



Bushell, Z. L., Broderick, C. A., Nattermann, L., Joseph, R., Keddie, J. L., Rorison, J. M., ... Sweeney, S. J. (2019). Giant bowing of the band gap and spin-orbit splitting energy in GaP<sub>1-x</sub>Bi<sub>x</sub> dilute bismide alloys. *Scientific Reports*, 9(1), [6835]. <https://doi.org/10.1038/s41598-019-43142-5>

Publisher's PDF, also known as Version of record

License (if available):  
CC BY

Link to published version (if available):  
[10.1038/s41598-019-43142-5](https://doi.org/10.1038/s41598-019-43142-5)

[Link to publication record in Explore Bristol Research](#)  
PDF-document

This is the final published version of the article (version of record). It first appeared online via Springer Nature at <https://doi.org/10.1038/s41598-019-43142-5> . Please refer to any applicable terms of use of the publisher.

## University of Bristol - Explore Bristol Research

### General rights

This document is made available in accordance with publisher policies. Please cite only the published version using the reference above. Full terms of use are available:  
<http://www.bristol.ac.uk/pure/about/ebr-terms>

# SCIENTIFIC REPORTS

OPEN

## Giant bowing of the band gap and spin-orbit splitting energy in GaP<sub>1-x</sub>Bi<sub>x</sub> dilute bismide alloys

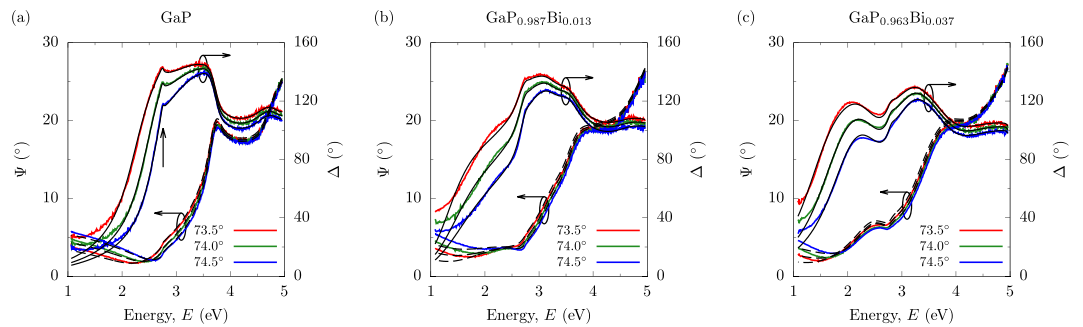
Zoe L. Bushell<sup>1</sup>, Christopher A. Broderick<sup>2,3</sup>, Lukas Nattermann<sup>4</sup>, Rita Joseph<sup>1</sup>, Joseph L. Keddie<sup>1</sup>, Judy M. Rorison<sup>3</sup>, Kerstin Volz<sup>4</sup> & Stephen J. Sweeney<sup>1</sup>

Using spectroscopic ellipsometry measurements on GaP<sub>1-x</sub>Bi<sub>x</sub>/GaP epitaxial layers up to  $x = 3.7\%$  we observe a giant bowing of the direct band gap ( $E_g^\Gamma$ ) and valence band spin-orbit splitting energy ( $\Delta_{SO}$ ).  $E_g^\Gamma$  ( $\Delta_{SO}$ ) is measured to decrease (increase) by approximately 200 meV (240 meV) with the incorporation of 1% Bi, corresponding to a greater than fourfold increase in  $\Delta_{SO}$  in going from GaP to GaP<sub>0.99</sub>Bi<sub>0.01</sub>. The evolution of  $E_g^\Gamma$  and  $\Delta_{SO}$  with  $x$  is characterised by strong, composition-dependent bowing. We demonstrate that a simple valence band-anticrossing model, parametrised directly from atomistic supercell calculations, quantitatively describes the measured evolution of  $E_g^\Gamma$  and  $\Delta_{SO}$  with  $x$ . In contrast to the well-studied GaAs<sub>1-x</sub>Bi<sub>x</sub> alloy, in GaP<sub>1-x</sub>Bi<sub>x</sub> substitutional Bi creates localised impurity states lying energetically within the GaP host matrix band gap. This leads to the emergence of an optically active band of Bi-hybridised states, accounting for the overall large bowing of  $E_g^\Gamma$  and  $\Delta_{SO}$  and in particular for the giant bowing observed for  $x \lesssim 1\%$ . Our analysis provides insight into the action of Bi as an isovalent impurity, and constitutes the first detailed experimental and theoretical analysis of the GaP<sub>1-x</sub>Bi<sub>x</sub> alloy band structure.

Highly-mismatched III-V semiconductor alloys containing dilute concentrations of bismuth (Bi) have attracted significant attention in recent years<sup>1</sup> since their unique electronic properties open up a range of possibilities for practical applications in semiconductor lasers<sup>2-16</sup>, photovoltaics<sup>17,18</sup>, spintronics<sup>19-21</sup>, photodiodes<sup>22-25</sup>, and thermoelectrics<sup>26</sup>. Research on dilute bismide alloys has primarily focused to date on GaAs<sub>1-x</sub>Bi<sub>x</sub>, where incorporation of Bi brings about a strong reduction of the direct  $\Gamma$ -point band gap ( $E_g^\Gamma$ )—by up to 90 meV per % Bi at low Bi compositions  $x^{27-31}$ —characterised by strong, composition-dependent bowing<sup>29,32</sup>. This unusual behaviour derives from the large differences in size (covalent radius) and chemical properties (electronegativity) between As and Bi: Bi, being significantly larger and more electropositive than As, acts as an isovalent impurity which primarily impacts and strongly perturbs the valence band (VB) structure<sup>30,33,34</sup>. This is in contrast to dilute nitride alloys, in which small electronegative nitrogen (N) atoms strongly perturb the conduction band (CB) structure in GaN<sub>x</sub>As<sub>1-x</sub> and related alloys<sup>35-38</sup>. Additionally Bi, being the largest stable group-V element, has strong relativistic (spin-orbit coupling) effects<sup>39</sup>. As such, the reduction of  $E_g^\Gamma$  in (In)GaAs<sub>1-x</sub>Bi<sub>x</sub> is accompanied by a strong increase in the VB spin-orbit splitting energy ( $\Delta_{SO}$ )<sup>16,30,31,40</sup>.

Epitaxial growth of GaP<sub>1-x</sub>Bi<sub>x</sub> alloys, via molecular beam epitaxy<sup>41,42</sup> and metal-organic vapour phase epitaxy<sup>43</sup> (MOVPE), has only recently been attempted. Here, we present the first detailed analysis of the GaP<sub>1-x</sub>Bi<sub>x</sub> electronic band structure. Early experiments on impurities in GaP can be traced back to the advent of semiconductors, with the initial experiments of Trumbore *et al.*<sup>44</sup> revealing that Bi dopants generate bound localised impurity states in GaP, i.e. Bi-related localised impurity states lying energetically within the GaP host matrix band gap. However, there is little further data available regarding the GaP<sub>1-x</sub>Bi<sub>x</sub> band structure. Here, we explicitly verify that the evolution of the main features of the GaP<sub>1-x</sub>Bi<sub>x</sub> VB structure with Bi composition  $x$  can be understood straightforwardly in terms of an  $x$ -dependent valence band-anticrossing (VBAC) interaction between the extended states of the GaP VB edge and localised bound impurity states associated with substitutional Bi impurities. The VBAC interaction between these extended and localised states produces a set of Bi-hybridised bands,

<sup>1</sup>Advanced Technology Institute and Department of Physics, University of Surrey, Guildford, GU2 7XH, UK. <sup>2</sup>Tyndall National Institute, Lee Maltings, Dyke Parade, Cork, T12 R5CP, Ireland. <sup>3</sup>Department of Electrical and Electronic Engineering, University of Bristol, Bristol, BS8 1UB, UK. <sup>4</sup>Materials Science Center and Faculty of Physics, Philipps-Universität Marburg, 35032, Marburg, Germany. Correspondence and requests for materials should be addressed to C.A.B. (email: [c.broderick@umail.ucc.ie](mailto:c.broderick@umail.ucc.ie)) or S.J.S. (email: [s.sweeney@surrey.ac.uk](mailto:s.sweeney@surrey.ac.uk))



