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Enhancing photovoltaic performance of perovskite solar cells with silica nanosphere antireflection coatings

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Organic-inorganic halide perovskite solar cells have enormous potential to impact the existing photovoltaic industry. As realizing higher power conversion efficiency (PCE) of the solar cell is still the most crucial task, a great number of schemes were proposed to minimize the carrier loss by optimizing the electrical properties of the perovskite solar cells (PSCs). Here, we focus on another significant aspect that is to minimize the light loss by using an antireflection coating (ARC) component to gain a high PCE for PSC devices. In our scheme, silica nanosphere based ARCs are employed to $\text{CH}_3\text{NH}_3\text{PbI}_3$ PSCs for enhancing the device efficiency. SiO_2 nanosphere based ARCs were grown by spin-coating of an aged silica sol. The microstructure and the thickness of the SiO_2 nanosphere based ARC were controlled by changing the spin-coating speed from 400 to 4000 rpm. The effect of SiO_2 nanosphere based ARCs on the photovoltaic performance of perovskite solar cells is systematically investigated. The optimized SiO_2 nanosphere ARC coating on cleaned glass substrate exhibited a maximum transmittance of 96.1% at $\lambda = 550$ nm wavelength, and averagely increased the transmittance by about 3.8% in a broadband of 400–800 nm. The optimized antireflection coating strongly suppressed broadband and wide-angle reflectance in typical PSC solar cells, significantly enhancing the omnidirectional photovoltaic (PV) performance of PSCs. As a result, the power conversion efficiency was improved from 14.81% for reference device without SiO_2 nanospheres to 15.82% for the PSC device with the optimized ARC. Also, the PV performance of the PSC device with the optimized SiO_2 nanosphere ARC revealed less angular dependence for incident light.

1. Introduction

Increasing energy demand, environmental issues and limited availability of fossil fuels are demanding sustainable and renewable energy resources. Solar photovoltaic (PV) technology provides an eco-friendly and renewable energy route to directly convert photon energy into electricity (Kerr, 2007). After decades of research silicon (Si) solar cells have achieved record device efficiencies above 26% (Green et al., 2018). Confirmed terrestrial module efficiencies of crystalline Si solar cells are above 24% and of multicrystalline Si solar cells are close to 20%. However, the efficient establishment of solar cell technology on a global scale requires development in terms of materials and device store to reduce the fabrication cost and increase the power conversion efficiency (PCE). As a new entrant in the PV field, organometallic halide perovskites based photovoltaic device offers the advantages of high-efficiency, low-cost, and facile solution processability, making it one of the most promising next-generation PV technologies for the future. In

perovskite solar cells (PSCs), halide perovskites are of the form ABX_3 (A is an organic ammonium ion: CH_3NH_3^+ (MA^+) or $\text{NH}_2\text{CHNH}_2^+$ (FA^+), B is Pb^{2+} or Sn^{2+} , X is the halogen ion: Cl, Br or I) as the efficient light harvester (Green et al., 2018; Kojima et al., 2009; Burschka et al., 2013; Liu et al., 2013; Yang et al., 2017; Zhao and Zhu, 2016). These perovskite materials possess some fascinating properties, such as tunable bandgap (Noh et al., 2013), high absorption coefficients (Xing et al., 2013), long charge carrier (electron–hole) diffusion length (Stranks et al., 2013), and low-temperature solution processability (Wang et al., 2014). Over the past few years, numerous efforts and extensive investigations have been made into the morphology control of each functional layer (Haque et al., 2017; Yang et al., 2016), compositional engineering of perovskite materials (Hao et al., 2014; Jeon et al., 2015), device structure design (Burschka et al., 2013; Lee et al., 2012; Gao et al., 2014; Cao and Xue, 2014; Tan et al., 2014; Zhang et al., 2017), crystallization modulation of perovskites (Jeon et al., 2014; Xiao et al., 2014; Im et al., 2014) and interfacial engineering (Zhou et al., 2014;

Zhang et al., 2017; Luo et al., 2017; Assadi et al., 2018). Consequently, the power conversion efficiencies (PCEs) of PSC devices have been boosted from 3.8% to 22.7% under 1 sun conditions (100 mW cm^{-2} AM 1.5G) (Green et al., 2018).

The utilization of highly transparent antireflection coating (ARC) in perovskite solar cells provides an important approach to reduce or suppress Fresnel reflection losses, to increase the amount of light entering the PV device and hence, to enhance the power conversion efficiency of PSCs. PSCs typically are formed on transparent conductive oxide (TCO) glasses, such as fluorine doped tin oxide (FTO) and indium tin oxide (ITO) slides. There is a trade-off in PSC devices between the sheet resistance of the TCO and its optical transmittance. Whilst the use of thicker TCOs can significantly reduce the sheet resistance, it will also result in a decrease in transmittance. Thus, the optical transmittance of FTO or ITO slides is limited, which has a considerable impact the efficiency of PSCs. Currently, the ARCs have been widely employed on silicon (Si) or transparent substrates (e.g., glasses, plastics, etc.) in different types of Si based or organic solar cells or modules to reduce the surface reflection losses over a broad range of incident wavelengths and angles (Xu et al., 2008; Sakhujia et al., 2014; Raut et al., 2011; Leem et al., 2014a, 2014b; Ballif et al., 2004).

