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Research Article Indium-Doped Zinc Oxide Thin Films as Effective Anodes of Organic Photovoltaic Devices

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Indium-doped zinc oxide (IZO) thin films were prepared by low-cost ultrasonic spray pyrolysis (USP). Both a low resistivity $(3.13 \times 10^{-3} \Omega \text{ cm})$ and an average direct transmittance (400~1500 nm) about 80% of the IZO films were achieved. The IZO films were investigated as anodes in bulk-heterojunction organic photovoltaic (OPV) devices based on poly(3-hexylthiophene) and [6,6]-phenyl C₆₁-butyric acid methyl ester. The device fabricated on IZO film-coated glass substrate showed an open circuit voltage of 0.56 V, a short circuit current of 8.49 mA cm⁻², a fill factor of 0.40, and a power conversion efficiency of 1.91%, demonstrating that the IZO films prepared by USP technique are promising low In content and transparent electrode candidates of low-cost OPV devices.

1. Introduction

Organic photovoltaics (OPVs) have attracted considerable attentions for use as the next generation of renewable energy sources, due to their simple cell structure, ease of fabrication, low cost, and flexibility [1-3]. To fabricate costefficient and large area OPVs, it is imperative to develop low-cost transparent conducting oxide (TCO) electrodes, which can substitute for the conventional indium tin oxide (ITO) electrodes and a highly efficient TCO deposition process. Considering the cost advantage of OPVs, ITO film is not a desirable anode material, due to the high cost and shortage of indium [4]. Therefore, extensive efforts are being dedicated to finding low/non-In TCOs which are inexpensive but have higher conductivity and a broader transparency window than ITO [5-12]. In addition, the conventional sputtering process is not acceptable in low-cost OPVs due to its low target usage, clumsy equipments, and necessity of vacuum-based condition. Recently, ultrasonic spray pyrolysis (USP) has received extra-attentions because of its simplicity, suitability for product industrialization, and atmosphericpressure solution deposition routes [13-16]. Wienke et al. have used USP to fabricate IZO thin film, which were

used in amorphous silicon solar cells as transparent anodes. However, there have been few studies on investigation of the fabrication of OPVs on the UPS-deposited IZO electrodes, which would be beneficial for the low-cost production of OPVs [17].

In this work, we report the electrical and optical properties of low indium IZO electrodes prepared by USP technique. An efficient OPV device fabricated on the IZOcoated glass was achieved. The performance of OPV device fabricated on both commercial ITO and USP-deposited IZO are compared.

2. Experimental Details

IZO thin films were prepared on glass substrates (Corning eagle 2000) by USP. A 0.2 mol/L solution of zinc acetate dehydrate $[Zn(CH_3COO)_2 \cdot 2H_2O]$ diluted in ethanol, and deionized water (3:1) was used for the film, and indium acetate $[In(CH_3COO)_3]$ was added to initial solution for indium doping. The In/Zn ratio was 1.5 at %. The glacial acetic acid was added into the precursor solution at 18 vol. % content. After the glass substrate was heated to 480°C, the aerosol of initial solution was nebulized by an ultrasonic



FIGURE 1: (a) Device structure and (b) energy diagram of OPVs.

agitation with 1.7 MHz and transported into the growth chamber by air at a flow rate of 11 L/min. The IZO film with ~600 nm thicknesses was measured by step profilometer (AMBIOS-XP2) and field emission scanning electron microscope (FE-SEM, JSM-6700F). The electrical properties were obtained by Hall-effect measurements using Van der Pauw method (Accent HL5500 PC). The surface morphology was observed by SEM. The direct transmittances of thin films were evaluated by UV-visible-NIR spectrometer (Cary 5000 UV-VIS produced by Varian Company). The root mean square roughness (RMS) was taken in a contact mode with an atomic force microscope (AFM, SPA-400 SPM UNIT product by Seiko Instruments Inc.).

