

Research Article

ZnTe Semiconductor-Polymer Gel Composites Electrolyte for Conversion of Solar Energy

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Nanostructured cubic p-type ZnTe for dye sensitized solar cells (DSSCs) was synthesized from 1 : 1 molar ratio of Zn : Te by 600 W and 900 W microwave plasma for 30 min. In this research, their green emissions were detected at the same wavelengths of 563 nm, the energy gap (E_g) at 2.24 eV, and three Raman shifts at 205, 410, and 620 cm^{-1} . The nanocomposited electrolyte of quasisolid state ZnO-DSSCs was in correlation with the increase in the J_{SC} , V_{OC} , fill factor (ff), and efficiency (η) by increasing the wt% of ZnTe-GPE (gel polymer electrolyte) to an optimum value and decreased afterwards. The optimal ZnO-DSSC performance was achieved for 0.20 wt% ZnTe-GPE with the highest photoelectronic energy conversion efficiency at 174.7% with respect to that of the GPE without doping of p-type ZnTe.

1. Introduction

Since the first report on low-cost dye sensitized solar cells (DSSCs) in 1991 [1], great efforts have been made to improve their performance [2–4]. In spite of their initial success of approximately 11% solar energy conversion efficiency, much effort to improve cell performance has not been made to major breakthrough since then [5]. DSSCs are very attractive and promising alternative for the development of new generation of photovoltaic devices. Recent research has focused on the use of p-type semiconductors and organic hole-transport materials [6–9] that have better mechanical properties and simple fabrication processes. The main difficulty is to form contacts at p-n junctions and the potential of DSSCs using solid state electrolyte is quite low [10]. Thus, both liquid and solid electrolytes are combined to form quasisolid (gel) electrolytes, including the replacement of liquid electrolytes by solid state hole conductors such as p-type semiconductors [11], CuI [12], CuSCN [13, 14], and NiO [15–17] and polymeric electrolytes [18, 19]. In this study, p-type ZnTe as hole collectors in a polymer electrolyte was used to fabricate solid state DSSCs.

2. Experimental Procedures

2.1. p-Type ZnTe. To synthesize ZnTe, Zn and Te powders (analytical grade, Fluka) were used without further purification. Four mixtures of 1 : 1 molar ratio Zn : Te were mixed by rotation for 1 h at room temperature and loaded into 11 mm I.D. \times 100 mm long silica boats. Each was placed in a horizontal quartz (HQ) tube at a time. The HQ tube was tightly closed and evacuated to 4.3 ± 1 kPa absolute pressure for removal of air and followed by gradual feeding of argon into this HQ tube. The procedure was repeated three times. Finally, argon in the HQ tube was evacuated to 4.3 ± 1 kPa constant absolute pressure. Each solid mixture was heated in a manner of 10 min batch run by a 2,450 MHz microwave plasma using 300, 450, 600, and 900 W, left to cool down in vacuum to room temperature, and thoroughly mixed. The process was repeated under the same condition until 30 min run completion. In the end, the products were collected for further characterization and for being selected for fabrication of DSSCs.

2.2. Gel Polymer Electrolyte. Basically, gel polymer electrolyte (GPE) contained 0.10, 0.20, 0.30, 0.40, 0.50, and 0.70 wt%

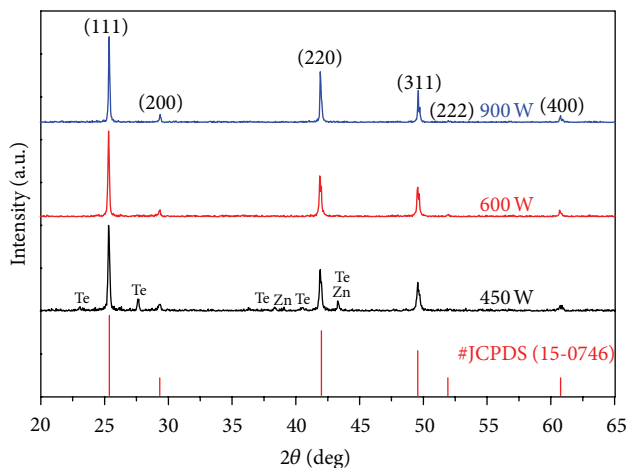


FIGURE 1: XRD patterns of ZnTe synthesized by 450, 600, and 900 W microwave plasma for 30 min.

p-type ZnTe in electrolytic solution of 0.5 mM LiI, 0.05 mM I, 0.15 nM PEG20000, and 10 mL acetonitrile (CH_3CN). The colloidal liquid was stirred at 60°C for 30 min until clear homogeneous quasisolid state was achieved.

