CHARACTERIZATION OF INDIUM BASED LOW TEMPERATURE SOLDER ALLOY AND THE EFFECT ON SURFACE FINISH

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by

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Declaration

I declare that this thesis is the result of my own research, that it does not incorperote

without acknowledgement any material submitted for a degree or diploma in any

university and that it does not certain any materials previously published, written or

produced by another person except where due reference is made in the text.

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LIST OF ABBREVIATIONS

DSC Differential scanning calorimetry

EDX Energy Dipersive X-ray

IBS In-Bi-Sn IBZ In-Bi-Zn

IMC Intermetallic compound

SAC Sn-Ag-Cu

SEM Scanning Electron Microscope

XRD X-ray Diffraction X-ray Diffraction

XRF X-ray fluorescence

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LIST OF PUBLICATIONS & SEMINARS

Wettability and strength of In-Bi-Sn lead-free solder alloy on copper substrate, <u>Ervina Efzan Mhd Noor</u>, Ahmad Badri Ismail, Nurulakmal Mohd Sharif, Cheong Kuan Yuew, Tadashi Ariga, and Zuhailawati Hussain, Journal of Alloys and Compounds, Volume 507, Issue 1,2010 (I.F.2.20)

Low Temperature of Bi-In-Zn Lead-free Solder Alloy, <u>Ervina Efzan Mhd Noor</u>, Zuhailawati Hussain, Ahmad Badri Ismail, Nurulakmal Mohd Sharif, Tadashi Ariga, Material, Science & Technology (MS&T), pp. 124-131 (ISBN 13:978-1-61503-006-4).

Characteristic of Low Temperature of Bi-In-Sn Solder alloy, <u>Ervina Efzan Mhd Noor</u>, Ahmad Badri Ismail, Nurulakmal Mohd Sharif, Tadashi Ariga, and Zuhailawati Hussain, 33rd International Electronics Manufacturing Technology Conference 2008, Penang, IEEE Publishing art. no. 5507865

Interfacial Reaction of Bi-In-Zn/Cu and Bi-In-Zn/Ni solder joint alloy, <u>Ervina Efzan Mhd Noor</u>, Ahmad Badri Ismail, Nurulakmal Mohd Sharif, Kuan Yew Cheong, Tadashi Ariga, and Zuhailawati Hussain, 4th International Conference on Recent Advances in Materials, Mineral& Environment and 2nd Asian Symposium On Materials & Processing 2009, Bayview Hotel, Penang

Effect of temperature on the formation of Interfacial between Bi-In-Sn and Bi-In-Zn solders on Cu Substrate, <u>Ervina Efzan Mhd Noor</u>, Ahmad Badri Ismail, Nurulakmal Mohd Sharif, Cheong Kuan Yuew, Tadashi Ariga, and Zuhailawati Hussain, for 17TH EMSM SCIENTIFIC Conference 2008, Selangor

Contact Angle of Bi-In-Zn lead-free solder system on Cu and tin surface finish, Ervina Efzan Mhd Noor, , Zuhailawati Hussain, Nurulakmal Mohd Sharif, Cheong Kuan Yuew, Y.T.Chin, Tadashi Ariga, and Ahmad Badrii Ismail, International Conference of Science and Technology, UITM, 2009, Penang.

Effect of temperature on the formation of Interfacial between Bi-In-Sn solder on Cu and Ni Surface Finish, <u>Ervina Efzan Mhd Noor</u>, Ahmad Badri Ismail, Nurulakmal Mohd Sharif, Tadashi Ariga and Zuhailawati Hussain, Malaysian Metallurgical Conference 2008 (MMC 2008),UKM Selangor.

Surface Tension and Wetting Behaviour of Bi-In-Sn Alloys, <u>Ervina Efzan Mhd Noor</u>, Ahmad Badri Ismail, T.K. Soong, Y.T.Chin and Luay Bakir Hussain, NMC 2007, Johor Bahru, Malaysia.

Effect of Temperature on the Formation of the Interfacial Between In-Bi-Zn solder on Cu Substrates, <u>Ervina Efzan Mhd Noor</u> and Zuhailawati Hussain, Infineon Sysmposium 2011, Malacca

