

# SCIENTIFIC REPORTS

OPEN

## High-pressure Raman scattering in bulk $\text{HfS}_2$ : comparison of density functional theory methods in layered $\text{MS}_2$ compounds ( $M = \text{Hf}, \text{Mo}$ ) under compression

J. Ibáñez<sup>1</sup>, T. Woźniak<sup>2</sup>, F. Dybala<sup>2</sup>, R. Oliva<sup>2</sup>, S. Hernández<sup>3</sup> & R. Kudrawiec<sup>2</sup>

We report high-pressure Raman-scattering measurements on the transition-metal dichalcogenide (TMDC) compound  $\text{HfS}_2$ . The aim of this work is twofold: (i) to investigate the high-pressure behavior of the zone-center optical phonon modes of  $\text{HfS}_2$  and experimentally determine the linear pressure coefficients and mode Grüneisen parameters of this material; (ii) to test the validity of different density functional theory (DFT) approaches in order to predict the lattice-dynamical properties of  $\text{HfS}_2$  under pressure. For this purpose, the experimental results are compared with the results of DFT calculations performed with different functionals, with and without Van der Waals (vdW) interaction. We find that DFT calculations within the generalized gradient approximation (GGA) properly describe the high-pressure lattice dynamics of  $\text{HfS}_2$  when vdW interactions are taken into account. In contrast, we show that DFT within the local density approximation (LDA), which is widely used to predict structural and vibrational properties at ambient conditions in 2D compounds, fails to reproduce the behavior of  $\text{HfS}_2$  under compression. Similar conclusions are reached in the case of  $\text{MoS}_2$ . This suggests that large errors may be introduced if the compressibility and Grüneisen parameters of bulk TMDCs are calculated with bare DFT-LDA. Therefore, the validity of different approaches to calculate the structural and vibrational properties of bulk and few-layered vdW materials under compression should be carefully assessed.

Transition metal dichalcogenides (TMDCs) have attracted a great deal of attention during the last few years due to their remarkable properties and great potential for photonics and optoelectronics applications<sup>1</sup>. TMDCs are van der Waals layered semiconductors of the type  $\text{MX}_2$ , where M stands for a transition metal atom such as Mo, W, or Hf, while X is a group-16 atom like S, Se, or Te. Due to their semiconducting character, TMDC monolayers emerge as an alternative to graphene, which does not normally exhibit an electronic bandgap. The optoelectronic properties (optical gaps and spin-orbit splittings) of TMDCs can indeed be tailored through strategically selecting their composition, i.e., alloying with different chalcogen or transition metal atoms<sup>2</sup>.

Despite a great deal of research has been devoted to the study of Mo and W-based TMDCs, relatively little attention has been paid to Hf-containing layered compounds. The latter are semiconductor materials with band gaps in the visible or infrared spectral range (the band gap decreases with increasing the chalcogen atomic number)<sup>3</sup>, which makes them attractive for a wide variety of optoelectronic devices like third-generation solar cells<sup>4</sup>. Recent advances on Hf-based TMDCs include the investigation of their promising thermoelectric properties<sup>5–7</sup>, their epitaxial growth on graphite or  $\text{MoS}_2$  substrates<sup>8</sup>, or the realization of different devices such as tunnel field-effect transistors (TFETs)<sup>9</sup> or few-layer transistors<sup>10</sup>.

Among  $\text{HfX}_2$  compounds, hafnium disulphide ( $\text{HfS}_2$ ) has been found to exhibit particularly high and fast electrical responses<sup>9,11</sup>. It has been calculated that the mobility of this compound might reach values as high as  $1800 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ,

<sup>1</sup>Institute of Earth Sciences Jaume Almera, Consell Superior d'Investigacions Científiques (CSIC), Lluís Solé i Sabarís s.n., 08028, Barcelona, Catalonia, Spain. <sup>2</sup>Faculty of Fundamental Problems of Technology, Wrocław University of Science and Technology, Wybrzeże Wyspiańskiego 27, 50-370, Wrocław, Poland. <sup>3</sup>Departament d'Electrònica-MIND2-UB, Universitat de Barcelona, Martí i Franquès 1, 08028, Barcelona, Catalonia, Spain. Correspondence and requests for materials should be addressed to R.O. (email: robert.oliva.vidal@pwr.edu.pl)

much higher than the value of  $400 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$  of  $\text{MoS}_2$ <sup>11</sup>. Also, it has been found that the sheet current density of tunneling field-effect transistors (TFETs) based on  $\text{HfS}_2$  is two orders of magnitude larger than that of  $\text{MoS}_2$ <sup>12</sup>. It is thus necessary to further investigate the fundamental properties of  $\text{HfS}_2$  and explore its possible use in future device applications.

Raman spectroscopy is a powerful analytical tool which provides, in a non-destructive manner, valuable information about several important aspects of semiconductor materials and structures, such as their lattice dynamics, crystal quality, strain state, composition, or electronic structure. In the case of layered compounds such as graphene or TMDCs, Raman scattering is being widely used to determine flake thicknesses of few-layer systems and also to probe the strain, stoichiometry, or polytypism and stacking order of the TMDC layers<sup>13,14</sup>. However, in order to fully understand the vibrational properties of monolayer and few-layered TMDCs, it is essential to thoroughly investigate the lattice dynamics of the bulk materials.

High-pressure experiments are commonly employed in semiconductor physics studies in order to further probe the optical, electronic and vibrational properties of semiconductors. In particular, high-pressure techniques provide a highly useful benchmark to test the results of existing theoretical models, like for instance density functional theory (DFT), for the calculation of the electronic and lattice-dynamical properties of semiconductors. This may be critical in the case of TMDCs, whose properties are strongly affected by the weak van der Waals bonds between layers. In the case of Raman-scattering investigations, the pressure behavior of the optical phonons provides useful information about phase stability and anharmonic effects. In combination with theoretical calculations, it may also help one to assign the features that show up in the Raman spectra.

