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Heteroepitaxial Beta-Ga₂O₃ on 4H-SiC for an FET With Reduced Self Heating

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ABSTRACT A method to improve thermal management of β -Ga₂O₃ FETs is demonstrated here via simulation of epitaxial growth on a 4H-SiC substrate. Using a recently published device as a model, the reduction achieved in self-heating allows the device to be driven at higher gate voltages and increases the overall performance. For the same operating parameters an 18% increase in peak drain current and 15% reduction in lattice temperature are observed. Device dimensions may be substantially reduced without detriment to performance and normally off operation may be achieved.

INDEX TERMS FET, gallium oxide, molecular beam epitaxy, normally-off, self-heating, silicon carbide, threshold voltage.

I. INTRODUCTION

Wide band-gap semiconductor materials commonly thought of for use in power electronic devices include silicon carbide (SiC) [1], [2], gallium nitride (GaN) [3] and diamond. SiC has shown the greatest potential to date in replacing silicon for DC power applications, indeed 4H-SiC diodes and MOSFETs are today commercially available. Diamond has been touted as the ideal material for high power electronics due to its large thermal conductivity, high breakdown field and high bulk carrier mobility. It has the potential to compete with both SiC for high voltage DC applications and GaN for RF power applications [4]–[6]. However despite recent advances, doping and substrate cost remains an issue [7], [8].

In contrast, beta-Gallium Oxide (β -Ga₂O₃) could offer an alternative to 4H-SiC for power applications. Its key intrinsic material properties are competitive with diamond, e.g., a band-gap of 4.8 eV and high breakdown field 8 MVcm⁻¹ [9], [10]. Large scale growth is also far less challenging and possible at a fraction of the cost [11].

Despite the material structure being first investigated in the 1960s [12] it is only much more recently this material has

been suggested for application in power electronic devices. Indeed, Ga₂O₃ thin-films have been successfully employed for a variety of applications including a window layer for solar cells and a transparent conductive oxide [13]–[15]. This holds promise for a larger degree of integration compared with traditional wide band-gap semiconductors, e.g., into display screens or transparent electronics [16].

In recent years basic FET devices have been demonstrated [17]–[20], although p-type doping (which is notoriously difficult in wide band-gap semiconductors) is still a practical challenge. The best oxide solution is yet to be settled upon with work to date mainly focusing on Al₂O₃, although in theory SiO₂ provides a greater band off-set and hence reduced gate leakage [21], [22].

In this paper a SILVACO Atlas non-isothermal model for β -Ga₂O₃ demonstrated in a recent paper is employed to recreate a device demonstrated by Higashiwaki *et al.* [18], [23]. The low thermal conductivity of β -Ga₂O₃ (~ 0.2 Wcm⁻¹K⁻¹) [10] can be regarded as perhaps the most severe drawback for very high voltage applications. We demonstrate via simulation the advantages

of engineering a thin film of β -Ga₂O₃ onto a good thermal conductor such as SiC ($\sim 5 \text{ Wcm}^{-1}\text{K}^{-1}$) to increase the thermal performance of β -Ga₂O₃ FETs.

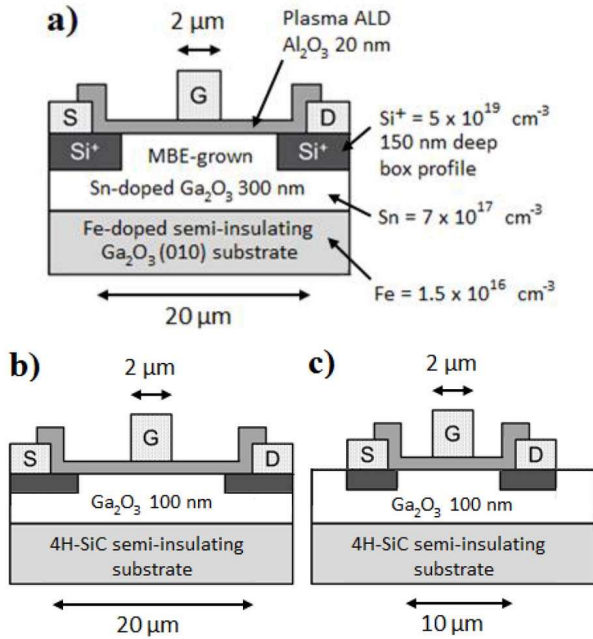


FIGURE 1. Schematic showing the structure and doping of a) original MOSFET device based on β -Ga₂O₃ [18] and b) & c) adaptations made in this paper.