**Figure 1.** Measured SE spectra for the MOVPE-grown (a) GaP (Bi-free,  $x = 0$ ), (b) GaP<sub>0.987</sub>Bi<sub>0.013</sub> ( $x = 1.3\%$ ), and (c) GaP<sub>0.963</sub>Bi<sub>0.037</sub> ( $x = 3.7\%$ ) samples described in the text and in ref.<sup>43</sup>. Solid red, green and blue lines respectively denote data measured for incident beam angles of 73.5°, 74.0° and 74.5°. Solid (dashed) black lines show the fits to the measured  $\Delta$  ( $\Psi$ ) spectra; the SE model and associated fitting procedure are outlined in the text.

with the GaP<sub>1-x</sub>Bi<sub>x</sub> alloy VB edge then being a primarily Bi-derived band possessing an admixture of GaP VB edge Bloch character<sup>30</sup>, enabling optical coupling to the comparatively unperturbed  $\Gamma_{6c}$  CB states. Comparison between theory and experiment highlights the emergence of this Bi-derived impurity band lying energetically within the GaP band gap, close in energy to the unperturbed GaP VB edge. Our analysis reveals the giant bowing of  $E_g^\Gamma$  and  $\Delta_{SO}$  due to this VBAC interaction:  $E_g^\Gamma$  ( $\Delta_{SO}$ ) decreases (increases) by  $\approx 200$  meV (240 meV) when 1% Bi is incorporated substitutionally in GaP.

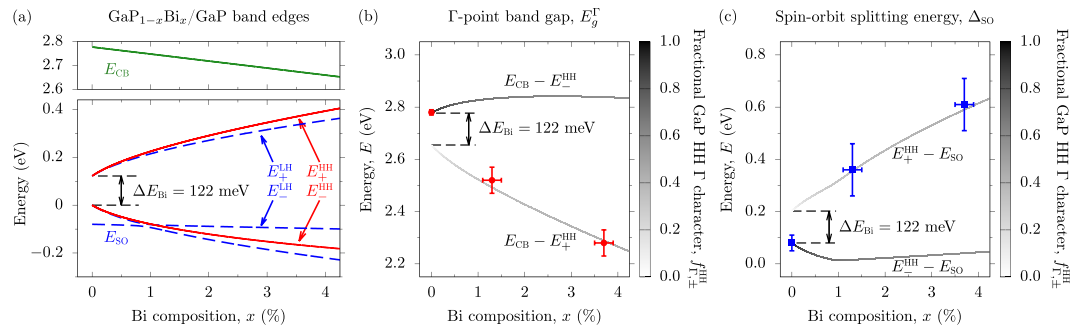
We begin here by describing our measurement procedure and the general features of our experimental results. Next, we describe our theoretical model, which is then applied to analyse the trends revealed by the experimental measurements. Finally, we use the results of this combined theoretical and experimental analysis to describe general features of the GaP<sub>1-x</sub>Bi<sub>x</sub> band structure.

## Experimental Measurements

Spectroscopic ellipsometry (SE) was used to study bulk-like GaP<sub>1-x</sub>Bi<sub>x</sub> epitaxial layers containing up to 3.7% Bi. The samples were grown on (001)-oriented GaP by MOVPE, with x-ray diffraction measurements confirming that the layers are in a state of compressive pseudomorphic strain. Full details of the sample growth and characterisation—including data from high-resolution x-ray diffraction, atomic force microscopy, secondary ion mass spectrometry, and scanning transmission electronic microscopy measurements—can be found in ref.<sup>43</sup>. The SE measurements were performed at room temperature using a J. A. Woollam Co. variable angle spectroscopic ellipsometer. Three incident beam angles were used to generate sufficient data to provide confidence in modelling fits to the measured spectra. Angles of 73.5°, 74.0° and 74.5° were chosen since they are close to the pseudo-Brewster angles of the samples under investigation, thereby ensuring that the phase change on reflection  $\Delta$  (measured relative to the sample normal) remained close to 90°. When  $\Delta$  is close to 0 or 180°, the level of noise in the measured SE spectra is increased, and the sensitivity of the ellipsometric parameters to small changes in the optical properties is decreased. Keeping  $\Delta$  close to 90° increases both the precision and accuracy of the measurements<sup>45</sup>. A carefully defined modelling and fitting procedure was used to extract the energies corresponding to critical points in the band structure from the measured SE data, allowing the energies corresponding to the  $E_g^\Gamma$  and  $E_g^\Gamma + \Delta_{SO}$  inter-band transitions to be extracted<sup>46</sup>. Full details of this fitting procedure are presented as Supplementary Material.

The solid red, green and blue lines in Fig. 1(a–c) respectively show the measured SE data—where  $\tan(\Psi)$  is related to the change in the ratio of the amplitudes of the  $p$ - and  $s$ -polarisations upon reflection<sup>45</sup>—in the GaP, GaP<sub>0.987</sub>Bi<sub>0.013</sub> and GaP<sub>0.963</sub>Bi<sub>0.037</sub> samples. Solid (dashed) black lines show the corresponding fits to  $\Delta$  ( $\Psi$ ). The GaP sample consists of an epitaxial GaP buffer layer grown on a GaP substrate, and was analysed first in order to obtain accurate input parameters for the SE fits. These parameters were then used to describe the substrate and buffer layer in the subsequent models of the Bi-containing samples. Following this procedure it was possible to achieve good fits to the key features observed in the measured GaP<sub>1-x</sub>Bi<sub>x</sub> SE spectra (cf. Fig. 1(b,c)). In Fig. 1(a) a clear feature associated with  $E_g^\Gamma$  is visible in the measured  $\Delta$  and  $\Psi$  spectra, which is well described by a modelling fit corresponding to a  $\Gamma$ -point GaP room temperature band gap  $E_g^\Gamma = 2.76$  eV. The slight deviation from the accepted value of 2.78 eV is attributable to the sample growth taking place on n-doped GaP substrates<sup>47</sup>.

Turning to Fig. 1(b,c) we note from the measured  $\Delta$  spectra that Bi incorporation gives rise to an additional feature at lower energy than  $E_g^\Gamma$  in GaP, which shifts to even lower energies with increasing  $x$ . This indicates a large reduction of  $E_g^\Gamma$ , in agreement with theoretical predictions<sup>30,48,49</sup>. The spectral features associated with  $E_g^\Gamma$  are significantly broader in the Bi-containing samples than in GaP. This is likely associated with the presence of Bi composition fluctuations across the samples, as well as short-range alloy disorder, associated with the formation of pairs and larger clusters of Bi atoms sharing common Ga nearest neighbours in a substitutional alloy<sup>50,51</sup>. Using the fitting procedure outlined in the Supplementary Material it was also possible to extract the energies associated with the  $E_g^\Gamma + \Delta_{SO}$  transitions in each sample. The values of  $E_g^\Gamma$  and  $\Delta_{SO}$  extracted in this manner are shown respectively in Fig. 2(b,c), using closed red circles and blue squares. We note that the uncertainties in these data are associated with the broadening of the corresponding features in the measured spectra<sup>46</sup>. Overall, the SE



**Figure 2.** (a) Calculated variation of the  $\Gamma$ -point band edge energies with  $x$  in pseudomorphically strained  $\text{GaP}_{1-x}\text{Bi}_x/\text{GaP}$ . Solid green and red lines respectively denote the CB- and HH-like band edge energies, while dashed blue lines denote the LH- and SO-like band edge energies. (b) Variation of the  $\text{GaP}_{1-x}\text{Bi}_x/\text{GaP}$   $\Gamma$ -point band gaps  $E_{\text{CB}} - E_{\pm}^{\text{HH}}$  with  $x$ , calculated (shaded lines) and extracted from SE measurements (closed red circles). The line shading is determined by the fractional GaP HH  $\Gamma$  character  $f_{\Gamma, \pm}^{\text{HH}}$  of the associated HH-like  $\text{GaP}_{1-x}\text{Bi}_x$  VB states  $E_{\Gamma, \pm}^{\text{HH}}$ . (c) Variation of the  $\text{GaP}_{1-x}\text{Bi}_x/\text{GaP}$  VB spin-orbit splitting energies  $E_{\pm}^{\text{HH}} - E_{\text{SO}}$  with  $x$ , calculated (shaded lines) and extracted from SE measurements (closed blue circles). The line shading is as in (b).

measurements indicate that incorporation of dilute concentrations of Bi is sufficient to cause a giant reduction (increase) and bowing of  $E_g^{\Gamma}$  ( $\Delta_{\text{SO}}$ ).