First-principle calculations have recently shown that, contrary to bulk  $\text{HfS}_2$ , its monolayer form exhibits a phononic gap which can be tailored by applying biaxial strain<sup>15</sup>. So far, however, the high-pressure effects on the vibrational properties of bulk  $\text{HfS}_2$  have been investigated neither theoretically nor experimentally. With regard to ambient pressure conditions, several works dealing with the resonant and non-resonant Raman spectrum of bulk 1T- $\text{HfS}_2$  have been published in the literature<sup>16–18</sup>. All the previous studies report the first-order  $A_{1g}$  and  $E_g$  Raman-active modes, but there is some discrepancy with regard to a weak feature that appears at  $\sim 325 \text{ cm}^{-1}$  as a low-frequency shoulder of the intense  $A_{1g}$  peak. While some authors<sup>16,18</sup> assign this feature to a Raman-forbidden  $E_u(\text{LO})$  mode, Roubi and co-workers<sup>17</sup> attribute it to a forbidden  $A_{2u}(\text{LO})$  mode. It should be noted, however, that half the frequency of this mode ( $\sim 162 \text{ cm}^{-1}$ ) is very close to the frequency of the zone-center  $E_u(\text{TO})$  measured by Lucovsky and co-workers ( $166 \text{ cm}^{-1}$ ) with infrared (IR) reflectance spectroscopy<sup>19</sup>. High-pressure measurements, in combination with lattice-dynamical calculations, could help to clarify the origin of all the Raman features and better assign these modes.

In the present work we report on Raman-scattering measurements under high hydrostatic pressures on bulk  $\text{HfS}_2$ . From the pressure dependence of the Raman-active modes we obtain the corresponding mode Grüneisen parameters. The experimental pressure coefficients are compared with theoretical values obtained with lattice-dynamical ab initio calculations performed with density functional theory (DFT), using different functionals and methodologies. We find that the best agreement between theory and experiment is found with DFT-GGA calculations including Van der Waals interaction. In contrast, we show that bare DFT-LDA, which is widely used to calculate the properties of 2D (bulk and few-layer) compounds at ambient conditions, does not correctly predict the behavior of bulk  $\text{HfS}_2$  upon compression. Similar conclusions are reached for the case of 2H- $\text{MoS}_2$ .

## Results and Discussion

Three different types of calculations were performed with ABINIT in order to find the best agreement between theoretical and experimental lattice constants of  $\text{HfS}_2$  at ambient pressure: (i) calculations within the generalized gradient approximation (GGA), using fully-relativistic projector augmented-wave (PAW) potentials with Perdew-Burke-Ernzerhof (PBE) exchange-correlation functionals<sup>20</sup>; (ii) PAW-PBE calculation including Grimme's D3 dispersion correction<sup>21</sup> to take into account long-range van der Waals (vdW) interactions; (iii) calculations within the Perdew-Wang local density approximation (LDA); DFT-LDA usually provides successful structural results in layered compounds because of a compensating effect between the overestimated covalent part of the interlayer bonding and the neglected vdW forces<sup>14</sup>.

At ambient pressure, we obtained the following lattice constants for GGA calculations with (and without) vdW corrections:  $a = 3.616$  (3.650) Å and  $c = 5.801$  (7.021) Å. While the bare GGA-PBE calculations clearly overestimate the  $c$  parameter, the vdW-corrected GGA results are in very good agreement with the experimental values ( $a = 3.630$  Å and 3.622,  $c = 5.854$  and 5.88 Å)<sup>19,22</sup>, thus reflecting the importance of vdW interactions in the structural properties of these layered materials. On the other hand and, as expected, the LDA calculations provide somewhat underestimated values ( $a = 3.556$  Å and  $c = 5.677$  Å), although in reasonable agreement with the experimental data.

Table 1 shows a summary of these results, together with bulk modulus ( $B_0$ ) values obtained after structural relaxation at different pressures and subsequent fitting of the resulting volume-pressure data, using a 3<sup>rd</sup> order Birch-Murnaghan equation of state. The Table also shows data for an additional calculation performed with the Quantum Espresso package using PBEsol functionals, i.e., the revised version of GGA-PBE for the solid state<sup>23</sup>. As can be seen in the table, a large compressibility (low bulk modulus) is obtained with the bare GGA calculations for  $\text{HfS}_2$ , which is clearly a consequence of the overestimated  $c$  parameter at zero pressure. In turn, the PBEsol approach provides significantly improved results for the  $c$  parameter. However, a relatively large compressibility is still obtained with this functional. In contrast, similar bulk modulus values are found with GGA-vdW and LDA, although the latter yields a somewhat lower compressibility. This result may be attributed to the fact that the overestimation of the interlayer covalent bonding in LDA calculations becomes larger upon compression. Unfortunately, no X-ray diffraction measurements as a function of pressure have been so far reported for  $\text{HfS}_2$ .

	<i>a</i> (Å)	<i>c</i> (Å)	<i>B</i> <sub>0</sub> (GPa)	<i>B</i> '
LDA	3.556	5.677	36.8	8.3
PBE	3.650	7.021	8.1	8.3
PBE + vdW	3.616	5.801	30.6	8.3
PBEsol	3.588	5.943	19.4	12.7
Experiment	3.630 <sup>a</sup> , 3.622 <sup>b</sup>	5.854 <sup>a</sup> , 5.88 <sup>b</sup>	—	—

**Table 1.** Calculated lattice parameters at room pressure, bulk modulus and its pressure derivative for bulk HfS<sub>2</sub> as obtained with different DFT functionals. Experimental values for the room-pressure lattice parameters are also given. <sup>a</sup>Ref.<sup>19</sup>, <sup>b</sup>ref.<sup>22</sup>.

Symmetry	Method	$\omega_{i0}$ (cm <sup>-1</sup> )	$a_{i0}$ (cm <sup>-1</sup> GPa <sup>-1</sup> )	$\gamma_i$
<i>A</i> <sub>1g</sub>	FD (LDA)	332.9	4.88	0.45
	FD (PBE)	316.5	4.63	0.45
	FD (PBE + vdW)	330.7	4.63	0.43
	DFPT (LDA)	345.7	4.82	0.43
	DFPT (PBE)	321.3	4.42	0.42
	DFPT (PBEsol)	324.6	4.97	0.47
	Experiment <sup>a</sup>	340.2	4.71	0.42
<i>E</i> <sub>g</sub>	FD (LDA)	266.8	1.34	0.15
	FD (PBE)	243.7	2.87	0.36
	FD (PBE + vdW)	250.9	2.37	0.29
	DFPT (LDA)	270.2	1.80	0.20
	DFPT (PBE)	253.3	1.93	0.23
	DFPT (PBEsol)	259.1	1.79	0.21
	Experiment <sup>a</sup>	260.2	2.33	0.27

**Table 2.** Theoretical and experimental Raman frequencies ( $\omega_{i0}$ ) and their pressure coefficients ( $a_{i0}$ ) and mode Grüneisen parameters ( $\gamma_i$ ) for the Raman-active (gerade) modes of HfS<sub>2</sub>. Theoretical values were obtained with the finite displacement (FD) method or within Density Functional Perturbation Theory (DFPT), using different functionals. <sup>a</sup>This work.

