

1 **Direct Measurement of the Exciton Binding Energy and Effective**
2 **Masses for Charge carriers in an Organic-Inorganic Tri-halide**
3 **Perovskite**

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16 **Solar cells based on the organic-inorganic tri-halide perovskite family of materials have**
17 **shown remarkable progress recently, offering the prospect of low-cost solar energy from**
18 **devices that are very simple to process. Fundamental to understanding the operation of**
19 **these devices is the exciton binding energy, which has proved both difficult to measure**
20 **directly and controversial. We demonstrate that by using very high magnetic fields it is**
21 **possible to make an accurate and direct spectroscopic measurement of the exciton binding**
22 **energy, which we find to be only 16 meV at low temperatures, over three times smaller**
23 **than has been previously assumed. In the room temperature phase we show that the**
24 **binding energy falls to even smaller values of only a few millielectronvolts, which explains**

25 **their excellent device performance due to spontaneous free carrier generation following**
26 **light absorption. Additionally, we determine the excitonic reduced effective mass to be**
27 **0.104m_e (where m_e is the electron mass), significantly smaller than previously estimated**
28 **experimentally but in good agreement with recent calculations. Our work provides crucial**
29 **information about the photophysics of these materials, which will in turn allow improved**
30 **optoelectronic device operation and better understanding of their electronic properties**

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33 The recent rapid development of perovskite solar cells is revolutionizing the photovoltaic (PV)
34 research field, with the latest certified power conversion efficiencies reaching over 20%¹.

35 Initially developed from the concept of the nanostructured excitonic solar cell where there is no
36 requirement for long range charge or exciton diffusion^[1-8], it has now become clear that due to

37 the remarkable properties of the inorganic-organic perovskite family of materials ABX₃

38 (A=CH₃NH₃⁺; B=Pb²⁺; and X = Cl⁻, I⁻ and/or Br⁻) these cells^[2-10] are capable of operating in a

39 comparable configuration and with comparable performance to the best inorganic

40 semiconductors,^[7, 9,10] where the solid absorber layer is sandwiched between n- and p-type

41 charge selective contacts in a planar heterojunction configuration^[7-8]. Despite this success

42 several fundamental properties of the organic lead tri-halide perovskites remain controversial and

43 poorly known. In particular the binding energy of the excitons (R*), bound electron-hole pairs

44 that are the primary photoexcited species created in the absorption process, is vital to

45 understanding the way that the cells function. The operating mechanisms depend upon what

46 fraction of excitons dissociate in the bulk material, giving rise to free charge transport, or what

47 fraction need to be dissociated at heterojunctions within the cells. Knowledge of the true exciton

48 binding energy is also crucial for interpreting spectroscopic measurements based on these
49 materials, such as time-resolved spectroscopy. Values for R^* reported in the literature cover a
50 broad range from 2 to 55 meV¹¹⁻¹⁷, with the larger values being initially adopted and a growing
51 number of reports suggesting a wide range of lower values¹⁵⁻¹⁷. In addition basic parameters such
52 as the effective masses of electrons and holes also remain to be directly measured in the
53 archetypical material $\text{CH}_3\text{NH}_3\text{PbI}_3$. A number of calculations of the band structure in the
54 literature are able to reproduce the observed band gaps^{14,18,19} and these suggest that the
55 conduction and valence bands are essentially isotropic and symmetrical^{14,18,19}. A direct
56 measurement of the exciton binding energy and effective masses is therefore crucial for our
57 current understanding and for future development of this remarkable class of materials.

58 In this work we describe the use of very high field inter-band magneto-absorption studies which
59 allow us, unlike previous PhotoLuminescence measurements²⁰, to make an accurate study of the
60 family of free exciton states, which are the relevant excitations created in PV devices. In the low
61 temperature orthorhombic phase we establish that the exciton binding energy is only 16 meV,
62 which is significantly smaller than has been previously assumed. We also investigate the room
63 temperature tetragonal phase, which occurs above 160K^{11,19}, where we find the striking result
64 that the binding energy falls to only a few millielectron volts. Our measurements give an
65 accurate and independent value for the reduced effective mass and in addition, show that the
66 essentially symmetric and isotropic conduction and valence bands of the organic-inorganic lead
67 tri-halide perovskites makes them model semiconductors to demonstrate the optical properties of
68 excitons in a high magnetic field.