## Theoretical Calculations

To understand this unusual behaviour we have used alloy supercell electronic structure calculations to analyse the contributions to the Bi-induced changes in the band edge energies, and to parametrise a suitable VBAC model for  $\text{GaP}_{1-x}\text{Bi}_x$ . This approach does not rely on post hoc fitting to alloy band structure data, thereby providing a predictive capability commonly lacking in models of this type<sup>32</sup>. In ref.<sup>30</sup> we employed an atomistic tight-binding (TB) model to analyse the electronic structure of ordered and disordered  $\text{GaP}_{1-x}\text{Bi}_x$  alloys. By directly constructing the  $T_2$ -symmetric localised states  $|\psi_{\text{Bi}}\rangle$  associated with an isolated, substitutional Bi impurity we predicted the presence of a VBAC interaction having a composition dependence  $\beta\sqrt{x}$ . In the dilute doping (large supercell) limit we determined that the Bi-related localised states in  $\text{GaP}:\text{Bi}$  lie approximately 120 meV above the unperturbed GaP VB edge, in good agreement with experiment<sup>44</sup>. Analysis of the electronic structure of ordered  $\text{Ga}(\text{P,As})_{1-x}\text{Bi}_x$  alloys elucidates the differences in the impact of Bi incorporation on the band structure: the natural VB offsets between GaP, GaAs and GaBi lead to the  $6p$  valence orbitals of Bi lying below the  $4p$  valence orbitals of As in energy, but higher in energy than the  $3p$  valence orbitals of P. As such, a substitutional Bi impurity forms a resonant localised state lying energetically below the VB edge in GaAs, but a bound localised state lying above the VB edge in energy in GaP<sup>30</sup>.

Building on our initial analysis of  $\text{GaP}_{1-x}\text{Bi}_x$  we have derived an extended basis set 12-band (VBAC)  $\mathbf{k} \cdot \mathbf{p}$  Hamiltonian to describe the dilute bismide band structure<sup>32</sup>. Using the TB model of ref.<sup>30</sup> we have directly evaluated the Bi-related parameters of this model, including the distinct VBAC, virtual crystal (VC) and strain-related contributions to the Bi-induced shifts in the band edge energies<sup>52</sup>. To analyse the SE measurements we focus on the band edge energies at the zone centre: at  $\Gamma$  the 12-band Hamiltonian diagonalises into decoupled blocks describing the CB, heavy-hole (HH), light-hole (LH) and spin-split-off (SO) band edges<sup>52</sup>. As in  $\text{GaAs}_{1-x}\text{Bi}_x$ , the energy of the  $\text{GaP}_{1-x}\text{Bi}_x$   $\Gamma$ -point CB state  $\Gamma_{6c}$  is well described as  $E_{\text{CB}}(x) = E_g^{\Gamma}(0) - \alpha x + \delta E_{\text{CB}}^{\text{hy}}$ , where the zero of energy has been chosen at the unperturbed GaP VB edge,  $E_g^{\Gamma}(0) = 2.78$  eV is the host matrix band gap,  $\alpha$  describes the VC shift of the CB edge energy, and  $\delta E_{\text{CB}}^{\text{hy}}$  is the energy shift associated with the hydrostatic component of the compressive pseudomorphic strain in a  $\text{GaP}_{1-x}\text{Bi}_x/\text{GaP}$  epitaxial layer<sup>52</sup>.

The energies of the HH-like alloy VB states are given in the 12-band VBAC model as the eigenvalues of the  $2 \times 2$  matrix<sup>30,52</sup>

$$\begin{pmatrix} \Delta E_{\text{Bi}} + \delta E_{\text{Bi}}^{\text{hy}} - \delta E_{\text{Bi}}^{\text{ax}} & \beta\sqrt{x} \\ \beta\sqrt{x} & \kappa x + \delta E_{\text{VB}}^{\text{hy}} - \delta E_{\text{VB}}^{\text{ax}} \end{pmatrix} \begin{pmatrix} |\psi_{\text{Bi}}^{\text{HH}}\rangle \\ |\psi_{\text{HH}}^{(0)}\rangle \end{pmatrix}, \quad (1)$$

where  $\kappa x + \delta E_{\text{VB}}^{\text{hy}} - \delta E_{\text{VB}}^{\text{ax}}$  describes the VC, hydrostatic and axial strain-induced shifts to the GaP HH band edge energy, and  $\Delta E_{\text{Bi}} + \delta E_{\text{Bi}}^{\text{hy}} - \delta E_{\text{Bi}}^{\text{ax}}$  is the energy of the HH-like Bi-related localised states relative to the zero of energy at the unperturbed GaP VB edge<sup>52</sup>. The energies of the LH- and SO-like VB states are given as the eigenvalues of a  $3 \times 3$  matrix which can be found, along with full details of the model, in ref.<sup>52</sup>.

The Bi-related band structure parameters computed for  $\text{GaP}_{1-x}\text{Bi}_x$  are summarised in Table 1 where, for comparative purposes, the corresponding parameters computed for  $\text{GaAs}_{1-x}\text{Bi}_x$  are provided. Comparing these two sets of VBAC parameters we firstly note that while a substitutional Bi impurity in GaAs leads to the formation of a resonant localised state lying 183 meV below the VB edge ( $\Delta E_{\text{Bi}} < 0$ ), in GaP it leads to the formation of a bound localised state lying 122 meV above the VB edge ( $\Delta E_{\text{Bi}} > 0$ ). In both cases the alloy VB edge consists of an admixture of extended (Bloch) and localised (Bi-related) character. However, in  $\text{GaAs}_{1-x}\text{Bi}_x$  the alloy VB edge states are primarily GaAs-derived, retaining significant Bloch character, while in  $\text{GaP}_{1-x}\text{Bi}_x$  the alloy VB edge states are primarily Bi-derived, having low overall Bloch character<sup>30</sup>. As we will describe in further detail below,

Parameter	GaP <sub>1-x</sub> Bi <sub>x</sub>	GaAs <sub>1-x</sub> Bi <sub>x</sub>
$\Delta E_{\text{Bi}}$ (eV)	0.122	-0.183
$\alpha$ (eV)	4.39	2.82
$\beta$ (eV)	1.41	1.13
$\gamma$ (eV)	0.24	0.55
$\kappa$ (eV)	1.47	1.01

**Table 1.** Bi-related parameters for the 12-band (VBAC)  $\mathbf{k} \cdot \mathbf{p}$  Hamiltonian of Ga(P,As)<sub>1-x</sub>Bi<sub>x</sub>, computed using atomistic TB calculations on ordered alloy supercells. The energy  $\Delta E_{\text{Bi}}$  of the Bi-related localised impurity states is given relative to the unperturbed Ga(P,As) host matrix VB edge<sup>30,32,52</sup>.

this key difference describes the fundamentally distinct nature of the perturbed VB structure in GaAs<sub>1-x</sub>Bi<sub>x</sub> and GaP<sub>1-x</sub>Bi<sub>x</sub> alloys, and has significant consequences for the electronic and optical properties. We note that this behaviour is qualitatively similar to the stark differences in the CB structure of the dilute nitride alloys GaN<sub>x</sub>As<sub>1-x</sub> and GaN<sub>x</sub>P<sub>1-x</sub>: a substitutional N impurity in GaAs creates a N-related localised state which is resonant with the CB, while in GaP it creates a bound impurity state lying energetically within the band gap<sup>35,38,53,54</sup>.