With regard to the vibrational properties, group theoretical analysis of the zone-center phonons of HfS<sub>2</sub> with the ideal 1*T*-structure (*D*<sub>3d</sub> point group, with 3 atoms in the primitive cell) shows that 9 vibrational modes are present in the total representation:  $\Gamma = A_{1g} + E_g + 2A_{2u} + 2E_u$ , where the odd modes (*u*, ungerade) are infrared-active while the even modes (*g*, gerade) are Raman-active. Infrared and Raman activities are not compatible because the crystal structure has a center of inversion symmetry (and one of the modes for each of the *u*-symmetries correspond to acoustic modes). The atomic displacements for the optical modes can be found in previous works<sup>15,17,19</sup>.

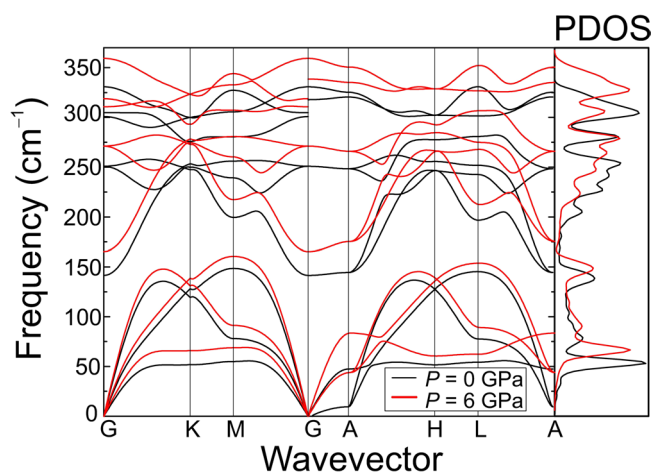
The zero-pressure frequency values ( $\omega_{i0}$ ), phonon pressure coefficients ( $a_i$ ) and mode Grüneisen parameters ( $\gamma_i$ ) obtained in this work with the Finite Displacement (FD) and Density Functional Perturbation Theory (DFPT) methods are shown in Tables 2 and 3. Table 2 shows the results for the gerade (*g*) modes of HfS<sub>2</sub> (see below), while Table 3 shows the data for the TO and LO ungerade (*u*) phonons. The Grüneisen parameters were obtained using the expression  $\gamma_i = B_0 a_i / \omega_{i0}$ , where, for the sake of comparison between different DFT methodologies, a constant value of  $B_0 = 30.6$  GPa as obtained from DFT calculations within GGA + vdW has been employed. These data will be later discussed in reference to the experimental results.

Figure 1 shows the calculated phonon dispersion and one-phonon density of states (1-PDOS) over the whole Brillouin zone as obtained with Phonopy for HfS<sub>2</sub> at 0 and 6 GPa, using the finite displacement (FD) method and a PBE functional including the vdW correction. The figure includes splitting between LO and TO modes, which was not taken into account in previous works<sup>15</sup>. It should be recalled, however, that DFT calculations tend to yield large errors in the case of LO frequencies as a consequence of the band-gap problem and the computation of the dielectric constants. This is particularly important in the case of DFT-LDA, although it is not substantially improved within GGA<sup>24</sup>.

The corresponding 1-PDOS curves are displayed in the right panel of the figure. As expected, all phonon branches show an overall upward frequency shift upon compression, without any softening of the modes. The results at zero pressure are consistent with those obtained by DFPT within PBE-GGA as reported in ref.<sup>5</sup>. As can be seen in the figure, no phononic gap exists in bulk HfS<sub>2</sub>, which suggests that reduced optical phonon lifetimes may be expected in this compound for all the first-order modes due to an increased number of pathways for phonon decay. Interestingly, many of the phonon branches, like for instance the acoustic phonons along  $\Gamma$ -K-M- $\Gamma$  or the optical phonons along the  $\Gamma$ -A direction, exhibit very low dispersion and therefore increased two-phonon density of states. As a consequence of this, second-order Raman scattering processes may be favored in HfS<sub>2</sub>.

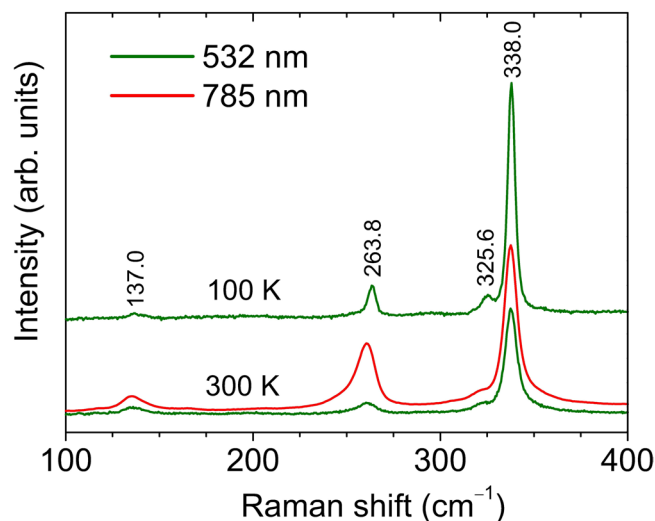
Symmetry	Method	$\omega_{i0}$ (cm <sup>-1</sup> )	$a_{i0}$ (cm <sup>-1</sup> GPa <sup>-1</sup> )	$\gamma_i$
A <sub>2u</sub> (TO)	FD (LDA)	306.8	2.86	0.29
	FD (PBE)	302.8	2.27	0.23
	FD (PBE + vdW)	300.9	2.93	0.30
	DFPT (LDA)	324.3	2.82	0.27
	DFPT (PBE)	302.4	2.46	0.25
	DFPT (PBEsol)	306.9	2.71	0.27
	Experiment	—	—	—
A <sub>2u</sub> (LO)	FD (LDA)	324.4	3.05	0.29
	FD (PBE)	314.4	2.94	0.29
	FD (PBE + vdW)	317.8	3.25	0.31
	DFPT (LDA)	341.4	3.00	0.27
	DFPT (PBE)	318.5	2.80	0.27
	DFPT (PBEsol)	323.5	3.02	0.29
	Experiment <sup>a</sup>	321.1	3.58	0.34
E <sub>u</sub> (TO)	FD (LDA)	151.7	1.94	0.39
	FD (PBE)	137.8	3.15	0.70
	FD (PBE + vdW)	141.5	3.06	0.66
	DFPT (LDA)	173.8	1.75	0.31
	DFPT (PBE)	148.1	2.44	0.50
	DFPT (PBEsol)	152.2	2.23	0.45
	Experiment <sup>b</sup>	166	—	—
E <sub>u</sub> (LO)	FD (LDA)	295.6	0.44	0.05
	FD (PBE)	290	1.5	0.16
	FD (PBE + vdW)	304.8	0.71	0.07
	DFPT (LDA)	308.4	0.30	0.03
	DFPT (PBE)	301.1	0.57	0.06
	DFPT (PBEsol)	300.5	0.29	0.03
	Experiment <sup>b</sup>	318	—	—