69

70 In Fig. 1 we summarize the results of transmission measurements of a ~300-nm thick
 71 polycrystalline film of CH₃NH₃PbI₃, deposited directly on a glass substrate, measured at 2 K in
 72 magnetic fields of up to 65 T in the Faraday configuration using a long (500ms) pulsed field
 73 magnet. Close to the band edge (~1.6 eV), the spectra are dominated by the hydrogen-like
 74 exciton states and at higher energies the free carrier behaviour of the conduction and valence
 75 bands gives rise to a series of interband transitions between the van Hove singularities at the
 76 bottom of the Landau levels with energies given by:

$$77 \quad E(B) = E_g + (N + 1/2)\hbar\omega_c \pm 1/2 g_{eff}\mu_B B,$$

78 where E_g is the energy gap, $N=0,1,2,3..$ is the Landau quantum number, $\omega_c = eB/m^*$, B is the
 79 applied magnetic field, e is the elementary charge, m^* is the reduced effective mass of the
 80 exciton given by, $1/m^*=1/m^*_e+1/m^*_h$, where m_e and m_h are the electron and hole effective
 81 masses respectively, g_{eff} is the effective g -factor for the Zeeman splitting, and μ_B is the Bohr
 82 magneton. In the present case we use unpolarised light so that the last term is ignored. In contrast
 83 to the early magneto-optical studies on this material^{12,13} our spectra show a clearly resolved 1s
 84 exciton at 1.64 eV, which has a large diamagnetic shift, significantly larger than the peak width
 85 and larger than the values observed previously^{12,13}. In addition, the spectra we show in Fig. 1
 86 demonstrate several important new features as the field is increased. By 65T there is a clear
 87 sequence of 5 well-resolved Landau level transitions which develop in magnetic fields above
 88 30T. Furthermore, a small shoulder develops on the high energy side of the 1s exciton peak
 89 which we identify as the 2s exciton absorption. The appearance of the Landau levels and the 2s
 90 exciton state can be seen more clearly by taking the ratio of the high field spectra to zero field as
 91 shown in Fig. 1b and 1c, where the resonant absorptions become minima in the ratio of
 92 transmission. This shows that the 2s state is clearly visible as a weak but gradually growing

93 absorption in the range 10 - 35 T. Above 20 T a second absorption grows much more rapidly
94 which we attribute to a combination of the 2p exciton and the first of the free-electron inter-band
95 Landau level transitions. The 2p exciton transition is forbidden at zero magnetic field but
96 becomes allowed at high fields due to the re-construction of the hydrogenic energy levels which
97 occurs once the cyclotron energy exceeds the exciton binding energy ($\hbar\omega_c > R^*$)²¹. At high
98 fields the conventional atomic quantum numbers (n, l) are no longer valid and this state is
99 renamed as (1,0) corresponding to the strongest bound state associated with the N=1 Landau
100 level.

101

102 The separation between the sequence of Landau levels visible directly in the absorption spectrum
103 at 65T immediately allows us to estimate the reduced effective mass to be $m^* \approx 0.1 m_e$ to within
104 a few per cent accuracy, while the separation of the 1s and 2s states at low field (15 meV at 10 T)
105 allows us to estimate an excitonic binding energy, R^* , of order 20 meV, which is consistent with
106 the condition $\hbar\omega_c > R^*$ for fields above 14T in agreement with our preliminary analysis of fig.
107 1b. We will present a more detailed fitting to the full family of transitions below, which allows
108 us to deduce a precise value of R^* being 16 ± 2 meV. The values for the effective mass are in
109 good agreement with recent calculations^{14,18} and somewhat smaller than previous experimental
110 estimates. The excitonic binding energy is much smaller than the early spectroscopic values^{12,13}
111 of 30-50 meV although in good agreement with a recently calculated theoretical value¹⁵ which
112 takes into account the frequency dependent refractive index of the perovskite.