Considering the VBAC coupling parameter  $\beta$ , we note that a larger value is calculated for GaP<sub>1-x</sub>Bi<sub>x</sub>. This describes that the VBAC interaction is more pronounced in GaP<sub>1-x</sub>Bi<sub>x</sub> than in GaAs<sub>1-x</sub>Bi<sub>x</sub>—i.e. that a substitutional Bi impurity more strongly perturbs the VB structure in GaP than in GaAs—reflecting the larger differences in size and electronegativity between P and Bi than between As and Bi. We note that  $\beta$  in GaP<sub>1-x</sub>Bi<sub>x</sub> is comparable to that calculated previously for the GaN<sub>x</sub>P<sub>1-x</sub> CB ( $\beta = 1.74$  eV)<sup>38,54</sup>. Turning our attention to the VC contributions to the band offsets—described via the parameters  $\alpha$ ,  $\kappa$  and  $\gamma$  for the CB, VB and SO offsets respectively—we again note that the calculated differences in these parameters for GaP<sub>1-x</sub>Bi<sub>x</sub> and GaAs<sub>1-x</sub>Bi<sub>x</sub> alloys reflect the associated trends in the lattice mismatch and natural band offsets between GaP, GaAs and the fictitious semimetallic zinc blende compound GaBi<sup>30,32,33,55</sup>. The calculated values  $\alpha = 4.39$  and 2.82 eV for GaP<sub>1-x</sub>Bi<sub>x</sub> and GaAs<sub>1-x</sub>Bi<sub>x</sub> respectively describe increases of approximately 44 and 28 meV per % Bi of the type-I zone-centre CB offset in free-standing GaP<sub>1-x</sub>Bi<sub>x</sub>/GaP and GaAs<sub>1-x</sub>Bi<sub>x</sub>/GaAs, reflecting the larger difference in energy between the  $\Gamma_{6c}$  CB edge states in GaP and GaBi compared to that between GaAs and GaBi. For the VB offset, the larger calculated value of  $\kappa$  for GaP<sub>1-x</sub>Bi<sub>x</sub> describes the larger VB offset between GaP and GaBi than between GaAs and GaBi. Similarly, the smaller calculated value of  $\gamma$  for GaP<sub>1-x</sub>Bi<sub>x</sub> describes the smaller SO band offset between GaP and GaBi than between GaAs and GaBi, and results from a combination of (i) a natural VB offset between GaP and GaAs which exceeds the difference in  $\Delta_{\text{SO}}$  between the two compounds<sup>56</sup>, and (ii) the extremely large predicted VB spin-orbit splitting in GaBi<sup>39</sup>.

Finally, since the III-V compound AlAs has approximately the same natural VB offset relative to GaAs as GaP<sup>56</sup>, we note that the VB structure of the AlAs<sub>1-x</sub>Bi<sub>x</sub> alloy—growth of which has also recently been established<sup>57</sup>—is expected to be qualitatively the same as in GaP<sub>1-x</sub>Bi<sub>x</sub>. That is, based on the known chemical trends for III-V compounds, it can be expected based on the analysis presented here that (i) a substitutional Bi impurity in AlAs leads to the formation of a bound Bi-related localised impurity state, having the same general character as that described here in GaP, and (ii) this leads in turn to a large decrease (increase) and composition-dependent bowing of  $E_g^{\Gamma}$  ( $\Delta_{\text{SO}}$ ) at dilute Bi compositions  $x$ <sup>58</sup>.

## Results

Figure 2(a) shows the calculated variation of the  $\Gamma$ -point band edge energies with  $x$  in pseudomorphically strained GaP<sub>1-x</sub>Bi<sub>x</sub>/GaP, for the CB ( $E_{\text{CB}}$ , solid black line), HH ( $E_{\pm}^{\text{HH}}$ , solid red lines), LH ( $E_{\pm}^{\text{LH}}$ , dashed blue lines) and SO ( $E_{\text{SO}}$ , dash-dotted green lines) states, respectively. The hydrostatic component of the pseudomorphic strain pushes the CB (HH, LH and SO) edge(s) upwards (downwards) in energy, while the axial component lifts the degeneracy of VB edge in the usual manner<sup>59,60</sup>. Since the GaP<sub>1-x</sub>Bi<sub>x</sub>/GaP epitaxial layers under investigation are in a state of compressive pseudomorphic strain, this splitting of the HH- and LH-like states leads in general to HH-like VB edge eigenstates<sup>52</sup>. We calculate that  $E_{\text{CB}}$  reduces linearly with increasing  $x$ , by 18 meV per % Bi. The VBAC interaction produces two Bi-hybridised HH-like bands, the energies  $E_{\pm}^{\text{HH}}$  of which vary strongly with  $x$ , displaying strong composition-dependent bowing. Beginning from  $E_{-}^{\text{HH}} = 0$  and  $E_{+}^{\text{HH}} = \Delta E_{\text{Bi}}$  at  $x = 0$ , we calculate that  $E_{-}^{\text{HH}}$  ( $E_{\pm}^{\text{HH}}$ ) decreases (increases) by 79 meV (103 meV) between  $x = 0$  and 1%. Similarly, the VBAC interaction produces a set of LH- and SO-like Bi-hybridised bands<sup>52</sup>, the energies of which are again strongly dependent on  $x$  and characterised by strong composition-dependent bowing. As  $E_{\pm}^{\text{LH}}$  moves downwards in energy towards  $E_{\text{SO}}$  with increasing  $x$  the coupling between the LH- and SO-like states—which is brought about by the axial component of the pseudomorphic strain<sup>52</sup>—leads to an anticrossing which is manifested in an abrupt increase in the rate at which  $E_{\text{SO}}$  decreases for  $x \gtrsim 1\%$ . We note also that this axial strain-induced anticrossing between the LH- and SO-like states leads to a change of the VB ordering at  $\Gamma$ , with  $E_{-}^{\text{LH}} > E_{-}^{\text{HH}}$  for  $x \gtrsim 1\%$ .

Since the SE measurements do not detect optically forbidden transitions, such as those between the  $\Gamma_{6c}$  CB edge states and the Bi-related localised states<sup>32,61</sup>, quantitative assessment of the evolution of  $E_g^{\Gamma}$  and  $\Delta_{\text{SO}}$  with Bi composition  $x$  depends critically on the character of the hybridised GaP<sub>1-x</sub>Bi<sub>x</sub>/GaP alloy VB edge eigenstates. As described above, since the GaP<sub>1-x</sub>Bi<sub>x</sub>/GaP epitaxial layers under investigation are in a state of compressive pseudomorphic strain, the highest energy alloy VB states are expected to be HH-like (cf. Fig. 2(a)). It is therefore sufficient to investigate the character of the HH-like alloy eigenstates  $|\psi_{\pm}^{\text{HH}}\rangle = a_{\text{HH}}^{(\pm)} |\psi_{\text{HH}}^{(0)}\rangle + a_{\text{Bi}}^{(\pm)} |\psi_{\text{Bi}}^{\text{HH}}\rangle$  of Eq.

(1)–corresponding respectively to the eigenvalues  $E_{\pm}^{\text{HH}}$ –which are formed of a linear combination of the extended HH band edge state  $|\psi_{\text{HH}}^{(0)}\rangle$  of the unperturbed GaP host matrix, and the HH-like Bi-related localised state  $|\psi_{\text{Bi}}^{\text{HH}}\rangle$ .