**Table 3.** Theoretical and experimental Raman frequencies ( $\omega_{i0}$ ) and their pressure coefficients ( $a_{i0}$ ) and mode Grüneisen parameters ( $\gamma_i$ ) for the Raman-inactive (ungerade) modes of HfS<sub>2</sub>. Theoretical values were obtained with the finite displacement (FD) method or within Density Functional Perturbation Theory (DFPT), using different functionals. <sup>a</sup>This work, <sup>b</sup>ref.<sup>19</sup>.



**Figure 1.** Ab initio calculation of the phonon band structure of bulk HfS<sub>2</sub>, including TO-LO splitting effects, along the main lines of symmetry at 0 and 6 GPa. The right panel shows the corresponding phonon density of states (PDOS).

In order to evaluate the validity of different functionals and methodologies based on DFT calculations, Raman measurements were performed at different pressures and temperatures. Figure 2 shows ambient-pressure Raman spectra at room temperature and low temperature (100 K) of HfS<sub>2</sub>, acquired with 532-nm excitation radiation. An additional room-temperature spectrum excited with  $\lambda = 785$  nm has also been included in the



**Figure 2.** Room-pressure Raman spectra, acquired at room temperature and low temperature (100 K), excited with  $\lambda = 532$  nm. For comparison, a Raman spectrum excited at room conditions with  $\lambda = 785$  nm is also shown.

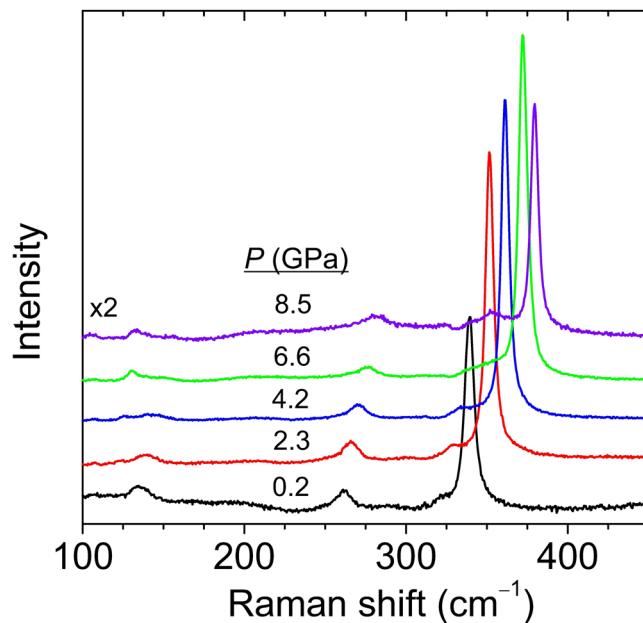
figure for comparison purposes. Four different modes around 137, 264, 326 and 338  $\text{cm}^{-1}$  clearly show up in the low-temperature spectrum. The Raman peaks at  $\sim 338$  and  $264$   $\text{cm}^{-1}$  can be unambiguously assigned to the  $A_{1g}$  and  $E_g$  modes, respectively<sup>16–18</sup>. It is interesting to note that these modes only involve S vibrations, and as a consequence they are not affected by isotopic anharmonic effects due to the strong isotopic variability of natural hafnium. In contrast, strong isotopic broadening may be expected for the  $u$ -phonons, since these do involve Hf vibrations.

As expected, all the features in Fig. 2 are broadened and shifted to lower frequencies in the room-temperature spectra. In particular, the mode at  $\sim 325$   $\text{cm}^{-1}$  only appears as a broad low-frequency shoulder of the  $A_{1g}$  mode. In turn, it is worth noting that the  $E_g$  mode at  $\sim 264$   $\text{cm}^{-1}$  is significantly enhanced, relatively to the rest of peaks, in the spectrum excited with  $\lambda = 785$  nm. As in other layered compounds<sup>25–27</sup>, this intensity enhancement may be attributed to resonance effects arising from exciton-phonon coupling involving particular exciton states of  $\text{HfS}_2$ . Given that atoms in the  $E_g$  phonon vibrate in-plane, the observed intensity increase in Fig. 2 may be ascribed to electronic transitions involving excitonic states with strong in-plane character. This is what occurs, for instance, is the case of the C exciton in  $\text{MoS}_2$ <sup>25</sup>. However, given that the combination modes in TMDCs exhibit strong resonance enhancement due to coupling between phonons of nonzero momentum and excitonic transitions<sup>27,28</sup>, it cannot be completely ruled out that the strong peak that appears at  $\sim 264$   $\text{cm}^{-1}$  for 785-nm excitation has contribution from a second-order Raman band. More work is required to fully understand excitonic resonance Raman effects in  $\text{HfS}_2$ .