PENCIRIAN PATERI SUHU RENDAH BERASASKAN INDIUM DAN KESANNYA KE ATAS KEMASAN AKHIR

ABSTRAK

Peningkatan penggunaan peranti elektronik meningkatkan penggunaan pateri. Plumbum, bahan utama di dalam kebanyakkan pateri sangat merbahaya kepada kesihatan dan persekitaran. Oleh itu, penggantian pateri Sn-37Pb kepada pateri tanpa plumbum merupakan isu yang sangat penting di dalam industry elektronik. Oleh itu, perbandingan ciri-ciri pateri In-Bi-Sn dan In-Bi-Zn telah dilakukan. Analisis Pembezaan Pengubahsuaian Kalometri menunjukkan pateri In-Bi-Sn dan In-Bi-Zn menunjukkan takat lebur yang rendah masing-masing iaitu 61.3°C dan 72.3°C. Tegangan permukaan dan sudut sentuhan aloi pateri In-Bi-Sn dan In-Bi-Zn diukur pada substrat Cu dan pelbagai kemasan akhir pada suhu 100, 120 dan 140°C. Ukuran sessile drop menunjukkan sudut sentuhan bergantung kepada suhu aliran. Sudut sentuhan menurun perlahan dari 30.76° kepada 17.25° apabila suhu aliran meningkat dari 100 kepada 140°C bagi aloi pateri In-Bi-Sn dan In-Bi-Zn ke atas Cu substrat, pada julat 58° kepada 7° selepas pembasahan ke atas Ni/Cu substrat pada julat suhu aliran yang sama (100-140°C). Analisis Tenaga Penyerapan X-Ray menunjukkan dua lapisan sebatian antaramuka pada In-Bi-Sn Cu₆Sn₅ (berbentuk scallop) dan Cu₁₁In₉ (warna cerah) ke atas substrat Cu dan Sn/Cu. Sebatian antaramuka antara aloi pateri In-Bi-Zn boleh dilihat adalah Cu₅Zn₈ (sambungan satah) dan sedikit lapisan Cu₁₁In₉ ke atas substrat Cu dan Sn/Cu. Walaubagaimanapun, hanya terdapat satu sebatian antaramuka terbentuk pada kedua-dua pateri (In-Bi-Sn dan In-Bi-Zn) ke atas substrat Ni/Cu iaitu InNi₂. Semakin suhu aliran meningkat, kekuatan ricih pada pateri In-Bi-Sn dan In-Bi-Zn ke atas Cu, Ni/Cu dan Sn/Cu juga meningkat disebabkan sudut sentuhan yang berkurangan dan perebakan kawasan yang besar.

CHARACTERIZATION OF INDIUM BASED LOW TEMPERATURE SOLDER ALLOY AND THE EFFECT ON SURFACE FINISH

ABSTRACT

The increased use of electronic devices has increased the usage of solder connections. Lead, the prime solder hitherto used, is hazardous to human health and the environment. Thus, replacing Sn-37Pb with a lead-free solder is one of the most important issues in the electronics industry. As such, the characteristics of In-Bi-Sn and In-Bi-Zn compared with that of the Sn-Ag-Cu solder alloy were studied. In the differential scanning calometry analysis, In-Bi-Sn and In-Bi-Zn system alloys presented a low melting temperature of 61.3 °C and 72.3 °C, respectively. Surface tension and contact angle of In-Bi-Sn and In-Bi-Zn lead-free solder alloys were measured on Cu substrate and different surface finishes at 100, 120 and 140 °C reflow. Sessile drop measurements showed that the contact angle depended on the reflow temperature. The contact angle gradually decreased from 30.76° to 17.25° as reflow temperature increased from 100 to 140 °C and for In-Bi-Sn and In-Bi-Zn solder alloy on Cu substrate, ranged from 58° to 7° after wetting on Ni/Cu substrate at the same reflow temperature range (100 to 140°C). Energy-dispersive X-ray analysis found two layers of intermetallic compound in the In-Bi-Sn solder alloy: Cu₆Sn₅ and Cu₁₁In₉ (scallop shaped) and Cu₁₁In₉ (brightly coloured) with Cu and Sn/Cu substrate. The IMC between the In-Bi-Zn solder alloy could be observed: Cu₅Zn₈ (continuous planar) and Cu₁₁In₉, a minor IMC layer with Cu and Sn/Cu substrate. However, only one type of IMC was formed between both solders (In-Bi-Sn and In-Bi-Zn) and Ni/Cu substrate, which was InNi₂. As the reflow temperature increased, the shear strength of the In-Bi-Sn and In-Bi-Zn solder alloys on Cu, Ni/Cu and Sn/Cu joints improved due to reduced contact angle and larger spreading area.

CHAPTER 1

INTRODUCTION

1.1 Introduction

Miniaturization is a key issue in achieving advanced performance from electronic devices. Rapid advances in microelectronic design and technology in recent decades have made modern day electrical and electronic equipment (EEE) obsolete within a very short period after their purchase resulting in mountains of electronic waste (e-waste) to be dealt with in many countries around the world. According to a United Nations estimate, collectively the world generates 20 to 50 million tons of e-waste every year (Schwarzer et al., 2005). The growing consumer demand and discarded electrical and electronic products (often called e-waste) has been estimated to already constitute 8% of municipal waste (Tsydenova & Bengtsson, 2011).

E-waste contains significant quantities of toxic materials, considered dangerous, as certain components of some of these electronic products contain materials such as lead, which are hazardous, depending on their condition and density. Consumer electronics devices (CEDs) contain numerous toxic materials, such as mercury, lead, zinc, and cadmium, which pose a threat to public and environmental health if CEDs are improperly discarded. These include metals such as lead, mercury, cadmium; and chemicals such as polychlorinated biphenyls. Among these metals, lead is one of major concern because of the extensive use of lead solder, especially eutectic tin-lead (Sn-37Pb) or near eutectic Sn-40Pb solder in the assembly of the modern electronic circuits (Saphores et al., 2009a).

