | Residue at 700 s (wt%) |
|---------|------------------------------|--------------------------------|-----------------------------|---------------------------------|---------------------------|
| PP | 1128.7 | 343.8 | 149.5 | 0.087 | 0 |
| PP-1 | 446.5 | 229.3 | 144.5 | 0.055 | 8.3 |
| PP-2 | 775.0 | 156.5 | 130.9 | 0.062 | 9.8 |
| PP-6 | 244.4 | 208.9 | 133.6 | 0.036 | 23.8 |

When the mass ratio of MPP to PEPA was 1:2, the LOI value of PP-6 reached 33.0% and the UL-94 test classed a V-0 rating, indicating that PEPA showed a good synergistic effect with MPP in flame retardant PP. This could be explained by the fact that, when being burned, PEPA could interact with MPP and formed an intumescent multicellular char, acting as a barrier which slowed down the heat and oxygen transfer, and thus prevented the polymer from burning. When the content of PEPA was not enough, MPP/PEPA formed a thin and weak intumescent char on the polymer because of the lack of charring source. On the contrary, when PEPA was excessive, there was not enough acid source to catalyze PEPA to form high quality intumescent char. When the mass ratio of MPP to PEPA was 1:2, the ratio of acid source, charring agent, and blowing agent was appropriate, and a high quality intumescent char layer was formed and covered on the surface of the polymer and thus effectively enhanced flame retardancy of PP.

3.3. Cone Calorimetric Analysis. The cone calorimetric test (CCT) is widely used to evaluate the combustion behavior of materials. Some important combustion parameters can be obtained from the test, including heat release rate (HRR), total heat release (THR), and mass loss rate (MLR) [21, 22]. The combustion performance and the characteristic data of CCT for flame retardant PP are presented in Figure 3 and Table 3.

Figure 3 shows the HRR and THR curves of PP and flame retardant PP. It was observed that the pure PP burnt out within 370 s after ignition. HRR reached a sharp peak with a peak heat release rate (PHRR) of 1128.7 kW/m², and the average heat release rate (AV-HRR) was 343.8 kW/m². For PP-1 and PP-2, the PHRR were 446.5 kW/m² and

TABLE 4: TGA characteristic parameters of PP and the flame retardant PP under air atmosphere.

| Samples | T _i (°C) | T _{max} (°C) | R _{max} (wt%/min) | Residue at 700°C (wt%) |
|---------|------------------------|--------------------------|-------------------------------|---------------------------|
| PP | 269 | 337 | 28.1 | 0 |
| PP-1 | 278 | 346 | 23.5 | 8.9 |
| PP-2 | 292 | 341 | 21.6 | 9.2 |
| PP-6 | 287 | 346 | 19.6 | 9.6 |

775.0 kW/m², and the AV-HRR was 229.3 kW/m² and 249.2 kW/m², respectively. For PP-6, the PHRR and AV-HRR were 244.4 kW/m² and 156.5 kW/m², which were 78.3% and 54.5% lower than those of the pure PP, respectively. These results indicated that the synergistic effect between MPP and PEPA could enhance the flame retardancy of PP, which was in accordance with the LOI and UL-94 test results. Compared with the pure PP, it was obvious that the burning time of PP-6 was prolonged remarkably. Moreover, it could be seen from Figure 3(b) that, after combustion for 300 s, the THR of the PP resin reached 144 MJ/m², while the value was only 59 MJ/m² for PP-6 at the same time. The reason might be related to the formation of continuous and compact intumescent char, which acted as barrier to prevent combustible gas from feeding the flame and protect the underlying matrix effectively from further burning [23].

As also listed in Table 3, the average mass loss rate (AV-MLR) for PP, PP-1, PP-2, and PP-6 were 0.087, 0.055, 0.062, and 0.036 g/s·m², respectively. At the end of burning, the residues of PP-1, PP-2, and PP-6 at 700 s were 8.3 wt%, 9.8 wt%, and 23.8 wt%, respectively, while there was almost no char left for the pure PP. The residue of PP-6 was more than those of PP-1 and PP-2, which further confirmed that the synergistic effect between MPP and PEPA could significantly improve the char-forming capability of PP.