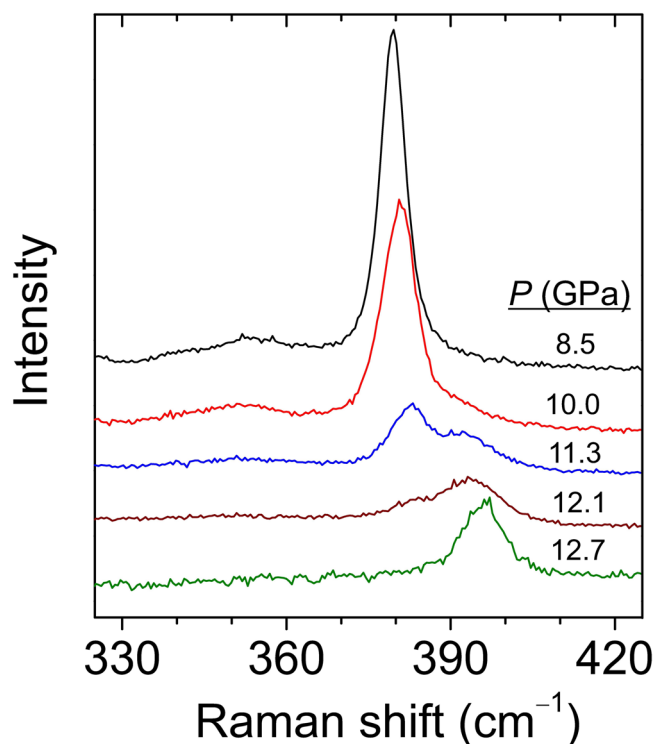
In spite of the Raman inactivity of  $u$ -modes, two different works<sup>16,18</sup> attribute the weak peak below the  $A_{1g}$  mode around 326  $\text{cm}^{-1}$  to an  $E_u(\text{LO})$  mode. In contrast, ref.<sup>17</sup> assigns this feature to an  $A_{2u}(\text{LO})$  mode. In that work, a weak, broad band around 155  $\text{cm}^{-1}$  is also attributed to an  $E_u(\text{TO})$  mode. However, such feature is not reported in any other previous work, whereas the  $E_u(\text{TO})$  phonon was found at 166  $\text{cm}^{-1}$  by means of infrared-reflectance spectroscopy<sup>19</sup>. Note that the peak at 326  $\text{cm}^{-1}$  could also correspond to a  $2E_u(\text{TO})$  mode, since this feature is not far from twice the frequency of the  $E_u(\text{TO})$  mode reported by Lucovsky and co-workers.

Leaving aside the previous works on  $\text{HfS}_2$ , several studies on layered compounds like  $\text{MoS}_2$ ,  $\text{MoSe}_2$  or  $\text{WS}_2$  have also reported Raman inactive  $u$ -modes<sup>27–29</sup>. In those works, ungerade modes were observed amid a large number of second-order modes excited resonantly, and were attributed to symmetry breaking of the selection rules through disorder or the participation strongly localized electronic states. Other works on TMDCs reported bulk-inactive  $g$ -modes in few-layer compounds like  $\text{MoTe}_2$ <sup>30</sup>, which was attributed to a loss of periodicity along the  $c$ -axis. In the case of  $\text{HfS}_2$ , it cannot be ruled out that the  $u$ -modes show up in the Raman spectra due to a contribution of few-layer domains in the investigated samples. Regardless of their origin, high-pressure Raman-scattering measurements, in combination with DFT lattice-dynamical calculations, may provide highly valuable information to identify the different features that appear in the Raman spectra of vdW compounds.

Figure 3 shows selected Raman spectra at different pressures, up to 8.5 GPa, for bulk  $\text{HfS}_2$ . As expected, all modes are found to blueshift with increasing pressure. Around 4 GPa, a very weak, broad feature is found to show up around 120  $\text{cm}^{-1}$ . This mode has never been reported in previous works and probably emerges due to a particular resonant excitation at this pressure range (note that, as can be seen in Fig. 1, 1T- $\text{HfS}_2$  has no first-order optical modes below  $\sim 150$   $\text{cm}^{-1}$ ). As already mentioned, second-order resonant Raman scattering in TMDCs may yield a large number of bands at some given excitation conditions. In this case, resonance enhancement may be achieved by pressure-induced bandgap shifts. In contrast, the mode at 137  $\text{cm}^{-1}$  progressively weakens upon compression and is hardly visible above 6 GPa. Around 10 GPa, as shown in Fig. 4, a new peak appears above the  $A_{1g}$  mode. This peak becomes dominant at 12.7 GPa and can be tentatively attributed to a high-pressure phase of  $\text{HfS}_2$ . The study of this possible new phase is beyond the scope of the present work.

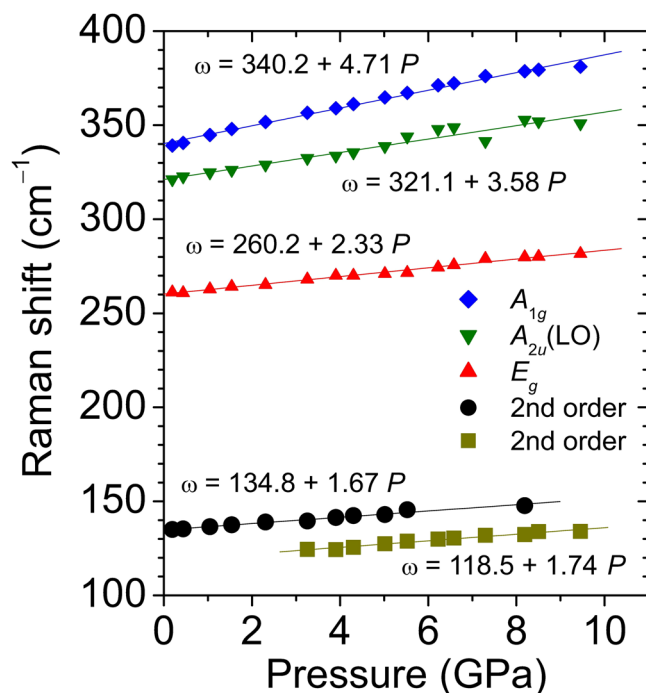


**Figure 3.** Raman spectra of bulk  $\text{HfS}_2$  acquired at different hydrostatic pressure values up to 8.5 GPa.



**Figure 4.** Raman spectra of bulk  $\text{HfS}_2$  acquired at different hydrostatic pressure values, up to 12.7 GPa, showing the onset of a possible first-order phase transition in this compound at around 11 GPa.

We have plotted in Fig. 5 the pressure dependence of all the observed Raman features up to 10 GPa, below the phase transition that presumably occurs in bulk  $\text{HfS}_2$ . The results of linear fits to all the experimental curves as a function of pressure ( $P$ ) are also shown in the figure. These were obtained by using the expression  $\omega_i(P) = \omega_i(0) + a_i P$ , where  $\omega_i(0)$  and  $a_i$  stand for the zero-pressure frequency and pressure-coefficient for the  $i$ -th mode, respectively. The experimental data are summarized in Tables 2 and 3, together with theoretical values obtained within DFT-based lattice-dynamical calculations of the zone-center phonons performed with the FD and DFPT methods. The corresponding mode Grüneisen parameters ( $\gamma_i = B_0 a_i / \omega_{i0}$ , with  $B_0 = 30.6$  GPa as obtained with the present DFT-GGA calculations including vdW interactions) are also given.



**Figure 5.** Pressure dependence of first-order optical phonons and second-order bands that appear in the Raman spectrum of bulk HfS<sub>2</sub>. The lines show the result of linear fits to the experimental data.