Conventional OPVs were fabricated on the USP-grown IZO electrode to investigate the effect of the sheet resistance, transmittance and work function of the USP-grown IZO electrode on the performance of the OPVs. To fabricate the OPVs, IZO electrodes were cleaned with detergent, ultrasonicated in acetone and isopropyl alcohol, and subsequently dried in an oven overnight. Conducting PEDOT:PSS (Baytron P VP AI 4083) was spin-cast (5000 rmp) with thickness about 40 nm from aqueous solution (after passing a $0.45 \,\mu\text{m}$ filter). A blend solution of $15 \,\text{mg}$ of poly(3hexylthiophere) (P3HT) and 12 mg of [6,6]-phenyl C_{61} butyric acid methyl ester (PCBM) in 1 mL of chlorobenzene was spin-coated on top of the respective PEDOT:PSS layers. Subsequently, the devices were pumped down in vacuum (10^{-4} Pa) , and an LiF(1 nm)/Al (100 nm) film used as the cathode was deposited on top of the active layer. The reference devices were fabricated on commercial ITO glass substrates in the same manner. OPVs with a structure of TCO/PEDOT:PSS/P3HT:PCBM/LiF/Al were completed. The schematic diagram is shown in Figure 1(a). The active area of the devices was 0.08 cm² defined by the shade mask. Device current density-voltage (J-V) characteristics were measured



FIGURE 2: Transmittance of IZO- and ITO-coated glass substrates from ultraviolet to near-infrared region. Inset indicates transmittance of IZO- and ITO-coated glass substrates with respect to the absorbance of the P3HT:PCBM films.

using a Keithley 2420 under 100 mW/cm^2 simulated AM 1.5 G solar illumination calibrated using a Si photodetector. All of the measurements were fabricated and tested in ambient conditions in air.

3. Results and Discussions

As shown in Figure 2, the transparency of IZO-coated glass extents well into the near infrared region (NIR) and exhibits transmittance >80% around 1500 nm. The high transparency of IZO film in the NIR suggests the potential





×30000

(a)

100 nm

10.0 kV

SEI

FIGURE 3: (a) Scanning electron microscope (SEM) image of IZO film prepared by ultrasonic spray pyrolysis. The topography of (b) IZO-coated glass and (c) ITO-coated glass substrates measured using atomic force microscopy (AFM). The root mean square surface roughness is 12.37 nm for IZO and 2.41 nm for ITO.

application in stacked cell where high NIR transmittance is required. In contrast, the transmittance of commercial ITO-coated glass decreases in the NIR (<1000 nm) due to the high degree of free-carrier absorption [18]. From Figure 2, it is perceptible that the transmission window of the IZO-coated glass is significantly larger than that of the ITO-coated glass. The high transparency is ascribed to the lower carrier density (1.54×10^{20} cm⁻³) according to the Drude model [19]. The measured resistance of IZO-coated glass is $3.14 \times 10^{-3} \Omega$ cm, higher than that of commercial ITO-coated glass ($4.58 \times 10^{-4} \Omega$ cm). These optical and electrical characteristics of the IZO films make this material suitable for solar cell applications.

A comparison of the surface roughness between the IZOand ITO-coated glass substrates is shown in Figures 3(b) and 3(c), respectively. The AFM images reveal that the IZO film forms large irregular crystals, resulting in a surface root mean square (RMS) roughness of 12.37 nm as shown in Figure 3(b). In contrast, the ITO film forms small crystal grains and has a comparatively smooth surface with a RMS roughness of 2.41 nm as shown in Figure 3(c). Although the rough surface of the IZO may induce direct contact between the oxide anode and the Al cathode, spin-coating a \sim 40 nm thick PEDOT:PSS buffer layer on the IZO film surface prior to spin-coating of the active layer smooths the rough surface and eliminates these shorts.

Device A with the structure of IZO/PEDOT: PSS/P3HT: PCBM/LiF/Al and Device B with the structure of ITO/ PEDOT:PSS/P3HT:PCBM/LiF/Al were fabricated, respectively. The *J*-*V* curves in the dark and under simulated AM 1.5 conditions (100 mW/cm²) of the devices A and B are shown in Figure 4. Under illumination, the device A yields a V_{oc} of 0.56 V, a J_{sc} of 8.49 mA/cm², an FF of 0.40, and a power conversion efficiency (PCE) of 1.91%, which is inferior to the performance of the device B with an efficiency of 3.24%. The slopes at the short circuit point and at the open circuit voltage are the inverse values of the shunt resistance (R_{sh}) and the series resistance (R_s) of the equivalent circuit scheme of a solar cell, respectively. The estimated values of R_s and R_{sh} are presented in Table 1.

The measured short circuit currents of the devices A and B were nearly identical, from which we may conclude that there was little change in the optical field distribution, or in charge collection efficiency. From the inset of Figure 2, it can be seen that the absorption domain of the blend of P3HT:PCBM, the photoconductive films of the organic cells, is mainly situated in the region of large transparency of IZO-coated glass. The average of transmittance of IZO is appreciably less than that of ITO in the range of 350~650 nm, which may result in the lower current density than that of device B.