3.4. Thermal Behavior. The interaction between MPP and PEPA could be revealed by the experimental and calculated TGA curves. Figure 4 shows the TGA and DTG curves of MPP, PEPA, and MPP/PEPA under air atmosphere. The experimental curve was directly obtained from the thermogravimeter, while the calculated curve was the sum of the normalized weight loss of MPP and PEPA. The distinct difference between the experimental and calculated results indicated a strong interaction between MPP and PEPA. When the temperature was 250°C, the mixture began to lose mass due to the esterification between MPP and PEPA, which eliminated some kinds of small molecule; thus the experimental curve had lower thermostability than the calculated one. As the temperature increased, the experimental curve exhibited much higher thermostability, which indicated that MPP could react with PEPA and form a more thermostable char layer. With the further increase of temperature, the mass loss rate of MPP/PEPA reached a maximum at about 430°C. This may be explained by the decomposition of MPP and PEPA, which would produce large amounts of NH₃ and some phosphorus-containing compounds at high

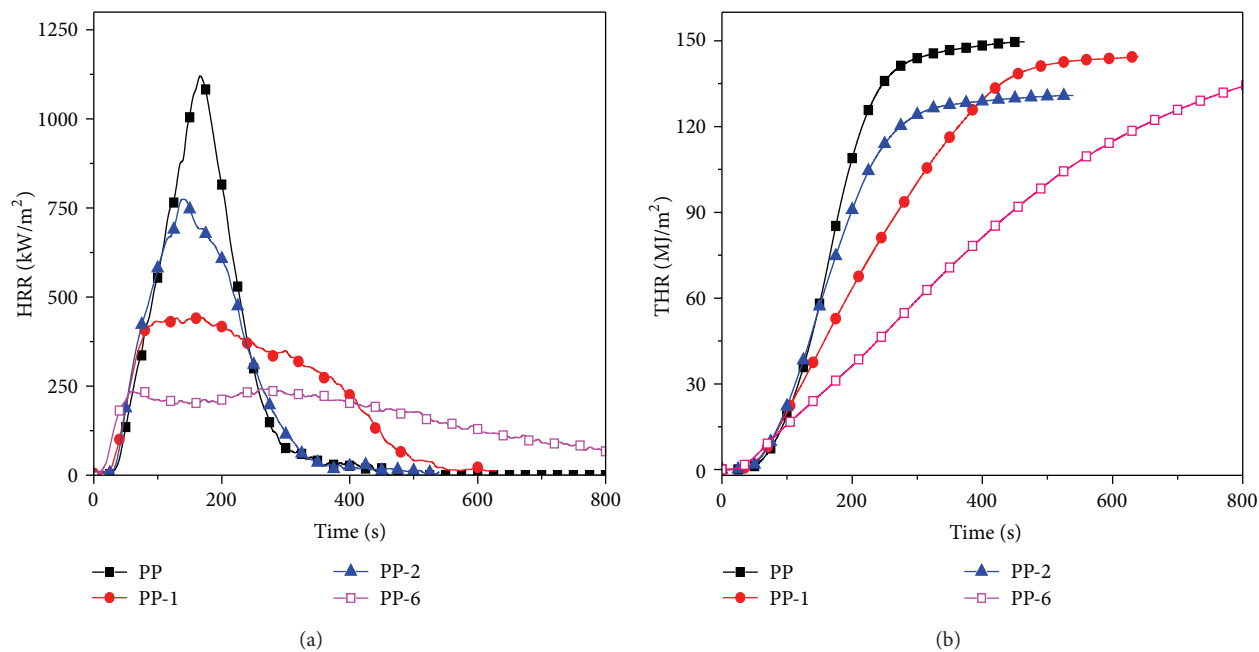


FIGURE 3: (a) HRR and (b) THR curves of PP and flame retardant PP at a flux of 35 kW/m².

