

Constitution, Microstructure and Mechanical Properties of Magnetron-sputtered TiC/a-C Nanocomposite Coatings

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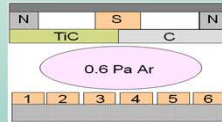
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Motivation

Carbon-based nanostructured composite coatings are attracting steadily increasing attention due to their promising properties. These materials often consist of nanocrystalline phases, clusters or particles homogeneously dispersed in an amorphous carbon matrix or covered by an amorphous carbon grain boundary phase. Advanced nanocomposite films providing simultaneously wear protection and low friction are required in engineering applications. We report on the synthesis and correlation between constitution, microstructure and properties of non-reactively magnetron-sputtered TiC/a-C nanocomposite coatings.

Deposition

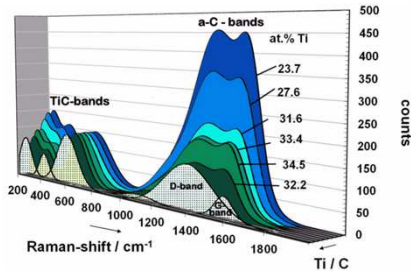
rf magnetron sputtering from a segmented TiC / Graphite target in an Ar atmosphere (0.6 Pa)
 1 power supply, Power density 11.3 W/cm²
 target-substrate distance 60 mm
 Substrates: cemented carbide (Pos. 1 - Pos. 6).
 rf substrate bias voltage: 0, -150 V or -300 V
 substrate temperature < 250°C



Analyses of mechanical properties, microstructure and constitution

EPMA: Cameca Camebax microbeam system
 XRD: Seifert PAD II using Cu K_α radiation (λ = 0.15418 nm)
 Micro Raman: Renishaw Raman System 1000 spectrometer, Ar ion laser, 514.5 nm, ≤ 1 mW
 AFM: Digital Instruments, Dimension 3100, Si tip (Nanosensors LFM-20; tip radius < 10 nm) in contact mode
 Nanoindentation: CSIRO Umis 2000, load 20 mN, 50 steps.

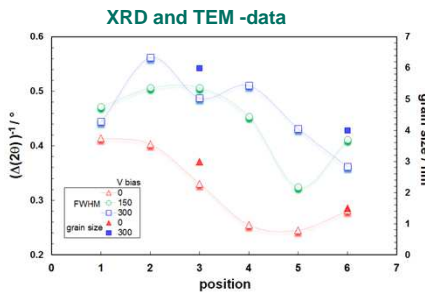
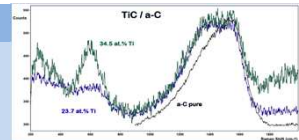
Raman spectroscopy of TiC/a-C coatings



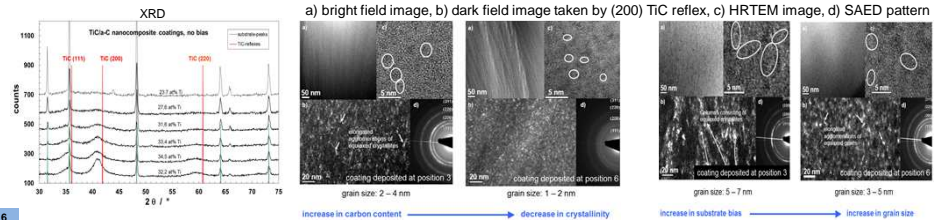
0 V substrate bias	Pos. 1	Pos. 2	Pos. 3	Pos. 4	Pos. 5	Pos. 6	a-C
Position G-Peak / cm ⁻¹	1569	1565	1565	1568	1565	1567	1567
I _D /I _G	2.16	2.22	2.00	2.09	2.12	1.99	1.31
FWHM G-Peak / cm ⁻¹	115	116	132	122	129	126	172
area ratio (TiC-bands / a-C-bands)	0.99	0.73	0.78	0.76	0.53	0.44	

- 150 V substrate bias	Pos. 1	Pos. 2	Pos. 3	Pos. 4	Pos. 5	Pos. 6	changes with increasing carbon content of the TiC/a-C-coatings	
Position G-Peak / cm ⁻¹	1580	1574	1570	1571	1572	1568	G-peak position unchanged → vibration frequency is related to the average C-C bond length → no change in a-C structure	
I _D /I _G	1.66	1.71	1.88	1.80	1.94	2.05	G-FWHM increases → FWHM is uniquely related to the sp ² clustering → decrease in crystallinity	
FWHM G-Peak / cm ⁻¹	110	118	117	121	121	125	I _D /I _G ratio → decreasing with progressive disordering of the sp ² -bonded rings and increase of sp ³ sites → bias influence	
area ratio (TiC-bands / a-C-bands)	0.47	0.48	0.47	0.44	0.35	0.28	area ratio (TiC-bands / a-C-bands) → directly related to Ti/C ratio → linear decrease with rising C content → decreasing TiC grain size	