In the case of the  $A_{1g}$  mode, excellent agreement is found between the experimental pressure coefficients and most of the approaches considered in this work. For this phonon mode, with atoms vibrating out-of-plane, all methods provide similar results because interlayer forces have a low effect on the resulting frequencies of vibration. As expected, among the DFPT values (note that vdW corrections are not yet implemented for DFPT phonon calculations in Quantum Espresso), LDA provides the best zero-pressure frequency value. In the case of FD calculations, LDA and GGA + vdW provide comparable results. The bare GGA value, however, yields a too low value as a consequence of the overestimated  $c$  parameter at zero pressure.

Similar conclusions can be reached for the IR-active  $A_{2u}$ (TO) mode, although in this case the bare GGA calculations tend to yield slightly reduced pressure coefficients. In turn, DFPT-LDA predicts a zero-pressure frequency that is ~6–7% larger than the rest of values. Unfortunately, no experimental results are available to perform a comparison between theory and experiment for this mode.

Interestingly, when the zone-center modes involve in-plane atomic displacements, as is the case of  $E_g$  and  $E_u$  modes, the LDA calculations predict fairly low pressure coefficients, below  $2\text{ cm}^{-1}/\text{GPa}$ , both with the FD and DFPT methods. In contrast, the GGA and GGA-vdW results predict much larger values. In particular, the GGA-vdW pressure coefficient obtained with the FD method for the  $E_g$  mode ( $2.37\text{ cm}^{-1}/\text{GPa}$ ) is in remarkable agreement with the experimental value ( $2.33\text{ cm}^{-1}/\text{GPa}$ ). This result indicates that the GGA-vdW calculations are better suited to model the high-pressure lattice-dynamical properties of 1T-HfS<sub>2</sub>.

Such a conclusion is not limited to the case of HfS<sub>2</sub>, but can be extended to other layered compounds. To further test this point, we have performed additional lattice-dynamical calculations for the case of the archetypical TMDC compound 2H-MoS<sub>2</sub> (see the results in Supplementary Table 1). In this case we also find that, although the LDA functional predicts remarkable zero-pressure Raman frequencies, the theoretical pressure coefficients for the  $E$ -symmetry modes, including the low-frequency shear mode, are significantly lower than those obtained with PBE functionals (with and without vdW corrections). Bearing in mind the relatively large dispersion of the experimental data reported in the literature (Supplementary Table 1), only the GGA-vdW calculations provide good agreement between theoretical and experimental values for both the zero-pressure Raman frequencies and their corresponding phonon pressure coefficients in bulk MoS<sub>2</sub>. Similar conclusions were also reached elsewhere<sup>31</sup> for the case of black phosphorous. In that work, although no direct comparison was shown regarding different methods to calculate the lattice dynamics of this layered material as a function of pressure, it was found that only the PBE-vdW calculations are able to simultaneously predict the structural and vibrational properties of black phosphorous under compression.

On the other hand, and as shown in Table 3, the theoretical pressure coefficients for the  $E_u$ (LO) mode of bulk HfS<sub>2</sub> are found to be very small, much lower than  $1\text{ cm}^{-1}/\text{GPa}$ . This result, together with the theoretical  $\omega_{i0}$  values found for this mode, allow us to discard the conclusions of refs<sup>16,18</sup> with regard to the origin of the peak at  $\sim 320\text{ cm}^{-1}$  (Fig. 2). In those works, this feature was attributed to an  $E_u$ (LO) mode. In contrast, the experimental pressure coefficient (zero-pressure frequency) for this band is  $3.58\text{ cm}^{-1}/\text{GPa}$  ( $321.1\text{ cm}^{-1}$ ), much larger than the values predicted by the different calculations for the  $E_u$ (LO) mode, e. g.  $0.71\text{ cm}^{-1}/\text{GPa}$  ( $304.8\text{ cm}^{-1}$ ) with GGA + vdW and the FD method. Note also that the  $E_u$ (LO) mode should involve in-plane vibrations and,

following the behavior of the  $E_g$  mode, resonant enhancement of this mode would have been expected for 785-nm excitation. In contrast, the theoretical values for the out-of-plane  $A_{2u}(\text{LO})$  [ $3.25 \text{ cm}^{-1}/\text{GPa}$  ( $317.8 \text{ cm}^{-1}$ ) with GGA + vdW and the FD method] are compatible with the experimental values. Note also that no combinations of first-order modes at different points of the Brillouin zone, using the phonon dispersion of Fig. 1, provide a satisfactory explanation for this feature. In particular, it cannot be attributed to a  $2E_u(\text{TO})$  mode, since all the present DFT calculations predict a much larger pressure coefficient for any second-order combination of optical modes. Therefore, given the good agreement between the experimental and theoretical (GGA + vdW) values, we conclude that the low-frequency shoulder of the  $A_{1g}$  mode can be assigned to the silent  $A_{2u}(\text{LO})$  phonon of  $\text{HfS}_2$ . More work would be necessary, however, to fully understand the actual mechanism giving rise to silent (symmetry-forbidden) modes in  $\text{HfS}_2$  and, in general, in layered TMDCs.

Finally, we use the theoretical phonon dispersion as a function of pressure (Fig. 1) in combination with the experimental pressure coefficients in order to tentatively assign the remaining Raman feature at  $137 \text{ cm}^{-1}$  as well as the weak band that shows up above 3 GPa (Fig. 2). First, we note that the experimental pressure coefficient of the latter, and also that of the feature at  $137 \text{ cm}^{-1}$  is just around  $1.7 \text{ cm}^{-1}/\text{GPa}$ , much smaller than the value predicted for the  $E_u(\text{TO})$  mode (Table 3). Hence, in contrast to the conclusions of ref.<sup>17</sup>, it can be ruled out that forbidden first-order Raman scattering by the  $E_u(\text{TO})$  mode occurs in  $\text{HfS}_2$ . Indeed, given that numerous phonon branches display low dispersion along relative large regions of the Brillouin zone (see Fig. 1), and bearing in mind the low experimental pressure coefficients of the observed bands, it is very likely that these two modes correspond to second-order difference modes. For instance, combinations like  $E_g(\text{L})$ - $\text{LA}(\text{L})$  or  $A_{2u}(\text{TO},\text{L})$ - $E_u(\text{TO},\text{L})$  might explain the pressure behavior of the peak at  $137 \text{ cm}^{-1}$ , while the band that shows up above 3 GPa at  $118 \text{ cm}^{-1}$  can be explained with a  $E_u(\text{TO},\text{M})$ - $\text{LA}(\text{M})$  or a  $E_u(\text{TO},\text{L})$ - $\text{LA}(\text{L})$  combination.

