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# Research Article

# Fabrication of TiO<sub>2</sub> Nanofilm Photoelectrodes on Ti Foil by Ti Ion Implantation and Subsequent Annealing

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The  ${\rm TiO_2}$  photoelectrodes fabricated on the substrate of Ti foils by Ti ions implantation and subsequent annealing at different temperatures were applied for water splitting. The size of  ${\rm TiO_2}$  nanoparticles increased with annealing temperatures, and the GIXRD patterns and Raman spectra demonstrate that the phase of  ${\rm TiO_2}$  turns to rutile at high temperature. The photoelectrochemical (PEC) and X-ray photoelectron spectroscopy (XPS) spectra of the valence band demonstrate that the samples annealed at 400 and 500°C show the n-type property. The sample annealed at 600°C shows the weak p-type  ${\rm TiO_2}$  property. For the sample annealed at 700°C, the negative photocurrent is main, which mainly performs the p-type property of  ${\rm TiO_2}$ . The IPCE values indicate that the absorption edges are red shifted with the increase of annealing temperatures.

#### 1. Introduction

The production of clean chemical fuels by solar conversion is an attractive and sustainable solution to the energy shortage. Since the solar water splitting on a  ${\rm TiO_2}$  photoelectrode was discovered by Fujishima and Honda [1], the hydrogen and electric power from water using solar energy has attracted considerable interest because it promises clean, environmentally friendly energy generation. The photoelectrochemical (PEC) reaction has emerged to convert solar energy into chemical energy [2, 3]. The PEC splitting of water into hydrogen and oxygen by the direct use of sunlight is an ideal method. As we know, a PEC cell is based on a semiconductor/liquid junction, where the minority charges generated on light absorption in the semiconductor are driven into the solution by the electric field at the junction, where they can drive a redox reaction [4–6].

In our previous work, we found that  ${\rm TiO_2}$  nanofilms can be formed on the surface of silica by the  ${\rm Ti}^+$  ions implantation and subsequent annealing [7]. However, the substrate is nonconductive and the formed  ${\rm TiO_2}$  films cannot be used as photoelectrode. In order to acquire  ${\rm TiO_2}$  nanofilms on the substrate of  ${\rm Ti}$  foil, we try to fabricate the  ${\rm TiO_2}$ 

by Ti ion implantation and subsequent annealing. The n-type properties of the as-prepared TiO<sub>2</sub> photoelectrode are detected at low temperatures of 400 and 500°C, while, for the samples annealed at 600 and 700°C, the current-potential curves show the property of p-type TiO<sub>2</sub>. The photocurrent of the sample annealed at 500°C under UV-Visible light illumination with applied potential of 0.8 V (versus SCE) is  $139.2\,\mu\text{A/cm}^2$ .

# 2. Experimental

We used high purity (99.9%) Ti foils in our experiment, and the thickness of the Ti foil is 0.5 mm. High purity Ti foils were cleaned in a mixture of hydrofluoric acid, nitric acid, and purified water for 5 minutes and then implanted with Ti ions at an accelerate voltages of 20 kV, to the fluence of 3  $\times$  10  $^{17}$  ions/cm² using a metal vapor vacuum arc (MEVVA) ion source implanter. The implanted samples were annealed at 400, 500, 600, and 700 °C for 6 hours in oxygen atmosphere. The surface morphologies of the annealed samples were examined by scanning electron microscope (SEM, FEI Versa 3D). Raman scattering spectrum measurements were performed by a MicroRaman Microscope (Jobin-Yvon LabRAM

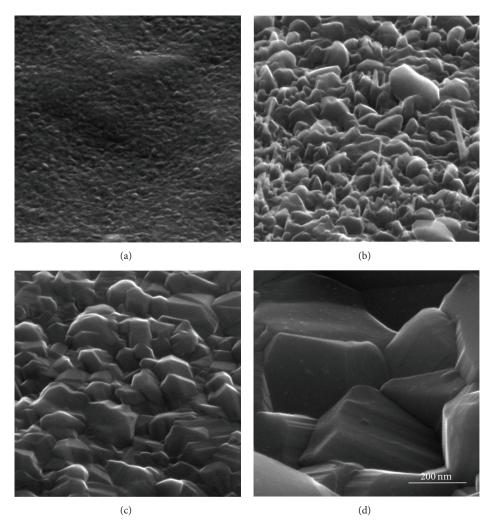


FIGURE 1: SEM image of the  $TiO_2$  formed by Ti implanted into Ti foil and subsequent annealing at 400 (a), 500 (b), 600 (c), and 700°C (d) for 6 hours in oxygen atmosphere, respectively.