An important and well-known phenomenon lowering the overall PCE of solar cell devices is Fresnel reflection, which takes place when a refractive index contrast exists between the ambient medium (air) and the solar cell device. The reflectance losses due to the refractive index mismatch between a material and the air can be reduced by applying an ARC with an intermediate refractive index between air and the substrate material (Prevo et al., 2005). The antireflection coatings can either be materials with an intermediate refractive index between the substrate and air or porous or refractive index-graded materials (Mahadik et al., 2015). Most of these antireflection coatings (ARCs) are manufactured by film deposition techniques such as chemical vapor deposition (CVD), sputtering, or evaporation, as well as possible lithography steps. These methods are costly and often involve high-temperature environments that are unsuitable for certain solar cell technologies (Wan et al., 2013; Ali et al., 2014; Lopez et al., 2013). Silicon dioxide (SiO_2) is the most ideal material for ARC due to its low refractive index, good durability and environmental resistance. Especially, SiO_2 nanospheres, which do not require vacuum-based or high-temperature fabrication processes, offer a promising nanophotonic ARC option due to their light trapping and scattering properties (Grandidier et al., 2011, 2012). Further, SiO_2 nanoscale structures are stable with respect to extreme temperatures and light illumination. Different techniques have been used to achieve layers of SiO_2 nanospheres on top of photoactive layers on flexible and rigid substrates, including the Langmuir–Blodgett method (Grandidier et al., 2013; Hsu et al., 2008), sedimentation (Jiang et al., 1999), and controlled evaporation (Mihi et al., 2011). Although these fabrication approaches are simple, scalable and inexpensive and could enable large-scale manufacturability of microresonator-based photovoltaics, the application of nanosphere based ARCs on photovoltaic devices are scarcely reported.

In this study, we demonstrate a facile and simple approach to prepare silica nanosphere based antireflection coatings. SiO_2 nanosphere based ARCs were grown by spin-coating of an aged silica sol. The structure and the thickness of the SiO_2 nanosphere based ARC were finely adjusted by changing the spin-coating speed from 400 to 4000 rpm. The microstructure and the effective refractive index of the SiO_2 nanosphere based ARC were optimized to achieve high transmittance. The effect of SiO_2 nanosphere based ARCs on the photovoltaic performance of perovskite solar cells is systematically investigated. The optimized SiO_2 nanosphere ARC coating on cleaned glass substrate exhibited a maximum transmittance of 96.1% at $\lambda = 550 \text{ nm}$ wavelength, and averagely increased the transmittance by about 3.8% in a broad wavelength range of 300–800 nm, therefore leading to a significant increase in the efficiency of the PSC device from 14.81% to 15.82%, demonstrating over 6.8% enhancement, compared with the

reference device without SiO_2 nanospheres. The present process provides a simple and cost effective method for the preparation of antireflection coatings, which benefit the development of PSC solar cells, as well as the light-harvesting scheme applicable to a wide variety of optoelectronic devices.

2. Experimental details

2.1. Preparation of the sols and deposition of the thin films

Tetraethoxysilane (TEOS) was procured from Sigma Aldrich. TEOS was used as a precursor for SiO_2 nanosphere ARC coatings. Ethanol also obtained from Sigma Aldrich was used as the solvent $\text{NH}_3 \cdot 3\text{H}_2\text{O}$ was purchased from Aladdin Industrial Corporation and used for the catalysis of the sol.

A hybrid silica sol was prepared by mixing the precursor materials (Jun et al., 2008). Initially, 1.5 mL of $\text{NH}_3 \cdot 3\text{H}_2\text{O}$ solution (10 wt%) was mixed with deionized (DI) water (6 mL), ethanol (70 mL) in a 100 mL flask by magnetic stirring. Then, 3 mL of TEOS was added drop by drop to the mixture. The final mixture was magnetically stirred for 30 min. The silica sol obtained was kept at room temperature (RT) for at least 1 day for aging and finally used for deposition on external-facing surfaces of TCO transparent substrates of PSCs.

2.2. Device fabrication

FTO slides (Pilkington TEC 15) were patterned by etching with Zn powders and 2 M HCl. The etched slides were then cleaned with liquid detergent, acetone, ethyl alcohol and DI water for 15 min, sequentially, to remove the organic or inorganic residues, and finally dried in a vacuum oven. Titanium isopropoxide (TTIP) (200 μL) and ethanol (5 mL) were mixed to prepare a clear precursor sol. The precursor sol was spin-coated onto the etched FTO substrate at 4500 rpm, followed by annealing at 500 °C to form a compact TiO_2 (c- TiO_2) layer. Diluted TiO_2 pastes were prepared by mixing TiO_2 paste (Dyesol 18NR-T, 0.1 g) and anhydrous ethanol (444 μL). The mesoporous TiO_2 (p- TiO_2) layer was deposited on the c- TiO_2 layer by spin-coating the diluted TiO_2 pastes at 2000 rpm for 30 s. The layers were then sintered at 500 °C for 30 min in air. After cooling down to the RT, the samples were treated using the TiCl_4 aqueous solution at 70 °C for 30 min and dried at 500 °C for 30 min. The perovskite MAPbI_3 absorber was grown via a spin-coating process using a GBL/DMSO solution of PbI_2 and MAI. Briefly, a pure perovskite-precursor solution (1.25 mol L^{-1}) was prepared by mixing MAI (0.1975 g) powders and lead iodide PbI_2 (0.5785 g) in GBL (700 μL) and DMSO (300 μL) at 60 °C for 12 h. The formed precursor solution was deposited onto the p- TiO_2 /c- TiO_2 /FTO sample via a successive two-step spin-coating process, at 2000 rpm for 30 s and at 3500 rpm for 50 s, respectively. Anhydrous diethyl ether was dripped onto the center of the sample in the second spin-stage during the spin-coating process (Luo et al., 2017). The perovskite-precursor coated sample was heated and dried on a hot plate at 110 °C for 30 min. The hole-transport material (HTM) was deposited on the grown perovskite absorber by spin-coating a spiro-OMeTAD solution at 4000 rpm for 30 s. 100 nm thick AgAl alloy film with an active area of 0.1 cm^2 was formed via evaporation on the Spiro-OMeTAD-coated film (Jiang et al., 2016). Finally, SiO_2 nanosphere based ARCs were deposited on the front glass surface of the as-prepared PSC solar cell by spin-coating of the aged silica sol with various spin-coating speeds from 400 to 4000 rpm.