Lead (Pb) containing solders have been widely used in the electronics industries for a long time due to its good solderability, low melting temperature, and satisfactory mechanical properties. It serves to form interconnections between electronic components and printed circuit boards (PCB) both electrically and mechanically (Chen et al., 2005). However, the presence of lead raises several environmental concerns, including the fate of the lead upon disposal of the discarded electronic device. The toxicity characteristic leaching procedure (TCLP) is the United State Environmental Protection Agency's (US EPA) method used to determine the status of a solid waste as a hazardous waste due to the toxicity characteristic (TC) (Timothy et al., 2008).

The European Parliament and the Council of the European Union (EU) adopted the Directive on the "Restriction of the Use of Hazardous Substances" (RoHS) (2002/95/EC) on 23 January 2003. This Directive bans certain hazardous substances in electrical and electronic equipment. It covers a broad range of uses for toxic chemicals, with three articles specifically addressing lead, as shown in the sidebar. Additionally, the European Union (EU) Directive 2002/95/EC—Restriction of Hazardous Substances (RoHS) restricts the use of specific hazardous materials found in electrical and electronic products including lead beginning 1 July 2006. Waste Electrical and Electronic Equipment Directive (WEEE Directive) mandates the producers of electrical and electronic equipment to finance the collection, recycling, or otherwise adopt nonhazardous disposal of their products after consumers are ready to discard them (Ogunseitan, 2007).

In Japan, Japanese law mandates the take-back of various household electrical appliances to reuse and recycle materials and substances in those electrical

appliances. Japanese Electronic Industry Development Institute Association (JEIDA) and Japanese Institute of Electronic Packaging (JIEP) followed the same steps to maintain trade with EU countries and marketing edge by promoting "green product" (lead-free) to their customers.

For these reasons, environmental regulations in various countries have emerged. Following the impetus of the EU directives, Japan (Kim et al., 2005a), China (Anonymous, 2006), and South Korea (Ogunseitan, 2007) have instituted similar regulations to limit the use of lead and other RoHS toxicants in electronic and electrical equipment. Although binary, ternary, and quaternary systems in solder investigated showed improving properties, none have met all standards, which included a low melting temperature, wettability, mechanical performance, and good manufacturability (Suganuma, 2001). Therefore, research on lead-free solders remains an important issue in the electronic industry (Abtew & Selvaduray, 2000; McCormack & Jin, 1994).

Therefore, an alternative solder free from Pb needs to be developed. For example, Sn–Ag solder alloy (melting temperature, 221 °C) and Sn–Ag–Cu solder alloy (melting temperature, 217 °C) have been investigated as replacements (Buyok & Maras, 2009). However, despite the general acceptance of these solders as the leading Pb free solders, they have a number of disadvantages, one of which is a relatively high melting temperature. High melting temperature requires high reflow temperature. Thus, low melting temperature solder is more suitable for an alternative solder.

Lowering reflow temperatures reduces thermal stress on adjacent components. In addition, heat-sensitive materials such as ferroelectric polymers or liquid crystals are increasingly introduced to integrated chips. These heat sensitive materials, in general, have limited thermal stabilities and some of them require low-temperature assembly of below 125 °C (Suh et al., 2008).

The metallurgical behaviour of low reflow temperatures solder joints with different surface finishes (Ni and Sn) and the related reliability of joints on surface finishes have not been sufficiently studied yet. Therefore, the present study was carried out to investigate the reliability of In-Bi-Sn and In-Bi-Zn solders with Cu substrate and various surface finishes.

1.2 Problem Statement

Among the six prohibited substances in the RoHS directive, the prohibition of lead has had the largest impact on the electronics industry. In response to the need to adopt lead-free process technology, companies have variously switched to other substitute alloys. Most companies in the industry have switched to a tin-based lead-free solder in recent years, and added other metals to reinforce solder characteristics. Numerous types of lead-free alloys are currently available in the market (Lin et al., 2007; Liou et al., 2009; Wang et al., 2008a; Yoon et al., 2009). Of these, Sn-Ag-Cu alloys are widely used in Europe, in the US, and in Japan.

Despite of the general acceptance of Sn-Ag (221 °C) and Sn-Ag-Cu solder alloy (217 °C)(free lead), the disadvantages of these alloys, which could not be ignored are the high melting points. A higher melting point leads to a 20–30 °C increase in the peak reflow temperature required for assembly compared with that required for Pb-Sn eutectic solder. Such an increased temperature is detrimental to many microelectronic components, making assembly very difficult. The increased

temperature reduces the integrity, reliability, and functionality of printed wiring boards (PWBs) and their other components (Kim et al., 2004).

Another alternative is Sn-Zn solder alloy (Zn about 9wt%) with a melting point of about 198 °C (Garcia et al., 2010; Hu et al., 2010; Mahmudi et al., 2009). This alloy offers significant benefits in terms of cost and good mechanical properties (Mahmudi et al., 2009; Garcia et al., 2010). However, this alloy suffers from low oxidation resistance and poor wettability (Mahmudi et al., 2009).

Despite of the development of Sn-Ag, Sn-Ag-Cu and Sn-Zn solders, the properties of these solders are still not good enough for these alloys to be employed in temperature-sensitive components, optoelectronics modules, step soldering processes, and thin printed wiring boards (PWBs) (Suganuma, 2004), all of which require a low melting temperature. Low-temperature soldering is necessary when electronic devices to be soldered are prone to thermal damage.

