

Morphology Control of Tin Oxide Nanostructures and Sensing Performances for Acetylene Detection

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Abstract: Morphology Control plays an important role in gas sensing properties of metal oxide semiconductor based gas sensors. In this study, various morphologies of SnO₂ nanostructures including nanobulks, nanospheres, nanorods, and nanowires were successfully synthesized via a simple hydrothermal method assisted with different surfactants. X-ray powder diffraction and scanning electron microscopy were employed to characterize the prepared products. Gas sensors were fabricated by screen-printing the as-prepared SnO₂ nanostructures onto planar ceramic substrates. Moreover, their gas sensing properties were systematically investigated towards acetylene gas (C₂H₂), an important fault hydrocarbon dissolved in power transformer oil. Experiments indicate that the SnO₂ nanowires based sensor exhibits excellent gas sensing properties, such as lower operating temperature, higher gas response, quicker response-recovery time and good stability than those of SnO₂ nanobulks, nanospheres and nanorods. These results imply SnO₂ nanowires a promising sensing morphology for C₂H₂ detection and provide us a feasible way to develop high-performance gas sensor by tailoring the microstructures and morphologies of the materials in further. *Copyright © 2013 IFSA.*

Keywords: SnO₂ nanostructures, Gas sensor, Acetylene, Gas sensing properties.

1. Introduction

Large-scale power transformers are expensive and significant electric apparatus in power transmission and distribution system [1, 2]. As one of the most important fault characteristic gases dissolved in transformer oil, acetylene (C₂H₂) can effectively reflect the incipient faults of partial discharge [3, 4]. Therefore, how to rapidly and accurately detect dissolved C₂H₂ in oil is currently the subject of intensive research [5].

In recent years, great efforts have been made to this field and numerous methods, including metal oxide semiconductor (MOS) [6], infrared spectroscopy [7], Raman spectroscopy [8] or photoacoustic spectroscopy [9] and carbon nanotube [10] have been employed for C₂H₂ detection. Owing to the remarkable advantages of simple design technology, easy fabrication process, low maintenance cost and long service life, MOS has proved to be a promising gas sensing material [11]. However, some limitations still need to be further

improved, such as high operating temperature, low gas response and poor long-term stability.

Recently, interest in morphology control of MOS has been greatly stimulated. And different morphologies of MOS including wires [12], flowers [13], rods [14], spheres [15], fibers [16] and tubers [17] have been prepared. However, to the best of our knowledge, reports on the synthesis of various SnO₂ nanostructures by hydrothermal method assisted with different surfactants have been rare and there are no literatures on morphology control of various SnO₂ nanostructures and its application as a C₂H₂ gas sensor.

In this study, we first reported a simple and facile hydrothermal method assisted with different surfactants to prepare variety morphologies of SnO₂ nanostructures. Then X-ray powder diffraction (XRD) and scanning electron microscopy (SEM) were employed to characterize the prepared samples and a possible growth mechanism was discussed in detailed. Finally, gas sensors were fabricated with a planar structure and C₂H₂ sensing properties were systematically measured.

2. Experimental

2.1. Synthesis of SnO₂ Nanostructures

All the raw chemicals were analytical-grade reagents purchased from Chongqing Chuandong Chemical Reagent Co. Ltd. and used as received without any further purification. Various morphologies of SnO₂ nanostructures including nanobulks, nanospheres, nanorods, and nanowires were successfully synthesized with a facile surfactant-assisted hydrothermal method. The detailed synthesis processes were as follows:

In a typical synthesis process of SnO₂ nanospheres: 1.4 g of SnCl₄·5H₂O, 1.24 g of NaOH and 0.16 g of citric acid were dissolved completely into 60 mL mixture of absolute ethanol and distilled water (1/1, V/V) in a 100 mL capacity beaker under intense magnetic stirring. Then the mixed solution was transferred into a 100 mL Teflon-lined stainless steel autoclave, sealed and maintained at 180 °C for 24 h in an electric furnace. When the reaction was completed, the autoclave was cooled to room temperature naturally. The product was harvested by centrifugation, washed with distilled water and absolute ethanol several times, respectively, and dried at 60 °C in air to remove the solvent.

SnO₂ nanorods and nanowires were obtained in a similar process except that 0.2 g of polyvinyl pyrrolidone (PVP, K30 g) and 0.1 g of polyethylene glycol (PEG, MW = 6,000) were added to replace citric acid, respectively. SnO₂ nanobulks were prepared by the following process: 1.4 g of SnCl₄·5H₂O, 0.6 g of CO(NH₂)₂, 0.15 g of glycine, 30 mL absolute ethanol and 30 mL distilled water were mixed together under intense magnetic stirring, and then the precipitation precursor was transferred

into a 100 mL Teflon-lined stainless steel autoclave, sealed and maintained at 160 °C for 20 h. The product was obtained by cooling, centrifugation, washing and drying.