2.3. Characterization

The morphologies of SiO_2 nanosphere based ARC films were characterized by a high-resolution field emission scanning electron microscope (FESEM, Hitachi S4800). Optical spectra of SiO_2 ARCs were examined and characterized by means of ultraviolet–visible light (UV–vis) spectrometer (Hitachi, U-3010). Photocurrent density–voltage (J - V)

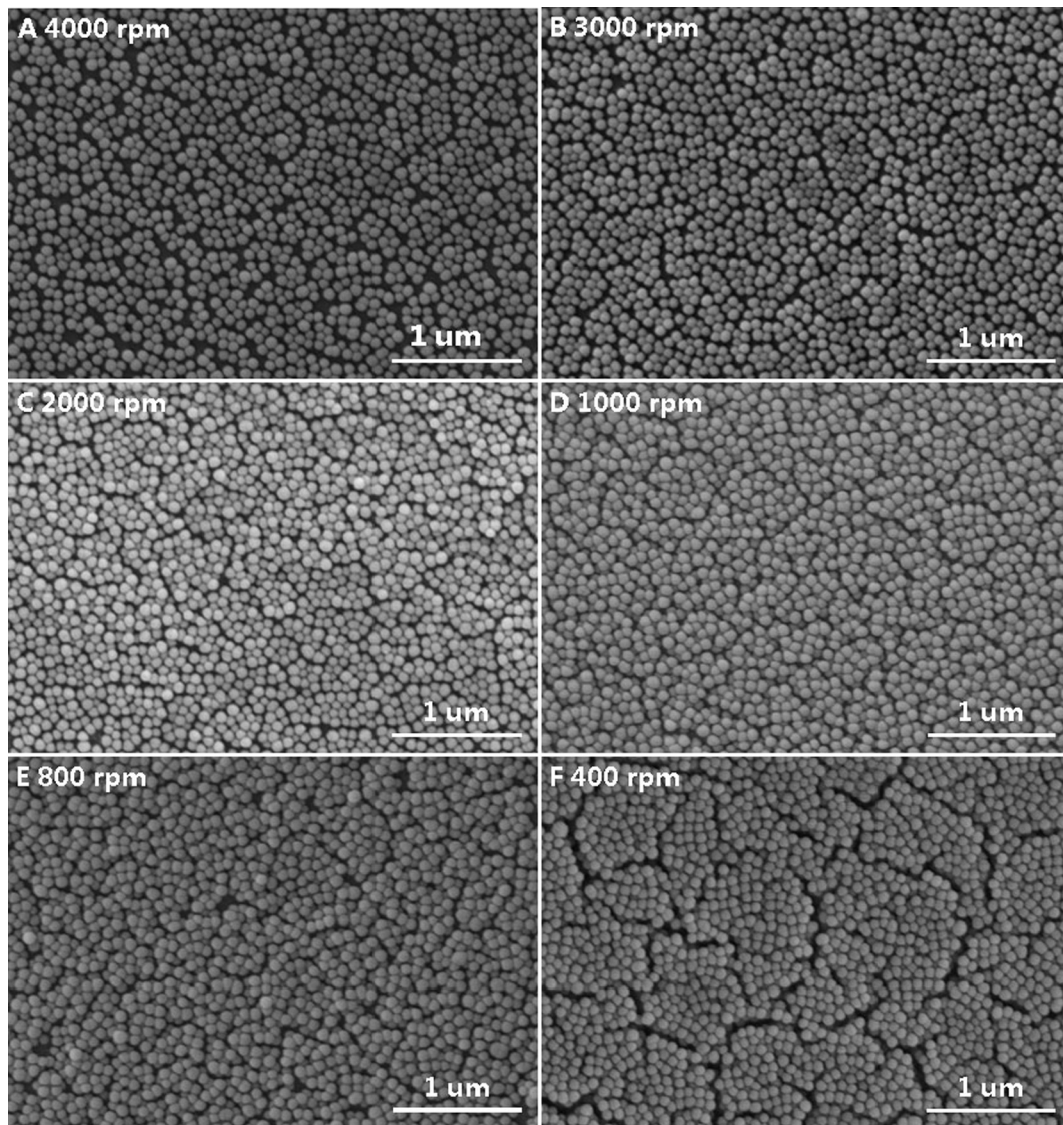


Fig. 1. FESEM image of the top surface morphology of SiO₂ sphere based films grown on glass substrates with different spin-coating speeds.

measurements were performed using an AM 1.5 solar simulator equipped with a 1000 W Xenon lamp (Model No. 91192, Oriel, USA). The solar simulator was calibrated by using a standard silicon cell (Newport, USA). The light intensity was 100 mW cm⁻² on the surface of the test cell. During device photovoltaic performance characterization, a metal aperture mask with an opening of about 0.09 cm² was used. External quantum efficiency (EQE) measurements (74125, Oriel, USA) were also carried out for these solar cells.