Although lead-free solder have been extensively studied, information about the In-31.6Bi-19.6Sn (In-Bi-Sn) and In-32.7Bi-0.5Zn (In-Bi-Zn) solders alloys are not yet available in the literature. Since In-Bi-Sn and In-Bi-Zn solder alloys have low liquidus temperature, it may be used for low-temperature soldering. Furthermore, its low liquidus temperature suits for use as an organic-based transistor such as liquid crystal display (145-178 °C), in the new generation of nano silicon chips (100-300 °C), and in polymeric conductive boards (140-400 °C). Using these alloys, outer space nano satellites to be used under cryogenic conditions (minus 147-447 °C) can be manufactured at low temperatures.

Compared with Sn-based solders, In-based solders possess advantages such as lower melting points, better wettability, and longer fatigue life (Chuang et al., 2009). Step-soldering, required for high-density packaging of multichip modules (MCMs), uses a number of solders with wide range of melting points. Indium was used as the joining material because it remains electrically conductive and ductile to cryogenic temperatures and helps relieve the strain between the substrate and the package (Chang & McCluskey, 2009).

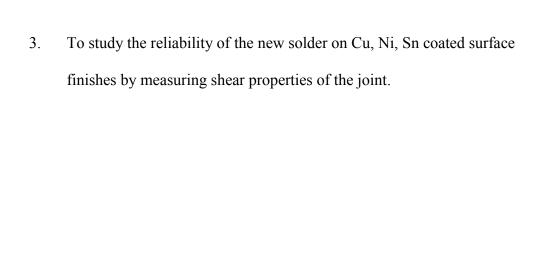
The presence of intermetallic compounds (IMCs) between solder alloys and the surface finishes ensures a good metallurgical bonding. Thus, many studies have been performed on the joint reliability and interfacial reaction between lead free solders and various surface finishes such as Ni and Sn (Gosh et al., 2000; Koo et al., 2005; Sharif et al., 2004; Yoon et al., 2009). Nevertheless, studies on the interfacial reactions and mechanical properties of In-Bi-Sn and In-Bi-Zn solder on Ni and Sn surface finish have not been available yet in the literature.

1.3 Objectives of the Study

In this work, the properties of In-Bi-Sn and In-Bi-Zn solder alloys were investigated.

The main objectives of this work include:

- 1. To develop and characterize physical properties of In-Bi-Sn and In-Bi-Zn new solders in terms of melting temperature, density, and microstructure.
- 2. To study the wetting characteristic of the new solders on Cu, Ni, Sn coated surface finishes at different reflow temperatures.



CHAPTER 2

LITERATURE REVIEW

2.1 Introduction

Joining processes, such as attaching two or more substrates or base materials, be they metals, ceramics, or plastics, can be considered as one of two generalized methodologies: filler material joining or fusion joining (Vianco, 1999a).

2.2 Soldering

The American Welding Society defines soldering as a joining process where a coalescence between metal parts is produced by heating to temperature below 425 °C using filler metals (solders) having melting temperatures below those of the base metal (Frear et al., 1994). Manko (2001) defined soldering as a metallurgical joining method using filler metal (the solder) with a melting point below 315 °C. Metallurgical bonds are connections between metals only (Manko, 2001). In general, they are those bonds in which metallic continuity from one metal to another is established. The roles of solder, the reaction itself is exothermic which means that it requires no energy input to proceed, once it has started. Soldering heat is needed to melt the solder, because solid solder can neither react with the substrate (or only very slow), nor flow into a joint (Strauss, 1998). Various types of solder have been used in the electronic industry lately. Examples are such as lead solder, hard solder, flux core solder, and lead-free solder. Previously, lead solder is the most widely used in the electronic industry, however, due to the toxicity of lead (Pb) present, alternative solders need to be considered (Yoon et al., 2008).

There are two levels of packaging where solders are used in electronics assembly. One of the levels is the bonding of the die (Chip) to a substrate, which is known as Level 1 packaging (Abtew & Selvaduray, 2000). In Level 2 packaging, the components are mounted in printed wiring board (PWB). A printed circuit board is used to connect electrically and mechanically to hold the electronic components using conductive pathways (Anonymous, 2009a). Figure 2.1 shows an actual PWB (Anonymous, 2010a). The two ways to connect the electronic equipment to the PWBs are pin through hole and surface mount technology (SMT). A graphical representation of the processes of pin through hole and SMT are shown in Figure 2.2 (Anonymous, 2009a).