2.2. Structure Characterization

The crystalline structures of the prepared products were investigated with X-ray powder diffraction (XRD) using a Rigaku D/Max-1200X diffractometer (Tokyo, Japan) with Cu K α radiation (40 kV, 200 mA and $\lambda = 1.5418 \text{ \AA}$) and 2θ ranging from 20° to 80°. The surface morphologies of the nanostructures were characterized with a Nova 400 Nano field emission scanning electron microscope (FE-SEM, FEI, Hillsboro, OR, USA). A surface area and porosimetry analyzer (V-Sorb 2800, Beijing Jinaipu General Instrument Co., Ltd, Beijing, China) was employed to test the specific surface area and open pore size of the nanostructures.

2.3. Fabrication of Gas Sensor

The synthesized SnO₂ nanostructures were firstly ground into fine powder and mixed with absolute ethanol and deionized water in a weight ratio of 8:1:1 to form a paste. Then the paste was screen-printed onto a planar ceramic substrate, where five pairs of Ag-Pd interdigital electrodes had been pre-deposited. Finally, the sensor was dried in air at 60 °C to volatilize the organic solvent and further aged in an aging chamber for 36 h. Fig. 1 shows the structural scheme of the planar sensor.

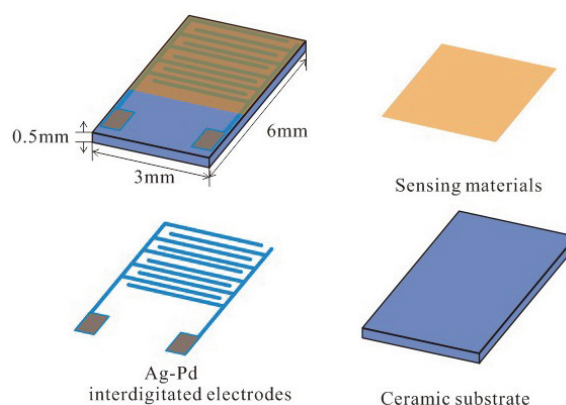


Fig. 1. Structure representation of the planar gas sensor.

2.4. Gas Sensing Property Measurement

Gas sensing properties of the prepared SnO₂ based sensors towards C₂H₂ were measured by a CGS-1TP intelligent gas sensing analysis system purchased from Beijing Elite Tech Co., Ltd, China which could offer an external temperature control

ranging from room temperature to 500 °C with an adjustment precision of 1 °C. Fig. 2 shows a top-view photograph of the operating platform.

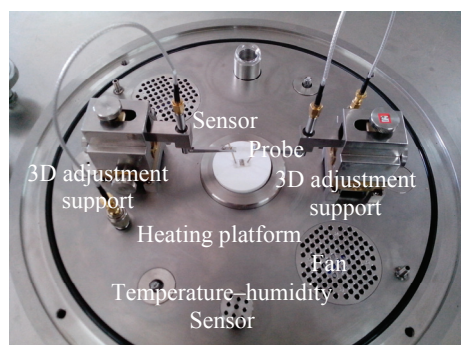


Fig. 2. A top-view photograph of the operating platform in CGS-1TP.

As seen in Fig. 2 the planar sensors were laid on the heating platform and two adjustable probes were pressed on the sensor electrodes to collect electrical signals. When the sensor was pre-heated at certain operating temperature for about 30 min and its resistance was stable, saturated target gas was injected into the test chamber (18 L in volume) by a micro-injector through a rubber plug. The saturated target gas was mixed with air in the test chamber by two fans. After its resistance value reached a new constant value, the test chamber was opened to recover the sensor in air.

The gas response S was defined as $S = R_a/R_g$, where R_a was the electrical resistance value of prepared planar sensor in air and R_g was that in a mixture of target gas and air. The time taken by the sensor to reach 90 % of the total resistance change was defined as the response time in the case of gas adsorption or the recovery time in the case of gas desorption. All measurements were repeated several times to ensure the repeatability and stability of the sensor.

3. Results and Discussion

3.1. Structural Characterization

X-ray powder diffraction (XRD) measurement was first employed to characterize the crystalline structure and chemical composition of the prepared nanostructures. As shown in Fig. 3 the prominent peaks (110), (101), (211) and other smaller peaks are highly consistent with the standard data file of rutile SnO₂ (JCPDS file no. 41-1445). No diffraction peaks from any impurities were observed, implying a high purity of our prepared samples.

