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Active site formation mechanism of carbon-based oxygen reduction catalysts derived from a hyperbranched iron phthalocyanine polymer

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Abstract

Carbon-based cathode catalysts derived from a hyperbranched iron phthalocyanine polymer (HB-FePc) were characterized, and their active-site formation mechanism was studied by synchrotron-based spectroscopy. The properties of the HB-FePc catalyst are compared with those of a catalyst with high oxygen reduction reaction (ORR) activity synthesized from a mixture of iron phthalocyanine and phenolic resin (FePc/PhRs). Electrochemical measurements demonstrate that the HB-FePc catalyst does not lose its ORR activity up to 900°C, whereas that of the FePc/PhRs catalyst decreases above 700°C. Hard X-ray photoemission spectra reveal that the HB-FePc catalysts retain more nitrogen components than the FePc/PhRs catalysts between pyrolysis temperatures of 600°C and 800°C. This is because the linked structure of the HB-FePc precursor has high thermostability against nitrogen desorption. Consequently, effective doping of active nitrogen species into the *sp*² carbon network of the HB-FePc catalysts may occur up to 900°C.

Keywords: Oxygen reduction reaction (ORR); X-ray photoemission spectroscopy; Electronic structure; Hyperbranched polymer; Iron phthalocyanine (FePc)

Background

The polymer electrolyte fuel cell (PEFC) is an efficient and clean energy conversion system that produces electricity from the redox reaction of H₂ and O₂ gases [1]. Because PEFCs operate at lower temperatures (approximately 80°C) than other types of fuel cells and the oxygen reduction reaction (ORR) at the cathode has sluggish kinetics, Pt is usually used as a catalyst to accelerate the ORR in PEFCs. However, the high cost and poor durability of Pt currently limit the commercialization of PEFCs [2]. Among studies on Pt-free cathode catalysts, carbon-based catalysts have attracted considerable attention as alternatives to Pt catalysts. In particular, carbon-based cathode catalysts synthesized by pyrolyzing transition metal (TM) chelates (such as phthalocyanine or

porphyrin complexes) and/or nitrogen precursors [3-19] have demonstrated promising activities. Based on a pioneering study on cobalt phthalocyanine as an oxygen reduction catalyst [3], it has been suggested that nitrogen-coordinated TMs such as TM-N₄ and TM-N₂ (TM = Fe, Co) are possible active sites of the ORR in carbon-based cathode catalysts [5-7]. Nitrogen species incorporated into a carbon network, such as pyridineand pyrrole-like nitrogen [9,10,12,13] and graphite-like nitrogen [11-13,15-17], also contribute to the activity of the ORR [8-17]. In previous studies, we analyzed the electronic structure of carbon-based cathode catalysts derived from mixtures of metallophthalocyanine and phenolic resin (TMPc/PhRs) [17,18]. We found that the residual TM atoms existed mostly as metallic nanoparticles after pyrolysis. It has also been revealed that these TM particles catalyze graphitization by the Yarmulke mechanism [20], serving as centers of graphitization in a manner similar to carbon nanotubes. We have elucidated that nitrogen functionalities in the graphitized carbon network surrounding these

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TM particles are responsible for the high ORR activity of TMPc/PhRs-derived catalysts [15-17,21]. It is therefore important to introduce more nitrogen into TMPc/PhRs systems to increase their ORR activity. However, there is a trade-off between activity and durability: although graphitization at higher temperature (above 600°C) improves the stability of the catalysts, nitrogen desorption catalyzed by TM atoms and/or desorption of the TM atoms occurs, and this desorption reduces ORR activity [6,13]. Therefore, to obtain catalysts with both high ORR activity and stability, it is essential to develop a process to fabricate highly graphitized carbon-based cathode catalysts with sufficient nitrogen content.

Here we report a novel carbon-based cathode catalyst synthesized from a hyperbranched iron phthalocyanine polymer (HB-FePc). HB-FePc contains numerous branched structures where FePc complexes are connected to each other by a specific linker [22,23]. These HB polymers can solve the trade-off problem mentioned above because they resist thermal decomposition and thus can suppress nitrogen desorption during pyrolysis [23]. In this paper, we explore spectroscopic study on the formation mechanism of ORR active sites in HB-FePc-derived catalysts by comparison of their properties with those of the iron phthalocyanine and phenolic resin (FePc/PhRs)-derived catalysts which do not contain a linker among FePc complexes. We conducted hard X-ray photoemission spectroscopy (HXPES) measurements to elucidate the roles of light elements and residual iron in the ORR. HXPES using synchrotron radiation has many advantages over conventional X-ray photoemission spectroscopy (XPS) because of its high energy resolution and long probing depth. In particular, the average diameter of the iron-centered nanoshell structure in carbon-based cathode catalysts is 20 to 30 nm [14,18,19], which matches the long probing depth of HXPES of over 30 nm well. Our experimental results reveal how the linked structure in the HB-FePc catalysts modifies the thermal process to form ORR active sites and contributes to retain the ORR activity of these catalysts pyrolyzed at temperatures up to 900°C.