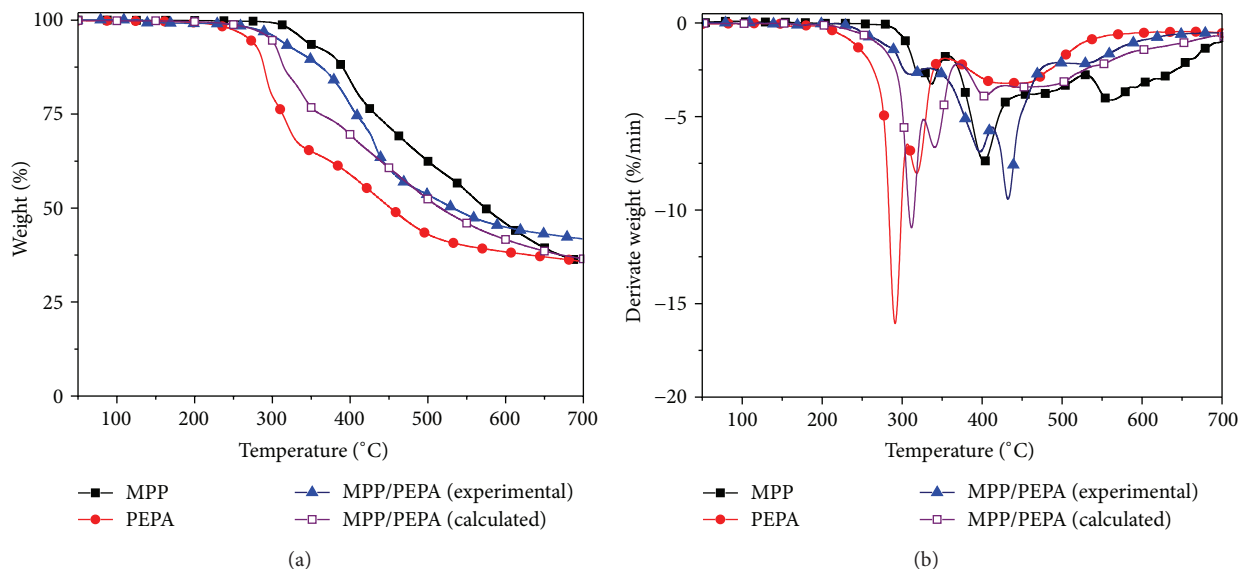


FIGURE 4: (a) TGA and (b) DTG curves of MPP, PEPA, and MPP/PEPA under air atmosphere. Note: the mass ratio of MPP to PEPA was 1 : 2.

temperature. Above 600°C, the experimental curve almost unchanged, indicating that the intumescent char formed by MPP/PEPA was thermostable. The experimental residue of MPP/PEPA at 700°C was 41.8 wt%, whereas it was only 36.5 wt% based on the calculation, which also confirmed that there was an outstanding char-forming capability of MPP/PEPA.

Figure 5 shows the FTIR spectra of MPP/PEPA after heat treatment at different temperature. As could be seen, the relative intensities of the characteristic bands for the MPP/PEPA mixture were almost unchanged below 250°C.

Above 250°C, the decrease of intensity of the broad band around 3200~3500 cm⁻¹ (ν_{O-H}) and the increase of intensity of the 1075 cm⁻¹ (ν_{P-O-C}) peak were caused by the dehydration between MPP and PEPA. With the increase of the temperature, the absorption of 863 cm⁻¹ (the caged bicyclic phosphate) and 1075 cm⁻¹ (ν_{P-O-C}) was significantly decreased [10, 15], which was attributed to the degradation of phosphate esters to form thermostable pyrophosphate. Simultaneously, the phosphorus-containing compound radicals including pyrophosphate radicals and phosphate ester radicals were generated by the decomposition of PEPA, which

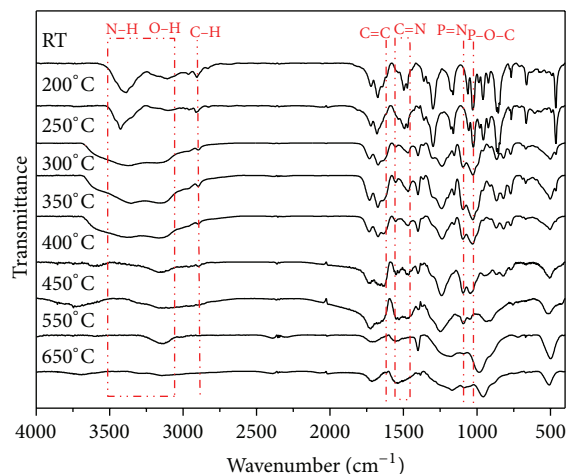


FIGURE 5: FTIR spectra of MPP/PEPA treated at different temperatures for 15 min. Note: the mass ratio of MPP to PEPA was 1:2.