Figure 2.1. Printed wiring board (PWB)

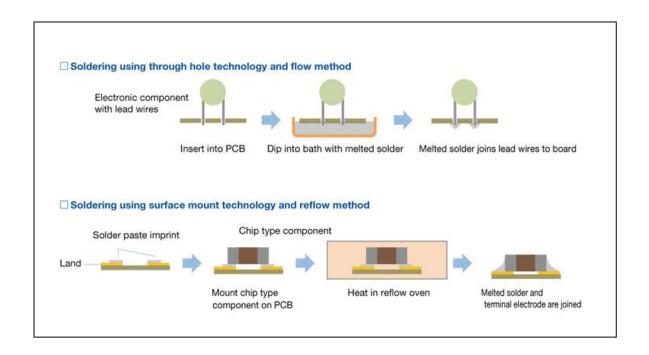


Figure 2.2. Process of pin through hole and surface mount technology (SMT) (Anonymous, 2010a)

2.3 Lead

Lead is a heavy metal found naturally in the earth that has been used in many ways for hundreds of years. Lead's atomic number may not be 1 (it is 82) but it ranks near the top when comes to industrial uses (Casas & Sordo, 2006). Lead (and its compounds) have been cited by the Environmental Protection Agency (EPA) as one of the top 17 chemicals posing the greatest threat to human life and the environment (Abtew & Selvaduray, 2000). Although lead is very soft in its pure state and can be scratched even with the fingernail, small additions of impurities such as antimony, arsenic, copper, zinc, and so forth, would make the lead much harder by solution hardening. These small traces of elements are enough to increase the hardness and strength appreciably (Manko, 2001).

Health concerns are relatively recent, and they came with the realization that very small amounts of lead have a profound effect on the human body. Lead binds strongly to proteins in the body and inhibits normal processing and functions of the human body (Abtew & Selvaduray, 2000). The common types of lead poisoning may be classified according to (a) alimentary, (b) neuromotor, and (c) encephalic. Lead poisoning commonly occurs following prolonged exposure to lead or lead compounds. The damage is often induced slowly (Braga et al., 2007; Lee, 2001).

In industry, the authorities have allowed lead containing dross, clippings, spent solder, and the like, to be recycled and not classified as "hazardous waste." However, still one lead-containing material has not been addressed by the cycle, namely discarded products such as electronics equipment which end up in the landfill (Manko, 2001).

2.4 Sn-Pb and Regulatory Trends for Leaded Solder

The term solder has become synonymous with the tin-lead system and widely used in the industry. The lead solder alloys most commonly used for electronics assembly contain tin (Sn) as a main composition with a standard eutectic composition being Sn63Pb37. These alloys consist of 60% tin (Sn) and 37% lead (Pb). The eutectic ratio of 63/37 corresponds closely to a formation of α and β eutectic compound.

Sn-Pb solders for metal interconnections have appreciated since at least Roman times, dating back to 2000 years ago. These solder materials serve as electrical, thermal, and mechanical connectors in Surface Mount Technology (SMT), such as ball grid arrays (BGAs), chip scale packages (CSPs), and flip chips (FCs)

(Lun, 2001). The development of compositions alloy developed provides many benefits, such as ease of handling, low melting temperatures, good workability, ductility, relatively inexpensive to produce, and using superior wettability and spreading characteristics on Cu and its alloys (Chen et al., 2006; Humpston & Jacobson, 2004; Kanchanomai et al., 2002; Suganuma, 2001).

In the electronics industry, the lead generated by the disposal of electronic assemblies is considered as hazardous to the environment (Abtew & Selvaduray, 2000). The elimination of lead from electronic devices as mandated by Restriction of Hazardous Substances (RoHS) is changing the playing field for all those involved in the printed circuit industry (Satyanarayan et al., 2011). The attempt to ban lead from electronic solder was initiated in the U.S. Congress. The European Union (EU) and RoHS set stringent limits on the amount of hazardous materials that could be present in electronics products. This has speeded up the market for a lead-free soldering system (Xin, 2007). Major Japanese companies also set voluntary timetable for the phasing out of lead (Wu et al., 2004).

On the PCB level, OEMs have the brunt of dealing with the change that designers are always pushing the limits of the industry's capabilities to meet the requirements of their newer designs (Milad & Orduz, 2007).

2.5 Lead-Free Solder and Its Requirement

Lead-free solder in the electronics industry is a segment of a global trend towards a lead-free environment, although the lead used in electronics soldering accounts for less than 1% of the total industrial use of this metal (Xin, 2007). To become a good solder material, lead-free solder needs to be reliable over long-term

use (Wu et al., 2004). The current processing equipment and conditions for electronics assembly are optimized for Sn–Pb solders. Any new conditions for lead-free alloys must ensure both productivity and reliability at least equivalent to the present level of Sn–Pb solders (Suganuma, 2001).

There are strict performance requirements for solder alloys used in microelectronics. According to Kang (2001), there are several requirements for new lead-free solder alloy to replace the lead solder Sn-Pb (Kang, 2001). In general, the solder alloy must meet the expected levels of electrical and mechanical performance, whist at the same time it must also have the desired melting temperature (Abtew & Selvaduray, 2000). One of the most sensitive eutectic solder systems parameter for the quality of soldered joints is soldering temperature (Suganuma, 2001). It must adequately wet common printed circuit board (PCB) lands, form inspectable solder joints, allow high volume soldering and rework of defective joints, provide reliable solder joints under service conditions, and it must not significantly increase assembly cost (Abtew & Selvaduray, 2000).