Methods

Sample preparation and characterization

Two different types of carbon-based cathode catalysts were prepared from pyrolysis of (i) a mixture of FePc/PhRs and (ii) HB-FePc with a specific linker (see Figure 1). The Fe content of the HB-FePc precursor was in the range of 2.2 to 2.7 wt% (estimated from thermogravimetry-differential thermal analysis (TG-DTA) measurements), while that of FePc/PhRs was adjusted to 3 wt%. The carbon-based cathode catalysts were synthesized at various temperatures (500°C to 900°C) under $\rm N_2$ flow for 5 h. These samples are denoted according to their pyrolysis temperature. For example, Fe600 and HB600 denote FePc/PhRs- and HB-FePc-derived carbon-based cathode catalysts pyrolyzed at 600°C, respectively. A mixture of FePc and PhRs that was not heat treated is denoted FePc/PhRs.

The ORR activities of these samples were evaluated by rotating disk electrode (RDE) voltammetry in 0.5 mol L $^{-1}$ $\rm H_2SO_4$ saturated with oxygen at room temperature. Linear sweep voltammograms were recorded by sweeping the potential from 1.1 to 0 V νs . a normal hydrogen electrode (NHE) at 0.5 mV s $^{-1}$, where the working electrode was rotated at a fixed speed of 1,500 rpm. As a measure of the ORR activity, we used the onset potential of the ORR $E_{\rm O2}$, which is defined as the voltage at which the reduction current density normalized by the area of a glass-like carbon disk electrode reached $-2~\mu A~cm^{-2}$. The current density at 0.5 V (νs . NHE) was also measured as an indicator of catalytic performance.

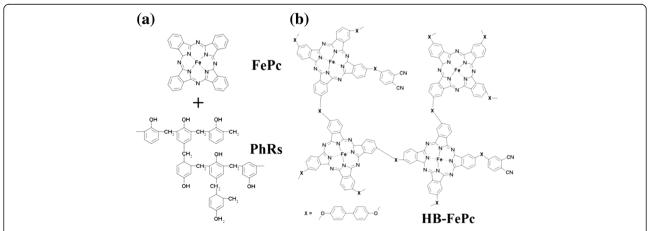


Figure 1 Structural formulae of precursors. (a) Iron phthalocyanine (FePc) and phenolic resin (PhRs), and (b) hyperbranched iron phthalocyanine polymer (HB-FePc). In this study, biphenyl was used to link HB-FePc (X).

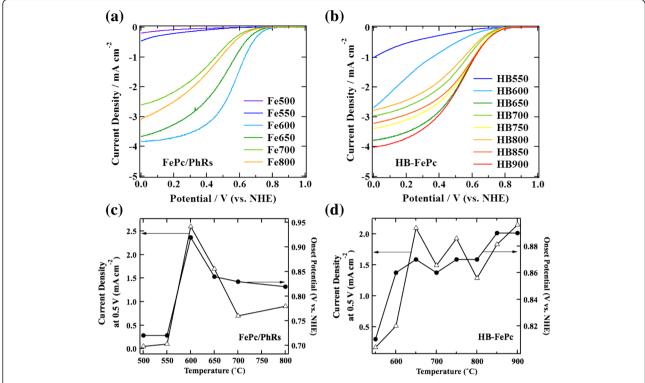


Figure 2 Electrochemical characterization of the catalysts. Linear sweep voltammograms of (a) FePh/PhRs and (b) HB-FePc catalysts and onset potential and current density plots as a function of pyrolysis temperature (bottom) of (c) FePc/PhRs and (d) HB-FePc catalysts.

In order to estimate the specific surface area of the catalysts, N2 adsorption measurements were conducted with a volumetric adsorption measurement instrument (Belsorp-mini II, BEL Japan, Inc., Osaka, Japan). X-ray diffraction (XRD) was measured with an X-ray diffractometer (Ultima IV, Rigaku, Tokyo, Japan) using Cu $K\alpha$ radiation operating at 40 kV and 40 mA. TG, DTA, and differential thermogravimetry (DTG) were performed in He gas flow (300 mL min⁻¹) using a TG-DTA system (Thermo Plus Evo, Rigaku, Tokyo, Japan) connected to a mass spectrometer (GC-MS-QP2010, Shimadzu, Kyoto, Japan). The reaction section was purged with He gas for 30 min before heating. The temperature was raised from room temperature to 900 K at a rate of 50 K min⁻¹, and outlet gas was analyzed by the mass spectrometer.

HXPES measurements

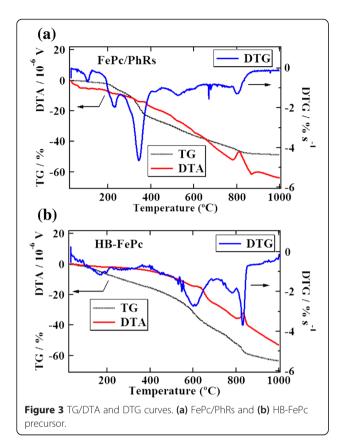
HXPES measurements were conducted in an ultrahigh-vacuum chamber at room temperature at BL46XU and BL47XU of SPring-8. The photon energy used for the photoemission measurements was 7.94 keV. Insulating FePc/PhRs catalysts Fe500 and Fe550 were mounted on indium sheets to avoid charging problems. In this study, the probing depth of HXPES for the C 1s, N 1s, O 1s, and Fe 2p edges is approximately 40 nm, which is about three

times larger than the inelastic mean free paths of photoelectrons [24]. HXPES spectra were recorded by an electron analyzer (VG Scienta, R-4000) with a pass energy Epass of 200 eV. The overall energy resolution estimated using the Fermi edge of an Au plate was approximately 0.25 eV, and the stability of the absolute energy was within 20 meV. All of the spectra shown in this paper are normalized to the peak height. For detailed chemical analysis, backgrounds of the core level spectra were subtracted by the Shirley method [25]. For the N 1s HXPES, each spectrum was fitted with a Voigt function [26], which is a convolution of Gaussian and Lorentzian functions, to decompose it into different chemical species of nitrogen.