2.3. Fabrication of DSSCs. ZnO films were deposited on two fluorine-doped tin oxide (FTO) glass plates by the microwave plasma technique. Each of ZnO photoelectrodes was soaked in the Eosin Y dye (the cheapest dye) solution for 24 h and dried. Concurrently, the FTO electrode was coated by platisol T solution until its sheet resistance was achieved at $10 \Omega/\text{cm}^2$. The GPE solution was cast on the dye-absorbed glass plate and the counter electrode over 1 cm^2 in area for fabrication of a solar cell. Then, the photoanode was partly placed on top of the counter electrode such that they were extended beyond, with the GPE sandwiched in between them. This solar cell was tightly held together by alligator clips. Each sandwich-type photoelectrochemical solar cell was composed of a dye-absorbed ZnO photoelectrode with 8 mm in thickness, a laminar film, and a counter electrode.

3. Results and Discussion

3.1. XRD. XRD is an effective method for determining phases and crystallite sizes of the as-synthesized products. XRD patterns (Figure 1) of the products synthesized by different powers of microwave were characterized. All the spectra were compared with the JCPDS database (reference codes: 15-0746 for ZnTe, 01-1238 for Zn, and 01-0714 for Te) [20]. At 300 W (result not shown), the product was composed of ZnTe and some impurities. Upon increasing the heating power from 300 W to 450, 600, and 900 W, the reactions of Zn and Te became more complete with reduction in impurities. At 450 W, Zn and Te impurities were still left in the main product. At 600 and 900 W, only cubic ZnTe was detected with no impurity detection. At these stages, the chemical reaction of Zn and Te was complete. Comparing among different heating powers, the XRD peaks at 900 W were the

sharpest, implying that this product was the best crystalline and that its atoms resided in normal lattice. The strongest intensity peak is at 2θ of 25.3 degrees and diffracted from the (111) crystallographic planes of the cubic ZnTe main product.

Generally, reaction rate is controlled by heating power and temperature. Therefore, increasing of the heating power has the influence to increase the driving force of reaction, which promotes the reaction rate to speed up and more product is synthesized. Calculated average crystallite sizes using the Scherrer formula [21] were 51.729 nm at 600 W and 51.734 nm at 900 W, due to the crystal growth process.

3.2. SEM and SAED. SEM image (Figure 2(a)) shows a number of ZnTe nanoparticles oriented in different directions with 50 nm in diameter at 600 W microwave heating. This product appears as facet nanoparticles in clusters, specified as crystal—in accordance with the above XRD analysis.

SAED pattern (Figure 2(b)) of the ZnTe product appeared as hollow concentric rings of bright spots, showing that this product was composed of a number of nanocrystals with different orientations. These rings were interpreted and found to correspond very well with the (111), (200), (220), (311), (400), and (511) planes of cubic ZnTe [20].

3.3. Photoluminescence (PL) and UV-Visible Absorption. PL emission of cubic ZnTe (Figure 3(a)) was green emission centered at around 563 nm (2.20 eV), possibly associated with point defects with their energy level above the valence band edge. The emission was in accordance with green emission centered at 560 nm wavelength (2.21 eV) of ZnTe nanowires reported by Meng et al. [22], and 552 nm wavelength (2.25 eV) of 14 nm nanocrystals reported by Lee et al. [23]. The present emission was slightly red shift comparing to 2.26 eV band gap of ZnTe (bulk) [22] caused by some defects residing in the product.

UV-visible absorption of ZnTe synthesized by the 600 and 900 W microwave plasma was recorded. The direct energy gap (E_g) was determined by extrapolation the linear portion of the curves (Figure 3(b)) to zero absorbance, corresponding to 2.24 eV for the threshold optical absorption. In this study, the photoluminescence energy is less than the energy gap due to the existence of defects.

3.4. Raman Spectra. Raman spectra (Figure 4) of ZnTe nanocrystals synthesized by different microwave powers were composed of three longitudinal optical (LO) modes at 205 cm^{-1} (fundamental), 410 cm^{-1} (1st overtone), and 620 cm^{-1} (2nd overtone) and were in good accordance with the available results at 198 cm^{-1} and 396 cm^{-1} modes of ZnTe nanowires reported by Meng et al. [22].

