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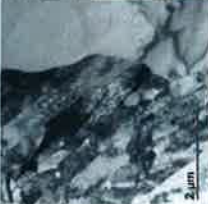
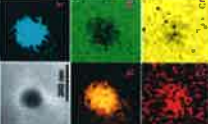
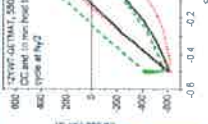


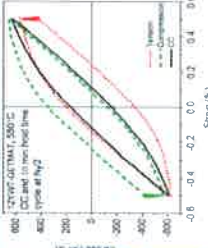
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## Creep-fatigue interaction in ODS steels

**Ankur Chauhan, Janir Aktaa**

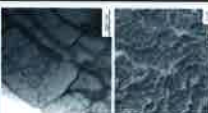
7<sup>th</sup> International conference on creep, fatigue and creep-fatigue interaction  
January 19-22, 2016  
IGCAR, India




100% Cyclic Fatigue, 250°C  
Creep-fatigue interaction  
452 cycles at  $\dot{\epsilon}_p = 10^{-5}$  s<sup>-1</sup>

Stress (MPa) vs. Strain (%)




HAADF image  
EDS maps

KIT – University of the State of Baden-Wuerttemberg and  
National Research Center of the Helmholtz Association



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## Objective

Investigating and characterizing the deformation and damage mechanisms of ODS steels

Mechanical Characterization  
 • Tensile  
 • Fatigue  
 • Creep-Fatigue


➔

Microstructural Characterization  
 • Damage and deformation mechanisms


➔

Modelling  
 • Constitutive equations

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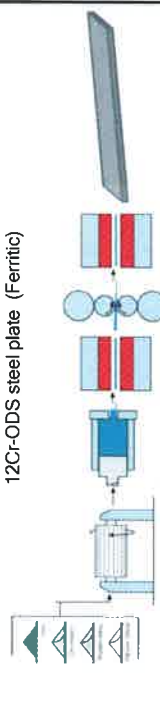


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## Material



12Cr-ODS steel plate (Ferritic)

Element	Cr	W	Ti	Si	Y <sub>2</sub> O <sub>3</sub>	Fe
wt. %	11.59	1.87	0.22	0.1	0.23	Bal.

Powder raw material: Mechanical alloying at atmosphere, 48 h, 250 rpm


Scaled Degassed: 400 °C, 1 Pa, 3 h

Annealing: 1150 °C, 1 h


Cold rolling: Enrolled/Hotrolled, 1150 °C

Annealing: 1200 °C, 1 h

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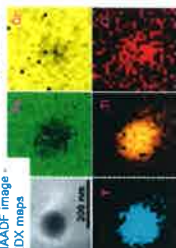


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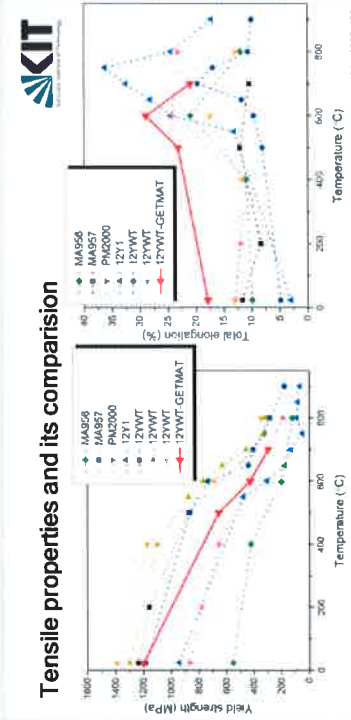
## Microstructure @ as-received state

- Bimodal microstructure
  - Heavily deformed elongated small grains (200 nm to 5 µm)
  - Deformation-free large elongated grains → partial recrystallization (up to 20 µm)
- Smallest grains are mostly oriented along <110> in extrusion direction
- Larger grains have no preferential orientation
- Complex Y-Ti-O particles : ~6 - 90 nm

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### Tensile properties and its comparison



◆ MA956  
 ● MA957  
 ▲ MA958  
 ▼ PM2000  
 ◆ 12Y1  
 ● 12YWT  
 ▲ 12YWT  
 ▼ 12YWT-GETMAT

◆ MA956  
 ● MA957  
 ▲ MA958  
 ▼ PM2000  
 ◆ 12Y1  
 ● 12YWT  
 ▲ 12YWT  
 ▼ 12YWT-GETMAT

Yield strength (MPa)  
 Temperature (°C)