3. Results and discussion

Figs. 1 and 2 show the top surface and cross-sectional SEM images of SiO₂ sphere based films grown on glass substrates with different spin-coating speeds. From Fig. 1(a), at the spin-coating speed high up to 4000 rpm, a monolayer of monodisperse silica spheres was formed on the glass substrate. The synthesis of monodisperse and spherical silica spheres were achieved by the ammonia-catalyzed hydrolysis and condensation of TEOS in mixed ethanol - water solvents (Teng et al., 2010). As can be seen from Figs. 1 and 2, the formed silica spheres have perfect spherical morphology and controllable monodisperse particle size. The size of SiO₂ spheres is about 75 nm in diameter. From the cross-sectional SEM pictures shown in Fig. 2, the spin-coating speed impacted the nanostructure and the thickness of the ARC coating. The sample

exhibited increasing film thickness with decreasing spin-coating speed. The top-down SEM micrographs show that, for either monolayer or thicker SiO₂ sphere based film, the packing fraction remained relatively uniform, consisting of three-dimensional (3D) close-packed spheres. At the higher speed from 2000 rpm to 4000 rpm, the ARC film became submonolayers, with increasing void space between SiO₂ spheres, as shown in Fig. 2(a-c). At the lower speed from 800 rpm to 400 rpm, the ARC film turned into thicker and more multilayers. At the speed of 400 rpm, the number of stacking layers increased to 6. From Figs. 1(d) and 2(d), with a speed of 1000 rpm, the ARC film developed into sub-double layers with uniformly distributed voids formed in the upper layer, leading to a 3D crystal architecture, which is more clearly shown in the insert of Fig. 2(d). The surface feature of the SiO₂ sphere based ARC deposited at 1000 rpm is similar to that of textured silicon or aluminum-doped zinc oxide surfaces applied in crystalline or amorphous silicon solar cells (Campbell and Green, 1987, 2001; Zhu et al., 2011). A single crystalline Si substrate can be textured by etching along the faces of the crystal planes, leading to a surface made up of random pyramids. Highly efficient crystalline silicon solar cells have been developed using anti-reflection and light trapping properties of pyramidally textured surfaces of Si wafers. Geometric textures of the SiO₂ sphere based ARC deposited at 1000 rpm can in principle perform as well as a random pyramid texture by trapping more light in Si based

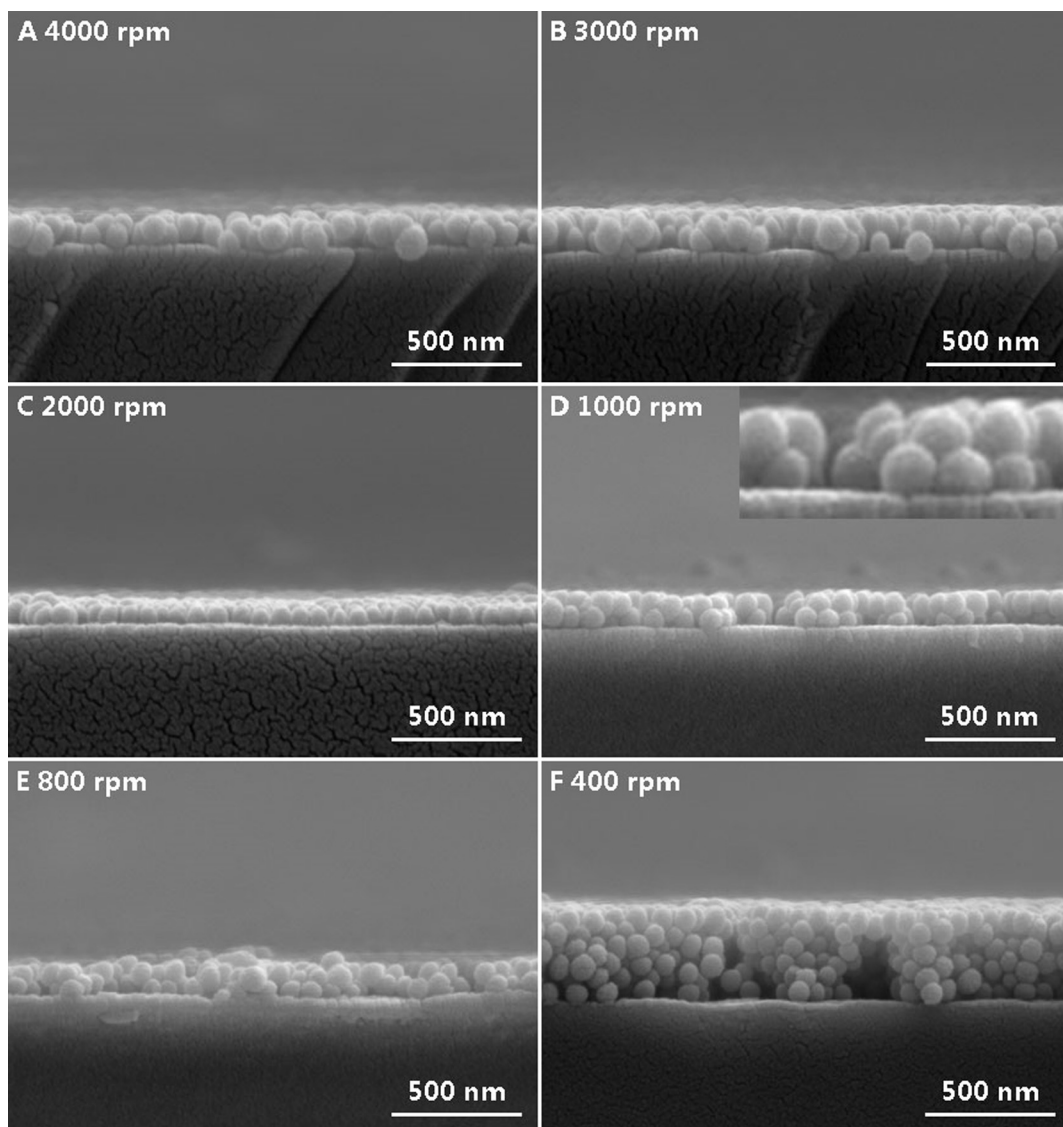


Fig. 2. Cross-sectional FESEM picture of SiO₂ sphere based films grown on glass substrates with different spin-coating speeds.

solar cell, with more energetic passing into the PV devices. Using the proposed approach in this work, the particle ARC film thickness and structure could be precisely predictable and controlled. Differently assembled structures of SiO₂ nanosphere based ARCs have various optical transmittance properties, and result in different PV performance of PSCs with SiO₂ nanosphere based ARCs.