II. SIMULATION

Fig. 1 shows a cross-section of the FET modelled here. The semi-insulating β -Ga₂O₃ substrate material is detailed in the original paper as having a doping of $1.5 \times 10^{16} \text{ cm}^{-3}$, while the dopant in the Molecular Beam Epitaxy (MBE) layer is $7 \times 10^{17} \text{ cm}^{-3}$ with half considered activated. The Si ohmic dopant is $5 \times 10^{19} \text{ cm}^{-3}$ with $3 \times 10^{19} \text{ cm}^{-3}$ activated. A 20 nm Al₂O₃ Atomic Layer Deposited (ALD) oxide layer covers the device with a source-drain spacing of 20 μm and gate length 2 μm . This device (Fig. 1a) suffers from self-heating effects due to the relatively poor thermal conductivity of β -Ga₂O₃ which reduces the saturated drain current value. One way to mitigate this is to use a substrate with a high thermal conductivity such as 4H-SiC. A recent study has shown the positive impact this can have even on Si based devices [24]. The lattice match of β -Ga₂O₃ to 4H-SiC is close at 3.04\AA compared to 3.07\AA respectively, allowing for low defect density materials to be grown and interfaces to be formed between β -Ga₂O₃ and 4H-SiC by growth methods such as MBE used in this instance.

A simple constant thermal conductivity and low-field mobility model was used for this study with main parameters summarised in Table 1.

As many β -Ga₂O₃ parameters are not fully established yet care is needed to not pick unrealistic values. Electron effective mass is taken to be $0.28 m_0$ giving a calculated local conduction band density of states $N_c = 3.72 \times 10^{18} \text{ cm}^{-3}$,

TABLE 1. Simulation parameters.

Material	Parameter	Value	
β -Ga ₂ O ₃	Bandgap Energy	4.8 eV	
	Thermal Conductivity	$0.13 \text{ Wcm}^{-1}\text{K}^{-1}$	
	Effective Mass	$0.28 m_0$	
	Local Conduction Band Density of States	$3.72 \times 10^{18} \text{ cm}^{-3}$	
	Epitaxial Layer Mobility	$118 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$	
	Epitaxial Layer Dopant Concentration	$7 \times 10^{17} \text{ cm}^{-3}$ ($3.5 \times 10^{17} \text{ cm}^{-3}$ activated)	
	Substrate Mobility	$20 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$	
	Substrate Dopant Concentration	$1.5 \times 10^{16} \text{ cm}^{-3}$	
	4H-SiC	Bandgap Energy	3.23 eV
		Thermal Conductivity	$3.7 \text{ Wcm}^{-1}\text{K}^{-1}$
Effective Mass		$0.41 m_0$	
Local Conduction Band Density of States		$5 \times 10^{18} \text{ cm}^{-3}$	
Substrate Mobility		$460 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$	
Substrate Dopant Concentration		$1.5 \times 10^{16} \text{ cm}^{-3}$	

β -Ga₂O₃ parameters taken from [12 & 25] whereas standard SILVACO parameters are used in the case of the 4H-SiC substrate

mobility of the channel layer and mobility of the semi-insulating substrate are set to be 118 and $20 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$ respectively [25].

A parameter (LAT.TEMP) was added to the model to account for poor heat flow in β -Ga₂O₃ material. The lattice temperature coefficient for the temperature dependence of electron mobility $\text{TMUN}=2.0$ was used as seen in Equation (1).

$$\mu_{n0} = \text{MUN} \left(\frac{T_L}{300} \right)^{-\text{TMUN}} \quad (1)$$

where μ_{n0} represents electron mobility adjusted for lattice temperature, MUN the originally input mobility value, T_L lattice temperature and TMUN the temperature dependence coefficient.

The material parameters of 4H-SiC are much better established and standard SILVACO parameters were used for this material.

III. RESULTS & DISCUSSION

Fig. 2 demonstrates the immediate benefit of switching to a 4H-SiC substrate for this technology, output characteristics are displayed for the original device as well as the modelled version on a 4H-SiC substrate simulated to have the same doping level ($1.5 \times 10^{16} \text{ cm}^{-3}$) as the original β -Ga₂O₃ substrate. Peak drain current increases for all gate bias points, an increase of 19% is seen for $V_{\text{gs}} = +8\text{V}$ with peak lattice temperature reducing by 15% at this bias point, far extending the realm of operation of the original device.

