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# Synthesis of Ce3+-doped Terbium Aluminum Garnet Phosphors from Nanostructured Oxides

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The paper describes the synthesis of Ce<sup>3+</sup>-doped terbium aluminum garnet (TAG) phosphors with the use of nanostructured oxides of aluminum and rare earths. Aluminum oxide nanoparticles were obtained by gaseous-disperse synthesis and characterized by X-ray diffraction, differential thermal analysis and scanning electron microscopy. The preparation of cerium-doped terbium oxide nanopowders and luminescent properties of the Tb<sub>3(1-x)</sub>Ce<sub>3x</sub>Al<sub>5</sub>O<sub>12</sub> phosphors are also discussed.

Keywords: Terbium Aluminum Garnet, Synthesis, Nanopowders, Luminescence.

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## 1. INTRODUCTION

The white light-emitting diodes (LED's) are one of the most promising alternatives to conventional electric light sources, i.e. incandescent lamps, Hg-containing fluorescent lamps. This concept has been commercially realized by using a combination of a blue LED emitting around 460 nm and yttrium aluminum garnet (Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub>, YAG) doped with Ce<sup>3+</sup> ions as a yellow phosphor. The restricting factors in using such white LED's for general lighting are a low color rendering index (Ra<80) that is caused by a deficit of red component in the emission spectrum of YAG:Ce<sup>3+</sup> [1,2]. The shortcomings of YAG:Ce<sup>3+</sup>-based white LED's stimulated the search for alternative compositions with the garnet structure [1,2] and, in particularly, the attempts to improve the color characteristics of LED's by substitution of Y<sup>3+</sup> with other rare earth (R) ions. Jang et al. [3] have studied the luminescent properties of Ce3+ ions in Tb<sub>3x</sub>Y<sub>3(1-x)</sub>Al<sub>5</sub>O<sub>12</sub> (TYAG) solid solutions. It was found that the Tb3+ substitution induces larger crystal field splitting of the Ce3+ 5d configuration and shifts the Ce<sup>3+</sup> emission band towards longer wavelengths. Since this emission is efficiently excited by photons in the 380-460 nm region, efficient white LED's were fabricated by using a combination of (In,Ga)N chips emitting around 460 nm and TYAG:Ce3+ as a yellow-orange phosphor. Also, several groups of authors have studied the luminescent properties of Ce<sup>3+</sup> ions in Tb<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> (TAG) upon excitation in the 240-550 nm region [4-6]. It was shown that there is an efficient energy transfer between Tb<sup>3+</sup> and Ce<sup>3+</sup> in TAG and this material can be also of interest for development of new scintillators [5]. As a rule, Ce-containing garnets phosphors are prepared by the direct reaction between starting solidphase reagents at temperatures, which typically exceed 1500°C. Even then, depending upon the preparation conditions some amounts of impurity phases such as RAlO<sub>3</sub>, CeO<sub>2</sub> can be revealed in the final products [2,7].

Beside this, an insufficient mixing and low reactivity of raw materials often result in the significant difference in the Ce<sup>3+</sup> concentrations within the grains and in the vicinity of grain boundaries [8]. In the present paper, we describe the synthesis of  $Tb_{3(1-x)}Ce_{3x}Al_5O_{12}$  (x= 0-0.03) luminescent materials by solid state reaction between aluminum oxide Al<sub>2</sub>O<sub>3</sub> and mixed rare earth oxides. Our approach included the use of a nanosized metastable Al<sub>2</sub>O<sub>3</sub>, the use of oxalate precursor-derived rare earth oxides, the firing of reaction mixtures in the temperature region of a metastable alumina phase to α-Al<sub>2</sub>O<sub>3</sub> transition. It was expected that an acceleration of the diffusion of Al3+ and O2- in temperature region of the Al<sub>2</sub>O<sub>3</sub> phase transformation would stimulate the solid state reaction. The luminescent properties of the obtained phosphors are also briefly discussed.

## 2. EXPERIMENTAL

All samples were characterized by X-ray diffraction (XRD) using Cu K<sub>α</sub> radiation (Rigaku Ultima IV). Differential thermal analysis (DTA) was performed on a thermal analyzer (LABSYS DSC/DTA/TG) in air at a heating rate of 20°C/min. Morphological investigations were carried out by scanning electron microscopy (SEM) on JEOL JSM 6390LV electron microscope. The emission and excitation spectra in UV-visible region were recorded at 77 K and room temperature using a Fluorolog Fl-3 (Horiba Jobin Yvon) spectrofluorometer equipped with a xenon lamp.