Fig. 3(a) shows the transmittance spectra of the non-coated pristine glass (reference sample) and SiO₂ sphere based ARCs grown on glass substrates at different speeds. Their Average (T_{mean}) and maximal (T_{max}) transmittance values in the wavelength region of 400 nm–800 nm are shown in Table 1. The non-coated pristine glass has an average transmittance (T_{mean}) of 91.5% in the wavelength region of 400–800 nm. For a low spin-coating speed of 400 rpm, the multilayered SiO₂ ARC shows a mean transmittance of 90.1% in this optical region, obviously lower than that of the pristine glass. In contrast, for a speed of 1000–4000 rpm, the formed single or double layered SiO₂ sphere coating exhibits the higher transmittance than the non-coated pristine glass over the whole UV–visible–near-infrared (NIR) region. For the sample prepared at a speed of 2000–4000 rpm, the average transmittance is up to 94%–95%. At the sample prepared at a speed of 1000 rpm, the average transmittance reaches a maximum value (95.3%). Thus, the silica coating showed an increased transmitted light of approximately 3.8% in the wavelength region of 400–800 nm,

appropriate for perovskite solar cells.

The transmittance property of an ARC film depends on its thickness and surface morphology. An ideal homogeneous single-layer ARC having an optical thickness of one-quarter of a wavelength achieves effectively 0% reflection at a specific wavelength when its refractive index, n_c , meets the condition of $n_c = \sqrt{n_a n_s}$, where n_a and n_s are the refractive indices of the air and the substrate, respectively (Prevo et al., 2005). The refractive index of soda lime glass is roughly 1.52 at 550 nm, which implies an ideal refractive index of 1.23 for an ARC coating. Moreover, for a single layer coating and normal incidence, according a quarter-wavelength-thick film gives the minimum reflection or maximum transmittance, the wavelength λ which corresponds to minimum reflection is calculated by: $\lambda = 4n \times d$, where, n and d represent the refractive index and layer thickness, respectively (Xiao et al., 2014). The refractive index of the SiO₂ AR film is 1.46, therefore the optimum thickness for the wavelength range of 400–800 nm is 69.0–138.0 nm. The result indicated that the silica sphere ARC films deposited at 1000–4000 rpm were of the proper quarter-wavelength film thickness for AR in the 400–800 nm wavelength region, considering that the diameter of SiO₂ spheres is about 75 nm. Based on the transmittance measurements shown in Fig. 3(a), compared to the case of the reference sample, we further calculate the increase percentage in transmittance ($\Delta T/T_{\text{ref}}$) with the addition of the SiO₂ nanosphere film in

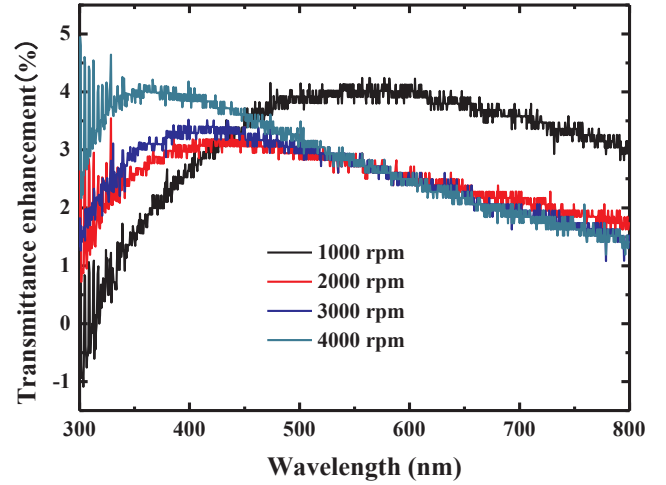
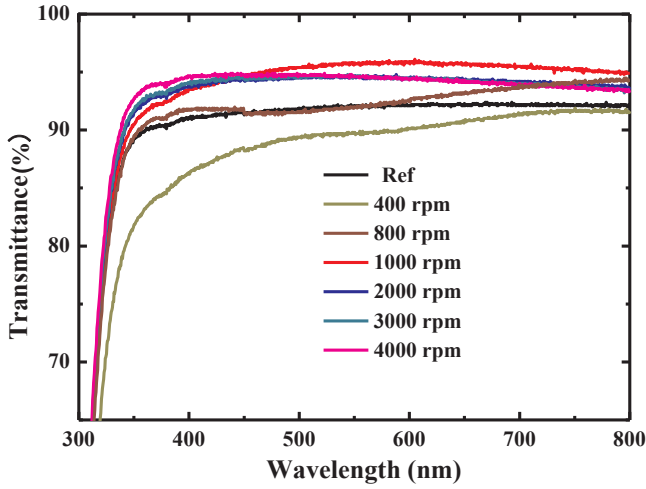
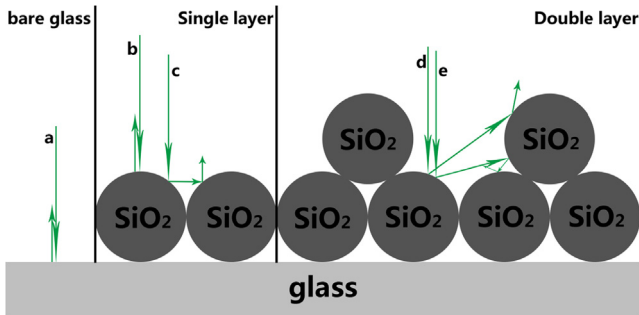


Fig. 3. (a) Transmittance spectra of the non-coated pristine glass (reference sample) and SiO₂ sphere based ARCs grown on glass substrates at different speeds. (b) Transmittance enhancements of SiO₂ sphere based ARCs grown on glass substrates at speeds of 1000 rpm–4000 rpm.