When trying to identify an alternative to the current Sn-Pb solders that are widely used, it is important to ensure that the properties of the replacement solder are comparable to or superior to Sn-Pb solders. The first step in finding suitable alloy candidates is to search for some nontoxic, low-melting temperature alloys that can replace this amount of Pb (Suganuma, 2001). The search for a global Pb-free replacement for Sn-Pb eutectic alloy has been an evolving process as the threat of a regional lead ban became a reality in July 2006.

The properties of solders that are important from a manufacturing and a longterm reliability standpoint are listed as in the following:

(a) Non-toxic

The main driving force for elimination of Pb is its toxic nature (Mayappan, 2007). The concern here is not just the toxicity of Pb; but the fact that when disposed of in landfills, Pb can eventually leach out and can contaminate underground water sources and subsequently finds its way into the human ecosystem in an ingestible form. In choosing alternatives for Pb, it must be ascertained ahead of time that the alternatives will not pose a similar hazard (Abtew & Selvaduray, 2000; Jasbir, 2007; Suganuma, 2001; Wu et al., 2004).

(b) Melting Temperature

The temperature margin beyond the melting temperature of solder is about 50 °C for reflow soldering (Abtew & Selvaduray, 2000). At present, some of the electronic components (such as capacitors and connectors) cannot withstand an increase in reflow temperature need to modify or develop processing conditions to incorporate heat-resistant components (Suganuma, 2001). The solder must also have a liquidus temperature below the melting point (solder temperature) of the parent materials and any surface metallization.

(c) Narrow pasty range

The pasty range is the temperature range between solidus and liquidus, where alloy is part solid and part liquid. When the pasty range is narrow, the solder needs shorter time to solidify. The acceptable level for pasty range is less than 30 °C by National Center for Manufacturing Science (NCMS). The Centre completed its lead-free solder project in 1997 (Jasbir, 2007).

(d) Acceptable wettability

Wettability behaviour is important to the interconnection of electronic packages like soldering (Wu et al., 2004). In reflow step, the molten solder contacts the substrate, wet the substrate, causing an interfacial reaction (Chen, 2010). This is necessary to allow capillary action to take place to fill a joint gap or to fill a plated-through hole (PTH) in a printed wiring board during soldering (Evans, 2007).

(e) Process economy

Cost is another important factor in the adoption of solders for practical electronic applications (Satyanarayan & Prabu, 2011). The microelectronics industry, wherever, is extremely cost conscious, and the industry is not willing to overlook increased costs; thus, higher-priced proprietary alloys will not be accepted. The history of the industry has been to produce continuously at higher performance with lower costs. Cost competitiveness in the electronics industry is maintained by reducing the cost of individual components to a minimum to maximize the overall cost reduction. In general, taking cost of raw metals into consideration, most lead-free solders cost about two to three times more than Sn–Pb solders (Suganuma, 2001). From a purely cost standpoint, an unpatented solder alloy, with multiple suppliers, with a stable price structure, and without geopolitical concerns, would be most desirable.

(f) Material manufacturability and acceptable processing temperature

Solder alloys must be compatible with standard surface finishes used today or have a universally acceptable finish of their own. This is a critical point used to having terminations finished with solderable coatings such as gold, platinum, silver, copper, and the solder itself. The solder alloy system must be stable and form as few intermetallic compounds as possible. The intermetallic compound (IMC) should not pose an embrittlement problem in the solder and the wetted interface (Manko, 2001).

The temperature range for processing temperature has to be good with hardware built, components, and substrates used. The lower temperatures would be acceptable for low-end commercial applications.

(g) Form reliable joints

One of the major concerns of area array packages, such as ball-grid-array (BGA) and flip chip, is the reliability of the solder joint (Hwang et al., 2004). Solder properties are important to solder-joint integrity in surface mount technology (Zeng & Tu, 2002). Recent studies have indicated that the interfacial intermetallic compounds (IMCs) between solder alloys and substrates have a significant effect on the reliability of solder joints (Yoon & Jung, 2006).

2.6 Common Solder Electronic Assemblies

There are three important considerations in the selection of solder alloys. They are the composition, the solidus temperature, and the liquidus temperature. The composition of the alloy has a direct effect on the joint strength and the wettability of the solder (Cheng et al., 2009; Chria & Ozvold, 2008). It is also a crucial factor in

determining whether the alloy is a suitable replacement as a lead-free solder. The solidus temperature is the point where 100% of the alloy is in solid crystals. The liquidus temperature is the point where 100% of the alloy is in liquid form (Chria & Ozvold, 2008). The solid and liquid phase will coexist when the temperature is in between the solidus and the liquidus temperature. The solder alloy is aeutectic alloy when the liquidus and the solidus temperature are the same (Kim et al., 2004).

A relatively large number of lead-free solders have been proposed, and they are summarized in Table 2.1 with their elemental compositions and melting temperatures. The melting temperatures are present in the solidus, Ts, liquidus, Tl, and eutectic temperatures, Te. The solder alloys are binary and ternary.