HR) using an Ar<sup>+</sup> laser (488 nm) as the excitation source to identify the crystalline phase of  ${\rm TiO_2}$ . The X-ray diffraction (XRD) patterns were obtained from a PANalytical X'pert MPD Pro diffractometer operated at 40 kV and 40 mA using Ni-filtered Cu K $\alpha$  irradiation (Wavelength 1.5406 Å). The chemical composition was obtained by X-ray photoelectron spectroscopy (Axis UltraDLD, Kratos) with monoaluminum K $\alpha$  radiation. The charge calibration was done by correcting C1s line of adventitious carbon setting to 284.8 eV to compensate the charge effect. UV-Vis diffuse reflection spectra (DRS) and monochromatic incident photon-to-electron conversion efficiency (IPCE) were detected to confirm the optical properties.

Photoelectrochemical measurements were carried out in a convenient three-electrode cell. Ti foils implanted by Ti ions onto a special designed electrode holder were used as the working electrodes. The surface areas exposed to electrolyte were fixed at  $0.785\,\mathrm{cm^2}$ . The PEC properties of all samples are characterized in  $0.5\,\mathrm{M}$  Na $_2\mathrm{SO_4}$  aqueous solution using three-electrode configuration with a Pt and an Ag/AgCl

electrode as a counter electrode and a reference electrode. An electrochemical workstation (CHI760D) and a 300 W Xe lamp ( $100 \, \text{mW/cm}^2$ ) as the solar irradiated simulator with light intensity set at  $100 \, \text{mW/cm}^2$  through an AM 1.5G filter were used for photocurrent-potential measurement. Incident photon-to-current conversion efficiency (IPCE) measurements were performed using a 300 W Xe lamp integrated with a computer-controlled monochromator, a photo chopper (PARC), and a lock-in amplifier used for photocurrent detection. IPCE measurements were performed in 0.5 M Na<sub>2</sub>SO<sub>4</sub> aqueous solution as electrolyte, and the applied potential was controlled at 0.8 V versus Ag/AgCl reference electrode.

#### 3. Results and Discussion

Figure 1 shows the SEM images of nanoparticles formed by Ti ions implanted into Ti foil and subsequent annealing at 400, 500, 600, and 700°C for 6 hours in oxygen atmosphere. In order to observe the nanoparticles clearly and present a three-dimensional image, we set the tilt angle to 52°. When

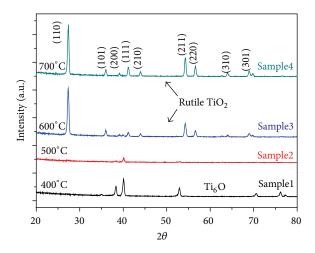


FIGURE 2: XRD patterns of  ${\rm TiO_2}$ , which are annealed at different temperatures.

the sample was annealed at 400°C, small titanium oxide nanoparticles were formed on the surface of the substrate, the average size of the nanoparticles is 25.8 nm, as shown in Figure 1(a). When the annealing temperature increases, the size of nanoparticles grows quickly. The average size of the nanoparticles grows to 93.3, 151.7, and 675 nm, for the annealing temperatures of 500, 600, and 700°C, respectively.

Figure 2 shows GIXRD patterns of the  ${\rm TiO_2}$  formed by Ti ions implantation into Ti foils and subsequent annealing at 400, 500, 600, and 700°C for 6 hours in oxygen atmosphere, respectively. It is found that the phase composition strongly depends on the annealing temperatures. With the increase of annealing temperature, the quality of titanium oxide changed from amorphous to rutile phase. The sample annealed at  $400^{\circ}{\rm C}$  is  ${\rm Ti_6O}$ . At the annealing temperatures of 600 and  $700^{\circ}{\rm C}$ , the titanium oxide transfers mostly to rutile phase.