The device made with IZO exhibited similar V_{oc} than did those made with ITO. The value of V_{oc} is also similar to the previous results. If both electrodes establish ohmic contacts with the active layer, V_{oc} in these bulk-heterojunction cells is related directly to the energy difference between the HOMO level of the donor (P3HT) and the LUMO level of the acceptor (PCBM) components. As shown in Figure 1(b),



FIGURE 4: Current density-voltage (*J*-*V*) characteristics of FTO/PEDOT:PSS/P3HT:PCBM/LiF/Al and ITO/PEDOT:PSS/P3HT:PCBM/LiF/Al photovoltaic devices (a) under simulated AM 1.5 conditions and (b) in the dark.

TABLE 1: Detailed performance parameters of the OPVs fabricated on the IZO- and ITO-coated glass substrates.

Device	Anode	$J_{\rm sc}~({\rm Ma}{\rm cm}^{-2})$	$V_{\rm oc}~({ m V})$	FF (%)	PCE (%)	$R_s \left(\Omega \mathrm{cm}^2\right)$	$R_{\rm sh} (\Omega {\rm cm}^2)$
А	IZO	8.49	0.56	40	1.91	32.25	284.45
В	ITO	8.95	0.59	61	3.24	8.08	584.79

the introduction of high work function of PEDOT:PSS as a buffer layer, ohmic contacts between TCO and active layer are achieved, resulting in better hole extraction. Therefore, PEDOT:PSS buffer layer not only planarizes the IZO surface, but also induces the ohmic contact between TCO and active layer. Slight variation in V_{oc} values may arise from pinholes, which short-circuits the photovoltaic cell resulting in large leak current. For a conventional *p*-*n* junction solar cell the V_{oc} is given by the formula

$$V_{\rm oc} = \left(\frac{nkT}{q}\right) \ln\left(\frac{J_{\rm sc}}{J_{\rm s}} + 1\right),\tag{1}$$

where J_s is the saturation current density, q is the elementary charge, k is Boltzmann's constant, T is temperature, and nis the ideality factor, J_{sc} is the short circuit current density. The values of V_{oc} derived from experimental light J-V curves of devices A and B fit to the above equation agree with the measured values well.

In Figure 4(b), the dark J-V characteristics of the devices A and B are shown on a logarithmic scale. The devices A and B almost present the same reverse biased current, which suggest the smooth effect of PEDOT:PSS on relatively rough IZO-coated glass. The influence of the buffer layer can be observed in the J-V curves under lower bias in the dark. The J-V curves were first symmetric, and then reached the injection limit at forward bias. At further increase of forward bias the current approached a space charge limited condition and the difference among the J-V curves of the diodes with different buffer layers is small. This is because the current in

this region mainly depends on the mobility of charge carriers in the active layers. Under forward bias, the difference of ohmic contributions and injection limit in diodes reflect the hole injection from anode to active layer via the buffer layer, because the electron injection conditions are the same in the devices A and B. This indicates that the injection capabilities of these buffer layers are the same as well. At high forward bias, the current density of device B is obviously larger than that of device A, which is ascribed to the lower resistance of ITO films, indicating superior FF of device B [2]. To overcome FF losses in solar cells, future research should concentrate on increasing the layer conductivity and enhancing the surface smoothness of the films without losing the transmittance.

4. Conclusion

In summary, this study investigates atmospheric-pressure solution deposition routes as an alternative to these traditional high-vacuum techniques. Solution processing is attractive due to its ease and potential to lower devicemanufacturing costs. We demonstrated that USP-grown IZO electrodes are promising low-cost anode materials. Due to the relatively low resistivity and high transmittance extended to NIR, which is a critical benefit for the fabrication of lowcost OPVs. More importantly, considering the materials cost and process efficiency, it is believed that low In IZO electrodes grown by the USP method are better suited to costefficient OPVs than the batch-type grown ITO electrodes. Furthermore, it was found that the performance of the OPV fabricated on the UPS-grown IZO electrodes depends mainly on the sheet resistance and surface morphology of the IZO electrodes, regardless of the work function, due to the existence of a PEDOT:PSS ohmic layer connecting the anode and the active layer. The presented results demonstrate that USP is a promising method for the deposition of IZO thin films and can be used as replacement for the ITO electrode. In addition, the IZO film with high transparency in the NIR appears very promising in stacked solar cells with absorption of full wavelength spectrum. This may lead to higher efficiency organic multijunction photovoltaic cells.

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