## Conclusion

We have presented a joint theoretical and experimental high-pressure Raman-scattering study of the vibrational properties of bulk 1T- $\text{HfS}_2$ . Comparison between the experimental phonon pressure coefficients and the results of DFT calculations with different functionals, with and without vdW interactions, indicate that GGA + vdW properly describes the pressure behavior of bulk  $\text{HfS}_2$ . In contrast, it is found that bare DFT-LDA fails to reproduce the structural and vibrational properties of this compound under compression. Similar conclusions can be reached in the case of 2H- $\text{MoS}_2$ . DFT-LDA is widely used for calculations of phonons and other properties in layered compounds due to a well-known compensating effect between the underestimated bond lengths and the overestimated interatomic forces. However, the present results suggest that DFT-LDA is not valid to predict the compressibility or the mode Grüneisen parameters of bulk or few-layered vdW materials. In particular, we have found that DFT-LDA gives rise to a sizable underestimation of phonons involving in-plane atomic displacements. From the pressure dependence of several Raman features and the theoretical data, it has been possible to shed additional light on the origin of different features that appear in the Raman spectrum of  $\text{HfS}_2$ . In particular, the low-frequency shoulder below  $A_{1g}$  peak has been assigned to a forbidden  $A_{2u}(\text{LO})$  mode. More work is required in order to understand the scattering mechanism responsible for the appearance of Raman-inactive ungerade modes in the Raman spectrum of TMDCs.

## Methods

**Samples.** For the present study, we used a commercial  $\text{HfS}_2$  bulk sample from 2DSemiconductors Inc. grown by the Bridgman technique. The indirect and direct optical band-gap of the sample, as measured by optical measurements at ambient conditions, are 1.39 and 2.09 eV, respectively<sup>3</sup>.

**High pressure measurements.** Small flakes of around  $50 \times 50 \mu\text{m}^2$  were detached from the bulk  $\text{HfS}_2$  sample and loaded into a gasketed membrane-type diamond anvil cell (DAC) with  $400 \mu\text{m}$  culet-size diamonds. A mixture of methanol-ethanol-water (16:3:1) was employed as pressure transmitting medium, and the ruby fluorescence method was used to evaluate the pressure applied to the flake. Room-temperature micro-Raman measurements were acquired during the upstroke cycle, up to 13 GPa, by using a HORIBA Jobin-Yvon LabRam-HR spectrometer coupled to a high-sensitive CCD camera. The spectra were excited with the second harmonic of a continuous-wave Nd:YAG laser ( $\lambda = 532 \text{ nm}$ ), and a  $\times 50$  long-working distance objective was employed to focus the laser light and to collect the backscattered radiation. To further probe resonance effects in  $\text{HfS}_2$  at room temperature, additional Raman spectra excited with  $\lambda = 785 \text{ nm}$  radiation were also acquired.

**First-principle calculations.** DFT calculations for bulk 1T- $\text{HfS}_2$  and 2H- $\text{MoS}_2$  were carried out with the ABINIT<sup>32</sup> and Quantum Espresso<sup>33</sup> packages. Details for the case of  $\text{MoS}_2$  can be found in the Supplementary Material. For all calculations on  $\text{HfS}_2$ , the plane wave basis cut-off was 30 Ha and the Monkhorst-Pack  $k$ -point grid was set to  $8 \times 8 \times 4$ . The structures were first fully relaxed until the interatomic forces were lower than  $10^{-5} \text{ eV}/\text{\AA}$ . Lattice-dynamical calculations were performed with the Phonopy software<sup>34</sup> for different pressure values, up to 10 GPa, within the FD method.  $2 \times 2 \times 2$  supercells were found to give convergence for all the phonon modes of both compounds. From these calculations, zone-center phonon frequencies for the Raman-active  $A_{1g}$  and  $E_{1g}$  modes and for the infrared-active  $A_{2u}$  and  $E_u$  modes of  $\text{HfS}_2$  were obtained. For the construction of the corresponding dynamical matrices, the forces associated to the selected finite displacements were calculated with ABINIT using the GGA, GGA-vdW and LDA methods. In order to calculate the splitting between transverse optical (TO) and longitudinal optical (LO) phonon modes, dielectric tensors and Born effective charges were calculated as a function of pressure with Quantum Espresso, using the DFPT approach implemented in this package. For this purpose, LDA and PBEsol functionals were used. For comparison purposes, zero-pressure frequencies and pressure coefficients for the zone-center modes including TO-LO splitting were also calculated within DFPT.