In order to confirm the ingredients of nanoparticles on the surface of the substrate, we show the Raman spectra of the TiO<sub>2</sub> films formed by Ti implanted into Ti foil and subsequent annealing. As shown in Figure 3, when the annealing temperature is 400°C, the Raman peak located at 140 cm<sup>-1</sup> is the  $E_a$  mode of the  $TiO_2$  in anatase phase [8, 9]. No Raman mode of the Ti<sub>6</sub>O is found in the Raman spectra, which may be not in the range of the measurement. The anatase TiO<sub>2</sub> has not been detected in the XRD patterns due to its too fewer amount. With the increase of annealing temperature, two rutile Raman peaks appear at the  $442 \,\mathrm{cm}^{-1}$  (E<sub>a</sub>) and  $603 \,\mathrm{cm}^{-1}$  (A<sub>1,a</sub>) [10, 11], which were the first-order Raman spectra of rutile TiO<sub>2</sub>. When the annealing temperature is 700°C, the second-order Raman peak at 234 cm<sup>-1</sup> becomes stronger [12]. At the same time, the intensity of the  $\mathbf{E}_g$  and A<sub>1a</sub> modes became much stronger.

According to the SEM images, the GIXRD patterns, and the Raman spectra, we can conclude that the main content of nanoparticles on the surface of the samples annealed at  $400^{\circ}$ C is  $\text{Ti}_6\text{O}$  with small amount of  $\text{TiO}_2$  nanoparticles. When the annealing temperature is  $500^{\circ}$ C the content of rutile  $\text{TiO}_2$  is increased. As the annealing temperature increases to 600 and

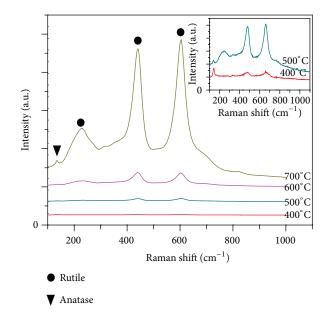


FIGURE 3: The Raman spectra of  ${\rm TiO_2}$  annealed at different temperatures.

700°C, the main particles are rutile  $\rm TiO_2$ . While annealing at low temperatures, implanted Ti atoms are slowly diffused out of the sample and are oxidized. Therefore, small  $\rm TiO_2$  particles are formed on the sample surface. Meanwhile, the near surface of substrate is also oxidized to  $\rm Ti_6O$ . Under higher annealing temperatures, the implanted Ti atoms are easier to diffuse to sample surface and the size of  $\rm TiO_2$  particles grows. In order to confirm the formation of the  $\rm TiO_2$  films, XPS analysis was performed for the  $\rm Ti$ -implanted sample to the fluence of  $\rm 3 \times 10^{17}$  ions/cm² and annealed at 400, 500, 600, and 700°C for 6 hours. As shown in Figure 4, the binding energy of  $\rm Ti2p_{1/2}$  locates at 458.4 eV belong to the binding energy of  $\rm Ti^{4+}$ .

In order to know the optical properties, we test the UV-Visible diffuse reflection spectra (DRS) and monochromatic incident photon-to-electron conversion efficiency (IPCE) of the  ${\rm TiO_2}$  films. As shown in Figure 5(a), with the increase of annealing temperature, the absorption of light from UV enlarged to visible light region. According to the GIXRD and Raman spectra, the main composition of the sample annealed at  $400^{\circ}{\rm C}$  is  ${\rm Ti_6O}$ . The broad absorption is possibly due to the absorption of  ${\rm Ti_6O}$ . The size of  ${\rm TiO_2}$  nanoparticles increased with the annealing temperatures, and the high temperature easily caused the phase change of  ${\rm TiO_2}$  nanoparticles to rutile phase, which are the reason that the absorption edges are red shifted.

The IPCE values are shown in Figure 5(b); for the sample annealed at  $400^{\circ}$ C, the maximum IPCE is 2.9% at the absorption wavelength of 320 nm and is 6.94% for the sample annealed at  $500^{\circ}$ C. The highest IPCE for the sample annealed at  $600^{\circ}$ C is 2.8% at the absorption wavelength of 370 nm. The absorption edges are red shifted with the increase of annealing temperatures, indicating the formation of TiO<sub>2</sub> nanoparticles. The further red shift of the absorption