Since the properties of the binary lead-free solders cannot achieve the requirements, the ternary and quarternary lead-free solders have been developed by the addition of alloying elements to the binary lead-free solders to improve the performance (Anderson et al., 2001). Many of the systems are based on adding small quantities of a third or fourth element to binary alloy systems to lower the melting point and increase wetting and reliability. With increasing amount of additive elements, (a) the melting point of system decreases, (b) the bond strength first rapidly decreases, then almost levels off, then decreases again, and (c) the wettability increases rapidly first, reaching the maximum at composition corresponding to midpoint of plateau of bond strength, then decreases (Lee, 2001). Various compositions can be formulated from the combination of the major elements: Sn, Ag, Cu, Bi, In Sb, and Zn in binary, ternary, or multicomponent systems (Abtew & Selvaduray, 2000; Kucharski & Fima, 2005; Lin et al., 2008; Moelans et al., 2003). Meanwhile, elements such as gold (Au) (Yoon et al., 2008), ferum (Fe)(Wang, 2009), germanium

(Ge) (Yen et al., 2011), lithium (La) (Shiue & Chuang, 2010), gallium (Ga) (Chen et al., 2010), magnesium (Mg), and nickel (Ni) can be added in a very small portion to refine the grain or strengthening mechanisms (Wu et al., 2003) (Wang et al., 2008).

Table 2.1 Common Compositions for Solder Alloys

Solder alloy, wt%	Solidus/liquidus temperature		
	°C	°F	Pattern
52In-48Sn	118/118	244/244	Indium
50In-50Sn	118/125	244/257	Indium
58Bi-42Sn	138/138	281/281	-
67Bi-33In	109/109	228/228	Indium
43Sn-43Pb-14Bi	143/163	289/325	Indium
58Sn-42In	118/145	244.4/293	Indium
97In-3Ag	143/143	289/289	
63Sn37Pb	183/183	361/361	(Control)
60Sn-40Pb	183/190	361/374	
96.5Sn-3.5Ag	233/240	451/464	Indium
80Sn-10In-10Bi	153/199	307.4/390.2	IBM
87Sn-8Zn-5In	175/188	347/370.4	AT&T

Note. Adapted from Lead-Free Soldering (p. 20-30), by A. A. Jasbir, 2007,

New York: Springer Science+Bussiness Media, LLC.

For binary and higher systems, a unique "melting temperature" can be expected only for eutectic compositions or for compositions that melt congruently. Other compositions can be expected to melt over a range of temperatures, with melting beginning at the solidus temperature and being complete at the liquidus temperature. Elements such as Bi, Cd, In, Zn, Au, Ti, Ga, Hg, Cu, Sb, and Ag are the candidates that can lower the melting temperature of Sn which is important to develop solder alloys that possess the require properties for electronic packaging and assembly (Braga et al., 2007; Hwang & Suganuma, 2004).

2.7 Phase Diagrams

Phase diagram is the keyword for initial selection of candidate solders and interaction between solder and substrate especially in wetting and spreading. The understanding of phase diagrams for alloy system is extremely important because there is a strong correlation between microstructure and mechanical properties. The development of microstructure of an alloy is related to characteristics of its phase diagram. In addition, the phase diagram is also frequently used to analyze the melting point or eutectic point of a multi-components system (Zhang et al., 2004).

2.7.1 Sn-Pb Phase Diagram

The binary phase diagram for Sn-Pb is a typical and a very well-known phase diagram. The most popular is 40Sn-60Pb. Figure 2.3 shows the Sn-Pb phase diagram. In industry, solders are called tin-lead.

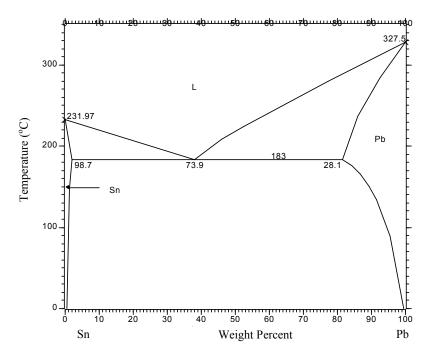


Figure 2.3. Sn-Pb phase diagram (Massalski, 1990).

The temperature of 183 °C is the eutectic temperature, T_{eut} , of the binary Sn-Pb system. At 183 °C, complete solidification takes place. At temperature a fraction of a degree lower than 183 °C, the solder comprises two solid phases: Sn-rich phase designated by β , having a composition 97.5 Sn and Pb-rich phase, α , having a composition 19Sn-81Pb.

2.7.2 Bi-In Phase Diagram

Four stable intermediate phases exist: (1) Bi-In, with a congruent melting point of 109.7 °C; (2) Bi₃.In₅, with peritectic melting point of 88.9 °C; (3) Bi-In₂, melting congruently at 89.5 °C; and (4) In-type ε, existing between 93.5 and 49 °C (Okamoto, 1990). Figure 2.4 shows Bi-In phase diagram.

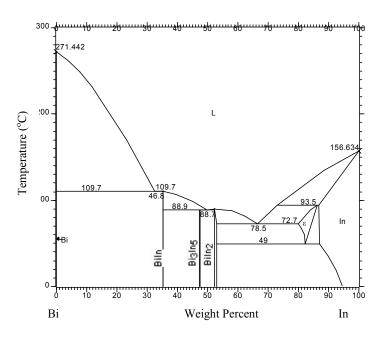


Figure 2.4. Bi-In phase diagram (Okamoto, 1990)

2.7.3 Bi-Sn Phase Diagram

Figure 2.5 shows that the Sn–Bi alloy has a eutectic composition of 42Sn–58Bi and a relatively low eutectic temperature of 139 °C. The room temperature equilibrium phases are Bi and Sn with about 4 wt.% Bi in solid solution (Mei & Morris, 1992).