## References

- Manzeli, S., Ovchinnikov, D., Pasquier, D., Yazyev, O. V. & Kis, A. 2D transition metal dichalcogenides. *Nature Reviews Materials* **2**, 17033 (2017).
- Wang, Z. *et al.* Strong interface-induced spin-orbit interaction in graphene on WS<sub>2</sub>. *Nature Communications* **6**, 8339 (2015).
- Zelewski, S. J. & Kudrawiec, R. Photoacoustic and modulated reflectance studies of indirect and direct band gap in van der Waals crystals. *Scientific Reports* **7**, 15365 (2017).
- Jiang, H. Structural and electronic properties of ZrX<sub>2</sub> and HfX<sub>2</sub> (X = S and Se) from first principles calculations. *The Journal of Chemical Physics* **134**, 204705 (2011).
- Yumnam, G., Pandey, T. & Singh, A. K. High temperature thermoelectric properties of Zr and Hf based transition metal dichalcogenides: A first principles study. *The Journal of Chemical Physics* **143**, 234704 (2015).
- Lv, H. Y. *et al.* Enhancing the thermoelectric performance of a HfS<sub>2</sub> monolayer through valley engineering. *arXiv:1608.05464* [cond-mat] (2016).
- Ding, G., Gao, G. Y., Huang, Z., Zhang, W. & Yao, K. Thermoelectric properties of monolayer M Se 2 (M = Zr, Hf): low lattice thermal conductivity and a promising figure of merit. *Nanotechnology* **27**, 375703 (2016).
- Yue, R. *et al.* HfSe<sub>2</sub> Thin Films: 2D Transition Metal Dichalcogenides Grown by Molecular Beam Epitaxy. *ACS Nano* **9**, 474–480 (2015).
- Gong, C. *et al.* Band alignment of two-dimensional transition metal dichalcogenides: Application in tunnel field effect transistors. *Appl. Phys. Lett.* **103**, 053513 (2013).
- Kanazawa, T. *et al.* Few-layer HfS<sub>2</sub> transistors. *Scientific Reports* **6**, 22277 (2016).
- Xu, K. *et al.* Ultrasensitive Phototransistors Based on Few-Layered HfS<sub>2</sub>. *Adv. Mater.* **27**, 7881–7887 (2015).
- Fiori, G. *et al.* Electronics based on two-dimensional materials. *Nat Nano* **9**, 768–779 (2014).
- Zhang, X., Tan, Q.-H., Wu, J.-B., Shi, W. & Tan, P.-H. Review on the Raman spectroscopy of different types of layered materials. *Nanoscale* **8**, 6435–6450 (2016).
- Molina-Sánchez, A., Hummer, K. & Wirtz, L. Vibrational and optical properties of MoS<sub>2</sub>: From monolayer to bulk. *Surface Science Reports* **70**, 554–586 (2015).
- Chen, J. Phonons in bulk and monolayer HfS<sub>2</sub> and possibility of phonon-mediated superconductivity: A first-principles study. *Solid State Communications* **237**, 14–18 (2016).
- Iwasaki, T., Kuroda, N. & Nishina, Y. Anisotropy of Lattice Dynamical Properties in ZrS<sub>2</sub> and HfS<sub>2</sub>. *J. Phys. Soc. Jpn.* **51**, 2233–2240 (1982).
- Roubi, L. & Carlone, C. Resonance Raman spectrum of HfS<sub>2</sub> and ZrS<sub>2</sub>. *Phys. Rev. B* **37**, 6808–6812 (1988).
- Cingolani, A., Lugarà, M. & Lévy, F. Resonance Raman scattering in HfSe<sub>2</sub> and HfS<sub>2</sub>. *Phys. Scr.* **37**, 389 (1988).
- Lucovsky, G., White, R. M., Benda, J. A. & Revelli, J. F. Infrared-Reflectance Spectra of Layered Group-IV and Group-VI Transition-Metal Dichalcogenides. *Phys. Rev. B* **7**, 3859–3870 (1973).
- Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* **78**, 1396–1396 (1997).
- Grimme, S., Antony, J., Ehrlich, S. & Krieg, H. A consistent and accurate ab initio parametrization of density functional dispersion correction (DFT-D) for the 94 elements H-Pu. *The Journal of Chemical Physics* **132**, 154104 (2010).
- Greenaway, D. L. & Nitsche, R. Preparation and optical properties of group IV–VI chalcogenides having the CdI<sub>2</sub> structure. *Journal of Physics and Chemistry of Solids* **26**, 1445–1458 (1965).
- Perdew, J. P. *et al.* Restoring the Density-Gradient Expansion for Exchange in Solids and Surfaces. *Phys. Rev. Lett.* **100**, 136406 (2008).
- Yakovkin, I. N. & Dowben, P. A. The problem of the band gap in lda calculations. *Surf. Rev. Lett.* **14**, 481–487 (2007).
- Carvalho, B. R., Malard, L. M., Alves, J. M., Fantini, C. & Pimenta, M. A. Symmetry-Dependent Exciton-Phonon Coupling in 2D and Bulk MoS<sub>2</sub> Observed by Resonance Raman Scattering. *Phys. Rev. Lett.* **114**, 136403 (2015).
- Bhatt, S. V., Deshpande, M. P., Sathe, V., Rao, R. & Chaki, S. H. Raman spectroscopic investigations on transition-metal dichalcogenides MX<sub>2</sub> (M = Mo, W; X = S, Se) at high pressures and low temperature. *J. Raman Spectrosc.* **45**, 971–979 (2014).
- Nam, D., Lee, J.-U. & Cheong, H. Excitation energy dependent Raman spectrum of MoSe<sub>2</sub>. *Scientific Reports* **5**, 17113 (2015).
- Lee, J.-U., Park, J., Son, Y.-W. & Cheong, H. Anomalous excitonic resonance Raman effects in few-layered MoS<sub>2</sub>. *Nanoscale* **7**, 3229–3236 (2015).
- Staiger, M. *et al.* Excitonic resonances in WS<sub>2</sub> nanotubes. *Phys. Rev. B* **86**, 165423 (2012).
- Yamamoto, M. *et al.* Strong Enhancement of Raman Scattering from a Bulk-Inactive Vibrational Mode in Few-Layer MoTe<sub>2</sub>. *ACS Nano* **8**, 3895–3903 (2014).
- Appalakondaiah, S., Vaitheeswaran, G., Lebegue, S., Christensen, N. E. & Svane, A. Effect of van der Waals interactions on the structural and elastic properties of black phosphorus. *Physical Review B* **86**, 035105 (2012).
- Gonze, X. *et al.* First-principles computation of material properties: the ABINIT software project. *Computational Materials Science* **25**, 478–492 (2002).
- Giannozzi, P. *et al.* QUANTUM ESPRESSO: a modular and open-source software project for quantum simulations of materials. *J. Phys.: Condens. Matter* **21**, 395502 (2009).
- Togo, A. & Tanaka, I. First principles phonon calculations in materials science. *Scripta Materialia* **108**, 1–5 (2015).

## Acknowledgements

Work supported by the Spanish Government through projects MAT2015-71035-R and FIS2017-83295-P, and by the National Science Centre (NCN) Poland POLONEZ 3 no. 2016/23/P/ST3/04278 and grant OPUS 11 no. 2016/21/B/ST3/00482. This project has received funding from the European Union's Horizon 2020 research and innovation programme under the Marie Skłodowska-Curie grant agreement No 665778. T.W. acknowledges support within the Maestro grant from NCN (no. 2014/14/A/ST3/0065). F. D. acknowledges the support within the FUGA grant from the NCN (No. 2014/12/S/ST3/00313). DFT calculations were performed at ICM at the University of Warsaw, and at ICTJA-CSIC.

## Author Contributions

J.I. wrote the manuscript and performed first-principle calculations, T.W. carried out first-principle calculations, F.D. and S.H. realized the high-pressure Raman experiments, R.O. realized the data analysis and figures, R.K. provided the samples and overall guidance. All authors discussed the results and commented on the manuscript.

## Additional Information

**Supplementary information** accompanies this paper at <https://doi.org/10.1038/s41598-018-31051-y>.

**Competing Interests:** The authors declare no competing interests.

**Publisher's note:** Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



**Open Access** This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2018