played an important role in the interruption of free radicals chain reaction [3]. Meanwhile, the increased band at about 1620 cm^{-1} was ascribed to $\text{C}=\text{C}$ structure [24, 25], which was also produced by PEPA and could promote the formation of thermostable residue [19]. When the temperature reached up to 400°C , there was a new peak around 1100 cm^{-1} appearing, which corresponded to $\text{P}-\text{N}$ absorption [26], indicating that the phosphorus-containing compounds could further crosslink with the triazine oligomer produced by MPP [3, 27]. With the further increase of pyrolysis temperature, the intensities of the bands at $3200\text{--}3500\text{ cm}^{-1}$ ($\nu_{\text{N-H}}$ and $\nu_{\text{O-H}}$) decreased gradually and nearly disappeared at 550°C to 650°C . However, the bands at 1075 cm^{-1} ($\nu_{\text{P-O-C}}$), 1100 cm^{-1} ($\nu_{\text{P-N}}$), and 1485 cm^{-1} ($\nu_{\text{C=N}}$) were almost invariable. These results suggested that MPP had reacted with PEPA and formed a cross-linked structure, containing $\text{P}-\text{O}-\text{C}$, $\text{P}-\text{N}$ and $\text{C}=\text{N}$ bonds and acting as a thermostable graphite-like char [14].

In order to further investigate the interaction between MPP and PEPA, TG-FTIR was used to analyze the pyrolysis products and gas products evolved during the thermal degradation. Figure 6 presents the FTIR spectra of pyrolysis gases of the MPP/PEPA mixture at various temperatures. It could be seen that there was almost no infrared signal below 250°C . With the increase of temperature, there was a weak signal which appeared belonging to H_2O at 1645 cm^{-1} in 250°C , which could be assigned to the esterification and dehydration of MPP and PEPA. When the temperature was raised to 350°C , the bands of $\text{P}-\text{O}-\text{P}$ (845 cm^{-1}) and $\text{P}-\text{O}-\text{C}$ (1075 cm^{-1}) appeared, which were attributed to the pyrophosphate fragment and phosphate ester fragment, respectively. Simultaneously, the appearance of NH_3 (950 cm^{-1}) indicated that MPP started to decompose to release large amount of NH_3 and generated the triazine oligomer [3, 27]. These phenomena were coincident with the TGA and FTIR spectra of MPP/PEPA results. With the further increase of temperature, the band of carbon oxide and H_2O became stronger and diluted the oxidizing and combustible gas phases because of

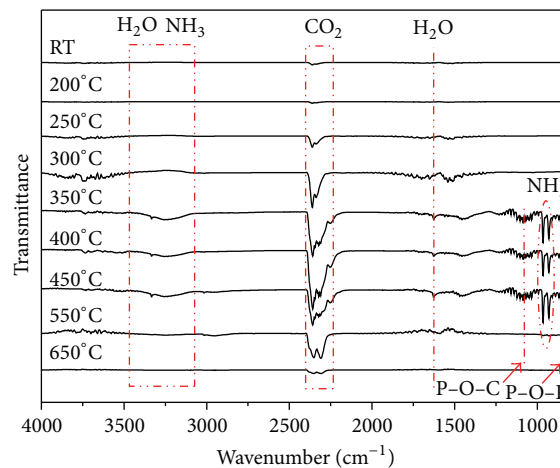


FIGURE 6: FTIR spectra of pyrolysis gases of the MPP/PEPA mixture at various temperatures. Note: the mass ratio of MPP to PEPA was 1:2.

the further degradation of the unstable partial char of the residue. As the temperature reached up to 650°C , all peaks nearly disappeared, meaning that the process of decomposition had finished and a thermostable char was formed.

Figure 7 presents the TGA and DTG curves of PP and the flame retardant PP under air atmosphere, and the detailed results are listed in Table 4. It could be seen that pure PP began to decompose at 269°C and the temperature at maximum mass loss rate (T_{max}) was 337°C with almost no residue left at 470°C . As expected, the incorporation of 20 wt% PEPA into PP resulted in the increase of T_i and T_{max} , which were 23°C and 4°C higher than those of PP, respectively. This could be explained by the fact that the phosphorus-containing compounds (mainly the $\text{P}-\text{O}\cdot$ radicals) which decomposed by PEPA could quench the free radicals like $\text{H}\cdot$ and $\text{OH}\cdot$ when PP composite was heating and improved the thermostability of PP [3, 28]. When combined with MPP, the T_i of PP-6 declined 5°C compared with PP/PEPA composite. This was because of the esterification reaction between MPP and PEPA. However, the residue of PP-6 at 700°C was 9.6 wt%, which was higher than that of PP-1 and PP-2, indicating that the synergistic effect between MPP and PEPA could improve the char-forming property of PP.