Total elongation (%)  
 Temperature (°C)

Kishitani et al., J. Nucl. Mater., 341, 2005, 103  
 Sokolov et al., J. Nucl. Mater., 367-368, 2007, 103  
 Rehmann/Chauhan et al., J. Nucl. Mater., 44, 2012, 888  
 Rehmann/Chauhan et al., J. Nucl. Mater., 468, 2015, 148

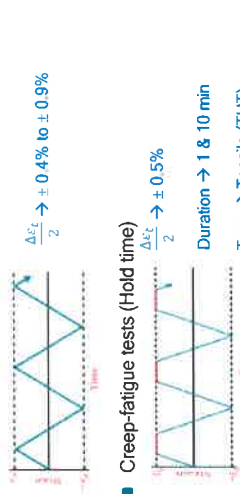
- Good compromise between strength and ductility
- Characteristic total elongation peak at 600°C → change in mechanism

**The improved ductility is an effect of bimodal microstructure**

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### Cyclic tests at elevated temperatures

- Two temperatures (550°C & 650°C)
- Performed in air,  $R = -1$ ,  $\dot{\epsilon} = 10^{-3} s^{-1}$
- Test types
  - Low cycle fatigue (LCF) or continuous cycling (CC) tests
  - Creep-fatigue tests (Hold time)

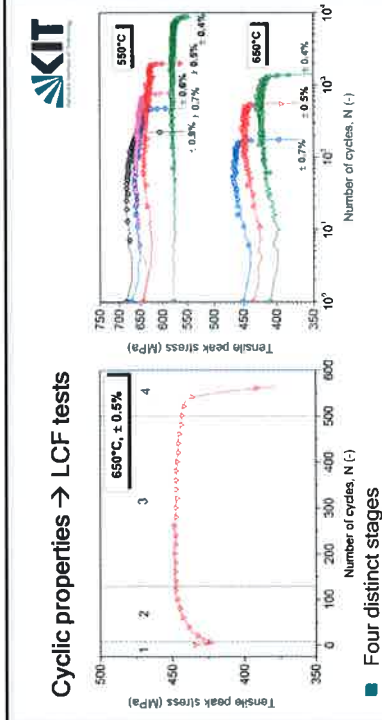


$\frac{\Delta \epsilon_c}{2} \rightarrow \pm 0.4\% \text{ to } \pm 0.9\%$   
 $\frac{\Delta \epsilon_t}{2} \rightarrow \pm 0.5\%$   
 Duration → 1 & 10 min  
 Type → Tensile (THT)  
 Compressive (CHT)  
 Tensile-compressive (THT-CHT)

Mini specimen (7.6x2mm)  
 MTS servo-hydraulic UTM

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### Cyclic properties → LCF tests

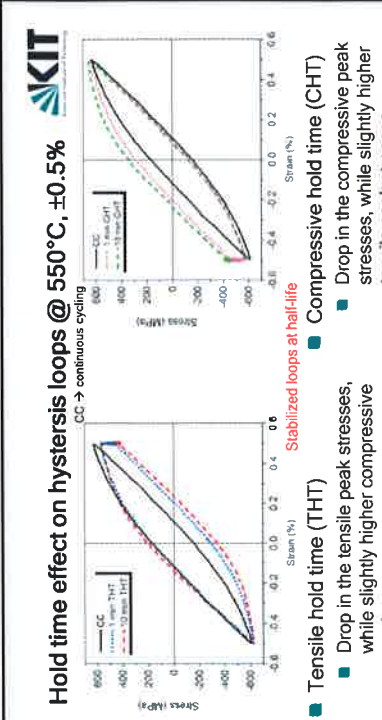


650°C, ± 0.5%  
 580°C  
 Tensile peak stress (MPa)  
 Number of cycles, N (-)

- Four distinct stages
  - Very brief rapid cyclic softening
  - Gradual cyclic hardening
  - Prolonged cyclic saturation
  - Crack initiation and growth stage

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### Hold time effect on hysteresis loops @ 550°C, ±0.5%



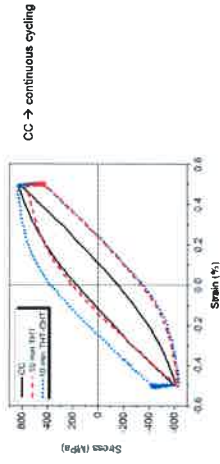
CC → continuous cycling  
 Stabilized loops at half-life  
 The drop is observed independently for tensile and compressive hold time