Table 1 Specific BET surface area of FePc/PhRs and HB-FePc catalysts calculated from nitrogen adsorption measurements

Temperature (°C)	FePc/PhRs ^a (m ² g ⁻¹)	HB-FePc (m ² g ⁻¹)		
900	-	409		
800	407	485		
700	444	409		
650	437	384		
600	515	337		
550	506	564		

^aData from ref. [14].

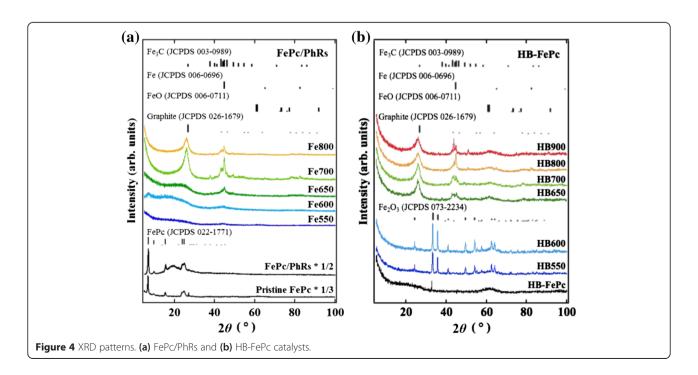


Results and discussion

Electrochemical and physical properties

Figure 2 shows linear sweep voltammograms of (a) the FePc/PhRs and (b) HB-FePc carbon-based cathode catalysts (hereafter denoted simply as FePc/PhRs and HB-FePc catalysts) measured by the RDE method. The corresponding onset potential and current density as a function of pyrolysis temperature for the FePc/PhRs and the HB-FePc catalysts are plotted in Figure 2c,d. Among the FePc/PhRs catalysts, Fe600 shows the highest onset potential (0.92 V) with the largest current density at 0.5 V. For pyrolysis temperatures above 600°C, the onset potential decreases slightly and the current density at 0.5 V degrades markedly. In contrast, the ORR activity of the HB-FePc catalysts increases at a pyrolysis temperature of around 650°C (onset potential = 0.87 V for HB650) and remains almost unchanged up to 900°C (0.89 V for HB900). It is remarkable that the ORR activity of the HB-FePc catalysts is retained at such high pyrolysis temperatures compared with that of the FePc/PhRs catalysts. Specific BET surface areas of both samples are summarized in Table 1. The values are between 337 to 564 m² g⁻¹, suggesting that the pore structures of these samples do not have significant impacts on the ORR activity.

TG-DTA/DTG curves of the FePc/PhRs and HB-FePc precursors are presented in Figure 3. In the case of the FePc/PhRs precursor, two large DTG mass loss peaks are observed at around 230°C and 340°C. From the mass spectrum (not shown), the molecular weight of these peaks corresponds to phenol (m/z = 94). The DTG curve of the HB-FePc precursor does not show any significant



peaks, implying that the biphenyl linker is stable below 500°C. Instead, the DTG curve of the HB-FePc precursor shows two large peaks at around 600°C and 830°C. The corresponding structure also appears in the DTG curve of the FePc/PhRs precursor, but with much smaller weight loss compared with that of the HB-FePc precursor. In other words, the main decomposition region of the HB-FePc precursor appears at a higher temperature (above 600°C) than that of the FePc/PhRs precursor. This may be because in the HB-FePc catalyst, the biphenyl linker connecting phthalocyanine moieties does not decompose, which suppresses liberation/sublimation of each component below 600°C. The DTA curve of the HB-FePc precursor has exothermal

peaks at 650°C and 830°C, suggesting that graphitization occurs in the same temperature range as the main decomposition of the HB-FePc precursor. In contrast, the DTA curve of the FePc/PhRs precursor shows a graphitization peak around 800°C. Taking into account both the DTA and DTG curves of the FePc/PhRs precursor, decomposition occurs gradually above 600°C, while graphitization occurs exclusively at around 800°C.

In the XRD patterns shown in Figure 4, quite strong diffraction peaks are observed for the pristine FePc precursor as a result of its crystalline nature [27], whereas the HB-FePc precursor does not show any diffraction peaks because it has a large, randomly linked structure

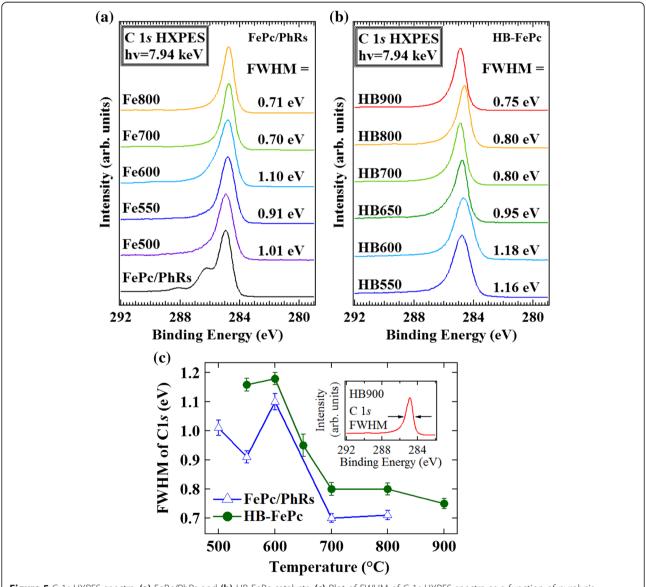


Figure 5 C 1s HXPES spectra. (a) FePc/PhRs and (b) HB-FePc catalysts. (c) Plot of FWHM of C 1s HXPES spectra as a function of pyrolysis temperature.