Figure 8 shows the 3D TG-FTIR spectra of pyrolysis gases of (a) PP and (b) PP-6 during the thermal degradation. Figure 9 presents the evolution curves of hydrocarbons, which are the main gaseous pyrolysis products of PP and PP-6. Pure PP began to release some hydrocarbons (2961 cm^{-1}) at 331°C . For PP-6, the initial hydrocarbons gases release occurred at 355°C , which was 24°C higher than that of the pure PP, indicating that the phosphorus-containing compounds could trap the free radicals and enhance the thermostability of PP [3, 28]. With the increase of temperature, the evolution of hydrocarbons for pure PP-6 reached a maximum at 551°C , which was 73°C higher than that of the pure PP (478°C). What is more, the intensity of the hydrocarbons released by PP-6 was much lower than that of pure PP. This was due to the fact that thermostable char

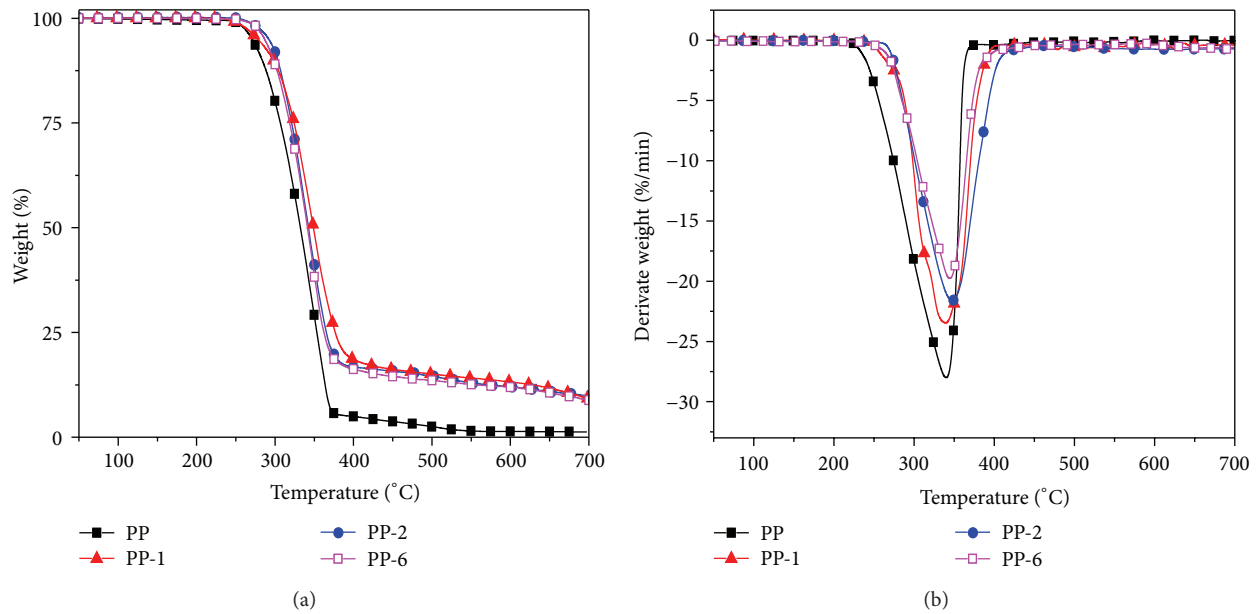


FIGURE 7: (a) TGA and (b) DTG curves of PP and the flame retardant PP under air atmosphere.