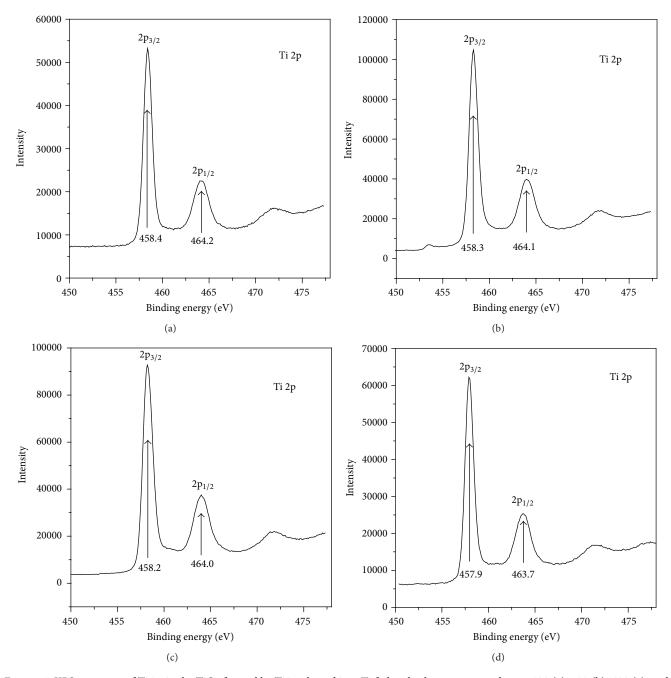


FIGURE 4: XPS spectrum of Ti 2p in the  $TiO_2$  formed by Ti implanted into Ti foil and subsequent annealing at 400 (a), 500 (b), 600 (c), and 700°C (d) for 6 hours in oxygen atmosphere, respectively.

wavelength after annealing at 700°C is due to the increase of the sizes of TiO<sub>2</sub> nanoparticles.

Figure 6(a) shows current-potential curves in  $0.5\,\mathrm{M}$  Na<sub>2</sub>SO<sub>4</sub> aqueous solution under UV-Visible light illumination for TiO<sub>2</sub> formed by Ti ion implantation and subsequent annealing at 400, 500, 600, and 700°C. The samples annealed at 400 and 500°C were n-type Ti<sub>6</sub>O. The photocurrent of the sample annealed at 500°C under UV-Visible light illumination with applied potential of  $0.8\,\mathrm{V}$  (versus SCE) is  $139.2\,\mu\mathrm{A/cm^2}$ . The samples annealed at 400 or 500°C show positive photocurrent; they present n-type property.

For the samples annealed at 600°C, which have the negative photocurrent at the negative bias and have a positive photocurrent when they are at the positive bias, show the property of both the p-type and n-type TiO<sub>2</sub>. Thus, the sample annealed at 600°C has weak p-type property. For the sample annealed at 700°C, the negative photocurrent is main, which mainly performs the p-type property of TiO<sub>2</sub>. Because the substrate is Ti foils, the Ti ions are abundant and the Ti atoms were diffused to the surface and oxidized during annealing under oxygen atmosphere, and the Ti<sub>6</sub>O was also gradually oxidized to TiO<sub>2</sub> with the increase of annealing temperature.

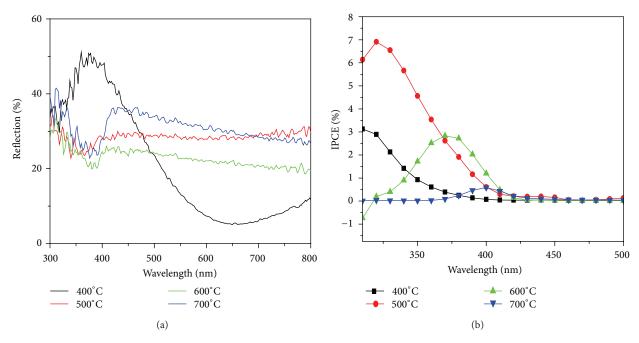


FIGURE 5: The UV-Vis diffuse reflection spectra (DRS) and monochromatic incident photon-to-electron conversion efficiency (IPCE) of the  $TiO_2$  films formed by Ti implanted into Ti foil and subsequent annealing.

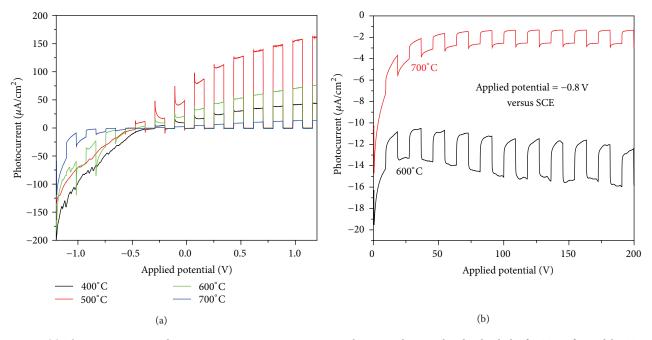


Figure 6: (a) The current-potential curves in  $0.5\,\mathrm{M}$  Na $_2\mathrm{SO}_4$  aqueous solution under simulated solar light for TiO $_2$  formed by Ti ion implantation and subsequent annealing at 400, 500, 600, and 700°C. (b) The i-t curve annealed at 600 and 700°C.