A visual comparison is shown in the form of a heat contour map for the original device and the 4H-SiC substrate version in Fig. 3. For the β -Ga₂O₃ device at a V_{gs} of +8V and V_{ds} of +40V the lattice temperature reaches a peak of 166°C ,

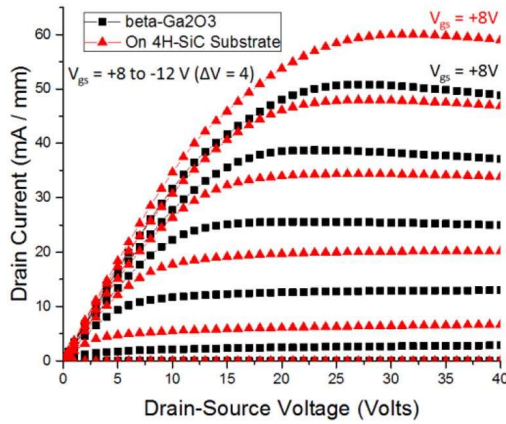


FIGURE 2. Comparison of the output characteristics from a β -Ga₂O₃ MOSFET and the same on a 4H-SiC substrate, both are simulated to have a 300 nm thick epitaxial layer.

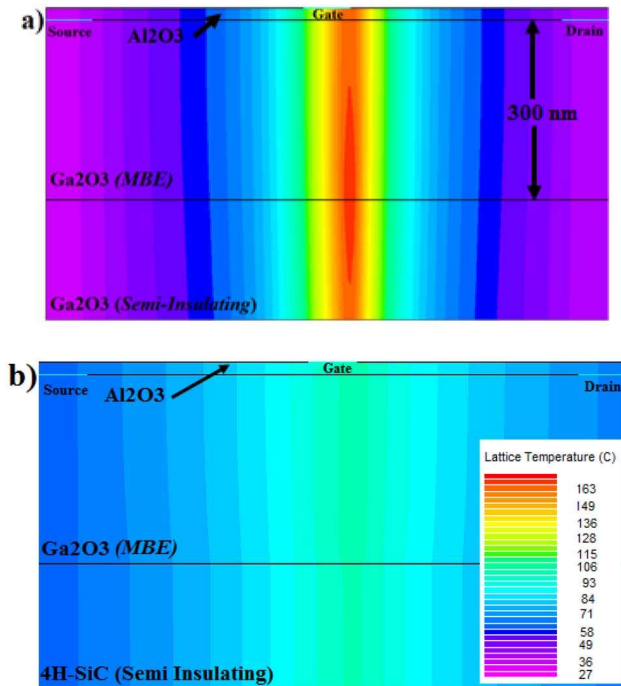


FIGURE 3. Lattice temperature at $V_{gs} = 8V$ for β -Ga₂O₃ (a) and 4H-SiC (b) substrate.

for the same bias on a 4H-SiC substrate the peak lattice temperature reaches only 98° C.

The 4H-SiC version remains normally-on although this can be shifted in both cases by a reduction in thickness of the epilayer. Fig. 4. Shows the shift in threshold voltage (V_T) for modelled devices as the simulated epitaxial layer thickness is reduced, there is also an associated reduction in temperature.

Fig. 5 shows the output characteristic of a 100 nm epitaxial layer FET on 4H-SiC device compared to the original β -Ga₂O₃ FET. Self-heating is virtually eliminated at this gate voltage (V_{gs}) of +8 V (lattice temperature = 53° C). Peak drain current is understandably reduced due to the more

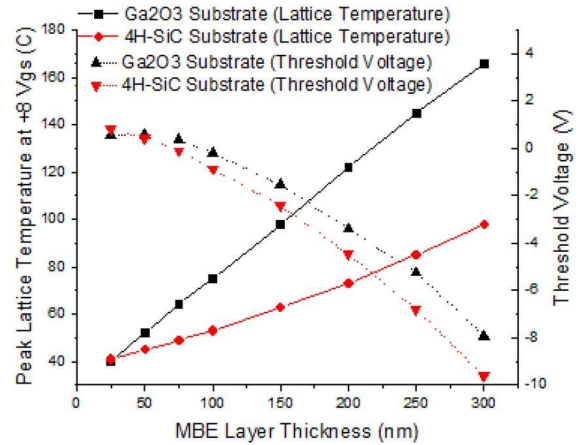


FIGURE 4. Comparison of transfer characteristics for β -Ga₂O₃ MOSFET and the same on a 4H-SiC substrate.

restricted and hence resistive route but the nature of this design and its improved thermal conductivity allows a much higher V_{gs} to be used.