113

114 We now perform a full fitting, where we also include data taken up to 150T using a fast pulse
115 single turn magnetic field system at a fixed photon energy. In order to extract accurate values for

116 the effective mass and exciton binding energy from the magnetic field dependence of the
117 transition energies it is important to use full numerical calculations for the magnetic field
118 dependent transitions as the data cover a wide range from the low to the high magnetic field
119 limits and no analytical solution exists for the hydrogen atom in a high magnetic field. To do this
120 we use the values calculated by Makado and Magill²¹ which scale the most strongly bound
121 excitonic energy levels $E_{n,0}(\gamma)$ by the use of the dimensionless parameter $\gamma = \hbar\omega_c/2R^*$. This
122 allows us to fit the complete range of magnetic field values. In addition to the excitonic
123 transitions, the higher energy transitions are known²² to become dominated by simple interband
124 transitions between free carrier Landau levels as described above. Fig. 2 shows the full fan
125 diagram of measured transition energies, together with the calculated transition energies for the
126 excitonic and free electron transitions and a schematic of the optically allowed transitions
127 observed. The values for the effective mass and exciton binding energy are then adjusted to
128 globally fit the data. In practice the two parameters are dominated by very different parts of the
129 data set and are not strongly interdependent.

130

131 The observation of the 2s transition places strong constraints upon the exciton binding energy,
132 while the slope and separation of the high field, high quantum number Landau levels strongly
133 constrain the reduced effective mass value. Further confirmation of the exciton binding energy
134 comes from our simultaneous fitting of the diamagnetic shift of the 1s state, as shown in the inset
135 to Fig. 2a, where the binding energy is now also strongly constrained by the accurately
136 determined effective mass. We conclude that the excitonic transitions dominate for the N=0
137 Landau level (1s, 2s) and the N=1 (1,0) level up to ~50T, and free electrons dominate for N=1
138 above 50T and for all higher Landau levels. In the intermediate region a weak splitting of the

139 N=1 level can just be detected where both transitions are occurring. This clear demonstration of
140 the interchange between excitonic and interband free carrier transitions is usually masked in
141 more conventional semiconductors by complications brought about by additional degeneracies,
142 such as the light and heavy holes present in III-V or II-VI materials^{22,23} and is an illustration that
143 in addition to their potential in applications, the organic-inorganic metal tri-halide perovskites
144 can also act as excellent model semiconductors. A further interesting observation from the fan
145 charts in Fig. 2 is that the Landau level separations are not significantly decreasing at higher
146 energies (and in a related observation the Landau level fans show good linearity). This suggests
147 that the approximation of a constant effective mass is good for a wide range of band energies and
148 non-parabolicity effects are relatively small compared with some predictions²⁴.

149 The value deduced for the effective mass of $m^*=0.104\pm 0.003 m_e$ is remarkably close to
150 that of $0.099 m_e$ predicted recently by Menendez et al¹⁴ who have adjusted the amount of
151 exchange coupling in order to match the experimentally measured values of the band gap, and
152 Umari et al¹⁸, who find $0.11 m_e$. We find the exciton binding energy (R^*) to be 16 ± 2 meV, in
153 contrast to the earlier values of 37 to 50 meV reported in^{12, 13}, also determined at 4K, which
154 were deduced by fitting only the 1s state without an independent measurement of the effective
155 mass and using only a low magnetic field approximation and with much poorer experimental
156 resolution. Our value here is also strongly supported by the extrapolation of the free electron
157 transitions to zero magnetic field, which fixes the excitonic continuum.

158 Several authors¹⁴⁻¹⁷ have pointed out that using lower mass values ($0.1 m_e$) and
159 depending on whether the low ($\epsilon = 25.7$) or high ($\epsilon = 5.6$) frequency dielectric constant is
160 used^{18, 24} the conventional Wannier-Mott Hydrogenic model gives values for the excitonic
161 binding energy ($R^*=m^*e^4/2\hbar^2\epsilon^2$) anywhere from 2 -50 meV. In practice the exciton binding

162 energies are comparable to several of the phonon modes²⁵ and so it is likely that the appropriate
163 dielectric constant at the equivalent frequency will be intermediate between the low and high
164 frequency limits or could approach the low frequency limit as discussed by Huang and
165 Lambrecht²³ for CsSnX₃ perovskites. Our values of R* and m* suggest a value of $\epsilon \approx 9$
166 assuming the hydrogenic model. In fact Even et al¹⁵ have recently argued from fitting the
167 lineshape of the low temperature absorption that the exciton binding energy should be on the
168 order of 15 meV, and that as a result of the new rotational motion of the CH₃NH₃⁺ cations in the
169 high temperature phase there is an additional contribution to the dielectric screening which
170 makes the exciton binding energy fall discontinuously to 5 meV at the transition to the high
171 temperature phase where the photovoltaic cells operate. In a similar analysis of the temperature
172 dependent absorption, Yamada et al¹⁶ conclude that the exciton binding energy decreases
173 continuously from 30 meV at 13K to 6 meV at 300K.