Table 1

Average (T_{mean}) and maximal (T_{max}) transmittance values of SiO₂ sphere based ARCs grown on glass substrates at different speeds in the wavelength region of 400 nm–800 nm.

Speed (rpm)	Glass substrate	4000	3000	2000	1000	800	400
T_{mean} (%)	91.5	94.3	94.3	94.2	95.3	92.7	90.1
T_{max} (%)	92.4	94.9	94.8	94.8	96.1	94.5	91.9



Scheme 1. Schematic of the reflection path of the incident light in the SiO₂ sphere based ARCs.

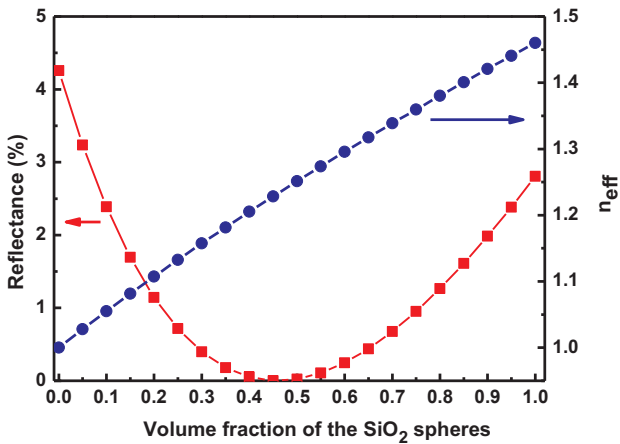


Fig. 4. Calculated reflectance for quarter-wave thickness silica coating on glass using Eq. (2). The effective refractive index n_{eff} of the coating from Eq. (1) is plotted on the right axis for reference.

From this figure, the peak of $\Delta T/T_{\text{ref}}$ changes with the speed of coating. The maximal transmittance enhancement was achieved at a wavelength of 360 nm, 400 nm, 405 nm and 580 nm for the SiO₂ ARC deposited at 4000 rpm, 3000 rpm, 2000 rpm and 1000 rpm, respectively. Thus, a red-shift of the peak of $\Delta T/T_{\text{ref}}$ is observed as the speed slows down. As shown in Fig. 1, the thickness of SiO₂ ARC increases as the speed slows down. According to the equation: $\lambda = 4n \times d$, the value of λ corresponding to the minimal reflection or the maximal transmittance will increase as the value of d increase, leading to the red-shift characteristics shown in Fig. 3(b).

A rich variety of optical structures can increase the transmittance property of the film, such as pyramid, inverted pyramid, sub-wavelength grating, nanowire array, biomimetic and so on (Chen and Hong, 2016; Haase and Stiebig, 2007; Esteban et al., 2010; Huang et al., 2012). For a randomly textured surface, the increased transmittance is explained by a second reflection of the incident light at the sidewalls. Different assembly structures of SiO₂ ARC films shown in Figs. 1 and 2 lead to different transmittance properties in Fig. 3. As shown in Scheme 1, for the surface of the bare glass, incident light (light a) reflected just once ($T_{\text{mean}} = 91.5\%$). For the surface of the SiO₂ close-packed sphere monolayer ARC, some areas of spheres are almost parallel to the original flat substrate surface, so incident light (light b) also reflected just once, and some areas of spheres have an angle from the original wafer surface, this allows the incident light (light c) to be reflected at least twice on the surface and can reduce the reflection loss ($T_{\text{mean}} \cong 94\%$). For the surface of the SiO₂ sphere multilayer ARC with a close-packed sphere feature, the reflection path of incident light are similar with that for the SiO₂ close-packed sphere monolayer, the transmittance would also be influenced by the thickness of the film ($T_{\text{mean}} = 90\%–92\%$). However, for the sub-double layer SiO₂ nanosphere based ARC film prepared at 1000 rpm, its top surface allows the incident light (light d and e) to be reflected more times than the case for the single layer SiO₂ nanosphere based ARC film, and thus the corresponding reflection loss is much lower ($T_{\text{mean}} = 95.3\%$).

The transmittance property of ARC films prepared with different speeds can also be explained by the changing of volume fraction of the SiO₂ nanosphere in the coating. If the ARC coating is a composite made of two or more materials dispersed in sub-wavelength domains, such as a porous thin film of particles tens of nanometers in diameter, the effective refractive index of the nanosphere coating, n_{eff} , can be estimated by a volume average as shown in the following equation (Prevo et al., 2005):