- Tensile hold time (THT)
  - Drop in the tensile peak stresses, while slightly higher compressive peak stresses
- Compressive hold time (CHT)
  - Drop in the compressive peak stresses, while slightly higher tensile peak stresses

Negligible effect on peak stresses  
 Hold time → ↑ Inelastic strain accumulated

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### Hold time effect on hysteresis loops @ 550°C, ±0.5%



- Tensile hold time (THT) while slightly higher compressive peak stresses
- Tensile-compressive hold time (THT-CHT) No effect on peak stresses
- Higher inelastic strain accumulated

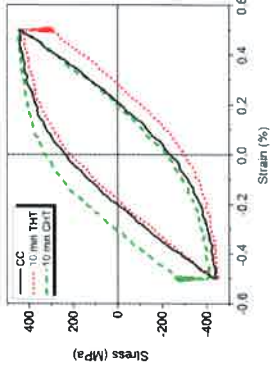
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### Hold time effect on hysteresis loops @ 650°C, ±0.5%

Similar observations were noticed



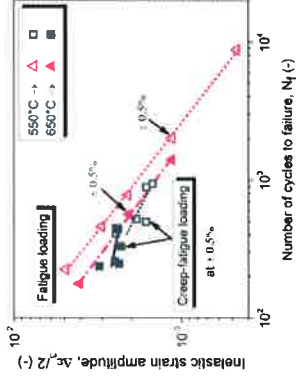
- The drop is observed independently for tensile and compressive hold time

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### Inelastic strain vs. Lifetime



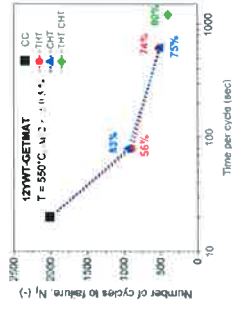
- Creep reduces the fatigue lifetime due to increase in the inelastic strain

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### Hold time type effect on lifetime




- Tensile and compressive hold time result in a similar cycle life
- Lowest cycle life is observed for the hold time in both tensile and compressive direction (THT-CHT)

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**Microstructure evolution**  
550 °C, ± 0.5 %, 10 min THT




500 nm

HAADF image

- Large grains containing high dislocation density and in-process LAGBs formation
- Planar slip character

**@ 550 °C**

550 °C, ± 0.5 %, 10 min THT



TEM BF image

500 nm

550 °C, ± 0.5 %, 10 min THT

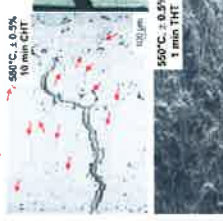
- Decrease in the dislocation density
- Large elongated grains are fractionated into smaller subgrains
- Small grains are stable and free of dislocations
- Pinning prevents subgrains development
- Both planar and wavy slip characters

**@ 650 °C**

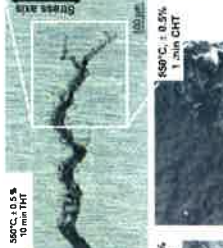
**Oxide particles homogeneity is critical**

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**Crack initiation and propagation @ Creep-fatigue tests**  
Longitudinal section



550 °C, ± 0.5 %, 10 min THT



Stress axis  
550 °C, ± 0.5 %, 10 min THT



650 °C, ± 0.5 %, 1 min THT

- High density of surface cracks
- @ 550 °C → Transgranular stage I crack initiation
- @ 650 °C → Transgranular stage II crack initiation → higher oxidation near surface
- Propagation is of mixed mode type → transgranular & intergranular

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**Summary**

- Creep-fatigue interaction in a bimodal 12Cr ferritic ODS steel is investigated and presented
- Tensile hold time results in a drop in the tensile peak stresses, while slightly higher compressive peak stress and vice-versa
- Hold time duration has minor effect on peak stresses
- Hold time in both direction causes no visible effect on the peak stresses
- Microstructure shows partial stability due to in-homogeneous distribution of oxide particles
- Transgranular stage I crack initiation at 550 °C
- Transgranular stage II crack initiation at 650 °C
- Mixed mode crack propagation → transgranular & intergranular

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**Thank you for your attention**

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