# 3. RESULTS AND DISCUSSION

Nanosized Al<sub>2</sub>O<sub>3</sub> was obtained by a gaseousdisperse synthesis. This method is based on the combustion of powdered metals due to exothermic oxidizing reactions between them and a gaseous oxidizer (typically O<sub>2</sub>). The details of experimental setup used for the synthesis of nanosized Al<sub>2</sub>O<sub>3</sub> can be found in the paper by Poletaev et al. [9]. The dispersed in N2 aluminum

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particles with an average size of 4.8 µm were injected through an inner tube into  $O_2$  stream. After ignition by an external source, a stable two-phase diffusion flame was observed. The resulting product was collected using a fabric filter. The XRD pattern of the as-prepared  $Al_2O_3$ , shown Fig. 1, indicates that the powder contains  $\gamma,\delta,\theta$ -phases of  $Al_2O_3$ , which are present in approximately the same quantities.

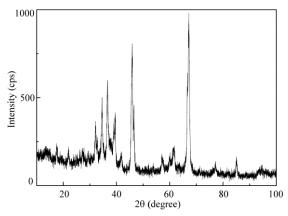


Fig. 1 - Powder X-ray diffraction pattern of as-prepared Al<sub>2</sub>O<sub>3</sub>

As can be seen from Fig. 2, its crystallites are of spherical in shape with 20-70 nm in diameter, and they exhibit a tendency to adhesion. DTA curve of the asprepared  $Al_2O_3$  sample revealed a strong exothermic effect in the range  $1200\text{-}1400^{\circ}\text{C}$  with a maximum at  $1296^{\circ}\text{C}$  (see Fig. 3), which corresponds to the metastable alumina phase— $\alpha\text{-}Al_2O_3$  phase transition [10]. This phase transformation has been extensively studied, and at present, it is well known that  $\alpha\text{-}Al_2O_3$  is formed through a nucleation and growth process, and depending upon chemical prehistory of the precursor, degree of its crystallinity, the presence of impurities etc. the transition temperature varies from 950 to 1350°C [10].

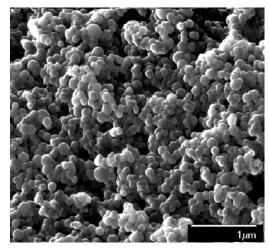


Fig. 2 – SEM photograph of as-prepared Al<sub>2</sub>O<sub>3</sub>

The second step in the preparation involved the formation of mixed rare earth oxides  $Tb_4O_7$ :Ce. To this end, the appropriate amounts of  $Tb_4O_7$  (99.99%) and  $CeO_2$  (99.99%) were dissolved in a dilute  $HNO_3$  solution. The hydrous mixed rare earth oxalates  $R_2(C_2O_4)_3$   $nH_2O$  were precipitated from the hot solution

(80 °C) by the slow addition of a concentrated  $\rm H_2C_2O_4$  solution. The precipitates were filtered, washed with distilled water, dried and then heated at 850 °C for 1 h in air. It is known that oxalate precursor-derived rare earth oxides typically consist of submicron aggregates of nanosized crystals [11]. After that the calculated amounts of Tb<sub>4</sub>O<sub>7</sub>:Ce and Al<sub>2</sub>O<sub>3</sub> were grinded and thoroughly mixed in ethanol, and the resulting mixtures were fired at a temperature of about 1300 °C for 3-5 h in a reducing medium created by burning activated carbon.

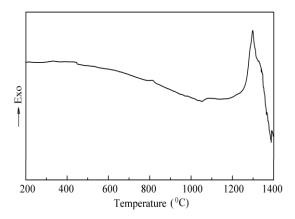
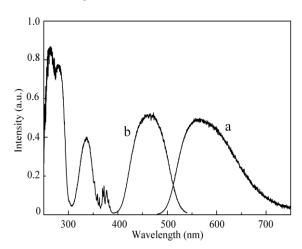


Fig. 3 - DTA curve of nanosized Al<sub>2</sub>O<sub>3</sub>

The XRD patterns of the as-prepared samples were well matched with JCPDS File No. 76-0111 for TAG. The synthesis procedure used was not further optimized for emission intensity or particle size distribution, but even without this highly efficient luminescent materials were obtained. The emission spectra of TAG:Ce<sup>3+</sup> depend on cerium concentration and excitation wavelength.