3.5. DSSCs. ZnO films were synthesized by the microwave plasma technique for using as photoelectrodes. In general, the operating principle of a DSSC with p-type ZnTe adding in polymer electrolyte is shown in Figure 5. It consisted of a transparent conductive oxide glass (F-doped SnO_2 glass, FTO), nanoporous ZnO layer, Eosin Y dye (the cheapest dye) monolayer bonded to ZnO nanoparticles, electrolyte

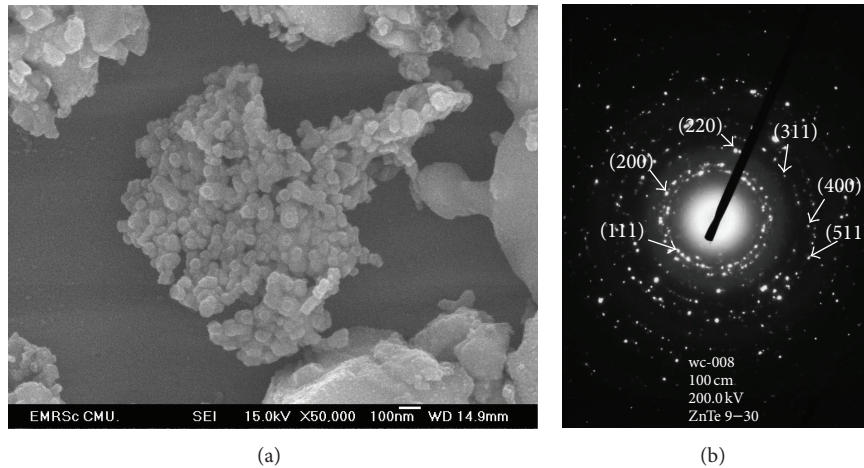


FIGURE 2: (a) SEM image and (b) SAED pattern of ZnTe synthesized by 600 W microwave plasma for 30 min.

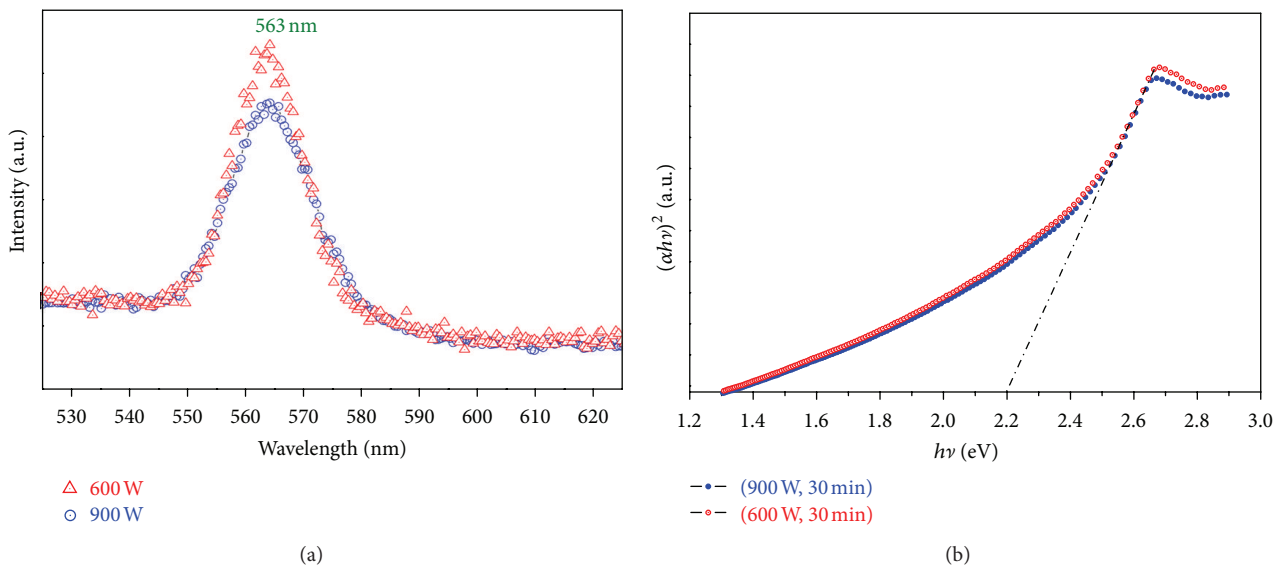


FIGURE 3: (a) PL emissions and (b) the $(\alpha h\nu)^2$ versus $h\nu$ plots of ZnTe synthesized by the 600 and 900 W microwave plasma for 30 min.