Field emission scanning electron microscopy (FESEM) was further used to investigate the morphologies and microstructures of the synthesized samples. Fig. 4 (a) shows a typical SEM image of

spherical SnO₂ nanostructures, which are uniform in size and shape and the average diameters range from 300 nm to 400 nm. Fig. 4 (b) shows the SEM image of SnO₂ nanobulks with average thickness of 150 nm. Fig. 4 (c) displays a SEM image of the synthesized SnO₂ nanorods. It can be clearly seen in Fig. 4 (c) that flower-like sample is composed of a large scale of nanorods with width of about 150 nm and length of 500-600 nm. The SEM image of SnO₂ nanowires is shown in Fig. 4 (d), where the lengths of the nanowires range from hundreds of nanometers to several ten micrometers and the mean diameters are in the scope of 80 to 120 nm. These results indicate that different morphologies of SnO₂ nanostructures have been successfully obtained under the current synthetic conditions.

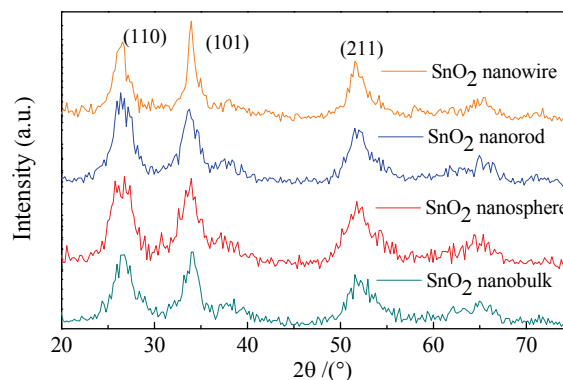


Fig. 3. XRD patterns of the prepared SnO₂ nanostructures.

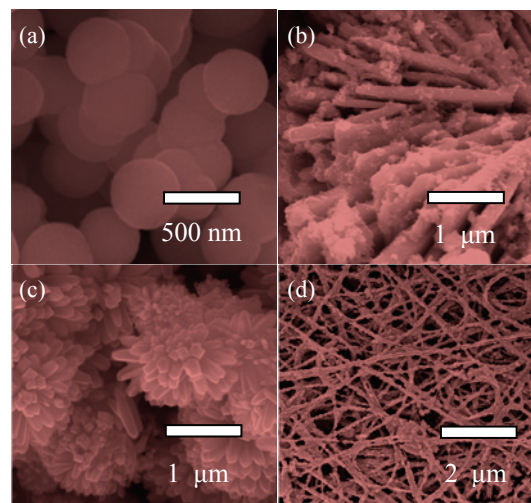


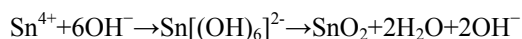
Fig. 4. FESEM images of the prepared SnO₂ nanostructures (nanobulks (a), nanospheres (b), nanorods (c), and nanowires (d)).

3.2. Growth Mechanism

According to the experiments of material synthesis and structural characterization, surfactant added in the precursor plays an important role in the

final morphologies of the as-prepared products. In order to reveal how the additive surfactants affect the morphologies of the prepared SnO₂ nanostructures, possible growth mechanisms are discussed in the following section.

The whole growth process can be divided into a nucleation stage and a self-assembly stage. The former is closely related to the reaction solvents, while the latter is mainly depends on the surface-active agents. A certain amount of Sn[(OH)₆]²⁻ ions has been reported recently as a prerequisite for hydrothermal synthesis of rutile SnO₂ nanostructures. The chemical reactions for the formation of the SnO₂ precursor in our programs can be described as:



At first a large number of tiny SnO₂ nanoparticles was formed spontaneously from the dehydration of abundant component Sn[(OH)₆]²⁻ ions. With the reaction time increasing, the tiny SnO₂ crystal nanoparticles gathered and develop into larger nanocrystals. Thus, SnO₂ nanospheres were observed in Fig. 4 (a) with no addition of any surfactant. With the addition of glycine, SnO₂ nanobulks were obtained as seen in Fig. 4 (b). It could be considered that glycine benefited the growth along one direction and agglomerated the nanoparticles to form smooth bulks.