that does not readily crystallize. HB550 and HB600 have amorphous signatures as a broad structure around 26° and multiple iron oxide (Fe₂O₃) peaks, while above HB650 graphitized carbon network around 26° as well as iron metal (Fe) and/or iron carbides (Fe₃C) around 45° appeared. Thus the formation of the graphitic structure and the iron metal and/or iron carbides are strongly correlated. This can be understood by considering the Yarmulke mechanism; i.e., formation of a graphitic carbon shell structure around reduced metal nanoparticles [20]. In the case of the FePc/PhRs catalysts, the intensity of the diffraction peaks at both 26° and 45° gradually increases from 600°C. These peaks are absent in the XRD pattern of HB600, indicating that the HB-FePc precursor has higher thermostability than the FePc/PhRs precursor. Instead, these peaks suddenly emerge at 650°C, and then their intensities remain almost unchanged up to 900°C. Therefore, the formation of reduced iron components and their subsequent graphitization may proceed in a different manner in the FePc/PhRs and HB-FePc catalysts against pyrolysis temperature, which is consistent with the TG-DTA/DTG results. The sudden appearance of the metallic iron components at 650°C in the XRD patterns of the HB-FePc catalysts can be explained by reduction of the iron moieties, mostly iron oxide (Fe₂O₃). The presence of the iron oxide in HB550 and HB600 is quite in contrast to the FePc/PhRs catalysts. As already seen in the TG-DTA/DTG results, HB-FePc precursors are very reactive around 600°C. We believe that the biphenyl linker in the HB-FePc catalyst may connect phthalonitrile fragments decomposed from FePc, while the residual iron atoms dissociate and then generate clusters, which are easily oxidized in air to form iron oxides.

C 1s HXPES

Figure 5a,b shows the C 1s HXPES spectra of the FePc/ PhRs and HB-FePc catalysts, respectively. All of the pyrolyzed samples show a single peak in the range from 284.3 to 284.9 eV, indicating that the catalysts predominantly consist of typical sp^2 carbon networks [7,10,28,29]. The FePc/PhRs precursor shows two main peaks at 284.9 and 286.2 eV corresponding to C-C and C-N bonds, respectively, and a small shake-up satellite at 288.1 eV (the same profile as in ref. [30]). The full width at half-maximum (FWHM) of the single peak was estimated by a rough fitting with a single Voigt function, which can be used to represent the degree of sp^2 carbon network formation [28,29]; i.e., a smaller FWHM indicates more development of the sp^2 carbon network. Figure 5c plots the FWHM of the single peak as a function of pyrolysis temperature. For the FePc/PhRs catalysts, the FWHMs increase by 0.2 eV from 550°C to 600°C, indicating decomposition of the Fe-N₄ centers of FePc [7,14] and appearance of disordered carbon structure. In contrast, the FWHMs of the HB-FePc catalysts exhibit a very slight (0.02 eV) increase from 550°C to 600°C, suggesting that the HB-FePc catalysts do not decompose or graphitize as much as the FePc/PhRs

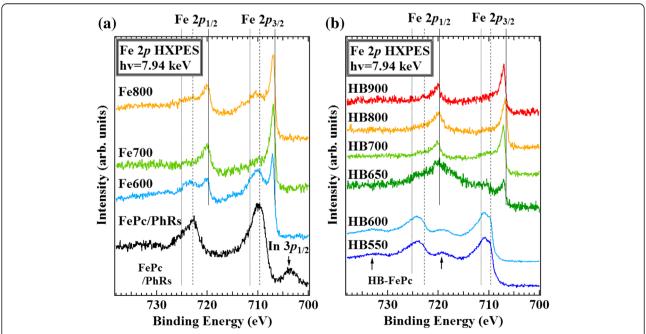


Figure 6 Fe 2p HXPES spectra. **(a)** FePc/PhRs and **(b)** HB-FePc catalysts. Solid, dashed, and dotted lines show the binding energies of Fe metal, Fe²⁺, and Fe³⁺, respectively. The arrows indicate satellite peaks of Fe₂O₃ $2p_{3/2}$ and $2p_{1/2}$, which emerge approximately 8 eV above pristine peaks [31]. In the spectrum of the FePc/PhRs precursor, a peak from In $3p_{1/2}$ is observed.

catalysts in this temperature range because of their higher thermostability. Details of the correlation between the active site formation mechanism and carbon structure of these catalysts will be discussed in the last section.