- 300 V substrate bias	Pos. 1	Pos. 2	Pos. 3	Pos. 4	Pos. 5	Pos. 6	bias influence	
Position G-Peak / cm ⁻¹	1578	1575	1566	1566	1573	1569	G-peak position: rising for -150 V, greater variance for -300 V → higher vibration frequency → shortening of the average bondlength	
I _D /I _G	1.69	2.25	2.22	1.76	2.19	2.02	G-FWHM: comparable for no bias or -150 V, rising for -300 V → broader spectra → less sp ² -clustering	
FWHM G-Peak / cm ⁻¹	125	128	143	147	141	145	I _D /I _G ratio: no change for 0 V, greater variance for -300 V, for -150 V increase with rising C content → more sp ² clustering, ring structures	
area ratio (TiC-bands / a-C-bands)	0.80	0.54	0.68	0.57	0.43	0.43	area ratio: for comparable TiC: lower for -150 V, max. for 0 V → TiC-crystallite size increasing with bias, grain size from TEM: Pos. 3: 2-4 nm for 0 V, 5-7 nm for -300 V, Pos. 6: 1-2 nm for 0 V, 3-5 nm for -300 V	



Microstructure of TiC/a-C coatings deposited without bias / at -300 V bias

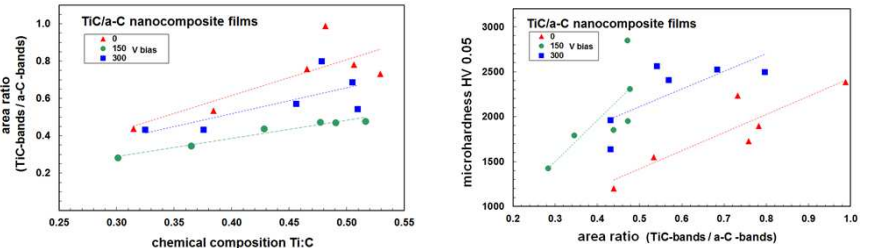


Composition and properties of TiC/a-C coatings

deposited at 0 V substrate bias	Pos. 1	Pos. 2	Pos. 3	Pos. 4	Pos. 5	Pos. 6
Ti content / at.-%	32.2	34.5	33.4	31.6	27.6	23.7
C content / at.-%	66.9	65.1	65.9	68.0	71.9	75.4
Ti:C ratio	0.48	0.53	0.51	0.47	0.38	0.31
microhardness HV0.05	2384	2235	1899	1730	1548	1204
nanohardness / GPa	26.1	24.6	22.0	20.3	19.0	16.2
red. Young's modulus / GPa	380	266	221	230	232	200
friction coefficient vs. 100Cr6	0.25	0.21	0.20	0.17	0.13	0.11
Roughness R _a / nm	174.5	170.2	99.6	87.5	80.6	114.0

deposited at -150 V substrate bias	Pos. 1	Pos. 2	Pos. 3	Pos. 4	Pos. 5	Pos. 6
Ti content / at.-%	32.4	33.3	31.4	29.2	25.9	22.4
C content / at.-%	66.2	64.5	65.9	68.1	71.1	74.3
Ti:C ratio	0.49	0.52	0.48	0.43	0.36	0.30
microhardness HV0.05	2844	2307	1950	1852	1791	1421
nanohardness / GPa	28.4	24.9	23.4	22.4	21.8	21.3
red. Young's modulus / GPa	385.9	360.6	295.6	234.4	252.3	246.6
friction coefficient vs. 100Cr6	0.22	0.22	1.23	0.18	0.16	0.13
Roughness R _a / nm	281.2	221.5	295.3	261.4	488.6	197.1

deposited at -300 V substrate bias	Pos. 1	Pos. 2	Pos. 3	Pos. 4	Pos. 5	Pos. 6
Ti content / at.-%	31.5	32.8	32.7	30.5	26.5	23.8
C content / at.-%	65.8	64.3	64.7	66.8	70.5	73.2
Ti:C ratio	0.48	0.51	0.51	0.46	0.38	0.32
microhardness HV0.05	2494	2560	2521	2402	1958	1634
nanohardness / GPa	32.5	27.4	28.6	26.8	24.1	23.9
red. Young's modulus / GPa	405	344	348	329	282	270
friction coefficient vs. 100Cr6	0.22	0.21	0.23	0.23	0.17	0.16
Roughness R _a / nm	456.2	341.9	308.9	330.1	385.0	216.5



Summary

From the results of the combination of Raman spectroscopy, XRD, TEM, AFM and nano-indentation it is concluded that at least 3 effects play together in the formation of nanocomposite TiC/a-C films:

- formation of both the crystalline TiC and amorphous carbon phase
- densification of the amorphous carbon network by the ion bombardment
- preferential resputtering of the C atoms.

The most interesting growth effects are observed at -150 V bias voltage at the center positions 3 and 4. In this case the structure-property correlation is influenced by the interplay of all three effects, whereas in the outer sample positions and for the other bias voltages, always one of the three effects is dominating the formation of the TiC/a-C nanocomposite structure.