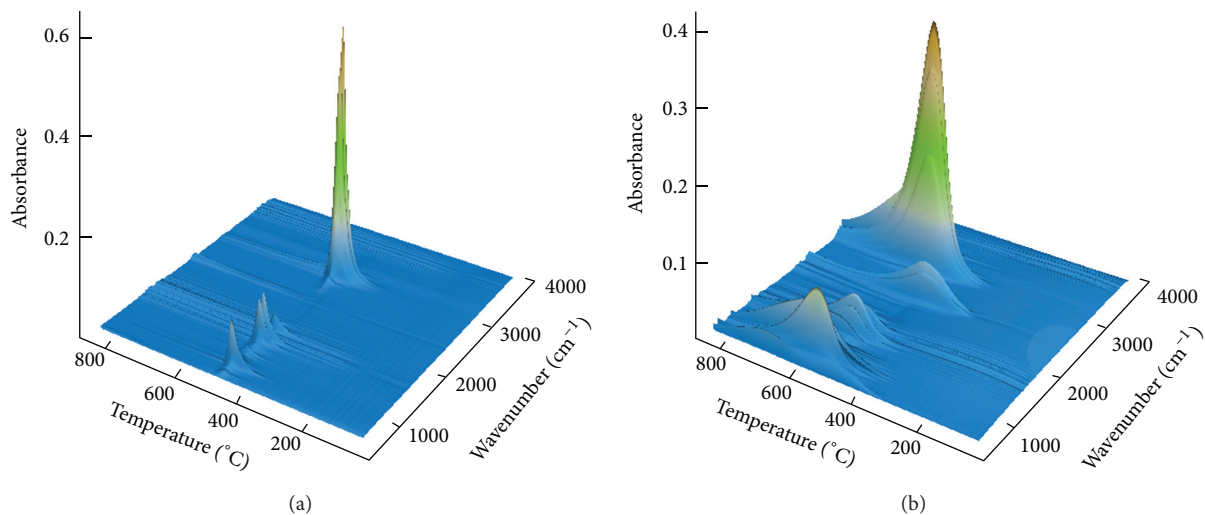


FIGURE 8: 3D TG-FTIR spectra of pyrolysis gases of (a) PP and (b) PP-6 during the thermal degradation.

layer formed by MPP/PEPA could effectively slow down the combustible gas and heat transfer and thus prevented the inner mixture from further decomposition.

3.5. Morphology and SEM-EDXS Analysis of the Residue.

Figure 10 displays the digital photographs of PP and the flame retardant PP after the cone calorimetric test. It could be clearly seen that PP was burnt completely with no residue left (Figure 10(a)), while there was some intumescent char left for all of the flame retardant PP. However, the char layers of PP-1 and PP-2 were thin and broken (Figures 10(b) and 10(c)), which could not effectively slow down either the combustible gas or heat transfer, resulting in the poor flame retardancy of the mixture [29]. When MPP was combined

with PEPA to flame-retardant PP, a swollen and compact char layer was formed by MPP/PEPA covered on the substrate (Figure 10(d)), which could effectively block the transmission of oxygen, combustible gases, and heat and thus prevented the matrix from burning [30]. The micromorphology and composition of the outer char layer of PP-6 were also observed by the scanning electron microscopy-energy dispersive X-ray spectrometry (SEM-EDXS). As shown in Figure 11, the surface of intumescent char layer for PP-6 was compact and dense. The composition of the char layer included 36.8 wt% carbon (C), 10.2 wt% nitrogen (N), 43.1 wt% oxygen (O), and 9.9 wt% phosphorus (P). The high C, N, and P contents of the char layer indicate that C atoms probably were cross-linked with N and P atoms, which could effectively promote the matrix to form a thermostable char [31].

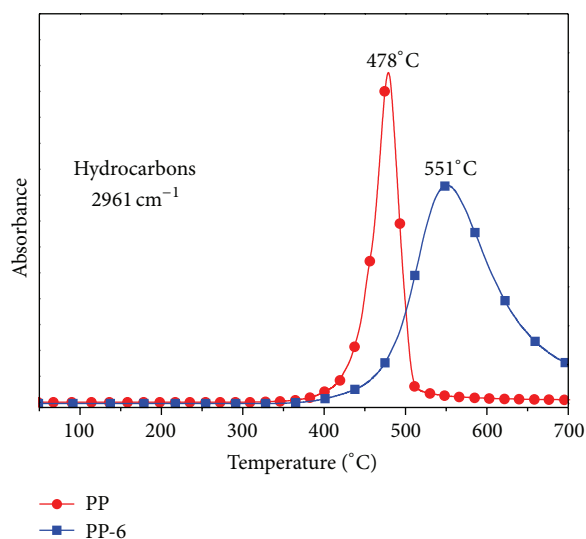


FIGURE 9: FTIR absorbance versus temperature curves of hydrocarbons evolved from PP and PP-6.