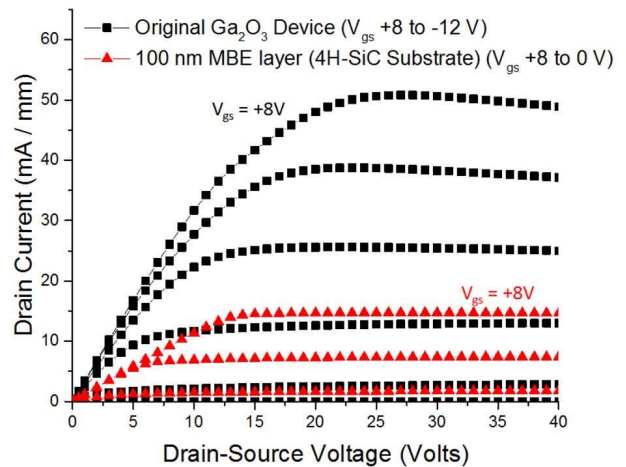


FIGURE 5. Output characteristic for β -Ga₂O₃ substrate with 100 nm layer thickness.

Fig. 6 shows how this device may be driven easily to $V_{gs} = 20$ V. A similar peak drain current to the original device is achieved although on-resistance of the device is increased. If however the source-drain gap is reduced to 10 μ m a similar output characteristic to the original device may be seen with a substantially reduced V_T and a lower lattice temperature (91° C at $V_{GS} = +8$ V).

Analyzing current density for the β -Ga₂O₃ FET at voltages close to V_T the current path is partially forced through the semi-insulating substrate, albeit at a much reduced level due to the lower mobility of this layer. Reducing the MBE layer thickness ensures this occurs at a more positive V_{gs} and impacts upon the threshold voltage. On the 4H-SiC substrate the heterojunction present means negligible charge will migrate in to this layer giving rise to the marginal difference in V_T between substrates. This is visualized in Fig. 7.

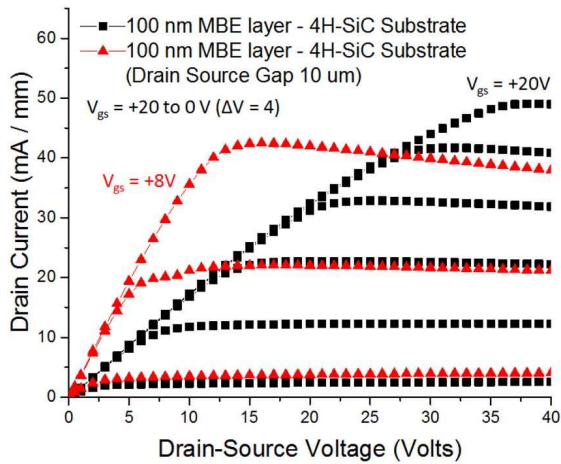


FIGURE 6. Output characteristic for 4H-SiC substrate with 100 nm layer thickness, V_{gs} up to +20 (lattice temp max 71° C).

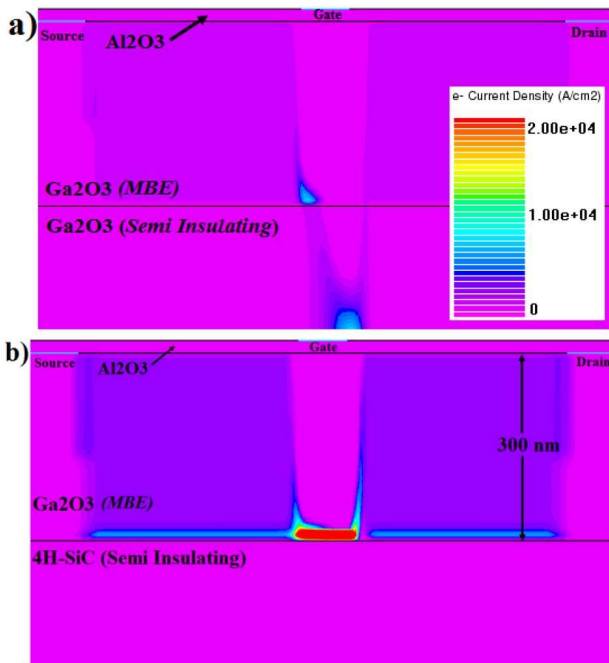


FIGURE 7. Current route at V_{gs} = -8V for β-Ga₂O₃ (a) and 4H-SiC (b) substrate.

Below a thickness of 100 nm the device becomes normally-off (below 75 nm for the 4H-SiC substrate). As the MBE layer thickness approaches 100 nm the shift in V_T plateaus and reducing beyond this point has little benefit and begins to severely limit current performance. With this reduced layer thickness less drain current is possible for the same gate voltage as the restricted route increases the resistance of the β-Ga₂O₃ MBE layer.