consisting of I^- and I_3^- redox species, and platinum coated on counter electrode. When the dye sensitizer was excited by solar radiation, electrons were released into conduction band of mesoporous oxide film and further diffused towards the anode and utilized at external loads prior to be collected by the electrolyte at the cathode to complete the cycle. To improve transmittance properties and electrical conductivity, a FTO conducting glass was used as the anode. Zinc oxide thin film deposited on the FTO glass can play the role in the exciton dissociation and the electron diffusion processes. Porosity of the oxide thin film is the important factor that controls the content of absorbed dye molecules and provides surface area of reaction sites on monolayer dye molecules deposited on top. The selection of dye should correspond to the solar energy range, long-term stability, and firm grafting. I^- and I_3^- couples as well as ZnTe-GPE were used, including a platinum catalyst deposited at the cathode [24–26].

The J - V characteristic curves of 0–0.70 wt% ZnTe-GPE DSSCs are shown in Figure 6, including their performance parameters in Table 1. The efficiency (η) was calculated by

$$\eta (\%) = \frac{J_{SC} \cdot V_{OC} \cdot ff}{P_{in}} \times 100, \quad (1)$$

where J_{SC} , V_{OC} , ff , and P_{in} are the short-circuit current density, open-circuit voltage, fill factor, and incident light power density, respectively [24, 27–30]. In this research, the efficiency of the DSSCs increases to the highest value with the increase in wt% ZnTe added to GPE and decreases afterwards. For 0.20 wt% ZnTe-GPE, its efficiency was the highest at 174.7% with respect to that of the GPE without doping of p-type ZnTe. Comparison with other researchers showed that performance of photovoltaic cells was controlled by several parameters according to the following. Premalal et al. [13]

TABLE 1: Performance of the DSSCs.

DSSCs	wt% ZnTe	J_{sc} (mA/cm ²)	V_{oc} (V)	ff	η (%)
ZT-000	0.00	1.113	0.476	0.343	100.0
ZT-001	0.10	1.631	0.487	0.274	119.8
ZT-002	0.20	1.635	0.508	0.383	174.7
ZT-003	0.30	1.510	0.512	0.402	170.3
ZT-004	0.40	0.893	0.465	0.363	83.0
ZT-005	0.50	0.539	0.476	0.306	42.9
ZT-007	0.70	0.077	0.349	0.130	2.2

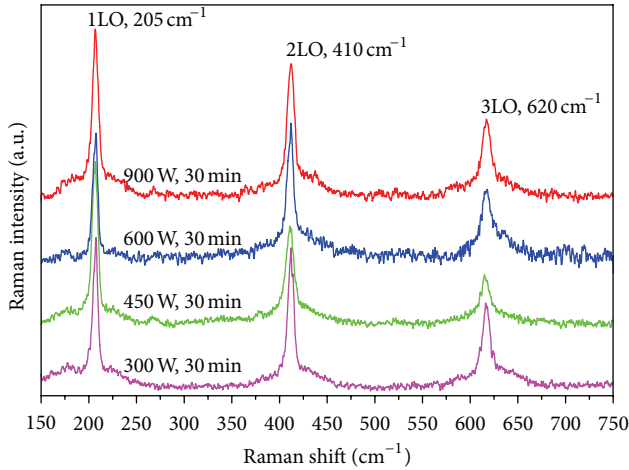


FIGURE 4: Raman spectra of ZnTe synthesized by 300, 450, 600, and 900 W microwave plasma for 30 min.