174

175 An important factor in understanding the band structure of the organic-inorganic lead tri-halide
176 perovskite is the presence of structural phase transitions from cubic (T>350K) to tetragonal
177 (T>145K) to orthorhombic (low T) which result in changes in band structure and band gap^{11,19}.
178 The tetragonal (room temperature) and orthorhombic phases have similar band structures, with
179 almost symmetric and isotropic direct band gaps at the Γ -point, with the band gap increasing by
180 ~100 meV at the phase transition to the orthorhombic phase^{11,15, 16, 27}, although the magnitude of
181 the band gap change is dependent on the growth process^{15, 16}. Although the band structures are
182 very similar the phase transition may be expected to produce significant changes in the phonon
183 structure and consequently may significantly affect the dielectric constant and hence the exciton
184 binding energy as discussed above.

185

186 In Fig. 3, we extend the interband magneto-optical spectra up to 150T using a fast pulse single
187 turn magnetic field system, which shows the magnetic field dependent transmission for a series
188 of temperatures at a fixed photon energy. From this data it is very easy to observe the influence
189 of the lower temperature phase change. Two strong transmission minima can be observed in the
190 low temperature data which begin to move towards lower field values as the temperature
191 increases. This is because the perovskite band gap increases¹³ with increasing temperature,
192 causing the interband magneto-optical transitions at a fixed energy to shift towards lower
193 magnetic field values. Above 140K there is a sudden rapid shift upwards of the resonance fields
194 caused by the structural phase transition to the high temperature tetragonal phase, which has a
195 lower band gap. We then repeat the magneto-optical study in this tetragonal phase using the fast
196 pulse system as we show in fig. 3b. At these temperatures, significantly fewer resonances are
197 resolved, but it is still possible to observe several different inter-Landau level transitions and it is
198 possible to construct a good fan diagram(fig. 4b). This shows that the effective mass is
199 essentially the same ($m^*=0.104\pm 0.005 m_e$) as at low temperature in the orthorhombic phase,
200 although the band gap is slightly (~ 50 meV) smaller than at 2K.

201

202 We show the optical density spectra taken in the long pulse system in Fig. 4a, which shows that
203 the 1s exciton has considerably more broadening and as a result it is much harder to make an
204 accurate analysis of the diamagnetic shift of the 1s transition energy, in comparison to that of the
205 lower temperature phase. The exciton binding energy is more difficult to fit precisely because the
206 absolute energy of the 1s exciton state is less certain and would need to be fitted with a
207 knowledge of the scattering processes and the dielectric function. A detailed examination of the

208 1s exciton absorption spectra, as in fig. 4a, shows that there is an anomalous behavior in which
209 the transition becomes much more resolved at higher magnetic fields and actually moves down
210 in energy at first (dashed line, fig. 4a). Both the increase in intensity and the reduction in
211 transition energy with increasing field suggest that the exciton binding energy is increasing with
212 the magnetic field. We attribute this to a decrease in dielectric constant resulting from the
213 additional magnetic binding of the exciton.

214

215 Using only the high field ($B > 50$ T) spectra where Landau levels can be observed and the 1s
216 exciton peak is well formed, we estimate a binding energy on the order of 10-12 meV *in high*
217 *magnetic fields*. We extrapolate the high energy free carrier Landau levels to a band edge energy
218 very close to the apparent 1s exciton peak at $B=0$ which allows us to conclude that the exciton
219 binding energy is much smaller at zero magnetic field, with a value less than our measurement
220 uncertainty of a few meV. This strongly supports the suggestion by Even et al¹⁵ that the binding
221 energy is reduced to values of order 5 meV above the phase transition to the orthorhombic phase
222 and the recent analysis of Yamada et al¹⁶ that there is a decrease to around 6meV at room
223 temperature.