Fe 2p HXPES

Fe 2p HXPES spectra of the FePc/PhRs and HB-FePc catalysts are illustrated in Figure 6. Solid, dashed, and dotted lines show the binding energies of Fe metal (706.5 and 719.7 eV), FeO (709.6 and 722.9 eV), and Fe₂O₃ (711.6 and 725.1 eV), respectively [31]. The FePc/PhRs precursor shows two peaks at 709.5 and 723.0 eV. These values are about 1 eV higher than those reported previously for FePc (708.8 eV at Fe $2p_{3/2}$) [30,32] and

similar to those of FeO. However, because we did not observe Fe oxides in the XRD pattern of the FePc/PhRs precursor (see Figure 4), the 1 eV shift could be caused by a chemical interaction between PhRs and FePc. In both catalysts pyrolyzed below 600°C, the main iron components are oxidized species. The peaks in the spectrum of Fe600 correspond to a mixture of FePc and Fe metal. In contrast, the peaks in the spectra of HB550 and HB600 can be mostly assigned to Fe₂O₃, consistent with the XRD results as already discussed. Reduced Fe metal peaks appear in the spectra of the HB-FePc catalysts pyrolyzed at temperatures at least 50°C higher than the FePc/PhRs catalysts, in good agreement with the XRD results. In both catalysts, the oxidized moieties are completely reduced to a metallic state (Fe°) at higher pyrolysis temperatures.

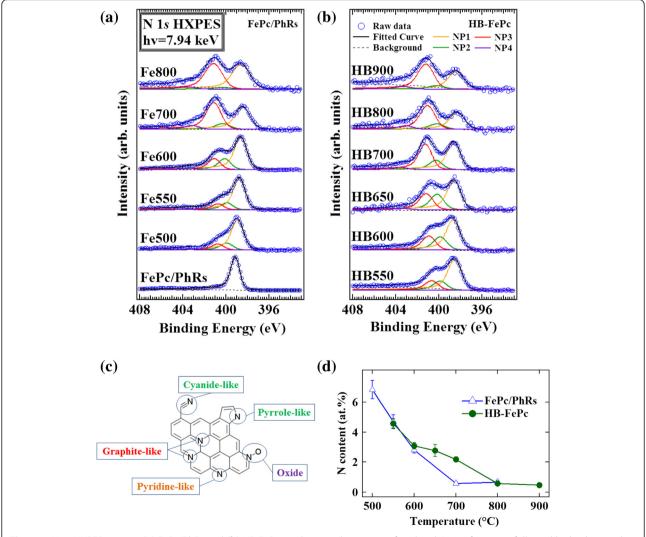


Figure 7 N 1s HXPES spectra. **(a)** FePc/PhRs and **(b)** HB-FePc catalysts, each spectrum fitted with Voigt functions followed by background subtraction by the Shirley method (dashed line). Orange, green, red, and purple solid lines are pyridine-like or FePc (NP1), pyrrole- or cyanide-like (NP2), graphite-like (NP3), and oxide (NP4) nitrogen components, respectively. **(c)** Structural formulae of four nitrogen components in graphite. **(d)** Plot of calculated nitrogen content as a function of pyrolysis temperature.

N 1s HXPES

Figure 7a,b shows N 1s HXPES spectra of the FePc/ PhRs and HB-FePc catalysts, respectively. To reveal the chemical states of nitrogen, each spectrum was fitted with a Voigt function and decomposed into different chemical species (Figure 7c). The Gaussian width was used as a common parameter among chemical species to fit the experimental data. The Lorentzian width and asymmetric parameter α were fixed to 0.25 eV and 0.10, respectively, where α is adjusted to the mean value for the C 1s HXPES spectra. By applying the above conditions, we were able to decompose the N 1s HXPES spectra into four peaks, NP1 to NP4 [7,30,33-37]. The resultant fitting parameters and relative composition ratios of the four nitrogen components of the FePc/PhRs and HB-FePc catalysts are summarized in Tables 2 and 3, respectively. The nitrogen content of FePc/PhRs was not calculated because of contamination with indium. Peak NP1 was observed at 398.2 to 398.6 eV except for FePc, where it appeared at 399.1 eV, and can be assigned to pyridine-like nitrogen (nitrogen atoms with two carbon neighbors in an aromatic ring) [7,33,35] or nitrogen of FePc [7,30]. Considering that the majority of the Fe-N₄ structure of FePc is decomposed below 600°C as revealed by Fe 2p XPS [33], NP1 is assigned to pyridine-like nitrogen. Peak NP2 (399.8 to 400.2 eV) may be a mixture of pyrrole-like nitrogen (nitrogen atoms with two carbon neighbors in a fivemembered aromatic ring) [7,33] and cyanide-like nitrogen (one neighboring carbon atom with a triple bond) [34]. Peak NP3 (400.6 to 401.1 eV) can be attributed to graphite-like nitrogen, which is nitrogen with three carbon bonds incorporated into an aromatic ring [7,35-37]. The binding energy of peak NP4 varies randomly for each catalyst (403.0 to 403.6 eV). Based on the previous reports [7,33], NP4 is assigned to N-oxide groups.

There is a small difference between the N 1s spectra of the two types of catalysts. The relative composition of graphite-like nitrogen (NP3) increases with pyrolysis temperature, suggesting that graphite-like nitrogen can be introduced and stabilized more easily at higher temperatures than pyridine-like nitrogen (NP1) [6,7,13]. The NP2 component is unstable and disappears at high temperatures. The composition ratio of NP4 (N-oxide group) is less than 8% (at most 0.1 at.% in total) at all pyrolysis temperatures. Taking into account that the amount of oxygen is more than 2.0 at.% in all of the catalysts, oxygen should be primarily bonded to carbon, not to nitrogen in the form of N-oxide.