$$n_{\text{eff}} = (\phi_p n_p^2 + (1-\phi_p) n_a^2)^{1/2} \quad (1)$$

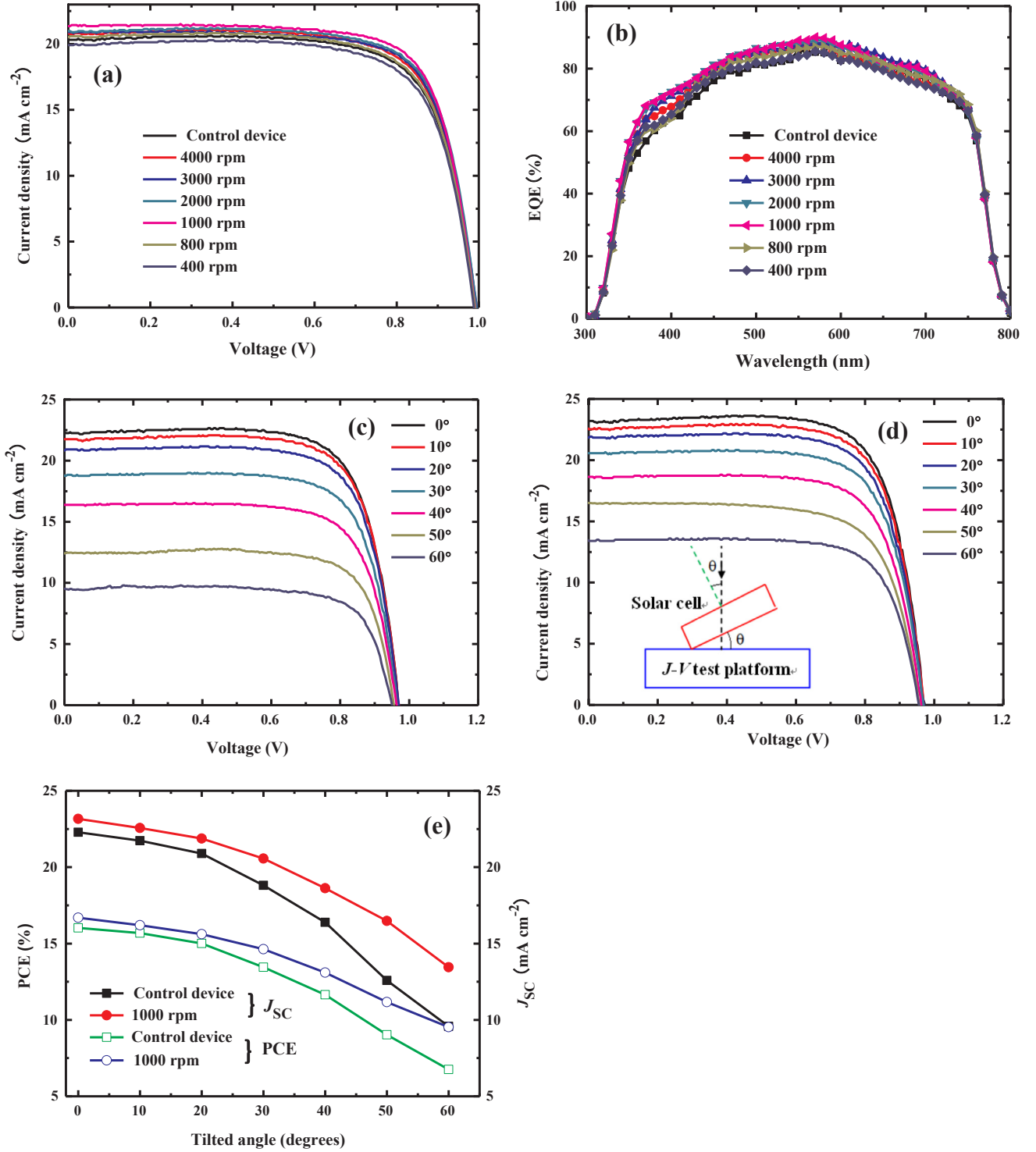


Fig. 5. (a) J - V characteristics and (b) EQE spectra of PSCs without (control device) and with the SiO₂ sphere based ARCs at different spin-coating speeds. J - V curves with respect to the tilted angle of (c) control device (0°) and (d) PSC with the SiO₂ sphere based ARC prepared at 1000 rpm. (e) J_{sc} and PCE values of the control device and the PSC with the SiO₂ sphere based ARC prepared at 1000 rpm as function of the tilted angle.

Here ϕ_p is the volume fraction of the particles in the coating, while n_p and n_a are the refractive indexes of the particles and the ambient medium (air in this case), respectively. From Eq. (1), n_{eff} increases monotonically as a function of ϕ_p from 1.0 (n_a) to 1.46 (n_p), an established value for the refractive index of silica particles. The reflectance (R) can be calculated for quarter-wave thickness silica coating on glass using Eq. (2).

$$R = \frac{(n_s - n_c)^2}{(n_s + n_c)^2} \quad (2)$$

Fig. 4 shows the calculated reflectance for quarter-wave thickness silica coating on glass using Eq. (2). The effective refractive index of the coating from Eq. (1) is plotted on the right axis for reference. As can be seen from Fig. 4, complete ARC at a wavelength of 555 nm is predicted for $\phi_p = 0.45$, which corresponds to $n_{eff} = 1.23$.

Table 2

Parameters of PSCs without (control device) and with the SiO₂ sphere based ARCs at different spin-coating speeds.

Device	V _{OC} (V)	J _{SC} (mA/cm ²)	Fill factor (%)	Efficiency (%)
Control	0.99	20.34	73.5	14.81
4000 rpm	0.99	20.82	73.7	15.23
3000 rpm	0.99	20.91	74.1	15.42
2000 rpm	0.99	20.97	74.5	15.51
1000 rpm	0.99	21.39	74.6	15.82
800 rpm	0.99	20.57	73.4	14.95
400 rpm	0.99	19.95	73.2	14.42

Thus, the factor limiting the ARC efficiency of these nanosphere coatings is not the thickness but the coating porosity which dictates the effective refractive index. Our SEM observations clearly exhibit voids distributed throughout the sub-monolayer or the sub-double layer coatings at high deposition speeds of 4000–1000 rpm, as shown in Figs. 1 and 2. The sphere volume fraction ϕ_p in the ARC coatings extracted from the SEM measurements changed with the spin-coating speed. For the sub-monolayer SiO₂ nanosphere ARC film deposited at 2000 rpm, 3000 rpm and 4000 rpm, ϕ_p is around 50%, 41% and 36%, respectively; hence $n_{\text{eff}} \sim 1.25, 1.21$ and 1.19 , respectively. For the SiO₂ nanospheres based ARC film grown at 800 rpm and 1000 rpm, ϕ_p is around 51%, and 44%, respectively; hence $n \sim 1.25$, and 1.22 , respectively. For the multilayer SiO₂ nanosphere based ARC film at 400 rpm, ϕ_p is around 58%; hence $n \sim 1.29$. The effective refractive index of the SiO₂ ARC film deposited at 1000 rpm is closest to 1.23, where the antireflective coating on the glass substrate achieves effectively 0% reflection at a wavelength of 550 nm. Therefore, the reflectance, particle packing fraction and effective refractive index data obtained from Eqs. (1) and (2) are consistent with the SEM findings.