224

225 The overall picture is that there is a critical collapse of the exciton binding energy as a function
226 of both increasing temperature and decreasing magnetic field. As the temperature increases any
227 fall in binding energy decreases the frequency of motion of the electron and hole bound in the
228 exciton leading to increased contributions to the dielectric constant from the many phonon
229 modes and molecular rotations present in these materials. The fall in binding energy leads to a
230 further increase in dielectric constant and the binding energy collapses to a value close to that

231 predicted by using the low frequency dielectric constant, giving a value of a few meV. When a
232 magnetic field is applied the process is partly reversed due to the additional cyclotron motion
233 which increases the binding energies. Similarly low exciton binding energies due to a frequency
234 dependent dielectric constant have been predicted²⁶ in the CsSnX₃ (X = I, Br, Cl) perovskite
235 halide semiconductors which have very similar band and crystal structures to the organic-
236 inorganic lead tri-halide perovskites studied here, previously leading to significant experimental
237 controversy^{26,28}. An immediate consequence of the temperature and magnetic field dependence
238 of the exciton binding energy is that the interpretation of the earlier literature measurements as
239 giving an exciton binding energy of 30-50 meV, which are based on the assumption of a constant
240 binding energy as a function of magnetic field^{12,13}, or temperature¹¹, will be invalidated. Hence,
241 our measurements show that the exciton binding energies are much smaller than previously
242 concluded and explain why the properties of the organic-inorganic perovskites in the room
243 temperature phase will be dominated by free carrier behavior as suggested by more recent time
244 resolved studies^{17,20,25,27,30,31}.

245
246 We also show in Fig. 4 the temperature dependence of the magnetic field position of the N=1
247 Landau level resonance, compared with the temperature dependence of the band gap as deduced
248 from the film absorption. The sudden shift in resonance position of 63T at the phase transition
249 temperature can be shown to be equivalent to a change in band gap of 105 meV from the slope of
250 the N=1 resonance (1.67 meV/T), consistent with the 100 meV change in band gap deduced from
251 absorption. This confirms that the magneto-spectroscopy is measuring the same fundamental
252 band structure as the optical absorption.
253

254 Our basic conclusion is that the excitonic binding energy in the low temperature phase of the
255 organic-inorganic perovskite, $\text{CH}_3\text{NH}_3\text{PbI}_3$ is much smaller (16 ± 2 meV) than has previously
256 been estimated and is comparable to conventional III-V semiconductors with a similar band
257 gap²⁹. Our measurements in the room temperature phase suggest that there is a critical collapse
258 of the exciton binding energies at higher temperatures, as has been predicted by Even et al¹⁵, due
259 to the frequency dependent dielectric constant. By room temperature the appropriate binding
260 energy will be only a few milli-electron-volts and the photovoltaic device performance is
261 essentially a free carrier phenomenon. This result conclusively shows that the very impressive
262 performance of PV devices using this material¹⁻¹⁰ can be attributed to the spontaneous generation
263 of free electrons and holes following photo-absorption, thereby also resolving the apparent
264 contradiction between initial reports of sizeable exciton binding energy values (30-50 meV) and
265 recent reports of free carrier behavior^{16,17,20,25,30-32}. The reason for the difference between our
266 observations and previous estimates of the binding energy is primarily the combination of higher
267 quality, much more crystalline films, and the use of very high magnetic fields. This has enabled
268 us to measure multiple excitonic transitions which allow precise spectroscopic measurements, in
269 contrast to previous works which have relied exclusively on measuring the 1s exciton and require
270 assumptions to be made about the dielectric constant and effective masses^{11-13, 17}. The effective
271 mass values of $\sim 0.1 m_e$, which we have determined, are in good agreement with recent
272 calculations^{14,18,19,24} but are also significantly lower than the earlier experimental estimates^{12,13}.

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363

364 **Author contribution**

365 Atsuhiko Miyata, Anatolie Mitioglu, P.P, O.P and R.J.N collected and analysed the data. J. T-
366 W. W. and S.D.S. prepared the samples. All authors contributed to the interpretation and the
367 manuscript preparation. R.J.N. supervised and initiated the project.

368

369 **Figure captions**

370

371 **Figure 1: Magnetic field dependence of the optical density for the perovskite $\text{CH}_3\text{NH}_3\text{PbI}_3$**
372 (a) A sequence of optical density ($\text{Log}(1/\text{transmission})$) spectra measured during a single pulse

373 of the magnetic field. To improve the resolution of smaller field dependent features (b) and (c)
374 show sequences of ratios of the transmission in magnetic field $T(B)$ to that measured at zero field
375 $T(0)$, where the resonant absorption features correspond to minima. For easier comparison
376 spectra are offset. The feature highlighted at 2.03 eV is a band edge absorption from a
377 previously undetected higher energy band edge and will be the subject of a future publication.