Formation mechanism of active sites

XRD and HXPES results indicate that the FePc/PhRs precursor decomposes gradually above 600°C. At the same time, thermal reduction of iron proceeds and metallic iron nanoparticles form [19]. These metallic particles accelerate the graphitization of the carbon network around them and catalyze effective nitrogen doping with considerable amounts of disordered surfaces and carbon edge structures. The XRD results support this graphitization mechanism; as shown in Figure 4, the diffraction peak of the graphitic carbon in Fe600 is quite broad, suggesting that the carbon in Fe600 mainly consists of small graphitized fractions. High ORR activity caused by the presence of such disordered, nitrogen-modified structure is consistent with the findings of other studies on carbon-based cathode catalysts derived from Fe/carbon/nitrogen precursors. Ozkan et al. reported that active carbon-based cathode catalysts have a large amount of edge planes [9,10]. Although Fe600 has a high proportion of edge structure,

Table 2 Fitting parameters and relative composition ratios of four nitrogen components in FePc/PhRs catalysts

Sample	Gaussian width (eV)	N content (at.%)	Binding energy (eV)/(composition ratio, %)				
			NP1	NP2	NP3	NP4	
Fe800	1.72	0.64	398.5	400.2	401.1	403.6	
			(46.2)	(3.2)	(46.5)	(4.1)	
Fe700	1.42	0.56	398.4	400.2	401.0	403.1	
			(38.9)	(9.7)	(45.1)	(6.3)	
Fe600	1.15	2.80	398.6	400.1	401.0	403.4	
			(59.0)	(19.6)	(19.5)	(1.9)	
Fe550	1.09	4.72	398.7	399.9	401.7	403.5	
			(70.3)	(15.5)	(13.0)	(1.2)	
Fe500	1.15	6.83	398.6	399.9	400.7	403.4	
			(68.7)	(16.1)	(13.7)	(1.5)	
FePc/PhRs	0.79		399.1	-	-	-	
			(100.0)	(0.0)	(0.0)	(0.0)	

Table 3 Fitting parameters and relative composition ratios of four nitrogen components in HB-FePc catalysts	Table 3 Fitting parameters and	l relative comp	osition ratios of fo	our nitrogen comi	ponents in HB-FePc catalysts
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Sample	Gaussian width (eV)	N content (at.%)	Binding energy (eV)/(composition ratio, %)				
			NP1	NP2	NP3	NP4	
HB900	1.53	0.45	398.3	400.1	401.1	403.2	
			(36.3)	(7.4)	(50.8)	(5.5)	
HB800	1.44	0.56	398.2	400.1	401.0	403.0	
			(34.4)	(11.1)	(47.6)	(7.0)	
HB700	1.30	2.17	398.6	400.2	401.1	403.5	
			(43.9)	(11.9)	(40.7)	(3.5)	
HB650	1.30	2.76	398.5	400.1	401.1	403.3	
			(47.7)	(24.8)	(25.5)	(2.0)	
HB600	1.30	3.09	398.6	399.8	400.9	403.3	
			(52.6)	(22.9)	(24.0)	(0.5)	
HB550	1.30	4.56	398.4	399.8	400.6	403.2	
			(62.1)	(18.5)	(19.0)	(0.4)	

its degree of graphitization is insufficient [6]. Above 700°C, further graphitization occurs and stabilizes the structure, while facilitating nitrogen desorption. In fact, Figure 7d reveals that the nitrogen content of the FePc/PhRs catalysts reduces to a constant value (approximately 0.6 at.%) above 700°C. This behavior is consistent with the decreased ORR activity of the FePc/PhRs catalysts at high temperatures, which is probably caused by a decrease of catalytically active nitrogen species through nitrogen desorption. In contrast, the nitrogen contents of the HB-FePc catalysts are higher than those of the FePc/PhRs catalysts between 600°C to 800°C, as shown in Figure 7d. These observations may be associated with the thermostability of the HB-FePc catalysts where their own components are trapped more effectively than the FePc/PhRs catalysts during pyrolysis. Meanwhile, the metallic iron components effectively facilitate graphitization and trap nitrogen moieties at 650°C. Thus, nitrogen-containing fragments are presumably present when the carbon network is formed by the Yarmulke mechanism [20] in this temperature region (600°C to 800°C).

The question is why the HB-FePc catalysts do not lose their ORR activity up to 900°C even though the residual nitrogen contents above 800°C are the same (approximately 0.6 at.%, see Figure 7d) for both the FePc/PhRs and HB-FePc catalysts. We consider that the simultaneous decomposition and graphitization in the HB-FePc catalysts may effectively produce the nitrogen sites doped at the carbon edge which is responsible for the high ORR activity. Indeed, as shown in Figure 5c, HB800 has a more disordered carbon structure than Fe800 and possibly has a higher proportion of nitrogen atoms at edge sites. The role of iron in the HB catalysts seems to be the formation of the stable but disordered carbon

network and the effective nitrogen uptake at high temperatures, while they do not directly catalyze ORR.