Since  $\Delta E_{\text{Bi}} > 0$  in  $\text{GaP}_{1-x}\text{Bi}_x$ , the higher energy  $E_{+}^{\text{HH}}$  eigenstate  $|\psi_{+}^{\text{HH}}\rangle$  of Eq. (1) is primarily Bi-derived ( $|a_{\text{HH}}^{(+)}|^2 < \frac{1}{2}$ ). Furthermore, given that the Bi-related localised states  $|\psi_{\text{Bi}}^{\text{HH}}\rangle$  do not couple optically to the  $\Gamma_{6c}$  CB edge states<sup>32</sup>, any optical transitions between  $|\psi_{+}^{\text{HH}}\rangle$  and the  $\Gamma$ -point CB edge, having energy  $E_{\text{CB}} - E_{+}^{\text{HH}}$ , result from the VBAC interaction imparting GaP HH fractional  $\Gamma$  character  $f_{\Gamma,+}^{\text{HH}} \equiv |\langle \psi_{\text{HH}}^{(0)} | \psi_{+}^{\text{HH}} \rangle|^2 = |a_{\text{HH}}^{(+)}|^2$  to  $|\psi_{+}^{\text{HH}}\rangle$ . Using Eq. (1),  $f_{\Gamma,+}^{\text{HH}}$  can be determined analytically as

$$f_{\Gamma,+}^{\text{HH}} = \frac{\beta^2 x}{\beta^2 x + (E_{+}^{\text{HH}} - \kappa x - \delta E_{\text{VB}}^{\text{hy}} + \delta E_{\text{VB}}^{\text{ax}})^2}. \quad (2)$$

As  $x$  increases the increase in the strength  $\beta\sqrt{x}$  of the VBAC interaction leads to  $|\psi_{+}^{\text{HH}}\rangle$  acquiring significant GaP HH  $\Gamma$  character which, despite being limited to values  $< \frac{1}{2}$ , is sufficient to produce appreciable optical coupling to the  $\Gamma_{6c}$  CB states. At  $x = 1\%$  we calculate  $f_{\Gamma,+}^{\text{HH}} = 0.315$ , indicating that the optical transition strength between  $|\psi_{+}^{\text{HH}}\rangle$  ( $|\psi_{-}^{\text{HH}}\rangle$ ) and  $\Gamma_{6c}$  in an ordered  $\text{GaP}_{0.99}\text{Bi}_{0.01}$  alloy should be close to one-third (two-thirds) of that between  $\Gamma_{8v}$  and  $\Gamma_{6c}$  in GaP. Thus, our analysis indicates the emergence of a hybridised alloy VB edge in the form of an impurity band: this band has (i) energy  $E_{+}^{\text{HH}}$  and has primarily Bi-related localised character, and (ii) optical coupling to the comparatively unperturbed  $\Gamma_{6c}$  CB edge states which increases with increasing Bi composition  $x$  (equivalently, decreasing Bi localised character,  $1 - f_{\Gamma,+}^{\text{HH}}$ ).

To reflect this admixture of GaP (Bloch) and Bi (localised) character we have calculated the four distinct energy gaps  $E_{\text{CB}} - E_{\pm}^{\text{HH}}$  and  $E_{\pm}^{\text{HH}} - E_{\text{SO}}$ . These energies represent the distinct separations evolving from the GaP direct band gap  $E_{\text{g}}^{\Gamma}$  and VB spin-orbit splitting energy  $\Delta_{\text{SO}}$ , as a direct result of the VBAC-induced hybridisation producing states having GaP HH character at distinct energies  $E_{\pm}^{\text{HH}}$ . The results of these calculations are shown, using shaded lines, in Fig. 2(b,c)–for  $E_{\text{g}}^{\Gamma}$  and  $\Delta_{\text{SO}}$ , respectively–where they are compared to the values of  $E_{\text{g}}^{\Gamma}$  and  $\Delta_{\text{SO}}$  extracted from the SE measurements of Fig. 1(a–c). To describe the optical activity of these transitions the lines denoting the calculated transition energies are shaded according to the GaP HH  $\Gamma$  character  $f_{\Gamma,\pm}^{\text{HH}}$  of the corresponding HH-like alloy VB edge states  $|\psi_{\pm}^{\text{HH}}\rangle$ , with solid black describing a purely GaP-like state having  $f_{\Gamma,\pm}^{\text{HH}} \equiv |a_{\text{HH}}^{(-)}|^2 = 1$ .

The SE measurements are sensitive to optically allowed inter-band transitions, and are hence capable in principle of detecting transitions between (i) the CB edge and the HH-like VBs, having transition energies  $E_{\text{CB}} - E_{\pm}^{\text{HH}}$ , and (ii) the CB edge and the SO band, having transition energy  $E_{\text{CB}} - E_{\text{SO}}$ . No clear features were distinguishable in the measured SE spectra close to the calculated transition energy  $E_{\text{CB}} - E_{-}^{\text{HH}}$ . However, we note that the calculated  $E_{\text{CB}} - E_{-}^{\text{HH}}$  transition energy for each sample is close to the  $E_{\text{CB}} - E_{\text{SO}}$  transition energy associated with the SO band. On the basis of the SE measurements it was not possible to determine whether there exist two distinct transitions in this energy range, although the presence of such transitions may explain the large spectral linewidths required to describe the measured SE spectra in this energy range (as reflected in the large estimated errors in the extracted values of  $\Delta_{\text{SO}}$ ).

The quantitative agreement between the calculated and measured data in Fig. 2(b,c) confirms that the extremely large observed reduction (increase) and bowing of  $E_{\text{g}}^{\Gamma}$  ( $\Delta_{\text{SO}}$ ) results from the emergence of an optically active band of primarily Bi-derived impurity states lying energetically within the GaP band gap. That this impurity band lies within the GaP band gap accounts quantitatively for the observed trends: the contribution of the strong, composition-dependent bowing of  $E_{+}^{\text{HH}}$  to the decrease (increase) of  $E_{\text{g}}^{\Gamma}$  ( $\Delta_{\text{SO}}$ ) is combined with the binding energy  $\Delta E_{\text{Bi}}$  of the Bi-related localised states. This behaviour is qualitatively distinct from that in  $\text{GaAs}_{1-x}\text{Bi}_x$ , where substitutional Bi atoms generate localised states which are resonant with the GaAs VB<sup>30,33,62,63</sup>, but similar to that in dilute nitride  $\text{GaN}_x\text{P}_{1-x}$ , where substitutional N atoms produce a band of primarily N-derived states lying deep within the GaP band gap<sup>37,38,53,54,64–67</sup>.

From the SE measurements we extract  $E_{\text{g}}^{\Gamma} = 2.52$  eV at  $x = 1.3\%$  ( $f_{\Gamma,+}^{\text{HH}} = 0.338$ ), an extremely large reduction of 240 meV compared to the measured GaP  $\Gamma$ -point band gap  $E_{\text{g}}^{\Gamma}(0) = 2.78$  eV. This is in good agreement with the calculated reduction of 284 meV in  $E_{\text{g}}^{\Gamma}$  between  $x = 0$  and 1.3% in a pseudomorphically strained, ordered  $\text{GaP}_{1-x}\text{Bi}_x/\text{GaP}$  alloy. Given the calculated reduction of 38 meV in  $E_{\text{CB}}$  between  $x = 0$  and 1.3%, we conclude that the majority (87%) of the reduction in  $E_{\text{g}}^{\Gamma}$  is associated with the emergence of the  $E_{+}^{\text{HH}}$  impurity band. Similarly, we measure an extremely large (>fourfold) increase of  $\Delta_{\text{SO}}$ , from 80 meV in GaP to approximately 360 meV at  $x = 1.3\%$ . This is again in excellent agreement with the calculated value  $\Delta_{\text{SO}} = 355$  meV, with the majority (92%) of the increase in  $\Delta_{\text{SO}}$  associated with the emergence of the  $E_{+}^{\text{HH}}$  band.