When PVP was added to the reaction solvent SnO₂ nanorods were firstly produced with the assistance of polarized functional groups (-C=O) [18], which were commonly present in PVP. These prepared nanorods subsequently aggregated with the common centre of PVP. Finally, flower-like SnO₂ nanostructures appeared [19]. As known to all PEG is a kind of template with long chains, and numerous of hydrophilic "-O-" and "-CH₂-CH₂-" groups exist along its long chains [20]. With the addition of PEG, numerous of tiny Sn[(OH)₆]²⁻ crystals embed and nucleated in the long-chains of PEG. Thus, net-like SnO₂ nanowires were successfully synthesized.

3.2. Gas Sensing Properties

In order to make clear how the morphologies of SnO₂ nanostructures influence the gas sensing behaviors of the SnO₂ based sensors, systematic C₂H₂ sensing experiments, including operating temperature, gas response, response-recovery time, stability and repeatability were measured.

The gas responses of the prepared SnO₂ based sensors to 100 μL/L of C₂H₂ as a function of operating temperature ranging from 100 to 430 °C are measured in Fig. 5 to find out the optimum operating temperature. As shown in Fig. 5 for each sensor curve, the gas response increases first at different degrees and reaches its maximum value, and then decreases rapidly with further increasing temperature. The optimum operating temperatures of nanobulks and nanospheres based sensors are tested

to be about 280 °C, while 250 °C for nanorods and nanowires, where the sensor exhibits the maximum gas responses of 20.34, 26.80, 34.04 and 46.96 respectively.

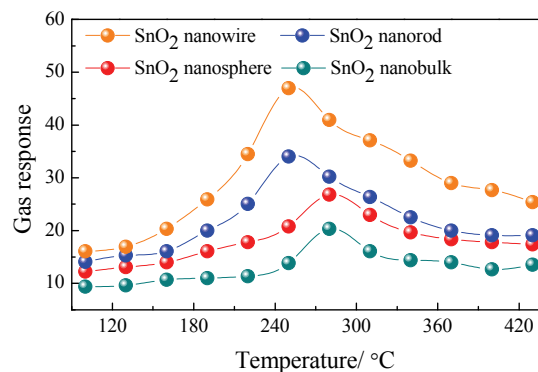


Fig. 5. Response versus operating temperature of the sensors to 100 μL/L of C₂H₂.

Fig. 6 shows the gas responses of the nanowire and nanobulk SnO₂ based sensors towards C₂H₂ with gas concentration ranging from 1 μL/L to 1000 μL/L. As can be seen, the gas response of the SnO₂ nanowires (operated at 250 °C) increases rapidly in linearity with increasing C₂H₂ concentration below 200 μL/L. And then the sensor shows a slower increase with further increasing the gas concentration, and finally reaches saturation at nearly about 1000 μL/L. Similar C₂H₂ response curves were also found for SnO₂ nanobulk based sensor (operated at 280 °C), while much weaker.

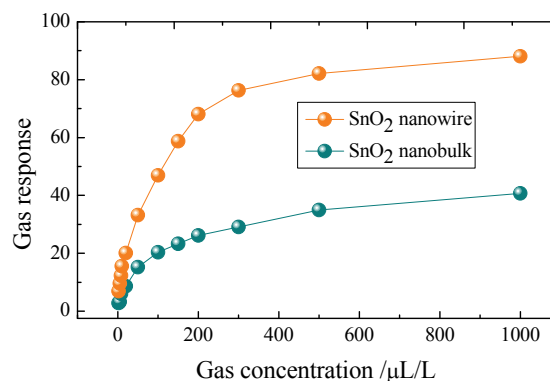


Fig. 6. Responses of SnO₂ nanowire and nanobulk sensor versus C₂H₂ concentration from 1 to 1000 μL/L.

Response and recovery time is another key gas sensing parameter of metal oxide semiconductor gas sensor. Fig. 7 shows the transient responses of various SnO₂ based sensors exposed to 20 μL/L of C₂H₂ with sensor working at its optimum operating temperature. The response-recovery time of nanobulks, nanospheres, nanorods, and nanowires based sensors were calculated to be about 15-25, 13-21, 11-18, 7-13 s, respectively.

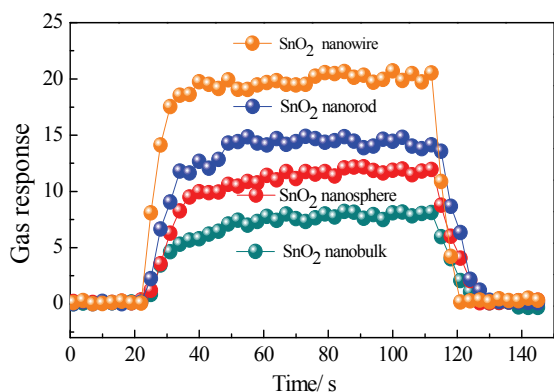


Fig. 7. Response and recovery characteristics of the sensor to 20 µL/L of C₂H₂.