The samples annealed at 600 and 700°C are rutile TiO<sub>2</sub>; some of the Ti ions locate in the interstitial positions, which lead to the property of p-type TiO<sub>2</sub>. Figure 6(b) shows the photoelectrochemical measurement of the samples annealed at 600 and 700°C, which were operated by the amperometric i-t curve to evaluate their ability for photocatalytic water splitting under visible light illumination. The photocurrent

was  $3.52 \,\mu\text{A/cm}^2$  at  $600^{\circ}\text{C}$  and  $1.60 \,\mu\text{A/cm}^2$  at  $700^{\circ}\text{C}$ , with applied potential of  $-0.8 \,\text{V}$  (versus SCE).

Additional evidence of p-type  $TiO_2$  is from XPS spectra of the valence band region. As shown in Figure 7, the valence band of  $TiO_2$  that was annealed at  $400^{\circ}$ C (Figure 7(a)) starts from about 2.13 eV. In contrast, the valence band of  $TiO_2$  starts from about 1.66 eV (Figure 7(c)). It means that

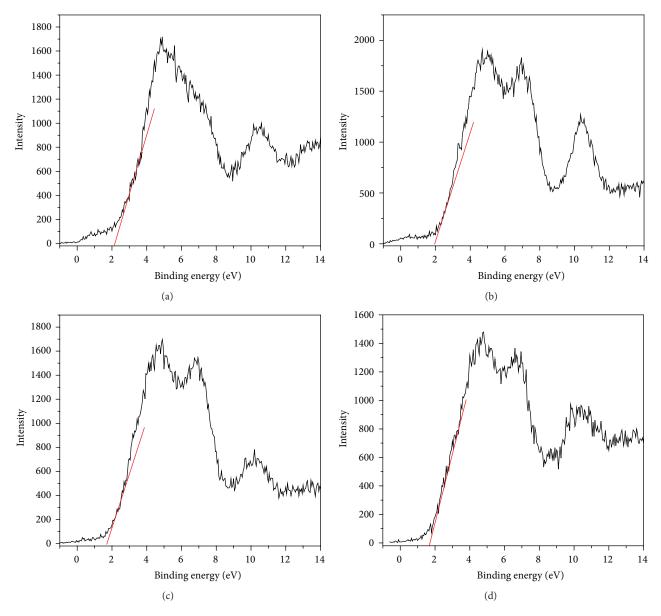


FIGURE 7: XPS spectra of the  $TiO_2$  thin film at the valence band region, annealing at 400 (a), 500 (b), 600 (c), and 700°C (d) for 6 hours in oxygen atmosphere, respectively.

the Fermi level of  ${\rm TiO_2}$  (0.0 eV) is closer to the valence band of the sample annealing at 600°C; it is reported that the preparation of p-type  ${\rm TiO_2}$  using the thermal approach needs high temperature [13]. The XPS spectra prove the formation of p-type  ${\rm TiO_2}$ .

#### 4. Conclusion

The  ${\rm TiO_2}$  nanofilms were formed on the substrate of Ti foils as the photoelectrochemical (PEC) electrode. The size of  ${\rm TiO_2}$  nanoparticles increased with the increasing of annealing temperatures, and the GIXRD patterns and Raman spectra

demonstrate that the phase of  ${\rm TiO_2}$  turns to rutile at high temperature. The IPCE values indicate that the absorption edges are red shifted with the increase of annealing temperatures. The n-type properties of the as-prepared  ${\rm TiO_2}$  photoelectrode are detected at low temperatures of 400 and 500°C. For the samples annealed at 600 and 700°C, the photocurrent shows the property of p-type  ${\rm TiO_2}$ .

### **Conflict of Interests**

The authors declare that there is no conflict of interests regarding the publication of this paper.

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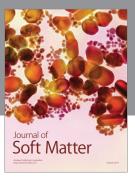
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