Increasing  $x$  from 1.3 to 3.7% we note that the change in  $E_{\text{g}}^{\Gamma}$  and  $\Delta_{\text{SO}}$  per % Bi is significantly reduced. The measured (calculated) value  $E_{\text{g}}^{\Gamma} = 2.28$  eV (2.289 eV) at  $x = 3.7\%$  ( $f_{\Gamma,+}^{\text{HH}} = 0.421$ ) represents a further reduction of 240 meV (207 meV) from that at  $x = 1.3\%$ , while the measured (calculated) value  $\Delta_{\text{SO}} = 1$  eV (0.590 eV) at  $x = 3.7\%$  Bi represents a further increase of 250 meV (235 meV) from that at  $x = 1.3\%$ . For  $E_{\text{g}}^{\Gamma}$  this change is only 73% of that between  $x = 0$  and 1.3%, despite occurring over a 2.4% increase in  $x$ . The measured and calculated changes of  $\Delta_{\text{SO}}$  between  $x = 1.3$  and 3.7% are approximately equal to those between  $x = 0$  and 1.3%, again representing a significantly reduced change per % Bi. These trends highlight the strong dependence of the bowing of  $E_{\text{g}}^{\Gamma}$  and  $\Delta_{\text{SO}}$  on  $x$ . Our measured and calculated variation of  $E_{\text{g}}^{\Gamma}$  and  $\Delta_{\text{SO}}$  with  $x$  differs from that predicted using first principles electronic structure calculations<sup>48</sup>, but is close to that calculated via a VBAC model using parameter estimates based on available data for related alloys<sup>49</sup>.

We now turn our attention to two key qualitative features of the GaP<sub>1-x</sub>Bi<sub>x</sub> electronic structure. Firstly, GaP has an indirect band gap due to the X<sub>6c</sub> CB states lying ≈0.5 eV below Γ<sub>6c</sub>, while in semimetallic GaBi the X<sub>6c</sub> states lie ≈2 eV above Γ<sub>6c</sub><sup>30,55</sup>. Applying the VC approximation in conjunction with the TB model we estimate that the X<sub>6c</sub> states shift downwards in energy by ≈12 meV per % Bi in free-standing GaP<sub>1-x</sub>Bi<sub>x</sub>. This is less than the 44 meV per % Bi reduction of the Γ<sub>6c</sub> state energy described by the VC parameter α (cf. Table 1), suggesting that Bi incorporation may bring about a direct band gap for sufficiently high x. Based on the calculated evolution of the GaP<sub>1-x</sub>Bi<sub>x</sub> CB structure with x (cf. Supplementary Material) we estimate that a direct band gap exists for high Bi compositions x ≳ 30%, suggesting that a direct band gap cannot be achieved at Bi compositions which are compatible with epitaxial growth. This highlights an important qualitative difference between the GaP<sub>1-x</sub>Bi<sub>x</sub> and GaN<sub>x</sub>P<sub>1-x</sub> band structures, since substitutional N in GaP generates localised states lying below the host matrix X<sub>6c</sub> states in energy, bringing about a quasi-direct band gap even at ultra-dilute N compositions<sup>37,53,54,64–66</sup>.

Secondly, our analysis in ref.<sup>30</sup> demonstrated that the VBAC description of the GaP<sub>1-x</sub>Bi<sub>x</sub> VB structure breaks down with increasing x in the presence of short-range alloy disorder. Bi clustering creates a distribution of Bi-related localised states with which the GaP VB edge states strongly hybridise. This leads to a distribution of GaP VB edge Γ character over a multiplicity of impurity levels, suggesting that there is no single band possessing sufficient Bloch character to allow for appreciable absorption or emission of light<sup>30</sup>. While our results above demonstrate that the VBAC model provides a useful approach to analyse the main features of the band structure, the details of the electronic structure are in practice determined primarily by the impact of short-range alloy disorder. The difficulty in obtaining photoluminescence from the samples studied here, despite their high crystalline quality<sup>43</sup>, supports this interpretation: the GaP<sub>1-x</sub>Bi<sub>x</sub> optical properties are intrinsically limited not solely by growth-related defects commonly associated with Bi incorporation, but by a combination of the indirect band gap and breakdown in VB edge Bloch character.

Despite having lattice constants commensurate with growth on Si, our analysis suggests that refinement of the epitaxial growth of GaP<sub>1-x</sub>Bi<sub>x</sub> alloys is unlikely to lead to efficient light emitters: the optical properties are expected to be intrinsically limited by the nature of the material band structure. However, just as quaternary GaN<sub>x</sub>As<sub>y</sub>P<sub>1-x-y</sub> alloys have found applications in III-V semiconductor lasers monolithically integrated on Si<sup>68</sup>, it is possible that similar progress could be made using As-rich quaternary GaP<sub>1-x-y</sub>As<sub>y</sub>Bi<sub>x</sub> alloys for, e.g., applications in multi-junction solar cells, due to the fact that these alloys can be grown lattice-matched to either GaAs or germanium (Ge) while having band gaps close to 1 eV<sup>69–71</sup>.

## Conclusion

In conclusion, we have presented a combined experimental and theoretical investigation of the GaP<sub>1-x</sub>Bi<sub>x</sub> band structure. Measurements performed on GaP<sub>1-x</sub>Bi<sub>x</sub>/GaP epitaxial layers reveal giant bowing of E<sub>g</sub><sup>Γ</sup> and Δ<sub>SO</sub>, whereby E<sub>g</sub><sup>Γ</sup> (Δ<sub>SO</sub>) decreases (increases) by approximately 200 meV (240 meV) between x = 0 and 1%. These changes are characterised by strong, composition-dependent bowing. Electronic structure calculations confirm that substitutional Bi in GaP generates localised impurity states lying energetically within the GaP band gap, and that the main features of the GaP<sub>1-x</sub>Bi<sub>x</sub> band structure can be understood in terms of a VBAC interaction between the extended states of the GaP VB edge, and highly localised Bi-related impurity states. A VBAC model was derived and parametrised directly from atomistic supercell calculations, allowing quantitative prediction of the evolution of the main features of the band structure with x. Our analysis suggests that the highest energy VB in GaP<sub>1-x</sub>Bi<sub>x</sub> is a hybridised impurity band: admixture of the GaP VB edge Γ character into this primarily Bi-derived band allows optical coupling to the comparatively unperturbed CB states. Aspects of the GaP<sub>1-x</sub>Bi<sub>x</sub> band structure are broadly comparable to GaN<sub>x</sub>P<sub>1-x</sub>, but key qualitative differences highlight the distinction between Bi and N as isovalent impurities in conventional III-V semiconductors.