Conclusions

Two types of carbon-based cathode catalysts were synthesized from FePc/PhRs and HB-FePc precursors, and characterized by electrochemical measurements, TG-DTA, XRD, and HXPES. The HB-FePc catalysts retained their ORR activity at high temperatures up to 900°C. TG-DTA and XRD results showed that the HB-FePc catalysts have a linked structure and high thermostability at these high temperatures. From the HXPES analysis, the FWHMs of the C 1s peaks of the FePc/PhRs catalysts exhibited a large increase (0.2 eV) at 600°C and then decreased above 600°C, possibly because of thermal decomposition of FePc macrocycles and subsequent graphitization. In contrast, that of the HB-FePc catalysts showed a negligible (0.02 eV) increase, so decomposition was suppressed by the biphenyl linker. In addition, the nitrogen content of the HB-FePc catalysts was higher than that of the FePc/PhRs catalysts between 600°C to 800°C. The retention of the ORR activity of the HB-FePc catalysts above 800°C may be related to the graphitization of the HB-FePc catalysts containing ORRactive nitrogen [38]. The unique active site formation mechanism in the HB-FePc catalysts provides a possibility to synthesize carbon-based cathode catalysts with improved ORR activity at high pyrolysis temperatures.

Abbreviations

DTA: differential thermal analysis; DTG: differential thermogravimetry; FePc/PhRs: iron phthalocyanine and phenolic resin; HB-FePc: hyperbranched iron phthalocyanine polymer; HXPES: hard X-ray photoemission spectroscopy; NHE: normal hydrogen electrode; ORR: oxygen reduction reaction; PEFC: polymer electrolyte fuel cell; RDE: rotating disk electrode; TG: thermogravimetry; TM: transition metal; TMPc/

PhRs: metallophthalocyanine and phenolic resin; XPS: X-ray photoemission spectroscopy; XRD: X-ray diffraction.

Competing interests

The authors declare that they have no competing interests.

Authors' contributions

YHi carried out all the measurements and drafted the manuscript. MS carried out the HXPES measurements and participated in the discussion of the study. HN and MKo carried out the HXPES measurement, participated in the discussion of the study, and helped to draft the manuscript. YHa participated in its coordination, discussion of the study and draft the manuscript. MO participated in its design and coordination, and helped to revise the manuscript. JK carried out the synthesis and elemental analysis of the hyperbranched polymer. YN and MKa conceived of the study, participated in its design, coordination, and supplied the hyperbranched polymer. All authors read and approved the final manuscript.

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References

- Wee J. Applications of proton exchange membrane fuel cell systems. Renew Sust Energ Rev. 2007;11:1720–38.
- Gasteiger HA, Kocha SS, Sompalli B, Wagner FT. Activity benchmarks and requirements for Pt, Pt-alloy, and non-Pt oxygen reduction catalysts for PEMFCs. Appl Catal B. 2005;56:9–35.
- 3. Jasinski R. A new fuel cell cathode catalyst. Nature. 1964;201:1212–3.
- 4. Jahnke H, Schönborn M, Zimmermann G. Organic dyestuffs as catalysts for fuel cells. Top Curr Chem. 1976;61:133–81.
- Lefevre M, Proietti E, Jaouen F, Dodelet J. Iron-based catalysts with improved oxygen reduction activity in polymer electrolyte fuel cells. Science. 2009;324:71–4.
- Faubert G, Lalande G, Cote R, Guay D, Dodelet J, Weng L, et al. Heat-treated iron and cobalt tetraphenylporphyrins adsorbed on carbon black: physical characterization and catalytic properties of these materials for the reduction of oxygen in polymer electrolyte fuel cells. Electrochim Acta. 1996;41:1689–701.
- Pylypenko S, Mukherjee S, Olson TS, Atanassov P. Non-platinum oxygen reduction electrocatalysts based on pyrolyzed transition metal macrocycles. Electrochim Acta. 2008;53:7875–83.
- 8. Yeager E. Electrocatalysts for ${\rm O_2}$ reduction. Electrochim Acta. 1984;29:1527–37.
- Matter P, Ozkan U. Non-metal catalysts for dioxygen reduction in an acidic electrolyte. Catal Lett. 2006;109:115–23.
- Matter P, Zhang L, Ozkan U. The role of nanostructure in nitrogen-containing carbon catalysts for the oxygen reduction reaction. J Catal. 2006;239:83–96.
- Iwazaki T, Obinata R, Sugimoto W, Takasu Y. High oxygen-reduction activity of silk-derived activated carbon. Electrochem Commun. 2009;11:376–8.
- Nallathambi V, Lee J, Kumaraguru SP, Wu G, Popov BN. Development of high performance carbon composite catalyst for oxygen reduction reaction in PEM proton exchange membrane fuel cells. J Power Sources. 2008;183:34–42.