A source-drain distance of 20 microns is relatively large even for a power FET and utilising the 4H-SiC substrate allows further device scaling to increase current performance. Halving this to 10 microns and biasing at +8 V_{gs} a similar current level to the original device is obtained, still

with a lower peak lattice temperature of 91° C. This scaling will of course put further stress on the dielectric layer. Simulation shows a maximum electric field of 1.22 MVcm⁻¹ even at +20 V_{gs} on the gate dielectric for the 100 nm on 4H-SiC device with reduced source-drain dimension. This is well within the remit of either Al₂O₃ or SiO₂ as a gate dielectric.

This demonstration allows many further routes to be taken in terms of scaling, the suitable lattice match of these two materials make it the perfect complement. Improvements in processing of this technology will undoubtedly yield more improvements from this promising material. It is accepted that 4H-SiC substrates are expensive however recent technological innovations such as those offered by Siltectra GmbH mean substrates may be split and recycled many times over [26]. The ability to grow high quality large area substrates of β-Ga₂O₃ also raises the possibility of wafer bonding to high thermal conductivity substrates.

IV. CONCLUSION

The replacement of the semi-insulating β-Ga₂O₃ substrate with a 4H-SiC alternative in this device yields two improvements. Firstly the reduction in self-heating due to the order of magnitude thermal conductivity of 4H-SiC compared to β-Ga₂O₃. Secondly due to this it is possible to scale the device in reducing the MBE layer thickness and reducing the source-drain gap, giving overall the same current performance with a drastically reduced V_T.

Future work should focus on the best gate oxide for this technology. Characterization of interface traps will shed some light on this matter and the best route to practically scaling these FETs. Attention also needs to be paid to ohmic metallization stacks as even in 4H-SiC devices stability of these at elevated temperature can still be an issue [27].

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From 2013 to 2014, he was a Post-Doctoral Research Assistant with the University of Glasgow researching diamond electronics with a focus on field-effect transistors for high-performance applications and has authored several papers on this topic. In 2014, he was a Research Fellow with the University of Warwick, where he was involved in high-voltage blocking (10 kV) silicon carbide MOSFETs.

Dr. Russell has recently been appointed as a Senior Research Fellow with the University of Warwick to continue work on this and other power semiconductor devices. More recently he has taken an interest in Gallium Oxide as a new material for power electronics. In 2017, he was awarded a feasibility study from the Advanced Propulsion Centre, U.K., to investigate on the above area.



AMADOR PÉREZ-TOMÁS received the Ph.D. degree in physics from the Autonomous University of Barcelona, Barcelona, Spain, in 2005.

From 1999 to 2001, he was with Hewlett-Packard. From 2006 to 2008, he was a Research Fellow with the University of Warwick, Coventry, U.K. From 2008 to 2014, he was with the Consejo Superior de Investigaciones Científicas Ramón y Cajal. In 2014, he joined the Institut Català de Nanociència i Nanotecnologia. He has authored or co-authored over 150 journal papers and conferences.

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His research interests include surfaces, interfaces, epitaxial growth of semiconducting materials, conventional, and novel oxide electronic materials. His research cluster uses MBE, PLD, and CVD growth facilities and also uses a range of surface and bulk sensitive techniques (including XRD, TEM, XPS, STM, and AFM) to extract physical and electronic structure information. He has pioneered surface studies of III–V's, III-nitrides, and several semiconducting oxide materials.

Mr. McConville was the recipient of an Erskine Fellowship in 2010, the Institute of Physics and British Vacuum Council's Senior Prize, and the John Yarwood Memorial Medal in 2007. He has raised over £11M in grant income; delivered over 120 invited lectures and seminars at national and international conferences, workshops, and research institutions world-wide; published over 250 papers in refereed journals along with three book chapters. He is the Co-Editor of the first book on III-nitride materials entitled *Indium Nitride and Related Alloys* (2009).



CRAIG A. FISHER was born in Warwickshire, U.K., in 1984. He received the M.Sc. degree in advanced electronics engineering and the Ph.D. degree in silicon carbide power electronics with a focus on three principle areas: novel edge termination solutions for high voltage power devices, carrier lifetime enhancement using high temperature processes, and the formation of robust ohmic contacts to p-type 4H-SiC, from the University of Warwick, Coventry, U.K., in 2010 and 2014, respectively.

He then spent 12 months as a Research Fellow with the School of Engineering, University of Warwick, where his research interests included the design, fabrication, and characterization of 4H-SiC MOSFETs and diodes for high voltage (>3.3 kV) applications. Since 2015, he has been involved in the power electronics industry, developing silicon carbide power devices and systems. He has authored (or co-authored) over 40 journal/international conference papers, and has acted as a Reviewer for the IEEE TRANSACTIONS ON ELECTRON DEVICES and the IET.



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