378

379 **Figure 2: Energy ‘Fan’ diagrams.** (a) Full fan using data from long pulse fixed field spectra
380 (black circles) and fixed energy fast field sweep data (red stars). The calculated transition
381 energies are shown for the free electron and hole levels (solid lines) and the excitonic transitions
382 (dashed lines). Inset to (a) lower fields measured using fixed field spectra. (b) shows a schematic
383 of the energy levels and transitions between the free electron and hole levels (solid lines) and the
384 excitonic transitions (dashed lines).

385

386 **Figure 3: Single turn coil results.** Plots of the magneto-transmission measured using a single
387 turn fast pulse magnetic field, with schematic of the experimental system. For better comparison
388 spectra are offset. (a) shows the temperature dependence with the arrows indicating the position
389 of the $N=1$ inter Landau level transition. The inset shows a schematic of the single turn
390 experimental system showing the coil and its firing circuit, the sample and the optical fiber
391 illumination and collection system. (b) shows the transmission spectra in the tetragonal
392 (intermediate temperature) phase with spectra measured for different wavelengths in the
393 temperature range 155 – 190 K. The linked arrows show the positions of the $N=1$ and $N=2$
394 transitions.

395 **Figure 4: Transmission in high temperature tetragonal phase** (a) optical density, with the
396 dashed line showing the anomalous behavior of the 1s exciton transition. (b) fan diagram. (c)
397 Temperature dependent band gaps and resonance positions for the N=1 inter-Landau level
398 transition.

399

400

401

402 **Methods**

403 Samples were prepared on glass substrates cleaned sequentially in hallmanex, acetone,
404 isopropanol and O₂ plasma. The polycrystalline CH₃NH₃PbI₃ perovskite films were deposited in
405 a nitrogen-filled glovebox following the interdiffusion preparation methods described
406 previously⁵. In brief, a PbI₂ layer was first deposited on cleaned glass by spin-coating
407 (speed/ramp = 6000rpm/6000 rpm/s, time = 35s) from a precursor solution of PbI₂ in DMF with
408 concentration of 450 mg/ml, followed by drying at 70°C for 5 min. Then the MAI layer was
409 deposited on dried PbI₂ film by spin-coating (speed/ramp = 6000rpm/6000 rpm/s, time = 35s)
410 from a precursor solution of MAI in isopropanol with concentration of 50 mg/ml, followed by
411 annealing at 100°C for 1 hour. The perovskite films were sealed by spin-coating a layer of the
412 insulating polymer poly(methyl methacrylate) (PMMA; 10 mg/ml, speed/ramp = 1000rpm/1000
413 rpm/s, time = 60s) on top in order to ensure air-and moisture-insensitivity.

414

415 The magneto-optical measurements have been performed using 70 T long-duration and 150T
416 short duration pulsed magnets in the high magnetic field laboratory in Toulouse. For the long

417 pulsed measurements (~500 ms duration) the sample was immersed in liquid or gaseous helium
418 in a cryostat. A tungsten halogen lamp was used to provide broad spectrum in the visible and
419 near infra-red range. The absorption was measured in the Faraday configuration in which k , the
420 wave propagation vector is parallel to the magnetic field B . A nitrogen cooled CCD array
421 coupled to a spectrometer collected the light transmitted through the sample. The exposure time
422 was 3 ms in order to limit variations in the magnetic field during acquisition. Thirty spectra were
423 taken during a 70 T shot of the magnetic field. The magnetic field was measured using a
424 calibrated pick-up coil. All spectra were normalized to both the incident intensity and by the zero
425 field transmission to produce absolute and differential transmission spectra. For the short
426 duration (10 μ s) pulsed measurements a series of diode and Ti-Sapphire laser lines was sent
427 through and collected from the sample using fiber optics and detected using a fast (100MHz)
428 silicon detector and high speed digital oscilloscope³³. The sample was mounted inside a non-
429 conducting helium flow cryostat and was cooled separately for each measurement. The magnetic
430 fields were generated by a semi-destructive single turn coil system using 10mm coils^{33,34}.