The long-time stability and repeatability of the sensors have been also measured as shown in Fig. 8. One can clearly see that the prepared SnO₂ nanowires sensor changes slightly and exhibits a nearly constant sensing response during the long experimental cycles, confirming the excellent stability and good repeatability of the SnO₂ nanowires sensor for C₂H₂ detection.

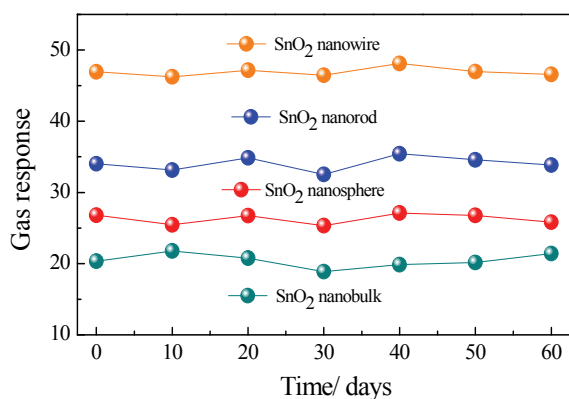
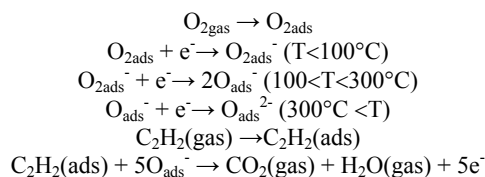


Fig. 8. Long-time stability of the prepared sensors to 100 µL/L of C₂H₂.

It is known to all that rutile SnO₂ is an important n-type semiconductor sensing material and the change in electrical resistance is dominantly controlled by the adsorption and desorption process of C₂H₂ molecules on SnO₂ surface. Due to the nonstoichiometry of the prepared SnO₂ nanostructures, many oxygen vacancies exist in their crystal lattices. When the SnO₂ based sensors were surrounded by air, oxygen molecules would be absorbed on the sensing surface to generate various kinds of chemical adsorbed oxygen species, including O₂⁻, O²⁻ and O⁻. Thus the sensor would exhibit a high electrical resistance. When the sensor was exposed certain concentration of C₂H₂ atmosphere, a typical reducing gas, a reaction happened between the adsorbed oxygen and C₂H₂ molecule. Meanwhile, the adsorbed electrons are released to SnO₂ surface,

which decreases the height of barrier in the depletion region and increases the carrier concentration. Therefore, a decreasing electrical resistance of the sensor was measured. A possible gas sensing process between the prepared SnO₂ nanostructures and C₂H₂ gas in this study can be expressed as follows [21].



To go insight into how the nanostructures and morphologies of the as-prepared SnO₂ samples influence C₂H₂ sensing behaviors, nitrogen adsorption and desorption measurements were further employed to estimate the texture properties. With the Barrett–Joyner–Halenda method [22], the BET surface area and pore structure parameters are calculated and shown in Table 1. With larger BET surface area and pores compared with those of nanobulks, nanospheres and nanorods, SnO₂ nanowires would provide much more reaction sites and promote charge transfer in the sensing process. Thus, excellent C₂H₂ sensing properties are observed in our measurements.

Table 1. BET surface area and pore structure parameters of the samples.

SnO ₂	S _{BET} (m ² g ⁻¹)	d _P (nm)
Nanowire	27.4	9.2
Nanorod	21.9	8.7
Nanobulk	17.6	6.8
Nanosphere	14.3	5.4

4. Conclusions

Various morphologies of SnO₂ nanostructures, including nanobulks, nanospheres, nanorods, and nanowires have been successfully prepared and performed with XRD and FESEM. Planar SnO₂ based gas sensors were fabricated with screen-printing and their gas sensing properties were systematically measured towards C₂H₂. The optimum operating temperatures of nanobulks, nanospheres, nanorods and nanowires are suggested to be about 280, 280, 250 and 250 °C. The corresponding response value of the sensors towards 100 µL/L of C₂H₂ are tested to be about 20.34, 26.80, 34.04 and 46.96 respectively and the response-recovery time are calculated to be about 15-25, 13-21, 11-18, 7-13 s, respectively. The prepared nanowires exhibit excellent gas sensing properties that those of nanobulks, nanospheres and nanorods, which implies a promising sensing morphology in fabricating high-performance C₂H₂ gas sensor.

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