The roughening features of the SiO₂ sphere based ARC at the suitable speed reduces reflection by increasing the chances of reflected light bouncing back onto the surface or into the PSC device, resulting in the enhanced PV performance of the PSC device, confirmed by the current density-voltage (*J-V*) and EQE measurements. Fig. 5(a) and (b) show *J-V* characteristics and EQE spectra of PSCs without (control device) and with the SiO₂ spheres based ARCs at different spin-coating speeds, respectively. The PV parameters obtained from corresponding PSCs including short circuit-current density (J_{SC}), open circuit voltage (V_{OC}), PCE and fill factor (FF) are tabulated in Table 2. The PV results displayed in Fig. 5 and Table 2 were from the same reference device. The PV performance measurements were first carried out for the as-prepared PSC device without SiO₂ nanospheres (control device). Then, a SiO₂ nanosphere based ARC was deposited on the front glass surface of the control device at a specific spin-coating speed (400–4000 rpm). A blue-green ARC coating was formed on the front glass surface and ready for the PV performance assessments of the PSC device with a SiO₂ spheres based ARC. After the PV performance measurements, the SiO₂ spheres based ARC was carefully removed from the front glass surface with DI water, ethanol and cotton swabs, and got ready for growth of a new SiO₂ nanosphere based ARC at other spin-coating speeds. From Fig. 5(b), in the 400 nm–750 nm range, the EQE value is increased in the order of device@400 rpm < control device < device@800 rpm < device@4000 rpm < device@3000 rpm < device@2000 rpm < device@1000 rpm. The EQE is a product of light harvesting efficiency, charge separation efficiency, and charge collection efficiency, and hence the EQE improvement is attributed to the improved light harvesting efficiency because the SiO₂ sphere ARC film was just attached on the external and front glass surface of the as-prepared PSC solar cell without any change with the device inner structure. From Fig. 5(a) and Table 2, the reference device exhibits a V_{OC} of 0.99 V, J_{SC} of 20.34 mA cm⁻² and FF of 73.5%, resulting in a total PCE of 14.81%. The increased trend of the J_{SC} and the PCE values in the PSC device with respect to the SiO₂ nanosphere based ARC is

consistent with the EQE findings. The improved device efficiency can be attributed to the improved J_{SC} due to the increased transmittance by the AR function of the SiO₂ sphere based ARC film shown in Fig. 3. The optimal cell was obtained with the SiO₂ nanosphere based ARC film deposited at 1000 rpm. It shows a PCE of 15.82%, demonstrating over 6.8% enhancement, compared with the reference device without SiO₂ nanospheres. The PCE enhancement for the device@1000 rpm comes from the greatly improved short-circuit photocurrent J_{SC} (from 20.34 to 21.39 mA cm⁻²). The *J-V* characteristics and EQE results are supported by the optical measurement data shown in Fig. 3.

Fig. 5(c) and (d) show the *J-V* curves with different tilted angles for the control device and the PSC with the SiO₂ sphere based ARC prepared at 1000 rpm, respectively. During the measurements, the AM 1.5 solar light vertically irradiated on the *J-V* test platform, and the PSC device was placed on the platform with various tilted angles ($\theta = 0^\circ\text{--}60^\circ$) above the horizontal during *J-V* characterization as shown in the insert of Fig. 5(d). The incident angle of solar light at the surface of the measured solar cell is equal to the tilted angle. As we observed, both J_{SC} and PCE values of the PSC with the optimized SiO₂ sphere based ARC smoothly decreased with respect to the tilted angle of the device, whereas those of the reference device more steeply and quickly decreased with the increase of the tilted angle. This trend can be seen more clearly in Fig. 5(e), where we have plotted the J_{SC} and PCE data as function of the tilted angle. The result might be attributed to the enhanced light absorption in the active layer of the device due to the increased effective optical path lengths (i.e., strong light scattering) caused by the SiO₂ sphere based ARC film. Furthermore, the omnidirectionally aligned microstructures of the optimized SiO₂ sphere based ARC allow for suppressing the surface reflection for obliquely incident light (Leem et al., 2013). Therefore, the PSC device with the optimized SiO₂ sphere based ARC can generate electricity more efficiently under the circumstance of varying the incident angle of light in a day and the seasons than the reference device. The SiO₂ nanosphere based ARC prepared at 1000 rpm strongly suppress broadband and wide-angle reflectance in typical PSC solar cells, significantly enhancing the omnidirectional PV performance of PSCs.

4. Conclusions

We have developed silica dielectric nanosphere ARCs that suppress broadband and wide-angle reflectance within CH₃NH₃PbI₃ PSC devices and ultimately increase the power conversion efficiency and enhance the omnidirectional photovoltaic performance of solar cells. SiO₂ nanosphere based ARCs were grown by spin-coating of an aged silica sol. The microstructure and the thickness of the SiO₂ nanosphere based ARC were controlled by changing the spin-coating speed from 400 to 4000 rpm. The optimized SiO₂ nanosphere ARC coating on cleaned glass substrate exhibited a maximum transmittance of 96.1% at $\lambda = 550$ nm wavelength, and averagely increased the transmittance by about 3.8% in a broadband of 400–800 nm. The power conversion efficiency was improved from 14.81% for reference device without SiO₂ nanospheres to 15.82% for the PSC device with the optimized ARC. Also, the PV performance of the PSC device with the optimized SiO₂ nanosphere ARC revealed less angular dependence for incident light. Because the layer of SiO₂ nanospheres can be grown with an easy, inexpensive, and scalable process, this type of ARC is an excellent candidate for replacing conventional ARC technologies that are inappropriate for certain photovoltaic technologies.

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