- Liu G, Li X, Ganesan P, Popov BN. Studies of oxygen reduction reaction active sites and stability of nitrogen-modified carbon composite catalysts for PEM fuel cells. Electrochim Acta. 2010;55:2853–8.
- Nabae Y, Malon M, Lyth SM, Moriya S, Matsubayashi K, Islam NM, et al. The role of Fe in the preparation of carbon alloy cathode catalysts. ECS Trans. 2009;25:463–7.
- Oshima M, Niwa H, Saito M, Kobayashi M, Horiba K, Harada Y, et al. Electronic structures of non-Pt carbon alloy catalysts for polymer electrolyte membrane fuel cells revealed by synchrotron radiation analyses. Mater Res Soc Symp Proc. 2011;1318:437.
- Niwa H, Horiba K, Harada Y, Oshima M, Ikeda T, Terakura K, et al. X-ray absorption analysis of nitrogen contribution to oxygen reduction reaction in carbon alloy cathode catalysts for polymer electrolyte fuel cells. J Power Sources. 2009;187:93–7.
- Niwa H, Kobayashi M, Horiba K, Harada Y, Oshima M, Terakura K, et al. Electronic structure analysis of carbon alloy cathode catalysts for polymer electrolyte fuel cells by hard X-ray photoemission spectroscopy. J Power Sources. 2011;196:1006–11.
- Kobayashi M, Niwa H, Harada Y, Horiba K, Oshima M, Ofuchi H, et al. Changes in electronic states of platinum–cobalt alloy catalyst for polymer electrolyte fuel cells by potential cycling. J Power Sources. 2011;196:8346–51.
- Kobayashi M, Niwa H, Saito M, Harada Y, Oshima M, Ofuchi H, et al. Indirect contribution of transition metal towards oxygen reduction reaction activity in iron phthalocyanine-based carbon catalysts for polymer electrolyte fuel cells. Electrochim Acta. 2012;74:254–9.
- 20. Dai H, Rinzler A, Nikolaev P, Thess A, Colbert D, Smalley R. Single-wall nanotubes produced by metal-catalyzed disproportionation of carbon monoxide. Chem Phys Lett. 1996;260:471–5.
- Kiuchi H, Niwa H, Kobayashi M, Harada Y, Oshima M, Chokai M, et al. Study on the oxygen adsorption property of nitrogen-containing metal-free carbon-based cathode catalysts for oxygen reduction reaction. Electrochim Acta. 2012;82:291.
- 22. Guo M, Yan X, Kwon Y, Hayakawa T, Kakimoto M, Goodson T. High frequency dielectric response in a branched phthalocyanine. J Am Chem Soc. 2006;128:14820–1.
- Nogami K, Kakimoto M, Hayakawa T, Yokomachi K, Seino M, Sakamoto K. Modification of anodized aluminum film by hyperbranched poly (siloxysilane)s in conductive polymer aluminum solid electrolytic capacitor. Chem Lett. 2006;35:134–5.
- 24. Tanuma S, Powell C, Penn D. IMFP is calculated using the TPP-2M equation in the following reference. Surf Interface Anal. 1991;17:911–26.
- Shirley DA. High-resolution X-ray photoemission spectrum of the valence bands of gold. Phys Rev B. 5(1972):4709–14.
- 26. Doniach S, Sunjic M. Many-electron singularity in X-ray photoemission and X-ray line spectra from metals. J Phys C: Solid State Phys. 1970;3:285–91.
- 27. Kuzmann E, Homonnay Z, Vertes A, Li S, Yin H, Wei Y, et al. Mössbauer studies of the interaction of oxygen with solid β -Fe(II) phthalocyanine. J Sol State Chem. 2003;170:118–23.
- 28. Schafer J, Ristein J, Graupner R, Ley L, Stephan U, Frauenheim T, et al. Photoemission study of amorphous carbon modifications and comparison with calculated densities of states. Phys Rev B. 1996;53:7762–74.
- Haerle R, Riedo E, Pasquarello A, Baldereschi A. sp²/sp³ hybridization ratio in amorphous carbon from C1s core-level shifts: X-ray photoelectron spectroscopy and first-principles calculation. Phys Rev B. 65(2002):045101.
- Ahlund J, Nilson K, Schiessling J, Kjeldgaard L, Berner S, Martensson N, et al. The electronic structure of iron phthalocyanine probed by photoelectron and X-ray absorption spectroscopies and density functional theory calculations. J Chem Phys. 2006;125:034709.
- Mills P, Sullivan J. A study of the core level electrons in iron and its three oxides by means of X-ray photoelectron spectroscopy. J Phys D: Appl Phys. 1983;16:723–32.
- 32. Uchida K, Soma M, Onishi T, Tamaru K. XPS studies of nitrosyl metal (iron, cobalt, and manganese) phthalocyanines. Inorg Chim Acta. 1978;26:L3–4.
- Pels J, Kapteijn F, Moulijn J, Zhu Q, Thomas K Evolution of nitrogen functionalities in carbonaceous materials during pyrolysis. Carbon. 1995;33:1641–53.
- Ray S, Pao C, Chiou J, Tsai H, Jan J, Pong W, et al. Electronic properties of a-CNx thin films: an X-ray-absorption and photoemission spectroscopy study. J Appl Phys. 2005;98:033708.
- 35. Casanovas J, Ricart J, Rubio J, Illas F, Jimenez-Mateos J. Origin of the large N 1s binding energy in X-ray photoelectron spectra of calcined carbonaceous materials. J Am Chem Soc. 1996;118:8071–6.

- Ripalda J, Diaz N, Roman E, Galan L, Montero I, Goldoni A, et al. Chemical shift resolved photoionization cross section of amorphous carbon nitride. Phys Rev Lett. 2000;85:2132–5.
- 37. Rodil S, Muhl S. Bonding in amorphous carbon nitride. Diamond Relat Mater. 2004;13:1521–31.
- Ikeda T, Boero M, Huang S, Terakura K, Oshima M, Ozaki J. Carbon alloy catalysts: active sites for oxygen reduction reaction. J Phys Chem C. 2008;112:14706–9.

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