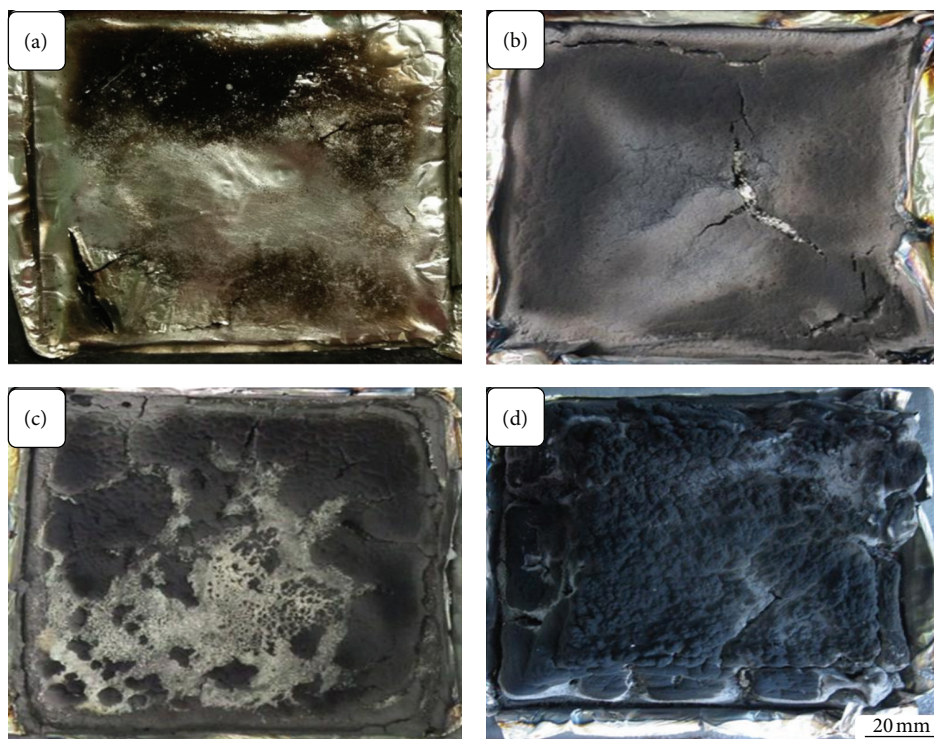


FIGURE 10: Digital photographs of PP and the flame retardant PP after the cone calorimetric test: (a) PP, (b) PP-1, (c) PP-2, and (d) PP-6.

3.6. Possible Flame-Retardant and Thermal Degradation Mechanism. Based on the results above, the possible flame-retardant and thermal degradation mechanism of MPP/PEPA for PP is shown in Figure 12. Above 250°C, the formation of P–O–C bond occurred from the esterification and dehydration reaction between MPP and PEPA. As the temperature increased, the cage structure of PEPA was destroyed to form phosphorus-containing compound radicals and C=C structure compound. Simultaneously, MPP was decomposed to form the triazine oligomer. Those

radicals could quench the free radicals (such as $-\text{CO}\cdot$ and $-\text{COO}\cdot$) produced by the degradation of PP and promoted the thermostability of PP. Then the C=C structure and phosphorus-containing compounds began to aromatize with the triazine oligomers to form a precursor char layer. After that, some incombustible gases like NH_3 , H_2O , and CO_2 were released to take away part of the heat and swell the precursor layer. With the increase of temperature, the further cross-linked reaction of MPP/PEPA formed P–N bonds and improved the formation of an intumescent and compact char

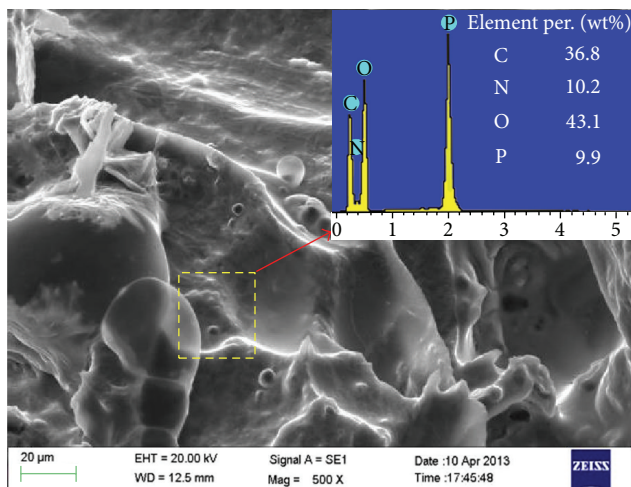


FIGURE 11: SEM-EDXS image of the outer char surface for the PP-6 after the cone calorimetric test.