**Fig. 4** – Emission and excitation spectra of  $Tb_{3(1-x)}Ce_{3x}Al_5O_{12}$  (x= 0.03). The emission spectrum (a) was recoded upon excitation at 450 nm and the excitation spectrum (b) was recorded for the  $Ce^{3+}$  emission at 560 nm

As can be seen from Fig. 4, upon excitation at 450 nm the emission band of  ${\rm Tb_{3(1-x)}Ce_{3x}Al_5O_{12}}$  (x=0.03) extends from 480 to 750 nm and has a maximum at about 563 nm. At 77 K the spectrum shows the doublet structure and can be reasonably decomposed into Gaussiantype bands with maxima at 534 and 583 nm, which are

due to transitions from the lowest Ce3+ 5d excited state to the 4f ground state levels 2F5/2 and 2F7/2. The positions of these maxima are in agreement with the results reported for the Ce3+ emission in TAG at 10 K [3]. The excitation spectrum of  $Tb_{3(1-x)}Ce_{3x}Al_5O_{12}$  (x =0.03) recorded for the Ce3+ emission at 560 nm is also shown in Fig. 4. No doubt that the broad band with a maximum at 465 nm is mainly caused by the 4f→5d transition to the lowest component of the Ce3+ 5d configuration, while the narrow ones are due to the 4f8→4f8 transitions of Tb3+ ions. The band at 334 nm is a superposition of bands with maxima at 325 and 338 nm arising from spin-forbidden 4f8→4f75d transition of Tb3+ ions and 4f→5d transition of Ce3+ ions, respectively, so that its shape and relative intensity depend on the emission wavelength and temperature. The spectrum also contains overlapping bands with maxima at 277 and 262 nm, which must be attributed to the  $4f8 \rightarrow 4f75d$  transitions of the Tb3+ ions [5]. The substitution of Tb3+ for Y3+ in the garnet structure results in broadening the emission band and shifting its maximum towards the longer wavelengths. At room temperature the full width at half maximum of the Ce3+ emission increases from 3460 cm<sup>-1</sup> (YAG) to ~3800 cm<sup>-1</sup> for TAG. When the larger Tb3+ ion occupies the Y3+ position, the dodecahedral site is expanded and distorted, so that these results can be explained by

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an increase in both the crystal field splitting of the Ce3+5d configuration and Stokes shift of the emission. Indeed, because the lowest Ce3+ excitation band of Tb<sub>3(1-x)</sub>Ce<sub>3x</sub>Al<sub>5</sub>O<sub>12</sub> (x=0.03) is situated at 465 nm (Fig. 4), the Stokes shift of the emission amounts to 2780 cm-1. This value is comparable but somewhat larger than that reported for the Ce3+ emission in YAG (2400 cm-1) [12]. The quantum efficiency of luminescence ( $\eta$ ) of the Tb<sub>3(1-x)</sub>Ce<sub>3x</sub>Al<sub>5</sub>O<sub>12</sub> (x=0.03) sample was determined as described in Refs. [1, 2] using a commercial YAG:Ce3+phosphor for LED's with  $\eta=0.90$  as a standard. For the excitation at 460 nm, the found value of  $0.83\pm0.04$  is comparable but somewhat larger than that ( $\eta=0.76$ ) reported in the literature for the Ce3+ emission in TAG prepared at 1500oC [4].

### 4. CONCLUSIONS

TAG: $Ce^{3+}$  phosphors have been successfully prepared with the use of nanostructured oxides of aluminum and rare earths. Spherical  $Al_2O_3$ -particles with a diameter ranging from 20 nm to 70 nm were obtained by gaseous-disperse synthesis. It was shown that the  $Ce^{3+}$  ions in TAG exhibit the efficient broad band emission with a maximum at about 563 nm and the quantum efficiency of luminescence of the  $Tb_{3(1-x)}Ce_{3x}Al_5O_{12}$  (x = 0.03) phosphor was found as high as 0.83.

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