## References

- Li, H. & Wang, S. M. (eds) *Bismuth-Containing Compounds* (Springer, New York, 2013).
- Sweeney, S. J., Batool, Z., Hild, K., Jin, S. R. & Hosea, T. J. C. The potential role of bismide alloys in future photonic devices. *In proceedings of the 13th International Conference on Transparent Optical Networks* **1** (2011).
- Broderick, C. A., Usman, M., Sweeney, S. J. & O'Reilly, E. P. Band engineering in dilute nitride and bismide semiconductor lasers. *Semicond. Sci. Technol.* **27**, 094011 (2012).
- Marko, I. P. *et al.* Temperature and Bi-concentration dependence of the bandgap and spin-orbit splitting in InGaBiAs/InP semiconductors for mid-infrared applications. *Appl. Phys. Lett.* **101**, 221108 (2012).
- Sweeney, S. J. & Jin, S. R. Bismide-nitride alloys: Promising for efficient light emitting devices in the near- and mid-infrared. *J. Appl. Phys.* **113**, 043110 (2013).
- Ludewig, P. *et al.* Electrical injection Ga(AsBi)/(AlGa)As single quantum well laser. *Appl. Phys. Lett.* **102**, 242115 (2013).
- Marko, I. P. *et al.* Physical properties and optimization of GaBiAs/(Al)GaAs based near-infrared laser diodes grown by MOVPE with up to 4.4% Bi. *J. Phys. D: Appl. Phys.* **47**, 345103 (2014).
- Fuyuki, T., Yoshida, K., Yoshioka, R. & Yoshimoto, M. Electrically pumped room-temperature operation of GaAs<sub>1-x</sub>Bi<sub>x</sub> laser diodes with low-temperature dependence of oscillation wavelength. *Appl. Phys. Express* **7**, 082101 (2014).
- Butkutė, R. *et al.* Multi-quantum well Ga(AsBi)/GaAs laser diodes with more than 6% of bismuth. *Electron. Lett.* **50**, 1155 (2014).
- Marko, I. P. *et al.* Optical gain in GaAsBi/GaAs quantum well diode lasers. *Sci. Rep.* **6**, 28863 (2016).
- Kim, H., Guan, Y., Forghani, K., Keuch, T. F. & Mawst, L. J. Laser diodes employing GaAs<sub>1-x</sub>Bi<sub>x</sub>/GaAs<sub>1-y</sub>P<sub>y</sub> quantum well active regions. *Semicond. Sci. Technol.* **32**, 075007 (2017).
- Wu, X. *et al.* 1.142 μm GaAsBi/GaAs quantum well lasers grown by molecular beam epitaxy. *ACS Photonics* **4**, 1322 (2017).
- Marko, I. P. & Sweeney, S. J. Progress towards III-V bismide alloys for near- and mid-infrared laser diodes. *IEEE J. Sel. Topics Quantum Electron.* **23**, 1501512 (2017).
- Jin, S. R. & Sweeney, S. J. InGaAsBi alloys on InP for efficient near- and mid-infrared light emitting devices. *J. Appl. Phys.* **114**, 213103 (2013).
- Broderick, C. A., Xiong, W. & Rorison, J. M. Theory of InGaBiAs dilute bismide alloys for highly efficient InP-based mid-infrared semiconductor lasers. *In proceedings of the 16th International Conference on Numerical Simulation of Optoelectronic Devices* **47** (2016).

16. Broderick, C. A., Xiong, W., Sweeney, S. J., O'Reilly, E. P. & Rorison, J. M. Theory and design of InGaAsBi mid-infrared semiconductor lasers: type-I quantum wells for emission beyond 3  $\mu\text{m}$  on InP substrates. *Semicond. Sci. Technol.* **33**(9), 094007 (2018).
17. Thomas, T. *et al.* Requirements for a GaAsBi 1 eV sub-cell in a GaAs-based multi-junction solar cell. *Semicond. Sci. Technol.* **30**, 094010 (2015).
18. Richards, R. D. *et al.* GaAsBi: an alternative to InGaAs based multiple quantum well photovoltaics. In *proceedings of the 43rd IEEE Photovoltaics Specialists Conference* **1135** (2016).
19. Mazzucato, S. *et al.* Electron spin dynamics and g-factor in GaAsBi. *Appl. Phys. Lett.* **102**, 252107 (2013).
20. Pursley, B., Luengo-Kovac, M., Vardar, G., Goldman, R. S. & Sih, V. Spin lifetime measurements in gaasbi thin films. *Appl. Phys. Lett.* **102**, 022420 (2013).
21. Simmons, R. A., Jin, S. R., Sweeney, S. J. & Clowes, S. K. Enhancement of the Rashba interaction in GaAs/AlGaAs quantum wells due to the incorporation of bismuth. *Appl. Phys. Lett.* **107**, 142401 (2015).
22. Lee, J. J., Kim, J. D. & Razeghi, M. Growth and characterization of InSbBi for long wavelength infrared photodetectors. *Appl. Phys. Lett.* **70**, 3266 (1997).
23. Hunter, C. J. *et al.* Absorption characteristics of GaAs<sub>1-x</sub>Bi<sub>x</sub>/GaAs diodes in the near-infrared. *IEEE Photon. Tech. Lett.* **24**, 2191 (2012).
24. Sandall, I. C. *et al.* Demonstration of InAsBi photoresponse beyond 3.5 mm. *Appl. Phys. Lett.* **104**, 171109 (2014).
25. Gu, Y. *et al.* Nearly lattice-matched short-wave infrared InGaAsBi photodetectors on InP. *Appl. Phys. Lett.* **108**, 032102 (2016).
26. Dongmo, P. *et al.* Enhanced room temperature electronic and thermoelectric properties of the dilute bismuthide InGaBiAs. *J. Appl. Phys.* **112**, 093710 (2012).
27. Francoeur, S. *et al.* Band gap of GaAs<sub>1-x</sub>Bi<sub>x</sub>, 0 < x < 3.6%. *Appl. Phys. Lett.* **82**, 3874 (2003).
28. Yoshida, J., Kita, T., Wada, O. & Oe, K. Temperature dependence of GaAs<sub>1-x</sub>Bi<sub>x</sub> band gap studied by photoreflectance spectroscopy. *Jpn. J. Appl. Phys.* **42**, 371 (2003).
29. Alberi, K. *et al.* Valence-band anticrossing in mismatched iii-v semiconductor alloys. *Phys. Rev. B* **75**, 045203 (2007).
30. Usman, M., Broderick, C. A., Lindsay, A. & O'Reilly, E. P. Tight-binding analysis of the electronic structure of dilute bismide alloys of GaP and GaAs. *Phys. Rev. B* **84**, 245202 (2011).
31. Batool, Z. *et al.* The electronic band structure of GaBiAs/GaAs layers: Influence of strain and band anti-crossing. *J. Appl. Phys.* **111**, 113108 (2012).
32. Broderick, C. A., Usman, M. & O'Reilly, E. P. Derivation of 12 and 14-band k · p hamiltonians for dilute bismide and bismide-nitride alloys. *Semicond. Sci. Technol.* **28**, 125025 (2013).
33. Zhang, Y., Mascarenhas, A. & Wang, L.-W. Similar and dissimilar aspects of III-V semiconductors containing Bi versus N. *Phys. Rev. B* **71**, 155201 (2005).
34. Deng, H.-X. *et al.* Band crossing in isovalent semiconductor alloys with large size mismatch: First-principles calculations of the electronic structure of bi and n incorporated gaas. *Phys. Rev. B* **82**, 193204 (2010).
35. Shan, W. *et al.* Band anticrossing in GaInNAs alloys. *Phys. Rev. Lett.* **82**, 1221 (1999).
36. Kondow, M. *et al.* GaInNAs: a novel material for long-wavelength semiconductor lasers. *IEEE J. Sel. Topics Quantum Electron.* **3**, 719 (1997).
37. Kent, P. R. C. & Zunger, A. Theory of electronic structure evolution in GaAsN and GaPN alloys. *Phys. Rev. B* **64**, 115208 (2001).
38. O'Reilly, E. P., Lindsay, A., Klar, P. J., Polimeni, A. & Capizzi, M. Trends in the electronic structure of dilute nitride alloys. *Semicond. Sci. Technol.* **24**, 033001 (2009).
39. Carrier, P. & Wei, S.-H. Calculated spin-orbit splitting of all diamondlike and zinc-blende semiconductors: Effects of  $p_{1/2}$  local orbitals and chemical trends. *Phys. Rev. B* **70**, 035212 (2004).
40. Fluegel, B. *et al.* Giant spin-orbit bowing in GaAs<sub>1-x</sub>Bi<sub>x</sub>. *Phys. Rev. Lett.* **97**, 067205 (2006).
41. Christian, T. M., Beaton, D. A., Alberi, K., Fluegel, B. & Mascarenhas, A. Mysterious absence of pair luminescence in gallium phosphide bismide. *Appl. Phys. Express* **8**, 061202 (2015).
42. Christian, T. M., Fluegel, B., Beaton, D. A., Alberi, K. & Mascarenhas, A. Bismuth-induced raman modes in GaP<sub>1-x</sub>Bi<sub>x</sub>. *Jpn. J. Appl. Phys.* **55**, 108002 (2016).
43. Nattermann, L. *et al.* MOVPE growth of Ga(PBi) on GaP and GaP on Si with Bi fractions up to 8%. *J. Cryst. Growth* **463**, 151 (2017).
44. Trumbore, F. A., Gershenson, M. & Thomas, D. G. Luminescence due to the isoelectronic substitution of bismuth for phosphorus in gallium phosphide. *Appl. Phys. Lett.* **9**, 4 (1966).
45. Fujiwara, H. *Spectroscopic Ellipsometry: Principles and Applications* (John Wiley & Sons, Chichester, 2007).
46. Bushell, Z. L. *et al.* Optical functions and critical points of dilute bismide alloys studied by spectroscopic ellipsometry. *J. Appl. Phys.* **123**, 045701 (2018).
47. Jain, S. C., McGregor, J. M. & Roulston, D. J. Band-gap narrowing in novel III-V semiconductors. *J. Appl. Phys.* **68**, 3747 (1990).
48. Polak, M. P., Scharoch, P. & Kudrawiec, R. First-principles calculations of bismuth induced changes in the band structure of dilute Ga-V-Bi and In-V-Bi alloys: chemical trends versus experimental data. *Semicond. Sci. Technol.* **30**, 094001 (2015).
49. Samadjar, D. P., Das, T. D. & Dhar, S. Valence band anticrossing model for GaSb<sub>1-x</sub>Bi<sub>x</sub> and GaP<sub>1-x</sub>Bi<sub>x</sub> using k · p method. *Mater. Sci. Semicond. Process.* **40**, 539 (2015).
50. Usman, M. *et al.* Impact of alloy disorder on the band structure of compressively strained GaBi<sub>x</sub>As<sub>1-x</sub>. *Phys. Rev. B* **87**, 115104 (2013).
51. Luo, G. *et al.* Understanding and reducing deleterious defects in the metastable alloy GaAsBi. *NPG Asia. Materials* **9**, 345 (2017).
52. Broderick, C. A. *et al.* Determination of type-I band offsets in GaBi<sub>x</sub>As<sub>1-x</sub> quantum wells using polarisation-resolved photovoltage spectroscopy and 12-band k · p calculations. *Semicond. Sci. Technol.* **30**, 094009 (2015).
53. Shan, W. *et al.* Nature of the fundamental band gap in GaN<sub>x</sub>P<sub>1-x</sub> alloys. *Appl. Phys. Lett.* **76**, 3251 (2000).
54. Harris, C., Lindsay, A. & O'Reilly, E. P. Evolution of N defect states and optical transitions in ordered and disordered GaP<sub>1-x</sub>N<sub>x</sub> alloys. *J. Phys.: Condens. Matter* **20**, 295211 (2008).
55. Janotti, A., Wei, S.-H. & Zhang, S. B. Theoretical study of the effects of isovalent coalloying of Bi and N in GaAs. *Phys. Rev. B* **65**, 115203 (2002).
56. Vurgaftman, I., Meyer, J. R. & Ram-Mohan, L. R. Band parameters for III-V compound semiconductors and their alloys. *J. Appl. Phys.* **89**, 5815 (2001).
57. Wang, C. *et al.* Molecular beam epitaxy growth of AlAs<sub>1-x</sub>Bi<sub>x</sub>. *Semicond. Sci. Technol.* in press (2018).
58. Alaya, R., Mbarkı, M., Rebey, A. & Postnikov, A. V. Ab initio predictions of structure preferences and band gap character in ordered AlAs<sub>1-x</sub>Bi<sub>x</sub> alloys. *Curr. Appl. Phys.* **16**, 288 (2016).
59. Van de Walle, C. G. Band lineups and deformation potentials in the model-solid theory. *Phys. Rev. B* **39**, 1871 (1989).
60. Krijn, M. P. C. M. Heterojunction band offsets and effective masses in iii-v quaternary alloys. *Semicond. Sci. Technol.* **6**, 27 (1991).
61. Broderick, C. A., Harnedy, P. E. & O'Reilly, E. P. Theory of the electronic and optical properties of dilute bismide quantum well lasers. *IEEE J. Sel. Top. Quant. Electron.* **21**, 1503313 (2015).
62. Joshya, R. S., Ptak, A. J., France, R., Mascarenhas, A. & Kini, R. N. Resonant state due to Bi in the dilute bismide alloy GaAs<sub>1-x</sub>Bi<sub>x</sub>. *Phys. Rev. B* **90**, 165203 (2014).
63. Alberi, K., Beaton, D. A. & Mascarenhas, A. Direct observation of the  $e^-$  resonant state in GaAs<sub>1-x</sub>Bi<sub>x</sub>. *Phys. Rev. B* **92**, 241201 (2015).