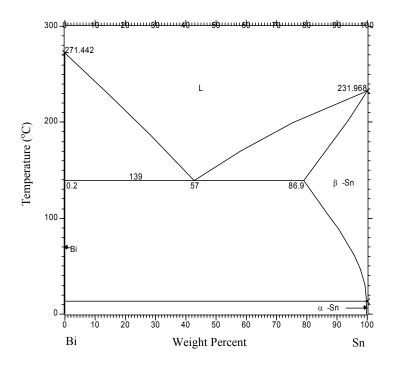


Figure 2.5.Bi-Sn phase diagram (Franke & Neuschütz, 2004)

2.7.4 Bi-Cu Phase Diagram

Figure 2.6 shows that the equilibrium phases of the Cu-Bi system are (1) the liquid, miscible in all proportions; (2) Cu, with restricted solubility of Bi; and (3) Bi, with negligible solubility of Cu. The assessed phase diagram for the Cu-Bi system is of the eutectic type with the eutectic point located close to pure Bi (Chakrabarti & Laughlin, 1990).

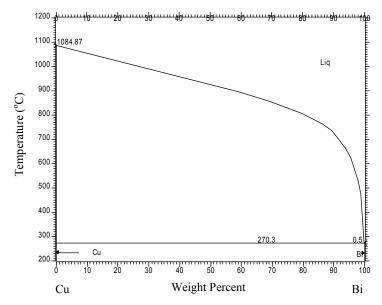


Figure 2.6. Bi-Cu phase diagram (Laughlin, 1990)

2.7.5 Bi-Zn Phase Diagram

The assessment of Okamoto (2000) based on existing experimental results is shown in Figure 2.7. The figure indicates a significant solubility of Zn in (Bi) (few wt% of Zn.

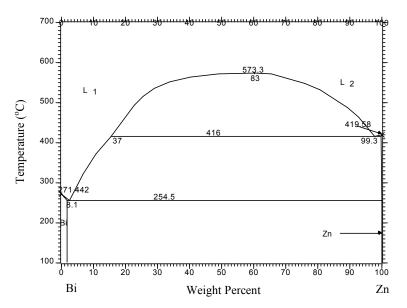


Figure 2.7. Bi-Zn phase diagram (Okamoto, 2000)

2.7.6 In-Sn Phase Diagram

Several In-Sn solder compositions are used in a number of soldering applications, including low-temperature service conditions. The In-Sn system is a binary alloy having an assortment of solid-solution phases. The phase diagram is shown in Figure 2.8. In-Sn system has a eutectic composition of 49.1In-50.9Sn and eutectic temperature of 120 °C. The eutectic composition is often cited as simply 50In-50Sn. The indium-based solder with the composition of In-48Sn is commonly used for surface mount technology (SMT) applications due to its substantially low melting temperature.

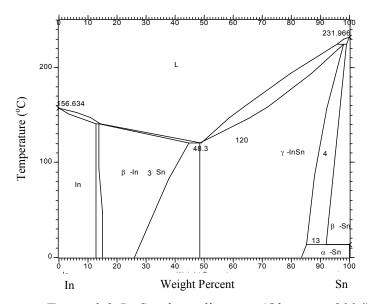


Figure 2.8. In-Sn phase diagram. (Okamoto, 2006)

The two phases formed are intermetallic phases and In-rich, pseudo body centred tetragonal phase, β , which has 44.8 wt.% Sn, and a hexagonal Sn-rich phase, γ , with 77.6 wt.% Sn (Glazer, 1995). Mei and Morris (1992) described the

microstructure of In-48Sn solder on a Cu substrate as having lamellar features. The Sn-rich phase is composed of equiaxed grains. The In-rich phase contains Sn precipitates.

2.7.7 In-Zn Phase Diagram

Figure 2.9 shows In-Zn phase diagram. The liquidus on the Zn side of the eutectic was calculated, assuming maximum solubility of In in (Zn) to be only 0.12at%In. Liquidus temperatures obtained from the thermodynamic properties are very close to the selected experimental values with an average deviation of \pm 6K (Dutkiewicz & Zakulski, 1990).

Tin-zinc eutectic structure consists of two phases: a body centred tetragonal Sn-matix phase and a secondary phase of hexagonal Zn containing less than 1% tin in solid solution (McCormack et al.,1994).

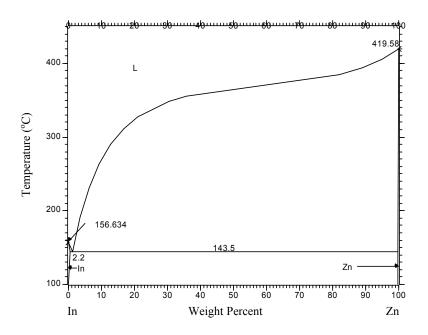


Figure 2.9. In-Zn phase diagram. (Dutkiewicz & Zakulski, 1990)