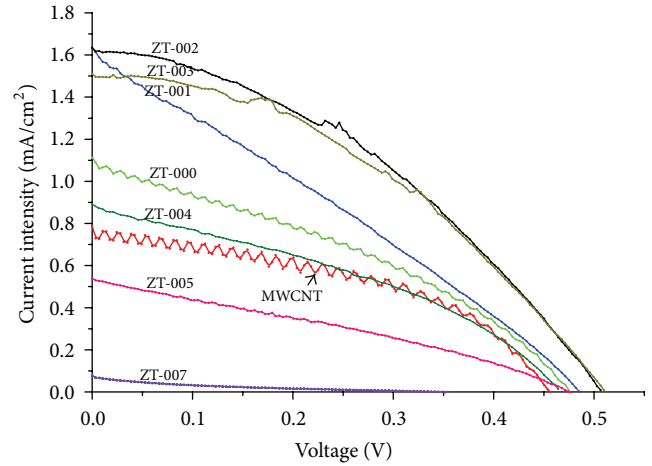
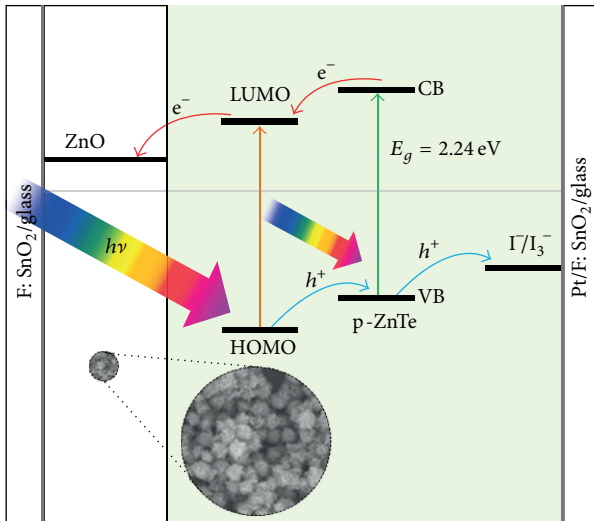
FIGURE 6: J - V characteristic curves of DSSCs with different contents of ZnTe in GPE.

FIGURE 5: A simplified energy level diagram of the ZnTe-GPE DSSCs.

demonstrated the enhancement of conductivity and better pore filling of modified CuSCN inside the TiO₂ matrix. The solar cell performance gradually increased to a maximum value beyond which it decreased. For the best results, the

conversion efficiency is 3.4% at AM 1.5 with 41.8% enhancement of a solid state DSSC using CuSCN as a hole transport material. It was 14 times higher than that obtained for the solid state DSSC of ordinary CuSCN. Chung et al. [27] found that the fresh and clean CsSnI₃ cell showed very good J - V characteristics of solid state solar cell: $J_{SC} = 8.82$ mA/cm², $V_{OC} = 0.638$ V, ff = 66.1%, $\eta = 3.72\%$, $J_{SC} = 12.2$ mA/cm², and $\eta = 5.62\%$ by doping with 5 mol% fluorine. Further improvement was obtained for 2 mol% SnF₂-CsSnI_{2.95}F_{0.05} which, respectively, provided an increase in 29% and 21% of J_{SC} (15.7 mA/cm²) and η (6.81%) comparing to the undoped CsSnI_{2.95}F_{0.05} with the optimum molar concentration of 5 mol% SnF₂. Similarly, the TiO₂ nanoporous film of the corresponding cell was pretreated by a fluorine plasma etching process to increase the size of the nanopores and nanochannels, including the possibility to reduce surface states and charged particle recombination. The results were $J_{SC} = 17.4$ mA/cm², $V_{OC} = 0.730$ V, ff = 72.9%, and $\eta = 9.28\%$. Kim et al. [31] fabricated a DSSC using titanium deposited with a dye-absorbed TiO₂ layer as a working electrode and a platinum deposited with a porous polymer electrolyte membrane as a counter electrode. Overall, the ionic conductivity was improved by increasing the P123 content. The photoelectrochemical characteristics were also controlled by the thickness of the TiO₂ layer and the twist pitch length of the counter electrode. The best performance

is $J_{SC} = 2.117 \text{ mA/cm}^2$, $V_{OC} = 0.6932 \text{ V}$, $ff = 0.7015$, and $\eta = 1.029\%$. Kumara et al. [32] found that the thickness of TiO_2 and ZnO layers can play the role in conversion efficiency of the photovoltaic cells. When the thickness and, hence, the sheet resistance of the TiO_2 dense layer were increased, the conversion efficiency was gradually increased up to 2.6% at a sheet resistance of 370Ω ($15 \mu\text{m}$ thick) beyond which it decreased. Upon the use of ZnO dense layer for the replacement of TiO_2 dense layer, the similar trend was also detected, and the best conversion efficiency of 3.2% was obtained at a sheet resistance of 2500Ω ($15 \mu\text{m}$ thick).

4. Conclusions

Pure cubic ZnTe nanocrystals were successfully synthesized by an inexpensive solid state process using 600 and 900 W microwave plasma for 30 min. Their photoluminescence was the same green emission centered at 563 nm (2.20 eV), possibly associated with point defect level above the valence band edge. The performance of quasisolid state ZnO DSSCs with different wt% ZnTe -GPE composited electrolytes was the highest efficiency for 0.20 wt% ZnTe adding.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

Acknowledgments

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