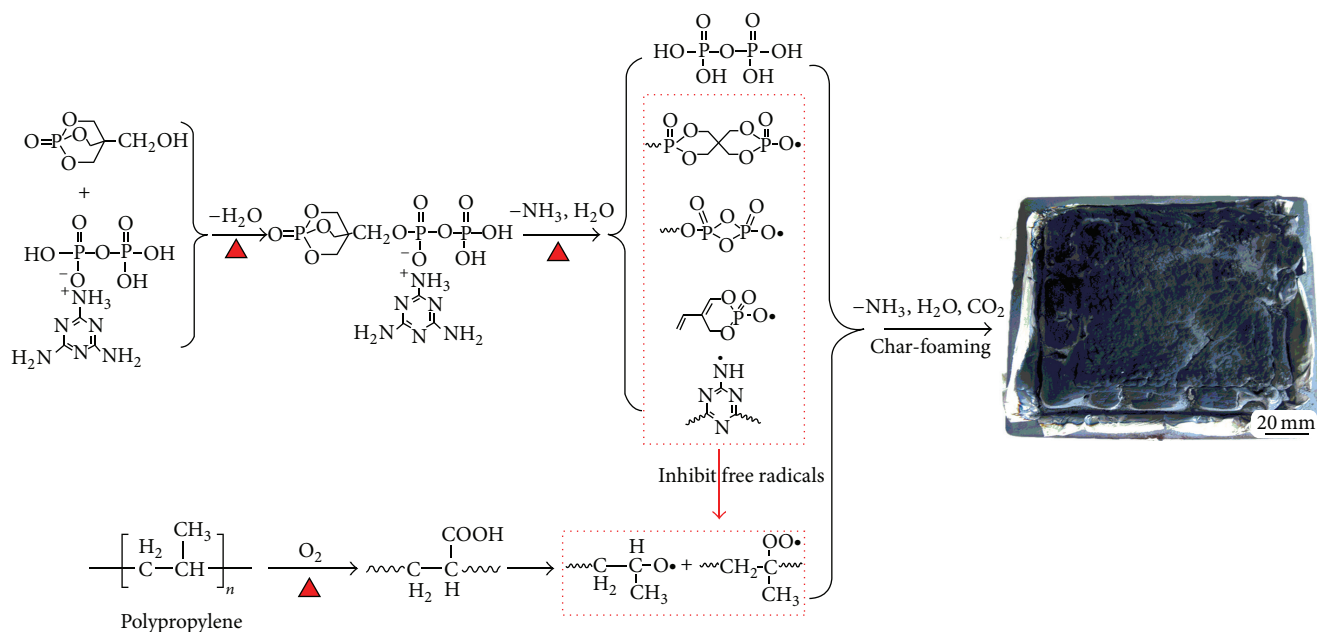


FIGURE 12: Possible flame-retardant and thermal degradation mechanism of MPP/PEPA for PP.

layer, which could slow down the heat and combustible gases transfer, and thus protected the polymer from degrading and burning.

4. Conclusions

An efficient caged phosphate charring agent named PEPA was synthesized successfully, and it showed an outstanding synergistic effect with MPP in flame retardant PP. When the content of PEPA was 13.3 wt% and MPP was 6.7 wt%, the LOI value of the PP/MPP/PEPA mixture was 33.0%, and the UL-94 test reached a V-0 rating. Meanwhile, the peak heat release rate (PHRR), total heat release (THR), and average

mass loss rate (AV-MLR) were also significantly reduced. The result of TGA, TG-FTIR, and SEM-EDXS revealed that, when being burned, MPP/PEPA could generate the phosphorus-containing compound radicals and the triazine oligomer radicals which captured the free radicals of PP chain scission, and thus inhibited the degradation and combustion of the polymer. Afterwards, the cyclization between the C=C structure, phosphorus-containing compounds, and the triazine oligomers could form a cross-linked precursor char containing P-O-C, P-N, and C=N bonds. Finally, a compact, continuous, and thermostable char layer was formed, which could effectively separate the heat, oxygen, and combustible gas from transferring, and thus prevented the substrate from degrading and burning.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

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