64. Zhang, Y., Fluegel, B., Mascarenhas, A., Xin, H. P. & Tu, C. W. Optical transitions in the isoelectronically doped semiconductor GaP:N: an evolution from isolated centers, pairs, and clusters to an impurity band. *Phys. Rev. B* **62**, 4493 (2000).
65. Wu, J. *et al.* Band anticrossing in GaP<sub>1-x</sub>N<sub>x</sub> alloys. *Phys. Rev. B* **65**, 241303(R) (2002).
66. Fluegel, B., Zhang, Y., Geisz, J. F. & Mascarenhas, A. Confirmation of the impurity band model for GaP<sub>1-x</sub>N<sub>x</sub>. *Phys. Rev. B* **72**, 073203 (2005).
67. Güngerich, M. *et al.* Experimental and theoretical investigation of the conduction band edge of GaN<sub>x</sub>P<sub>1-x</sub>. *Phys. Rev. B* **74**, 241202 (2006).
68. Liebich, S. *et al.* Laser operation of Ga(NAsP) lattice-matched to (001) silicon substrate. *Appl. Phys. Lett.* **99**, 071109 (2011).
69. Forghani, K. *et al.* GaAs<sub>1-y-z</sub>P<sub>y</sub>Bi<sub>z</sub>, an alternative reduced band gap alloy system lattice-matched to GaAs. *Appl. Phys. Lett.* **105**, 111101 (2014).
70. Nattermann, L. *et al.* MOVPE growth and characterization of quaternary Ga(PAsBi)/GaAs alloys for optoelectronic applications. *Applied Materials Today* **5**, 209 (2016).
71. Nattermann, L., Ludewig, P., Sterzer, E. & Volz, K. Exploiting strain to enhance the Bi incorporation in GaAs-based III/V semiconductors using MOVPE. *J. Cryst. Growth* **470**, 15 (2017).

## Acknowledgements

This work was supported by the Engineering and Physical Sciences Research Council, U.K. (EPSRC; Project Nos EP/H005587/1, EP/N021037/1, and EP/K029665/1), by Science Foundation Ireland (SFI; Project No. 15/IA/3082), and by the German Science Foundation (DFG; Project No. GRK 1782). Z.L.B. acknowledges support from the University of Surrey Marion Redfearn and Advanced Technology Institute Scholarships. The data associated with this work are available from the University of Surrey publications repository at <https://doi.org/10.5281/zenodo.2635732>.

## Author Contributions

Z.L.B. led the experimental measurements and analysis of the experimental data, and contributed to the writing of the manuscript. C.A.B. performed the theoretical calculations and analysis of the theoretical data, and led the writing of the manuscript. L.N. performed the growth and characterisation of the material samples investigated. R.J. and J.L.K. assisted with the experimental measurements and analysis of the experimental data. K.V. and J.M.R. secured funding to support this work, and oversaw and contributed respectively to the sample growth and characterisation, and theoretical analysis. S.J.S. secured funding to support this work, oversaw and contributed to the experimental analysis, and contributed to the writing of the manuscript. All authors reviewed the manuscript prior to submission.

## Additional Information

**Supplementary information** accompanies this paper at <https://doi.org/10.1038/s41598-019-43142-5>.

**Competing Interests:** The authors declare no competing interests.

**Publisher's note:** Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



**Open Access** This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2019